

## Article

# Thermal Cycling Behavior of Air Plasma-Sprayed and Low-Pressure Plasma-Sprayed Environmental Barrier Coatings

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**Abstract:**  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  environmental barrier coatings (EBCs) were produced by air plasma spray (APS) and low-pressure plasma spray (LPPS) processes. The phase composition, microstructure, and bonding strength of APS and LPPS EBCs were investigated. Thermal cycling tests were performed in air and in steam atmosphere respectively at 1316 °C for both APS and LPPS EBCs. There is no coating failure in air atmosphere for both APS and LPPS EBCs after 900 cycles. In contrast, APS EBCs have an average life of 576 cycles in a steam cycling test in 90%  $\text{H}_2\text{O}$  + 10% air at 1316 °C while LPPS EBCs survived 1000 cycles without failure. The superior durability of the LPPS EBCs compared to APS EBCs in the same steam cycling environment is attributed to the significantly reduced thermally grown oxide (TGO) growth rate because of the denser and crack-free microstructure, higher bonding strength, and reduced coefficient of thermal expansion (CTE) mismatch (less  $\text{Yb}_2\text{SiO}_5$  phase) in the LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  EBCs.



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**Keywords:**  $\text{Yb}_2\text{Si}_2\text{O}_7$ ; environmental barrier coatings; air plasma spray; low-pressure plasma spray; thermal cycling; thermally grown oxide (TGO)

## 1. Introduction

There is always a continuous demand for higher operating temperatures for gas turbine engines in order to improve fuel efficiency and engine thrust. The temperature capability of current Ni-base superalloys has reached their limit [1]. Silicon carbide ceramic matrix composites (SiC/SiC CMCs) have been considered as the material for next-generation aircraft engine components due to their excellent high-temperature strength and lower specific weight [1–3]. Due to the high heat resistance, CMCs components require less air cooling and therefore enable engines to run hotter and more efficiently at higher thrust. Recently, silicon carbide-based CMCs have been successfully implemented in hot section components of LEAP and GE<sub>9x</sub> aircraft engines [3]. One key issue with SiC/SiC CMCs is that they suffer from recession in steam environments due to the volatilization of silica scale by water vapor, and so CMCs components must be protected with environmental barrier coatings (EBCs) to prevent recession in engine operating environments. Current EBCs consist of a silicon bond coat and oxide based upper layers such as mullite, barium–strontium–aluminum–silicates (BSAS), and rare earth silicates [2,4–6]. Among them,  $\text{Yb}_2\text{Si}_2\text{O}_7$  disilicate is the leading candidate for environmental barrier coatings (EBCs) for SiC-based CMCs due to its high-temperature phase stability and excellent thermal expansion match with SiC [5–8].

The air plasma spray process (APS) has been widely used for the deposition of EBCs due to its relatively low cost and high deposition rates [2,5,7,9]. EBCs coatings such as mullite [4], rare earth silicates [6,9], and BSAS ( $\text{BaO}\cdot\text{SrO}\cdot\text{Al}_2\text{O}_3\cdot 2\text{SiO}_2$ ) [10] have been deposited using the APS process. In contrast to the coating deposition in air atmosphere, a plasma spray torch can also be operated in a controlled and low-pressure atmosphere. The reduced pressure compared to atmospheric conditions enlarges the plasma plume from

approximately 50 to 500 mm in length and from 10 to 40 mm in diameter. A larger plume results in a bigger spray spot. Therefore, particle velocities and temperatures are more homogeneously distributed over the cross-section of the plume. This allows for coatings with a homogenous coating microstructure, even on parts with complex geometries—such as engine blades and vanes. In addition, due to the reduced pressure, the plasma gas velocities as well as the resultant particle velocities are much higher than those in APS. Higher particle velocities at impact are favorable to produce denser coatings, which is a critical property for hermetic EBCs. Therefore, it is of significant value to investigate and compare the properties of EBCs deposited using APS and LPPS processes.

In our previous work,  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  EBCs had been deposited using the LPPS process [11]. The isothermal oxidation kinetics of LPPS EBCs in a steam environment was also evaluated. However, the thermal cycling behavior of LPPS EBCs in steam has not been investigated.

In this study,  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  EBCs were produced by APS and LPPS processes. The phase composition, microstructure, and bonding strength of APS and LPPS EBCs were investigated. The thermal cycling life as well as TGO growth behavior of these coatings in air and in water vapor conditions were also evaluated.

## 2. Experimental Procedures

Metco 4810 Si powder with a size distribution of  $-75/+15\ \mu\text{m}$  was used for bond coat deposition and Metco 6157  $\text{Yb}_2\text{Si}_2\text{O}_7$  powder ( $\approx 3\ \text{vol.}\%$   $\text{Yb}_2\text{SiO}_5$  minor phase) with a size distribution of  $-90/+11\ \mu\text{m}$  was used for top coat deposition. Both of them are produced by Oerlikon Metco (Westbury, NY, USA). The  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  EBCs were deposited onto  $\phi 25.4\ \text{mm} \times 3\ \text{mm}$  Hexoloy™  $\mu\text{-SiC}$  substrates (Saint Gobain Ceramics, Niagara Falls, NY, USA) using APS and LPPS process, respectively. Before coating deposition, SiC substrate was grit basted using 24 mesh SiC grits. For APS EBCs, the coatings were sprayed using the SinplexPro plasma torch. For LPPS EBCs, the coatings were sprayed using an O3CP torch in a 30 mbar chamber pressure. Both APS and LPPS EBC consist of a  $\approx 200\ \mu\text{m}$  thick top coat and  $\approx 175\ \mu\text{m}$  thick Si bond coat. Only one side of the SiC substrate was coated.

All of the as-sprayed coatings were further annealed at  $1300\ ^\circ\text{C}$  for 10 h under ambient conditions. X-ray diffraction (XRD, Ultima IV, Rigaku Corporation, Tokyo, Japan) analysis was carried out on the as-sprayed and annealed coatings. The relative phase composition of the annealed coating was analyzed semi-quantitatively by comparing the crystalline peak intensities [12]. Standard metallographic practices were used to polish the coatings cross-sections, and their microstructures were characterized using scanning electron microscopy (SEM, Hitachi S-3400N, Hitachi America, Ltd., Tarrytown, NY, USA). The coating porosity was measured on the cross-sections of as-prepared coatings using the image analysis method. Coating bonding strength was measured according to ASTM C633 standard test method.

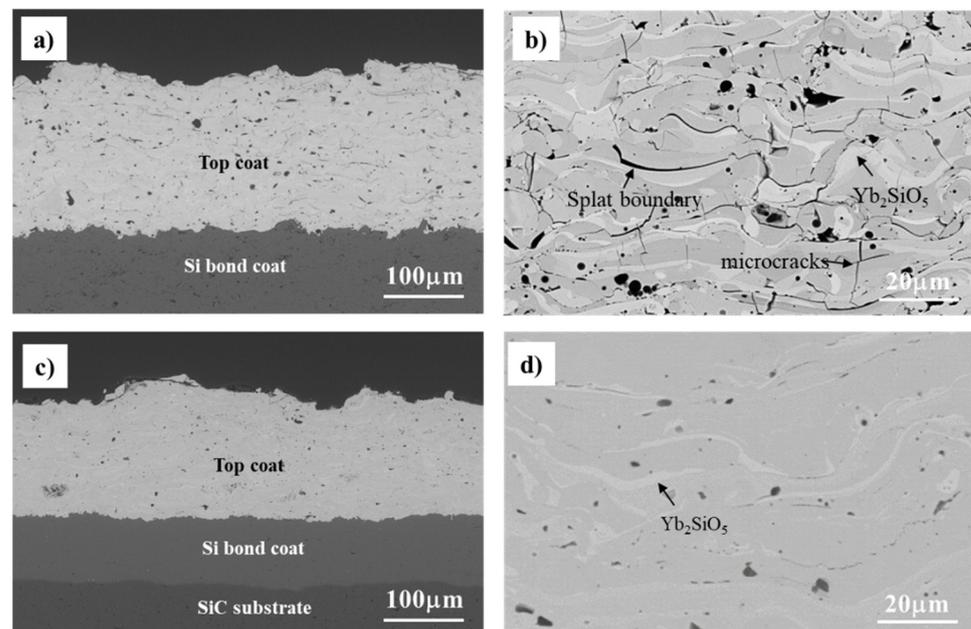
The EBCs thermal cycle durability was evaluated in air atmosphere and steam atmosphere at  $1316\ ^\circ\text{C}$ , respectively. A thermal cycling test in the ambient conditions was performed at  $1316\ ^\circ\text{C}$  using a CM™ rapid temperature furnace (CM Bloomfield, Bloomfield, NJ, USA). The cycling conditions consisted of three steps: a 10-min heating up to test temperature of  $1316\ ^\circ\text{C}$ , a 40 min hold at the test temperature, and a 10 min cooling that brought the substrate temperature to around  $100\ ^\circ\text{C}$ . Please note that the cycle start and end temperatures were not room temperature, to maintain the cycling time. The conditions for a similar test in steam were different. The thermal incursion had a 5 min heat-up to the test temperature of  $1316\ ^\circ\text{C}$ , a 60 min hold at the test temperature, and a 15 min air cooling until the substrate temperature reached  $\sim 100\ ^\circ\text{C}$ . In both the air and steam furnace cycle testing, the samples were pulled out of the furnace hot zones to allow them to cool under air environment. To create steam environment, a controlled mixture of water vapor and air is introduced to the furnace. The steam consisted of  $\approx 90\ \text{vol.}\%$   $\text{H}_2\text{O}$  (g) +  $\approx 10\ \text{vol.}\%$  air. The gas velocity is  $\approx 4\ \text{cm/s}$  at the hot zone of the furnace chamber. A total of eight samples were cycled for each system. One sample was removed after

200, 400, 600, and 800 cycles for TGO growth rate evaluation. A total of 1000 cycles were performed. The coating was considered to be at failure when spallation was observed. For these thermal cycled samples, in order to avoid damage during metallography sample preparation, coatings were sectioned with a slow diamond saw cutting speed of 0.01 mm/s. Then, the cut samples were mounted and polished for SEM characterization.

### 3. Results and Discussion

#### 3.1. Microstructure and Phase Composition

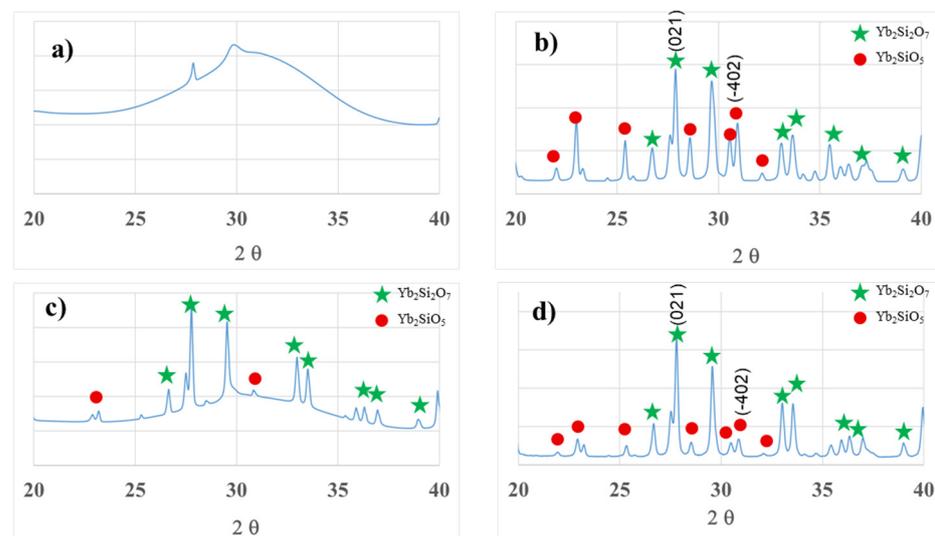
The cross-sections of APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  coating microstructures are shown in Figure 1. Both the Si bond coat and  $\text{Yb}_2\text{Si}_2\text{O}_7$  top coat exhibit dense microstructures with measured porosity less than 5%. High magnification analysis shows that the APS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coating contains some microcracks and splat boundaries (Figure 1b). In contrast, there are no microcracks observed in the LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coatings (Figure 1d). The LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coating is  $\approx 2\%$  denser than the APS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coating. A previous study indicated that the particle velocity in the LPPS process is about four to six times faster than the APS process [13]. The high particle velocity during impingement on the substrate will result in better adhesion between molten splats, which results in a denser coating in the LPPS process. Two phase microstructures, with dark ( $\text{Yb}_2\text{Si}_2\text{O}_7$ ) and bright ( $\text{Yb}_2\text{SiO}_5$ ) appearances, are evident from the contrast difference in backscatter electrons mode of SEM images. These two phases can be found in both APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coatings. The amount of bright phase ( $\text{Yb}_2\text{SiO}_5$ ) in the LPPS coating is much less than that in the APS coating.



**Figure 1.** Typical annealed coating microstructures at low and high magnifications. (a,b) APS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coating; (c,d) LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coating.

Figure 2 shows the XRD patterns of as-sprayed and annealed APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coatings. The as-sprayed APS coating is mainly composed of amorphous phases (Figure 2a). In contrast, the LPPS coating is composed of a mixture of amorphous phases, crystalline  $\text{Yb}_2\text{Si}_2\text{O}_7$  phase, as well as a small amount of crystalline  $\text{Yb}_2\text{SiO}_5$  phase (Figure 2c). The phase difference can be attributed to the coating deposition temperature difference in APS and LPPS processes. In the LPPS process, the coating surface temperature is  $\approx 900$  °C (recorded by pyrometer), which is significantly higher than that in the APS process ( $\approx 100$  °C). The higher deposition temperature promotes some degree of in situ crystallization of solidifying particles. After 1300 °C/10 h heat treatment, the amorphous phase

transforms to crystalline  $\text{Yb}_2\text{Si}_2\text{O}_7$  and  $\text{Yb}_2\text{SiO}_5$  phases in both APS and LPPS coatings (Figure 2b,d). The appearance of the  $\text{Yb}_2\text{SiO}_5$  phase in both APS and LPPS coatings can be attributed to  $\text{SiO}_2$  evaporation loss from the molten particle in the high-temperature plasma plume.



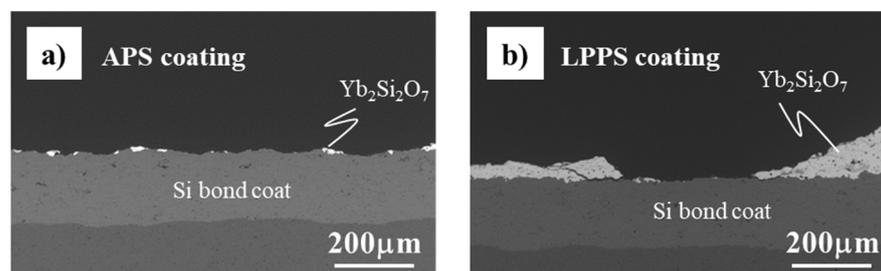
**Figure 2.** XRD patterns of APS and LPPS coatings. (a) As-sprayed APS coating; (b) 1300 °C/10 h heat treated APS coating; (c) as-sprayed LPPS coating; (d) 1300 °C/10 h heat treated LPPS coating.

The relative phase content of  $\text{Yb}_2\text{SiO}_5$  and  $\text{Yb}_2\text{Si}_2\text{O}_7$  in the annealed coatings was estimated by the intensity ratio of the  $(-402)$  and  $(021)$  primary diffraction peaks corresponding to  $\text{Yb}_2\text{SiO}_5$  and  $\text{Yb}_2\text{Si}_2\text{O}_7$  phases, respectively [12]. The APS coating contains about 32.8 vol.%  $\text{Yb}_2\text{SiO}_5$  and 67.2 vol.%  $\text{Yb}_2\text{Si}_2\text{O}_7$  phases, while the LPPS coating contains 15.5 vol.%  $\text{Yb}_2\text{SiO}_5$  and 84.5 vol.%  $\text{Yb}_2\text{Si}_2\text{O}_7$  phases. Some of the  $\text{Yb}_2\text{SiO}_5$  phases in both the APS and LPPS coatings are likely from the feedstock powders which contain  $\approx 3$  vol.%  $\text{Yb}_2\text{SiO}_5$ . The significant difference in the amount of  $\text{Yb}_2\text{SiO}_5$  phase can be attributed to the very different particle melting states (e.g., particle temperature and velocity) in APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7$  coating deposition. Since the particle velocity in the APS process is four to six times slower compared to that in the LPPS process [13], it is estimated that the particle dwelling time in the APS process is  $\approx 1.5$  times that in the LPPS process. The longer dwelling time will result in more  $\text{SiO}_2$  evaporation in the APS process.

### 3.2. Bonding Strength

The bonding strength of APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  coatings were measured. The APS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  coating has a bonding strength of  $17.6 \pm 1.6$  MPa, while the LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  coating has a bonding strength of  $29.1 \pm 0.75$  MPa. The bonding strength of LPPS EBCs is about 65% higher than that of APS EBCs. The cross-section microstructures after the bonding strength test are shown in Figure 3. Both APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  coatings primarily failed at the interface between the top coat and bond coat, with the observation that LPPS coatings had some degree of cohesive failure, while APS coatings were more similar to classic bond coat–top coat failure. In both the cases, SEM observations revealed that the Si bond coat has a strong adherence to the substrate. It had been reported [14] the splat–substrate and inter-splat contact is dramatically improved at higher deposition temperatures, leading to reduced porosity and increased adhesion strength. The deposition temperature in the LPPS process ( $\approx 900$  °C) is significantly higher than that in the APS process ( $\approx 100$  °C); thus, it can be expected that the inter-splat bonding as well as the splat–silicon bonding will be improved in LPPS EBCs compared to that in APS EBCs. In addition, the higher kinetic energy due to the higher particle velocity in LPPS EBCs will

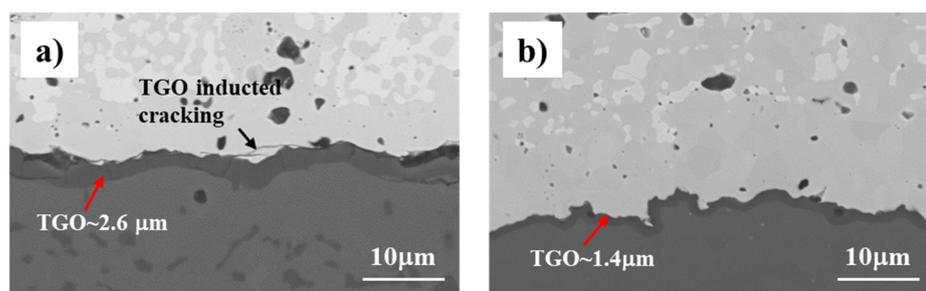
further enhance the adhesion between splats and the silicon bond coat. Therefore, a higher bonding strength was observed in the LPPS EBCs.



**Figure 3.** Cross-section microstructures after bonding strength test for (a) APS and (b) LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  EBCs.

### 3.3. Thermal Cycle Behavior in Air

A furnace cycle test (FCT) in air at 1316 °C was performed, and the test ran for 900 cycles ( $\approx 600$  h hot time at 1316 °C). There were no coatings spallation/failure for both APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  EBCs after 900 cycles. Both APS and LPPS  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  EBCs showed excellent thermal cycle behavior in air atmosphere. The coating cross-section microstructures after 900 FCT cycles are shown in Figure 4. A thin layer of thermally grown oxides (TGO) formed at the  $\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  interface. The average TGO thickness for APS EBCs is about 2.6  $\mu\text{m}$ , while it is about 1.4  $\mu\text{m}$  for LPPS EBCs. In the APS EBCs, horizontal microcracks are observed between the TGO and top coat, and vertical microcracks are also shown in the TGO layer. In contrast, there are no microcracks observed at the top coat and TGO interface in the LPPS EBCs. The LPPS top coat continued to adhere very well to the TGO/Si bond coat.

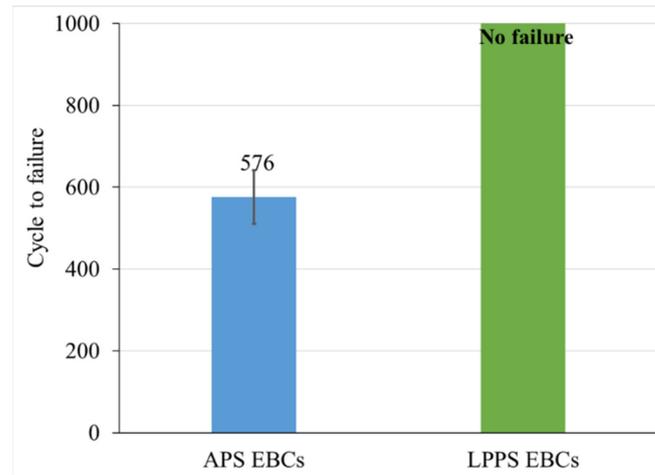


**Figure 4.** Coating microstructures after 900 FCT cycle tests in air. (a) APS EBCs; (b) LPPS EBCs.

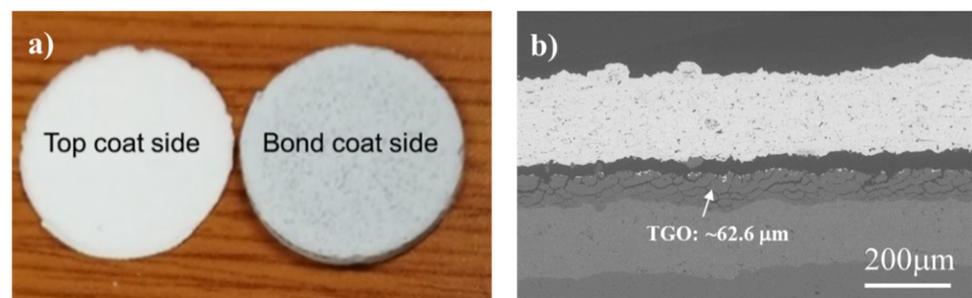
### 3.4. Thermal Cycle Life in Steam

Both APS and LPPS EBCs were also thermally cycled in 90 vol.%  $\text{H}_2\text{O}$  + 10 vol.% air environment at 1316 °C. APS EBCs had an average lifetime of  $576 \pm 65$  cycles, while there was no failure for LPPS EBCs after 1000 cycles (Figure 5). A typically failed APS EBC coating after 600 cycles in steam was shown in Figure 6. The coating was failed by delamination of the entire ceramic top coat (as a one piece) from the substrate, and the failure occurred at the top coat/TGO interface. In SEM characterization, a fish-scale-like cracking TGO layer with a thickness of  $\approx 63$   $\mu\text{m}$  is observed in the APS EBCs after 600 steam cycles (arrow indicated in the Figure 6b). Compared to the TGO thickness in air atmosphere (Figure 4a), it is obvious that the TGO layer of APS EBCs in steam is much thicker than that in air with a similar exposure duration at 1316 °C. The results indicated that TGO growth for APS EBC is accelerated under the steam cycling conditions. It has been demonstrated that water vapor is the primary oxidant for Si oxidation in a steam environment [6]. Deal and Grove [15] have shown that the oxidation rate of silicon was much higher in  $\text{H}_2\text{O}$  vapor than in dry  $\text{O}_2$ . Therefore, the accelerated TGO growth of a Si bond coat was expected

in steam cycling for the APS EBCs. The rapid TGO growth and failure of APS EBCs in a steam environment compared to what was observed in air atmosphere clearly indicated that the thermal cycling of EBCs in air could result in false positive results.

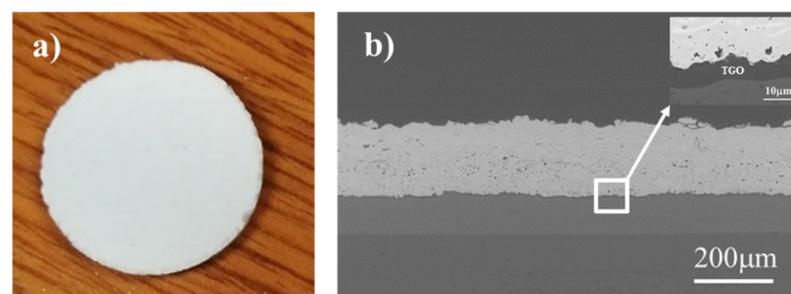


**Figure 5.** Thermal cycling life of APS and LPPS EBCs in steam environment at 1316 °C. APS EBCs have an average life of  $576 \pm 65$  cycles, while LPPS EBCs survived 1000 cycles without failure.



**Figure 6.** Photograph (a) and SEM image (b) of failed APS EBC after 600 steam cycling.

In contrast, the LPPS coatings remained adherent after 1000 cycles test in steam, and the SEM cross-section shows a good bonding between the top coat and bond coat (Figure 7). Only a thin TGO layer was formed (insert in Figure 7b). This points to the fact that LPPS EBCs have superior durability over APS EBCs in steam cycling.

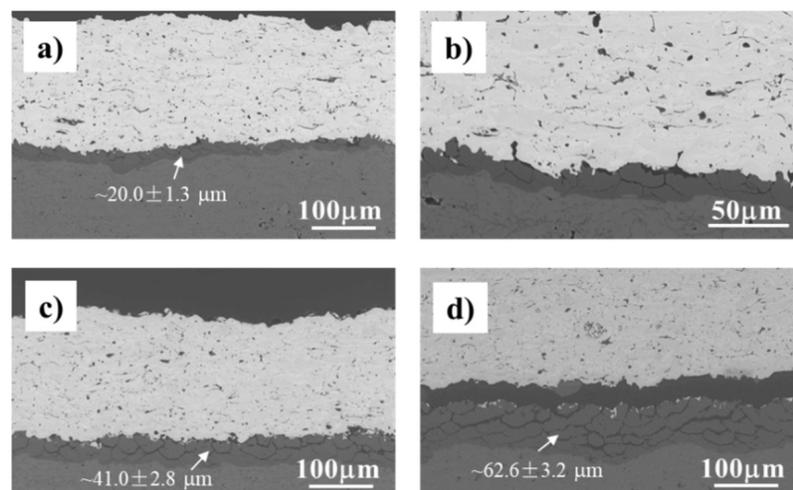


**Figure 7.** Photograph (a) and SEM image (b) of LPPS EBC after 1000 steam cycles (insert showing the high magnification of the TGO layer).

### 3.5. TGO Growth Rate of APS and LPPS EBCs in Steam

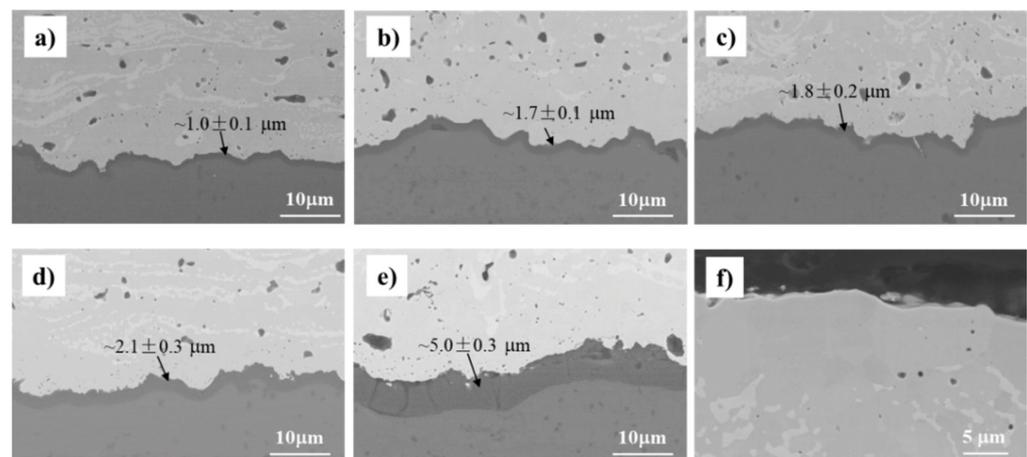
The coating cross-section microstructures after various cycles at 1316 °C in 90% H<sub>2</sub>O (g) + 10% air for APS and LPPS EBCs are shown in Figures 8 and 9, respectively. A fish-scale-like cracking TGO layer is observed even after only 200 cycles in APS EBCs

(Figure 8a,b). Microcracks/gaps between TGO and the top coat are also observed. These microcracks/gaps may be caused by thermal stress due to the coefficient of thermal expansion (CTE) mismatch between the top coat and TGO/Si bond coat during thermal cycling. The average TGO thickness is  $\approx 20 \mu\text{m}$  after 200 cycles, and it grows to  $\approx 41 \mu\text{m}$  after 400 cycles and reached  $\approx 63 \mu\text{m}$  after 600 cycles at failure. A linear TGO growth trend with respect to the number of steam cycles is revealed. XRD analysis on surface of the failed sample indicated that the fish-scale-like TGO layer was mostly composed of an  $\alpha$  cristobalite phase. Richards et al. also confirmed the  $\text{SiO}_2$  TGO as  $\alpha$ -cristobalite phase via Raman analysis in a steam cycling test of plasma-sprayed  $\text{Si}/\text{Yb}_2\text{Si}_2\text{O}_7$  EBC [4]. It has been established that the oxide growth involves the formation of a thermodynamically stable  $\beta$ -cristobalite ( $\text{SiO}_2$ ) structure at high temperature, and as the test conditions change to a cooling step, a phase transformation (from  $\beta$ -to- $\alpha$ ) of this phase occurs at around  $220^\circ\text{C}$ , which is calculated to cause  $\approx 4.9\%$  volume change [16]. This volume change introduces severe cracking in the TGO layer, as shown in Figure 8, which provides a short diffusion path to the environmental oxidants such as oxygen and water vapor. The continuous exposure of the coating to the test conditions (high temperature and steam) enables further interaction of the Si bond coat and the oxidants, causing an accelerated oxidation of the bond coat. The successive cycling of the EBCs continues to introduce further cracking and leads to an eventual failure.



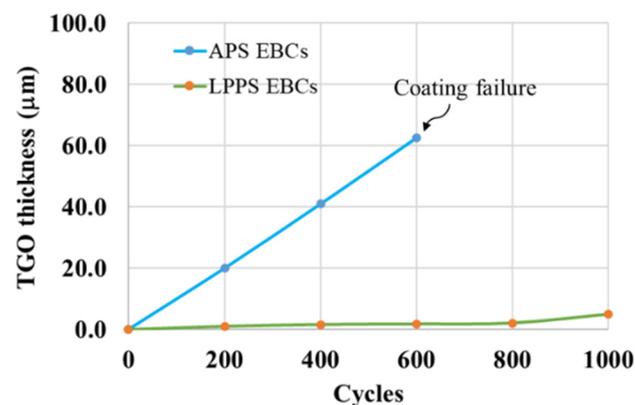
**Figure 8.** SEM images of APS EBCs in steam cycling showing the TGO growth. (a,b) 200 cycles; (c) 400 cycles; (d) 600 cycles.

For LPPS EBCs, a thin and dense  $\text{SiO}_2$  TGO layer is observed for all samples at various numbers of cycles (Figure 9). The average TGO thickness is  $\approx 1.0 \mu\text{m}$  (against  $\approx 20 \mu\text{m}$  for APS EBC) after 200 cycles, and it gradually grows to  $\approx 2.1 \mu\text{m}$  after 800 cycles and then starts to grow rapidly to  $\approx 5.0 \mu\text{m}$  at 1000 cycles. Vertical cracks through the full TGO thickness are observed in the sample after 1000 cycles but are not shown in other samples. The cracking of the TGO as its thickness increased was attributed to the increased strain energy release rate when the TGO thickness exceeds a certain value (7). The accelerated TGO growth from 800 cycles to 1000 cycles can be attributed to the appearance of vertical cracks in the TGO, which provide a fast diffusion path for oxidants. The top coat microstructure after 1000 steam cycles is shown in Figure 9f. No recession is observed on the coating surface, indicating the excellent steam recession resistance of the LPPS EBCs.



**Figure 9.** Microstructures of LPPS EBCs for various steam cycles. (a) 200 cycles; (b) 400 cycles; (c) 600 cycles; (d) 800 cycles; (e) 1000 cycles; (f) Top coat cross-section microstructure after 1000 cycles at 1316 °C in 90% H<sub>2</sub>O (g) + 10% air.

TGO thickness as a function of cycles for APS and LPPS EBCs at 1316 °C in 90% H<sub>2</sub>O-10% air is plotted in Figure 10. The TGO growth rate in APS EBCs is over 20 times higher than that in LPPS EBCs. Possible reasons for the fast TGO growth rate in the present APS EBCs will be discussed in Section 3.6.



**Figure 10.** TGO thickness as a function of cycles for APS and LPPS EBCs at 1316 °C in 90% H<sub>2</sub>O-10% air.

### 3.6. Failure Mechanism

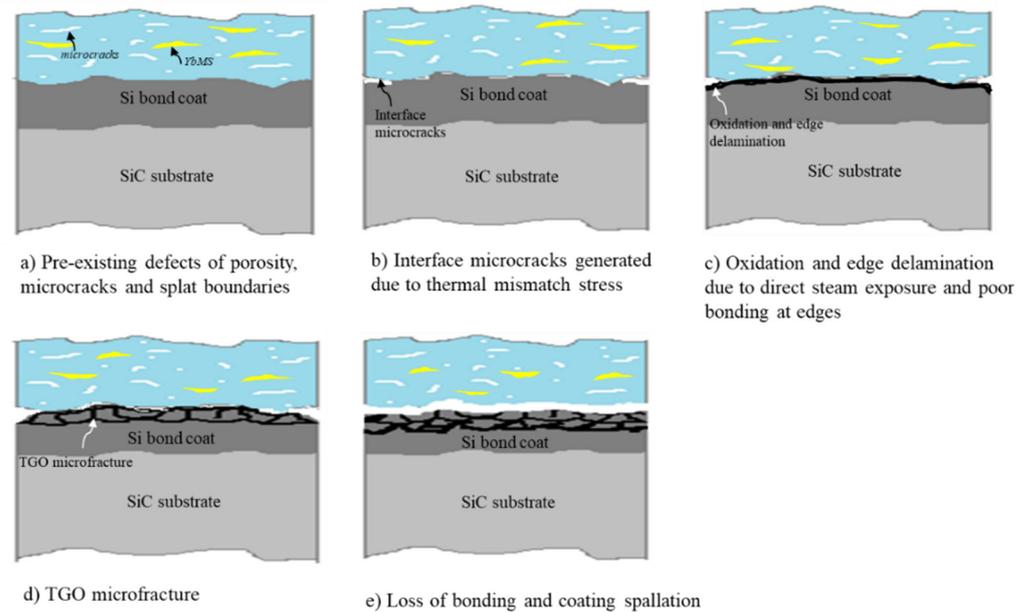
The failure mechanism of various EBCs systems in a steam environment had been investigated. Kimmel et al. [10] evaluated the BSAS/mullite + BSAS/Si EBC on SiC/SiC combustion liners in a gas turbine engine. The TGO layer with  $\approx 60$ – $100 \mu\text{m}$  thickness was formed in the inner liner after the 13,937 h field test at  $\approx 1200 \text{ }^\circ\text{C}$ . Localized EBCs spallation was observed in the area with a TGO thickness over  $\approx 200 \mu\text{m}$ . The thick TGO layer raised the middle and top layers, thus causing the EBCs coating to buckle/delamination. Richards et al. [4] examined the thermal cycling properties of APS Yb<sub>2</sub>SiO<sub>5</sub>/mullite/Si tri-layer EBCs on a SiC substrate at 1316 °C in 90% H<sub>2</sub>O-10% O<sub>2</sub>. It was found that partial delamination occurred with the fracture plane located within a thermally grown oxide (TGO) at the mullite/Si interface. The formation of a cristobalite TGO was a major contributing factor to the poor steam cycling durability of this tri-layer EBC system. Richards et al. [7] studied thermal cycling of APS Yb<sub>2</sub>Si<sub>2</sub>O<sub>7</sub>/Si EBCs in water vapor and found that the delamination cracking initiated from the EBCs sample edge where the TGO was first formed due to the direct exposure of the Si bond coat to water vapor. A laboratory steam oxidation test by Lee [6] also showed that the EBCs failed at TGO when TGO reaches

a critical thickness. A common feature of the above-mentioned engine and laboratory rig tests indicated that the formation of TGO played a major role for the failure of EBCs.

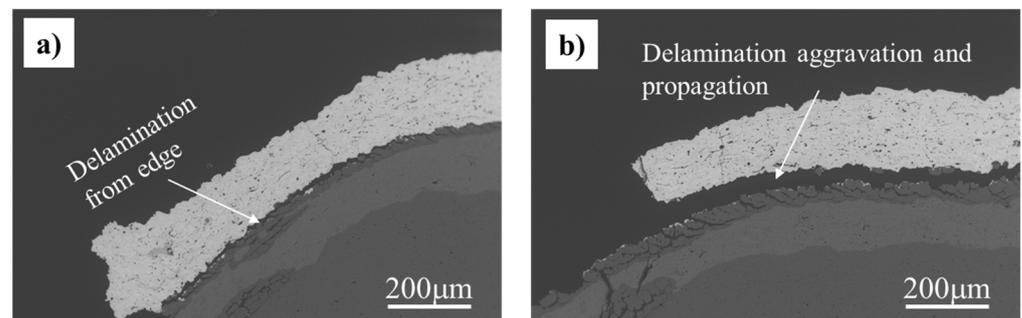
In the present study, a much shorter life and faster TGO growth had been observed in APS EBCs compared to the LPPS EBCs under the same steam cycling condition. The major driving forces for TGO growth and resultant short life in APS EBCs could be attributed to the following factors. The corresponding coating damage evolutions are schematically shown in Figure 11.

1. Pre-existing EBC defects such as porosity, microcracks, and splat boundaries in APS EBCs (Figure 1b) provided a diffusion path for oxidants ( $H_2O/O_2$ ) to reach the surface of the Si bond coat and resulted in the oxidation of silicon.
2. High volume percentage of the larger CTE  $Yb_2SiO_5$  phase in the APS  $Yb_2Si_2O_7$  EBCs will cause larger thermal mismatch stress during thermal cycling. The higher thermal stress coupled with the lower adhesion strength between the top coat and bond coat may generate horizontal microcracks at the interface between the top coat and Si bond coat at the earliest few cycles (Figure 11b). The interface microcracks provide a fast diffusion path for oxidant gas species.
3. Off-angle spray at the sample edge in the APS process will generally result in a poor coating bonding. Delamination will likely occur first from the coating edges during thermal cycling due to the weak bonding strength at the edges. Oxidation will progress inwards along delaminated edges (Figure 11c). In the experiment, the coating delamination from edges was observed after 200 cycles (Figure 12a). The delamination gets aggravated after 400 cycles (Figure 12b). Similar edge delamination has also been observed by Richards et al. [7].
4. Oxidation of the Si bond coat will generate TGO growth stress due to the  $\approx 2.2 \times$  volume expansion during the conversion of Si to  $SiO_2$ . The TGO expansion will create the out-of-plane strain and tensile stress in the top coat and tends to raise the top coat.
5. Repeated  $\beta \leftrightarrow \alpha$  cristobalite TGO phase transformation during thermal cycling and the accompanied large volume contraction and expansion will result in a severe microfracture of the TGO layer (Figure 11d). The TGO microcracks provide a fast diffusion path for oxidants to reach the Si surface to accelerate its oxidation rate.
6. Thermal stress due to the CTE mismatch between  $\alpha$  cristobalite ( $\approx 30 \times 10^{-6}/^\circ C$ ) [4] and the EBC top coat ( $(4-6) \times 10^{-6}/^\circ C$ ) will aggravate the cracks formation in the TGO layer (Figure 11d).
7. Fish-scale-like cracks and thick TGO lead to a weak interface between the top coat and TGO layer (Figure 11e).
8. The loss of bonding strength between the top coat and TGO in conjunction with the severe thermal stress during the thermal cycling results in the ultimate coating spallation (Figure 11e).

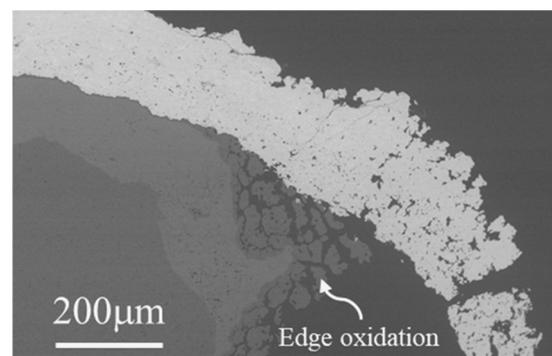
From the above analysis, it can be concluded that in order to improve EBCs' durability, it is critical to reduce the TGO growth rate by the deposition of dense and crack-free coatings with high bonding strength and reduced CTE mismatch (less  $Yb_2SiO_5$  phase). The highly durable LPPS EBCs with a slow TGO growth rate further supports this viewpoint. It is interesting to note that severe coating edge oxidation/damage in the LPPS EBCs after 1000 cycles was also observed (Figure 13). However, the damage by water vapor/oxygen attack is only limited to the coating edges. Unlike the thermal cycling behavior in APS EBCs, the severe oxidation/damage from the edge did not propagate toward the coating center. This remarkable difference is probably due to the high bond strength and close CTE match of the LPPS  $Yb_2Si_2O_7$  coating with substrate. LPPS  $Yb_2Si_2O_7/Si$  EBCs showed promising application for SiC protection in a water vapor environment.



**Figure 11.** Schematic of damage evolution of APS EBCs in a steam cycling test. (a) Pre-existing defects of porosity, microcracks and plant boundaries, (b) Interface microcracks generated due to thermal mismatch stress, (c) Oxidation and edge delamination due to direct steam exposure and poor bonding at edges, (d) TGO microfracture, (e) Loss of bonding and coating spallation.



**Figure 12.** SEM images of APS EBCs coating edges after steam cycling. (a) 200 cycles; (b) 400 cycles.



**Figure 13.** SEM image of LPPS EBCs coating edge after 1000 steam cycles.

It is worth mentioning that the current state-of-the-art EBC systems with silicon as the bond coat are limited to the application/testing temperature of  $\approx 1316$  °C due to the low melting temperature of silicon ( $T_m = 1410$  °C). For next-generation EBC-CMC, the temperature of the EBC-CMC interface is expected to exceed the melting point of Si; therefore, new bond coat materials are needed. Our current work demonstrated that the

LPPS process has the capability for the deposition of highly durably dense EBCs due to the high particle temperature and velocity, which would be the preferred process for the deposition of next-generation EBCs with an advanced bond coat.

#### 4. Conclusions

$\text{Yb}_2\text{Si}_2\text{O}_7/\text{Si}$  bilayer environmental barrier coatings (EBCs) were produced by air plasma spray (APS) and low-pressure plasma spray (LPPS) processes. LPPS EBCs showed denser and crack-free microstructures and higher bonding strength than APS EBCs. Thermal cycling behaviors were tested in air and in steam atmosphere at 1316 °C for both APS and LPPS EBCs. There is no coating failure in air atmosphere for both APS and LPPS EBCs after 900 cycles. In contrast, APS EBCs have an average life of 576 cycles in a steam cycling test in 90%  $\text{H}_2\text{O}$  + 10% air at 1316 °C while LPPS EBCs survived 1000 cycles without failure. Failure of the APS EBCs occurred at the interface between the TGO and top coat. The TGO growth rate is the major driving force for EBCs failure. It was found that the TGO growth rate in APS EBCs is over 20 times higher than that in LPPS EBCs. The LPPS process is more promising than the APS process for producing highly durable EBCs with a far slower TGO growth rate.

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