



# Article Screening of Carbon-Supported Platinum Electrocatalysts Using Frumkin Adsorption Isotherms

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Abstract: An important stage in the development of platinum electrocatalysts on carbon support is the analysis of their basic parameters. Cyclic voltammetry is an effective tool for analyzing the structural and electrochemical properties of such electrocatalysts. Using Frumkin adsorption isotherms, the contribution of the platinum surface to the hydrogen adsorption region was well described by three peaks corresponding to different crystal structures. The screening was carried out for platinum black and platinum electrocatalysts supported by carbon black, reduced graphene oxide (RGO), carbon nanotubes (CNTs), and nanofibers (CNFs). For most samples, the peak contribution to the electrochemical surface area (ESA) and corresponding hydrogen adsorption energies had close values, but the parameters deviated for Pt black and RGO-based samples was observed. The dependence of the calculated peak parameters on the number of accelerated stress test cycles was used to evaluate the effect of the type of carbon support on the stability of the electrocatalyst and the structure of platinum nanoparticles. The experimental results indicate a high degree of stability and differences in the degradation mechanisms of electrocatalysts based on nanostructured carbon compared to carbon black, which are explained by differences in the metal-support interaction and corrosion resistance of nanostructured carbon supports.

**Keywords:** hydrogen adsorption; cyclic voltammetry; adsorption isotherm; platinum electrocatalyst; surface structure; accelerated stress testing; carbon nanostructure; reduced graphene oxide

# 1. Introduction

Today, the fourth energy transition and decarbonization of the energy sector are actively pursued, and electrochemical systems, specifically fuel cells and water electrolyzers with a polymer electrolyte membrane (PEM), play an important role [1]. Their effective integration with energy installations based on renewable energy sources is important [2].

One of the main components of PEM electrochemical systems is a carbon-supported platinum electrocatalyst. Structural features of active nanoparticles and carbon support have a fundamental impact on the performance, durability, and efficiency of the catalyst and the electrochemical system as a whole [3]. Various electrocatalyst carriers can be used, such as carbon black [4], carbon nanotubes (CNTs) [5–9] and nanofibers (CNFs) [10–12], graphene and reduced graphene oxide (RGO) [13–17], as well as less studied carbon structures, such as "nano-onions" [18,19] and spheres [20,21].

The type of carbon support influences the morphology of the deposited Pt nanoparticles. The authors of [22] studied the mobility and morphology of three sizes of platinum nanoparticles (130, 249, and 498 atoms) on various carbon supports. It was shown that for all particle sizes, the diffusion coefficient of Pt nanoparticles was the lowest for carbon nanotube supports. Nanoparticles on carbon nanotubes may be subjected to less sintering



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**Copyright:** © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). and agglomeration than nanoparticles on graphite. The results of this work suggest that it is possible to adapt the catalytic properties of supported metal nanoparticles by selecting and functionalizing the support.

To optimize the synthesis method, catalyst concentration, and support type, electrocatalyst parameter screening methods are necessary. For quantitative analysis of the electrochemical surface area (ESA) and determination of the Pt structure of electrocatalysts, cyclic voltammogram (CV) data are used [23].

To study the hydrogen evolution reaction (HER) and the hydrogen oxidation reaction (HOR) on model metal catalysts at the atomic level, many experiments and theoretical studies were carried out for the elucidation of the processes of adsorption [24–26], diffusion [27–30], and desorption [31–33] of hydrogen on metal surfaces of single crystals. For example, Pt as a model catalyst effectively promotes the release of hydrogen with a high cathodic current density at low overvoltage [34–37]. The first experimental results clearly show that the order of activity of hydrogen release on Pt surfaces is as follows: Pt(110) > Pt(100) > Pt(111), and the kinetic current density on the surface (110) is about three times greater than the current density on the surface (111) [35].

In the case of platinum, the surface properties can be determined using the CV region corresponding to hydrogen adsorption; since the total adsorption charge is directly proportional to the number of atoms on the surface, it can be used to calculate the real surface area [38]. Also, the distribution of charge among different adsorption peaks gives an initial estimate of the presence of different regions of active sites on the surface of the Pt electrocatalyst [39].

Different empirical adsorption isotherms are traditionally used for model descriptions of adsorption phenomena. Processes of adsorption of charged particles are described with high accuracy by the Frumkin isotherm. The Frumkin isotherm is one of the most common in electrochemistry, especially when describing the adsorption of surface-active ions and molecules on the surface of electrodes in an electrolyte solution. In fact, the Frumkin isotherm equation considers the change in the adsorption enthalpy depending on the surface coverage, where the change is caused by lateral interaction of particles in the adsorption layer [40]. The CV curve of the hydrogen underpotential deposition reaction on the platinum surface may be considered a sum of distinct voltammetric peaks of adsorption states [23]. As a result, the Frumkin analysis is a powerful tool for determining the energy and charge associated with hydrogen adsorption on various platinum crystal surfaces [38].

The purpose of the present paper is to determine the effect of the type of carbon supports on the crystal structure of the surface of platinum nanoparticles and their stability during the accelerated stress test (AST) at high potentials. A series of electrocatalysts, such as platinum black and platinum electrocatalysts based on various carbon supports, including carbon black, CNTs, CNFs, and RGO, were synthesized using the polyol method. The obtained samples were subjected to AST, during which CV curves were recorded. To analyze the change in the structure of electrocatalysts, deconvolution of the hydrogen adsorption peaks was carried out using the Frumkin isotherms. The resulting parameters of the adsorption isotherms were used to analyze structural changes in electrocatalysts during the AST cycles.

# 2. Results and Discussion

Figure 1 shows the CV obtained for samples of electrocatalysts.

Pronounced differences are observed on the CV curves for the current density of the double-layer region, which characterizes different types of support. For the RGO-based sample, the current density is maximum in this region, which indicates a more developed surface of this support due to the presence of a large number of defects [41]. For samples based on CNT and CNF, the values are close, reflecting the similarity of their morphology.

The ESA values and the values of the average particle diameter calculated by formulas (11) and (12), respectively, are presented in Table 1.



**Figure 1.** CVs of Pt black, 40 wt.%-Pt/C (Pt<sup>40</sup>/C), 20 wt.%-Pt/C (Pt<sup>20</sup>/C), 20 wt.%-Pt/CNT (Pt<sup>20</sup>/CNT), 20 wt.%-Pt/CNF (Pt<sup>20</sup>/CNF) and 20 wt.%-Pt/RGO (Pt<sup>20</sup>/RGO) recorded in N<sub>2</sub>-saturateted 0.5 M H<sub>2</sub>SO<sub>4</sub> solution at the potential sweep rate of 20 mV/s. The current density of the double-layer region characterizes carbon support.

Table 1. ESA and average particle size (d) of the electrocatalysts.

Since the synthesis of platinum particles was carried out under the same conditions, the differences between the ESA values for electrocatalysts with the same platinum content are caused by specific features of the support. Relatively low ESA values for samples based on complex carbon structures were obtained due to the use of a synthesis procedure optimized for a support in the form of carbon black, which was necessary to exclude the effect of synthesis conditions on the structure of platinum particles.

The change in the shape of the curves with an increase in the number of AST cycles is shown in Figure S1. The stability of the electrocatalysts can be evaluated from the change in the ESA. The ESA values before and after the 3000 AST cycles, as well as the magnitude of the ESA drop, are shown in Figure 2 and in Tables S1–S6. The  $Pt^{40}/C$  sample is more stable in the AST than the  $Pt^{20}/C$  one due to the greater resistance of larger particles to degradation [42]. The minimal change in the ESA corresponds to the Pt black due to the absence of a carbon support in its composition and the large particle size. The increased stability of electrocatalysts based on complex carbon structures is explained by the greater corrosion resistance of the nanostructured support [43,44], as well as better adhesion of particles to the support surface due to the strong metal–support interaction [45].

According to previous studies [46,47], the greatest contribution to the decrease in the ESA during the AST is made by such processes as (1) dissolution of platinum nanoparticles, accompanied by re-deposition of platinum on larger particles at the expense of smaller ones (Ostwald ripening), (2) agglomeration of platinum nanoparticles, (3) corrosion of carbon support material with detachment of platinum particles, and formation of carbon oxidation products that may adsorb on the Pt active sites. When evaluating the change in the ESA, it is impossible to distinguish the individual contributions of these processes to the change in the electrochemical and structural properties of the electrocatalyst. For this purpose, the

hydrogen region of CV can be considered from the standpoint of spectroscopic methods. The use of a sulfuric acid solution instead of perchloric acid as an electrolyte appears more convenient in this case, since it allows registering more pronounced adsorption peaks [23].



Figure 2. ESA values (a) and ESA drop (b) before and after 3000 AST cycles.

Figure 3 shows the  $Pt^{40}/C$  CV with characteristic regions corresponding to electrochemical reactions.



Figure 3. Pt<sup>40</sup>/C CV with the region of hydrogen adsorption fitted by three Frumkin isotherm peaks.

The hydrogen adsorption and desorption curves have pronounced peaks that can be described using the Frumkin isotherm peaks. In the region of hydrogen desorption, the "third peak" is observed in addition to the structure-dependent peaks [48,49]. The nature of this peak is not fully understood; however, according to a previous study [50], this peak can be associated with the replacement of adsorbed hydrogen by adsorbed -OH groups. This peak is absent in the region of hydrogen adsorption; therefore, to simplify the method, we will consider the adsorption region.

The hydrogen adsorption peak includes the contribution of different types of crystal structure on the platinum surface [23,51]. The main types of structure include the surface with Miller indices (100), (110), and (111), as shown in Figure 4.

As a result of the procedure of fitting the peaks to the experimental curves, it was found that the predominant contribution of surface structures to the hydrogen adsorption peak can be described with sufficient accuracy by three peaks of the Frumkin isotherms (A1, A2, and A3). The most intense peak (A1) is in the region of lower potentials; it corresponds to a greater extent to the structure with the low Miller index (110), namely, weak adsorption of hydrogen on step-shaped sites [23,53]. Peak A3 includes a contribution to

the strong adsorption of hydrogen on the steps of (100) structure. Peak A2 corresponds to low-coordinated atoms, such as steps and kinks, as well as defects on the (110) and (111) structures formed after oxygen adsorption [53]. Structure (111), namely, well-oriented terraces, have a broad and uniform low-intensity peak over the entire region of hydrogen adsorption within the considered potential range [37,50]. This peak makes similar contributions to all three adsorption peaks under consideration and, therefore, should not have a significant effect on the relative change in the obtained parameters.



Figure 4. Crystal structure of a platinum nanoparticle [52].

In the potential region below the A1 peak, there is also a region of molecular hydrogen evolution. The beginning of this region on the CV corresponds to the bending of the polarization curve at about 0 V relative to the reversible hydrogen electrode (see Figure 3). For accurate measurement of the ESA, it is important to exclude this region; therefore, the use of the adsorption isotherm makes it possible to determine the beginning of the hydrogen evolution process with a higher degree of accuracy [54].

Figure 5 shows the contributions of the peaks to the ESA as well as the values of the adsorption energies corresponding to the peaks. The parameters of the peaks are close in value, but there is a greater deviation between the parameters of Pt black and Pt/RGO. In the case of the ESA, the Pt black sample has the lowest A1 and A2 peak values but the highest A3 peak value. Since with an increase in the particle size the percentage of sites with the lattice of type 110, as well as edges and corners on their surface, considerably decreases, the contribution of structures of this type to the hydrogen adsorption peak is less for larger particles than for smaller ones [55,56].



**Figure 5.** Normalized ESA (**a**) and adsorption energy (**b**) corresponding to the adsorption peaks A1, A2, and A3.

The RGO-based sample shows a relatively low ESA value compared to other samples on carbon supports. This feature may be due to the influence of the graphene surface on the structure of nanoparticles or to the inaccuracy in the estimation of the total ESA and the particle size calculated from the former, which is caused by the agglomeration of graphene sheets and the blocking of a part of the platinum active sites from electrolyte exposure.

The lower hydrogen adsorption energies of carbon-supported electrocatalysts can be attributed to the significant difference in size and surface structure in comparison with Pt black. Although the platinum–support interaction may also cause the decrease in the adsorption energy and can explain the low values for the RGO-based sample, which demonstrates the strong metal–support interaction due to the presence of surface functional groups and defects [44,45].

During the AST cycles (see Supplementary Material), for most samples, there is an observed shift in the peaks towards the region of lower potentials, on average by 10–20 mV. The potential shift can be explained by a more probable detachment of platinum atoms with higher free energy, which leads to a decrease in the average hydrogen adsorption energy of the final surface.

To study processes occurring during the AST, we considered such parameters as the ESA of a certain peak and the corresponding adsorption energy, the value of which was calculated from the peak position and the interaction constant of the peak using Equation (10). Figure 6 shows variation of the ESA and adsorption energy with an increase in the number of AST cycles for Pt black,  $Pt^{40}/C$ , and  $Pt^{20}/C$  samples.



**Figure 6.** Variation of the ESA and adsorption energy for peaks A1, A2, and A3 normalized to initial values with an increasing number of the AST cycles for the Pt black (**a**,**d**),  $Pt^{40}/C$  (**b**,**e**), and  $Pt^{20}/C$  (**c**,**f**) samples. The trends obtained characterize the stability of the electrocatalyst and changes in the structure of platinum nanoparticles during the AST cycles.

In the case of the Pt black sample, there are no significant changes in the ESA or hydrogen adsorption energy, which can be explained by the absence of a carbon support, which is susceptible to corrosion, and by the large size of the particles, which turn out to be much more resistant to dissolution and agglomeration. For samples based on carbon black, a proportional drop in the ESA of all peaks is observed; however, for the  $Pt^{20}/C$ , the drop for the A1 peak in percentage terms is less pronounced compared to the A2 and A3 peaks. Despite the fact that for the  $Pt^{40}/C$ , the drop in the A1 peak should be less

pronounced than for the  $Pt^{20}/C$  due to size effects, the second mechanism associated with CO adsorption on platinum active sites may be involved here. Carbon species, like CO, can be formed as a result of corrosion of the carbon support during cycling in the region of high potentials [57] and can be irreversibly adsorbed on structures of the 110 type to a lesser extent than on the surface structures of other types [58]. Since the carbon support to platinum ratio is higher for the  $Pt^{20}/C$ , the effect of CO adsorption is more pronounced.

The variations in the normalized adsorption energy for the  $Pt^{20}/C$  and  $Pt^{40}/C$  samples are similar, but more pronounced changes are observed for the  $Pt^{20}/C$  sample because of the size effect. The largest change corresponds to the A1 peak, due to the presence of a larger number of atoms with a low coordination number. For the A2 peak, on the contrary, an increase in the adsorption energy is observed in the initial region, which can be explained by the formation of defects as a result of surface reconstruction during oxygen adsorption; however, with an increase in the number of AST cycles, the effect of particle agglomeration and growth predominates, and as a result, the particles are less prone to defect formation.

The ESA and adsorption energy curves corresponding to adsorption peaks for the samples with carbon nanostructures are shown in Figure 7.



**Figure 7.** Variation of the ESA and adsorption energy for peaks A1, A2, and A3 normalized to initial values with an increasing number of the AST cycles for the  $Pt^{20}/RGO(a,d)$ ,  $Pt^{20}/CNF(b,e)$ , and  $Pt^{20}/CNT(c,f)$  samples. The trends obtained characterize the stability of the electrocatalysts based on complex carbon nanostructures and changes in the structure of platinum nanoparticles during the AST cycles.

The ESA value of the A3 peak for the Pt<sup>20</sup>/RGO sample decreases with an increase in the number of AST cycles much faster than for the other peaks. This effect may be associated with the specifics of the particle growth and dissolution on the surface of RGO sheets and requires further study. In addition, for peaks A1 and A2, a significant change in adsorption energy is observed as the number of AST cycles increases. This effect can be explained by the formation of a large number of platinum sites with a higher adsorption energy on the

corresponding surface types during the synthesis of the RGO-based electrocatalyst. During AST cycles, a more intensive reconstruction of these sites occurs.

The  $Pt^{20}/CNF$  and  $Pt^{20}/CNT$  samples show high stability of the ESA peaks during the AST and a rate and amount of change in these parameters similar to the  $Pt^{40}/C$  sample, despite the lower content of platinum. This effect may be explained by the high stability and corrosion resistance of complex carbon supports. However, in the case of the  $Pt^{20}/CNF$ , there is a loss in the ESA of the A2 peak comparable to that of the  $Pt^{40}/C$ , suggesting a weak influence of the support type on the stability of the corresponding surface structure of Pt active sites.

The curves of the adsorption energy versus number of AST cycles for the A1 and A3 peaks are similar in shape for all samples with a nanostructured support. Peak A2 shows a similar trend for the  $Pt^{20}/RGO$  and  $Pt^{20}/CNF$  samples, with the value of the adsorption energy decreasing over the entire range of the AST cycles. The difference between results for the nanostructured electrocatalysts and samples based on carbon black may indicate a smaller contribution of the metal–support interaction.

For the  $Pt^{20}/CNT$  sample, there is a reverse trend towards a change in the adsorption energy of the A2 peak, which is probably due to the influence of the support structure on the process of particle growth during the particle association and Ostwald ripening and its surface reconstruction, which cause the formation of relatively small particles with a large number of surface defects.

### 3. Materials and Methods

#### 3.1. Synthesis of Electrocatalysts

In this work, electrocatalysts with the compositions Pt<sup>40</sup>/C, Pt<sup>20</sup>/C, Pt<sup>20</sup>/CNT, Pt<sup>20</sup>/CNF, and Pt<sup>20</sup>/RGO (where 20 and 40 are wt % of platinum, C—carbon black Vulcan XC-72, CNT—lab-synthesized carbon nanotubes, CNF—commercial carbon nanofibers, and RGO—commercial reduced graphene oxide) and Pt black were synthesized using the polyol method in ethylene glycol as described in our previous work [59].

#### 3.2. Electrochemical Studies

Catalytic inks were prepared by mixing 10 mg of electrocatalyst with 1 mL of a 1:1 isopropanol/water solution, followed by ultrasonication for half an hour at room temperature. Then, the resulting catalytic ink aliquot with a volume of 25  $\mu$ L was pipetted onto a polished and cleaned titanium foil electrode with a geometric area of 0.5 cm<sup>2</sup>, forming a thin catalyst layer. After that, the titanium electrode was dried until the solvent was completely removed. The final catalyst loading was about 0.5 mg cm<sup>-2</sup>. The platinum loading for Pt<sup>20</sup>/C, Pt<sup>20</sup>/CNT, Pt<sup>20</sup>/CNF and Pt<sup>20</sup>/RGO samples was 0.1 mg<sub>Pt</sub> cm<sup>-2</sup>. According to the catalyst loading, the thickness of the catalyst layer was estimated to be about 0.8  $\mu$ m [60,61].

The cyclic voltammograms (CVs) were measured in N<sub>2</sub>-saturated 0.5 M H<sub>2</sub>SO<sub>4</sub> at 25 °C using a conventional three-electrode glass cell, an Ag/AgCl reference electrode, and a Pt wire counter electrode. The measurements were performed using a CorrTest CS350 electrochemical workstation (CorrTest Instruments Corp., Ltd., Wuhan, China). Before CV registration, the electrode was activated in the potential range of 0.05 to 1.20 V vs. RHE at the 50 mV/s potential sweep rate for about 30 cycles until a stable CV was obtained. The CVs of all samples were recorded at the same sweep rate of 20 mV/s in the same potential range and then normalized according to the total mass of Pt in the catalyst layer.

The electrocatalysts were subjected to AST [62,63] by cycling the potential in the range of 0.8-1.4 V vs. the RHE electrode with a sweep rate of 100 mV/s for 3000 cycles. The working 0.5 M H<sub>2</sub>SO<sub>4</sub> solution was pre-saturated with oxygen before AST cycles. The CVs were measured initially and every 300 cycles. Before the CVs measurement, the working solution was saturated with N<sub>2</sub>. The degree of degradation of the catalyst was estimated from the change in the ESA calculated from the CVs.

#### 4. Theory

The reaction of electrochemical adsorption of hydrogen on the surface of platinum is described by the following equation

$$Pt + H_3O^+ + e^- \rightleftharpoons Pt - H + H_2O \tag{1}$$

At equilibrium, the adsorption process in the simplest case is described by the Langmuir isotherm, according to which the electrochemical potential of the reaction can be written as

$$\mu_{ads} = \mu_{ads}^0 + RTln\left(\frac{\theta}{1-\theta}\right) \tag{2}$$

where  $\theta$  is the fraction of occupied active sites on the platinum surface;  $\mu_{ads}^0$  is the equilibrium electrochemical potential corresponding to the case of a half-filled platinum surface ( $\theta = 0.5$ ); *R* is the universal gas constant; and *T* is temperature.

To take into account the interaction between ions on the platinum surface, Frumkin introduced the interaction constant *g*. The expression for the electrochemical potential then takes the following form

$$\mu_{ads} = \mu_{ads}^0 + RT \left( ln \left( \frac{\theta}{1 - \theta} \right) + g\theta \right)$$
(3)

The constant *g* takes on a positive value when the adsorbed particles repel each other and a negative value in the case of attraction. For the Frumkin isotherm, the value of the equilibrium electrochemical potential no longer corresponds to the case of  $\theta_r = 0.5$ . The equilibrium value of  $\theta_r$  can be found from the condition of equality to zero of the last summand in Equation (3).

For a three-electrode cell with a reversible hydrogen electrode as the reference electrode, the Frumkin isotherm for reaction (1) occurring at the working electrode will have the following form [64]

$$\frac{\theta}{1-\theta}\exp(g\theta) = f_{H_2}^{1/2}\exp(-EF/RT)\exp\left(-\Delta G_{ads}^0/RT\right)$$
(4)

where  $f_{H_2}$  is the fugacity of  $H_2$ , E is the potential at the working electrode vs. RHE, F is the Faraday constant, and  $\Delta G_{ads}^0$  is the standard Gibbs free energy corresponding to equilibrium value  $\theta_r$ .

The double-layer corrected current density of the CV can be expressed as follows

$$j = \sigma_H \nu \frac{d\theta}{dE} \tag{5}$$

where  $\sigma_H$  is the specific charge required for the formation of a monolayer of adsorbed hydrogen on the active sites of the platinum surface, and  $\nu$  is the linear rate of potential sweep *E* [38].

Using expressions (4) and (5), the dependence of the current density on  $\theta$  can be deduced as

$$j = \sigma_H \nu \frac{F}{RT} \frac{(1-\theta)\theta}{1+g(1-\theta)\theta}$$
(6)

The dependence of the potential on  $\theta$  has the following form

$$E = -\frac{RT}{F} \left( ln \left( \frac{\theta}{1 - \theta} \right) + g\theta + \frac{\Delta G_{ads}^0}{RT} \right)$$
(7)

The resulting expressions can be represented as

$$j = j_m (4+g) \frac{1}{((1-\theta)\theta)^{-1} + g}$$
(8)

$$E = E_m + \frac{RT}{F} \left( ln \left( \frac{1-\theta}{\theta} \right) + g \left( \frac{1}{2} - \theta \right) \right)$$
(9)

The expressions obtained correspond to the symmetric peak of the polarization curve, where  $j_m$  and  $E_m$  are the coordinates of the peak maximum, which is reached at  $\theta = 0.5$  [65].

According to expressions (7) and (9), the adsorption energy can be expressed from the potential of the peak maximum as

$$-\frac{\Delta G_{ads}^0}{RT} = \frac{F}{RT} E_m + \frac{g}{2} \tag{10}$$

The peak parameters were fitted to the experimental values using the curve\_fit function of the SciPy software package of the Python programming language (version 3.10, Python Software Foundation, Wilmington, DE, USA, 2021), which is an implementation of the nonlinear least squares method. The boundary and initial values, as well as the number of peaks, were determined sequentially starting from the most intense peak, with subsequent subtraction of the approximating peak and search for parameters for the next highest peak [65]. According to the report [23], the attraction constant was assumed to be negative, and the full width at half maximum (FWHM) was taken to be less than 90 mV.

The ESA value of the electrocatalyst samples was calculated from the obtained hydrogen adsorption peaks according to the equation

$$ESA = \frac{\sum_{i} Q_{H_i}}{Q_0 M_{Pt}}$$
(11)

where  $Q_{H_i}$  is the hydrogen adsorption charge corresponding to the *i*-th peak,  $Q_0 = 2.1 \text{ C/m}^2$  is the hydrogen monolayer adsorption charge per unit surface area of polycrystalline platinum, and  $M_{Pt}$  is the mass of platinum in the electrocatalyst sample deposited on the electrode. Based on the ESA value, the size of spherical particles can be estimated from the expression

$$d = \frac{6000}{\rho_{Pt} ESA} \tag{12}$$

where  $\rho_{Pt}$  is the density of platinum ( $\rho_{Pt} = 21.45 \text{ g/cm}^3$ ).

#### 5. Conclusions

In the present work, we propose a method for screening electrocatalysts based on processing the hydrogen adsorption region of the CV by peak deconvolution using Frumkin isotherms. The predominant contribution of the platinum surface to the hydrogen adsorption region was well described by three peaks of the Frumkin isotherm corresponding to different crystal structures. For most samples, the peak contribution ratio to the ESA was about 3:1:1, and the hydrogen adsorption energy values were 8, 16, and 21 kJ mol<sup>-1</sup>, respectively. The grater deviation of these parameters was observed for the Pt black, which demonstrates the largest particles compared to other samples, and for the Pt<sup>20</sup>/RGO due to the presence of functional groups and defects on the RGO surface. The proposed method also allows for the evaluation of the primary degradation mechanisms in the AST process. The ESA drop for the  $Pt^{20}/C$  and  $Pt^{40}/C$  samples is more related to particle growth and CO adsorption; the hydrogen adsorption energy trend is determined by the surface reconstruction during oxygen adsorption. The electrocatalysts based on nanostructured carbon showed different behavior and 15–20% higher stability, which is explained by differences in metal-support interaction and the corrosion-resistant properties of these types of supports. However, for a more accurate determination of the predominant degradation mechanisms

of electrocatalysts, additional methods of physical, electrochemical, and model analysis are needed.

Supplementary Materials: The following supporting information can be downloaded at: https:// www.mdpi.com/article/10.3390/inorganics11030103/s1, Figure S1: Cyclic voltammograms (CVs) of Pt black (a),  $Pt^{40}/C$  (b),  $Pt^{20}/C$  (c),  $Pt^{20}/RGO$  (d),  $Pt^{20}/CNF$  (e) and  $Pt^{20}/CNT$  (f), recorded after each 300 cycles of the AST in N2 saturated 0.5 M H2SO4; Figure S2: CVs hydrogen adsorption region and deconvoluted peaks of Pt black before (a) and after 3000 AST cycles (b); Table S1: Computed values of adsorption isotherm parameters of Pt black: Em-the peak position; FWHM-the peak full width at half maximum; g—the interaction constant;  $\Delta G_{ads}^{0}$  —the hydrogen adsorption energy, corresponding to the *i*-th peak;  $ESA_i$ —the contribution of the *i*-th peak to the ESA,  $P_i$ —the normalized contribution of the *i*-th peak to the ESA; Figure S3: Hydrogen adsorption peaks of Pt black before and after 3000 AST cycles. The current density is normalized to the corresponding total ESA; Figure S4: CVs hydrogen adsorption region and deconvoluted peaks of Pt<sup>40</sup>/C before (a) and after 3000 AST cycles (b); Table S2: Computed values of adsorption isotherm parameters of Pt<sup>40</sup>/C; Figure S5: Hydrogen adsorption peaks of Pt<sup>40</sup>/C before and after 3000 AST cycles. The current density is normalized to the corresponding total ESA; Figure S6: CVs hydrogen adsorption region and deconvoluted peaks of Pt<sup>20</sup>/C before (a) and after 3000 AST cycles (b); Table S3. Computed values of adsorption isotherm parameters of Pt<sup>20</sup>/C; Figure S7: Hydrogen adsorption peaks of Pt<sup>20</sup>/C before and after 3000 AST cycles. The current density is normalized to the corresponding total ESA; Figure S8: CVs hydrogen adsorption region and deconvoluted peaks of Pt<sup>20</sup>/RGO before (a) and after 3000 AST cycles (b); Table S4: Computed values of adsorption isotherm parameters of Pt<sup>20</sup>/RGO; Figure S9: Hydrogen adsorption peaks of Pt<sup>20</sup>/RGO before and after 3000 AST cycles. The current density is normalized to the corresponding total ESA; Figure S10: CVs hydrogen adsorption region and deconvoluted peaks of Pt<sup>20</sup>/CNF before (a) and after 3000 AST cycles (b); Table S5: Computed values of adsorption isotherm parameters of Pt<sup>20</sup>/CNF; Figure S11: Hydrogen adsorption peaks of Pt<sup>20</sup>/CNF before and after 3000 AST cycles. The current density is normalized to the corresponding total ESA; Figure S12. CVs hydrogen adsorption region and deconvoluted peaks of Pt<sup>20</sup>/CNT before (a) and after 3000 AST cycles (b); Table S6. Computed values of adsorption isotherm parameters of Pt<sup>20</sup>/CNT; Figure S13. Hydrogen adsorption peaks of Pt<sup>20</sup>/CNT before and after 3000 AST cycles. The current density is normalized to the corresponding total ESA.

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#### Abbreviations

Abbreviation	Meaning
PEM	Polymer electrolyte membrane
CNTs	Carbon nanotubes
CNFs	Carbon nanofibers
RGO	Reduced graphene oxide
ESA	Electrochemical surface area
CV	Cyclic voltammogram

HER	Hydrogen evolution reaction
HOR	Hydrogen oxidation reaction
AST	Accelerated stress testing
RHE	Reversible hydrogen electrode
FWHM	Full width at half maximum

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