



# Proceedings Inhibition of Polymer Photodegradation by Incorporation of Coffee Silverskin<sup>+</sup>

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**Abstract:** Over the last years, the trend associated with the incorporation of materials from renewable resources into polymer technology is getting significantly more vital. Researchers are trying to transfer the properties of natural raw materials into the polymer world. Therefore, different natural materials are more often investigated as potential additives for polymers. Such an effect is noted for the coffee industry byproducts, such as coffee silverskin. Because of the relatively high contents of compounds showing antioxidant activity, such as caffeine, polyphenols, tannins, or melanoidins, this byproduct could be considered not only as a filler, but also as a potential modifier for polymer materials. Its antioxidant activity is comparable to commercially available antioxidants applied in polymer technology. Therefore, in the presented paper, we examined the influence of the coffee silverskin (from 1 to 20 wt %) on the thermal and mechanical performance of polyethylene-based composites. Moreover, materials were subjected to accelerated aging tests in the UV chamber, which revealed that coffee silverskin could inhibit the photodegradation of the polymer matrix. Therefore, this byproduct should be considered as an exciting alternative for the conventional lignocellulosic fillers, which could provide additional features to polymer composites.

**Keywords:** coffee silverskin; antioxidant activity; accelerated aging; polymer composites; mechanical properties

# 1. Introduction

Coffee silverskin is one of the byproducts generated by the coffee industry. It is not the most problematic one, because it stands only for ~4.2 wt % of coffee beans, so it is not generated in massive amounts [1]. Moreover, it is generated during roasting of coffee, performed at relatively high temperatures, resulting in the very low moisture content, which makes it very interesting for the potential industrial applications. The chemical composition of coffee silverskin is relatively similar to the materials conventionally applied as fillers for wood-polymer composites. It usually contains between 60 and 75% of fiber, mostly insoluble, including up to 30% of lignin [2]. Coffee silverskin also contains 15–19% of proteins, which can provide additional features to the polymer composites [3]. They may act as plasticizers, but due to the presence of additional functional groups may also provide additional possibilities for interfacial interactions with polar polymer matrices.

Nevertheless, the most interesting is the content of compounds showing antioxidant activity, such as caffeine, polyphenols, tannins, or melanoidins generated during roasting of coffee [4]. Generally, silverskin is considered as an excellent source of these com-

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**Copyright:** © 2020 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (http://creativecommons.org/licenses /by/4.0/). pounds, however, their content, and to be more precise, the yield of their recovery strongly depends on the extraction method or selected solvents [5]. Therefore, reported values of the total phenolics content varied from 2.3 to even 36.0 mg gallic acid/g, resulting in the antioxidant activity, determined by the (2,2-diphenyl-1-picryl-hydrazyl-hydrate) (DPPH) assay, in the range of 9.21–18.24  $\mu$ mol TR/g [6]. The main compounds showing antioxidant activity present in coffee silverskin are caffeine, epicatechin, as well as caffeic, caffeoylquinic, *p*-coumarcylquinic, ferulic, feruloylquinic, gallic, syringic, *trans*-cynnamic, and vanillic acids [7]. Other, less popular include hyperoside, kaempferol, naringin, quercetin, or quinine.

The compounds mentioned above might potentially act as antioxidants for various polymer matrices. Therefore, the presented paper aimed to investigate the impact of the coffee silverskin content (from 1 to 20 wt %) on the thermal and mechanical properties of polyethylene-based composites. The performance of the composites was also analyzed after accelerated aging tests in the UV chamber to determine the efficiency of the coffee silverskin as a potential antioxidant for polymer composites.

# 2. Experimental

# 2.1. Materials

High-density polyethylene (HDPE), type M300054, obtained from SABIC (The Netherlands) was applied as a matrix for the preparation of investigated composites. According to producer data, its density equaled 0.954 g/cm<sup>3</sup>, and it was characterized by a melt flow rate of 30 g/10 min (190 °C, 2.16 kg). Coffee silverskin was obtained from the local coffee roastery.

#### 2.2. Preparation of Polymer Composites

The composites were prepared by mixing in a molten state. To facilitate the mixing process, the HDPE pellets were pulverized into a fine powder using a Tria 25-16/TC-SL high-speed knife grinder. Then the polymeric powder was preliminary mixed with 1, 2, 5, 10, or 20 wt % of coffee silverskin. Preliminary to mixing in molten state compositions were dried in laboratory cabined dryer Memmert ULE 500 for 12 h at 70 °C. The mixtures were processed using a ZAMAK EH16.2D co-rotating twin-screw extruder operating at 100 rpm with the maximum temperature of the process of 190 °C. The obtained materials were cooled in forced airflow and pelletized. The resulting composites were then compression molded at 170 °C and 4.9 MPa for 2 min and then kept under pressure at room temperature for another 5 min to enable solidification of the material. The unfilled HDPE was processed along with its composites. The specimens were named in reference to their filler content, as PE and PE/XŁK, where X stands for the filler content.

#### 2.3. Accelerated Aging of Composites

After seven days of conditioning at room temperature, samples were subjected to the accelerated aging process with the use of using TestAn Xentest 2200 xenon light accelerated aging chamber. The aging process was conducted under 0.55 W/m<sup>2</sup> irradiance at 340 nm UV light intensity, with constant max. temperature up to 63 °C during dry UV light exposure. The samples were subjected to UV light exposure in two time periods of 100 and 200 h, which was used for sample description in the further part of the manuscript.

#### 2.4. Measurements

Oxidation induction time (OIT) of analyzed composites was determined by the differential scanning calorimetry (DSC) analysis. The 5 mg samples were placed in aluminum crucibles with pierced lids. They were heated from 20 to 190 °C with a heating rate of 20 °C/min in nitrogen, then kept at 190 °C for 5 min in nitrogen, and then gas was switched to oxygen and the time required for sample oxidation was measured. The measurements were conducted using a Netzsch 204F1 Phoenix apparatus.

To determine the crystallization and melting temperatures, as well as the crystalline structure of analyzed composites, DSC analysis was applied. The 5 mg samples were placed in aluminum crucibles with pierced lids. They were heated from 20 to 250 °C with a heating rate of 10 °C/min and then cooled back to the initial temperature with a cooling rate of 10 °C/min. The heating/cooling cycle was performed twice in order to erase the thermal history of the polymers during the first heating. The measurements were conducted using a Netzsch 204F1 Phoenix apparatus, in an inert atmosphere of nitrogen. The crystallinity degree  $X_{cr}$  of the samples was calculated using formula (1):

$$X_{cr} = (\Delta H_m/((1 - \varphi) \cdot \Delta H_{m100\%})) \cdot 100\%$$

where:  $\Delta H_m$ —melting enthalpy of a sample,  $\Delta H_{m100\%}$ —melting enthalpy of 100% crystalline polyethylene,  $\Delta H_{m100\%}$  = 293.6 J/g [8],  $\varphi$ —filler weight fraction.

The tensile strength and elongation at break were estimated following ISO 527 for dumbbell samples type 5B. Tensile tests were performed on a Zwick/Roell Z020 apparatus with a cell load capacity of 20 kN at a constant speed of 5 mm/min.

# 3. Results

#### 3.1. Thermal Properties

To determine the potential antioxidant activity of coffee silverskin, and its effect on the thermooxidative stability of prepared composites, the oxidation induction time was determined by DSC analysis. In Figure 1, there are presented thermograms for unaged samples, as well as for PE and PE/20ŁK depending on aging time. Moreover, in Table 1, there are presented values of oxidation induction time for all samples, along with the results of "conventional" DSC analysis. It can be seen that the introduction of coffee silverskin caused a significant delay in composites' oxidation process, confirming the antioxidant activity of this byproduct. Values of OIT were over 2.5 times bigger, even for 1 wt % content of filler, compared to 20 min for neat polyethylene matrix. For higher contents, OIT values reached even 130–140 min. Above 5 wt %, the effect was not as distinct. Accelerated aging of samples significantly reduced the thermo-oxidative stability of analyzed composites. Nevertheless, sample PE/20ŁK still showed higher stability than neat PE, even after 200 hours of aging, which points to the very high activity of antioxidant compounds present in coffee silverskin. In Figure 1b, it can also be seen that magnitude of signals was significantly reduced after aging, indicating partial oxidation of the material.

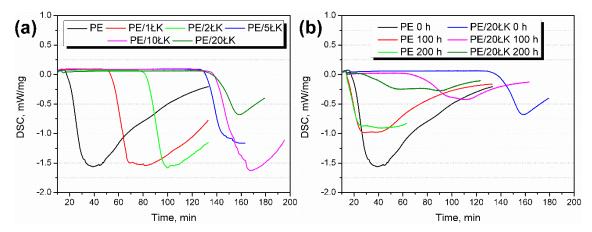


Figure 1. Thermograms of (a) unaged composite samples, and (b) PE and PE/20ŁK composites depending on aging time.

Incorporation of the coffee silverskin into the polyethylene matrix did not substantially affect the thermal properties of unaged samples. Temperatures of melting and crystallization were hardly affected by the filler incorporation. The decrease of the melting enthalpy ( $\Delta$ Hm) was noted, which was associated with the reduced share of the polymer matrix. Nevertheless, the degree of crystallinity was slightly increased from 63.8 to 66.6%, which was related to the nucleating activity of filler. This phenomenon can be related to the previously described cases of the interaction of lignocellulosic fillers as heterogeneous nucleants of the PE [9].

Aging of samples did not cause noticeable changes in the  $T_m$  and  $T_{cr}$  values, but affected the melting enthalpy and degree of crystallinity, especially for unmodified polyethylene and sample PE/1ŁK. After 200 h of aging, the degree of crystallinity for these samples was significantly reduced, pointing to the partial decomposition of the material.

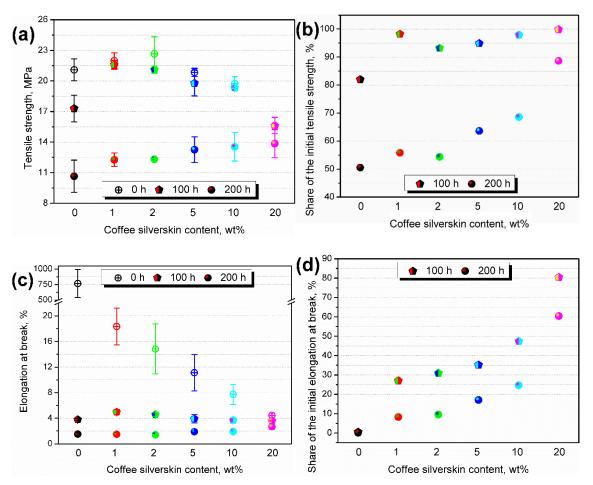
Sample	Aging Time, h	T <sub>m</sub> , °C	T <sub>cr</sub> , °C	∆H <sub>m</sub> , J/g	Xcr,%	OIT, min
PE	0	134.1	112.8	187.4	63.8	20.0
	100	132.5	112.4	197.8	67.4	13.5
	200	133.2	110.6	146.6	49.9	13.5
PE/1ŁK	0	133.2	113.1	190.5	65.5	54.8
	100	132.7	112.3	189.1	65.1	13.5
	200	133.8	113.5	152.1	52.3	13.5
PE/2ŁK	0	134.1	113.0	188.9	65.7	83.9
	100	132.8	112.9	181.8	63.2	15.1
	200	132.7	112.8	182.1	63.3	13.5
PE/5ŁK	0	133.3	113.3	186.5	66.9	131.9
	100	133.6	113.1	174.8	62.7	15.8
	200	132.0	113.8	177.9	63.8	16.5
PE/10ŁK	0	133.0	113.4	174.5	66.0	139.5
	100	132.3	112.7	170.9	64.7	17.9
	200	132.4	112.9	170.7	64.6	16.7
PE/20ŁK	0	132.6	112.5	156.4	66.6	140.4
	100	133.6	112.2	148.5	63.2	76.8
	200	131.6	113.6	159.2	67.8	30.1

Table 1. Results of differential scanning calorimetry (DSC) analysis for prepared composites.

# 3.2. Mechanical Properties

In Figure 2, there are presented tensile properties of the analyzed PE/ŁK composites before and after the accelerated aging. The initial values of tensile strength were at a similar level for neat polyethylene matrix and composites containing up to 5 wt % of coffee silverskin. For higher loadings of filler, drop of the tensile strength was noted. Similar effects were observed in our other work [10]. Such a phenomenon is associated with the insufficient interfacial adhesion between filler and matrix, which was especially pronounced for the elongation at break. This parameter was drastically decreased when solid particles were introduced into the polymer matrix.

The aging of composite samples resulted in noticeable changes in their mechanical performance. Nevertheless, a very beneficial influence of coffee silverskin can be noted. For the neat polyethylene matrix, a drop of tensile strength, and especially elongation at break was very significant after aging. Similar effects were noted by other researchers [11]. It can be seen that the incorporation of even 1 wt % of coffee silverskin caused a noticeable improvement of the mechanical performance. The share of the initial values of mechanical properties was increasing with the loading of coffee silverskin. It confirms the results of the thermal properties analysis indicating the elongation of oxidation induction time with the addition of applied coffee industry byproduct.



**Figure 2.** Plots showing (**a**) tensile strength, (**b**) share of the initial tensile strength, (**c**) elongation at break, and (**d**) share of the initial elongation at break for PE/ŁK composites depending on the filler content.

# 4. Conclusions

In the presented work, the polyethylene-based composites modified with natural waste filler in the form of coffee silverskin were produced and tested. It has been shown that incorporated filler shows novel functionality, in the form of an antioxidant effect, resulting from the active compounds it contains. The introduction of a lignocellulosic filler with a complex structure and chemical composition made it possible to suppress the aging effects caused by exposure to UV radiation in controlled conditions. Thermal analysis, performed by differential scanning calorimetry, was correlated to and remained in agreement with the results of mechanical studies performed using tensile test evaluation. Based on the obtained results, it can be concluded that coffee silverskin can be used as a low-cost filler for the production of highly-filled composites with additional functionality in comparison to different lignocellulosic waste fillers, as well as an effective additive, which influences negative effects of oxidation induced by aging processes.

**Author Contributions:** A.H. and M.B. conceived and designed the experiments; A.H., M.B., P.K. and O.M. performed the experiments; A.H., M.B., P.K. and O.M. analyzed the data; A.H. and M.B. contributed reagents/materials/analysis tools; A.H., M.B., P.K. and O.M. wrote the paper. All authors have read and agreed to the published version of the manuscript.

Conflicts of Interest: The authors declare no conflict of interest.

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