



Article In Situ Dilatometry Measurements of Deformation of Microporous Carbon Induced by Temperature and Carbon Dioxide Adsorption under High Pressures

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Abstract: Adsorption-based carbon dioxide capture, utilization, and storage technologies aim to mitigate the accumulation of anthropogenic greenhouse gases that cause climate change. It is assumed that porous carbons as adsorbents are able to demonstrate the effectiveness of these technologies over a wide range of temperatures and pressures. The present study aimed to investigate the temperatureinduced changes in the dimensions of the microporous carbon adsorbent Sorbonorit 4, as well as the carbon dioxide adsorption, by using in situ dilatometry. The nonmonotonic changes in the dimensions of Sorbonorit 4 under vacuum were found with increasing temperature from 213 to 573 K. At T > 300 K, the thermal linear expansion coefficient of Sorbonorit 4 exceeded that of a graphite crystal, reaching 5×10^{-5} K at 573 K. The CO₂ adsorption onto Sorbonorit 4 gave rise to its contraction at low temperatures and pressures or to its expansion at high temperatures over the entire pressure range. An inversion of the temperature dependence of the adsorption-induced deformation (AID) of Sorbonorit-4 was observed. The AID of Sorbonorit-4 and differential isosteric heat of CO₂ adsorption plotted as a function of carbon dioxide uptake varied within the same intervals of adsorption values, reflecting the changes in the state of adsorbed molecules caused by contributions from adsorbate-adsorbent and adsorbate-adsorbate interactions. A simple model of nanoporous carbon adsorbents as randomly oriented nanocrystallites interconnected by a disordered carbon phase is proposed to represent the adsorption- and temperature-induced deformation of nanocrystallites with the macroscopic deformation of the adsorbent granules.

Keywords: adsorption; adsorption-induced deformation; microporous carbon; adsorbent; carbon dioxide; high pressure; thermal expansion; thermodynamics of adsorption

1. Introduction

Carbon dioxide originated as a result of nature and anthropogenic activity. It is believed that increased carbon dioxide emissions contribute to climate change [1]. Three strategies to mitigate carbon dioxide emissions are proposed: (1) improvement of fossil fuel utilization and power efficiency of engines; (2) carbon dioxide sequestration; (3) carbon dioxide utilization in valuable branches of human activities [2]. Indeed, carbon dioxide has many applications. For example, it is used in medicine as a vasodilating agent [3]; intraoperative carbon dioxide insufflation was found to help to prevent surgical complications, including surgical site infection [4]. Carbon dioxide is utilized as renewable feedstock to produce chemicals and plastics [5,6] and unconventional biomass-derived fuels and biomaterials, following the concept of sustainable bioenergy [7–9]. Injecting carbon dioxide into depleted shale reservoirs or unminable coal seams is an important method to enhance oil and natural gas recovery [10–13]. This method is based on the dynamic displacement processes governed by the pressure gradient and competitive adsorption: gas shale and coal bed have a higher adsorption affinity for carbon dioxide compared to that of methane [11,12]. For example, the so-called "huff-n-puff" technology involves the



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Copyright: © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). underground injection of CO_2 followed by displacing previously adsorbed methane and forming a CO_2 – CH_4 mixture that causes a subsequent increase in the concentration of free methane within the rock [11,13,14]. It should be noted that the injection of CO_2 in shale is one of the methods used to store CO_2 in porous media [15,16].

All of the aforementioned uses of CO_2 include adsorption processes. Most of these processes take place at high pressures. Therefore, in making efforts to develop new high-pressure adsorption-based technologies and to improve the existing ones, researchers need to consider the adsorption-induced deformation (AID) of adsorbents. This phenomena should be taken into account both when predicting the performance of the adsorbent, including its integrity [17,18], and evaluating the thermodynamic state functions of adsorption processes [19,20].

From almost a century's worth of research, which began with the pioneering works of Mechan [21] and Bangham [22–24], it can be stated with confidence that there is a wealth of experimental evidence to suggest that an adsorbent, being a source of adsorption forces, experiences changes during adsorption. According to numerous macro- and microscopic experimental studies and computer simulations (see the reviews [25-27]), the adsorption of various gases and vapors can result in both expansion (as observed for the first time by Mechan and Bangham for charcoals [21–24]) and contraction (as reported by Lakhanpal and Flood for activated carbons [25]). The AID sign depends on the ratio of the sizes of pores and adsorbate molecules, as well as on the thermodynamic conditions of the adsorption process [25]. Therefore, a precise prediction of the AID sign and magnitude requires knowledge about the mechanism of this effect based on experimental data for various adsorbents. The deformation effects induced by CO_2 adsorption have been explored for activated carbons [21,22,28–31], zeolites [32,33], zeolite membranes [34], and flexible metal-organic frameworks [35,36]. In order to establish the general pattern of AID phenomena, one needs to employ microscopic (X-ray diffraction [37], small angle X-ray scattering [38], small-angle neutron scattering [31]) and macroscopic (dilatometry [39], ellipsometric porosimetry [40]) experimental techniques. These methods offer the possibility of registering changes in the adsorbent state at different scales. At the same time, it is not possible to directly compare the results of microscopic and macroscopic measurements because these two methods exhibit different sensitivities to anisotropic AID effects [27,41] and the pore size distribution [42].

It would make sense to thoroughly investigate the AID effect in an adsorbent with a certain type of pores over a wide range of pressures and temperatures. For this purpose, we employed the macroscopic dilatometry method to examine the AID effect in a carbon adsorbent, which contains predominantly slit-like micropores, from the early stage of adsorption to the high-pressure region. We selected carbon dioxide and a carbon adsorbent in the context of the relevance of these effects for the above-mentioned technologies for enhancing oil and natural gas recovery and CO_2 geological storage.

We started with a thorough characterization of the adsorbent using scanning electron microscopy (SEM) and low-temperature nitrogen adsorption. We then carried out the carbon dioxide measurements within a wide temperature range, including sub- and supercritical states of carbon dioxide up to a pressure of 0.1 Mpa. We used experimental data and an approach based on Dubinin's theory of the volume filling of micropores [43,44] to calculate the carbon dioxide adsorption up to high pressures. We employed a dilatometer to assess the changes in the sizes of the adsorbent induced by temperature in order to take into account the thermal expansion when measuring the AID of the carbon adsorbent at different temperatures up to 6 Mpa.

Finally, we calculated the thermodynamic state functions of the carbon dioxide adsorption on the microporous carbon adsorbent using the well-known formula by Bakaev, which includes the effects of the non-ideality of a gaseous phase and the non-inertness of an adsorbent [45]. Here, the non-inertness of the adsorbent characterizes its inability to maintain its dimensions during adsorption or with a change in temperature, i.e., adsorptionor temperature-induced deformation.

2. Materials and Methods

2.1. Adsorbent

We studied a commercial carbon adsorbent, Sorbonorit 4, produced by Cabot Norit Nederland B.V. (Amersfoort, The Netherlands) from a peat-based carbon-rich raw material via thermochemical activation [46]. Sorbonorit 4 cylindrical pellets, which were 6.8 in length and 3.7 mm in diameter, were used in the adsorption, temperature- and adsorption-induced deformation measurements.

2.2. Adsorptive

In the experiments, we used high purity carbon dioxide (99.995%), purchased from Linde Gas Rus (Balashikha, Moscow Region, Russia), as an adsorptive. The basic physicochemical properties of carbon dioxide are summarized in Table 1.

Table 1. Physicochemical properties of carbon dioxide [47–49].

M	T _b , K	T _{cr} , K	$ ho_{ m cr}$, kg/m ³	P _{cr} , MPa	T_{λ} , K	d/l _m , Å/Å	d _{kin} , Å
44,011	194.65	304.19	468	7.382	216.55	3.1/4.1	3.3
		λ <i>(</i> · 1 1	6 1 1 1		·1· · · · · · [1/] ·	1 1.	

M is a molecular mass of carbon dioxide; T_b [K] is the boiling point; T_{cr} [K] is the critical temperature; P_{cr} [MPa] is the critical pressure; ρ_{cr} [kg/m³] is the critical density; T_λ [K] is the ternary point; *d* [Å] is the critical diameter along the smallest axis of a molecule (it is commonly used to assess a possibility of an adsorptive molecule to penetrate into the pores of an adsorbent and its value is measured experimentally by the ability to be adsorbed in window-cage type zeolites [48]) and l_m [Å] is the length of a molecule; d_{kin} [Å] is the kinetic diameter of a molecule.

2.3. Methods

2.3.1. Characterization of Adsorbent

The porous structure characteristics of Sorbonorit 4 were acquired from the data on nitrogen vapor adsorption at 77 K, obtained with the use of a Quantachrome IQ commercial porous structure analyzer. Dubinin's theory of volume filling of micropores (TVFM) [43,44] was used to calculate the following microporosity parameters of the adsorbent: the micropore volume W_0 [cm³/g], the standard characteristic energy of adsorption E_0 [kJ/mol], and the micropore width X_0 [nm].

The microporosity parameters were evaluated from the experimental data on low-temperature nitrogen vapor adsorption using the general TVFM equation, namely the Dubinin–Astakhov equation [43,44]:

$$a = a_0 exp\left[-\left(A/E\right)^n\right],\tag{1}$$

where $A = RT \ln (P_S/P)$ [kJ/mol] is the differential molar work of adsorption; a_0 [mmol/g] is the limiting value of adsorption at the temperature T [K], which depends on the specific volume of micropores W_0 ; E [kJ/mol] is the characteristic energy of gas adsorption. It was found that, for activated carbons, the index of power (n), which depends on surface chemistry and the extent of uniformity of the pore structure, is equal to 2. Therefore, we used Equation (1) with n = 2, i.e., the Dubinin–Radushkevich (D-R) equation.

According to TVFM applied to a model of slit-like pores, the characteristic energy of adsorption of the test adsorptive is related to the effective half-width of micropores by the following formula:

$$E(kJ/mol) = \frac{12\beta}{X_0},$$
(2)

where β is the affinity coefficient for the test adsorptive (in this case, nitrogen vapors) with respect to the reference benzene vapors:

$$\beta = \frac{E}{E_0} = \frac{\Pi}{\Pi_0}.$$
(3)

Note that E_0 is the standard characteristic energy of adsorption for benzene vapors; Π and $\Pi_0 [(J^{1/4} \times cm^{5/2})/mol]$ are the parachors of the test adsorptive (in this case, nitrogen vapors) and standard benzene vapors, respectively, which can be calculated as follows [50]:

$$\Pi = \frac{M \cdot \sigma^{\frac{1}{4}}}{\rho_{\rm L} - \rho_{\rm vapor}},\tag{4}$$

Here, σ [J/m²] is the surface tension; ρ_L and ρ_{vapor} are the densities of liquid and vapor at the boiling temperature.

The limiting value of adsorption on the line of saturation vapor pressure at the boiling point temperature T_b and the pressure $P_s(T_b)$ is determined under the condition that, on the line of saturation and at $T = T_b$, the density of the adsorbate ρ_{ad} is close to the density of the liquid ρ_L :

$$a_0(T) = W_0 \rho_{\rm ad} \approx W_0 \rho_{\rm L}.$$
(5)

The differential work of adsorption is calculated as follows:

$$A = RTln\left(\frac{f_{\rm S}}{f}\right),\tag{6}$$

Here, *R* is the universal gas constant; f_S is the fugacity of the saturated adsorbed gas (vapor); *f* is the fugacity of the equilibrium phase.

The Brunauer–Emmett–Teller (BET) method [51] combined with the Rouquerol criterion [52] was employed to calculate the specific surface area S_{BET} [m²/g]. The pore size distribution (PSD) function was derived using the quenched solid density functional theory (QSDFT) [53] for micropores. The mesopore structure of the adsorbent was characterized using a Kelvin equation-based approach [54]; the PSD function for mesopores was estimated by applying the Barret–Joyner–Halenda (BJH) method [55].

The surface morphology of Sorbonorit 4 and its elemental composition were studied via scanning electron microscopy (SEM) using a Quanta 650 FEG microscope (FEI Company, US) equipped with an Oxford Energy Dispersive X-ray (EDX) detector. The images were obtained at an accelerating voltage of 30 kV. The elemental composition was estimated as an average of three measurements.

The isotherms of carbon dioxide adsorption onto Sorbonorit 4 were measured gravimetrically using a custom-designed setup [56,57]. The measurements were carried out over the range of sub- and supercritical states of the adsorptive within a temperature range from 216.4 K (slightly below the ternary point of carbon dioxide) to 393 K up to a pressure of 0.1 MPa. The accuracy of the experimental measurements was calculated following the State Standards [58], and its value did not exceed 1.5 % with a confidence level of 0.95.

2.3.2. Measurements of the Sorbonorit 4 Deformation Induced by the Carbon Dioxide Deformation and Temperature

The isotherms of the deformation of the adsorbent granules induced by adsorption and temperature were measured using a custom-made dilatometer bench [39]. The scheme of the bench and the principle of its operation have been described in detail in previous works [39,59]. Here, we only provide a brief description. The bench consists of an inductiontype dilatometer connected to a vacuum post and a gas supply system. The induction-type dilatometer is capable of performing measurements over a wide range of pressures (from 0.1 Pa to 10 MPa). The main element of the dilatometer is an induction-type displacement sensor, which consists of a reciprocating rod, a magnetic core, and an inductive coil arrangement outside the dilatometer housing. The magnetic core is connected to the rod mounted in the guidelines. The rod rests on a support plate contiguous to the adsorbent granules, which are arranged vertically in a pressure-tight measuring cell. Before a set of measurements, the Sorbonorit 4 granules were subjected to an adsorption stabilization procedure consisting of 25 adsorption/desorption cycles over a pressure range of 0.1 to 5 MPa at room temperature. By stabilizing the porous structure, we ensured the reproducibility of the experiments. Before each measurement of the set, the adsorbent granules were regenerated at elevated temperatures (up to 573 K) under vacuum for at least 6 h. Then, the cell with the adsorbent was filled with carbon dioxide under pressure. After reaching thermodynamic equilibrium in the measuring cell, the adsorption-induced deformation of Sorbonorit 4 caused the rod to move, thereby actuating the magnetic core. This displacement caused changes in electrical inductance within the inductive coil arrangement. The changes in the signal were used to determine the changes in the linear size of the Sorbonorit 4 granules at the specified temperature and pressure. It should be noted that pressure and temperature values were recorded. Thus, by progressively increasing the portions of injected carbon dioxide, we evaluated the changes in the size of the adsorbent granules as a function of the pressure at a given temperature.

The CO₂ adsorption-induced deformation of Sorbonorit 4 was measured within a temperature range from 216.4 to 393 K and two pressure ranges corresponding to the subcritical and supercritical states: from 0.1 Pa up to the saturated vapor pressure at the corresponding temperature and from 0.1 Pa to 6 MPa, respectively. The deformation of the adsorbent was assumed to be isotropic, and this statement is in agreement with previously published estimates [60] and confirmed by experimental data [41]. The confidence level for estimating the magnitude of linear deformation of the adsorbent was $\pm 5 \cdot 10^{-7}$ m, which is in agreement with the State Standards [61]. On average, the measurement uncertainty of the relative linear deformation of the adsorbent was $u_c(\eta) = 2.0\%$, and the expanded uncertainty was $U(\eta) = 6.0\%$ at a confidence level of 0.95 [58]. The deformation of the Sorbonorit 4 granules induced by temperature was investigated under vacuum conditions at a residual pressure less than 1 Pa within a temperature range from 213 K to 573 K. Prior to the measurements, the sample was also regenerated under the same conditions—573 K and <0.1 Pa.

3. Results and Discussion

3.1. Characterization of the Adsorbent

Figure 1 shows the isotherms of standard nitrogen vapor adsorption/desorption at 77 K, which were used to calculate the structure and energy characteristics of the adsorbent.



Figure 1. Standard isotherms of adsorption (light symbols)/desorption (dark symbols) of nitrogen vapors onto Sorbonorit 4 at 77 K represented in linear coordinates and semi-log coordinates (in the inset).

The Γ-shaped (or type I) isotherm [62] of adsorption of nitrogen vapors at 77 K represented in linear coordinates within the initial values of relative pressure P/P_S up to 0.3 indicates the presence of a developed micropore volume in Sorbonorit 4. A weakly

defined H4 loop of the capillary condensation hysteresis [62] was observed in the pressure range close to the saturated vapor pressure, which was attributed to mesopores.

Table 1 summarizes the parameters of porous structure of Sorbonorit 4, which were evaluated from the low-temperature nitrogen adsorption isotherm.

Figure 2 shows the pore size distributions in Sorbonorit 4 evaluated for micropores (a) and mesopores (b) using the QSDFT and the BJH methods, respectively.



Figure 2. The PSD function calculated by the QSDFT method for micropores (**a**) and the BJH method for mesopores (**b**) in Sorbonorit 4.

As follows from Figure 2, the pore structure of Sorbonorit 4 is characterized by the PSD function, with one well-defined peak corresponding to the micropore size d_{MAX}^{NLDFT} of 1.22 nm and a relatively wide peak corresponding to the mesopore size d_{MAX}^{BJH} of 3.9 nm. The average pore width calculated by applying the TVFM equations amounted to 1.1 nm (see Table 2), which is close to the value of d_{MAX}^{NLDFT} . As follows from Figure 2b and Table 2, the mesopore volume is less than 5% of the total pore volume. Dubinin showed that if the adsorbent pore structure involves mesopores with a surface area less than 25 m²/g, the adsorption in mesopores is insignificant compared to that in micropores [48], meaning that it can be neglected in the calculation of adsorption processes. Therefore, Sorbonorit 4 is primarily microporous, with a few mesopores acting as transport pores.

Table 2. The parameters of porous structure of Sorbonorit 4 evaluated from the standard isotherm of nitrogen vapors at 77 K.

Adsorbent	S _{BET} , m ² /g	W ₀ , cm ³ /g	<i>X</i> ₀ , nm	E ₀ , kJ/mol	W _S , cm ³ /g	W _{ME} , cm ³ /g	S _{ME} , m²/g	$d_{\mathrm{MAX}}^{\mathrm{NLDFT}}$, nm	d ^{BJH} MAX' nm
Sorbonorit 4	1200	0.49	0.55	21.8	0.51	0.02	10	1.22	3.9

Here, S_{BET} is the specific BET surface [51,52]; W_0 is the micropore volume; X_0 is the micropore half-width; E_0 is the characteristic energy of adsorption [43,44]; W_S is the total pore volume; W_{ME} is the mesopore volume; S_{ME} is the specific mesopore surface [54]; $d_{\text{MAX}}^{\text{NLDFT}}$ is the micropore size, which corresponds to the maximum of the PSD function evaluated by the QSDFT method elaborated for micro-mesoporous adsorbents [53]; $d_{\text{MAX}}^{\text{BJH}}$ is the mesopore size, which corresponds to the maximum of the PSD function provided by the use of the BJH method [55].

The SEM investigations of Sorbonorit 4 (Figure 3a,b) revealed that its surface is composed of various particle aggregates ranging in size from 10 to 30 μ m. At higher magnification (Figure 3b), the stacked grains with a size of 1–5 μ m are viewable. These irregular grains have a surface that is uneven and covered with cracks and cavities.



Figure 3. SEM images of the surface of Sorbonorit 4 at a scale bar of 90 µm (a) and 9 µm (b).

Table 3 presents the results of the EDX study on the elemental composition of Sorbonorit 4.

Table 3. The elemental con	position of Sorbonorit	4 according to the EDX data
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Element	Amount, at.%		
С	89.35		
0	9.11		
Κ	0.23		
Other impurities	1.21		

As follows from Table 3, Sorbonorit 4 is mostly carbon. The presence of other impurities is caused by the diverse composition of peat as a raw material and the activation method. There are many experimental studies evidencing that the presence of oxygen in carbon adsorbents has a direct effect on the adsorption capacity, including carbon dioxide [63,64].

3.2. Carbon Dioxide Adsorption onto Sorbonorit 4

The experimental data of carbon dioxide adsorption onto Sorbonorit 4 measured within a temperature range from 216.4 to 393 K were used to plot the isosteres. As follows from Figure 4, the isostere plots $\ln P = f(1/T)|_{a=\text{const}}$ for carbon dioxide adsorption are linear throughout a wide range of temperatures and pressures, including the ranges of temperature: $T_{\lambda} < T < T_{cr}$, and $T_{cr} < T$. It should be noted that the linearity of the isosteres of adsorption, which is preserved upon transition to the supercritical ($P > P_{cr}$, $T > T_{cr}$) state of the adsorbing substance, is an intrinsic feature of adsorption in microporous adsorbents [65–67]. The linearity of the isosteres of adsorption makes it possible to calculate the adsorption equilibrium from the minimal set of experimental data.

To calculate the carbon dioxide adsorption equilibrium under sub- and supercritical conditions up to 20 MPa, we used the linearity of the isosteres and the TVFM equations.

We applied Equations(1)–(6) to the structural and energy characteristics of Sorbonorit 4 deduced from the data on low-temperature nitrogen vapor adsorption (see Table 2) and the physicochemical characteristics of carbon dioxide (Table 1) to calculate the carbon dioxide adsorption equilibrium over the sub- and supercritical conditions.



Figure 4. Isosteres of carbon dioxide adsorption onto Sorbonorit 4 at the values of adsorption *a*, mmol·g⁻¹: 0.05 (1), 0.1 (2), 0.15 (3), 0.2 (4), 0.4 (5), 0.6 (6), 0.8 (7), 1.0 (8), 1.4 (9), 1.8 (10), 2.2 (11), 2.6 (12), 3.0 (13), 3.4 (14), 3.8 (15). Dashed red lines separate the regions of sub- and supercritical state of adsorbed molecules: $T_{\lambda} < T < T_{cr}$ and $T > T_{cr}$, respectively.

As indicated in Section 2.3.1, at boiling point, the limiting adsorption can be calculated using Equation (5). However, within a temperature range from the boiling point T_b to the critical value T_{cr} , the limiting adsorption was evaluated using the Dubinin–Nikolaev formula [68]:

$$a_0(T) = a_0(T_{\rm b})exp(-\alpha(T - T_{\rm b})),$$
(7)

Here, α [1/K] is the thermal coefficient of limiting adsorption: $\alpha > 1$, i.e., the adsorption values decrease as the temperature increases. Within a temperature range $T_b \leq T \leq T_{cr}$, the value of α is assumed to be constant [68]:

(

$$\alpha = -\frac{d(\ln a_0)}{dT} = \text{const.}$$
(8)

The isotherms of carbon dioxide adsorption at temperatures $T > T_{cr}$ (supercritical state) were also calculated using Equation (1), where the saturated pressure P_s was determined by the linear extrapolation of a function:

$$\ln P_{\rm s}(T) = M - \frac{N}{T}.\tag{9}$$

Here, *M* and *N* are the constants, which may be evaluated for carbon dioxide from two characteristic points: (T_b , P_s) and (T_{cr} , P_{cr}). The limiting adsorption a_0 at the saturation pressure was found by linear extrapolation of the Dubinin–Nikolaev Equation (7).

When calculating the adsorption equilibrium in the region of supercritical temperatures using Equation (1), we also considered the temperature dependence of both the characteristic energy E and the exponent n [69].

Figure 5 shows the experimental data and the results of calculations of carbon dioxide adsorption over temperatures ranging from 216.4 to 393 K and at pressures up to 20 MPa. As can be seen, the results of calculations using the TVFM equations (solid lines) coincide with both the experimental data on carbon dioxide adsorption (dark symbols) and the results obtained from the extrapolation of linear isosteres (light symbols).



Figure 5. The carbon dioxide adsorption onto Sorbonorit 4 versus pressure represented in semilog coordinates at temperatures *T*, K: 216.4 (1), 243 (2), 273 (3), 293 (4), 333 (5), and 393 (6). The dark symbols show the experimental data; the light symbols are the values of adsorption evaluated based on the linearity of the isosteres of adsorption; the solid lines are the results of calculation using Equations (1)–(9); the dashed lines separate the data obtained by these three methods.

It should be noted that the isotherms of carbon dioxide adsorption onto Sorbonorit 4 represented as $a = f (\ln p)$ have a characteristic S-shape. With increasing temperature, the curves move almost in parallel towards high pressures.

3.3. Temperature-Induced Deformation of Sorbonorit 4

Figure 6 depicts the results of the temperature-induced strain of Sorbonorit 4 measured under vacuum; the constant error determined from the measurements with a quartz mockup was taken into consideration according to a route described in recent work [20]. One can see that, within the entire temperature range from 213 to 573 K, an increase in temperature causes a non-monotonic change in its value (up to 267–280 K), while the relative deformation magnitude decreases to 0.08–0.10%. However, the subsequent rise in temperature causes the expansion amplitude to increase by up to 1.5 %—observed at 593 K.



Figure 6. The temperature-induced strain of Sorbonorit 4 measured under vacuum. The bars show the expanded uncertainty of measurements.

The thermal linear expansion coefficient (TLEC) of Sorbonorit 4, α_T [1/K], was calculated using the experimental plot $\Delta l/l_0 = f(T)$:

$$\alpha_T = \frac{1}{l_0} \frac{dl}{dT} = \frac{1}{\Delta T} \left(\frac{\Delta l}{l_0} \right)$$
(10)

Figure 7 shows the results of these calculations compared with those for a silicon carbide-derived activated carbon (denoted as ASC) [20] and the data for a graphite crystal in a plane parallel and perpendicular to a main crystallographic axis [70,71]: α_{II} , and α_{\perp} , respectively.



Figure 7. Temperature dependence of TLEC of Sorbonorit 4 under vacuum conditions (violet line) compared with the data for a graphite crystal in a plane parallel and perpendicular to the main crystallographic axis: α_{II} and α_{\perp} (respective black lines) [70,71]; microporous carbon adsorbent ASC (dashed black line) [20].

As follows from Figure 7, when T > 300 K, $\alpha_{\perp} < \alpha_{ASC} < \alpha_{II} < \alpha_{Tsorbonorit}$. It is well known that exceptionally low thermal expansion, wide variation of magnitude, and high structure sensitivity are intrinsic features of the carbon structures [72]. Therefore, the discrepancy between the TLEC values of Sorbonorit 4 and graphite can be attributed to the textural properties of the carbon adsorbent: (1) polycrystalline structure; (2) presence of the amorphous phase; (3) developed porosity, which includes intracrystalline micropores (crystal defects) and intercrystallite meso- and macropores. Indeed, according to estimations [41], the granules of activated carbons (including Sorbonorit 4) contain 10¹⁶ graphite nanocrystallites with a size of 0.6–3 nm and 10²⁰ randomly oriented slit-like pores per gram. The orientation of the nanocrystallites in the granules is equiprobable, and the nanocrystallites are surrounded by an amorphous carbon phase [73,74].

The discrepancy between the TLEC values of Sorbonorit 4 and ASC can be attributed to the differences in their structure, chemical composition, and density inherited from the raw materials and activation methods. The pyrolysis of amorphous peat with a loose structure at 973–1173 K results in the formation of the intermediary product composed of graphite-like nanocrystallites and residual amorphous phase. The subsequent thermochemical activation of the char is accompanied by the formation of intracrystalline slit-like micropores and intercrystallite mesopores [75,76]. The ASC carbon adsorbent prepared by the thermochemical leaching of Si atoms from the wurtzite-type silicon carbide crystals [77] is characterized by a higher density, a more homogeneous chemical and phase composition, and a relatively narrow pore size distribution [78]. The high rigidity of the ASC adsorbent is inherited from its crystalline precursor. Indeed, the activation process leads to a contraction of the sample by no more than 2% compared to the precursor [77]. Therefore, unlike Sorbonorit 4, the rigid carbon structure of the ASC adsorbent is less sensitive to variations in temperature over the entire range of studied temperatures.

Nevertheless, the TLEC values for Sorbonorit 4 do not exceed 5×10^{-6} 1/K, so we ignored this effect when studying the adsorption-induced deformation of Sorbonorit 4 at different temperatures.

3.4. Adsorption-Induced Deformation of Sorbonorit 4

Figure 8a,b show the relative changes in the size of the Sorbonorit 4 granules upon the adsorption of carbon dioxide as a function of pressure within the entire range of pressures from 1 Pa to 6 MPa (a) and at the initial stage of adsorption over a pressure range of from 0.1 Pa to 1 MPa (b). The AID isotherms were measured at temperatures ranging from 213 to 393 K.



Figure 8. The adsorption-induced strain of the Sorbonorit 4 adsorbent versus CO₂ pressure (**a**) from 0.1 Pa to 6 MPa and (**b**) from 0.1 Pa to 1 MPa, measured at a temperature T, K: 216.4 (1), 243 (2), 273 (3), 293 (4), 333 (5), and 393 (6). Symbols are the experimental data on deformation obtained during adsorption (light symbols) and desorption (dark symbols). Solid lines are the spline-approximation. The dashed lines (**a**) mark the pressure of saturated vapor at the corresponding temperature. The bars on curve 5 (**b**) show the expanded measurement uncertainty.

As can be seen from Figure 8, at low temperatures (curves 1–4), the AID magnitude is negative at low pressures (for example, p < 0.15 MPa—curve 1), indicating the contraction of Sorbonorit 4 at the initial stage of adsorption. However, with increasing pressure, the contraction magnitude reached a maximum value (a minimum on the AID curves) and then became smaller. Finally, the contraction of the adsorbent changed, resulting in its expansion. The initial contraction magnitude and the magnitude of the expansion at the final adsorption stage are comparable (Figure 8a). At the beginning of the adsorption process (Figure 8b), the contraction magnitudes fixed at the same pressure decreased with increasing temperature up to 0 (curves 1–4), whereas the expansion became more pronounced afterward (curves 5 and 6). At high pressures (Figure 8a), an increase in temperature led to a decrease in the expansion magnitude. As follows from Figure 8a,b, this inversion of the AID curves occurs within the pressure range of 0.4 to 1 MPa. The inversion of the AID curves was also disclosed for other carbon adsorbents, but the region of pressure was different. For example, the inversion of the AID curves for the ASC carbon adsorbent during the adsorption of CO_2 and CH_4 was observed within the range of 0.2 to 0.5 MPa [41] and 0.1 to 0.3 MPa [78], respectively. It should be noted that the two carbon adsorbents have almost the same micropore volume (0.49 and 0.51 cm³/g [41,78]), but these adsorbents differ in terms of pore width ($X_0 = 0.55$ and 0.41 nm), characteristic energies of adsorption (21.8 and 29.0 kJ/mol), PSDs, and densities. Therefore, the energy parameters, which dictate the adsorption process at the initial stage and, most likely, the mechanical properties of the adsorbents (e.g., compressibility) determine the extension of the inversion region.

The transition from the initial contraction of Sorbonorit 4 observed at temperatures below 293 K to its expansion with increasing pressure can be considered a result of a change in the balance of the intermolecular interactions in the adsorption system during the process. Indeed, the contraction of the adsorbent can be considered a manifestation of the attraction between the CO₂ molecules and the carbon atoms that form the walls of a micropore. It should be noted that the pore width in Sorbonorit 4 ($2X_0$ or d_{MAX}^{NLDFT}) and the dimensions of a CO₂ molecule (d, d_{kin} , and l_m) are comparable (see Tables 1 and 2). An increase in pressure, and thus the number of adsorbed molecules, leads to a decrease in the average distance between the adsorbed molecules and an enhancement of the repulsion. As a result, the contraction effect mitigates and transforms into the expansion of the adsorbent.

As the temperature rises, the region of initial contraction becomes narrower so that, at $T \sim 300$ K, the adsorbent expands within the entire range of pressures. Moreover, at $T \sim 333$ and 393 K, Sorbonorit 4 first abruptly expands by 0.04 and 0.08%, respectively, (see Figure 8b, curves 5 and 6). This effect can be perceived as a consequence of the influence of increasing molecular mobility on the balance of the interaction forces in the adsorption system. A further increase in pressure further expanded the adsorbent, the magnitude of which is determined by both the internal pressure of the adsorbate and the temperature-dependent compressibility of the adsorbent.

It should be noted that similar temperature-dependent contraction–expansion effects were found by Folman and Yates for porous glasses during the adsorption of polar molecules (CH₃Cl, SO₂, CCl₂F₂, and NH₃) [32] and by Pulin et al. for zeolite during CO₂ adsorption [33,79].

The contraction–expansion of the adsorbent induced by the CO₂ adsorption can be analyzed as a function of the amounts of adsorbed molecules. For this purpose, we plotted the AID magnitude $\Delta l/l_0$ versus the value of CO₂ adsorption a = f(P). The results are shown in Figure 9.



Figure 9. The AID value of Sorbonorit-4 (symbols) versus the carbon dioxide uptake at temperatures T, K: 216.4 (1), 243 (2), 273 (3), 293 (4), 333 (5), 393 (6). The solid lines are the spline approximation.

As follows from Figure 9, the adsorption of the first CO_2 molecules (i.e., when a = 0.1-0.3 mmol/g) onto Sorbonorit 4 resulted in its jump-like contraction or expansion depending on temperature. The jump-like deformation was not observed only at a temperature of about 300 K. We interpret this effect as a manifestation of the energy transition from a one-component system containing the only adsorbent to a two-component system containing the adsorbent and the adsorbate. It was found that, over the range of the investigated temperatures, the magnitude of the initial jump-like deformation at a constant number of adsorbed CO_2 molecules (at a = 0.3 mmol/g) is a linear function of the reciprocal temperature (see Figure 10).



Figure 10. The initial jump-like AID of Sorbonorit-4 (symbols) at the CO₂ adsorption a = 0.3 mmol/g versus reciprocal temperature. The solid line is the linear approximation. The dashed line shows the zero deformation.

The macroscopic manifestation of the AID effects can be interpreted by using a model of carbon adsorbent [44,73] represented schematically in Figure 11. As mentioned above, carbon adsorbent can be conceived of as a mixture of randomly oriented graphite-like nanocrystallites connected by an amorphous phase consisting of complex aromatic-aliphatic carbon radicals. The proportions of ordered and amorphous phases are determined by the nature of the raw materials and the activation method. The nanocrystallites include from 2 to 4–5 graphite layers with micropores; hence, the size of nanocrystallites ranges from 0.7 to 3 nm. Aromatic-aliphatic radicals form the less dense domains, and their size is commensurate with the size of the molecules between the crystallites. These domains serve as an interlinkage between the nanocrystallites. It should be noted that such a random arrangement of nanocrystallites linked via disordered matter is associated with the presence of voids (meso- or macropores).



Figure 11. Schematic representation of macroscopic deformation of a model carbon microporous adsorbent: adsorbent in a pre-stressed state under vacuum after regeneration at a certain temperature, white arrows indicate the forces inside and outside the nanocrystallite (**a**); the initial contraction of nanocrystallites (at low temperatures) (**b**); the expansion of nanocrystallites (at elevated temperatures and/or high pressures) (**c**). Red arrows indicate the direction of nanocrystallites (**b**,**c**).

After regeneration, the microporous carbon adsorbent under vacuum is in a so-called pre-stressed state, which is determined by its textural properties, including the porosity and the temperature [80]. Figure 11 illustrates a simplified model of a microporous carbon adsorbent as a set of nanocrystallites (depicted as grey-colored square regions) with aligned micropores. The nanocrystallites are inclined to each other at a certain angle ψ . The junctions of the nanocrystallites are in a disordered phase; shown as corner-sharing black circles. The mutual arrangement of the nanocrystallites or, more precisely, the angle ψ between them depends on the temperature-dependent volume compression modulus [41] (Figure 11a). The voids (meso- and macropores) between the nanocrystallites act as transport pores for adsorbate molecules, allowing them enter micropores in the nanocrystallites. At low temperatures, the first adsorbed molecules exert the van der Waals attractive forces upon the carbon atoms of the opposite micropore walls. The contraction of micropores results in the shrinkage of the nanocrystallites accompanied by their rotation relative to each other and a decrease in the angle ψ due to the bonding of the nanocrystallites and the low thermal mobility (Figure 11b). As a result, one can observe the macroscopic contraction of the adsorbent sample. At low temperatures, the initial jump-like deformation and the magnitude of the maximum contraction are both larger than those at elevated temperatures. Indeed, with rising temperature (Figure 11c), the intensity of thermal motions both of adsorbed molecules and carbon atoms of the pore walls increase, and the interactions between the carbon layers in the nanocrystallites become weaker. Therefore, the entry of the first adsorbed molecules into the pore leads to the expansion of the pore and the entire nanocrystallite. The expansion of the nanocrystallites is accompanied by their rearrangement with an increase in the angle ψ and, as a result, the macroscopic expansion of the granule (see Figure 11c). It should be noted that this simple model is also suitable for interpreting the temperature-induced deformation of carbon adsorbents.

In addition to this mechanism, various defects of the pore walls, such as intercalated heteroatoms, unburned layers of carbon, and functional groups located at the edges of graphite-like crystallite layers, contribute to the initial adsorption-induced deformation. Firstly, in the early stages of adsorption, the presence of impurities and defects in graphite crystallites in a carbon adsorbent, which serve as the adsorption centers for carbon dioxide [81], increases the amount of adsorbed carbon dioxide molecules. As a result, the impact of the adsorbate on the adsorbent, which is referred to as the adsorption stress [82], is also augmented. Secondly, the presence and number of intercalated heteroatoms and unburned carbon layers can affect the stiffness of the carbon adsorbent and, as a consequence, its response to the adsorption stress. Future comparative investigations of different carbon adsorbents using structural methods may support this general assertion with accurate quantitative data.

It is worth noting that despite the smallness of the deformation induced by adsorption in micropores, this effect may be essential for the above-mentioned CO₂-CH₂ gas exchange process applied for enhancing coal-bed methane extraction [83]. The swelling of the coal induced by the adsorption in micropores can close the transport meso- and macropores, thereby reducing the carbon dioxide injectivity of the coal bed.

The adsorption-induced deformation data of nanoporous materials are often used for evaluating their elastic properties, which are essential for practical applications. In the previous study [41], the triaxial compression modulus of Sorbonorit 4 was evaluated. Its value was found to be temperature-dependent, as evidenced by the fact that it decreased from 42 to 10 GPa as the temperature increased from 216.6 to 293 K. The obtained values of the compression modulus of Sorbonorit 4 are close to those reported by Neimark for microporous activated carbon prepared from silicon carbide [84].

3.5. Thermodynamic Behaviors of the CO₂–Sorbonorit 4 Adsorption System

As shown above, the effects of the initial jump-like deformation of the Sorbonorit 4 carbon adsorbent induced by the carbon dioxide adsorption are determined by many factors, including the intermolecular interactions between the carbon dioxide molecules and

the pore walls. Information on the specific features of the adsorbent–adsorbate interactions can be obtained from the analysis of the differential isosteric molar heat of adsorption as a function of the adsorption value. Indeed, isosteric differential heat of adsorption is often interpreted as the energy required for the adsorbate molecules to move from the equilibrium bulk phase to the adsorbed state under the following specified thermodynamic conditions: constant temperature, pressure, and amount of adsorbed substance or value of adsorption. The fundamental problem of describing the properties of the adsorbed state is solved by dividing the adsorption system into the bulk and adsorption phases, since the latter is characterized by the corresponding state parameters in relation to those of the bulk phase. It is assumed that molecules in the bulk phase are outside the adsorption field created by the adsorbent [85,86]. According to the definition by Hill, the differential molar isosteric heat of adsorption, $q_{\rm st}$ [kJ·mol⁻¹], is the difference between the molar enthalpy of the equilibrium gas phase $h_{\rm g}$ and the differential molar enthalpy of the adsorption phase $\left(\frac{\partial H_1}{\partial a}\right)_{\tau}$ [85,86]:

$$q_{\rm st} = h_{\rm g} - \left(\frac{\partial H_1}{\partial a}\right)_T = h_{\rm g} - h_1,\tag{11}$$

As noted in the Introduction, the most complete formula for calculating the differential molar isosteric heat of adsorption over a wide range of temperatures and pressures was proposed by Bakaev [45]:

$$q_{\rm st} = -R \cdot Z \cdot \left[\frac{\partial (\ln P)}{\partial (1/T)}\right]_a \cdot \left[1 - \left(\frac{\partial V_a}{\partial a}\right)_T / \nu_{\rm g}\right] - \left(\frac{\partial P}{\partial a}\right)_T \cdot \left[V_a - T \cdot \left(\frac{\partial V_a}{\partial T}\right)_a\right]$$
(12)

Here, R [J·(mol·K)⁻¹] is the universal gas constant; $Z = P \cdot v_g / (RT)$ is the compressibility of the equilibrium gas phase under specified P,T-conditions; v_g [m³·kg⁻¹] is the specific gas phase volume; $V_a = V_0(P,T)/m_a$ [cm³·g⁻¹] is the reduced volume of the adsorbent– adsorbate system; V_0 is the volume of the regenerated adsorbent with pores; and m_a is the mass of the regenerated adsorbent. Data on the compressibility Z(P,T) and the specific volume of the equilibrium CO₂ gas phase v_g are available in the NIST Chemistry WebBook [87].

Obviously, Bakaev's formula considers all of the factors that affect the value of differential molar isosteric heat of adsorption: isothermal adsorption-stimulated deformation $(\partial V_a/\partial a)_T$, temperature isosteric deformation $(\partial V_a/\partial T)_a$, the slopes of the isotherm of adsorption $(\partial P/\partial a)_T$ and isostere $[\partial \ln P/\partial (1/T)]_a$, and the non-ideality of a gas phase Z(P,T).

If the impacts of the non-ideality of the gas phase, thermal, and adsorption non-inertness (deformation) of the adsorption system are negligible, Equation (12) is reduced as follows:

$$q_{\rm st} = -R \cdot Z \cdot \left[\frac{\partial (\ln P)}{\partial (1/T)}\right]_a - \left(\frac{\partial P}{\partial a}\right)_T \cdot V_a,\tag{13}$$

As follows from Figure 12, the largest discrepancy between the curves 1 and 0 is observed at high pressures. Let us compare the contributions of the non-ideality of the gaseous phase and the non-inertness of the adsorbent to the isosteric heat of adsorption at various temperatures using the data summarized in Table 4.

As follows from Table 4, the compressibility of the gaseous phase produces the greatest effect on the calculated isosteric heat of adsorption, up to ~85% at 333 K. The effects of temperature- and adsorption-induced deformations on the values of q_{st} are almost commensurable. Their cumulative contribution does not exceed 4% and can be neglected.

It is known that changes in the heat of adsorption during the adsorption process reflect variations in the balance of the adsorbate–adsorbent and adsorbate–adsorbate interactions. At the beginning of adsorption, the value of q_{st} drops sharply by ~1.5 kJ/mol upon the pore loading with CO₂ to a value of 0.3 mmol/g. This effect is associated with the occupation of high-energy adsorption sites by adsorbing molecules, resulting in the jump-like deformation of Sorbonorit 4 (see Figure 9). The synchronicity of these processes points to the fact



Figure 12. The differential molar isosteric heat of carbon dioxide adsorption onto Sorbonorit 4 versus the adsorption value at temperatures, K: 216.4 (1), 243 (2), 273 (3), 293 (4), 333.0 (5), 393 (6), calculated by Equation (12). The curve 0 shows the results of the calculation of q_{st} by Equation (12), i.e., without corrections for the non-ideality of the gaseous phase and the non-inertness of the adsorbent at 216.4 K.

Table 4. The maximum percent contribution of the non-ideality of the CO_2 gaseous phase (Δ_Z) and non-inertness of the Sorbonorit 4 adsorbent exhibited as the temperature- and adsorption-induced deformations, $\Delta_{\eta(T)}$ and $\Delta_{\eta(a)}$, and their summarized contribution ($\Delta_{\eta(a+T)}$) to the differential molar isosteric heat of carbon dioxide adsorption onto Sorbonorit 4 within a temperature ranging from 216.4 to 393 K and up to a pressure of 6 MPa.

Т, К		$\Delta q_{ m st}$, %		
	Δ_{Z} , %	$\Delta_{\eta(T)}$, %	$\Delta_{\eta(a)}$, %	$\Delta_{\eta(a+T)}$, %
216.6	-2.92	0.02	-0.05	-0.03
243	-2.91	0.02	-0.02	0
273	-19.32	0.21	-0.67	-0.46
293	-42.65	0.74	-1.38	-0.64
333	-84.94	2.59	-0.19	2.40
393	-41.51	3.81	0.04	3.85

The subsequent decrease in the heat of CO₂ adsorption from 29 to 23 kJ·mol⁻¹ up to the adsorption value a = 2 mmol/g (see Figure 12) points to the non-uniformity of the adsorption sites. In fact, a carbon dioxide molecule has a rather large quadruple, which allows it to be involved in strong electrostatic interactions with heteroatoms incorporated into the carbon framework. Ma et al. revealed that the oxygen-containing functional groups present in porous carbon materials tightly entrap CO₂ through electrostatic interactions and contribute about 37% to the carbon dioxide capture [81]. On the other hand, micropores with a width close to the adsorbate molecule size play an important role as adsorption sites for carbon dioxide through dispersion interactions.

A further increase in the pressure leading to the progressive loading of pores of Sorbonorit 4 does not result in significant changes in the heat of CO₂ adsorption at low temperatures (curves 1–3, Figure 12). Rather, it is accompanied by a slow decrease in its value to 20–15 kJ·mol⁻¹ at elevated temperatures (curves 4–6) within a range of CO₂ uptake of 2–8 mmol·g⁻¹. This behavior can be attributed to the increased mutual attraction between the adsorbate molecules, which results in the formation of so-called adsorption associates [88]. Then, at high adsorption values, e.g., $a > 8 \text{ mmol·g}^{-1}$, the function $q_{st}(a)$ exhibits a steep decrease at $T \ge 273$ K, indicating a growth in repulsive forces caused by a

reduction in average distance between the adsorbed molecules and pore walls. With rising temperatures, these effects are observed at the lower pore fillings, which can be attributed to an increased contribution of thermal molecular motion.

It should be noted that the isosteric heat of adsorption does not depend on the temperature at the initial stage of the adsorption process. However, the changes in the value of q_{st} become temperature-dependent, as the carbon dioxide molecules occupy the high-energy adsorption centers. The divergence of the $q_{st}(a)$ curves can be attributed to changes in the temperature-dependent contributions from the gas phase compressibility *Z* and the derivative $(\partial P/\partial a)_T$ in the second term of the equation that determines the steepness of the isotherm (see Equation (12)).

When a molecule passes from the bulk phase to the adsorbed phase, a significant loss in the differential molar entropy s_1 takes place. By examining the entropy as a function of CO₂ uptake, one can gain insight into the state of the adsorbate molecules in the pore under the influence of adsorption potential created by the pore walls. The differential molar entropy of the CO₂–Sorbonorit 4 adsorption system was calculated from the following well-known formula [86]:

$$s_1 = s_g - q_{\rm st}/T, \tag{14}$$

where s_g is the entropy of the bulk gas.

The values of s_1 plotted as a function of CO₂ uptake in Figure 13 reflect the changes in the state of CO₂ molecules during their adsorption caused by the interaction with the adsorption sites compared to the bulk phase.



Figure 13. The differential molar isosteric entropy of the CO₂–Sorbonorit 4 adsorption system versus the CO₂ uptake at temperatures, K: 216,4 (1); 243 (2); 273 (3); 293 (4); 333 (5); 393 (6).

As follows from Figure 13, there is a sharp decrease in entropy at the adsorption value of about 0.3 mmol/g, indicating the transition of the CO₂ molecules to a bound state at the high-energy adsorption sites. In the range of adsorption values of 2–10 mmol/g, the entropy drop slows down significantly (or even increases slightly at low temperatures) with the gradual filling of the pore space with CO₂, resulting in the enhancement of the CO₂–CO₂ interactions. The difference in the behaviors of the *s*₁(*a*) values observed at *a* > 8 mmol/g for curves 1 and 2, on the one hand, and curves 3–6, on the other hand, can be explained by different degrees of thermal molecular motion averaging of adsorbate–adsorbent and adsorbate–adsorbate interactions at low and elevated temperatures.

The heat capacity of the adsorption system is a crucial parameter that determines the thermal stability of the system during adsorption/desorption cycles. Unlike the heat

capacity of any bulk phase, the heat capacity of the adsorption system depends not only on pressure and temperature but also on the number of adsorbed molecules. Thus, a complete thermodynamic description of the adsorption system includes an analysis of the differential molar isosteric heat capacity, C_a , as a function of all these parameters. The C_a value was calculated using the Kirchhoff equation [88]:

$$C_a = \left(\frac{\partial h_g}{\partial T}\right)_a - \left(\frac{\partial q_{\rm st}}{\partial T}\right)_{a'} \tag{15}$$

Here, $h_{\rm g}$ is the enthalpy of the gaseous phase.

Figure 14 shows the temperature dependence of the differential molar isosteric heat capacity of the CO₂–Sorbonorit 4 adsorption system for various amounts of adsorbed carbon dioxide.



Figure 14. Calculated temperature dependences of the differential molar heat capacity, C_a , of the CO₂–Sorbonorit 4 adsorption system for various values of CO₂ uptake, mmol/g: 1.0 (1); 1.5 (2); 2.0 (3); 2.5 (4); 3.0 (5); 4.0 (6); 5.0 (7); 6.(8); 7.0 (9); 8.0 (10); 9.0 (11); 10.0 (12).

It is clear from the figure that increased CO₂ uptake enhances the temperature dependence of C_a for the CO₂–Sorbonorit 4 adsorption system. At the value of adsorption a = 1 mmol/g, the heat capacity increases from 0.037 to 0.075 kJ/mol/K as the temperature rises from 216.4 to 373 K, and when the CO₂ uptake reaches 10 mmol/g, the value of C_a increases by ~6 times, with an increase in temperature by 100 °C. The obtained results can be interpreted as follows: At the early stage of adsorption, q_{st} does not depend on temperature (see Figure 12). Hence, the initial growth of C_a with rising temperature is determined by the temperature dependence of the enthalpy of carbon dioxide in the gaseous phase. However, as the amount of adsorbed carbon dioxide increases, the function q_{st} becomes strongly dependent on temperature, as illustrated by Figure 12, and this dependence makes a dominant contribution to $C_a(T)$.

4. Conclusions

In situ dilatometry measurements revealed that the microporous peat-derived carbon adsorbent, which was in a pre-stressed state after regeneration under vacuum, exhibited non-monotonic temperature-dependent deformation behaviors both during temperature change under vacuum and during carbon dioxide adsorption over a wide pressure range.

By comparing the thermal linear expansion coefficient of Sorbonorit 4 with that of silicon carbide-derived activated carbon, we found that the temperature-induced deformation behaviors of carbon porous materials are related to the phase and chemical composition, pore size distribution, and mechanical properties inherited from their precursors and affected by the synthesis route.

A detailed examination of the deformation of Sorbonorit 4 at the early stages of carbon dioxide adsorption revealed a jump-like deformation of Sorbonorit 4, the magnitude of which was found to be linearly dependent on reciprocal temperature. The derived linear relationship can be used for predicting the initial AID magnitude of the adsorbent at any temperature.

The initial jump-like changes in the sizes of Sorbonorit 4 during CO_2 adsorption coincided with the abrupt decrease in the differential molar isosteric heat of CO_2 adsorption, indicating the endothermic contribution of the adsorption-induced deformation. Both the dependencies of the adsorption-induced deformation of Sorbonorit 4 and the differential isosteric heat of carbon dioxide adsorption on the amounts of adsorbed molecules reflect the variations in the state of carbon dioxide molecules determined by the changes in the ratio between the adsorbate–adsorbent and adsorbate–adsorbate interactions.

The inversion of the adsorbent contraction–expansion transition during adsorption, occurring within the pressure range of 0.4 to 1 MPa, was interpreted in terms of the temperature-dependent effect of enhanced molecular mobility both on the intermolecular interactions of the adsorbed carbon dioxide molecules with the micropore walls and the adsorbent skeleton rigidity.

A simple qualitative model, which represented a carbon adsorbent as a set of graphitelike nanocrystallites connected by amorphous matter, made it possible to qualitatively explain the relationship between the adsorption- and temperature-induced deformation of nanocrystallites and the macroscopic deformation of the adsorbent granules.

The observed effects of contraction and expansion of carbon adsorbents and their temperature-dependent inversion during carbon dioxide adsorption should be taken into consideration when designing facilities for adsorption-based technologies for carbon dioxide capture and storage using commercial microporous carbon adsorbents and the thermal modes of their operation.

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