



## Article

# Concave Ni(OH)<sub>2</sub> Nanocube Synthesis and Its Application in High-Performance Hybrid Capacitors

Nan Cong <sup>1</sup>, Pan Li <sup>2</sup>, Xuyun Guo <sup>3</sup>  and Xiaojuan Chen <sup>1,\*</sup><sup>1</sup> Beijing Academy of Quantum Information Sciences, Beijing 100193, China; congnan@baqis.ac.cn<sup>2</sup> Institute of Analysis and Testing, Beijing Academy of Science and Technology, Beijing 100089, China; cnulp87@126.com<sup>3</sup> Centre for Research on Adaptive Nanostructures and Nanodevices (CRANN) & Advanced Materials Bio-Engineering Research Centre (AMBER), School of Chemistry, Trinity College Dublin, D02PN40 Dublin, Ireland; xguoaa@connect.ust.hk

\* Correspondence: chenxj@baqis.ac.cn

**Abstract:** The controlled synthesis of hollow structure transition metal compounds has long been a very interesting and significant research topic in the energy storage and conversion fields. Herein, an ultrasound-assisted chemical etching strategy is proposed for fabricating concave Ni(OH)<sub>2</sub> nanocubes. The morphology and composition evolution of the concave Ni(OH)<sub>2</sub> nanocubes suggest a possible formation mechanism. The as-synthesized Ni(OH)<sub>2</sub> nanostructures used as supercapacitor electrode materials exhibit high specific capacitance (1624 F g<sup>−1</sup> at 2 A g<sup>−1</sup>) and excellent cycling stability (77% retention after 4000 cycles) due to their large specific surface area and open pathway. In addition, the corresponding hybrid capacitor (Ni(OH)<sub>2</sub>/graphene) demonstrates high energy density (42.9 Wh kg<sup>−1</sup> at a power density of 800 W kg<sup>−1</sup>) and long cycle life (78% retention after 4000 cycles at 5 A g<sup>−1</sup>). This work offers a simple and economic approach for obtaining concave Ni(OH)<sub>2</sub> nanocubes for energy storage and conversion.

**Keywords:** Ni(OH)<sub>2</sub> nanocube; chemical etching; Prussian blue analogs; hybrid capacitor



**Citation:** Cong, N.; Li, P.; Guo, X.; Chen, X. Concave Ni(OH)<sub>2</sub> Nanocube Synthesis and Its Application in High-Performance Hybrid Capacitors. *Nanomaterials* **2023**, *13*, 2538. <https://doi.org/10.3390/nano13182538>

Academic Editor: Jung Woo Lee

Received: 11 August 2023

Revised: 31 August 2023

Accepted: 6 September 2023

Published: 11 September 2023



**Copyright:** © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

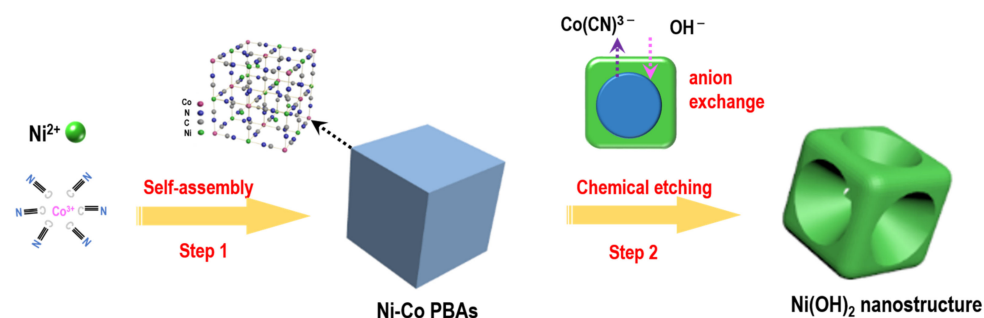
## 1. Introduction

Alkaline metal ion batteries and supercapacitors are two primary electric energy storage and conversion devices that have been widely applied in electrical vehicles, wearable electronics, and power backups [1–3]. To achieve high energy density, high rates, and long cycle life in a single device, hybrid capacitors with a battery-type faradaic cathode and a capacitor-type anode have been rationally designed [4–6]. It is well known that the performance of supercapacitors is highly dependent on the chemical composition and morphology of the electrode materials [7]. Transition metal compounds and conducting polymers frequently exhibit a combination of capacitive and faradaic electrochemical properties. Ni(OH)<sub>2</sub> is the only metallic compound that exhibits faradaic reactions in KOH electrolyte, analogous to battery-type materials [8–10]. Ni(OH)<sub>2</sub> is regarded as a prospective battery electrode material because of its high theoretical capacity, cost effectiveness, and diverse morphologies [11–13]. In terms of morphology, nanostructures with a large surface area, low density, and high surface permeability have the potential to significantly improve the efficiency of energy storage devices [14,15]. Therefore, it is essential to exploit materials with suitable porous structures for energy storage devices.

Prussian blue analogs (PBAs) are an intriguing class of crystalline porous materials that have gained considerable attention due to their uniform sizes, abundant types, and flexible compositions [16–19]. PBA derivatives (such as metal oxides, sulfides, and phosphides) with hollow structures (including frame, box, and porous) have been prepared by chemical corrosion/annealing processes [20–22], and showed tremendous potential for energy storage [23,24]. For instance, Chen et al. fabricated crystalline Mn(OH)<sub>2</sub> and

$\text{Co(OH)}_2$  from PBAs using ion exchange reactions, both of which exhibited outstanding electrochemical properties [25,26]. Although the  $\text{Ni(OH)}_2$  nanocubes derived from PBA have both been reported [27], the chemical etching process is entirely distinct with  $\text{Ni(OH)}_2$  nanocages using  $\text{Cu}_2\text{O}$  templates [28]. The synthesis of hollow  $\text{Ni(OH)}_2$  from PBA is still unknown [21,29].

In this study, size-controlled concave  $\text{Ni(OH)}_2$  nanocubes were successfully synthesized through a chemical etching method under an ultrasound condition. The synthetic route is illustrated in Figure 1. First, Ni-Co PBA precursors were prepared via a modified self-assembly strategy. The XRD pattern confirmed the precursors as  $\text{Ni}_3[\text{Co(CN)}_6]_2 \cdot 9\text{H}_2\text{O}$  (JCPDS #89-3738). As shown in Figure S1, the precursors exhibited a uniform nanocube shape with a size (side length) of  $\sim 400$  nm. Then, the PBAs were converted to concave  $\text{Ni(OH)}_2$  nanocubes via an anion exchange reaction in NaOH solution. The corresponding chemical reaction can be described as:  $\text{Ni}_3[\text{Co(CN)}_6]_2 (\text{s}) + 6\text{OH}^- (\text{aq}) = 3\text{Ni(OH)}_2 (\text{s}) + 2[\text{Co(CN)}_6]^{3-} (\text{aq})$  [15]. The electrochemical measurements indicate that the concave  $\text{Ni(OH)}_2$  nanocubes have a high specific capacity ( $1624 \text{ F g}^{-1}$  at  $2 \text{ A g}^{-1}$ ) as well as an exceptional rate capability. Moreover, an optimized hybrid capacitor concave  $\text{Ni(OH)}_2$  nanocube//graphene with high energy density was fabricated.



**Figure 1.** Schematic illustration of the synthesis route of concave  $\text{Ni(OH)}_2$  nanocubes.

## 2. Materials and Methods

### 2.1. Synthesis of Ni-Co PBA Nanocubes

A total of 132 mg of  $\text{C}_6\text{CoK}_3\text{N}_6$  was first added into 30 mL of DI water and stirred to make solution A, and 264 mg of  $\text{C}_6\text{H}_5\text{Na}_3\text{O}_7 \cdot 2\text{H}_2\text{O}$  and 174 mg of  $\text{Ni(NO}_3)_2 \cdot 6\text{H}_2\text{O}$  were added into 20 mL of DI water and stirred to make a clear green solution, B. Then, solution B was poured into solution A under magnetic stirring for 3 min. The mixed solution was maintained at room temperature for 7 day. After being centrifuged, washed with ethanol numerous times, and vacuum-dried for 24 h, a white-blue powder was achieved.

### 2.2. Synthesis of Concave $\text{Ni(OH)}_2$ Nanocubes

Ni-Co PBA nanocubes (40 mg) were ultrasonically dispersed into 30 mL of DI water to make a homogenous white-blue suspension. Then, NaOH (150–200 mg) was added. The solution turned light green after 6 h of ultrasonic bathing (40 kHz, 150 W) and mechanical vibration at  $40^\circ\text{C}$ . Finally, the as-prepared products were rinsed with ethanol and vacuum-dried overnight at  $80^\circ\text{C}$ .

### 2.3. Material Characterizations

The structure, chemical composition, and morphology of samples were characterized by XRD (X'Pert Pro MPD system), XPS (ESCALAB MK II), SEM (S4800) equipped with EDS, and TEM (JEM-2200F) equipped with EDS for HRTEM images, SAED patterns, and HAADF-STEM images.

### 2.4. Electrochemical Measurements

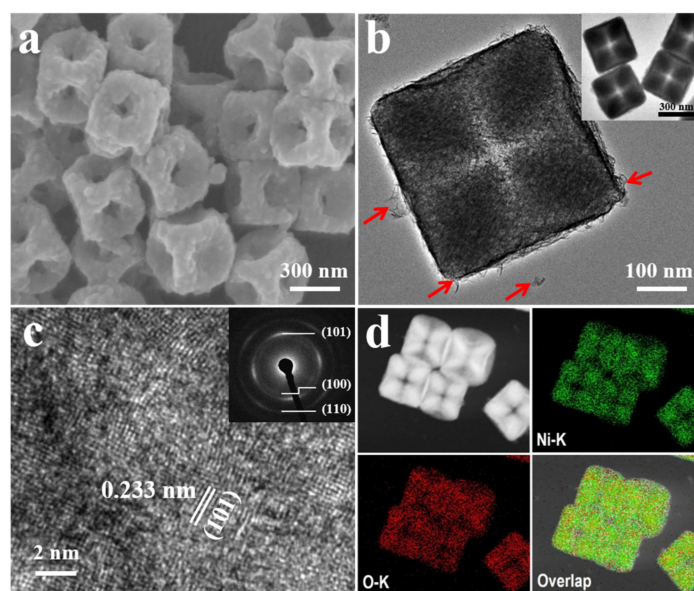
The working electrode was prepared by mixing active materials (70%), carbon black (20%), and polyvinylidene fluoride (10%) in the N-methyl-2-pyrrolidinone. Then, the

mixture was pressed on nickel foam (NF) (area  $\sim 1 \text{ cm}^2$ ) and vacuum-dried overnight at  $80^\circ\text{C}$ . The electrochemical property of an individual electrode was measured in a three-electrode system with a 6 M KOH aqueous electrolyte solution, in which a Pt wire and a saturated calomel electrode (SCE) were used as counter and reference electrodes, respectively. The hybrid capacitor electrochemical test was performed in a two-electrode system with concave  $\text{Ni}(\text{OH})_2$  nanocubes pressed on NF as anodes and graphene pressed on NF as cathodes. All electrochemical measurements were carried out by an electrochemical workstation (CHI660D).

### 3. Results

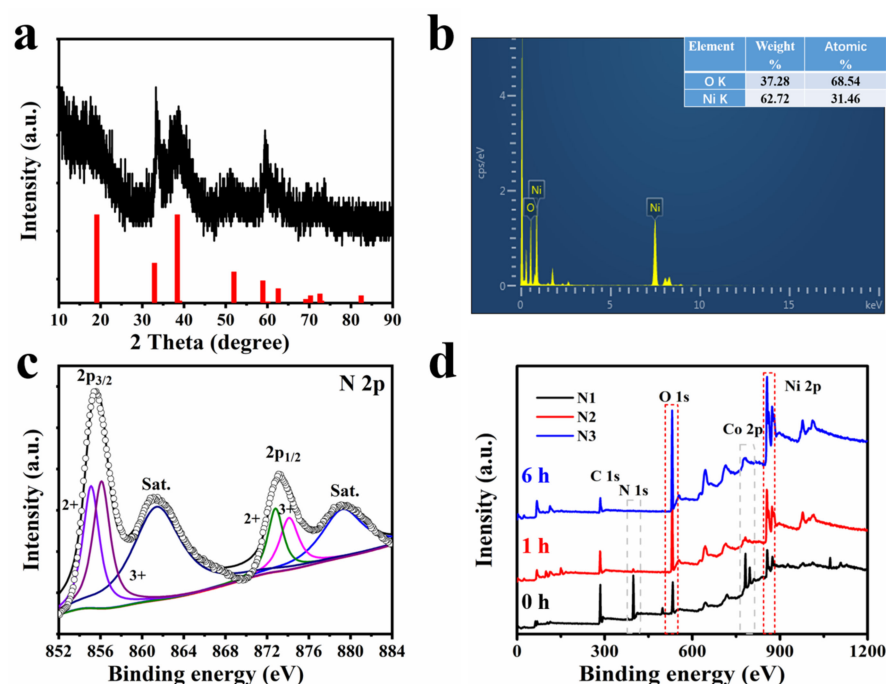
#### 3.1. Characterization of Concave $\text{Ni}(\text{OH})_2$ Nanocubes

An overview SEM image (Figure 2a) reveals that the samples are concave nanocubes with a cavity at each face-center position. The concave nanocubes are approximately 350 nm in size, which is evidently smaller than Ni-Co PBA nanocubes (Figure S1a). In addition, the surface of the concave nanocubes is quite rough, and some layered structure on the surface of the concave nanocubes can be observed clearly, as depicted by the red arrows in Figure 2b. This hollow and porous structure increases the specific surface area and shortens the ion transport path, promoting a rapid redox reaction under the conditions of a high scan rate or large current. Figure 2c shows an HRTEM image of the internal lattice structure of a concave  $\text{Ni}(\text{OH})_2$  nanocube, where the interplanar spacing of 0.23 nm corresponds to the (101) plane. The inset of Figure 2c depicts the selected area electron diffraction (SAED) for a concave  $\text{Ni}(\text{OH})_2$  nanocube, in which three diffraction rings are clearly observed and indexed to the (100), (101), and (110) lattice planes of  $\text{Ni}(\text{OH})_2$ , indicating its polycrystalline nature. A high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) image and the corresponding EDS elemental mapping of Ni and O for nanocages are shown in Figure 2d, indicating a homogeneous distribution of Ni and O elements. Moreover, 100 nm of concave  $\text{Ni}(\text{OH})_2$  nanocubes were also obtained through a similar method, as shown in Figure S2, demonstrating that this ultrasound-assisted etching strategy can be applied to Ni-Co PBA nanocubes with different sizes.



**Figure 2.** (a) SEM image of concave  $\text{Ni}(\text{OH})_2$  nanocubes. (b) TEM image of an individual concave  $\text{Ni}(\text{OH})_2$  nanocube, the red arrow marks the layered structure from concave  $\text{Ni}(\text{OH})_2$  nanocubes. (c) HRTEM lattice image of a concave  $\text{Ni}(\text{OH})_2$  nanocube with the corresponding SAED pattern image (inset). (d) HAADF-STEM image of concave  $\text{Ni}(\text{OH})_2$  nanocubes and the corresponding elemental mapping by STEM.

Figure 3a displays the sample's crystallographic information determined by XRD. The three diffraction peaks appearing at  $33.7^\circ$ ,  $38.4^\circ$ , and  $51.2^\circ$  may correspond to the (100), (101), and (102) crystal planes of  $\text{Ni}(\text{OH})_2$  (JCPDS No.14-0117). The broad diffraction peak located at  $59^\circ$ – $60.5^\circ$  may be attributed to the overlapping of the (110) and (003) crystal planes of  $\text{Ni}(\text{OH})_2$  (JCPDS No.14-0117). The peak at  $33.7^\circ$  is slightly higher than the typical  $\text{Ni}(\text{OH})_2$  (JCPDS No.14-0117) at  $33.0^\circ$ , which is likely due to the intercalation of ions during the synthesis process [30]. The atomic ratios of O/Ni for the concave  $\text{Ni}(\text{OH})_2$  nanocubes were determined by EDS to be 2.2 (Figure 3b). The chemical compositions of concave  $\text{Ni}(\text{OH})_2$  nanocubes were also investigated using XPS measurements. Figure 3c shows the high-resolution Ni 2p spectrum of  $\text{Ni}(\text{OH})_2$ . The Ni 2p spectrum was fitted with four main peaks and two satellite peaks using a Gaussian–Lorentz fitting method. The fitting peaks at 855.1 eV and 872.8 eV corresponded to  $\text{Ni}^{2+}$ , whereas the peaks at 856.1 eV and 874.1 eV corresponded to  $\text{Ni}^{3+}$  [31]. Two strong shakeup-type peaks of nickel were detected at 861.3 and 879.4 eV. The analysis of Ni 2p nanomaterials was also added to the revised manuscript [32,33]. Figure 3d depicts three broad XPS spectra of the products at three distinct etching stages: 0 (N1), 1 (N2), and 6 (N3) hours. The peaks at 398.4, 534.6, 782.7, and 858.3 eV in the N1 curve for  $\text{Ni}_3[\text{Co}(\text{CN})_6]_2 \cdot 9\text{H}_2\text{O}$  correspond to the N, O, Co, and Ni element signals, respectively. As the etching reaction progressed from N1 to N3, the peaks at N 1s and Co 2p (denoted by the grey dashed box) diminished and eventually disappeared, whereas the Ni and O peaks (denoted by the red dashed box) are relatively robust. This result indicates that Co and N from  $[\text{Co}(\text{CN})_6]^{3-}$  were progressively consumed.



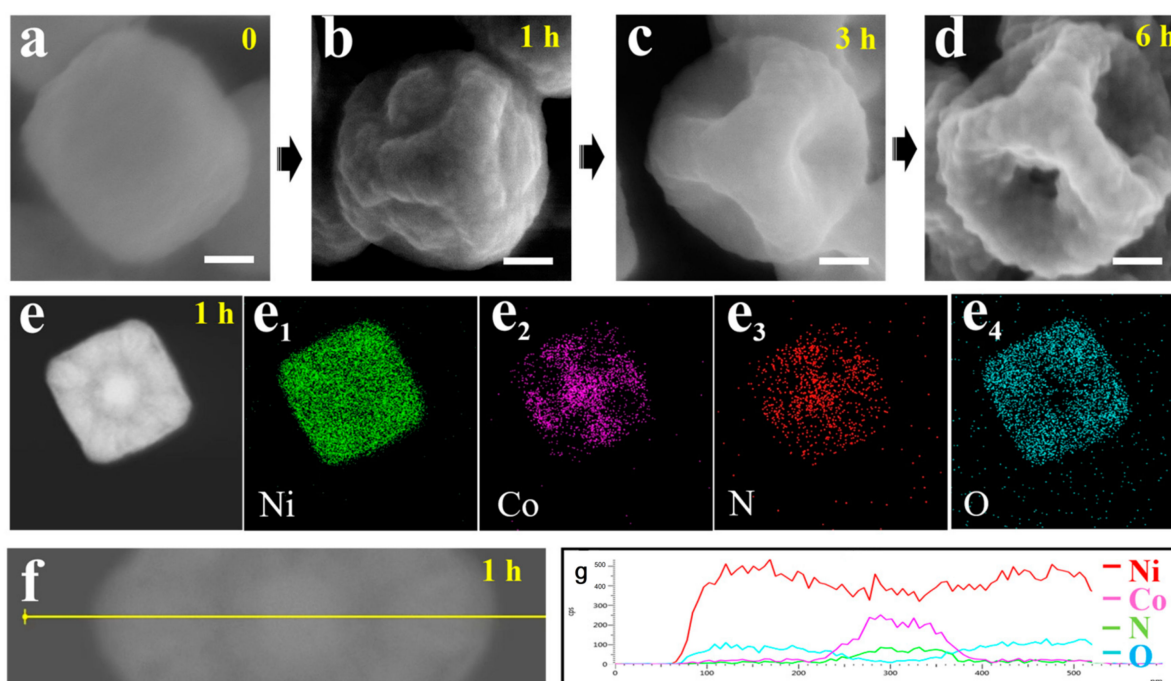
**Figure 3.** (a) The XRD pattern of concave  $\text{Ni}(\text{OH})_2$  nanocubes. (b) The EDS spectra of concave  $\text{Ni}(\text{OH})_2$  nanocubes. (c) High-resolution XPS spectra of Ni 2p for concave  $\text{Ni}(\text{OH})_2$  nanocubes. (d) XPS survey spectra of three samples at different etching stages (0 h, 1 h, and 6 h).

### 3.2. Formation Mechanism of Concave $\text{Ni}(\text{OH})_2$ Nanocubes

To explore the formation mechanism of concave  $\text{Ni}(\text{OH})_2$  nanocubes, a series of morphological and chemical composition characterizations of intermediates collected from different reaction stages was performed, as shown in Figure 4a–d. When the etching reaction reached 1 h (Figure 4b), all of the sharp corners and edges of the nanocube were dissolved, and a ring island emerged on each face of the nanocube. Arriving at 3 h (Figure 4c), the inner ring island vanished, resulting in the formation of an open hole on each face. Finally, a concave nanocube was discovered (6 h). As the reaction



progressed from 0 h to 6 h, the Co/Ni atomic ratio decreased from 2/3 to 0 (Figure S3), indicating that the Co in  $[\text{Co}(\text{CN})_6]^{3-}$  was completely replaced by  $\text{OH}^-$ . Besides the SEM and EDS analyses, a key chemical composition characterization of intermediates (reaction occurring at 1 h) was carried out. As displayed in Figure 4e–e4, the STEM image and corresponding EDS mapping of the intermediates show that Ni and O elements are uniformly distributed on the nanocube but weak in the face-center region, whereas the Co and N constituents from  $\text{Ni}_3[\text{Co}(\text{CN})_6]_2 \cdot 9\text{H}_2\text{O}$  only appear in the face-center region. Figure 4g depicts the comprehensive distribution of Ni, O, Co, and N along the yellow line (Figure 4f) on the sample. The EDS line results also prove that the edges of the nanocube are converted to  $\text{Ni}(\text{OH})_2$ , whereas the face-center regions retain their original composition of  $\text{Ni}_3[\text{Co}(\text{CN})_6]_2 \cdot 9\text{H}_2\text{O}$ .



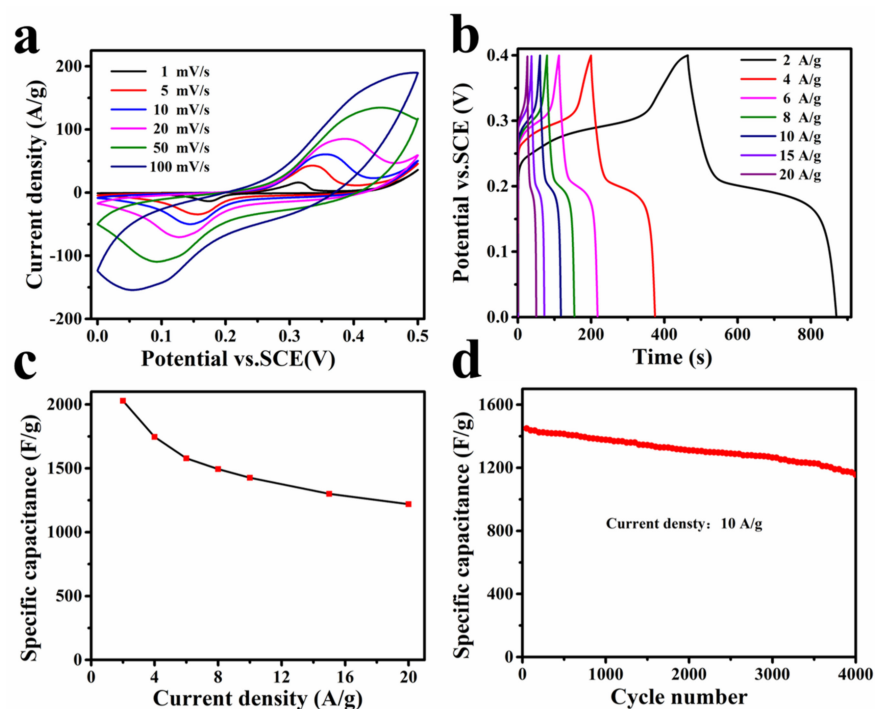
**Figure 4.** SEM images of the intermediate products at different reaction stages: (a) 0 h, (b) 1 h, (c) 3 h, and (d) 5 h. Scale bars: 100 nm. (e) STEM image of the samples at 1 h and the corresponding EDX mapping showing (e1–e4) Ni, Co, N, and O. (f) STEM image of the samples at 1 h and the corresponding EDS line scan (g) along the yellow line.

Based on the above discussions, a possible formation mechanism for concave  $\text{Ni}(\text{OH})_2$  nanocubes is proposed. The edges of Ni-Co PBA nanocubes first undergo ion exchange reactions due to their high curvature and defects. Due to the formation of thick  $\text{Ni}(\text{OH})_2$  layers that protect the interior of the edges from further etching and surfactants that protect the face-center region from dissolution, the etching route shifts to the boundary between the etched edge and the resting face, resulting in the formation of a ring channel on each face. As the channel becomes deeper, the face-center region is completely separated from the structure. After the remaining  $\text{Ni}_3[\text{Co}(\text{CN})_6]_2 \cdot 9\text{H}_2\text{O}$  in the body-center is completely dissolved from six different directions, the concave nanocube structure is formed.

### 3.3. Electrochemical Property of Concave $\text{Ni}(\text{OH})_2$ Nanocubes

In a standard three-electrode electrochemical setup, the capacity behavior of concave  $\text{Ni}(\text{OH})_2$  nanocubes of approximately 350 nm in size was investigated. Figure 5a shows the CV curves of the  $\text{Ni}(\text{OH})_2$  electrode for various scan rates within the potential window range of 0 to 0.5 V versus SCE. All CV curves exhibit a pair of redox peaks caused by the faradaic reaction  $\text{Ni}(\text{OH})_2 + \text{OH}^- \rightleftharpoons \text{NiOOH} + \text{H}_2\text{O} + \text{e}^-$  [31,34], implying the battery-type materials of the  $\text{Ni}(\text{OH})_2$  electrode [35–38]. As the scan rate increases, the anodic and

cathodic peaks shift toward higher and lower potentials, respectively, without a significant change in the shape of the CV profiles, revealing that the  $\text{Ni}(\text{OH})_2$  electrode enables rapid redox reactions. Figure 5b presents the typical galvanostatic charge–discharge (GCD) voltage vs. time curves for various current densities. All curves exhibit an asymmetrical shape with well-defined plateaus, implying its battery-type nature. According to the discharge profiles, the specific capacitance of the  $\text{Ni}(\text{OH})_2$  electrode is calculated to be 1624, 1396, 1260, 1196, 1142, 1044, and 980  $\text{F g}^{-1}$  at 2, 4, 6, 8, 10, 15, and 20  $\text{A g}^{-1}$ , respectively (Figure 5c). Even at the current density of 20  $\text{A g}^{-1}$ , a capacitance of 980  $\text{F g}^{-1}$  can be achieved. The  $\text{Ni}(\text{OH})_2$  electrode demonstrates a higher specific capacity than previously reported  $\text{Ni}(\text{OH})_2$  nanostructures, due to the larger Brunauer–Emmett–Teller (BET) surface areas of  $187 \text{ m}^2 \text{ g}^{-1}$ , as shown in Figure S4 [39,40]. This high-rate capability is attributed to the nanocage structure, which has a greater surface area with a porous structure, allowing ions to move quickly. Figure 5d shows the cycle performance of the  $\text{Ni}(\text{OH})_2$  electrode measured at 10  $\text{A g}^{-1}$ . After 4000 cycles, the specific capacitance retained almost 77% of the maximum capacitance. Figure S5 presents the Nyquist plot of the  $\text{Ni}(\text{OH})_2$  electrode. The equivalent series resistance (ESR) of the  $\text{Ni}(\text{OH})_2$  electrode was evaluated to be  $0.52 \Omega$  from the intercept with the real axis.

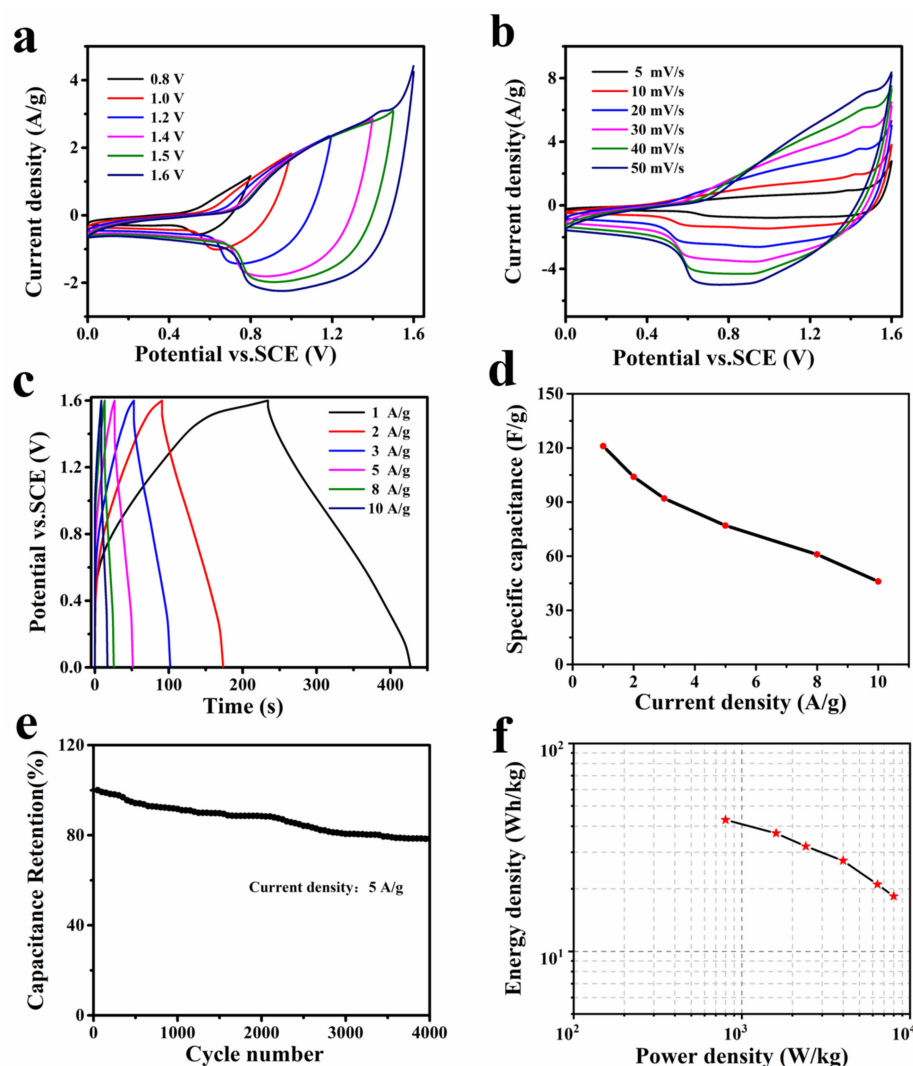


**Figure 5.** (a) CV curves and (b) GCD curves of concave  $\text{Ni}(\text{OH})_2$  nanocubes. (c) Specific capacity of concave  $\text{Ni}(\text{OH})_2$  nanocubes at different current densities. (d) Cycle performance of concave  $\text{Ni}(\text{OH})_2$  nanocubes at 10  $\text{A g}^{-1}$ .

For practical applications, we constructed a hybrid capacitor device with concave  $\text{Ni}(\text{OH})_2$  nanocubes as the positive electrode and graphene as the negative electrode. As shown in Figure S6, the CV and CP behaviors of graphene were investigated. As expected, it exhibited a characteristic of a double-layer electric capacitor with a specific capacitance of  $193 \text{ F g}^{-1}$  at  $1 \text{ A g}^{-1}$ . To acquire the maximum capacitance of the hybrid capacitor, the optimal mass ratio is  $m(\text{concave } \text{Ni}(\text{OH})_2 \text{ nanocubes})/m(\text{graphene electrodes}) = (C^- \times \Delta V^-)/(C^+ \times \Delta V^+) = 0.095$ .

Figure 6a shows the CV curves of the hybrid capacitor measured at a scan rate of  $10 \text{ mV s}^{-1}$  for various voltage windows. The operating potential window can extend to  $1.6 \text{ V}$  near the voltage of an oxygen evolution reaction in aqueous solution. Figure 6b presents the CV curves of the hybrid capacitor in the potential window of  $0\text{--}1.6 \text{ V}$  at various scan

rates. As can be seen, all CV curves have a distorted rectangular shape, indicating that the hybrid device contained both a faradaic capacitor and an electric double-layer capacitance. The voltage–time curves of the hybrid capacitor at varied current densities are depicted in Figure 6c. Figure 6d depicts the calculated specific capacities based on the total mass of the active materials. The maximal specific capacity can reach up to  $121 \text{ F g}^{-1}$  at  $1 \text{ A g}^{-1}$ . With repeated charge–discharge measurements at a current density of  $5 \text{ A g}^{-1}$ , the cycling stability of the hybrid capacitor was investigated. Figure 6e illustrates the capacity retention variation of the hybrid capacitor along the cycling number, which reveals a capacity loss of 22% after 4000 cycles. The final products are shown in Figure S8, the active materials were partially destroyed and aggregated.



**Figure 6.** (a) CV curves of the concave  $\text{Ni}(\text{OH})_2$  nanocubes//graphene hybrid capacitor with varying potential windows (at  $10 \text{ mV s}^{-1}$ ). (b) CV and (c) GCD curves of the hybrid capacitor. (d) Specific capacity of the hybrid device as a function of current densities. (e) Cycle performance measured at  $5 \text{ A g}^{-1}$  for 4000 cycles. (f) Ragone plots of the hybrid capacitor.

Figure 6f presents the Ragone plot with regard to the energy density and power density of our concave  $\text{Ni}(\text{OH})_2$  nanocubes//graphene hybrid capacitor. The maximal energy density was  $42.9 \text{ Wh kg}^{-1}$  at a power density of  $800 \text{ W kg}^{-1}$ , and remained at  $18.3 \text{ Wh kg}^{-1}$  at  $8 \text{ kW kg}^{-1}$ . The energy density of our hybrid capacitor is comparable to that of numerous analogous works, such as flower-like  $\text{Ni}(\text{OH})_2$ //AC ( $25.5 \text{ Wh kg}^{-1}$  at  $1.28 \text{ kW kg}^{-1}$ ) [41],  $\beta$ - $\text{Ni}(\text{OH})_2$ //AC ( $36.2 \text{ Wh kg}^{-1}$  at  $100.6 \text{ W kg}^{-1}$ ) [42],  $\text{Ni}(\text{OH})_2$ -ECF//AC/NF ( $31.3 \text{ Wh kg}^{-1}$

at  $800 \text{ W kg}^{-1}$  [43], NiCo-LDHs/Zn<sub>2</sub>SnO<sub>4</sub>//AC ( $23.7 \text{ Wh kg}^{-1}$  at  $284.2 \text{ W kg}^{-1}$ ) [44], and Ni(OH)<sub>2</sub>/graphene//RuO<sub>2</sub>/graphene ( $48 \text{ Wh kg}^{-1}$  at  $230 \text{ W kg}^{-1}$ ) [45].

#### 4. Conclusions

In summary, we reported concave Ni(OH)<sub>2</sub> nanocubes through the treatment of Ni-Co PBA nanocubes with NaOH solution in the presence of ultrasound. Different reaction activity between the edges and face-centered regions of Ni-Co PBAs resulted in the formation of a unique hollow structure. When these concave Ni(OH)<sub>2</sub> nanocubes were used as battery-type electrode materials, they presented high specific capacity and a long cycle life. Moreover, a hybrid capacitor (concave Ni(OH)<sub>2</sub> nanocubes//graphene) was fabricated and demonstrated outstanding performance with high energy density and good cycling stability. This study provides a simple and efficient method for synthesizing hollow nanomaterials for energy storage applications.

**Supplementary Materials:** The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/nano13182538/s1>, Figure S1: SEM image and XRD pattern of Ni-Co PBAs nanocubes with a size of 400 nm; Figure S2: SEM and TEM images of concave Ni(OH)<sub>2</sub> nanocubes with size of 350 nm and 100 nm, respectively; Figure S3: EDS spectra of samples under different reaction stages; Figure S4: EIS curve for concave Ni(OH)<sub>2</sub> nanocubes; Figure S5: The electrochemistry properties of graphene; Figure S6: EIS plot for concave Ni(OH)<sub>2</sub> nanocubes//graphene hybrid capacitor. Figure S7: Nyquist plot for concave Ni(OH)<sub>2</sub> nanocubes//graphene hybrid capacitor; Figure S8: SEM image of concave Ni(OH)<sub>2</sub> nanocubes and graphene after 4000 cycles.

**Author Contributions:** Conceptualization, X.C.; methodology, X.C.; investigation, X.C.; data curation, N.C.; writing—original draft preparation, X.C.; writing—review and editing, N.C.; visualization, X.G.; supervision, X.G.; project administration, X.C.; funding acquisition, P.L. All authors have read and agreed to the published version of the manuscript.

**Funding:** This research was supported by the Scientific Research Program of BJUST (No.23CB024).

**Data Availability Statement:** Not applicable.

**Conflicts of Interest:** The authors declare no conflict of interest.

#### References

1. Winter, M.; Brodd, R.J. What Are Batteries, Fuel Cells, and Supercapacitors? *Chem. Rev.* **2004**, *104*, 4245–4270. [[CrossRef](#)]
2. Simon, P.; Gogotsi, Y. Materials for electrochemical capacitors. *Nat. Mater.* **2008**, *7*, 845–854. [[PubMed](#)]
3. Cook, T.R.; Dogutan, D.K.; Reece, S.Y.; Surendranath, Y.; Teets, T.S.; Nocera, D.G. Solar energy supply and storage for the legacy and nonlegacy worlds. *Chem. Rev.* **2010**, *110*, 6474–6502. [[PubMed](#)]
4. Lu, X.; Yu, M.; Wang, G.; Tong, Y.; Li, Y. Flexible solid-state supercapacitors: Design, fabrication and applications. *Energy Environ. Sci.* **2014**, *7*, 2160–2181.
5. Dubal, D.P.; Ayyad, O.; Ruiz, V.; Gomez-Romero, P. Hybrid energy storage: The merging of battery and supercapacitor chemistries. *Chem. Soc. Rev.* **2015**, *44*, 1777–1790. [[CrossRef](#)] [[PubMed](#)]
6. Cherusseri, J.; Sambath Kumar, K.; Choudhary, N.; Nagaiah, N.; Jung, Y.; Roy, T.; Thomas, J. Novel mesoporous electrode materials for symmetric, asymmetric and hybrid supercapacitors. *Nanotechnology* **2019**, *30*, 202001–202019. [[PubMed](#)]
7. Yu, Z.; Tetard, L.; Zhai, L.; Thomas, J. Supercapacitor electrode materials: Nanostructures from 0 to 3 dimensions. *Energy Environ. Sci.* **2015**, *8*, 702–730.
8. Qin, Q.; Ou, D.; Ye, C.; Chen, L.; Lan, B.; Yan, J.; Wu, Y. Systematic study on hybrid supercapacitor of Ni-Co layered double hydroxide//activated carbons. *Electrochim. Acta* **2019**, *305*, 403–415. [[CrossRef](#)]
9. Liang, Z.; Qu, C.; Zhou, W.; Zhao, R.; Zhang, H.; Zhu, B.; Guo, W.; Meng, W.; Wu, Y.; Aftab, W.; et al. Synergistic Effect of Co-Ni Hybrid Phosphide Nanocages for Ultrahigh Capacity Fast Energy Storage. *Adv. Sci.* **2019**, *6*, 1802005–1802011. [[CrossRef](#)]
10. Yang, X.; Fu, K.; Mao, L.; Peng, W.; Jin, J.; Yang, S.; Li, G. Bio-mediated synthesis of  $\alpha$ -Ni(OH)<sub>2</sub> nanobristles on hollow porous carbon nanofibers for rechargeable alkaline batteries. *Chem. Eng. Sci.* **2019**, *205*, 269–277.
11. Nai, J.; Wang, S.; Bai, Y.; Guo, L. Amorphous Ni(OH)<sub>2</sub> nanoboxes: Fast fabrication and enhanced sensing for glucose. *Small* **2013**, *9*, 3147–3152.
12. Hu, B.; Qin, X.; Asiri, A.M.; Alamry, K.A.; Al-Youbi, A.O.; Sun, X. Fabrication of Ni(OH)<sub>2</sub> nanoflakes array on Ni foam as a binder-free electrode material for high performance supercapacitors. *Electrochim. Acta* **2013**, *107*, 339–342.
13. Xie, M.; Xu, Z.; Duan, S.; Tian, Z.; Zhang, Y.; Xiang, K.; Lin, M.; Guo, X.; Ding, W. Facile growth of homogeneous Ni(OH)<sub>2</sub> coating on carbon nanosheets for high-performance asymmetric supercapacitor applications. *Nano Res.* **2017**, *11*, 216–224.



14. Li, Y.; Shi, J. Hollow-structured mesoporous materials: Chemical synthesis, functionalization and applications. *Adv. Mater.* **2014**, *26*, 3176–3205. [[PubMed](#)]
15. Li, L.; Tan, L.; Li, G.; Zhang, Y.; Liu, L. Self-Templated Synthesis of Porous Ni(OH)<sub>2</sub> Nanocube and Its High Electrochemical Performance for Supercapacitor. *Langmuir* **2017**, *33*, 12087–12094.
16. Qian, J.F.; Wu, C.; Cao, Y.L.; Ma, Z.F.; Huang, Y.H.; Ai, X.P.; Yang, H.X. Prussian Blue Cathode Materials for Sodium-Ion Batteries and Other Ion Batteries. *Adv. Energy Mater.* **2018**, *8*, 1702619–1702643.
17. Ma, F.; Li, Q.; Wang, T.Y.; Zhang, H.G.; Wu, G. Energy storage materials derived from Prussian blue analogues. *Sci. Bull.* **2017**, *62*, 358–368.
18. Karyakin, A.A. Prussian Blue and its analogues: Electrochemistry and analytical applications. *Electroanalysis* **2001**, *13*, 813–819.
19. Nai, J.; Zhang, J.; Lou, X.W. Construction of Single-Crystalline Prussian Blue Analog Hollow Nanostructures with Tailorable Topologies. *Chem* **2018**, *4*, 1967–1982.
20. Feng, Y.; Yu, X.Y.; Paik, U. Nickel cobalt phosphides quasi-hollow nanocubes as an efficient electrocatalyst for hydrogen evolution in alkaline solution. *Chem. Commun.* **2016**, *52*, 1633–1636.
21. Han, L.; Yu, X.-Y.; Lou, X.W.D. Formation of Prussian-Blue-Analog Nanocages via a Direct Etching Method and their Conversion into Ni-Co-Mixed Oxide for Enhanced Oxygen Evolution. *Adv. Mater.* **2016**, *28*, 4601–4605. [[CrossRef](#)]
22. Yu, X.Y.; Yu, L.; Wu, H.B.; Lou, X.W. Formation of nickel sulfide nanoframes from metal-organic frameworks with enhanced pseudocapacitive and electrocatalytic properties. *Angew. Chem. Int. Ed. Engl.* **2015**, *54*, 5331–5335.
23. Zhang, L.; Shi, L.; Huang, L.; Zhang, J.; Gao, R.; Zhang, D. Rational Design of High-Performance DeNO<sub>x</sub> Catalysts Based on Mn<sub>x</sub>Co<sub>3-x</sub>O<sub>4</sub> Nanocages Derived from Metal–Organic Frameworks. *ACS Catal.* **2014**, *4*, 1753–1763.
24. Wang, D.; Qi, X.; Gao, H.; Yu, J.; Zhao, Y.; Zhou, G.; Li, G. Fabricating hierarchical porous ZnCo<sub>2</sub>O<sub>4</sub> microspheres as high-performance anode material for lithium-ion batteries. *Mater. Lett.* **2016**, *164*, 93–96.
25. Zhang, J.; Luan, Y.; Lyu, Z.; Wang, L.; Xu, L.; Yuan, K.; Pan, F.; Lai, M.; Liu, Z.; Chen, W. Synthesis of hierarchical porous delta-MnO<sub>2</sub> nanoboxes as an efficient catalyst for rechargeable Li–O<sub>2</sub> batteries. *Nanoscale* **2015**, *7*, 14881–14888.
26. Zhang, J.; Lyu, Z.; Zhang, F.; Wang, L.; Xiao, P.; Yuan, K.; Lai, M.; Chen, W. Facile synthesis of hierarchical porous Co<sub>3</sub>O<sub>4</sub> nanoboxes as efficient cathode catalysts for Li–O<sub>2</sub> batteries. *J. Mater. Chem. A* **2016**, *4*, 6350–6356.
27. Nai, J.; Yin, H.; You, T.; Zheng, L.; Zhang, J.; Wang, P.; Jin, Z.; Tian, Y.; Liu, J.; Tang, Z.; et al. Efficient Electrocatalytic Water Oxidation by Using Amorphous Ni-Co Double Hydroxides Nanocages. *Adv. Energy Mater.* **2015**, *5*, 1401880–1401886.
28. Tian, L.; Yang, T.; Pu, W.; Zhang, J. Synthesis of Cubic Ni(OH)<sub>2</sub> Nanocages Through Coordinating Etching and Precipitating Route for High-Performance Supercapacitors. *Nanoscale Res. Lett.* **2019**, *14*, 264–272. [[CrossRef](#)] [[PubMed](#)]
29. Zheng, C.; Cao, C.; Chang, R.; Hou, J.; Zhai, H. Hierarchical mesoporous NiCo<sub>2</sub>O<sub>4</sub> hollow nanocubes for supercapacitors. *Phys. Chem. Chem. Phys.* **2016**, *18*, 6268–6274.
30. Wang, G.; Yan, Z.; Ding, Y.; Xu, Z.; Li, Z. Hierarchical core-shell nickel hydroxide@nitrogen-doped hollow carbon spheres composite for high-performance hybrid supercapacitor. *J. Colloid Interface Sci.* **2022**, *628*, 286–296. [[CrossRef](#)]
31. Ji, J.; Zhang, L.L.; Ji, H.; Li, Y.; Zhao, X.; Bai, X.; Fan, X.; Zhang, F.; Ruoff, R.S. Nanoporous Ni(OH)<sub>2</sub> Thin Film on 3D Ultrathin-Graphite Foam for Asymmetric Supercapacitor. *ACS Nano* **2013**, *7*, 6237–6243. [[CrossRef](#)] [[PubMed](#)]
32. Chen, F.; Wang, H.; Ji, S.; Linkov, V.; Wang, R. Core-shell structured Ni<sub>3</sub>S<sub>2</sub>@Co(OH)<sub>2</sub> nano-wires grown on Ni foam as binder-free electrode for asymmetric supercapacitors. *Chem. Eng. J.* **2018**, *345*, 48–57. [[CrossRef](#)]
33. Zhao, B.; Zhang, L.; Zhang, Q.; Chen, D.; Cheng, Y.; Deng, X.; Chen, Y.; Murphy, R.; Xiong, X.; Song, B.; et al. Rational Design of Nickel Hydroxide-Based Nanocrystals on Graphene for Ultrafast Energy Storage. *Adv. Energy Mater.* **2018**, *8*, 1702247. [[CrossRef](#)]
34. Wang, Y.; Yin, Z.; Li, X.; Guo, H.; Zhang, D.; Wang, Z. Smartly tailored Co(OH)<sub>2</sub>-Ni(OH)<sub>2</sub> heterostructure on nickel foam as binder-free electrode for high-energy hybrid capacitors. *Electrochim. Acta* **2019**, *309*, 140–147. [[CrossRef](#)]
35. Albohani, S.; Minakshi Sundaram, M.; Laird, D.W. Egg shell membrane template stabilises formation of β-NiMoO<sub>4</sub> nanowires and enhances hybrid supercapacitor behaviour. *Mater. Lett.* **2019**, *236*, 64–68. [[CrossRef](#)]
36. Barmi, M.J.; Minakshi, M. Tuning the Redox Properties of the Nanostructured CoMoO<sub>4</sub> Electrode: Effects of Surfactant Content and Synthesis Temperature. *ChemPlusChem* **2016**, *81*, 964–977. [[CrossRef](#)]
37. Verma, M.L.; Minakshi, M.; Singh, N.K. Synthesis and Characterization of Solid Polymer Electrolyte based on Activated Carbon for Solid State Capacitor. *Electrochim. Acta* **2014**, *137*, 497–503. [[CrossRef](#)]
38. Wickramaarachchi, K.; Minakshi, M. Status on electrodeposited manganese dioxide and biowaste carbon for hybrid capacitors: The case of high-quality oxide composites, mechanisms, and prospects. *J. Energy Storage* **2022**, *56*, 106099–106126. [[CrossRef](#)]
39. Mullangi, D.; Dhavale, V.; Shalini, S.; Nandi, S.; Collins, S.; Woo, T.; Kurungot, S.; Vaidhyanathan, R. Low-Overpotential Electrocatalytic Water Splitting with Noble-Metal-Free Nanoparticles Supported in a sp<sup>3</sup> N-Rich Flexible COF. *Adv. Energy Mater.* **2016**, *6*, 1600110–1600116. [[CrossRef](#)]
40. Nandi, S.; Singh, S.K.; Mullangi, D.; Illathvalappil, R.; George, L.; Vinod, C.P.; Kurungot, S.; Vaidhyanathan, R. Low Band Gap Benzimidazole COF Supported Ni<sub>3</sub>N as Highly Active OER Catalyst. *Adv. Energy Mater.* **2016**, *6*, 1601189–1601199. [[CrossRef](#)]
41. Tang, Y.; Liu, Y.; Yu, S.; Zhao, Y.; Mu, S.; Gao, F. Hydrothermal synthesis of a flower-like nano-nickel hydroxide for high performance supercapacitors. *Electrochim. Acta* **2014**, *123*, 158–166. [[CrossRef](#)]
42. Huang, J.; Xu, P.; Cao, D.; Zhou, X.; Yang, S.; Li, Y.; Wang, G. Asymmetric supercapacitors based on β-Ni(OH)<sub>2</sub> nanosheets and activated carbon with high energy density. *J. Power Sources* **2014**, *246*, 371–376. [[CrossRef](#)]

43. Kazemi, S.H.; Kazemi, H.; Kabiri-Samani, M.; Kiani, M.A. Highly stable asymmetric supercapacitors based on  $\text{Ni}(\text{OH})_2$  deposited on electro-etched carbon paper. *J. Iran. Chem. Soc.* **2018**, *15*, 2117–2122. [[CrossRef](#)]
44. Wang, X.; Sumboja, A.; Lin, M.; Yan, J.; Lee, P.S. Enhancing electrochemical reaction sites in nickel-cobalt layered double hydroxides on zinc tin oxide nanowires: A hybrid material for an asymmetric supercapacitor device. *Nanoscale* **2012**, *4*, 7266–7272. [[CrossRef](#)]
45. Wang, H.; Liang, Y.; Mirfakhrai, T.; Chen, Z.; Casalongue, H.S.; Dai, H. Advanced asymmetrical supercapacitors based on graphene hybrid materials. *Nano Res.* **2011**, *4*, 729–736. [[CrossRef](#)]

**Disclaimer/Publisher’s Note:** The statements, opinions and data contained in all publications are solely those of the individual author(s) and contributor(s) and not of MDPI and/or the editor(s). MDPI and/or the editor(s) disclaim responsibility for any injury to people or property resulting from any ideas, methods, instructions or products referred to in the content.