



# CO<sub>2</sub> Adsorption Property of Amine-Modified Amorphous TiO<sub>2</sub> Nanoparticles with a High **Surface** Area

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Received: 23 May 2018; Accepted: 3 July 2018; Published: 5 July 2018



Abstract: Carbon dioxide capture and storage (CCS) technologies have attracted a great deal of attention as effective measures to prevent global warming. Adsorption methods using porous materials seem to have several advantages over the liquid absorption methods. In this study, we have developed a synthesis method of new amorphous titanium dioxide (TiO<sub>2</sub>) nanoparticles with a diameter of 3 nm, a high surface area of 617  $m^2/g$  and a large amount of OH groups. Next, the surface of the amorphous TiO<sub>2</sub> nanoparticles was modified using ethylenediamine to examine whether CO<sub>2</sub> adsorption increases. Amorphous TiO<sub>2</sub> nanoparticles were successfully modified with ethylenediamine, which was used in excess due to the presence of a large amount of hydroxyl groups. The amorphous  $TiO_2$  nanoparticles modified with ethylenediamine show a higher  $CO_2$  adsorption capacity (65 cm<sup>3</sup>/g at 0 °C, 100 kPa) than conventional TiO<sub>2</sub> and mesoporous SiO<sub>2</sub>. We discuss the origin of the higher  $CO_2$  adsorption capacity in terms of the high specific surface area of the amorphous TiO<sub>2</sub> nanoparticles and the modification with ethylenediamine on the surface of the amorphous TiO<sub>2</sub> nanoparticles. The optimization of the amount of ethylenediamine bound on the particles increased the CO<sub>2</sub> adsorption capacity without pore blocking.

Keywords: titanium dioxide; amorphous; nanoparticles; ethylenediamine modification; CO<sub>2</sub> adsorption

# 1. Introduction

Carbon dioxide capture and storage (CCS) technologies have been well studied over the last decade to decrease CO<sub>2</sub> emission in the atmosphere which might contribute to global warming. Various CCS methods including solvent absorption, membrane separation, cryogenics fractionation and adsorption using solid adsorbents have been proposed and developed so far [1–4]. Currently, a liquid phase absorption method using amine (e.g., monoethanolamine) solution has been put into practical use [5]. However, this process has several problems such as corrosion of equipment, degradation of the solution, and, in addition, it requires heat regeneration. Meanwhile, adsorption methods using solid porous materials have attracted more attention over the liquid method [4]. For instance, the adsorption process has higher cycle stability and does not cause corrosion of equipment. Moreover, the adsorption process by using pressure differences is of advantage to the reduction of energy consumption for CO<sub>2</sub> regeneration.

Porous materials including mesoporous alumina, silica, activated carbon and zeolites [6–9] have been studied because these solid absorbent materials have high surface areas. Moreover, to improve CO<sub>2</sub> adsorption capacity, surface modification of the porous materials has been studied, including amine modification [10-17]. By using the amine-modified porous adsorbents, we can expect an



advantage of the chemical adsorption in addition to the physical adsorption, which arises from chemical reactions between the amine groups and CO<sub>2</sub>.

Titanium dioxide has attracted the attention of researchers because of its remarkable properties. It has been applied to photocatalyst and dye-sensitized solar cells [18,19]. In order to improve these functions, many studies have been reported on controlling the structure of  $TiO_2$  and increasing the surface area. The reported  $TiO_2$  has a large variety of structures such as particulate, tube, rod, sheet, and sponge [20–25]. Although these many structures of  $TiO_2$  with high surface area have been reported, there have been fewer reports that amine-modified  $TiO_2$  is applied to  $CO_2$  adsorbents compared to the other porous materials such as  $SiO_2$  [26–31]. This is because  $TiO_2$  generally has small amount of OH groups on the surface to adsorb the amine species and it has been difficult for  $TiO_2$  to be modified with large amount of amines. Amine modification on the  $TiO_2$  requires the presence of OH groups on the surface.

We have developed a synthesis method of new amorphous  $TiO_2$  nanoparticles with high surface area and with higher concentration of OH groups. The amorphous  $TiO_2$  is more reactive with cations such as Li<sup>+</sup> compared to conventional anatase and rutile crystal  $TiO_2$ . The amorphous  $TiO_2$ nanoparticles having OH groups and a high surface area, is considered to enable to be modified with amine used in excess. In this study, we describe the modification of the amorphous  $TiO_2$  nanoparticles with ethylenediamine and their  $CO_2$  adsorption capacity.

#### 2. Materials and Methods

#### 2.1. Synthesis of Amorphous TiO<sub>2</sub> Nanoparticles

According to our previous report [32], 1.4 mL of titanium tetraisopropoxide (TTIP) was mixed with 30 mL of THF and the mixture was stirred at room temperature for 1 h. Next, 1.6 mL of water was added to cause hydrolysis reactions and white precipitates of TiO<sub>2</sub> was immediately formed. Finally, amorphous TiO<sub>2</sub> was collected by centrifugation and dried at 90 °C. The sample is labeled as a-TiO<sub>2</sub>-THF. As a reference, we prepared commercially available P25-TiO<sub>2</sub> and titanium dioxide synthesized from TTIP without THF solvent. They are labeled as P25-TiO<sub>2</sub> and TiO<sub>2</sub>-solventless, respectively.

# 2.2. Preparation of Amine-Modified TiO<sub>2</sub> Nanoparticles

The titanium dioxide powder was modified with amines by an impregnation process. The  $TiO_2$  samples (0.2 g) were added into 15 mL of 75 wt % ethylenediamine (EDA) solution in ethanol and stirred at room temperature for 3 h. Then the amine-modified samples were centrifuged and washed by using 10 mL of ethanol. After that, they were dried at 90 °C. The final products are labeled as a-TiO<sub>2</sub>-THF-EDA, P25-TiO<sub>2</sub>-EDA and TiO<sub>2</sub>-solventless-EDA, respectively. In addition, mesoporous silica (MCM-41), which had been well studied as a porous support, was also modified with EDA by the same process and it is labeled as MCM-41-EDA.

#### 2.3. Charactarization

The crystallinity of the TiO<sub>2</sub> samples were evaluated by an X-ray diffraction (XRD) pattern using PANalytical X'Pert PRO with Cu K $\alpha$  X-ray (1.54 Å). The particle size and the morphology were measured by the transmission electron microscopy (TEM) images which were recorded on Hitachi H800 electron microscope (Tokyo, Japan) at an acceleration voltage at 200 kV. Nitrogen adsorption-desorption isotherms were measured at 77 K using BELSORP-max (MicrotracBEL Corp., Osaka, Japan). The specific surface area was calculated by the Brunauer-Emmett-Teller (BET) method from nitrogen adsorption isotherms. The pore size distribution and pore volume were calculated by the Brunauer-Joyner-Halenda (BJH) method. Fourier transform infrared (FT-IR) spectra were recorded on a Shimadzu IRAffinity-1 (Kyoto, Japan) in transmission mode with a scan number of 100. The amount of EDA loaded on the samples were measured by thermogravimetry (TG) analysis under air atmosphere at 20–800 °C at a heating rate of 5 °C/min. The adsorption isotherms of CO<sub>2</sub> were measured at 0 °C with BELSORP-max.

#### 3. Results and Discussion

The XRD patterns of P25-TiO<sub>2</sub>, TiO<sub>2</sub>-solventless and a-TiO<sub>2</sub>-THF are shown in Figure 1. The commercial P25-TiO<sub>2</sub> is composed of a mixture of anatase and rutile structures of TiO<sub>2</sub>. The TiO<sub>2</sub>-solventless sample synthesized without organic solvent of THF exhibited an anatase crystal structure of TiO<sub>2</sub>. On the other hand, the sample synthesized with THF did not show any peaks, indicating that a-TiO<sub>2</sub>-THF was amorphous phase. A possible reason would be that THF inhibited the formation of TiO<sub>2</sub> crystal structure. The molecules of TTIP were surrounded by the THF molecules in the solvent. Upon addition of water for hydrolysis reaction, the THF molecules would hinder aggregation of TiO<sub>2</sub> particles and particle growth. According to XRD pattern of the TiO<sub>2</sub> samples after amine modification, the XRD diffraction peaks were similar to the ones before amine modification, indicating that the crystal structures was not changed.



**Figure 1.** XRD patterns of TiO<sub>2</sub> samples (a) amorphous TiO<sub>2</sub> (b) TiO<sub>2</sub> synthesized without solvent; (c) commercially available P25-TiO<sub>2</sub>.

Figure 2 shows TEM images of the TiO<sub>2</sub> samples. Nanoparticles with 3 nm in size were observed for a-TiO<sub>2</sub>-THF, while the particle size of TiO<sub>2</sub>-solventless and P25-TiO<sub>2</sub> were about 10 and 20–60 nm, respectively. Figure 3 shows N<sub>2</sub> adsorption isotherms and pore size distribution of TiO<sub>2</sub> samples. Their specific surface areas calculated by the BET method from the N<sub>2</sub> adsorption isotherms are summarized in Table 1. The TiO<sub>2</sub> sample synthesized with THF solvent had the highest surface area of about  $S_{BET} = 617 \text{ m}^2/\text{g}$  which is 25 times higher than P25-TiO<sub>2</sub> ( $S_{BET} = 63 \text{ m}^2/\text{g}$ ). Specific surface area of TiO<sub>2</sub>-solventless was 241 m<sup>2</sup>/g. According to N<sub>2</sub> adsorption isotherm, TiO<sub>2</sub>-THF showed a high adsorbed amount at low relative pressure suggesting the presence of micropores in addition to mesopores. The high specific surface area of a-TiO<sub>2</sub>-THF could be caused by the micropores. When TiO<sub>2</sub>-THF was synthesized, the THF molecules could surround TTIP molecules. The THF molecules stabilized TTIP molecules and inhibited the hydrolysis reaction of TTIP. The TiO<sub>2</sub> particles did not grow well, thus smaller nanoparticles were formed. When THF was dried, vacancy from THF molecules would become micropores. On the other hand, TiO<sub>2</sub>-soventless and P25-TiO<sub>2</sub> had mesopores and macropores, respectively.



**Figure 2.** TEM images of  $TiO_2$  samples (**a**) amorphous  $TiO_2$  (**b**)  $TiO_2$  synthesized without solvent; (**c**) commercially available P25-TiO<sub>2</sub>.



**Figure 3.** (A)  $N_2$  adsorption/desorption isotherms and (B) pore size distributions of TiO<sub>2</sub> samples (a) amorphous TiO<sub>2</sub> (b) TiO<sub>2</sub> synthesized without solvent (c) commercially available P25-TiO<sub>2</sub>.

	S <sub>BET</sub> (m <sup>2</sup> /g)	V (cm <sup>3</sup> /g)	Amount of Loaded Amine (wt %)	Amount of Loaded Amine (mg/m <sup>2</sup> )
a-TiO <sub>2</sub> -THF a-TiO <sub>2</sub> -THF-EDA	617	1.582	-	-
	472	1.134	15.1	0.245
TiO <sub>2</sub> -solventless	241	0.356	-	-
TiO <sub>2</sub> -solventless-EDA	201	0.330	4.7	0.197
P25-TiO <sub>2</sub> P25-TiO <sub>2</sub> -EDA	63	0.486	-	-
	40	0.502	1.1	0.175
MCM-41	978	0.504	-	-
MCM-41-EDA	360	0.148	12.2	0.125

Table 1. Specific surface area, pore volume and amount of loaded amine of TiO<sub>2</sub> samples and MCM-41.

The specific surface area and pore volume decreased with increasing amount of loaded amine. According to pore size distributions calculated from the nitrogen adsorption isotherms of a-TiO<sub>2</sub>-THF-EDA and TiO<sub>2</sub>-solventless-EDA after amine modification, the peak did not shift from the samples before amine modification and only the pore volume was reduced (see supporting information Figure S1). This results suggest that TiO<sub>2</sub> was not uniformly modified with EDA on the micropores and mesopores surface, but TiO<sub>2</sub> was rather modified with EDA as if EDA blocked some of the pores.

FT-IR spectra of TiO<sub>2</sub> and TiO<sub>2</sub>-EDA samples were measured to confirm the presence of OH groups of TiO<sub>2</sub> and amine modification onto TiO<sub>2</sub> samples, respectively. The absorption peaks at 400–1000 cm<sup>-1</sup> of all the samples in Figure 4 were ascribed to lattice vibration of TiO<sub>6</sub> octahedral crystal. The strong absorption peaks at 1630 and around 3400 cm<sup>-1</sup> were attributed to the bending vibration and the stretching vibration of O–H bonds, respectively. The peak intensity of a-TiO<sub>2</sub>-THF was stronger than TiO<sub>2</sub>-solventless and P25-TiO<sub>2</sub>, indicating that a-TiO<sub>2</sub>-THF samples had more amount of OH groups on the surface than TiO<sub>2</sub>-solventless and P25-TiO<sub>2</sub>. The peaks attributed C–H and C–O vibration at 2974 cm<sup>-1</sup> and 1126 cm<sup>-1</sup> respectively were observed for a-TiO<sub>2</sub>-THF. These peaks were derived from OC<sub>3</sub>H<sub>7</sub> groups of TTIP. The hydrolysis reaction of TTIP to form TiO<sub>2</sub> is as follows:

# $Ti(OC_{3}H_{7})_{4} + xH_{2}O \rightarrow Ti(OC_{3}H_{7})_{4-x}(OH)_{x} + xC_{3}H_{7}OH$ $\equiv Ti-OH + HO-Ti \equiv \rightarrow \equiv Ti-O-Ti \equiv + H_{2}O$



**Figure 4.** FT-IR spectra of TiO<sub>2</sub> samples (**A**) before amine modification (a) amorphous TiO<sub>2</sub> (b) TiO<sub>2</sub> synthesized without solvent (c) commercially available P25-TiO<sub>2</sub>, and (**B**) those after amine modification (d) a-TiO<sub>2</sub>-THF-EDA (e) TiO<sub>2</sub>-solventless-EDA (f) P25-TiO<sub>2</sub>-EDA.

When the hydrolysis reaction of TTIP occurred in the presence of the THF solvent, the intermediates  $Ti(OC_3H_7)_{x-4}(OH)_x$  were stabilized by an interaction with oxygen atoms of THF molecules. This interaction would be weak but enough to stabilize the intermediates. In consequence, unreacted groups such as OH or  $OC_3H_7$  remained on the surface of nanoparticles after THF was removed by drying.

After EDA modification, the new peaks emerged at 1031, 1515 cm<sup>-1</sup> were assigned to C–N stretching vibration and N–H<sub>2</sub> vibration in the primary amine group (RNH<sub>2</sub>) respectively, indicating the presence of EDA on the surface of a-TiO<sub>2</sub>-THF and TiO<sub>2</sub>-solventless. Weak peaks at 1031 cm<sup>-1</sup> attributed amine group were observed for P25-TiO<sub>2</sub>, indicating that little amounts of amine were loaded on P25-TiO<sub>2</sub> since P25-TiO<sub>2</sub> did not have enough OH groups on the surface for adsorption of amine species. The intensity of the absorption peaks ascribed to OH groups were not much decreased by the EDA modification for all samples, suggesting that OH groups remained from the amine treatment. The absorption peaks at 1330 cm<sup>-1</sup> could be ascribed to skeletal vibration of –NCOO by adsorbed gaseous CO<sub>2</sub> in the atmosphere [33].

The specific surface area, pore volume and the amount of EDA loaded on the samples measured by TG analysis were summarized in Table 1 and compared with those from MCM-41. The loaded amount of EDA was 15.1 wt % (a-TiO<sub>2</sub>-THF-EDA), 4.7 wt % (TiO<sub>2</sub>-solventless-EDA), 1.1 wt % (P25-TiO<sub>2</sub>-EDA)

and 12.2 wt % (MCM41-EDA), respectively. Amorphous  $TiO_2$  nanoparticles were modified with the largest amount of amine. One of the reasons seems to be the fact that a-TiO<sub>2</sub>-THF has a high specific surface area. Table 1 also shows the values of the loaded amount of amine per surface area. In per unit surface area, amine-modified amount onto a-TiO<sub>2</sub>-THF was the highest too. Since there is a large amount of OH groups on the surface of a-TiO<sub>2</sub>-THF, many amines could be loaded there.

The adsorption isotherms of CO<sub>2</sub> at 0 °C are shown in Figure 5. The effect of the modification with amine on the enhancement of CO<sub>2</sub> adsorption capacity was more largely for a-TiO<sub>2</sub>-THF than TiO<sub>2</sub>-solventless and P25-TiO<sub>2</sub>. This result is due to the higher contents of OH groups of a-TiO<sub>2</sub>-THF than TiO<sub>2</sub>-solventless and P25-TiO<sub>2</sub> with anatase and rutile crystal structure. Two schemes of enhanced CO<sub>2</sub> adsorption onto amine-modified material containing Ti and OH groups have been proposed [34]. First, –NH<sub>2</sub> groups of amines react with CO<sub>2</sub> to form carbamate species according to the equation shown below.



$$CO_2 + 2RNH_2 \leftrightarrow RNH_3^+ + RNHCOO^-$$

**Figure 5.**  $CO_2$  adsorption isotherms of TiO<sub>2</sub> samples MCM-41 at 0 °C (open symbols) before amine modification (closed symbols) after amine modification.

Second,  $CO_2$  adsorption is promoted by electrostatic force between  $CO_2$  molecules and amine molecules and –OH groups of TiO<sub>2</sub> surface. The high CO<sub>2</sub> adsorption capacity of a-TiO<sub>2</sub>-THF is possibly due to the enhancement of the second scheme because a-TiO<sub>2</sub>-THF has high surface area and a high surface concentration of -OH groups. In addition, a-TiO<sub>2</sub>-THF-EDA exhibited a higher CO<sub>2</sub> adsorption capacity than MCM-41-EDA even though the specific surface area of a-TiO<sub>2</sub>-THF was lower than that of MCM-41 (978 m<sup>2</sup>/g). In the CO<sub>2</sub> adsorption isotherm, the rise at the low-pressure side is due to adsorption by the reaction of amines and  $CO_2$  described as above the first scheme. Generally, adsorption volume at 20–100 kPa is considered as physical adsorption volume. In that case, the slope of the adsorption isotherms before and after amine modification should be similar at the high pressure side (20–100 kPa). However, the slope of the adsorption isotherm of MCM-41-EDA was smaller than that of the MCM-41. This is possibly because the effective surface area for the physical adsorption was decreased by amine modification. On the other hand, the slope of the adsorption isotherm of a-TiO<sub>2</sub>-THF sample was not decreased and rather increased. The improvement of adsorption capacity can be explained by the above second scheme. The OH groups should be exposed on the surface. Therefore, with regard to the MCM-41 which had few OH groups (see supporting information Figure S2), the physical adsorption mainly occurs. On the other hand, as described above, according to FT-IR

measurement of a-TiO<sub>2</sub>-THF, a part of OH groups was present on the surface of TiO<sub>2</sub>, which enhances adsorption of the second scheme process. Along with increasing the loaded EDA amount, CO<sub>2</sub> adsorption capacity decreased (see supporting information Figure S3). Excessive amine modification caused a pore blocking which reduces the surface area. In addition, OH groups on the surface was covered with amine molecules. In this study, the optimized amine loading effectively could increase the CO<sub>2</sub> adsorption capacity without pore blocking.

## 4. Conclusions

This work demonstrates synthesis methods of amorphous  $TiO_2$  nanoparticles with a high surface area and large amount of OH groups. The amorphous  $TiO_2$  nanoparticles were synthesized by using THF as a solvent in hydrolysis reaction of TTIP. According to the results of the TEM observation, the amorphous  $TiO_2$  had particle size of 3 nm, which is the smallest size among the conventional ones. The nitrogen adsorption isotherm had an initial rise derived from the micropores at a low-pressure area. The specific surface area calculated from the nitrogen adsorption isotherm was 617 m<sup>2</sup>/g, which was 10 times larger than the commercially available P25-TiO<sub>2</sub>.

Next, our amorphous  $TiO_2$  nanoparticles were modified by ethylenediamine. For modification of amorphous  $TiO_2$  nanoparticles, amine was used in excess because of not only the high specific surface area but also many OH groups which adsorb amine molecules. The amine-modified amorphous  $TiO_2$  nanoparticles showed the highest  $CO_2$  adsorption capacity among those of the amine-modified  $TiO_2$  supports and mesoporous silica MCM-41. The possible reasons for the high  $CO_2$  adsorption capacity are (1) the high specific surface area of the amorphous  $TiO_2$  nanoparticles which contributes to the physical immobilization with  $CO_2$ ; (2) the high loading of amine molecules which react with  $CO_2$  effectively and (3) the tripartite hydrogen bonding interactions among the amine molecules,  $CO_2$  and OH groups on the  $TiO_2$  surface. The new amorphous  $TiO_2$  nanoparticles having OH groups and a high surface area is a promising material for  $CO_2$  adsorption.

**Supplementary Materials:** The following are available online at http://www.mdpi.com/2504-5377/2/3/25/s1. Figure S1: Nitrogen adsorption isotherms and pore size distributions after amine modification; Figure S2: FTIR spectra of MCM41; Figure S3: CO<sub>2</sub> adsorption volume at 100 kPa of amine-modified samples synthesized by using various ethylenediamine/ethanol concentration; Figure S4: TG curves of amine-modified samples; Table S1: micro-, meso- and macro pore volume of TiO<sub>2</sub> samples.

Author Contributions: M.O. and N.N. conceived and designed the experiments; M.O. performed the experiments, analyzed the data and wrote the paper; Y.H. and Y.U. contributed to critical revision of the manuscript.

Funding: This work was supported by JSPS KAKENHI Grant Number 16K14458.

**Acknowledgments:** The TEM measurements were carried out by using a facility in Research Center for Ultrahigh Voltage Electron Microscopy, Osaka University.

Conflicts of Interest: The authors declare no conflict of interest.

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