

Article

# Effect of 1.5 MeV Proton Irradiation on Superconductivity in FeSe<sub>0.5</sub>Te<sub>0.5</sub> Thin Films

Toshinori Ozaki <sup>1,\*</sup>, Takuya Kashihara <sup>1</sup>, Itsuhiro Takeya <sup>2</sup> and Ryoya Ishigami <sup>3</sup> 

<sup>1</sup> Department of Nanotechnology for Sustainable Energy, Kwansei Gakuin University, 2-1 Gakuen, Sanda 669-1337, Japan; ete03144@kwansei.ac.jp

<sup>2</sup> Department of Electronic Science and Engineering, Kyoto University, Nishikyo-ku, Kyoto 615-8510, Japan; takeya@kuee.kyoto-u.ac.jp

<sup>3</sup> The Wakasa Wan Energy Research Center, Nagatani, Tsuruga 914-0192, Japan; rishigami@werc.or.jp

\* Correspondence: tozaki@kwansei.ac.jp

**Abstract:** Raising the critical current density  $J_c$  in magnetic fields is crucial to applications such as rotation machines, generators for wind turbines and magnet use in medical imaging machines. The increase in  $J_c$  has been achieved by introducing structural defects including precipitates and vacancies. Recently, a low-energy ion irradiation has been revisited as a practically feasible approach to create nanoscale defects, resulting in an increase in  $J_c$  in magnetic fields. In this paper, we report the effect of proton irradiation with 1.5 MeV on superconducting properties of iron–chalcogenide FeSe<sub>0.5</sub>Te<sub>0.5</sub> films through the transport and magnetization measurements. The 1.5 MeV proton irradiation with  $1 \times 10^{16}$  p/cm<sup>2</sup> yields the highest  $J_c$  increase, approximately 30% at 5–10 K and below 1 T without any reduction in  $T_c$ . These results indicate that 1.5 MeV proton irradiations could be a practical tool to enhance the performance of iron-based superconducting tapes under magnetic fields.

**Keywords:** superconductor; irradiation; critical current



**Citation:** Ozaki, T.; Kashihara, T.; Takeya, I.; Ishigami, R. Effect of 1.5 MeV Proton Irradiation on Superconductivity in FeSe<sub>0.5</sub>Te<sub>0.5</sub> Thin Films. *Quantum Beam Sci.* **2021**, *5*, 18. <https://doi.org/10.3390/qubs5020018>

Academic Editors: Akihiro Iwase and Lorenzo Giuffrida

Received: 31 March 2021

Accepted: 25 May 2021

Published: 4 June 2021

**Publisher's Note:** MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



**Copyright:** © 2021 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

## 1. Introduction

Iron-based superconductors have a reasonably high superconducting transition temperature  $T_c$ , very high upper critical magnetic fields  $H_{c2}$ , quite a small anisotropy  $\gamma$  and larger critical grain boundary angle than cuprate superconductors, which make them promising for high-field applications such as superconducting magnet and generators [1–5]. The use of superconducting materials for high field applications is limited by the critical current density  $J_c$  in magnetic fields, which can be sustained by pinning the vortices (flux pinning) at structural defects with nano-meter sizes such as cracks, voids, grain boundaries and secondary phases [6,7]. The ion irradiation is a useful tool to generate the desired defect structure. Depending on the ion species, ion energy and the properties of the target materials, ion irradiation enables the creation of defects with well-controlled morphology and density, such as point [8], cluster [9–12] and columnar [13–15] defects. Early works on the ion irradiation of cuprate (Cu–O based) high- $T_c$  superconductors (HTS) for improving  $J_c$  in the magnetic field have mostly focused on the high-energy, over hundreds of MeV, heavy ion irradiation [13–15]. At this energy range, the irradiation of superconducting materials by the swift heavy ion mainly causes electronic excitation and ionization of the target atoms. As a result, continuous amorphous tracks are formed in a process that can be described as the rapid melting and solidification of nm-sized columns in the path of an ion. Even though the heavy ion tracks proved to be very effective pinning defects, this approach has been limited to fundamental studies of the vortex matter.

Recently, ion irradiation of HTS with a low energy has received a renewed interest as a practical method for increasing  $J_c$  in magnetic fields, due to the compact accelerator, lower radioactivity and less costly operation [9–12]. Low-energy ion irradiation utilizes a

different mechanism for the creation of vortex pinning defects. The electronic excitation and ionization are low enough so the heat can dissipate without damaging the materials. The low-energy ion irradiation leads to the collision of the ion with the target atom nuclei, resulting in cascade, point and cluster defects. Matsui et al. demonstrated that 3 MeV Au<sup>2+</sup> ion irradiation to 700 nm thick YBCO films yielded an enhancement in the in-field  $J_c$  at 77 K of up to a factor of 4 [9]. Equally impressive results in YBCO commercial tape have been reported by Jia et al. using 4 MeV proton [10]. Recently, we reported a route to raise both  $T_c$  and  $J_c$  in iron-based superconducting FeSe<sub>0.5</sub>Te<sub>0.5</sub> (FST) thin films by low-energy (190 keV) proton irradiation [16,17]. The 190 keV proton irradiation yields the increase in  $T_c$  due to the nanoscale compressive strain induced by cascade defects. The irradiation also induced a near doubling of  $J_c$  at 4.2 K from the self-field to 35 T through strong vortex pinning by the cascade defects and surrounding nanoscale strain.

In this paper, we report the effect of 1.5 MeV proton irradiation on iron–chalcogenide FST superconducting films. We report the performance of irradiated samples at different temperatures in a magnetic field up to 9 T. We show that 1.5 MeV protons clearly enhance  $J_c$  in magnetic fields <1 T with no subsequent reduction in  $T_c$ . However, we did not observe a reproducible positive effect in the magnetic fields >1 T. The results are discussed in terms of the spatial distribution of defects produced by fast protons.

## 2. Materials and Methods

All films in this study were deposited by the pulsed laser deposition (PLD) method using a Nd:YAG laser ( $\lambda = 266$  nm). We first grew a CeO<sub>2</sub> layer with a thickness of about 80–100 nm on SrTiO<sub>3</sub> single-crystal substrate at a substrate temperature of 600–650 °C and oxygen partial pressure of ~115 mTorr. Then, 100–130 nm thick FST films were grown on CeO<sub>2</sub> buffer layers. During the deposition of FST films, the substrate temperature and oxygen partial pressure were kept at 300–360 °C and  $\sim 1 \times 10^{-6}$  Torr, respectively.

Superconducting transport properties were measured using the conventional four-probe method in a physical property measurement system (PPMS, Quantum Design).  $T_{c,10}$  and  $J_c$  were determined from the  $\rho T$  and  $I$ – $V$  curves using 0.1  $\rho_n$  and 1  $\mu\text{V}/\text{cm}$  criteria, respectively. Here,  $\rho_n$  means the normal state resistivity above the transition temperature. The current was applied perpendicularly to the magnetic field. The magnetization was measured using a superconducting quantum interference device (SQUID, Quantum Design) magnetometer. Two FST films (sample A and B) were fabricated under the same deposition condition for different irradiation conditions. Each FST film was cut into 3 pieces: one for magnetization measurement before and after irradiation with same film, another for transport measurement before irradiation (pristine) and the other for transport measurement after irradiation (irradiated).

The FST films were irradiated with 1.5 MeV proton doses of  $1 \times 10^{15}$  and  $1 \times 10^{16}$  p/cm<sup>2</sup> in vacuum at room temperature using the 5 MV tandem accelerator of the Wakasa Wan Energy Research Center (WERC). The samples were mounted on a copper plate with a double-faced carbon tape. The incident angle of ions was set as normal to the film surface. The flux was kept around  $3.2 \times 10^{12}$  p/cm<sup>2</sup>·s, corresponding to a beam current density of  $\sim 500$  nA/cm<sup>2</sup>. The surface temperature was monitored by a thermocouple. The surface temperature during the irradiation remained below 40 °C.

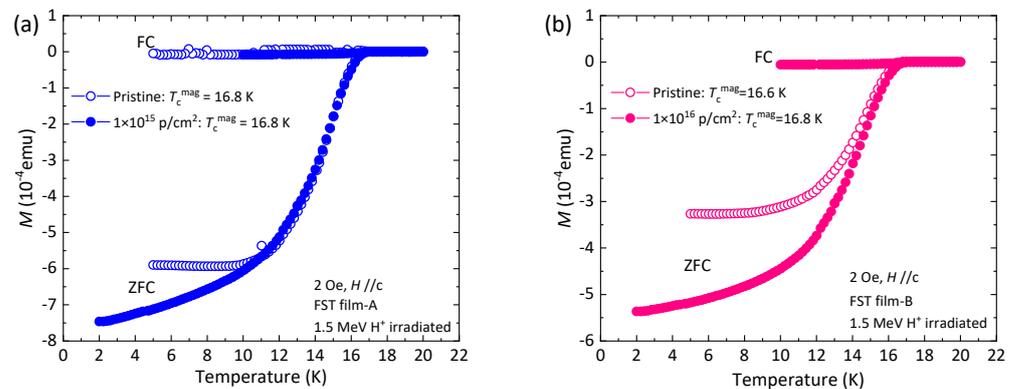
Prior to the ion irradiation experiment, we ran Stopping and Range of Ions in Matter (SRIM) [18] to estimate ion range and damage profile in our experiment. Based on the simulation results,  $1 \times 10^{15}$  and  $1 \times 10^{16}$  p/cm<sup>2</sup> are estimated to be  $\sim 3.2 \times 10^{-5}$  and  $\sim 3.2 \times 10^{-4}$  dpa (displacement per atom), respectively.

## 3. Results and Discussion

### 3.1. Magnetic Measurements

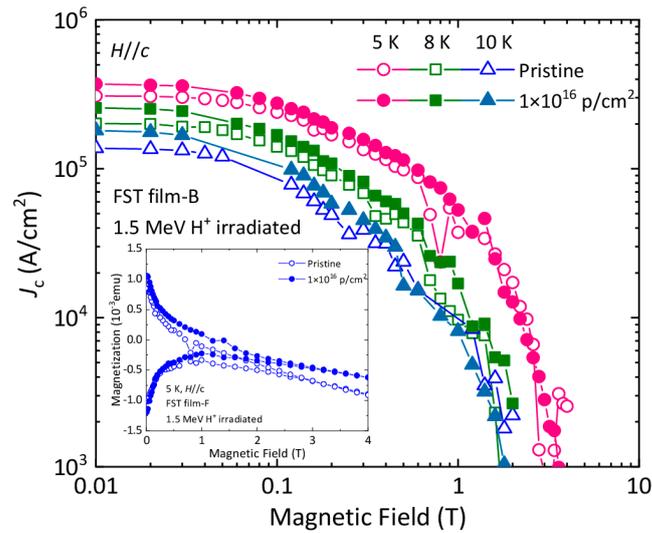
Figure 1a,b compare the temperature dependence of magnetic moment  $M$  with  $H//c$  for two FST films (film-A and film-B) before and after irradiation with  $1 \times 10^{15}$  and  $1 \times 10^{16}$  p/cm<sup>2</sup> dose, respectively. Both the zero-field-cooled (ZFC) and field-cooled

(FC) magnetizations in 2 Oe magnetic field parallel to the  $c$ -axis indicate the appearance of superconductivity (obtained by the bifurcation of ZFC and FC) in pristine FST films at 16.8 K for film-A and 16.6 K for film-B. After the irradiation, the superconducting transitions occurred at 16.8 K for film-A and 16.8 K for film-B, indicating that 1.5 MeV proton irradiations with  $1 \times 10^{15}$  and  $1 \times 10^{16}$  p/cm<sup>2</sup> dose have little impact on  $T_c^{\text{mag}}$ . However, the diamagnetic signal was enhanced with a sharper superconducting transition in the FST film-B irradiated with  $1 \times 10^{16}$  p/cm<sup>2</sup> dose. A degradation of  $T_c$  after the ion irradiation is commonly reported in iron-based superconductors [19], although there have been a few reports on an increased  $T_c$  in iron-based superconductors irradiated with proton and electron [16,20,21]. In previous work, the Fe(Se,Te) films were covered by Al foil with 80  $\mu\text{m}$  thickness and irradiated with 3.5 MeV protons at doses of  $2.68 \times 10^{16}$  and  $5.35 \times 10^{16}$  p/cm<sup>2</sup>, corresponding to  $2.30 \times 10^{-3}$  and  $4.59 \times 10^{-3}$  dpa, respectively [22–24]. The average bombarding energy of the protons on the Fe(Se,Te) film was calculated to be  $1.43 \pm 0.07$  MeV. As a result, the irradiations to doses of  $2.68 \times 10^{16}$  and  $5.35 \times 10^{16}$  p/cm<sup>2</sup> slightly suppressed  $T_c$  from 17.7 K for pristine film to 17.3 K and 17.1 K, respectively. Given these results, the primary reason of the almost same  $T_c$ s before and after the irradiation in our study would be a lower fluence than that in the previous works.



**Figure 1.** Temperature dependences of magnetic moment  $M$  for both zero-field-cooled (ZFC) and field-cooled (FC) process at a magnetic field of  $H = 2$  Oe applied along the  $c$ -axis for FST films before and after 1.5 MeV proton irradiation with (a)  $1 \times 10^{15}$  and (b)  $1 \times 10^{16}$  p/cm<sup>2</sup> dose, respectively.

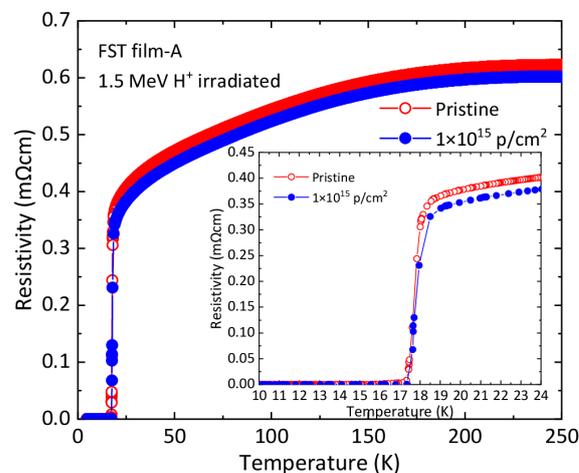
Figure 2 shows the magnetic field dependence of  $J_c$  for the FST film-B at 5, 8, 10 K before and after 1.5 MeV proton irradiation at a dose of  $1 \times 10^{16}$  p/cm<sup>2</sup>. The  $J_c$  was estimated from the magnetization hysteresis ( $M$ – $H$ ) loops using the critical-state Bean model [25,26]. For a rectangular prism-shaped crystal of dimensions  $a < b$ , we obtained the in-plane critical current density  $J_c^{ab}$  in the magnetic field parallel to the  $c$ -axis as  $J_c^{ab} = 20\Delta M / (a(1 - a/3b))$ , where  $\Delta M$  is the difference in magnetization  $M$ (emu/cm<sup>3</sup>) between the top and bottom branches of the  $M$ – $H$  loop. In the inset of Figure 2, the  $M$ – $H$  loop in FST film-B at 5 K before and after the irradiation of a dose of  $1 \times 10^{16}$  p/cm<sup>2</sup> is plotted. A large irreversibility is noticeable up to around 4 T at 5 K. We attained a 30% increase in  $J_c$  in the magnetic field below 1 T, which indicates that the irradiation defects contribute to vortex pinning. In contrast, we observed almost no change in the in-field  $J_c$  above 1 T. Irradiation with MeV protons could produce mostly random point defects and nanocluster [27] due to ion–nucleus collisions. Sylva et al. reported that 3.5 MeV proton irradiation with  $6.40 \times 10^{16}$  p/cm<sup>2</sup> dose (corresponding to  $2.27 \times 10^{-3}$  dpa) yields  $J_c$  improvement of about 40% at 4.2 K and 7 T with respect to the pristine film almost without a decrease in  $T_c$  [22]. On the contrary,  $J_c$  of 3.5 MeV proton irradiated Fe(Se,Te) films covered with 80  $\mu\text{m}$  thick Al foil decreased by up to 80% after irradiation at 4.2 K. The in-field  $J_c$  performance in the irradiated FST films in our study could be attributed to the small number of vortex pinning defects created by the irradiation at low fluence.



**Figure 2.** Magnetic field dependence of critical current density  $J_c^{ab}(H)$  at 5, 8 and 10 K calculated using the critical-state Bean model for FST film-B pre- and post- 1.5 MeV proton irradiation with  $1 \times 10^{16}$  p/cm<sup>2</sup> dose. The inset shows magnetic hysteresis loop under  $H//c$  at 5 K.

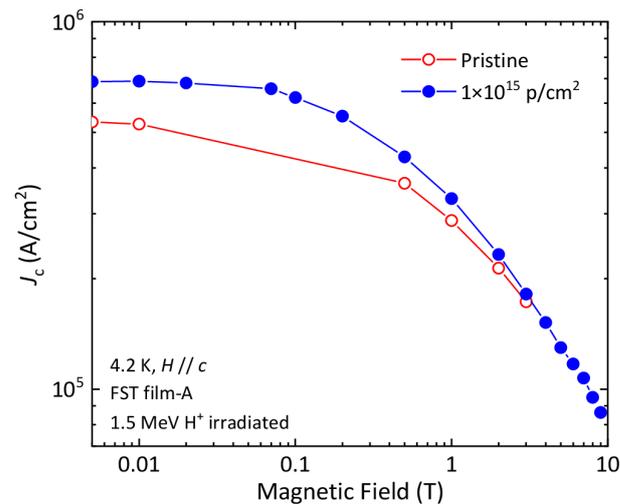
### 3.2. Transport Measurement

In transport measurements, the current is forced to flow through the sample in a particular direction, enabling the direct characterization of superconductivity as a function of temperature, applied magnetic field and field angle. However, we observed an obvious degradation of superconducting properties in the transport measurement of the FST film-B. This could be due to sample degradation, sample handling during mounting and unmounting in a measurement system and possible damage by the laser cutting for patterning the bridge on FST films. In this section, we refer to the FST film-A. Figure 3 presents the temperature dependence of the electrical resistivity before and after irradiation for FST film-A with  $1 \times 10^{15}$  p/cm<sup>2</sup> dose of 1.5 MeV proton. The FST films before and after the irradiation showed metallic behavior below 200 K. Additionally, 1.5 MeV proton irradiation with  $1 \times 10^{15}$  p/cm<sup>2</sup> dose has little effect on normal-state resistivity due to the low dpa. On the contrary, the normal-state resistivity shows nearly upwards parallel-shift upon 6 MeV Au-ion irradiation with a dose of  $1 \times 10^{12}$  Au/cm<sup>2</sup>, corresponding to  $6.42 \times 10^{-3}$  dpa [11]. We observed no change in  $T_{c,10}$  (=17.5 K) before and after the 1.5 MeV protons irradiation with  $1 \times 10^{15}$  p/cm<sup>2</sup> dose. This could be due to the low fluence, i.e., low dpa.



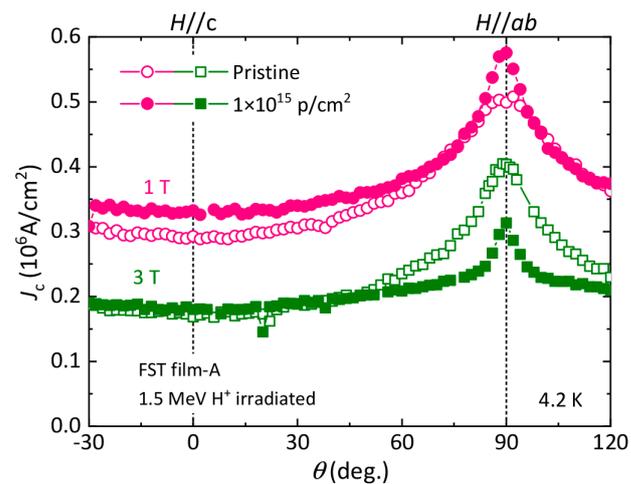
**Figure 3.** Temperature dependences of electrical resistivity at 0 T for the FST film-A before and after 1.5 MeV proton irradiation with  $1 \times 10^{15}$  p/cm<sup>2</sup> dose. Inset shows a magnified temperature region near  $T_c$ .

Figure 4 presents the magnetic field dependence of transport critical current density  $J_c$  with  $H//c$  for the FST film-A before and after irradiation with 1.5 MeV protons to a dose of  $1 \times 10^{15}$  p/cm<sup>2</sup> at 4.2 K. Comparing  $J_c$ s obtained from magnetization and transport measurements, the values of  $J_c$  obtained from transport measurement are larger than those of  $J_c$  calculated from magnetization measurement. This would come from the difference of criterion to determine the  $J_c$  values. The positive effect of the proton irradiation on  $J_c$  at 4.2 K is unambiguous in the magnetic field below 1 T. As the magnetic field increased, the difference between pristine and the irradiated FST film became smaller. Similar behavior was observed in  $J_c(H)$  (calculated from magnetization measurement in Figure 2) for FST film-B irradiated with  $1 \times 10^{16}$  p/cm<sup>2</sup> dose.



**Figure 4.** Magnetic field dependence of critical current density  $J_c$  obtained from transport measurement at 4.2 K for FST film-A pre- and post-1.5 MeV proton irradiation with  $1 \times 10^{15}$  p/cm<sup>2</sup> dose.

A more detailed representation of the pinning efficiency can be obtained from the angular dependence of  $J_c$ . We show  $J_c(\theta)$  for the FST film-A irradiated with  $1 \times 10^{15}$  p/cm<sup>2</sup> dose of 1.5 MeV proton beam under 1 and 3 T at 4.2K in Figure 5. The pristine film has a less-anisotropic  $J_c$  angular dependence at 1 and 3 T without a prominent  $J_c$  peak at  $H//c$ , which is often observed in YBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub> films [28]. A small  $J_c$ -anisotropy,  $\gamma_{Jc}$  ( $J_c^{H//ab}/J_c^{H//c}$ ), of 1.7 is observed at 1 T. This value is smaller than the value of Fe(Se,Te) films grown on Fe-buffered MgO substrates ( $\gamma_{Jc} = 2.6$ ) [29] while it is larger than the value of Fe(Se,Te) films grown on CaF<sub>2</sub> substrates [30,31]. These differences might arise from the difference of the substrate and buffer layer. Upon irradiation with 1.5 MeV proton, the  $J_c$  increases for most of the field orientations, retaining a small  $\gamma_{Jc}$  of 1.7 at 1 T, indicating that the vortex pinning defects would be less anisotropic and randomly distributed. At 3 T, there is a significant decrease in  $J_c$  in the angular range  $\pm 30^\circ$  from  $H//ab$ . Iron-based and cuprate high-temperature superconductors commonly possess inherent layered structures, consisting of alternating conducting and insulating atomic planes. In general, the strong  $J_c$  peak for  $H//ab$  could be ascribed to the vortex pinning by the intrinsic pinning and planar defects such as intergrowths and stacking faults, parallel to the  $ab$  plane [32–35]. In the iron–chalcogenide Fe(Se,Te) compound, which is composed of only the Fe–Se(Te) layer,  $J_c(\theta)$  has a maximum at  $H//ab$  due to intrinsic pinning from the Fe–Se(Te) intralayer and Van der Waals interlayer couplings [29,34,35]. Hence, the  $J_c$  suppression at around  $H//ab$  would occur because of the reduction in the density of intrinsic pinning upon the irradiation.



**Figure 5.** Angular field dependence of the critical current density  $J_c$  obtained from transport measurement for FST film-A before and after 1.5 MeV proton irradiation with  $1 \times 10^{15}$  p/cm<sup>2</sup> dose measured at 4.2 K under 1 and 3 T.

#### 4. Conclusions

We conclude a study on the effect of 1.5 MeV proton irradiation on superconducting properties of FST films. Upon the irradiation up to  $1 \times 10^{16}$  p/cm<sup>2</sup> dose,  $T_c$  remains virtually unchanged in magnetization as well as in transport measurement. An approximately 30% enhancement of  $J_c$  in the magnetic field below 1 T is observed using 1.5 MeV proton irradiation with  $1 \times 10^{16}$  p/cm<sup>2</sup>. Transport properties of a pristine film and an irradiated film with a fluence of  $1 \times 10^{15}$  p/cm<sup>2</sup> show a small anisotropy of  $J_c$  in the applied magnetic field range at 4.2 K. The enhancement of  $J_c$  for almost all the field orientations was accomplished by the irradiation at a dose of  $1 \times 10^{15}$  p/cm<sup>2</sup> at 4.2 K and 1 T. These results indicate that 1.5 MeV proton irradiation is effective in providing less anisotropic pinning defects in the magnetic field below 1 T in iron–chalcogenide superconducting films. Additionally, by fine tuning an irradiation fluence of proton, superconducting properties can be further improved.

**Author Contributions:** Conceptualization, T.O.; sample preparation, T.K. and T.O.; ion irradiation, R.I., T.K. and T.O.; transport measurement, T.K., I.K. and T.O.; magnetization measurement, T.O. and T.K.; data curation, T.K. and T.O.; writing—original draft preparation, T.O.; writing—review and editing, I.K. and R.I. All authors have read and agreed to the published version of the manuscript.

**Funding:** This work was partly supported by Foundation of Kinoshita Memorial Enterprise.

**Acknowledgments:** This research has been performed under the collaboration program between Kwansai Gakuin University, Kyoto University and Wakasa Wan Energy Research Center.

**Conflicts of Interest:** The authors declare no conflict of interest.

#### References

- Putti, M.; Pallechi, I.; Bellingeri, E.; Cimberle, M.R.; Tropeano, M.; Ferdeghini, C.; Palenzona, A.; Tarantini, C.; Yamamoto, A.; Jiang, J.; et al. New Fe-based superconductors: Properties relevant for applications. *Supercond. Sci. Technol.* **2010**, *23*, 034003. [[CrossRef](#)]
- Gurevich, A. Iron-based superconductors at high magnetic fields. *Rep. Prog. Phys.* **2011**, *74*, 124501. [[CrossRef](#)]
- Katase, T.; Ishimaru, Y.; Tsukamoto, A.; Hiramatsu, H.; Kamiya, T.; Tanabe, K.; Hosono, H. Advantageous grain boundaries in iron pnictide superconductors. *Nat. Commun.* **2011**, *2*, 409. [[CrossRef](#)] [[PubMed](#)]
- Si, W.; Zhang, C.; Shi, X.; Ozaki, T.; Jaroszynski, J.; Li, Q. Grain boundary junctions of FeSe<sub>0.5</sub>Te<sub>0.5</sub> thin films on SrTiO<sub>3</sub> bi-crystal substrates. *Appl. Phys. Lett.* **2015**, *106*, 032602. [[CrossRef](#)]
- Iida, K.; Hänisch, J.; Yamamoto, A. Grain boundary characteristics of Fe-based superconductors. *Supercond. Sci. Technol.* **2020**, *33*, 043001. [[CrossRef](#)]

6. Larbalestier, D.; Gurevich, A.; Feldmann, D.M.; Polyanskii, A. High- $T_c$  superconducting materials for electric power applications. *Nature* **2001**, *414*, 368. [[CrossRef](#)] [[PubMed](#)]
7. Foltyn, S.R.; Civale, L.; MacManus-Driscoll, J.L.; Jia, Q.X.; Maiorov, B.; Wang, H.; Maley, M. Materials science challenges for high-temperature superconducting wire. *Nat. Mater.* **2007**, *6*, 631. [[CrossRef](#)] [[PubMed](#)]
8. Kirk, M.A. Structure and flux pinning properties of irradiation defects in  $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ . *Cryogenics* **1993**, *33*, 235. [[CrossRef](#)]
9. Matsui, H.; Ogiso, H.; Yamasaki, H.; Kumagai, T.; Sohma, M.; Yamaguchi, I.; Manabe, T. 4-fold enhancement in the critical current density of  $\text{YBa}_2\text{Cu}_3\text{O}_7$  films by practical ion irradiation. *Appl. Phys. Lett.* **2012**, *101*, 232601. [[CrossRef](#)]
10. Jia, Y.; LeRoux, M.; Miller, D.J.; Wen, J.G.; Kwok, W.K.; Welp, U.; Rupich, M.W.; Li, X.; Sathyamurthy, S.; Fleshler, S.; et al. Doubling the critical current density of high temperature superconducting coated conductors through proton irradiation. *Appl. Phys. Lett.* **2013**, *103*, 122601. [[CrossRef](#)]
11. Ozaki, T.; Wu, L.; Zhang, C.; Si, W.; Jie, Q.; Li, Q. Enhanced critical current in superconducting  $\text{FeSe}_{0.5}\text{Te}_{0.5}$  films at all magnetic field orientations by scalable gold ion irradiation. *Supercond. Sci. Technol.* **2018**, *31*, 024002.
12. Zhang, Y.; Rupich, M.W.; Solovyov, V.; Li, Q.; Goyal, A. Dynamic behavior of reversible oxygen migration in irradiated-annealed high temperature superconducting wires. *Sci. Rep.* **2020**, *10*, 14848. [[CrossRef](#)]
13. Sueyoshi, T.; Kotaki, T.; Furuki, Y.; Fujiyoshi, T.; Semboshi, S.; Ozaki, T.; Sakane, H.; Kudo, M.; Yasuda, K.; Ishikawa, N. Strong flux pinning by columnar defects with directionally dependent morphologies in GdBCO-coated conductors irradiated with 80 MeV Xe ions. *Jpn. J. Appl. Phys.* **2020**, *59*, 023001. [[CrossRef](#)]
14. Civale, L. Vortex pinning and creep in high-temperature superconductors with columnar defects. *Supercond. Sci. Technol.* **1997**, *10*, A11. [[CrossRef](#)]
15. Kirk, M.A.; Yan, Y. Structure and properties of irradiation defects in  $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ . *Micron* **1999**, *30*, 507. [[CrossRef](#)]
16. Ozaki, T.; Wu, L.; Zhang, C.; Jaroszynski, J.; Si, W.; Zhou, J.; Zhu, Y.; Li, Q. A route for a strong increase of critical current in nanostrained iron-based superconductors. *Nat. Commun.* **2016**, *7*, 13036. [[CrossRef](#)] [[PubMed](#)]
17. Ozaki, T.; Wu, L.; Gu, G.; Li, Q. Ion irradiation of iron chalcogenide superconducting films. *Supercond. Sci. Technol.* **2020**, *33*, 094008. [[CrossRef](#)]
18. Ziegler, J.F.; Biersack, J.P.; Littmark, U. *The Stopping and Range of Ions in Solids*; Pergamon: Oxford, UK, 1985.
19. Eisterer, M. Radiation effects on iron-based superconductors. *Supercond. Sci. Technol.* **2018**, *31*, 013001. [[CrossRef](#)]
20. Teknowijoyo, S.; Cho, K.; Tanatar, M.A.; Gonzales, J.; Böhmer, A.E.; Cavani, O.; Mishra, V.; Hirschfeld, P.J.; Bud'ko, S.L.; Canfield, P.C.; et al. Enhancement of superconducting transition temperature by pointlike disorder and anisotropic energy gap in FeSe single crystals. *Phys. Rev. B* **2016**, *94*, 064521. [[CrossRef](#)]
21. Mizukami, Y.; Konczykowski, M.; Matsuura, K.; Watashige, T.; Kasahara, S.; Matsuda, Y.; Shibauchi, T. Impact of Disorder on the Superconducting Phase Diagram in  $\text{BaFe}_2(\text{As}_{1-x}\text{P}_x)_2$ . *J. Phys. Soc. Jpn.* **2017**, *86*, 083706. [[CrossRef](#)]
22. Sylva, G.; Bellingeri, E.; Ferdeghini, C.; Martinelli, A.; Pallecchi, I.; Pellegrino, L.; Putti, M.; Ghigo, G.; Gozzelino, L.; Torsello, D.; et al. Effects of high-energy proton irradiation on the superconducting properties of Fe(Se, Te) thin films. *Supercond. Sci. Technol.* **2018**, *31*, 054001. [[CrossRef](#)]
23. Leo, A.; Sylva, G.; Braccini, V.; Bellingeri, E.; Martinelli, A.; Pallecchi, I.; Ferdeghini, C.; Pellegrino, L.; Putti, M.; Ghigo, G.; et al. Anisotropic Effect of Proton Irradiation on Pinning Properties of Fe(Se, Te) Thin Films. *IEEE Trans. Appl. Supercond.* **2019**, *21*, 6601904. [[CrossRef](#)]
24. Leo, A.; Grimaldi, G.; Nigro, A.; Ghigo, G.; Gozzelino, L.; Torsello, D.; Braccini, V.; Sylva, G.; Ferdeghini, C.; Putti, M. Critical current anisotropy in Fe(Se, Te) films irradiated by 3.5 MeV protons. *J. Phys. Conf. Ser.* **2020**, *1559*, 012042. [[CrossRef](#)]
25. Bean, C.P. Magnetization of Hard Superconductors. *Phys. Rev. Lett.* **1962**, *8*, 250. [[CrossRef](#)]
26. Bean, C.P. Magnetization of High-Field Superconductors. *Rev. Mod. Phys.* **1964**, *36*, 31. [[CrossRef](#)]
27. Haberkorn, N.; Maiorov, B.; Usov, I.O.; Weigand, M.; Hirata, W.; Miyasaka, S.; Tajima, S.; Chikumoto, N.; Tanabe, K.; Civale, L. Influence of random point defects introduced by proton irradiation on critical current density and vortex dynamics of  $\text{Ba}(\text{Fe}_{0.925}\text{Co}_{0.075})_2\text{As}_2$  single crystals. *Phys. Rev. B* **2012**, *82*, 180520.
28. Civale, L.; Maiorov, B.; Serquis, A.; Willis, J.O.; Coulter, J.Y.; Wang, H.; Jia, Q.X.; Arendt, P.N.; MacManus-Driscoll, J.L.; Maley, M.P.; et al. Angular-dependent vortex pinning mechanisms in  $\text{YBa}_2\text{Cu}_3\text{O}_7$  coated conductors and thin films. *Appl. Phys. Lett.* **2004**, *84*, 2121. [[CrossRef](#)]
29. Iida, K.; Hänisch, J.; Schulze, M.; Aswartham, S.; Wurmehl, S.; Büchner, B.; Schultz, L.; Holzapfel, B. Generic Fe buffer layers for Fe-based superconductors: Epitaxial  $\text{FeSe}_{1-x}\text{Te}_x$  thin films. *Appl. Phys. Lett.* **2011**, *99*, 202503. [[CrossRef](#)]
30. Yuan, P.; Xu, Z.; Ma, Y.; Sun, Y.; Tamegai, T. Angular-dependent vortex pinning mechanism and magneto-optical characterizations of  $\text{FeSe}_{0.5}\text{Te}_{0.5}$  thin films grown on  $\text{CaF}_2$  substrates. *Supercond. Sci. Technol.* **2016**, *29*, 035013. [[CrossRef](#)]
31. Braccini, V.; Kawale, S.; Reich, E.; Bellingeri, E.; Pellegrino, L.; Sala, A.; Putti, M.; Higashikawa, K.; Kiss, T.; Holzapfel, B.; et al. Highly effective and isotropic pinning in epitaxial Fe(Se, Te) thin films grown on  $\text{CaF}_2$  substrates. *Appl. Phys. Lett.* **2013**, *103*, 172601. [[CrossRef](#)]
32. Spechta, E.D.; Goyal, A.; Li, J.; Martin, P.M.; Li, X.; Rupich, M.W. Stacking faults in  $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ : Measurement using x-ray diffraction and effects on critical current. *Appl. Phys. Lett.* **2006**, *89*, 162510. [[CrossRef](#)]
33. Civale, L.; Maiorov, B.; MacManus-Driscoll, J.L.; Wang, H.; Holesinger, T.G.; Foltyn, S.R.; Serquis, A.; Arendt, P.N. Identification of Intrinsic ab-Plane Pinning in  $\text{YBa}_2\text{Cu}_3\text{O}_7$  Thin Films and Coated Conductors. *IEEE Trans. Appl. Supercond.* **2005**, *15*, 2808. [[CrossRef](#)]

- 
34. Iida, K.; Hänisch, J.; Reich, E.; Kurth, F.; Hühne, R.; Schultz, L.; Holzapfel, B. Intrinsic pinning and the critical current scaling of clean epitaxial Fe(Se, Te) thin films. *Phys. Rev. B* **2013**, *87*, 104510. [[CrossRef](#)]
  35. Grimaldi, G.; Leo, A.; Nigro, A.; Pace, S.; Braccini, V.; Bellingeri, E.; Ferdeghini, C. Angular dependence of vortex instability in a layered superconductor: The case study of Fe(Se, Te) material. *Sci. Rep.* **2018**, *8*, 4150. [[CrossRef](#)] [[PubMed](#)]