

Article

Antibiofilm Activity of Epoxy/Ag-TiO₂ Polymer Nanocomposite Coatings against *Staphylococcus Aureus* and *Escherichia Coli*

Santhosh S. M. and Kandasamy Natarajan *

Applied Polymer Materials Laboratory, Department of Chemistry, R. V. College of Engineering, Mysore Road, Bangalore 560059, India; E-Mail: santhosh.s.m.rao@gmail.com

* Author to whom correspondence should be addressed; E-Mail: knatarajan.rvce@gmail.com; Tel.: +91-80-6717-8038; Fax: +91-80-2860-0337.

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Abstract: Dispersion of functional inorganic nano-fillers like TiO₂ within polymer matrix is known to impart excellent photobactericidal activity to the composite. Epoxy resin systems with Ag⁺ ion doped TiO₂ can have combination of excellent biocidal characteristics of silver and the photocatalytic properties of TiO₂. The inorganic antimicrobial incorporation into an epoxy polymeric matrix was achieved by sonicating laboratory-made nano-scale anatase TiO₂ and Ag-TiO₂ into the industrial grade epoxy resin. The resulting epoxy composite had ratios of 0.5–2.0 wt% of nano-filler content. The process of dispersion of Ag-TiO₂ in the epoxy resin resulted in concomitant in situ synthesis of silver nanoparticles due to photoreduction of Ag⁺ ion. The composite materials were characterized by DSC and SEM. The glass transition temperature (T_g) increased with the incorporation of the nanofillers over the neat polymer. The materials synthesized were coated on glass petri dish. Anti-biofilm property of coated material due to combined release of biocide, and photocatalytic activity under static conditions in petri dish was evaluated against Staphylococcus aureus ATCC6538 and Escherichia coli K-12 under UV irradiation using a crystal violet binding assay. Prepared composite showed significant inhibition of biofilm development in both the organisms. Our studies indicate that the effective dispersion and optimal release of biocidal agents was responsible for anti-biofilm activity of the surface. The reported thermoset coating materials can be used as bactericidal surfaces either in industrial or healthcare settings to reduce the microbial loads.

Keywords: epoxy nanocomposite; Ag-TiO₂ nanoparticles; photobactericidal; antibiofilm acitivity; crystal violet assay

1. Introduction

Biofilms are defined as communities of microorganisms that are developed on material surfaces. Prevention of microbial biofilm formation over the surface of materials is a technological imperative in health care. Many bacteria capable of forming biofilms on abiotic surfaces are menacing problems in medical and industrial systems. The biofilm forming ability of the opportunistic human pathogens Staphylococcus aureus and Escherichia coli, is a crucial step for sustenance and growth in above said environments [1]. Biofilms are a major source of biofouling in industrial water systems, and biofilm based industrial slimes also pose major problems for various industrial processes. Biofilm forming microbial cells attached to any surface in a moist environment can survive and proliferate. Pathogenic and resilient biofilms are difficult to eradicate with conventional disinfectants [2]. The interest in inorganic disinfectants such as metal oxide nanoparticles (NPs) is increasing. In the last decade, many studies describing the photocatalytic inactivation of bacteria using doped and undoped TiO₂ coated on different substrates have been reported, including silver doped TiO₂ [3–6]. A majority of these articles is focused on powder materials and thin films of TiO2 or doped TiO2. Unfortunately, most bare TiO2 coated films lose their efficiency of photocatalysis due to mass transfer [7,8]. However, only a fraction of studies deal with stemming of mass transfer of immobilized TiO₂ or doped-TiO₂ photocatalyst films. The most promising approach to overcome this disadvantage is by immobilization of TiO₂ in the porous polymer matrix such as epoxides, the most important classes of compounds used in the coating industry. These epoxy composites provide thin-layer durable coatings having mechanical strength and good adhesion to a variety of substrates [9]. Antimicrobial epoxy based surface coatings of walls and floors can fight the nosocomial menace [10] in hospitals.

The antibacterial function of a TiO₂ photocatalyst is markedly enhanced even with weak UV light, such as fluorescent lamps and with the aid of either silver or copper, which is harmless to the human body [11]. TiO₂ nano-fillers improve mechanical properties like crack resistance, surface characteristics and can also contribute to the photostability of the host material. The photostability and photocatalytic activity of epoxy/nano-TiO₂ coatings under UV irradiation has been reported by Calza *et al.* [12]. While doping TiO₂ with silver can synergistically enhance photobactericidal acitivity of TiO₂, a considerable improvement in mechanical properties can also be achieved by introducing very low amount of nano-fillers into resin system [13]. In addition, photo-stability of epoxy resin can be improved by the presence of nano-TiO₂ by its UV absorption properties [14]. Thus, modification of polymers with TiO₂ and subsequent coupling with Ag⁺/Ag NP enhance the photocatalytic and antimicrobial property of the material. Nanoparticles are generally introduced into epoxy matrix using various approaches like, *in situ* synthesis by reacting the precursors or physical dispersion of pretreated nano-fillers by mechanical stirring and subsequently processed by ultrasonication [15,16]. Successful dispersion of nanoparticles within the polymer matrix is determined by factors like particle size, particle modifications, specific surface area, particle load and the particle morphology.

Broadly there are two methods to impregnate a biocidal agent in order to achieve antibacterial polymeric materials. That is, either by introduction of aleaching biocidal agent into the polymer to form a composite or by covalent functionalization of the polymer with the pendent groups that confer antimicrobial activity. Such materials have displayed potent and broad spectrum antimicrobial activity [17]. The polycaprolactone-titania nanocomposites have been shown to decrease surface colonization of *Escherichia coli* and *Staphylococcus aureus* [18]. Similarly, introduction of (+)usnic acid, a natural antimicrobial agent into modified polyurethane prevented biofilm formation on the polymer surface by *Staphylococcus aureus* and *Pseudomonas aeruginosa* [19]. The poly(ethylene terephthalate) (PET) was surface functionalized with pyridinium groups possessing antibacterial properties, as shown by their effect on *Escherichia coli* [20]. Highly potent antibacterial activity toward both Gram-positive and Gram-negative bacteria was demonstrated by composites consisting of a cationic polymer matrix and embedded silver bromide nanoparticles [21].

There are very few empirical reports that quantitatively assess inhibition of biofilm formation on polymer surfaces by employing indicator dyes (crystal violet/fluorescent dye). Crystal violet (hexamethyl pararosaniline chloride) is such a dye, which binds proportionately to the peptidogly and can be a component of bacterial cell walls. It has been used by Kwasny and Opperman [22] to evaluate the amount of biofilm formed by staining the thick peptidoglycan layer of Gram-positive bacteria, the thin peptidoglycan layer of Gram-negative bacteria. In this study, anti-biofilm activity of polymeric surfaces was measured by protocol adoption as described by Kwasny and Opperman with minor modifications. The optical density of destaining solution after washing crystal violet adsorbed onto biofilm was measured with a multi-well plate spectrophotometer (using a 96 well titer plate). The color intensity of destaining solution after washing has been shown to be proportional to the quantity of biofilm formed. This method makes more practical high-throughput screening of polymer surfaces for their antibiofilm activity.

Metallic silver/TiO₂ and silver ion doped TiO₂ system in the form of films, deposition and its antibacterial performance in visible/UV light have been reported frequently [23–25]. To the best of our knowledge, there have been limited reports on the synthesis of polymers loaded with silver doped titania, for durable photobactericidal coatings that is compatible with many substrates to fight biofilms. In this work, composite materials suitable for coating was obtained by the addition of Ag-TiO₂ nanoparticles into epoxy resin system, with the aim to achieve "*in situ*" formation of silver species by photoreduction. The antibiofilm activity of this composite system is exhibited by the actions of photokilling and release of biocide (Ag^+/Ag^0) upon contact with aqueous environment.

2. Experimental Section

2.1. Preparation of Nanocrystalline TiO2 and Ag-TiO2

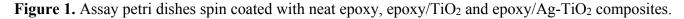
Ethanol 99.9%, Titanium(IV) butoxide, silver nitrate and acetic acid were of analytical grade and procured from Sigma Aldrich (Bangalore, India). About 1.5 wt% of Ag^+ ion doped nanocrystalline anatase TiO₂ was prepared by homogeneous hydrolysis of titanium butoxide-ethanolic solution using acetic acid-water as acid catalyst. The stoichiometric amount of AgNO₃ was dissolved in aqueous acetic acid and then added drop wise into the titania sol with stirring for 30 min at room temperature, and allowed to stand for two days at room temperature. Undoped TiO₂ gel was prepared by the same

procedure without the addition of AgNO₃. All the gels were isochronally annealed initially at 100 °C for 2 h then at 500 °C for 4 h.

2.2. Nanocomposite Preparation and Coating

The commercial grade resins, Lapox[®] L-12 [liquid epoxy resin based on bisphenol-A, (4.4'-Isopropylidenediphenol, oligomeric reaction products with 1-chloro-2,3-epoxypropane)] and reactive diluent, Lapox[®] XR-19 (Diglycidyl ether of polypropylene glycol) were procured from Atul Ltd., Ahmedabad, India. Diethylenetriamine (DETA) as a curative agent from Sigma-Aldrich was employed. The low molecular weight epoxy Lapox[®] XR-19, was added as diluents to lower the viscosity of the base resin and improve the initial physical dispersion of TiO₂ in the epoxy. The nanocomposites were prepared as follows: (i) the resin mixture was prepared (resin + diluant); (ii) the resin solution was diluted with ethanol to further decrease the viscosity of the resin mixture at 1:5 ratio; (iii) different amount of TiO₂ or Ag-TiO₂ was mixed into the diluted resin mixture. Then, the mixtures were sonicated under water bath for 30 min and degassed under vacuum. The resin-to-curative ratio in the material preparation at 10% of resin mixture weight was added. The mixtures were spin coated into the 50 mm \times 12 mm (outer dia \times height) size Borosil[®] S-Line petri plate on flat bottom dish and allowed to dry at room temperature for 24 h. The coatings were postcured at 100 °C for 2 h. Six different material samples were coated-neat epoxy resin, undoped TiO2/epoxy composite with 1 wt% loading and Ag-TiO₂/epoxy composite with 0.5, 1.0, 1.5 and 2.0 wt% loading Figure 1. The epoxy/Ag-TiO₂ composite turned pale brown indicating the formation of silver nanoparticles due to photoreduction. The coated substrates were sterilized by autoclaving at 121 °C, for 15 min before the start of experiments.





2.3. Physicochemical Characterization

Powder X-ray diffraction (PXRD) measurements were recorded by Bruker D8 Advance (Bruker AXS Inc., Madison, WI, USA) X-ray diffractometer with Cu K α radiation (1.5418 Å) at a 40 kV accelerating voltage and 30 mA. Raman measurements were performed with Renishaw Raman Microspectrometer (RM1000 System, Renishaw, Tokyo, Japan) of spectral resolution of 1 cm⁻¹ and spatial resolution of ~2.5 nm (using 50X Objective and 514.5 nm laser line). Scanning electron microscopy (SEM) images were captured using a Philips XL30 CP microscope equipped with EDX (energy dispersive X-ray) (Philips, Eindhoven, The Netherlands). The Brunauer–Emmett–Teller (BET) surface area (calculated from nitrogen adsorption data) was measured on a Quantachrome NOVA 1000 system at –180 °C.

UV-Vis diffuse reflectance spectra (DRS) were recorded using Analytik Jena Specord S600 spectrometer (Analytik Jena AG, Jena, Germany) (diffuse reflectance accessory with integrating sphere) by using BaSO₄ as a reference. All the above charecterizations were performed for the prepared nanocrystalline TiO₂ and Ag-TiO₂. The thermal property of composite materials was investigated by differential scanning calorimetry using Mettler-Toledo DSC823e (Mettler-Toledo AG, Schwerzenbach, Switzerland), and scans were performed at 5 °C/min for each composite under nitrogen flow and T_g value was extrapolated from the curves of second run.

2.4. Quantitative Determination of Biofilm

Bacteria used in this study were biofilm-proficient S. aureus ATCC 6538 and E. coli K-12 strains. Biofilm formation was measured under static condition by adopting quantitative crystal violet (CV) binding assay of Kwasny and Opperman with modifications [22]. In the current study, the flat inner surface of glass petri dish coated with prepared composites and resin was overlaid with 4 mL of sterile nutrient broth (composition is tabled in Supplementary Materials), so that the total area of the coating was covered. Then, 0.1 mL of logarithmic phase cultures of either E. coli or S. aureus grown over night to an optical density of *ca*. 0.1, at 595 nm in the appropriate growth media, were inoculated into sterile media in coated bottom plates prepared as above. Inoculated bottom plates were incubated in a bacteriological incubator at 37 °C under UV-A irradiation with intensity of 0.2 mW/cm² with λ_{max} around 365 nm (which is harmless to cause bacterial reduction), for different exposure durations. Later, the broth with planktonic cells was discarded by decantation. The plates were washed twice by gentle swirling with 2 mL of sterile phosphate-buffered saline to remove any non-adherent cells. Cells which remained adherent (biofilm mass) to the surface of polymer coated bottom plate were fixed by heating in a hot air oven at 60 °C for 60 min. Later plates were cooled to room temperature and stained with 1 mL of 0.06% (w/v) solution of crystal violet which was allowed to stand at room temperature for 5 min. Then plates were washed several times with phosphate-buffered saline to remove excess CV staining. Biofilm bound CV was eluted by vortexing with 1 mL of 30% acetic acid (destaining solution) for 10 min. The 0.2 mL aliquots of the wash solution with eluted crystal violet were transferred to 4 different wells of 96-well microtiter plates for the purpose of measuring the absorbance at 600 nm. Results were expressed as inhibition percentages of biofilm development. The percent inhibition of biofilm growth produced by each nanocomposite surface was calculated with the formula,

$$\left\{1 - \begin{bmatrix} \text{CV OD}_{600} \text{ (composite)} \\ \text{average CV OD}_{600} \text{ (negative control)} \end{bmatrix}\right\} \times 100$$
(1)

where CV OD₆₀₀ is OD of crystal violet destaining solution obtained at λ_{max} 600 nm. The results are presented as the average of four individual replicates. To check the binding affinity of CV to the prepared composites and neat epoxy, a similar assay with 48 h of UV exposure was conducted as above with the plain broth which was not inoculated with bacteria. The OD of destaining solution when measured was found to be insignificant to interfere with the experimental results. Then, the resulting silver concentrations in the same plain broth were also quantified by atomic absorption spectroscopy (AAS) analysis, released into the exposed media by the composites of different Ag-TiO₂ loadings. AAS analysis of released silver concentration was carried out with a 7700X instrumentation (Agilent, Santa Clara, CA, USA), using different standard concentrations. The reduction in biofilm colonization on composite was also determined in terms of CFU (colony forming unit), by sonicating assayed composite plate with 5 mL PBS for 5 min to remove adherent bacteria. The PBS suspension of released cells was then diluted appropriately, and spread on nutrient agar plate. The bacterial CFUs per milliliter of PBS that formed upon the medium was determined after incubation for 48 h at 37 °C. The experiment was repeated two times under identical conditions along with negative control (neat epoxy). The biofilm log reduction values were determined as difference between Log₁₀ CFU/plate recovered from the treated plates and Log₁₀ CFU/plate recovered from control plate (neat epoxy). Each experiment was conducted with three replications for each composite plates and colonies were enumerated to obtain the log reduction.

3. Results and Discussion

3.1. Characterization of Materials

Sol-gel derived nanocrystalline TiO₂ were subjected to the XRD analysis to determine crystalline phase and crystallite size. Titania exists in three crystalline polymorphs-anatase, rutile and brookite forms. Among these, anatase titania has been shown to exhibit higher antimicrobial activity than the other two and thus pure anatase phase content is a desirable feature [26]. The PXRD of titanias synthesized in this work had the peaks characteristic of anatase phase Figure 2a. (JCPDS No. 21-1272). From the X-ray diffraction patterns, the size of anatase TiO₂ materials prepared were in the nanometric scale Table 1. The average crystallite size was determined from the (101) plane in the PXRD pattern using Scherer's formula. The calculated value of undoped TiO2 had bigger crystallite size while Ag-doped TiO₂ showed a decrease in the crystallite size. A good correlation between the Raman and PXRD was also observed Figure 2b. The changes in the crystallite size of TiO₂ nanocrystals upon Ag-doping are closely correlated to the broadening and shifts of the Raman bands with decreasing particle size [27]. Similar observations were made for the titania systemesised in the present work. During annealing process, silver nitrate thermally decomposes into silver. Bigger ionic radii of Ag⁺ (0.75 Å) compared to Ti⁴⁺ (0.605 Å) prevents it from entering the crystal lattice of anatase TiO₂ because of a high energy barrier. Thus, it gets distributed uniformly on the surface of TiO₂. However, the PXRD pattern of Ag-TiO₂ did not reveal any Ag or Ag-containing phases. This may be due to the low concentration of Ag incorporated which is below the detection limit of the PXRD analysis.

Doping with Ag^+ ion also resulted in increase in the BET surface area of TiO₂ (48 m²/g), while that of undoped TiO₂ showed BET surface area of 27 m²/g. Thus, large surface area to volume ratio of Ag-doped TiO₂ was advantageous for the release of Ag^+ ion. From the energy dispersive X-ray (EDS) analysis at two locations (see Figure 3a), done during the SEM confirms silver is dispersed uniformly in TiO₂ host. Figure 3b shows the changes in the absorbance of Ag-doped TiO₂ in comparison to undoped TiO₂ and Degussa P 25 titania. Ag doped TiO₂ (calcined in ambient air at 500 °C) was found to have higher visible absorbance. In contrast, pure TiO₂ prepared under similar experimental conditions, had its absorbance slightly shifted towards the visible region as compared to Degussa P25 titania (Figure 3b). The DRS spectra showed a characteristic absorption band at about 500 nm, due to the surface plasmon resonance of silver [28]. Using the different absorbance onsets, it was found that the Ag-TiO₂ had a bandgap of ~2.8 eV while both of the undoped titania samples had wider band gaps estimated at \sim 3.1 eV for TiO₂ and \sim 3.2 eV for the Degussa P25 TiO₂ sample. Similar observations from previous studies can be confirmed [29].

Composite type	Nanocrystalline-TiO ₂		Epoxy-TiO2 composite		Amount of Ag released
	Crystallite size (nm)	BET surface area (m²/g)	Glass transition temperature <i>T</i> g (°C)	Weight of the coated composite (gm)	by the composite (μg/mL) *
Neat Epoxy	n/a	n/a	93	1.08	Nil
1.0 wt% Epoxy/TiO ₂	36	27	90	0.99	Nil
0.5 wt% Epoxy/Ag-TiO ₂	18	48	94	1.09	6.6
1.0 wt% Epoxy/Ag-TiO ₂	18	48	97	0.95	10.2
1.5 wt% Epoxy/Ag-TiO ₂	18	48	106	1.05	14.6
2.0 wt% Epoxy/Ag-TiO ₂	18	48	97	1.10	16.8

Table 1. Physio-chemical properties of nanofiller, T_g , weight of coated composite material and amount of silver ion released.

* Concentration of silver in the exposure media as determined by Atomic Absorption Spectroscopy (AAS), after 48 h.

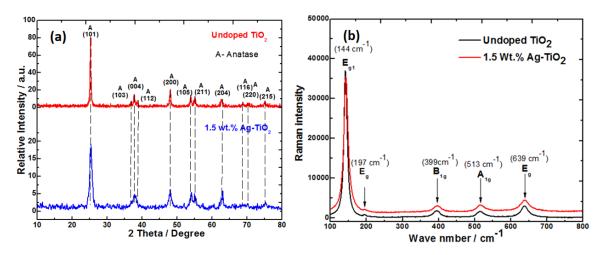


Figure 2. (a) Powder X-ray diffraction (XRD) and (b) Raman spectra of TiO₂ and Ag-TiO₂.

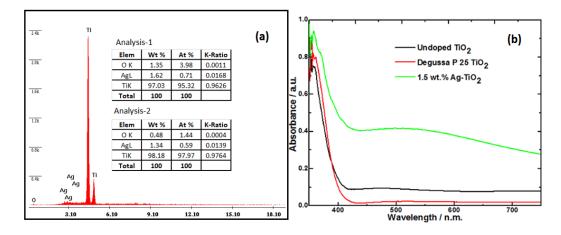


Figure 3. (a) Elemental analysis (EDS) of the silver doped TiO₂ showing the presence of Ti and Ag species; (b) UV-Vis diffuse reflectance spectra (DRS) of Ag-doped TiO₂, TiO₂ and Degussa P25 titania.

The homogeneous distribution of nano-filler in a polymer matrix has major influence on the composite performance. The morphology of synthesized titania nanoparticles and their dispersion in epoxy matrix were examined by SEM analysis Figure 4. The primary particle size of undoped and silver doped titania are different, varying from nanometer to micron size for the same magnification as seen in SEM micrographs Figure 4a,b. The undoped sample exhibited a nanostructure consisting of spherical clusters with a diameter of 50-500 nm, which are extensively agglomerated with an average crystallite size of 36 nm. However, silver doped titania showed bigger aggregates and smaller segregated particles consisting of primary anatase nanocrystals of 18 nm size (Figure 4b). Dispersion is an important factor in determining a nanocomposite's properties. Composites with the same weight percent (1 wt%) of nanofiller showed different degree of dispersion Figure 4c,d. The unmodified TiO₂ although thoroughly distributed in the matrix, yet particles agglomerated densely as shown in Figure 4c giving scattered hill lock like appearance on the surface of the composite. The size of these agglomerates varied from nanometers to micrometers. However, the Ag-TiO₂ particles Figure 4d, showed a lesser degree of agglomeration; interparticle distance are clearly visible between the TiO₂ particles. This indicates that the presence of silver enable good dispersion due to the interaction of oxidized silver ions with surface hydroxyl groups (titanol groups, Ti-OH) of TiO₂ and increase its wettability in apolar media like epoxy (hydrophobic polymer matrix). While Figure 4e shows the fractured surface of the composite, dispersion in the bulk is similar to distance between agglomerates as on surface. This suggests that the doped nano-fillers have better dispersion due to surface modifications, which improve the interactions between particles and polymer matrix. Use of reactive diluant also significantly reduced viscosity of epoxy resin during preparation and optimized the dispersion along with sonication.

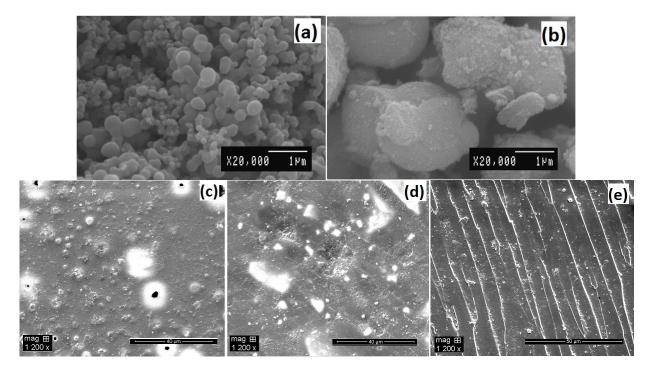


Figure 4. Scanning electron microscopic (SEM) characterization of (**a**) sol-gel synthesized TiO₂; (**b**) 1.5 wt% silver doped TiO₂; (**c**) 1 wt% epoxy/TiO₂ composite; (**d**) 1 wt% epoxy/Ag-TiO₂ composite; (**e**) Fractured surface of 1 wt% epoxy/Ag-TiO₂ composite.

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The glass transition temperature (T_g) of the samples were determined from the tangents of DSC spectra as a function of temperature. The DSC curves of the neat epoxy and nanocomposites with 1 wt% of TiO₂ and Ag-TiO₂ nanofiller from the second run are shown in the Figure 5a. For thermosetting resin glass transition temperature (T_g), values can shift due to reasons like cross-linking density, intermolecular interaction and chain length. The addition of nanometer sized TiO₂ particles in epoxy resulted in increase in the T_g from 93 °C for neat epoxy to 97 °C at 1 wt% loading of Ag-TiO₂. Whereas, T_g of composite shifts to lower temperature with undoped TiO₂ (1 wt% loading) due to poor dispersion and agglomeration as evident in the SEM micrograph. Nanocomposites with Ag-TiO₂ exhibited maximum T_g value at 1.5 wt% loading (107 °C) (Figure 5b). A further increase in the nano-filler content to 2 wt% led to the drop in the T_g value, this is due to their easy agglomeration arising from van der Waals attraction between particles.

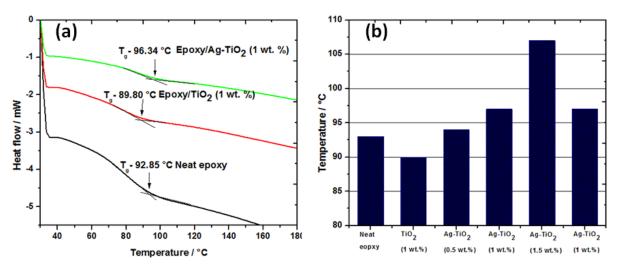


Figure 5. (a) DSC thermograms of neat epoxy and nanocomposites with 1 wt% of TiO₂ and Ag-TiO₂; (b) Variations in T_g values of neat resin and nanocomposites at different wt% of TiO₂/Ag-TiO₂ loading.

It can be seen from Figure 5b that the T_g value increases steadily then value drops; this corroborates with the trend observed by other investigators [13,30]. With our study, the degree of dispersion and nanofiller loading affected the shifts in T_g for epoxy/Ag-TiO₂ composites. The size, loading and dispersion state of the nanofillers are the factors that impact the glass-transition temperature. The T_g value increases due to polymer chain-filler (organic-inorganic interfacial contact) that are immobilized by cohesive interactions at the interface of nanofiller in the bulk of the material. On the other hand, higher loading of nanofiller or their agglomeration can result in mobile moieties within the matrix which significantly decrease the glass transition temperature. Very high T_g values are not achievable by room temperature curing agents, and the composites reported here can find their applications at temperature conditions below their T_g . These synthesized epoxy composites may be cross linked by means of any conventional hardener at room temperature, without the decomposition of incorporated biocides.

3.2. Antibiofilm Activity on the TiO₂ and Ag-TiO₂ Nanocomposite Coatings

Antibacterial epoxy coatings for antibiofilm properties were tested against *S. aureus* and *E. coli* under static conditions in glass petri dish with UV-A irradiation, on the surfaces of TiO₂ and Ag-TiO₂

composites (both with 1 wt% loading). Both *S. aureus* and *E. coli* were able to form biofilm on neat epoxy resin surface (negative control) and composites, *i.e.*, biofilm formation was independent of the underlying composite substrates. In the absence of TiO₂, epoxy resin showed higher growth of biofilm than that of epoxy/TiO₂ composite. Anti-boifilm activity appeared to increase significantly for Ag-TiO₂ composite.

The biofilm inhibition by composites does not seem to be restricted to specific strains or growth conditions; *E. coli* and *S. aureus* varied in their ability to produce biofilm on the surface of the composites as shown in Figure 6. In all assays, the amount of crystal violet eluted from *E. coli* biofilms was lower than that of *S. aureus* biofilms, because *E. coli*, being a Gram negative organism binds lesser dye than Gram positive organisms like *S. aureus*. The OD₆₀₀ of CV eluates from both biofilms was in the range of 0.121 to 2.8. Among the bacterial pathogens, *E. coli* was more susceptible for biofilm inhibition than *S. aureus* on these surfaces.

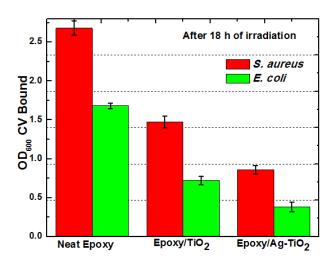


Figure 6. Spectrophotometric analysis (OD₆₀₀) of solubilized crystal violet of *E. coli* and *S. aureus* biofilm at 18 h irradiation time on the surfaces of TiO₂ and Ag-TiO₂ composite with similar loading (1 wt%).

To confirm the activity of TiO₂/Ag-TiO₂ on the surface of nanocomposite for the photokilling, we conducted the experiments under both dark and irradiated conditions as shown in Figure 7, and we found that higher inhibition of biofilm under irradiated conditions as shown in Figure 7b. The Ag-TiO₂ composite (1 wt%) showed 24% and TiO₂ composite (1 wt%) showed 6% biofilm inhibition of *E. coli* after 18 h of incubation in the dark as shown in Figure 7a. For the same conditions with UV irradiation *E. coli* biofilm showed 56% inhibition for epoxy/TiO₂ and 77% inhibition for epoxy/Ag-TiO₂, while that of *S. aureus* biofilm showed 43% and 67% ihibition, for epoxy/TiO₂ and epoxy/Ag-TiO₂ composites respectively. It is, therefore, the bactericidal activity of silver on biofilm that is rendered more likely in the absence of photokilling by Ag-TiO₂ with the dark experiment data. However, enhanced antibiofilm response of Ag-TiO₂ composite under UV irradiation can be attributed to the silver surface plasmon band favoring UV light absorption along with nanometer sized silver particles which exhibited a striking degree of synergy. The antibacterial feature was diminished for epoxy/TiO₂ composite in the dark experiment. However, the bare TiO₂ particles which are non-photo-activated on the surface also supported minor antibacterial activity, even in the dark. This is due to direct attack of cells upon contact

with TiO₂ nanoparticles which disrupt the integrity of the bacterial membrane [31,32]. This is also in agreement with reported experimental findings by Gogniat *et al.* [33] who also showed a loss of bacterial culturability after contact with TiO₂ nanoparticles even in the dark. These data show that the nature of epoxy resin makes it suitable host for dispersion of photocatalyst like TiO₂ for bacteriacidal activity.

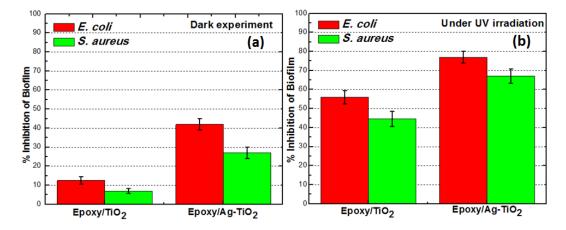


Figure 7. Mean values of quadruplicate experiments showing percent inhibition of *E. coli* and *S. aureus* bio-film formation on $epoxy/TiO_2$ and $epoxy/Ag-TiO_2$ composite surfaces calculated relative to the neat epoxy (negative control), under (**a**) dark and (**b**) UV irradiated conditions.

The release of the antimicrobial species $(Ag^+, Ag^0 \text{ and ROS})$ from a composite occurs due to the interaction of the diffused water molecules with TiO₂ and dispersed silver within the matrix during UV exposure; upon submerging it in the culture media [34,35]. Silver ions resident within the metal oxide nanofiller can diffuse to the surface of the epoxy matrix. The leaching of Ag⁺ ions was confirmed by AAS analysis of the bacterial media from blank experiments (without inoculums as explained in the experimental section). The Ag⁺ ion concentration of the same media was determined by atomic absorption spectrophotometer (AAS), which strongly suggests Ag⁺/Ag⁰ are associated noncovalently with cross-linked polymeric host and has leached to aqueous medium. By AAS analysis, the silver concentration (Ag⁺/Ag⁰) in the exposed media for the different epoxy/Ag-TiO₂ composite, showed a nonlinear increase that approached a maximum for the composite with 2.0 wt% of Ag-TiO₂ loading Table 1.

The valence band "electrons" can be excited to the conduction band (e_{cb}^{-}) , leaving positive "holes" in the valence band (h_{vb}^{+}) to form an e^{-}/h^{+} couple that react with aqueous environment and oxygen, to generate reactive oxygen speces (ROS) such as OH⁻, HO₂⁻⁻ and O₂⁻⁻, which are responsible for the mechanistic photo-biocidal activity [36,37]. The photoexcitation of non-leachably associated TiO₂ occurs when it absorbs light equal to or greater than band-gap energy near-ultraviolet light region. While Ag NP and Ag⁺ could act as efficient electron scavengers, and significantly enhanced the visible light responsiveness of TiO₂ to generate more oxygen free radicals by improving the quantum efficiency of a charge pair generated [35]. At the same time, these oxygen species can reduce Ag⁺ ions to form Ag nanoparticles. The smaller Ag⁺ ions can easily penetrate the cell wall and thus can hasten antimicrobial activity.

The attack of Ag⁺ on disulfide or sulfhydryl (thiol) groups present in the membrane protein result in formation of stable S–Ag bond with –SH groups thus inhibiting enzyme-catalyzed reactions and the

electron transport chain that are necessary for biofilm formation [38]. We speculate that the outer membrane of the bacterial cell is attacked by photocatalytic oxidation enabling the antimicrobial metal ions/particles to diffuse to interior of the cell thus becoming much more lethal to the bacterium. Thus, capability of photoactiveTiO₂ and leachable silver in destabilizing the biofilm matrix is enhanced by synergistic approach.

3.3. Effect of Exposure Duration on Formation of S. Aureus and E. Coli Biofilms

Figure 8 shows OD₆₀₀ values of eluted dye solution by *E. coli* and *S. aureus* for different duration of exposure (6 h, 9 h, 12 h, 15 h, 18 h, 20 h, 22 h and 24 h) of neat epoxy, epoxy/TiO₂ (1 wt%) and epoxy/Ag-TiO₂ (1 wt%). The biofilm ODs presented are averages of four independent experiments. Time course studies showed bactericidal ability of prepared composite surface up on contact and effectiveness in restraining bacterial biofilm formation. *S. aureus* biofilm formation response to time increased gradually, but it declined over a longer incubation period. It is plausible that this is due to biosorption of minerals and metals by microbial biofilms from the environment with which they are in contact [39,40]. When higher levels of silver is reached or with chronic exposure, it should be possible to limit the ability of the biofilm biosorption capacity, silver would then inhibit biofilm formation during prolonged exposure.

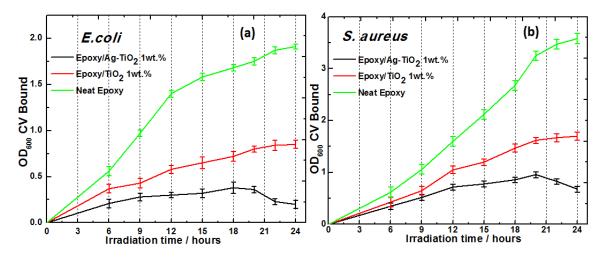


Figure 8. Growth curve for biofilm formation on neat resin, epoxy/TiO₂ and epoxy/Ag-TiO₂ composite of (a) *E. coli* and (b) *S. aureus*.

3.4. Effect of Ag-TiO₂ Loading on Biofilm Inhibition

The results showed that biofilm formation was highly inhibited in a dose dependent manner as shown in Figure 9. Increasing the load of Ag-TiO₂ resulted in shorter inhibition time *i.e.*, antibiofilm activity of composite is directly proportional to Ag-TiO₂ loading. Exposure of the composite with 1.5 wt% Ag-TiO₂ for 24 h. resulted in a inhibition of 100% (as per crystal violet binding assay) of both *E. coli* and *S. aureus*. The higher activity of these composites against *E. coli* a Gram-negative bacterium is attributed to its thinner peptidoglycan cell wall compared to *S. aureus* a Gram-positive bacterium. Complete inhibition of biofilm was achieved with 24 h of irradiation time with composite of Ag-TiO₂ with 1.5 wt% loading, in case of both *E. coli* and *S. aureus* (see Figure 9a,b).The antibacterial activity could also have effect on planktonic bacteria due to silver that has diffused to media from the matrix.

The bactericidal efficacy of these composite is through the diffusion of photogenerated ROS and Ag⁺ particles (acting as a leaching biocide) to the surface from the bulk of the polymer where such species/particles attack proteins and membrane lipids in bacterial cell wall. The driving force for silver particle diffusion is determined by a concentration gradient, which forms between the bulk of the composite material and the surface. The diffusion behavior depends on several factors including the structure of the material, environmental osmolarity and temperature.

We have quantified the silver release characteristics at 37 °C for the composites loaded with the 0.5 wt% to 2.0 wt% Ag-TiO₂ filler Table 1. And observed that non linear increase in the release of silver on increase of Ag-TiO₂ loading. The total released silver from the coatings was 6.6 to 16.8 µg/mL (16.8 ppm) after 48 h by epoxy/Ag-TiO₂ composites in the culture media without inoculum. From this observation it can be concluded that all the Ag-TiO₂ containing composites can have antibacterial activity even in the dark due to release of silver. However, presence of UV light will hasten the bactericidal activity of the composite due to photogeneration of ROS. Similar observations were made by Akhavan and Ghaderi [41] who investigated bactericidal activity of the anatase-TiO₂, the Ag thin film and the Ag-TiO₂/anatase-TiO₂ nanocomposite thin film against E. coli at dark and under UV exposure. In addition, they found superior antibacterial activity of Ag-TiO₂/anatase-TiO₂ nanocomposite thin film under the UV irradiation due its photocatalytic capability when compared to non-photocatalytic bare Ag and TiO₂ films and the silver ions released by Ag-TiO₂/anatase-TiO₂ nanocomposite thin film became saturated after 20 days at ~2 nM/mL. It is also possible to regulate the release of silver to the desired concentration by varying the nano-filler load incorporated into polymer composites and by tuning Ag-TiO₂ structure/composition during the sol-gel incorporation process. Antibiofilm activity of these composite remained unchanged at least for 5-6 cycles when we challenged during experiment through replications, this is due to continuous and uniform diffusion of the antimicrobial agents (ROS and silver species).

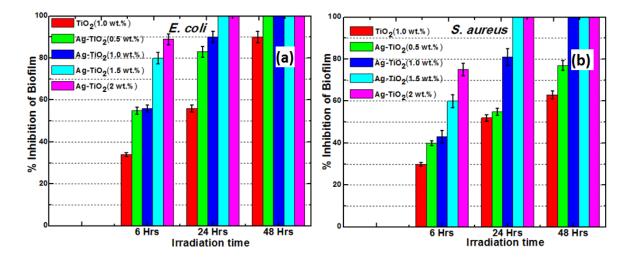


Figure 9. Biofilm inhibitory effect of Ag-TiO₂ loading (dose response) after 6, 24 and 48 h of irradiation on (**a**) *E. coli* and (**b**) *S. aureus*.

3.5. Quantitative Comparisions

There is no general consensus evolved for the comparison of efficiency of antibacterial activity of polymers surfaces between the research groups. However, most studies on antibacterial activity are interpreted by the number of surviving colony forming unit CFU/mL⁻¹ or per unit area. Kubacka et al. [42], studied the antibacterial effect of isotactic polypropylene (iPP) polymeric matrix incorporated with anatase-TiO₂ against Pseudomonas aeruginosa (Gram negative) and Enterococcus faecalis (Gram positive). They reported a maximum reduction by ca. 8-9 log in 30 min in case of P. aeruginosa. Francolini et al. [19] evaluated the effect of (+)-usnic acid incorporated into modified polyurethane surfaces on the biofilm forming ability of S. aureus. After three days postinoculation, they found culturable biofilm cell concentration of S. aureus on the untreated polymer was 7.3 log₁₀ CFU/cm² compared to 0.9 log₁₀ CFU/cm² on the (+)-usnic acid-containing polymer. Cen et al. [20] introduced pyridinium groups at 15 nmol/cm² on the surface of poly(ethylene terephthalate) (PET) film and demonstrated its bactericidal effect against Escherichia coli. Jansen et al. introduced silver ions by plasma-induced grafting onto polyurethane films which was found to reduce adherent viable bacteria from initial 10⁴ cells/cm² to zero within 48 h [43]. Jiang et al. [44] coated silver on silicon rubber substrates and showed decline in number of L. monocytogenes cells post 6 h. After 12 h, there was a reduction of over 2-log10 CFU/chip, and no viable bacteria were detected on both types of silver-coated SR after 18 and 24 h. Sambhy et al. [21] demonstrated antibacterial activity of composites consisting of poly(4-vinyl-N-hexylpyridinium bromide) (NPVP) embedded with silver bromide nanoparticles. They observed no biofilm formation on 1:1 AgBr/21% NPVP-coated surfaces after 72 h when incubated for 24-72 h with P. aeruginosa suspension (107 CFU/mL) in LB broth. Pant et al. [45] have demonstrated the ability to eliminate up to 99.9% of pathogenic bacteria on the surface of siloxane epoxy system containing quaternary ammonium moieties. In another work involving epoxy system, Perk et al. [46] observed fungicide, carbendazim supported on poly (ethylene-co-vinyl alcohol) and epoxy resin coating showed the antifungal activity contingent upon release from their polymer supports.

Coatings and thin films based on titania photoctalysts (Ag⁺-doped TiO₂/Ag-TiO₂/TiO₂) that kills microbes under UV and visible light illumination, also have been actively investigated in recent years. Studies by Necula et al. [47], with TiO₂-Ag composite coating prepared by plasma electrolytic oxidation on implantable titanium substrate, showed the ability to completely kill methicillin-resistant S. aureus (MRSA) within 24 h. In yet another investigation by Necula et al. [25], they examined the ion release and antibacterial activity of porous TiO2-Ag coating on biomedical alloy disk. Each evaluated samples could release 20.82 and 127.75 µg of Ag⁺ per disk and showed markedly enhanced killing of the MRSA inoculums with 98% and >99.75% respectively within 24 h of incubation, while their silver free counterpart sample allowed the bacteria to grow up to 1000-fold. The non-cumulative release of silver ions of 0.4 ppm, 0.26 ppm and 0.005 ppm for 1 h, 24 h and 7 days respectively after immersion in water, from nanometer scale Ag-TiO₂ composite film was demonstrated by Yu et al. [34] and they also reported that 0.4 ppm released silver from Ag-TiO₂ composite film is sufficient to cause almost 100% killing of E. coli when exposed to UV for 1 h. Studies by Jamuna-Thevi et al. [48], reported nanostuctured Ag⁺ doped TiO₂ coatings deposited by RF magnetron on stainless steel, with overall Ag⁺ ion release measured between 0.45 and 122 ppb. They also noted that at least 95 ppb Ag⁺ ion released in buffered saline was sufficient for 99.9% of reduction against S. aureus after 24 h of incubation. Biological activity

of silver-incorporated bioactive glass studies conducted by Balamurugan et al. [49] assessed in vitro antibacterial bioactive glass system elicited a rapid bactericidal action. Antimicrobial efficacy of these silver-incorporated bioglass suspension at 1 mg/mL for E. coli was estimated to be >99% killing, and the amount of Ag⁺ released from silver-incorporated glass was up to 0.04 mM after 24 h. In yet another study involving silver ions release by Liu et al. [35], the amount of silver released form the mesoporous TiO₂ and Ag/TiO₂ composites was measured to be 1.6×10^{-8} mol after 20 days. The photo-bactericidal activity on composite films was extremely high and displayed bactericidal activity even in the dark; they further reported that the survival rate was only 9.2% in the dark, and the E. coli cells were totally killed in UV light. Sun et al. [50] reported killing of bacteria on Ag-TiO2 thin film, even in the absence of UV irradiation against S. aureus and E. coli with significant antibacterial rate about 91% and 99% after 24 h respectively due to release of silver, and the concentration of silver ions released from the Ag-TiO₂ film was 0.118 µg/mL during 192 h. Akhavan [51], reported that a concentration of 2.8 to 2.5 nM/mL completely killed 10⁷ CFU/mL E. coli with visible light response photocatalytic Ag-TiO₂/Ag/a-TiO₂ material in 110 min. However, in most of the cases reports are based on planktonic studies and the release of silver is dependent upon the method employed for coating, thickness, conditions for gradient formation and silver source used. Nevertheless, release of silver ions frombare Ag/TiO2 composite layers reported above, obtained by methods viz., impregnation, deposition and nano-coatings gradually diminish over the time.

Bacterial biofilms are often more difficult to eradicate unlike planktonic cells. Until now, there have been very few reports that shown to resist biofilm formation on titania based polymer-nanocomposites. In one such study, Kubacka *et al.* [52] have demonstrated photocatalysis using ethylene-vinyl alcohol copolymer (EVOH) embedded with Ag-TiO₂ nanoparticles (*ca.* 10^{-2} wt%) that showed outstanding resistance to biofilm formation by bacteria and yeast, upon ultraviolet (UV) light activation. In the present study, although the release kinetics of silver was not established but comparing to above studies which established the antimicribial threshold concentration of silver and efficacy of killing with different bare Ag-TiO₂ (Ag/Ag-TiO₂ nanofilms), the polymer composite system reported here which released 6.4 to 16.8 µg/mL of silver seems adequate [53], when the overall biocidal ability (to prevent bacterial attachment) of the composite during 48 h period in combination with radical-mediated photocatalytic action. Practically, the added strengths of the polymer-based Ag-TiO₂ nanocomposite coatings as compared to bare TiO₂/Ag-TiO₂ coatings are its wear stability, flexibility, permeability and optical properties.

But the main objective of the disinfection technology in ensuring microbiological safety is to; set a standard for achieving a required logarithm of reduction of the microbial consortia. The microbial cells, which are not inactivated by the antimicrobial coatings adhering onto the testing surface over the different irradiation time, were able to grow on the agar plates. Quantifying their reduction in number (for quantitative assessment) of surviving CFU on a bactericidal surface compared with a non-bactericidal (neat epoxy) surface revealed reduction of microbial cells. In the present study, epoxy/Ag-TiO₂ with 1.0 wt% loading was found to cause a reduction of CFU on agar plates by approximately 6-log in case of *E. coli* and the same effected *ca.* 4-log reduction in case of *S. aureus* after 48 h of incubation, while epoxy/TiO₂ with 1.0 wt% loading exhibited lesser inhibition of biofilm formation, see Table 2.

There was an initial slower decrease in bacterial load by all the composites, *i.e.*, below 1-log reduction observed up to 18 h exposure followed by a rapid microbial decrease up to 6-log in 48 h for both

1.0 wt% of TiO₂ and Ag-TiO₂ loaded epoxy composites. Incomplete inhibition of biofilm formation was observed with lesser Ag-TiO₂ loading, but complete inhibition of both *E. coli* and *S. aureus* was possible for composites with above 1.5 wt% of Ag-TiO₂ after 24 h with UV irradiation. Strikingly, for the composite coating with 2.0 wt% epoxy/Ag-TiO₂ showed highest antibiofilm effectiveness with 1-log reduction in 18 h, *i.e.*, the shortest period with maximum inhibition. In addition, after 48 h of irradiation against both *S. aureus* and *E. coli* with very few surviving CFUs and complete inhibition (biofilm formation) and 7-log reduction was observed, relative to that in control plates as shown in Table 2. However, the present study results take into consideration only biofilm phase inhibition, and the obtained concentration levels) to radically prevent microbial cell viability. The polymer-based nanocomposite reported here obtained by dispersion of the Ag-TiO₂ nanoparticles into epoxy manifest a real potential as photobiocidal coatings in a wide variety of settings that prevents biofilm formation by a wide range of Gram-positive and Gram-negative bacteria.

Table 2. Different nanocomposite materials and their antibiofilm efficacy for 18 and 48 h irradiation time.

	E. co	li (G ^{-ve})	S. aureus (G ^{+ve})					
Composite type –	% Inhibition ^a	Log ₁₀ Reduction ^b	% Inhibition ^a	Log ₁₀ Reduction ^b				
% Biofilm inhibition and Log CFU reduction after 18 h								
1wt% Epoxy/TiO ₂	57.2 (±1.5)	<1.0 (±0.03)	46.0 (±1.4)	<1.0 (±0.02)				
1wt% Epoxy/AgTiO ₂	77.0 (±1.4)	<1.0 (±0.02)	68.5 (±2.0)	<1.0 (±0.02)				
2wt% Epoxy/AgTiO ₂	90.0 (±1.3)	1.0 (±0.2)	90.0 (±1.4)	1.0 (±0.03)				
% Biofilm inhibition and Log CFU reduction after 48 h								
1wt% Epoxy/TiO ₂	90.0 (±1.0)	1.0 (±0.2)	63 (±0.9)	1.0 (±0.2)				
1wt% Epoxy/AgTiO ₂	100	6.0 (±0.18)	99.9 (±0.1)	4.0 (±0.11)				
2wt% Epoxy/AgTiO ₂	100	7.0 (±0.19)	100	7.0 (±0.2)				

^a Percent reduction in biofilm formation as determined by Crystal Violet assay; ^b Mean value \pm SD for the group Log₁₀ reduction in CFU/plate.

4. Conclusions

The investigation relates the preparation of antibiofilm composite coatings containing both photocatalytic non-leaching Ag-doped TiO₂ and leaching silver biocide for production of potent oxidants (ROS) and silver species at the surface. The antimicrobial activity of these composite surfaces was quantified based on the inhibition of biofilm formation using crystal violet assay, which can be adopted more conveniently in high-throughput experiments. These antimicrobial materials are capable of killing microorganisms upon contact by inhibiting the biofilm formation in the aqueous environments. Both epoxy/TiO₂ and epoxy/Ag-TiO₂ nanocomposites exposed to UV irradiation exhibited antibiofilm activity against *S. aureus* (Gram-positive) and *E. coli* (Gram-negative). Although the optimal antimicrobial conditions remain to be fully established, the results highlight a better antibiofilm activity of Epoxy/Ag-TiO₂ compared to Epoxy/TiO₂. The role of different silver species could be that Ag⁺ as an active species found to enhance the catalytic activity, in contrast, Ag⁰ species showing strong antibacterial activity. This material may find potential applications in designing self-disinfecting surfaces, especially for hospitals and food industries where hygiene is a high priority.

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Author Contributions

S.S.M. prepared and charecterised the materials, performed the experiments, analyzed the data and designed the structure of manuscript. N.K. gave technical advice and reviewed the manuscript.

Conflicts of Interest

The authors declare no conflict of interest.

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