

# Hybrid Oxidation of Titanium Substrates for Biomedical Applications <sup>†</sup>

Jaroslav Jan Jasinski

Department of Innovation and Safety Systems, Czestochowa University of Technology (CUT),  
42-200 Czestochowa, Poland; jaroslav.jasinski@wz.pcz.pl; Tel.: +48-343250865

<sup>†</sup> Presented at the 2nd Coatings and Interfaces Web Conference, 15–31 May 2020; Available online:  
<https://ciwc2020.sciforum.net/>.

Published: 15 May 2020

**Abstract:** Titanium oxidation for biomedical applications is still a challenge in obtaining favorable mechanical and physicochemical properties of thin oxide layers, as well as the required high bioactivity. Interesting techniques for TiO<sub>2</sub> layer formation are electrochemical, plasma and diffusive methods. Each method aims to create a thin oxide layer characterized by thermal stability and re-passivation in the presence of a simulated body fluid SBF environment. However, an important aspect here is also the phase composition of oxide layers, essential for osseointegration. Accordingly, the research carried out aims to produce such a titanium substrate, where the surface zone is a Ti<sub>α</sub>(O) solid solution formed with fluidized bed (FB) diffusion process (640 °C, 8 h) and the top layer is TiO<sub>2</sub> produced by physical vapour deposition PVD—magnetron sputtering. The effects of such hybrid oxidation on titanium surface properties were investigated with scanning electron microscopy SEM/scanning transmission electron microscopy STEM/ Raman spectroscopy RS and nanoindentation tests. The results showed that hybrid oxidation made it possible to generate a favorable synergetic effect between FB and PVD oxide layers and to reduce the stresses at their interface. In turn, a variable share of TiO<sub>2</sub> phases (rutile + anatase mixture) obtained at the titanium surface allowed for the significant enhancement of hydroxyapatite compound growth, which was confirmed by a 14-day Kokubo test.

**Keywords:** titanium oxidation; PVD magnetron sputtering; thin TiO<sub>2</sub> layer; bioactivity

---

## 1. Introduction

Titanium oxide thin layers are still the subject of numerous studies due to their highly interesting properties for biomedicine and implantology, especially for third-generation biomaterials, which are produced to stimulate specific cell responses and tissue regeneration [1–3]. In fact, thin TiO<sub>2</sub> oxide layers are obtained with the use of several methods, such as: anodizing, laser treatment PLD, physical methods PVD and diffusion methods, which result in reduced thickness and poor adhesion, which depends on many factors, including surface preparation for the oxidation, the phase composition of surface oxides and the substrate's chemical and strength properties [4–7]. Nevertheless, each method aims to create a passive oxide coating, which is characterized by homogeneity, low thermal conductivity, chemical stability and the ability to re-passivate after being defected in the presence of a corrosive environment. The mechanism of titanium oxidation differs from the oxidation of other metals. This is due to stability of the material: Ti<sub>α</sub>—stable up to 882 °C and Ti<sub>β</sub>—stable over 882 °C. At room temperature, on the Ti surface, a thin (5–15 nm) passive nanolayer is formed, where, in turn, the oxidation of titanium at a high temperature (>400 °C) leads to the formation of the crystalline layer with a TiO/inter-Ti<sub>2</sub>O<sub>3</sub> layer and a TiO<sub>2</sub> layer (rutile or anatase) zone structure. The oxide coating formed at room temperature is stable and adheres well to the substrate, but is too thin. In turn, at

high temperatures, titanium oxidizes rapidly and forms thick oxide layer which is often porous and poorly bonded (anchored) to the substrate, and thus delaminates and cracks [8–11]. An important role of the oxide layer on the titanium surface, in addition to the aforementioned properties, is to provide the required osseointegration process kinetics by forcing the biochemical activity of the layers leading to accelerated interaction with the body's tissues [12–16]. Initially, it was thought that the titanium substrates are inert to the body. However, when in direct contact with the tissues of organisms, titanium can release Ti ions into the body's environment, which causes the occurrence of edema and inflammation, generates health problems for patients and ultimately the rejection of the implant. The biocompatibility and bioactivity of titanium are directly related to the physicochemical properties of the substrate surface. To improve the bioactivity of titanium substrates, the best-known solutions are single-stage surface treatment and the production of multilayers [17–20]. However, surface methods, due to the conditions of rapid chemical interaction between the atmosphere and substrate, have very limited influence on oxygen diffusion processes towards the substrate surface layer and the formation of a  $Ti_{\alpha}(O)$  diffusion layer with good strength properties. Thus, it is difficult to obtain substrates with the following arrangement: I.  $Ti_{\alpha}$  substrate/II.  $Ti_{\alpha}(O)$  solid solution/III. thin  $TiO_2$  oxide layer, with both stable oxide phases at the surface, a low hardness gradient between the matrix and layer and a reduced state of stress (compressive stresses required) at the interface [21–24]. Accordingly, highly bioactive titanium materials (i.e., third-generation metallic biomaterials) might be produced by the adequate functionalization of the thin oxide layers (tailored phase composition morphology and adhesion to the substrate) together with the control of the substrate surface stress state and structure. Therefore, the research carried out by the author aims to develop such a titanium substrate, where on the diffusion oxide layer ( $Ti_{\alpha}(O)$  solid solution), a homogenous, tight and smooth thin  $TiO_2$  layer is formed by surface treatment, i.e., PVD magnetron sputtering. Such a hybrid method uses the advantages of continuous substrate activation and defect by the influence of a fluidized bed aeromechanical factor and non-equilibrium PVD surface oxidation. There is the expectation that a combination of  $TiO_2$  layers will ensure a synergistic effect in the improvement of the titanium substrate's biofunctional properties.

## 2. Materials and Methods

The substrates used for hybrid oxidation were made of  $Ti_{\alpha}$  single-phase commercially pure titanium manufactured by Kobe Steel LTD in accordance with ASTM 8348, with the chemical composition presented in Table 1.

**Table 1.** The chemical composition of commercially pure titanium used for hybrid oxidation (in accordance with ASTM 8348) (mass %).

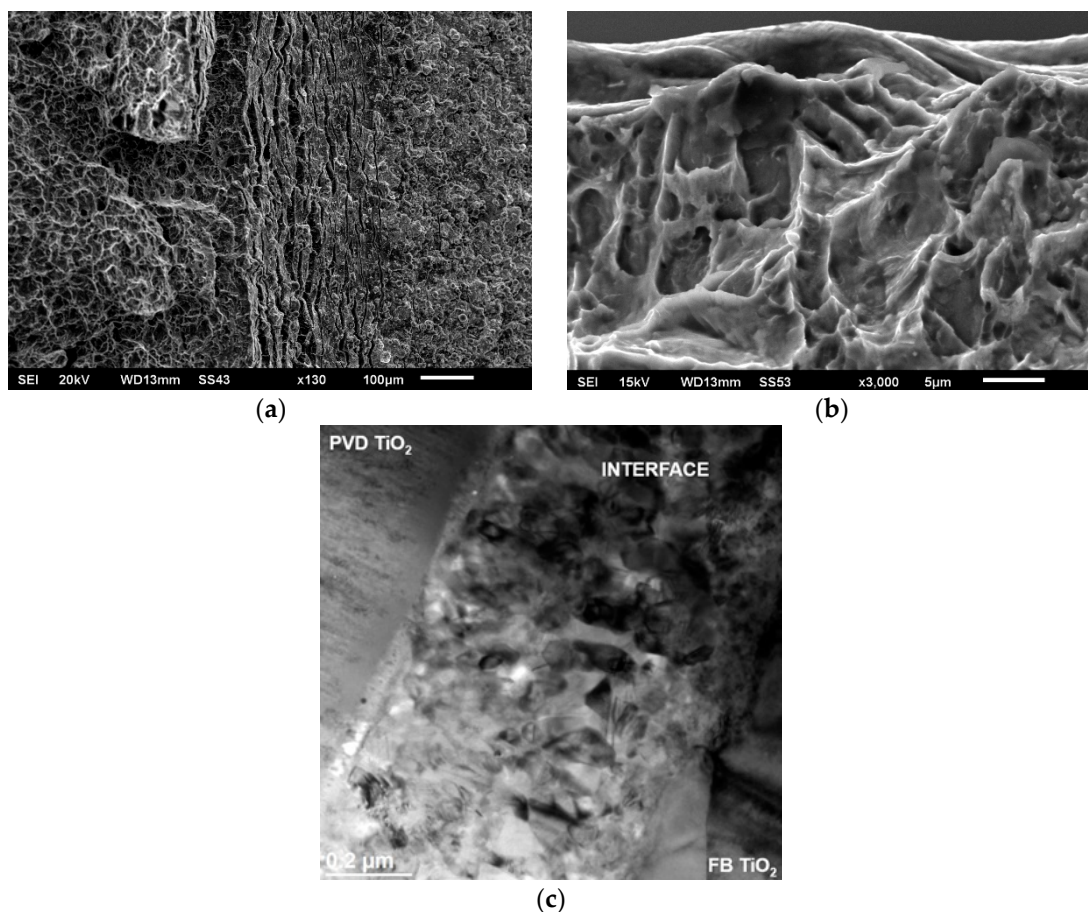
Material	Chemical Composition					
KOBE Steel LTD Titanium Grade 2 (ASTM 8348)	O	N	C	H	Fe	Ti
	0.20	0.03	0.10	0.015	0.30	rest

Before hybrid oxidation, the substrates were mechanically activated by blasting with a mixture of  $Al_2O_3 + ZrO_2 + Ti$ . Diffusive oxidation was carried out in a fluidized bed (FB) reactor with  $Al_2O_3$  grain material at 640 °C for 8 h in an air atmosphere. After the FB treatment, substrates were cooled down in air. A further oxidation process was conducted with PVD magnetron sputtering using a  $TiO_2$  target, with a pressure of  $3 \times 10^{-2}$  mbar, an Ar (99.95%) atmosphere, constant power mode  $P = 350$  W, target–substrate distance 60 mm and deposition time 20 min. The thin  $TiO_2$  oxide layer structure and interface were analyzed by a scanning electron microscopy (SEM) (FEI E-SEM XL30 microscope) and scanning transmission electron microscopy (STEM) (FEI S/TEM TITAN 80-300) method. The surface morphology of the substrates was evaluated by confocal laser scanning microscopy (CLSM) (LEXT OLS4000 microscope OLYMPUS, Tokyo, Japan). Phase analysis of the  $TiO_2$  layers was conducted by Raman spectroscopy (RS) (LabRAM HR micro-Raman spectrometer equipped with a CCD detector HORIBA Scientific France SAS), under an excitation wavelength of 532 nm and an intensity of ca. 10 mW. The acquisition time was set at 30 s. The precise determination of the oxide layers' hardness,

Young's modulus and elastic and plastic energy was realized with nanoindentation mechanical tests (NanoTest Vantage Micro Materials Quantum Design Europe, United Kingdom). The bioactivity response of the titanium substrates was evaluated by a 14-day Kokubo test using *c-SBF2* solution. It was found that the hybrid oxidation method (FB+PVD) led to the formation of tight, homogeneous thin  $\text{TiO}_2$  layers, which highly improves the bioactivity of the titanium surface in the aspect of biomedical applications.

### 3. Results and Discussion

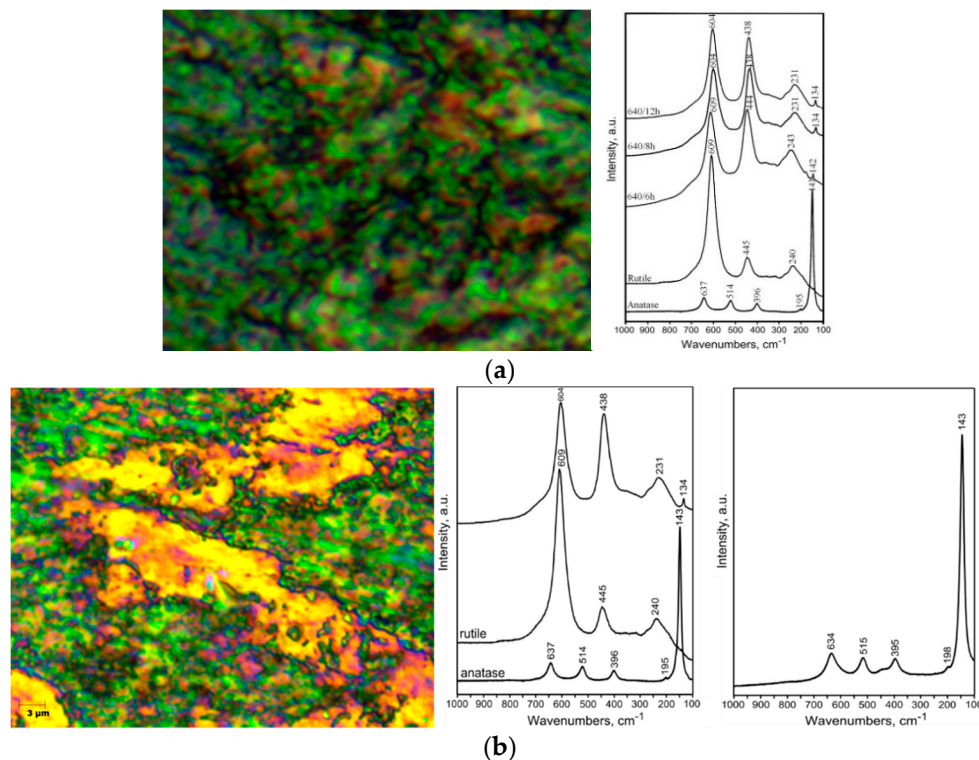
Titanium oxidation, realized by a two-stage hybrid process (FB + PVD), allowed for the production of strong substrates with rutile and anatase  $\text{TiO}_2$  thin layers at the top surface. The first stage of the oxidation process was conducted in a fluidized bed (FB), which allowed for the production of a  $\text{Ti}_6\text{O}$  diffusion layer with a thickness of 11  $\mu\text{m}$  and a ca. 2  $\mu\text{m}$  nano-porous  $\text{TiO}_2$  oxide layer. The saturation of titanium with oxygen atoms leads to the strengthening of the substrate matrix and improves its hardness. Furthermore, a fine-grained diffusion zone under a nano-porous oxide layer aimed to reduce the stress gradient between the matrix and the  $\text{TiO}_2$  layer. The second stage of the oxidation process was PVD magnetron sputtering, which resulted in the deposition of a thin  $\text{TiO}_2$  oxide layer with a thickness of ca. 0.8–1  $\mu\text{m}$ . The plasma interaction with the FB substrate involved the continuous bombardment of nano-porous  $\text{TiO}_2$  and enhanced local heat transfer to control chemical reactions (physisorption) when forming thin  $\text{TiO}_2$  PVD layers. Hybrid oxidation also produced a stable and fine FB  $\text{TiO}_2$ /PVD  $\text{TiO}_2$  interface with a thickness of ca. 600–620 nm (Figure 1).



**Figure 1.** SEM/STEM images of the titanium substrate microstructure and interface after hybrid oxidation (a) fluidized bed FB 640 °C/8 h, (b) fluidized bed FB + PVD magnetron sputtering and (c) FB  $\text{TiO}_2$ /PVD  $\text{TiO}_2$  interface.

At the interface zone, there were visible areas of nano-pores, which were free gaps for further anchoring the  $\text{TiO}_2$  layers deposited by the PVD method. The next step of the research was substrate

surface morphology and phase analysis conducted by confocal laser scanning microscopy and Raman spectroscopy (Figure 2).



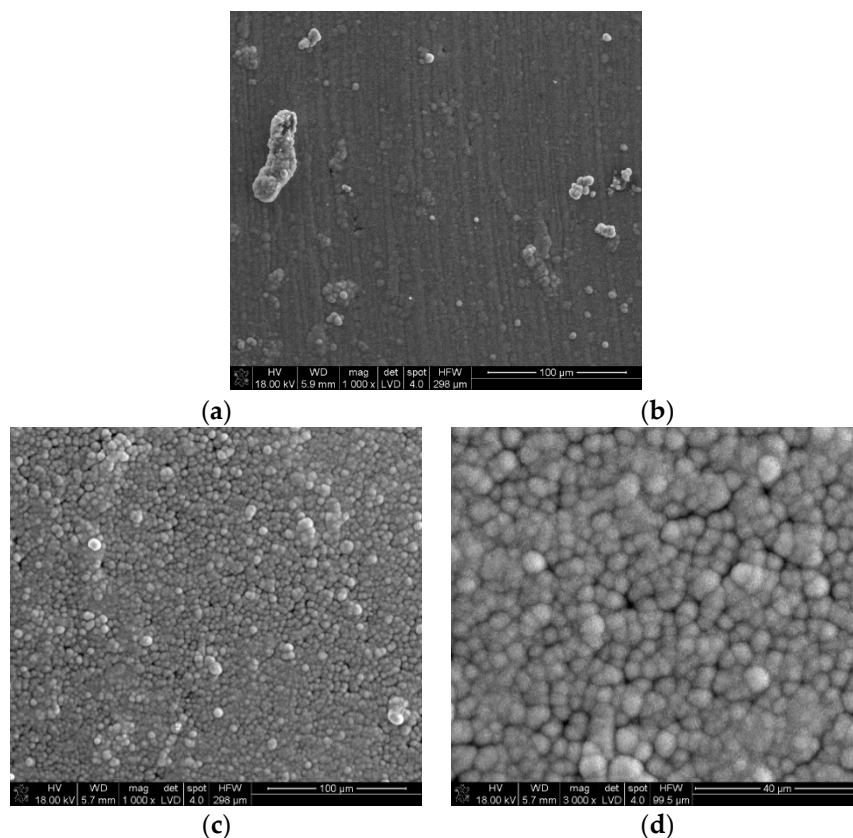
**Figure 2.** Confocal laser scanning microscopy image and Raman spectra of the titanium surface after hybrid oxidation, (a) FB 640 °C/8 h and (b) FB + PVD magnetron sputtering.

The Raman spectra obtained for titanium after FB oxidation showed the presence of the strongest peaks coming from the rutile phase. Hardly noticeable bands located at wave numbers of 143–146 cm<sup>-1</sup> were obtained for the anatase phase. However, hybrid oxidation FB + PVD showed the presence of visible TiO<sub>2</sub> rutile (wave numbers: 604 cm<sup>-1</sup>, 438 cm<sup>-1</sup> and 231 cm<sup>-1</sup>) and anatase bands in the wave numbers of 143 cm<sup>-1</sup>, 395 and 515 cm<sup>-1</sup>. In addition, it was also observed that the bands shifted towards lower wave numbers, suggesting the occurrence of compressive stresses in the TiO<sub>2</sub> thin layer. The rutile phase of TiO<sub>2</sub> layers plays an important role in inducing the apatite deposition as a result of the crystal lattice matching between rutile and apatite. In fact, there are some literature data which show biomedical properties of anatase, including the author's previous works [25,26]. The author tried to define and precisely indicate the favorable phase share of the rutile and anatase titanium oxide mixture at the surface, which promises to have a great influence on the bioactive behavior of the substrates. Such a phase gradient (between rutile and anatase) has a great influence on the osteogenesis and bioactivity of titanium substrates. The next step of the research was the nanomechanical investigation of the PVD thin TiO<sub>2</sub> oxide layer. The results showed the favorable strength properties of the layers. A series of indentations (at nano- and micro-scales) was performed on pure titanium (raw substrates) and the specimens after hybrid oxidation. The results of the nanoindentation tests are shown in Table 2.

**Table 2.** Nanoindentation test results of titanium substrates before and after hybrid oxidation (FB + PVD).

Substrate Type	Hardness, H (GPa)		Reduced Young's Modulus, $E_R$ (GPa)		Calculated Young's Modulus, $E$ (GPa)		Maximum Depth (nm)		Plastic Depth (nm)	
	Value	SD	Value	SD	Value	SD	Value	SD	Value	SD
Titanium Grade 2 (ASTM 8348)	9.33	4.14	160.00	60.30	148.34	55.91	204.08	57.41	167.17	54.36
Titanium after hybrid oxidation FB + PVD	15.21	6.04	281.83	87.79	261.28	81.39	144.90	28.87	119.20	27.37

Results allowed for finding a correlation between the mechanical parameters measured at the nano- and micro-scales for the substrates. Special attention was devoted to the mechanical properties of the FB + PVD interface, which plays a crucial role in the integrity of the whole hybrid system. The nanoindentation hardness and Young's modulus measured for the FB + PVD  $\text{TiO}_2$  were  $H = 15.21$  GPa and  $E = 261$  GPa, which were slightly higher values than the results of sputtered  $\text{TiO}_2$  layers reported in the literature [27]. The nanoindentation results confirmed that hybrid oxidation affects the improvement of titanium surface hardness and strength. From the point of view of the application of the obtained substrates as biomaterials, it is necessary to determine their bioactivity. Such important results were obtained after the 14-day Kokubo test in simulated body fluid SBF [28,29] (Figure 3).



**Figure 3.** SEM image of the effect of hydroxyapatite on the growth of titanium substrates after hybrid oxidation—14-day simulated body fluid SBF Kokubo test, (a) FB 640 °C/8 h and (b) FB + PVD rutile + anatase phase.

The intensive growth of globular hydroxyapatite compounds was visible at the surface of FB + PVD  $\text{TiO}_2$  thin rutile/anatase layers. Such an improvement in biochemical activity was reached both through stabilization and the reduction of stresses at the FB  $\text{TiO}_2$ /PVD  $\text{TiO}_2$  interface, and the tailoring

of the TiO<sub>2</sub> phase composition at the surface. The Kokubo test results confirmed that the hybrid oxidation significantly enhances the bioactivity and allows for the biofunctional modification of titanium substrates.

#### 4. Conclusions

- A. Diffusion oxidation in a fluidized bed (FB) leads to the formation of a highly defected Ti<sub>α</sub>(O) diffusion zone with good strength properties and nano-porous TiO<sub>2</sub>. Such a system plays a role as a foundation for the subsequent deposition of thin TiO<sub>2</sub> layers by PVD magnetron sputtering.
- B. The hybrid oxidation treatment applies two types of surface activation, I: mechanical, as an impact of an aeromechanical factor in FB; II: sputtering, with simultaneous oxidation by PVD. Activation increases the number of active centers, and enhances oxygen mass transport to finally form homogenous thin TiO<sub>2</sub> layers. The layers are characterized by a high level of homogeneity and resistance to cracking and delayering.
- C. In hybrid oxidation, the interface between nano-porous FB TiO<sub>2</sub> and PVD TiO<sub>2</sub> has a favorable state of stress and further influences the formation of a bioactive rutile and anatase mixture, which improves the rate of osseointegration.
- D. The presented hybrid oxidation is a promising surface treatment for biomedical applications, indicating the directions of forming bioactive layers on titanium substrates. The solution corresponds with the new trends in biomaterials and surface engineering to combine different processing techniques in order to improve implants and medical devices.

#### 5. Patents

Patent no PL 221053: Method for modifying the surface layer of titanium alloy implants. P. Podsiad, J.J. Jasinski, J. Jasinski, R. Czyz.

**Funding:** This research was funded by the National Science Centre, Poland.

**Conflicts of Interest:** The author declares no conflict of interest.

#### References

1. Rack, H.J.; Qazi, J.I. Titanium alloys for biomedical applications. *Mater. Sci. Eng. C* **2006**, *26*, 1269–1277.
2. Kokubo, T.; Kim, H.M.; Kawashita, M.; Nakamura, T. Bioactive metals: preparation and properties. *J. Mater. Sci. Mater. Med.* **2004**, *15*, 99–107.
3. Rahimi, N.; Pax, R.A.; Mac, A.; Gray, E. Review of functional titanium oxides. I: TiO<sub>2</sub> and its modifications. *Prog. Solid State Chem.* **2016**, *44*, 86–105.
4. Zhou, B.; Jiang, X.; Shen, R.; Rogachev, A.V. Preparation and characterization of TiO<sub>2</sub> thin film by thermal oxidation of sputtered Ti film. *Mater. Sci. Semicond. Process.* **2013**, *16*, 513–519.
5. Radmanesh, M.; Kiani, A. Bioactivity enhancement of titanium induced by Nd:Yag laser pulses. *J. Appl. Biomater. Funct. Mater.* **2016**, *14*, 70–77.
6. Wu, B.; Yu, Y.; Wu, J.; Shchelkanov, I.; Ruzic, D.N.; Huang, N.; Len, Y.X. Tailoring of titanium thin film properties in high power pulsed magnetron sputtering. *Vacuum* **2018**, *150*, 144–154.
7. Heinrichs, J.; Jarmar, T.; Wiklund, U.; Engqvist, H. Physical Vapour Deposition and Bioactivity of Crystalline Titanium Dioxide Thin Films. *Trends Biomater. Artif. Organs* **2008**, *22*, 104–110.
8. Shannon, R.D.; Pask, J.A. Kinetics of the anatase-rutile transformation. *J. Am. Ceram. Soc.* **1965**, *48*, 391–398.
9. Aniolek, K. The influence of thermal oxidation parameters on the growth of oxide layers on titanium. *Vacuum* **2017**, *144*, 94–100.
10. Satoh, N.; Nakashima, T.; Yamamoto, K. Metastability of anatase: size dependent and irreversible anatase-rutile phase transition in atomic-level precise titania. *Sci. Rep.* **2013**, *3*, 1959.
11. Pradhan, S.S.; Sahoo, S.; Pradhan, S.K. Influence of annealing temperature on the structural, mechanical and wetting property of TiO<sub>2</sub> films deposited by RF magnetron sputtering. *Thin Solid Films* **2010**, *518*, 6904–6908.



12. Ochsenbein, A.; Chai, F.; Winter, S.; Traisnel, M.; Breme, J.; Hildebrand, H.F. Osteoblast responses to different oxide coatings produced by the sol-gel process on titanium substrates. *Acta Biomater.* **2008**, *4*, 1506–1517.
13. Barfeie, A.; Wilson, J.; Rees, J. Implant surface characteristics and their effect on osseointegration. *Br. Dent. J.* **2015**, *218*, E9.
14. Niinomi, M.; Nakai, M.; Hieda, J. Development of new metallic alloys for biomedical applications. *Acta Biomater.* **2012**, *8*, 38883–38903.
15. Forsgren, J.; Svahn, F.; Jarmar, T.; Engqvist, H. Formation and adhesion of biomimetic hydroxyapatite deposited on titanium substrates. *Acta Biomater.* **2007**, *3*, 980–984.
16. Rosales-Leal, J.I.; Rodríguez-Valverde, M.A.; Mazzaglia, G.; Ramon-Torregrosa, P.J.; Diaz-Rodriguez, L.; Garcia-Martinez, O.; Vallecillo-Capilla, M.; Ruiz, C.; Cabrerizo-Vilchez, M.A. Effect of roughness, wettability and morphology of engineered titanium surfaces on osteoblast-like cell adhesion. *Colloids Surf. A Physicochem. Eng. Asp.* **2010**, *365*, 222–229.
17. Yamaguchi, S.; Nath, S.; Sugawara, Y.; Divakarla, K.; Das, T.; Manos, J.; Chrzanowski, W.; Matsushita, T.; Kokubo, T. Two-in-one biointerfaces—Antimicrobial and bioactive nanoporous gallium titanate layers for titanium implants. *Nanomaterials* **2017**, *7*, 229.
18. Ding, Z.; Hu, X.; Yue, P.L.; Lu, G.Q.; Greenfield, P.F. Synthesis of anatase TiO<sub>2</sub> supported on porous solids by chemical vapor deposition. *Catal. Today* **2001**, *68*, 173–182.
19. Sabetrasekh, R.; Tiainen, H.; Lyngstadaas, S.P.; Reseland, J.; Haugen, H. A novel ultra-porous titanium dioxide ceramic with excellent biocompatibility. *J. Biomater. Appl.* **2011**, *25*, 559–580.
20. Sengottuvelan, A.; Balasubramanian, P.; Will, J.; Boccaccini, A.R. Bioactivation of titanium dioxide scaffolds by ALP-functionalization. *Bioact. Mater.* **2017**, *2*, 1081–1015.
21. Li, D.; Ferguson, S.J.; Beutler, T.; Cochran, D.L.; Siting, C.; Hirt, H.P.; Buser, D.J. Biomechanical comparison of the sandblasted and acid-etched and the machined and acid-etched titanium surface for dental implants. *J. Biomed. Mater. Res.* **2002**, *60*, 325–332.
22. Lubas, M.; Sitarz, M.; Jasinski, J.J.; Jelen, P.; Klita, L.; Podsiad, P.; Jasinski, J. Fabrication and characterization of oxygen-Diffused titanium using spectroscopy method. *Spectrochim. Acta A* **2014**, *133*, 883–886.
23. Sarvadai, S.Y.; Gatin, A.K.; Kharitonov, V.A.; Dokhlikova, N.V.; Ozerin, S.A.; Grishin, M.V.; Shub, B.R. Oxidation of Thin Titanium Films: Determination of the Chemical Composition of the Oxide and the Oxygen Diffusion Factor. *Crystals* **2020**, *10*, 117.
24. Toptan, F.; Alves, A.C.; Pinto, A.M.P.; Ponthiaux, P. Tribocorrosion behavior of bio-functionalized highly porous titanium. *J. Mech. Behav. Biomed.* **2017**, *69*, 144–152.
25. He, J.; Zhou, W.; Zhou, X.; Zhong, X.; Zhang, X.; Wan, P.; Zhu, B.; Chen, W. The anatase phase of nanotopography titania plays an important role on osteoblast cell morphology and proliferation. *J. Mater. Sci. Mater. Med.* **2008**, *19*, 3465–3472.
26. Jasinski, J.J.; Lubas, M.; Kurpaska, L.; Napadlek, W.; Sitarz, M. Functionalization of Ti99.2 substrates surface by hybrid treatment investigated with spectroscopic methods. *J. Mol. Struct.* **2018**, *1164*, 4124–4119.
27. Pang, M.; Bahr, D. Thin-film fracture during nanoindentation of a titanium oxide film–titanium system. *J. Mater. Res.* **2001**, *16*, 2634–2643.
28. Kokubo, T.; Takadama, H. How useful is SBF in predicting in vivo bone bioactivity. *Biomaterials* **2006**, *27*, 2907–2915.
29. Oyane, A.; Onuma, K.; Ito, A.; Kim, H.M.; Kokubo, T.; Nakamura, T. Formation and growth of clusters in conventional and new kinds of simulated body fluids. *J. Biomed. Mater. Res.* **2003**, *64*, 3393–3348.

