



Article

The Effect of Temperature and Local pH on Calcareous Deposit Formation in Damaged Thermal Spray Aluminum (TSA) Coatings and Its Implication on Corrosion Mitigation of Offshore Steel Structures

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Academic Editors: Niteen Jadhav and Andrew J. Vreugdenhil

Received: 2 December 2016; Accepted: 5 April 2017; Published: 11 April 2017

Abstract: This paper is based on experimental data and provides better understanding of the mechanism of calcareous deposit formation on cathodically polarized steel surfaces exposed to synthetic seawater at 30 °C and 60 °C. The study comprises measurement of the interfacial pH of thermally sprayed aluminum (TSA) coated steel samples with and without a holiday (exposing 20% of the surface area). Tests were conducted at the corrosion potential for up to 350 h. It was experimentally determined that the local pH adjacent to the steel surface in the holiday region reached a maximum of 10.19 and 9.54 at 30 °C and 60 °C, respectively, before stabilizing at about 8.8 and 7.9 at the two temperatures. The interfacial pH on the TSA coating at 30 °C was initially 7.74 dropping to 4.76 in 220 h, while at 60 °C it increased from pH 6.41 to the range pH 7.0–8.5. The interfacial pH governed the deposition of brucite and aragonite from seawater on the steel surface cathodically polarized by the TSA. This mechanism is likely to affect the performance of TSA-coated offshore steel structures, especially when damaged in service.

Keywords: pH; thermally sprayed aluminum (TSA); offshore

1. Introduction

Carbon steel is used in offshore structures where corrosion, mainly caused by the constituents of seawater, is often exacerbated by fluctuating temperature and oxygen availability [1,2]. Oxygen availability is considered a key parameter influencing the corrosion of steel. Its supply to the steel surface is determined by the oxygen concentration in seawater, movement of the seawater, oxygen diffusion coefficient in seawater, and corrosion product formation [3,4]. Advances in offshore and drilling engineering allow extraction of oil and gas from deeper waters where higher temperatures along the pipelines and conduits are expected [5].

It is generally recognized that calcareous deposits will form on cathodically polarized steel surfaces when exposed to seawater as long as conditions allowing the nucleation and growth of these deposits are met. The calcareous deposits are beneficial to the structure since they act as a barrier to diffusion of dissolved oxygen [5]. Seawater is mildly alkaline (pH 7.8-8.3) and it is reported that a minimum pH of 9.5 and 7.5 is required at 25 °C for the deposition of Mg(OH)₂ and CaCO₃ layers, respectively [5–9]. A higher pH is required for the deposition of Mg(OH)₂ since seawater is usually unsaturated with Mg(OH)₂, while surface seawater is supersaturated with CaCO₃ [10].

The pH values, however, are mainly specified through theoretical approaches, with few experimental data available to support it so far, especially at higher temperatures.

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Dexter and Lin [11] estimated the local pH on steel subjected to cathodic polarization by a galvanostatic method, considering the buffering capacity and ionic strength of seawater at 25 °C, and compared the model values with their experimental values using a microelectrode technique by applying current densities of 0, 20, and 100 μ A/cm². From the modelling, they obtained a maximum interfacial pH of 9.9, while the experimental value reached 10.2 at the highest current density. They highlighted that the difference is related to the distance of the micro-pH electrode from the steel surface, estimated to be 50 and 100 μ m while the modelling gave values right at the steel surface. Also, the modelling did not consider the hydrogen evolution, only the oxygen reduction as cathodic reaction. Salgavo et al. [8] analyzed alloy 600 (UNS N06600) sample cathodically polarized to -0.8 V to -1.2 V (Ag/AgCl) in natural seawater at temperatures between 13 °C and 27 °C. The probe was 1 mm from the metal surface and the maximum pH value recorded was 10.7 at -1.2 V (Ag/AgCl). Lewandowski et al. [12] found pH values between 9.5 and 10 at -1.0 V (SCE) on 304 stainless steel (AISI 304SS) surface in synthetic seawater at room temperature. The distance between the pH microelectrode and the stainless steel surface was not specified.

It is evident from the above discussion that the local pH depends on cathodic polarization which ultimately governs the formation of calcareous deposits in seawater. However, experimental studies related to local pH at temperatures above 30 °C were not found in the literature. Moreover, polarization from a sacrificial anode such as TSA, at the temperatures applied in this study has not been covered in publications so far. This study aims to address this knowledge gap and, for such purpose, experiments were designed to acquire local pH data on TSA coatings and steel surfaces cathodically polarized by a TSA coating, while both considering exposure to seawater at 30 °C and 60 °C. The implication of deposit formation on the corrosion performance of TSA-coated steel in seawater is also discussed.

2. Materials and Methods

2.1. Specimen Preparation

Four carbon steel coupons (conforming to BS EN 10027-1 S355J2G4) were sprayed with commercially pure aluminum using a twin-wire arc spray system (TWAS) with a 528 gun (Metallisation Ltd., Dudley, UK). Table 1 shows both the steel and Al wire composition, while Table 2 gives the parameters used for the coating production. Holidays (Ø 20 mm, 0.8 mm depth, exposing 20% of the specimen surface area) were drilled on two coated specimens to expose the underlying steel (Figure 1). The dimension of the specimens was $40 \times 40 \times 6 \text{ mm}^3$ and their back and edges were covered with a polymeric resin.

Table 1. Composition of the substrate steel (S355) and coating consumable (wt %).

Component	С	Mn	Si	S	P	Fe	N	V	Cu	Al
EN10025S355J2G3	0.12	1.39	0.39	0.019	0.014	Bal.	0.003	0.065	_	_
Al wire (coating)	_	< 0.01	0.07	_	_	0.21	0.01	_	< 0.01	Bal.

Table 2. The spray parameters used for coating production.

Wire Diameter (mm)	Wire Feed	Spray	Increment	Traverse	Nominal
	Rate (g/min)	Distance (mm)	Step (mm)	Speed (m/s)	Thickness (µm)
2.3	98.7	95	15	0.5	300

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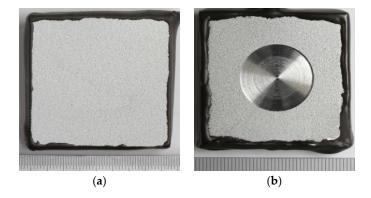


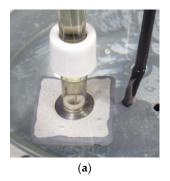
Figure 1. Thermally sprayed aluminum (TSA)-coated carbon steel specimens used for the local pH test: (a) without defect; and (b) with 20% holiday exposing the steel substrate.

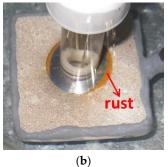
2.2. Exposure Tests and pH Measurements

The specimens were exposed to synthetic seawater (composition given in Table 3) at two temperatures: $30 \,^{\circ}\text{C}$ and $60 \,^{\circ}\text{C}$. A flat tip pH electrode was positioned, touching the sample's surface and in the case of the coupon with the holiday, the pH electrode was centered in the holiday region touching the steel surface (Figures 2a and 3a at $30 \,^{\circ}\text{C}$ and $60 \,^{\circ}\text{C}$, respectively).

Table 3. Synthetic seawater composition [13].

Compound	Concentration (g/L)	Compound	Concentration (g/L)
NaCl	24.53	NaHCO ₃	0.201
$MgCl_2$	5.20	KBr	0.101
Na ₂ SO	4.09	H_3BO_3	0.025
$CaCl_2$	1.16	SrCl ₂	0.025
KCl	0.695	NaF	0.003





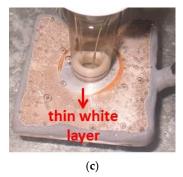


Figure 2. Photographs showing the specimen (with holiday) exposed to 30 $^{\circ}$ C seawater. (a) At the beginning of the test showing the position of the probe on the sample; (b) after 18 h showing formation of rust on the edges of the holiday at pH 10.08; and (c) after 82 h showing a thin white layer in the holiday at pH 9.84.

A pH-meter was used for data acquisition and a temperature compensator was also employed. The temperature was set by adjusting the water bath heater settings. The evaporative loss was replenished by adding deionized water. At 30 $^{\circ}$ C, this was done once every two days with a few milliliters of water since the evaporation rate was low. At 60 $^{\circ}$ C, the evaporation loss was much greater and required replenishment twice a day, i.e., in the morning and in the evening. The deionized water was heated to the approximate temperature of the bulk seawater prior to addition, however, a small fluctuation in pH was observed during the addition.

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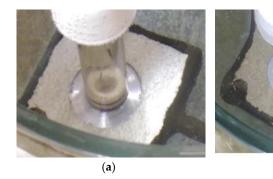


Figure 3. Photographs showing the specimen (with holiday) exposed to $60 \,^{\circ}$ C seawater. (a) At the beginning of the test showing the position of the probe on the sample; (b) after 17 h showing the holiday covered by a visible deposit of calcareous matter, pH 9.2.

The tests were monitored until reasonably stable pH values were reached, which in most cases took up to 350 h. The pH of the bulk seawater (away from the sample surface) was recorded daily. Prior to each experiment, the pH electrode was calibrated in buffer solutions of pH 7 and 10. Each test was carried out in separate reactors to avoid contamination. For comparison, a reactor filled with synthetic seawater only (without any specimen) had its pH measured twice a day in order to record the pH change without the influence of any sample.

2.3. Microstructural Characterization and Phase Identificiation

After testing, the specimens were dried and photographed. The samples with holidays had their top surfaces and cross sections analyzed by ZEISS 1455EP scanning electron microscope (SEM, Carl Zeiss AG, Oberkochen, Germany) and energy dispersive X-ray (EDX, Carl Zeiss AG, Oberkochen, Germany) in order to identify the chemical elements present. Prior to analysis, the samples were sputter coated with Au to provide a conductive layer. X-ray diffraction (XRD) was also carried out on the powder collected from the holiday region and the TSA coating surface. A Bruker D8-Advanced (Bruker Corporation, Billerica, MA, USA) was used to identify the crystalline phases. Bragg-Brentano geometry was employed to measure the diffraction intensity from 5° to 80° (20°) with 0.01° step size. CuK α radiation (λ = 1.541 Å) was used and the X-ray unit was operated with an accelerating voltage of 40 kV. The software Diffrac.Suite.Eva (Version 4.1.1, Bruker AXS GmbH, Karlsruhe, Germany) was used for analyzing the peaks and background correction was applied.

The EDX analyses were performed in different areas of the samples in top view and cross section. In order to simplify, different labels were used to identify the different areas with the same composition.

3. Results

3.1. Visual Inspection

Figure 4 shows the images of the samples after exposure to synthetic seawater for up to 350 h. The change in the appearance of specimen surface due to the deposition of corrosion product or calcareous matter was expected. However, some differences between the specimens were also observed. The circles formed on surfaces were caused by the contact of the pH probe tip on the specimen. Although the pH probe may have perturbed the growth of calcareous deposit in that region just below the probe, it is not expected to modify the characteristics in the whole steel surface away from the probe (in the exposed holiday region).

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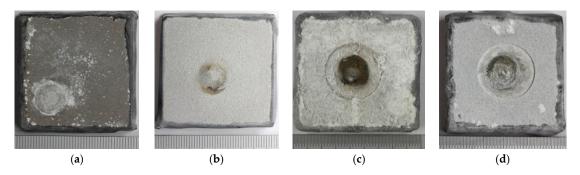


Figure 4. Photographs of TSA-coated specimens after up to two weeks exposure to synthetic seawater showing surface without holiday at 30 $^{\circ}$ C (**a**), and 60 $^{\circ}$ C (**b**); and surface with holiday at 30 $^{\circ}$ C (**c**), and 60 $^{\circ}$ C (**d**).

3.2. Microstructural Characterization

Figure 5 shows the microstructure of the deposit formed on the steel surface at 30 $^{\circ}$ C and 60 $^{\circ}$ C in the top view and cross-section. In the top view (Figure 5a,b) it is possible to see the needle structure, typical of aragonite [9,14]. In the cross-sections (Figure 5b,e) it was possible to identify two layers: A very thin dark grey layer below a thicker lighter grey layer. The regions were labeled according to their composition, as evidenced by EDX.

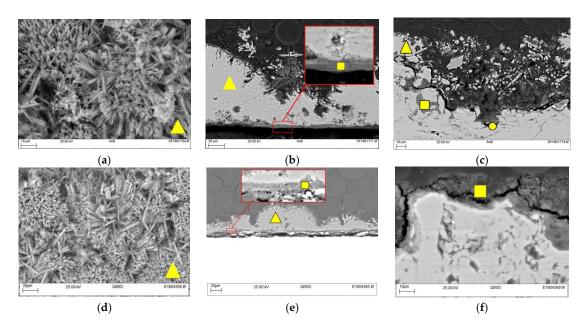


Figure 5. Calcareous deposits formed on the steel surface (in the holiday region) of the TSA coated sample: (a) 30 °C sample—holiday area in top view; (b) 30 °C sample—holiday area in cross-section; (c) 30 °C sample—TSA area in cross-section; (d) 60 °C sample—holiday area in top view; (e) 60 °C sample—holiday area in cross-section; (f) 60 °C sample—TSA area in cross-section.

It is important to highlight that the microstructural analysis was performed in a region away from the pH probe where the growth of the deposit was not affected by the presence of the probe. The constituents were identified by EDX analysis from both temperatures, where the main peaks of Ca, C, and O (Figure 6) and Mg and O (Figure 7) appeared. Al and O peaks were also observed on TSA (Figure 8).

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The XRD analyses confirmed the presence of $CaCO_3$ and $Mg(OH)_2$ (Figure 9) and $Al(OH)_3$ plus K-Al₂O₃ (Figure 10). Some constituents of synthetic seawater appeared in the XRD pattern, but were not always detected in the EDX spectra because the latter was performed in localized regions of the specimens while the XRD window is a bit wider with information collected from a greater volume of the deposit. The XRD results also show the presence of other compounds (like NaCl, CaSO₄, etc.) normally found in synthetic seawater (Figures 9 and 10).

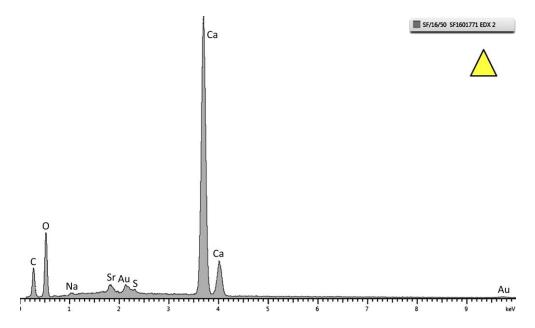


Figure 6. Energy dispersive X-ray (EDX) pattern of the top layer marked as a triangle in Figure 5. Strong peaks of Ca, C, and O are observed. This specific spectrum is from the sample tested at $30 \,^{\circ}$ C.

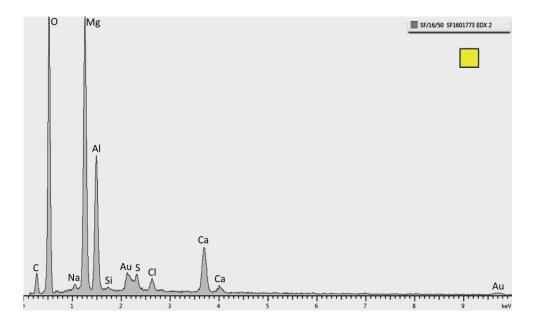


Figure 7. EDX pattern of the layer marked as a square in Figure 5. Strong peaks of Mg and O are observed. This specific spectrum is from the sample tested at 30 °C.

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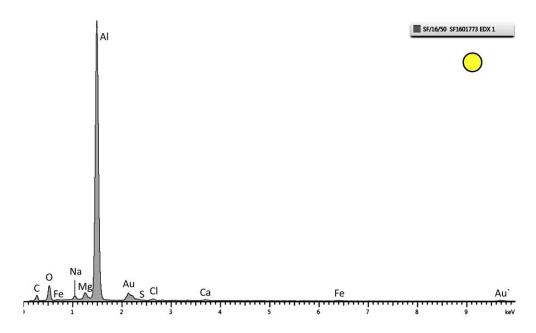


Figure 8. EDX pattern of the layer marked as a circle in Figure 5. Strong peaks of Al and O are observed. This specific spectrum is from the sample tested at 30 °C.

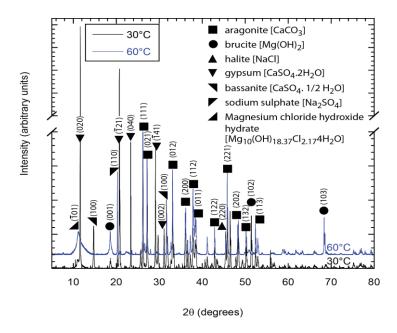


Figure 9. X-ray diffraction (XRD) patterns showing the predominant presence of brucite and aragonite in deposits from the holiday region of samples exposed to synthetic seawater at 30 °C and 60 °C.

The CaSO₄ products found by XRD were in the basanite and gypsum forms. The CaSO₄ has similar solubility values at 30 °C and 60 °C (0.209 and 0.204/100 g water, respectively) [15]. It has been reported that gypsum does not precipitate in a single step, occurring instead after basanite (CaSO₄·0.5H₂O) nucleation as part of a multi-stage crystallization process. Therefore, it is possible that at 30 °C the basanite nucleation was favorable which facilitated further formation of gypsum. This apparently did not happen at 60 °C [16]. Different crystal orientations are marked in the XRD pattern and it can be observed that the same chemical compounds appear in different patterns. This phenomenon is named polymorphism, where solids with the same chemical composition can exist in different crystal structures or different phases. This transformation changes with temperature

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and/or pressure [17]. Therefore, the presence of different patterns for $Mg(OH)_2$ and $CaCO_3$ at different temperatures is not unexpected.

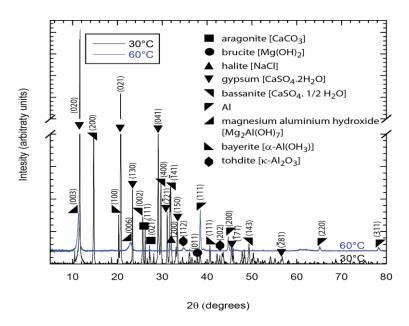


Figure 10. X-ray diffraction (XRD) patterns showing the presence of aluminum oxide in deposits from the TSA region of samples exposed to synthetic seawater at 30 °C and 60 °C. Al in the deposit is from the TSA.

3.3. pH Profiles

3.3.1. Synthetic Seawater at 30 $^{\circ}$ C and 60 $^{\circ}$ C

The seawater pH profile as a function of time is shown in Figure 11. It can be seen that the pH profile of synthetic seawater changed slightly when at 30 $^{\circ}$ C compared to 60 $^{\circ}$ C. The measured seawater pH value was 8.17 at 19.5 $^{\circ}$ C, prior to starting the test at 30 $^{\circ}$ C and 60 $^{\circ}$ C. At 30 $^{\circ}$ C, the seawater pH was 8.13 at the start which decreased to 7.70 after approximately 300 h. At 60 $^{\circ}$ C, the pH was 7.96 at the start and dropped to 7.6 after 240 h. Both cases presented a point in time when the decrease in pH occurred rapidly. This happened after 190 h and 120 h at 30 $^{\circ}$ C and 60 $^{\circ}$ C, respectively.

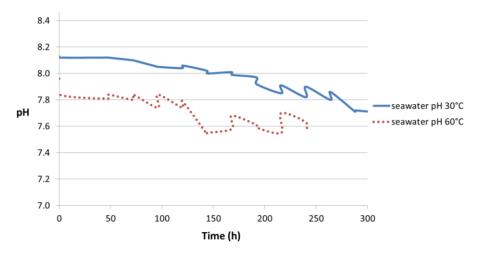


Figure 11. pH profiles of synthetic seawater in air at 30 °C and 60 °C without any specimen.

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3.3.2. TSA-Coated Steel Specimen

The pH profile from the TSA coating is shown in Figure 12. At 30 $^{\circ}$ C it shows an initial pH of 7.74 which started decreasing after 120 h and reached a value of 4.76 after 220 h. The bulk pH for the same test dropped from 8.23 to 7.57 during the same period. At 60 $^{\circ}$ C, an initial pH of 6.41 was recorded which later was found to fluctuate between 7.0 and 8.5 after 200 h of test. The pH of bulk seawater changed from 7.93 to 7.54 in 240 h.

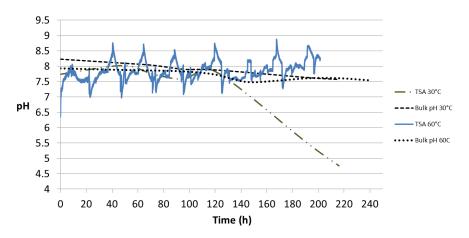


Figure 12. pH profiles obtained from the flat probe at 30 °C and 60 °C. The probe was in contact with the TSA-coated specimen surface without any holiday. The pH of the bulk seawater is also portrayed.

3.3.3. TSA-Coated Steel Specimen with Holidays

Samples with holiday showed a clear difference in the pH profile at the two temperatures tested, namely 30 $^{\circ}$ C and 60 $^{\circ}$ C (Figure 13). The local pH at 30 $^{\circ}$ C was monitored for 350 h and the initial pH of 7.2 increased to 9.87 followed by a slight decrease to around 9.6 before reaching a maximum of 10.19 in approximately 20 h, after which a smooth slope occurred until stabilization was achieved at pH 8.8. The sample exposed to 60 $^{\circ}$ C presented an initial pH of 7.51 and reached a peak in value of 9.54 in 4 h, followed by a sharper decrease until stabilization at pH 7.9 after 220 h of exposure. The low peaks observed at this temperature seem to be associated with the addition of deionized water to replenish the evaporation loss. At both temperatures, stabilization in local pH is preceded by a drop and subsequent stabilization of the bulk pH.

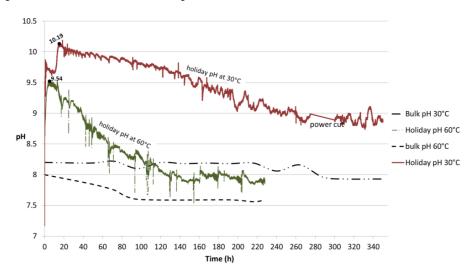


Figure 13. pH profiles obtained from the flat probe at 30 °C and 60 °C. The probe was in contact with the holiday region of the TSA-coated specimen. The pH of the bulk seawater is also portrayed.

The method used to measure pH may raise some concerns regarding the effect of obstruction to oxygen diffusion by the pH probe to the steel surface and thus giving incorrect pH data. However, prior to the test at 30 $^{\circ}$ C, which was the first to be run, the approximate values expected from previous studies had already been achieved. Although data at 30 $^{\circ}$ C was not available in the literature, the 25 $^{\circ}$ C literature data was not expected to be too different. The values in our 30 $^{\circ}$ C experiment were similar to reported room temperature data, already described in this paper. The comparison between modelling results (with no contact of the probe with the sample surface) from the literature and our experimental data were comparable. Although the past studies employed applied potentials, our experiments at open circuit potential with TSA coating polarizes the steel without an external potential being applied. Thus, no discrepancy in the results of the experiment at 30 $^{\circ}$ C was expected when compared to past studies at 25 $^{\circ}$ C. The local morphology of the calcareous deposit was possibly altered directly at the pH probe location, but not in its surroundings. Moreover, it should be highlighted that although a very careful pH probe calibration was performed prior to each test, the results are expected to be approximate values.

Bearing the above in mind it was possible to register some events. Firstly, at 30 °C (Figure 2b,c) some Fe-containing corrosion products (rust-like appearance) may have formed on the edges of the exposed steel area, which means that initial steel corrosion might have taken place. The picture was taken after 18 h of testing. Similar observations were also reported by Yang et al. [18] who indicated that on a cathodically polarized steel surface, the compound that is initially precipitated next to the iron-containing corrosion products is possibly brucite. The presence of brucite in the vicinity of iron hydroxides was explained by the co-precipitation of brucite during the formation of corrosion products when steel corrodes. The possibility of co-precipitation is based on the work by Packter and Derby [19] who established and examined the mechanisms of co-precipitation of magnesium and iron hydroxides from aqueous solution using potentiometric titration.

The initial corrosion of steel did not seem to proceed further as the Fe-containing corrosion product layer was limited to the initial area seen after 18 h. This did not extend any further even after 82 h. However, after 82 h a thin white deposit covering the steel surface was noted. This seems to have prevented the steel from corroding further. It is important to highlight that the time when the pictures were acquired does not necessarily represent the start of the event, but the time at which it was observed.

At 60 $^{\circ}$ C, a thick layer covering the whole holiday was observed after 17 h of exposure (Figure 3b). A much faster kinetic response is expected at elevated temperature (60 $^{\circ}$ C) as compared to 30 $^{\circ}$ C. This will be further discussed in Section 4.3.2.

4. Discussion

4.1. Bulk Seawater pH Profile

Figure 11 shows the pH of synthetic seawater decreasing with time. This can be explained by the change in solution chemistry in air. The properties of seawater are dependent on temperature and solution chemistry. The latter is often defined by the concentration of dissolved gases and salts [16]. The salinity of seawater is generally 3.5 wt % and the pH varies between 7.8 and 8.3 at room temperature. The pH is mainly controlled by the reactions involving the dissolution of CO_2 from air and the presence of un-dissociated boric acid (H_3BO_3) due to the buffering capacity of these compounds [4,20,21]. Also, the seawater evaporation leads to a saturation/supersaturation of some salts and their subsequent precipitation. The removal of some salts from solution also leads to changes in the pH. The first salt to be precipitated is $CaCO_3$, but only a small amount as predicted by the relatively low concentration (0.014% by weight) of bicarbonate and carbonate ions in seawater. When the volume of seawater is reduced to 19% of its original volume, calcium sulfate is precipitated either as an anhydrite ($CaSO_4$) or as gypsum ($CaSO_4 \cdot 2H_2O$). Sodium chloride in the form of halite (NaCl) is formed in higher amount when the volume is reduced to 9.5% of the original one. The final

volume (4%) may contain the highly soluble chlorides of potassium and magnesium [22]. The pH will show a variation as the compounds "salt out" of seawater.

The pH difference in the beginning of the test observed in Figure 11 (8.13 at 30 $^{\circ}$ C and 7.96 at 60 $^{\circ}$ C) can be explained by the change in the ionic product of water ($K_{\rm w}$) which increases with temperature (values are given in Section 4.3.3). This means that self-dissociation of water is increasing and a greater number of H⁺ ions (or more accurately, H₃O⁺) are present in solution at 60 $^{\circ}$ C. It must be noted that the electrical neutrality of water is not altered as equal number of OH⁻ ions are also present.

However, for the current study, the presence of CO_2 is also significant in determining the pH. The CO_2 present in air (400 ppmv) will dissolve in the surface water depending on the temperature and the presence of other dissolved ions, following Henry's law. The dissolution of CO_2 in water can be represented by Equations (1) and (2) and results is an increase in acidity (generation of H^+) or a decrease in the pH. Further discussion of the effect of temperature on CO_2 solubility in water is given in Section 4.3.2.

$$CO_{2(aq)} + H_2O \rightleftharpoons H_2CO_3 \rightleftharpoons H^+ + HCO_3^-$$
 (1)

$$HCO_3^- \rightleftharpoons H^+ + CO_3^{2-}$$
 (2)

Thus, the dissolution of CO_2 in seawater plays an important role in the gradual decrease of pH observed in Figure 11.

4.2. Local pH Profiles on the TSA Surface

Aluminum alloys of the 1xxx, 3xxx, 5xxx, and 6xxx series show good resistance to seawater as the passive film formed can resist attack by most seawater constituents. However, the tenacity of the surface oxide is dependent on the pH, temperature, chloride concentration etc. If the pH shifts outside the range 4.0 to 8.5 the stability of the oxide is compromised [2]. As shown in Figure 11, the local pHs, both at 30 °C and 60 °C were between 4.5 and 8.5 until 200 h exposure and in this pH range passivation is expected at 25 °C because of the formation of $Al_2O_3 \cdot H_2O$ [23]. However in this study the specimens were also exposed to seawater at 60 °C. An increase in temperature is likely to lead to an increase in the reaction kinetics, but the corrosion process is also dependent on the formation of precipitates and the change in the oxide film formed. Thus, the corrosion rate is unlikely to increase significantly unless the oxide layer dissolves rapidly [1,23].

The passive layer on aluminum is composed of (i) an inner amorphous oxide/hydrated oxide layer (adjacent to the aluminum surface) whose thickness depends on the temperature and (ii) a thicker outer layer of hydrated oxide [23]. As shown previously in Figure 10, two different types of aluminum oxide were formed: $Al(OH)_3$ at 30 °C and $K-Al_2O_3$ at 60 °C.

At 30 $^{\circ}$ C the rapid drop in pH is probably linked to the effect of the Cl $^{-}$ ions acting on the aluminum oxide layer. The effect of these ions on the passive film is not entirely clear, but so far the following has been discussed in the literature—breakdown of the passive film and its dissolution due to the attack of Cl $^{-}$ ions in the discontinuity of the film where a pit is formed, dissolving aluminum at the bottom (anode area) as shown in Equation (3) [1,23,24]:

$$Al \rightarrow Al^{3+} + 3e^{-} \tag{3}$$

$$Al^{3+} + 3H_2O \rightarrow Al(OH)_3 + 3H^+$$
 (4)

The generation of H^+ will decrease the pH as observed. Chloride ions migrate within the pit and form $AlCl_3$ which dissolves in the solution. However, for the pit to be able to propagate, a critical bulk chloride concentration of 1.6 M is needed, which is greater than that in seawater (0.57 M) [24]. However, the local chloride concentration in the pit may reach such values if evaporation of seawater occurs.

In the interfacial region, AlCl₃ is also likely to form by the reaction between the chloride ions and the passive film following Equation (5). This equilibrium is important since it will determine if the pit

will keep growing or cease to grow. The H⁺ formed in Equation (4) is also involved in the formation of AlCl₃ from hydrated alumina, which is generally formed when Al is in contact with water. If AlCl₃ is formed, the pit will keep growing [25]. The reaction can be written as:

$$Al_2O_3 + 6H^+ + 6Cl^- \leftrightarrow 2AlCl_{3(aq)} + 3H_2O$$
 (5)

However, more research is required to understand the role of the Cl⁻ ions in destabilization and dissolution of the passive film [24].

On the other hand, if Al_2O_3 is formed, the pit will passivate and become inactive and this happens when the solution within the pit is reverted to the composition of the bulk solution [3,24]. Thus, the relatively constant pH at 60 °C between 7 and 8.5 may have occurred due to passivation of Al_2O_3 .

4.3. Local pH Profiles in the Holiday Region

4.3.1. General Remarks on the Precipitation of CaCO₃ and Mg(OH)₂ from Seawater

For a better understanding of the mechanism of calcareous deposit formation, some relations and interdependencies must be considered. One would expect that the corrosion rate will increase with the rise in temperature and that the oxygen content will play an important role in seawater corrosivity. However, in this work the TSA coating acted as an anode to the steel and a calcareous deposit formed and acted as a barrier to the oxygen diffusion to the initially exposed steel surface in the holiday region [25]. Two main events happened which led to the precipitation of calcareous matter; (i) high local production of hydroxyl ions (OH⁻) at the steel surface/seawater interface (Equation (6)); and (ii) ionic products of the ions comprising the deposits exceeding the solubility product of magnesium and calcium compounds, leading to precipitation [9,18,25]. The exchange of electrons allows hydrogen evolution as shown in Equation (7).

$$O_2 + 2H_2O + 4e^- \rightarrow 4OH^-$$
 (6)

$$2H^{+} + 2e^{-} \rightarrow H_{2}$$
 (7)

According to the literature, $Mg(OH)_2$ will precipitate as long as the local pH reaches 9.5 (at room temperature), following the equilibrium shown in Equation (8) [5–9,26].

$$Mg(OH)_2 \leftrightarrow Mg^{2+} + 2OH^-$$
 (8)

Such an increase in OH^- ions (Equation (6)) also affects the calco-carbonate equilibrium (which is governed by the dissolution and subsequent dissociation of CO_2 in seawater) at the steel surface, shifting the equilibrium in Equation (9) to the right where the formation of $CaCO_3$ will be further favored following Equation (10), as long as a minimum pH of 7.5 at room temperature is reached [5,8–10,14,27,28].

$$OH^{-} + HCO_{3}^{-} \leftrightarrow CO_{3}^{2-} + H_{2}O$$
 (9)

$$CaCO_3 \leftrightarrow CO_3^{2-} + Ca^{2+} \tag{10}$$

Thus, the formation of CaCO₃ is dependent on the calco-carbonic equilibrium shown above. The reactions are initiated by the absorption of carbon dioxide into the surface water, which lowers the pH by dissociation and decreases the saturation of the carbonates [25].

4.3.2. Effect of Temperature on Solubility

The lower solubility of calcium carbonate at higher temperature favors its precipitation, as shown in Table 4 (temperatures close to the test conditions are reported due to the limited availability of data in the literature). However, as mentioned previously, the presence of gases such as CO_2 will

also contribute to the formation of CaCO $_3$. Figure 14 shows the solubility of pure CO $_2$ in water at different temperatures up to 60 °C, expressed in the amount of substance that is soluble in 100 g of water. The solubility of CO $_2$ at 30 °C is 0.1257 g/100 g, while at 60 °C it is 0.0576 g/100 g and it can be speculated that the increase in temperature and the consequent reduction of CO $_2$ solubility may also decrease the probability of CaCO $_3$ formation [15,29]. Although the data above is for pure CO $_2$ at 1 atm and the amount of CO $_2$ in air is 400 ppmv, the logic is still likely to be valid. Thus, the lack of copious calcium carbonate on TSA may be due to (i) the pH at the TSA surface not being ideal for precipitation; (ii) the consumption of the carbonates formed from dissolved CO $_2$ took place before the possibility of forming calcium carbonate outside the holiday region; and (iii) the kinetics of formation of Al corrosion product on the TSA surface outweighed the kinetics of deposition of CaCO $_3$.

Compound	Temperature (°C)	Solubility (g/100 mL)
C-CO	25	0.0153
CaCO ₃	75	0.0019
M-(OII)	35	0.00098
$Mg(OH)_2$	72	0.00069

Table 4. Solubility of CaCO₃ and Mg(OH)₂ in water at different temperatures [15,29].

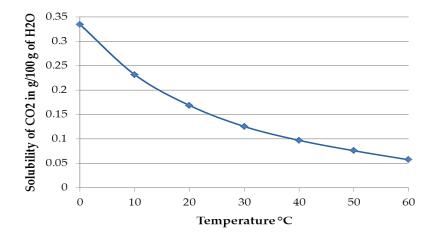


Figure 14. Solubility of pure CO₂ in water up to 60 °C. Data taken from Perry and Green [15].

Table 4 also suggests that a higher amount of brucite would form at higher temperature, primarily due to (i) the greater amount of Mg^{2+} in synthetic seawater solution compared to Ca^{2+} and the lesser solubility of $Mg(OH)_2$ and (ii) the lesser availability of dissolved carbonates (due to reduced solubility of CO_2). This could be observed in Figure 5 where more brucite was formed outside the holiday region, on the TSA coating surface.

It can be observed in Figure 13 that maxima in the peaks were reached very quickly followed by a drop. At first, the cathodic reaction increases the $[OH^-]$ at the steel surface (Figure 2b) which results in a pH increase. Such an increase is hindered when the local pH is high as at 10.19 at 30 °C and 9.54 at 60 °C. Those high pHs are favorable for $Mg(OH)_2$ precipitation, which will be further proved in Section 4.3.3 [30]. This is a very thin base layer (measured as 6 μ m) as observed in Figure 5b,e. Although very thin, this layer is able to hinder the diffusion of oxygen towards the steel surface in some of the areas, where it starts to form. The layer serves as a base layer for the further growth of $CaCO_3$, which is dependent on CO_2 diffusion as mentioned earlier and therefore on the calco-carbonate equilibrium (Equations (1) and (2)). Thus, the CO_2 is also being consumed to form $CaCO_3$, delaying the pH stabilization as the $CaCO_3$ is still being formed. Once both layers are fully formed, it acts as a natural barrier against further steel corrosion [6,9,27,30]. In summary, two events are taking place

even after the maximum peak at 30 $^{\circ}$ C; (i) local formation of OH $^{-}$ and (ii) consumption of CO $_{2}$ for the formation of CaCO $_{3}$.

At 60 °C, the calcareous deposit was already formed within 17 h (Figure 3b), straight after the maximum peak was reached. This faster limitation (of hindrance) of oxygen towards the steel surface, whose content is already less than at 30 °C [31], rapidly blocks further cathodic reaction or the generation of OH^- , decreasing the pH faster than at 30 °C. Considering the less availability of CO_2 in solution at 60 °C; it seems that the pH was more dependent on the local $[OH^-]$ at 60 °C.

4.3.3. Minimum Local pH at 30 °C and 60 °C for the Precipitation of Deposits

In order to better understand the environment that leads to the growth of the calcareous deposit on a steel surface, it is important to know the favorable pH for the precipitation to occur. These values are known at 25 °C, but information is lacking for the temperatures used in this study. The minimum pHs for precipitation of brucite and aragonite were calculated from thermodynamic data of the different species associated with the precipitation of brucite and aragonite (as presented in Equations (8) and (10)), given in Table 5. The ΔG° for the reactions were then calculated at 30 °C and 60° using Equation (11). From the data it was possible to calculate the solubility product ($K_{\rm sp}$) of each compound shown in Equation (12), where K is the ideal gas constant.

Table 5. Thermodynamic properties of different species in aqueous solution at 25 °C [32,33].

Element	ΔH° _f (kJ/mol)	S [◦] (J/mol K)
CO ₃ ²⁻ Ca ²⁺	-677.1	-56.9
Ca^{2+}	-542.8	-53.1
CaCO ₃ (aragonite)	-1207	88.7
Mg^{2+}	-466.85	-138.07
OH^-	-230	-10.8
$Mg(OH)_2$	-924.53	63.18

$$\Delta G^{\circ} = \Delta H - T \Delta S \tag{11}$$

$$K_{\rm sp} = \exp^{\frac{-\Delta G^{\circ}}{RT}} \tag{12}$$

From Equation (8), we can write the solubility product equilibrium in Equation (13) which shows the relationship of the Mg(OH)₂ solubility product with the concentration of its ionic constituents:

$$K_{sp}^{Mg(OH)_2} = [Mg^{2+}][OH^{-}]^2$$
 (13)

which leads to Equation (14):

$$[OH^{-}] = \sqrt{\frac{K_{\rm sp}^{\rm Mg(OH)_{2}}}{[{\rm Mg}^{2+}]}}$$
 (14)

From the ionic product of water (K_w) it is possible to draw the correlation in Equation (15). Values of K_w are available in the literature [34,35]. K_w is defined as:

$$K_{\rm w} = [{\rm OH}^-][{\rm OH}^+] \text{ or } [{\rm H}^+] = \frac{K_{\rm w}}{[{\rm OH}^-]}$$
 (15)

which can be used to calculate the pH as in Equation (16), where the relationship for the minimum precipitation pH is shown. In synthetic seawater, the concentration of Mg^{2+} was calculated from the data in Table 3 as 0.054 M. The solubility product data was taken from Table 6.

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Compound	Temperature (°C)	ΔG° (J·mol ⁻¹)	$K_{\rm sp} \ ({\rm mol}^3 \cdot {\rm L}^{-3})$
Mg(OH) ₂	30	65,024.77	6.24×10^{-12}
Wig(O11)2	60	71,922.47	5.28×10^{-12}
C-CO	30	47,335.90	6.97×10^{-9}
CaCO ₃	60	53,296.90	4.39×10^{-9}

Table 6. Calculated values of ΔG° and K_{sp} of brucite and aragonite at 30 °C and 60 °C.

$$pH_{Mg(OH)_{2}} = -\log[H^{+}] = -\log\frac{K_{W}}{\sqrt{\frac{K_{Mg(OH)_{2}}^{Mg(OH)_{2}}}{[Mg^{2+}]}}}$$
(16)

In the same way, the precipitation pH of $CaCO_3$ can be calculated, however, here Equation (9) must be considered which relates the formation of $CaCO_3$ to the concentration of OH^- . The molar concentration of OH^- is equal to the concentration of CO_3^{2-} (following Equation (9)) and therefore the relationship in Equation (17) can be derived which further results in Equation (18). The concentration of Ca^{2+} in synthetic seawater was calculated from the data in Table 3 as 0.010 M.

$$[CO_3^{2-}] = \frac{K_{\rm sp}^{\rm CaCO_3}}{\left[{\rm Ca}^{2+}\right]} \text{ or } [{\rm OH}^{-}] = \frac{K_{\rm sp}^{\rm CaCO_3}}{\left[{\rm Ca}^{2+}\right]}$$
 (17)

$$pH_{CaCO_3} = -\log[H^+] = -\log\frac{K_W}{\frac{K_{sp}^{CaCO_3}}{[Ca^{2+}]}}$$
(18)

With the data and expressions arranged, it was possible to calculate the minimum pH for the precipitation of both compounds: Mg(OH)₂ and CaCO₃. The results are summarized in Table 7.

Table 7. Summary	of values and	l calculated	d minimum	pH for the	precipitation of	brucite and	aragonite.
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Temperature (°C)		M) in Synthetic TM D1141 [13]	Ionic Product of Water (K _w)	Minimum pH for Precipitation		
(°C)	[Mg ²⁺]	[Ca ²⁺]	or water (rw)	pH _{Mg(OH)2}	pH _{CaCO3}	
30	0.054	0.010	1.47×10^{-14}	8.86	7.61	
60	0.054	0.010	9.6×10^{-14}	8.01	6.66	

The results show that the conditions for the precipitation of compounds were governed by the environment created in this study as the local pH reached values higher (as 10.19 at $30\,^{\circ}$ C and 9.54 at $60\,^{\circ}$ C) than the required minimum (Figure 13). The exact pH when the compounds started precipitating is difficult to define as it depends on complex interactions between different factors. The presence of Mg^{2+} in solution, for example, was reported to inhibit the nucleation of $CaCO_3$ [14]. Neville and Morizot [9] considered that the primary function of Mg^{2+} is to form a Mg-containing layer which makes the conditions favorable for the formation of $CaCO_3$. In addition, Barchiche et al. [6,36] mentioned that the presence of sulfate ions can hinder the deposition of $CaCO_3$ and favor the precipitation of magnesium hydroxide although the mechanism is not entirely clear. These considerations are important to understand in that prior to the formation of the brucite layer (which consumes the Mg^{2+} ions from the solution by forming a precipitate) no aragonite should be formed in the present experiments. These previous experimental data may help us understand the reasons for the delay in the deposition of $CaCO_3$ and why it is only formed at a higher pH than the calculated values.

Lee and Ambrose [30] mentioned that although the amount the dissolved oxygen content in water increases with a decrease in temperature, the diffusion coefficient of oxygen is diminished as well as the hydrogen evolution reaction, which can explain the lower pH necessary to precipitate $Mg(OH)_2$ at 60 °C compared to 30 °C.

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4.3.4. CaCO₃ Precipitation at 30 °C and 60 °C

One can wrongly assume that $CaCO_3$ will precipitate in acidic media at $60\,^{\circ}$ C, once its precipitation pH is found to be 6.6. However, it should be noted that the neutral point of water is defined as 7 at room temperature (25 $^{\circ}$ C), which is shifted to a different value at higher temperature. The neutral point of water at 30 $^{\circ}$ C and $60\,^{\circ}$ C can be easily found using the respective K_w values already mentioned. The neutrality occurs when

$$[OH^-] = [H^+] \tag{19}$$

and thus, as shown in Equation (20), it can be rewritten as

$$K_{\rm w} = [{\rm H}^+]^2$$
 (20)

From the above equation the neutral point of water at 30 $^{\circ}$ C is calculated to be 6.92 and at 60 $^{\circ}$ C it is 6.51. This means that the precipitation of all the compounds tookplace at a pH greater than the neutral point of water (alkaline environments) at the temperature concerned.

Yang et al. [18] calculated the minimum precipitation pH of calcium carbonate and magnesium hydroxide based on the solubility product data as 7.58 and 10 at 20 °C. No experimental pH measurement was carried out. The pH value found in the current study for the precipitation of $CaCO_3$ at 30 °C, which was 7.61, was very close to the value calculated by Yang et al. [18]. The slight difference is likely due to difference in the solubility values at the two temperatures. This may suggest that at temperatures up to 30 °C, the aragonite precipitation is not very sensitive to temperature change (between the temperature range 20 °C to 30 °C), while brucite had its minimum pH decreased from 10 to 8.86 within a 10 degree centigrade difference. However at 60 °C, the aragonite presented a higher fall in the pH value for precipitation compared to brucite. The results show that complex relations exist between the temperature and the pH required for the precipitation of the compounds.

5. Summary and Conclusions

The study covered the pH measurement in synthetic seawater, on a TSA surface and on the steel surface under cathodic protection at 30 $^{\circ}$ C and 60 $^{\circ}$ C. The pH of synthetic seawater changed with an increase in temperature due to (i) the change in dissociation constant of water which ultimately lowers the neutral pH of water with temperature; (ii) dissolution and dissociation of CO_2 from air in synthetic seawater; and (iii) the possible deposition of salts on the reactor wall. The local pH of the TSA coating increased with time at 60 $^{\circ}$ C and the opposite behavior was observed at 30 $^{\circ}$ C, in a mechanism related to the formation of Al(OH)₃ and hydrolysis of Al³⁺ in the presence of Cl⁻ ions. The AlCl₃ possibly formed in the chloride solution when hydrolysis leads to acidification and decreases the pH.

The pH in the holiday region, which represents the local pH of a cathodically polarized steel, showed a peak of 10.19 and stabilized at 8.8 at 30 °C. At 60 °C, the maximum peak occurred at 9.54 with the final pH measured at 7.9. The pH probe was touching the steel surface, so it is not expected that a correction in the values needs to be considered. However, a small gap between the actual measured region and the steel-seawater interface is likely.

Calculations showed that the minimum pH necessary for the precipitation of $Mg(OH)_2$ (brucite) in synthetic seawater is 8.86 at 30 °C and 8.01 at 60 °C, while for $CaCO_3$ the values are 7.61 and 6.66 for the above temperatures. It must be noted that these pH values are not the values when the compounds actually started to precipitate, but values theoretically predicted. The first deposit to be formed was a Mg-containing layer, which favored the growth of an aragonite ($CaCO_3$) layer on it, although other factors also influence $CaCO_3$ formation, for example the dissolution and subsequent dissociation of CO_2 in seawater.

The effect of temperature on solubility was discussed in this paper. The CO_2 solubility in seawater, and $CaCO_3$ and $Mg(OH)_2$ solubility in water were taken into account to calculate the minimum local pH required for the precipitation of deposits on the cathodically polarized steel surface. Thus, it is

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clear that the surface temperature of offshore pipelines carrying hot fluids will dictate the growth mechanism of calcareous deposits. Moreover, further studies at higher temperatures than 60 $^{\circ}$ C, possibly at elevated pressures need to be explored to understand the behavior of deposits under deep sea conditions. These could not be carried out in this research due to the limitation in the temperature and pressure capability of the available pH probes.

The formation of these layers (Ca-and Mg-containing deposits) on exposed steel has serious implications on the life of polarized steel structures. Generally, one would use galvanic anodes to polarize offshore steel structures, but the use of TSA coating as an evenly distributed anode on the steel surface has its advantages; firstly, as a barrier, and secondly, as a sacrificial coating when damaged. The ability of TSA to protect steel when damaged even at 60 °C is important for designers of structures which may experience hot seawater in service. If the Ca-and Mg-containing deposits restrict the ingress of corrosive species to the steel surface (exposed by coating damage), they can reduce the loss of TSA significantly, thus improving the life of the structure. The mechanism of deposit formation in the holiday region, enhanced by increase in local pH as elucidated here will, hopefully, help our understanding of the protective function TSA offers. Further, this will help us understand the general mechanism of protection of steel structures when cathodic protection (CP) is employed.

Acknowledgments: The work was funded by TWI. Nataly Ce would like to thank CNPQ and BG for the PhD scholarship. The authors also acknowledge the contribution of TWI staff, especially Mike Bennett, Ryan Bellward, and Sheila Stevens.

Author Contributions: Shiladitya Paul conceived and designed the tests; Nataly Ce carried out the tests, microstructural characterization, and wrote the paper; Shiladitya Paul reviewed the paper, made final changes, and contributed with technical knowledge.

Conflicts of Interest: The authors declare no conflict of interest.

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