



Article Modifying Electronic and Elastic Properties of 2-Dimensional [110] Diamond by Nitrogen Substitution

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Abstract: One type of two-dimensional diamonds that are derived from [111] direction, so-called diamane, has been previously shown to be stabilized by N-substitution, where the passivation of dangling bonds is no longer needed. In the present work, we theoretically demonstrated that another type of two-dimensional diamonds derived from [110] direction exhibiting a washboard conformation can also be stabilized by N-substitution. Three structural models of washboard-like carbon nitrides with compositions of C_6N_2 , C_5N_3 , and C_4N_4 are studied together with the fully hydrogenated washboard-like diamane (C_8H_4). The result shows that the band gap of this type structure is only open the dangling bonds that are entirely diminished through N-substitution. By increasing the N content, the C_{11} and C_{22} are softer and the C_{33} is stiffer where their bulk modulus are in the same order, which is approximately 550 GPa. When comparing with the hydrogenated phase, the N-substituted phases have higher elastic constants and bulk modulus, suggesting that they are possibly harder than the fully hydrogenated diamane.

Keywords: diamane; carbon nitrides; 2D materials; Ab initio; superhard materials; semiconductor

1. Introduction

Diamond, a carbon allotrope, is a material that has been reported to be the first hardest material and it also possesses the highest thermal conductivity. Its hardness originates from characteristic of sp³ hybridization of C atoms, which has a tetrahedral symmetry with high bonding strength. In fact, the sp² hybridization of C atoms is stronger than sp³ hybridization, for example, graphene. Graphene can persist with a high load of an indentation without breaking bonds showing remarkably high tensile strength [1]; however, its hardness cannot be measured. Other sp³ carbon allotrope, i.e., lonsdaleite, is predicted to be harder than diamond, but its existence is under discussion [2–4].

Since the two-dimensional (2D) materials has come into the spot light, the path for creating a thin diamond-like film has been examined [5,6]. A 2D diamond that is derived from the bulk diamond thinned down in [111] direction or diamane is a promising material for adopting the property of its bulk counterpart, diamond, but its thickness is only 4 C atoms stacking [7–10]. In order to create the sp³ carbon film, the layers of the sp² carbon films, e.g., bilayer graphene and few-layer graphene, must be closer, because its interlayer distance is much larger than a bond length of the sp³ carbon film [11]. The sp² carbon film can be indented by a sharp indenter creating the sp³ hybridization and showing that the indented multi-layer graphene is harder if its number of layers is fewer [10,12,13]. Moreover, diamane can also be synthesized at high pressure, where it is expected to be stable at 5 GPa [6,14].

Even though the diamane can be created, its sp³ hybridization cannot persist without an external auxiliary i.e., indentation, high-pressure [6,10]. The surfaces of diamane that are



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full of dangling bonds need to be passivated with atoms or molecules; otherwise, diamane will transform back to a sp² carbon film [11]. Several works have theoretically studied on the stability of diamane that is passivated by other atoms and molecules and also its band gap and thermal conductivity, depending on passivating molecules [15,16]. The diamane can be synthesized by exposing a bilayer graphene in order to chemically react with H or F atoms [17–20].

Lately, we have shown that the passivation of dangling bonds of the surficial C atoms is not the only method for stabilizing diamane. It can also be stabilized by substituting the N atom, which has one electron more than C atoms on its surfaces [21]. The remaining electrons that form lone electron pairs behave like the passivation that stabilizes the structure and opens a band gap. Moreover, the surface reconstruction that occurs on the outermost layer to have a particular structure can promote the stabilization of the sp³ structure of the 2D carbon [21–23]. Beneficially, the reconstructing surface also enhances a magnetism of the structure, which is typically absent in diamane [21,24].

In this work, 2D diamond-like carbon nitrides $(C_{8-x}N_x)$, where their structures are derived from the structure of diamond thinned down in [110] direction until it has eight atoms in a unit cell (see Figure 1), are studied where x = 2, 3, 4. Because its surfaces are washboard conformations [25], it will be represented as W-diamane for a phase with x = 0 and N-subsituted W-diamanes for phases with x = 2, 3, 4. Their elastic and electronic properties are investigated, due to a number of N atoms in the structure. It will be shown that an accretion of N atoms on the surfaces of W-diamane improves the dynamical stability and enhances the stiffness of the structure, which are even stiffer than hydrogenated 2D diamond.



Figure 1. Structures of (a) C_8H_4 , (b) C_6N_2 , (c) C_5N_3 , and (d) C_4N_4 where the black, red and green balls are C, H, and N atoms, respectively. The θ indicates to the angle measured between surficial and inner atoms. (e) Paths between high symmetry points in the Brillouin zone.

2. Computational Method

The VASP package [26,27], based on the density functional theory, was performed in order to study the properties of the 2D diamond. The projector augmented-wave method [28], which is an efficient method for reducing a cumbersome calculation of the core electrons, was used for the pseudopotential. The exchange-correlation functionals, Perdew-Burke-Ernzerhof (PBE) and Heyd–Scuseria–Ernzerhof (HSE06), were employed [29,30]. The former was used for the structural relaxation in order to find the optimum structural parameters and atomic positions at 0 GPa and 0 K, evaluating the elastic constants, and calculating phonon modes. The latter was used for the electronic density of states (DOS) and band structure calculations. The energy cutoff was set as 600 eV in all calculations. The k-point mesh was $11 \times 5 \times 1$ for the relaxation and calculation of the elastic constants. In order to neglect the interaction between layers of 2D materials, the *c*-axis of every structure was set to be 20 Å at the start of the relaxation and asserted to be larger than 15 Å after the relaxation. The phonon modes were calculated while using Phonopy package [31] with $5 \times 3 \times 1$ supercell and $1 \times 1 \times 1$ k-point mesh. Every DOS was calculated by using the tetrahedron method to integrate over the Brillouin zone [32].

In calculations of the elastic constants, lattice parameters of the studied structures were applied by 1% and 2% compressive and tensile strains, while the atomic positions were relaxed. For the elastic constants corresponding to a direction that is perpendicular to the layer, *z*-positions of the top and bottom atoms were fixed during the atomic relaxation to constrain a *c*-parameter of the strained lattice, which is a distance between top and bottom atoms. Otherwise, the top and bottom atoms would be relaxed to new *z*-positions, yielding unintended strained lattice, which should be fixed to be 1% or 2% strains [21].

3. Results

3.1. 2-Dimensional Structures

The structure of W-diamane is relaxed without any passivations or N substitutions. As expected, W-diamane with entire dangling bonds on its surfaces cannot be stabilized after relaxation, but it can be stabilized if the dangling bonds are passivated through hydrogenation (C_8H_4 , see Figure 1a). We find that, instead of hydrogenation, the substitution of N atoms for surficial C atoms can also result in the structural stabilization of W-diamane. Nevertheless, the substitution is needed on both surfaces of W-diamane in order to keep the washboard-like shape stable on two sides after relaxation. For C_7N , an amount of N atoms is not enough to substitute on both of its surfaces, hence it is not considered. Noting that the studied phases have N atoms arranging in highly order by considering the substitution in the unit cell of W-diamane with eight atoms, while the configurational disorder of C and N atoms in W-diamane is beyond the scope and, thus, not considered in this work.

Table 1 reports the relaxed structures of C_8H_4 , C_6N_2 , C_5N_3 , and C_6N_4 . Howeve, noting that *c*-axes were not fixed during the relaxation, the relaxed *c*-axes are long enough to keep all of structures as 2D materials. The lattice parameters, *a* and *b*, are smaller as the number of N atoms increasing, while their thicknesses from top to bottom layers (d_0) are thicker, as shown in Table 1. Two parameters presenting the characteristics of the structures are bond length and bond angle. Typically, the C-N and N-N bond lengths are shorter than C-C bond length, so average bond lengths of N-substituted W-diamanes are shorter if the number of N atoms is higher. For an illustration, the average C-C bond length of C_8H_4 is 1.558 Å, while the average bond lengths of C_6N_2 , C_5N_3 , and C_4N_4 are 1.508, 1.495, and 1.506 Å, respectively. On the contrary, bond angles of atoms between layers of C_6N_2 , C_5N_3 , and C_4N_4 (see Figure 1) are 101.85°, 110.08°, and 112.37°, respectively, showing an upward trend that corresponds to the number of N atoms.

Figure 2 shows phonon dispersions of hydrogenated and N-substituted W-diamanes, where the negative value of phonon frequency represents an imaginary mode of the phonon. C_8H_4 and C_4N_4 , which are fully passivated and N substituted, respectively, have no imaginary modes, so they are dynamically stable. On the other hand, C_6N_2 and C_5N_3 , which are partially N-substituted, have some amounts of the imaginary phonon, so they are not dynamically stable. Figure 2b–d show the evolution of the imaginary phonon, which is fewer with respect to the number of N substitution in W-diamanes. This is clearly indicating that the stabilization of such a structure can be promoted by suppressing a number of the dangling bonds on the surfaces.

Phase	Space Group	Lattice Parameters	Atomic Species	Wyckoff Sites	Positions
C ₈ H ₄	Pmna	a = 2.509 Å	C _{in}	4 h	(0.000, 0.626, 0.027)
		b = 3.762 Å	C _{sf}	4 h	(0.000, 0.367, 0.079)
		c = 23.282 Å	Н	4 h	(0.000, 0.519, 0.120)
		$d_0 = 5.57 \text{ Å}$			
C_6N_2	$Pmn2_1$	a = 2.428 Å	C _{in1}	2 a	(0.000, 0.616, 0.538)
		b = 3.614 Å	C_{in_2}	2 a	(0.000, 0.359, 0.463)
		c = 17.521 Å	C _{sf}	2 a	(0.000, 0.333, 0.601)
		$d_0 = 3.58 \text{ Å}$	N_{sf}	2 a	(0.000, 0.628, 0.397)
C_5N_3	Pm	a = 2.408 Å	C _{in1}		(0.000, 0.126, 0.154)
		b = 3.529 Å	C_{in_2}		(0.500, 0.377, 0.155)
		c = 18.239 Å	C _{in3}		(0.500, 0.624, 0.226)
		$\alpha = 90.13^{\circ}$	C_{in_4}		(0.000, 0.881, 0.226)
		$d_0 = 3.69 \text{ Å}$	C _{sf}		(0.000, 0.851, 0.092)
			$N_{ m sf_1}$		(0.500, 0.636, 0.090)
			N_{sf_2}		(0.500, 0.374, 0.293)
			N _{sf1}		(0.000, 0.190, 0.292)
C_4N_4	Pmna	a = 2.391 Å	C _{in}	4 h	(0.000, 0.378, 0.031)
		b = 3.468 Å	N _{sf}	4 h	(0.000, 0.625, 0.089)
		c = 21.189 Å			
		$d_0 = 3.76 \text{ Å}$			





Figure 2. Phonon dispersions of (a) C_8H_4 , (b) C_6N_2 , (c) C_5N_3 , and (d) C_4N_4 .

3.2. Electronic Property and Bonding

In the sp³ 2D carbons, the dangling bonds on their surfaces form energy bands across the Fermi level (E_F) closing the band gap, but the band gap is open if the dangling bonds are all passivated [11]. For example, the dangling bonds of NCCN and CNCN phases on the surfaces that are substituted by N atoms are replaced by lone pair electrons that open the band gap wildly. C_8H_4 and C_4N_4 are semiconducting with HSE06 band gap, 3.2 and 3.5 eV, respectively, while C_6N_2 and C_5N_3 are metallic. Thus, the band gap of W-diamane is open if no dangling bond is left to be passivated.

The right side of each figure shown in Figure 3 shows the electronic DOS corresponding to the electronic band dispersion on the left side of each structure. C_8H_4 , where its dangling bonds of C_{sf} atoms are fully passivated by H atoms, has the p_z -orbital of C_{sf} atoms hybridizing with the s-orbital of H atoms at energy state -1 eV below the E_F . The electronic density of states (DOS) being occupied near the E_F are of the p_x -orbitals of C_{sf} atoms in majority and of the p_x -orbitals of C_{in} next below. C_4N_4 , instead fully substituted by N atoms on the surfaces, has the electronic DOS, which is C_{in} - N_{sf} hybridization near the E_F in contrast to C_8H_4 , where its C-H hybridization is at a lower energy level. The p_z -orbital of N_{sf} atoms obviously dominates the valence states near the E_F .

The electronic band structures of C_6N_2 , C_5N_3 , and C_4N_4 have similar dispersing feature across the reciprocal space (see Figure 3b–d). One that makes the electronic property of these three structures different is top two valence bands. C_6N_2 , C_5N_3 , and C_4N_4 have two, one, and none valence bands, respectively, crossing the E_F . Despite that, their valence states are dominated by p_z -orbital of C_{sf} atoms for C_6N_2 and C_5N_3 , and p_z -orbital of N_{sf} atoms for C_4N_4 . This is opposite to the diamane that its partial N-substituted phase, CNCN, has an opening band gap, while the p_z -orbital of C_{sf} also dominates the valence states [21]. Moreover, the lowest conduction state of C_4N_4 is on a Γ -X path, and its highest valence state state is at Γ point. Thus, C_4N_4 has an indirect band gap, while C_8H_8 has a direct band gap.



Figure 3. The electronic band dispersions (**left**) and DOS (**right**) of (**a**) C_8H_4 , (**b**) C_6N_2 , (**c**) C_5N_3 and (**d**) C_4N_4 . The DOS of each atom are presented by blue, yellow, green and red lines for C_{in} , C_{sf} , H and N atoms, respectively. The DOS of each orbital are presented by solid, dashed, dotted, and dash-dotted lines for s, p_y , p_z , and p_x orbitals, respectively.

It is worth noting that N-substituted W-diamanes have no magnetism, while the CNCN phase has a tiny magnetization [21]. The magnetism occurs in 2D carbons when the 2D carbons are structurally distorted or defected, for example, hydrogenated graphene, 2D carbon nitrides that are porous structure, diamond surface with Pandey's reconstruction [24,33–37]. The latter needs HSE06 for calculation in order to obtain magnetism [24], while magnetism can be acquired using PBE for the former two [33,34]. However, the N-substituted W-diamanes investigated while using PBE and HSE06 are non-magnetism.

3.3. Elastic Constants

The hardness is one of the distinguishing properties of diamond and carbon nitrides, where their 2D counterparts are expected to be adopted. However, the hardness of 2D materials, to the best of our knowledge, has not been well-defined. Therefore, the elastic constants that implicitly reflect the hardness are herein considered. Despite the fact that the elastic constants can be calculated by a second derivative of energy with respect to applied strain and devided by a volume of non-strained structure, the 2D materials, such as graphene, are lacking a third dimension. The elastic constants defined for three-dimensional (3D) materials are thus reduced to 2D elastic constants [38,39],

$$C_{ij}^{2\mathrm{D}} = \frac{1}{A_0} \left(\frac{\partial^2 E(\varepsilon_1, \dots, \varepsilon_6)}{\partial \varepsilon_i \partial \varepsilon_j} \right),\tag{1}$$

where A_0 is an unstrained area of $C_{1-x}N_x$, and ε_i is an applied strain up to $\pm 2\%$ in order to fit the C_{ij}^{2D} . The 2D elastic constants of W-diamanes are reported in Table 2, by comparing with other 2D sp³ carbons and carbon nitrides. However, Pakornchote et al. [21] discussed that the 2D elastic constants cannot be used in order to compare across the 2D materials that have different thickness. Therefore, the 2D elastic constants have to be divided by d_0 of the 2D materials in order to make the values become intrinsic [21,40],

$$C_{ij} = \frac{1}{d_0 A_0} \Big(\frac{\partial^2 E(\varepsilon_1, \dots, \varepsilon_6)}{\partial \varepsilon_i \partial \varepsilon_j} \Big), \tag{2}$$

where d_0 is reported in Table 1. The C_{ii}^{2D} and C_{ij} are reported in Tables 2 and 3, respectively.

As the number of N atoms in the unit cell of W-dimane increasing, the C_{11} of Nsubstituted W-diamanes are lower, but their C_{22} and C_{33} are higher. Figure 1 shows that, in the *x* direction, the surficial atoms are bonding with the surficial atoms, which are at the same level, but, in the *y* direction, the surficial atoms are bonding with the inner atoms, which are at the lower level. Thus, the N substitution in the surficial layers can enhance (reduce) the stiffness in the direction that the atoms are bonding in the different (same) level. Although, the Voigt bulk modulus (K_V) of C_6N_2 , C_5N_3 , and C_4N_4 , which are 569, 553, and 533 GPa, respectively, are approximately the same if they are compared with the K_V of NCCN and CNCN phases that are above 600 GPa. The result is in accordance with the simulation in Ref. [10], showing that diamane has the tensile strength higher than other conformations. Noting that the K_V is typically valid for polycrystals and might not be valid for 2D materials, so the values of K_V reported in this work are used for the purpose of comparison.

For C_8H_4 , its C_{ij} , except C_{33} , are smaller than those of N-substituted W-diamanes. Therefore, K_V of C_8H_4 is 310 GPa, which is much smaller that K_V of $C_{1-x}N_x$. One might argue that d_0 of C_8H_4 is much larger than d_0 of $C_{1-x}N_x$, so C_{ij} is unsurprisingly small. However, even if we consider C_{ij}^{2D} , C_8H_4 has C_{11}^{2D} , C_{12}^{2D} , C_{23}^{2D} , and C_{66}^{2D} smaller than $C_{1-x}N_x$. The result is in accordance with the result in the H-diamane, which is a fully hydrogenated diamane that its C_{11}^{2D} and C_{ij} are smaller than that of NCCN and CNCN phases. Thus, it can be concluded that the hydrogenation seem to soften the 2D diamond, while the N substitution makes the 2D diamond stiffer.

Phases	C_{11}^{2D}	C_{12}^{2D}	C_{13}^{2D}	C_{22}^{2D}	C_{23}^{2D}	C_{33}^{2D}	C_{66}^{2D}
C_8H_4	577	41	22	384	8	451	210
C_6N_2	709	64	27	362	148	300	245
C_5N_3	666	73	26	392	120	335	256
C_4N_4	645	71	17	429	87	375	270
NCCN [21]	568	66	51			217	243
CNCN [21]	526	61	38			170	220
<i>π</i> -C ₃ N [21]	595	106	27	510	16	159	244
H-diamane [11]	487	38					

Table 2. The two-dimensional (2D) elastic constants of four phases of W-diamanes are reported in N/m unit by comparing with those of other 2D sp³ carbons and carbon nitrides.

Table 3. The elastic constants and Voigt bulk modulus (K_V) of four phases of W-diamanes are reported in GPa unit by comparing with those of other 2D sp³ carbons and carbon nitrides.

Phases	<i>C</i> ₁₁	<i>C</i> ₁₂	<i>C</i> ₁₃	<i>C</i> ₂₂	C ₂₃	C ₃₃	<i>C</i> ₆₆	K_V
C_8H_4	1036	73	40	688	15	810	378	310
C_6N_2	1963	177	76	1002	410	830	678	569
C_5N_3	1808	199	71	1062	3217	909	694	553
C_4N_4	1715	190	56	1140	226	996	719	533
NCCN [21]	2191	253	198			836	939	718
CNCN [21]	1968	227	142			635	825	618
<i>π</i> -C ₃ N [21]	2030	361	91	1739	199	541	831	624
H-diamane [21]	1026	81						

Moreover, the elastic constants can be calculated from group velocities of phonons for cross-validation [41]. In the orthorhombic system, the sound waves can be expressed as [42]

$$u_x \rho \omega^2 = u_x [q_x^2 C_{11} + q_y^2 C_{66} + q_z^2 C_{55}] + u_y q_x q_y (C_{12} + C_{66}) + u_z q_x q_z (C_{13} + C_{55})$$
(3)

$$u_y \rho \omega^2 = u_y [q_x^2 C_{66} + q_y^2 C_{22} + q_z^2 C_{44}] + u_x q_x q_y (C_{12} + C_{66}) + u_z q_y q_z (C_{23} + C_{44})$$
(4)

$$u_z \rho \omega^2 = u_z [q_x^2 C_{55} + q_y^2 C_{44} + q_z^2 C_{33}] + u_x q_x q_z (C_{13} + C_{55}) + u_y q_y q_z (C_{23} + C_{44}),$$
(5)

where $\mathbf{q} = q_x \hat{i} + q_y \hat{j} + q_z \hat{k}$ is a wave vector, $\mathbf{u} = u_x \hat{i} + u_y \hat{j} + u_z \hat{k}$ is a phonon eigenvector, $\omega(\mathbf{q}) = |\mathbf{q}| v_{[q_x q_y q_z]}$ is a phonon eigenvalue that is limited to acoustic modes, and ρ is a 3D density of W-diamanes. By solving Equations (3)–(5), each \mathbf{q} yields three values of sound velocities, where [q00] and [0q0] are herein considered,

$$v_{[q00]}^{(1)} = \frac{1}{|\mathbf{q}|} \sqrt{\frac{q^2 C_{11}}{\rho}}, \quad v_{[q00]}^{(2)} = \frac{1}{|\mathbf{q}|} \sqrt{\frac{q^2 C_{55}}{\rho}}, \quad v_{[q00]}^{(3)} = \frac{1}{|\mathbf{q}|} \sqrt{\frac{q^2 C_{66}}{\rho}}$$
(6)

$$v_{[0q0]}^{(1)} = \frac{1}{|\mathbf{q}|} \sqrt{\frac{q^2 C_{22}}{\rho}}, \quad v_{[0q0]}^{(2)} = \frac{1}{|\mathbf{q}|} \sqrt{\frac{q^2 C_{44}}{\rho}}, \quad v_{[0q0]}^{(3)} = \frac{1}{|\mathbf{q}|} \sqrt{\frac{q^2 C_{66}}{\rho}}.$$
 (7)

The above two equations lead to a condition that

$$v_{[q00]}^{(3)} = v_{[0q0]}^{(3)}.$$
(8)

Superscripts of the velocities are just numbering, but have not yet been assigned to any phonon modes. The sound velocities can be computed from three acoustic phonons while using Phonopy package that takes the derivative on a dynamical matrix with respect to **q** divided by $2\omega(\mathbf{q})$ at q = 0.05. Noting that only C₄N₄ and C₈H₄ are discussed, since the

dispersion of acoustic phonons of C_6N_2 and C_5N_3 are dropping, causing the instability of the structures (see Figure 2).

In Table 4, the C_{ij} were calculated from the group velocities of three acoustic phonons, ZA, TA, and LA modes while using Equations (6) and (7). If we assigned v^3 to be $v^{(TA)}$, this satisfies an unrestricted condition of Equation (8) that $v_{[q00]}^{TA} \approx v_{[0q0]}^{TA}$. Therefore, from Equations (6) and (7), C_{66} is either 725 or 716 GPa, which is similar to the 719 GPa of C_{66} that was reported in Table 3. Thus, if we assigned $v^{(1)}$ to be v^{LA} , $C_{11} = \rho(v_{[q00]}^{LA})^2 = 1666$ GPa and $C_{22} = \rho(v_{[0q0]}^{LA})^2 = 1097$ GPa. These two values are similar to C_{11} and C_{22} reported in Table 3. By these assignments, C_{ij} calculated using Equations (2), (6) and (7) are in correspondence. Noting that, because of a convex dispersion of ZA mode around Γ -point, v^{ZA} abruptly changes along **q**, yielding inconsistent values of C_{44} and C_{55} , so it needs to be further investigated in the future work.

Table 4. The elastic constants in GPa unit calculated from the group velocities of acoustic phonons at q = 0.05.

Phases	$ ho(v^{ZA}_{[q00]})^2$	$ ho(v_{[q00]}^{TA})^2$	$ ho(v^{LA}_{[q00]})^2$	$ ho(v^{ZA}_{[0q0]})^2$	$ ho(v_{[0q0]}^{TA})^2$	$ ho(v^{LA}_{[0q0]})^2$
C_8H_4	102	370	1003	39	394	692
C_4N_4	147	725	1666	60	716	1097

3.4. Formation Energy

An equation that is used to calculate a formation energy is

$$\Delta E_{form}(\mathbf{C}_x \mathbf{N}_y) = E(\mathbf{C}_x \mathbf{N}_y) - \frac{x E(\mathbf{C}) + y E(\mathbf{N})}{x + y},\tag{9}$$

where $E(C_x N_y)$ is the energy of 2D carbon nitrides, E(C) is an energy of graphene, and E(N) is an energy of N₂ molecule. The formation energies of N-substituted W-diamanes plotted in circle symbols in Figure 4 are 612, 626, and 630 meV for C₆N₂, C₅N₃, and C₄N₄, respectively. They are relatively high by comparing with the formation energy of synthesizable phases, triazine and polyheptazine [43], which are 171 and 126 meV, respectively.

On the one hand, the NCCN phase has the lowest formation energy among 2D diamond-like carbon nitrides. Its formation energy is positive at 0 GPa, but the formation energy is on a convex hull at 10 GPa if only layered phases of carbon nitrides are considered. Hence, it is possible to be synthesized if the precursor is limited to be 2D materials [21]. On the other hand, the N-subsituted W-diamanes have the formation energy as triple the NCCN phase (see Figure 4), which is too high for synthesizing such materials. Nw starting materials other than graphene and N₂ molecule must be examined in order to find the possible pathway to synthesize W-diamane, but this is beyond the scope of the present work.



Figure 4. Formation energy of two-dimensional (2D) carbon nitrides is plotted with respect to the N concentration in the structures, where pure C phase is graphene and pure N phase is N_2 molecule. The formation energies of NCCN, CNCN, π -C₃N, triazine, and polyheptazine phases are from Ref. [21].

4. Conclusions

We have shown that the stiffness and K_V of N-substituted W-diamane increase with the N content. The band gap of W-diamane is only open if the N atoms are entirely substituted on two surfaces. Even though their formation energies is positive when evaluated with respect to graphene and N2 molecule, which are both assumed to be precursors for W-diamane in the present work, the N-containing diamane is likely to possess higher stiffness than that of (hydrogenated) diamane. However, further investigation regarding a possible pathway to fabricate N-containing diamane, such as W-diamane, must be deserved in order to serve as guidelines for furture experimental synthesis of the material.

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Abbreviations

The following abbreviations are used in this manuscript:

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