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# Sub-Wavelength Scale Si Inverted Pyramid Fabrication with Enhanced Size Control by Using Silica Sphere Lithography Technique

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**Abstract:** In this paper, we present a novel silicon (Si) subwavelength-scale surface structure (SWSS) fabrication process using the silica sphere (SS) lithography technique, which allows controllable geometries. The process involves a new cost-effective solvent-controlled spin-coating method that deposits SS on large Si surface areas with enhanced SS monolayer coverage and uniformity compared to conventional methods. The larger areas and rapid, low cost processing allow colloidal sphere lithography to be realistically used for solar cells. We successfully demonstrate 1.57  $\mu$ m diameter SS on a 2-inch round Si wafer with more than 95% SS monolayer coverage and great uniformity. Using these deposited SS, a SWSS fabrication process was designed and successfully demonstrated Si inverted pyramid structures with dimension on the order of 1.1  $\mu$ m, thus potentially providing a new technique for effective light-management of thin crystalline Si solar cells.

Keywords: inverted pyramid; silica sphere lithography; sub-wavelength; spin-coating; solar cell

## 1. Introduction

Continued improvements in both solar cell cost and efficiency can be realized by continuing to decrease solar cell thickness. Reduced thickness has contributed to recent high efficiency and high voltage silicon (Si) solar cells, reaching over 750 mV open circuit voltage ( $V_{OC}$ ) on <100 micron thick wafers [1,2]. Further improvements in both optical control (light trapping) and reduced reflection leading to further efficiency enhancements require low-cost, controllable surface geometries. Previous reports have demonstrated that subwavelength-scale surface structures (SWSS) can offer an improved anti-reflection behavior over a broad wavelength range even at wide incident angles, resulting from the enhanced gradient in a refractive index in the structured layer [3]. Moreover, SWSS in an optimal scale and shape can also provide significant light path-length enhancement inside the Si absorber layer because of enhanced light trapping [4–6]. A recent report from Mavrokefalos, et al., demonstrated that an optimized subwavelength-scale inverted pyramid on a 10 µm thick c-Si absorber performs as a highly effective anti-reflection (AR) layer and achieves an improved absorption close to the Lambertian limit [7]. In addition to these optical advantages, thin Si PV enabled by SWSS can provide an enhanced open-circuit voltage (V<sub>OC</sub>) with improved photo-generated carrier collection [8]. However, despite the advantages of SWSS for thin c-Si PV, its commercial application is impractical using current fabrication techniques such as e-beam [9,10] and laser interference [11,12] lithography. These processes are time-consuming and relatively expensive and, as a result, are limited to research use and small cell areas. Therefore, a cost-effective and large area applicable SWSS fabrication process needs to be developed for the realization of future thin Si PV.

In this paper, we describe a novel SWSS fabrication process that enables the processing of Si inverted pyramid structures in a controlled scale on large Si surface areas. The Si surface patterning has been done with a silica sphere (SS) lithography technique to replace the conventional multi-step photolithography method. We have developed a solvent-controlled SS spin-coating method that enables deposition of a highly uniform SS monolayer on large surface areas [13]. Previous SS spin-coating methods based on water as a solvent require additional surfactant mixtures for (1) enhanced SS dispersion in solution, and (2) delayed solvent evaporation rate because of the properties of water [14]. Although surfactant addition offers significantly enhanced solution properties compared to pure water for uniform SS monolayer spin-coating; it is successful only for small area applications (e.g., 1-inch  $\times$  1-inch). Other conventional SS deposition methods including Langmuir–Blodgett coating and substrate surface functionalized spin-coating are not suitable for manufacturing due to their slowness and/or sensitivity to the processing environment [15-17]. To realize large area SS lithography with a simple and low-cost spin-coating approach, we use N,N-dimethyl-formamide (DMF) as a solvent and confirm that DMF can provide (1) a highly uniform close-packed SS monolayer with (2) minimal sensitivity to the spin-coating environment (e.g., vibration, humidity, and temperature), in addition to (3) large area SS deposition capability. Here, we demonstrate more than 95% SS monolayer coverage with outstanding uniformity on a 2-inch round Si wafer just with a simple spin-coating process performed in a non-environment controlled atmosphere.

One major advantage of SS lithography for SWSS fabrication is that SS offers highly flexible dimensional control of the structures. The period of fabricated structures can be easily changed by introducing different size of SS. In addition, the size of structure itself also can be adjusted by SS size reduction with dry etch process after SS spin-coating. In this work we utilized an optimized reactive-ion etching (RIE) condition designed mainly to etch the SS (i.e.,  $SiO_2$ ) based on a CHF<sub>3</sub> plasma assisted RIE process. We further demonstrate effective SWSS fabrication, especially, a well-arrayed inverted pyramid structure with dimension on the order of 1.1 µm and conclude that this will provide a more simplified and large-area applicable SWSS fabrication approach compared to conventional methods.

## 2. Experimental

Single side polished 2-inch diameter silicon wafers were used as substrates. Dry silica spheres (SS) sized at 1.57  $\mu$ m were obtained from Bang Lab (10% possible size variation indicated by Bang Lab, Fishers, IN, USA). The common solvents (acetone and methanol) used in the experiment were purchased from Chemical Strategies, Inc. (Los Angeles, CA, USA). H<sub>2</sub>O<sub>2</sub> (30 wt% in water) was purchased from Honeywell (Morris Plains, NJ, USA), H<sub>2</sub>SO<sub>4</sub> (purity 96.0%) from KMG Chemicals, Inc. (Fort Worth, TX, USA), and *N*,*N*-dimethyl-formamide (DMF, Purity > 99%) from Sigma-Aldrich (Saint Louis, MO, USA).

Prior to SS spin-coating, the Si substrates were cleaned with acetone, methanol, and de-ionized (DI) water in that order, and submerged in a piranha solution ( $H_2O_2$ : $H_2SO_4$  = 1:4 volume ratio) for 15 min followed by a DI-water rinse and then dried in a spin rinse dryer at 1000 rpm for 10 min to provide clean and hydrophilic Si surfaces. The substrate when placed on the spin-coater is exposed to air under normal ambient laboratory conditions (Temperature: 20–23 °C, Humidity: 25–35%). The spin-coater was programmed with an acceleration speed of 100 rpm/s and maximum speed of 2000 rpm/s for 120 s. 300 µL of SS solution (630 mg/mL SS in DMF) was dispensed on the Si surface for spin-coating.

SS spin-coated Si were then moved to the reactive ion etching(RIE) chamber (Oxford PlasmaLab 80+) and the SS size reduction was performed using a  $CHF_3/Ar$  mixed gases at 25/25 sccm flow rate with 200 Watts of RF plasma power under 75 mTorr chamber pressure. After RIE, ultra-violet ozone (UVO) surface cleaning treatment was followed to remove an organic passivating layer on the Si surface resulting from the  $CHF_3/Ar$  RIE, and a Cr mask layer was deposited by using an e-beam evaporator (Lesker PVD-75). The SS layer on Si surface were then lift-off by dipping in a buffered oxide etch (10:1 BOE) solution for 10-min. The Cr masked Si substrate was then etched to fabricate

inverted pyramids using a 1% KOH solution at 85°C with 4% iso-propanol (IPA) added to improve etching uniformity [18,19].

Scanning electron microscopy (SEM, FEI XL-30, Pillips, Amsterdam, The Netherlands) was used to observe the Si surface and the SEM image was investigated to calculate the coverage percentage of the SS monolayer by direct counting of the SS coated area through "Image J" (National Institutes of Health, Bethesda, MD, USA) image analysis software [20], which has been widely used for particle related surface analysis [21–26]. Then, the contact angle of solvent on a Si surface was also measured for wettability analysis by using the EasyDrop contact angle measurement system (KRUSS, Hamburg, Germany).

#### 3. Results and Discussion

#### 3.1. Deposition of Silica Sphere Monolayer

Prior to SWSS fabrication, deposition of a uniform and high coverage SS monolayer must be realized. We, therefore, developed a DMF solvent introduced spin-coating method which makes SS solution preparation easy and results in enhanced SS coverage and uniformity even on expanded Si surface areas [27]. Unlike conventional SS deposition techniques such as Langmuir–Blodgett (LB) dip-coating [15,16] and water-based spin-coating methods [21,28], our spin-coating process offers greatly reduced environmental sensitivity to, for example, vibration, humidity, and temperature along with reduced processing time. This is because DMF has more spin-coating oriented solvent properties compared to the conventional solvent, water. Figure 1 schematically compares the behavior of DMF and water during the various stages of spin-coating. First, from Figure 1b, DMF can provide outstanding wetting ability on a Si surface. From our contact angle measurements, the contact angle of DMF on a piranha cleaned Si surface was immeasurably small. This is an essential solvent property for fast, uniform, and omni-directional SS spread-out during spinning, which is also followed by less material loss due to its low centrifugal force caused by low surface tension at the liquid/solid interface, schematically shown in Figure 1d. Second, DMF also offers a significantly lower vapor pressure (VP<sub>DMF</sub> = 2.7 Torr at 20 °C) compared to that of water (VP<sub>water</sub> = 17.54 Torr at 20 °C) providing increased SS deposition and distribution time during spin-coating due to its relatively slow evaporation rate, schematically illustrated in Figure 1f. This is also very important for spin-coating closely-packed SS assembly in a long range order (Figure 1h) since slower solvent evaporation produces intensified convective flux  $(J_w)$  of SS to the initially assembled SS (Figure 1f inset) [29]. However, water, due to its high vapor pressure (VP), requires humidity and temperature control or additional surfactant mixtures to delay the evaporation rate as previously reported [14,21]. Combining these two major advantages of DMF provides even less sensitivity to the spin-coating surface area, and we demonstrated more than 95% monolayer coverage even with much smaller SS (310 nm diameter) on a 2-inch round Si wafer, which is more challenging for spin-coating due to the stronger inter-particle interaction [13]. More details of DMF effect on SS spin-coating can be found in our previous study [27].

Figure 2 shows SEM images and a picture of a uniform spin-coated SS (1.57  $\mu$ m diameter) monolayer on the Si surface with more than 95% coverage that has been achieved through the use of the DMF solvent without any modification of the surface [17] or spin-coating environment [21]. Therefore, with our novel solvent-controlled spin-coating method, we provide highly uniform and enhanced SS monolayer coverage even on large Si surface areas just by using a simple and low-cost spin-coating method.



**Figure 1.** Schematic illustration to compare *N*,*N*-dimethyl-formamide (DMF) and water for spin-coating application.



**Figure 2.** (**a**) Picture of silica sphere (SS) coated Si wafer (2-inch diameter) and (**b**) scanning electron microscope (SEM) images of SS coated Si surface.

#### 3.2. Silica Sphere Size Control by RIE

One of the biggest advantages of utilizing SS for the nano-/micro-sphere lithography process is that silica material (i.e., SiO<sub>2</sub>) can provide high etching selectivity to other semiconductor materials, such as Si to adjust the scale of the surface structures. Currently, most reported nano-/micro-sphere lithography (NSL) processes utilize a polystyrene sphere (PS), which reveals a soft material nature that can be problematic to produce repeatable dimensions and well-defined patterns on the lithographic surface. In this report, we successfully demonstrated an enhanced SS size controllability with a RIE process and we believe that this can offer a great opportunity to fabricate SWSS in the desired dimensions.

For effective size reduction of SS, two criteria are required. They are: (1) high etching selectivity of SS to Si substrate; and (2) effective horizontal direction etching of SS followed by minimized vertical etching. The high etching selectivity of SS to Si would prevent excess etching of Si surface, which is crucial to produce SWSS fabricated surface geometry in a desired way. In addition, well-controlled horizontal direction etching of SS is required to adjust the dimension of a projected pattern after mask

layer deposition, but vertical direction etching must be minimized to permit clean SS removal and mask layer lift-off. Based on these considerations, a  $CHF_3/Ar$  gas mixture is used for the SS etching process and has optimized its etching conditions by changing gas flow rates, RF power, and chamber pressure. After the investigation of each RIE result, we found that the chamber pressure is the most critical knob to produce high SS etching selectivity and a low vertical/horizontal etching ratio. For the experiments, three different chamber pressures were studied, which are 50, 75, and 100 mTorr, and the results are shown in Table 1. From Table 1, we observed that noticeable Si substrate etching was made after etching at 50 mTorr, but as chamber pressure increases to 75 and 100 mTorr, the Si substrate etching was significantly reduced, which is possibly due to the decreased mean-free path length of ionized-species at higher pressure [30]. However, even though significantly reduced Si substrate etching was observed for both pressures of 75 and 100 mTorr, 100 mTorr revealed a higher vertical/horizontal etching ratio of 3.9 compared to 2.6 from 75 mTorr. As mentioned in the earlier section, this excess etching rate in the vertical direction is not ideal for SS lithography because fast vertical diameter reduction of SS compared to that of the horizontal direction restricts our choice on the pattern sizes, which weaken the ability of the SS to pattern the surface as SS size decreases. As shown in Figure 3, for the same SS target horizontal diameter, 800 nm, 100 mTorr chamber pressure produced a reduced vertical SS diameter of 260 nm, compared to 360 nm from 75 mTorr, while the horizontal diameters were sustained at 800 nm for both. Therefore, we decided to use 75 mTorr chamber pressure for SS size reduction with RIE.



Figure 3. SEM images of SS (a) as-deposited, and after reactive-ion etching (RIE) process at (b) 75 mTorr, and (c) 100 mTorr.

In Figure 4, we show 1.57 µm diameter SS after different RIE process times (10-min, 13-min, 16-min, 19-min) under our optimized RIE condition. From Figure 4a–d, we confirmed that highly effective SS size reduction is possible with minimal Si substrate etching. In addition, from Figure 4e, the measured inter-particle spacing with different etching times reveals that SS size can be easily reduced in a desired scale due to its reasonably linear relation between decreased SS diameter and etching time. Moreover, as we observed from Figure 4a–d, after RIE, there was still highly well-defined circle shape of SS without any rugged shape at the edge, which is a well-known issue with a soft polystyrene sphere [31,32]. Therefore, by using our optimal RIE condition, highly controlled SWSS in an intended structure dimension and period by introducing SS in a different diameter.



**Figure 4.** SEM images of RIE processed SS after different etching time (**a**) 10 min, (**b**) 13 min, (**c**) 16 min, and (**d**) 19 min, (**e**) graph indicating inter-particle spacing with etching time (All samples were etched at the same RIE condition which are 75 mTorr pressure, 200 Watts RF power, and 25/25 sccm gas flow rate except etching time) (scale bar:1 μm).

Table 1. SS shape profile ch	nange with CHF <sub>3</sub> /Ar	gas RIE after 10-min etc	hing under three different
chamber pressures.			
PressureProperty	50 mTorr	75 mTorr	100 mTorr

PressureProperty	50 mTorr	75 mTorr	100 mTorr
Cross-sectional SEM Image (Scale-bar:500 nm)		201	XX
Vertical Etching Rate (nm/min)	90.3	80.3	70.3
Horizontal Etching Rate (nm/min)	34.8	30.3	18.1
Vertical/Horizontal Etching Ratio	2.6	2.6	3.9

#### 3.3. Mask Layer Deposition and SS Removal

In the earlier sections, we demonstrated that  $CHF_3/Ar$  RIE could provide very effective selective etching of SS to the Si substrate and well controlled SS size reduction.  $CHF_3/Ar$  RIE process, however, also convey an inevitable issue for SS layer elimination after metal mask layer deposition. The challenge of this SS layer removal has been originated from the thin  $SiC_xF_y$  layer formation on the all exposed surfaces during RIE [33]. This carbon included  $SiC_xF_y$  compound layer is very stable, especially in a hydrofluoric acid (HF) based solution like BOE. As a consequence, removal of  $SiC_xF_y$  covered SS in BOE solution cannot be achieved. The  $SiC_xF_y$  layer effect for SS removal from Si surface is schematically shown in Figure 5c,e. As illustrated in Figure 5e with an actual SEM image, the presence of  $SiC_xF_y$  layer on the surface of SS resulted in a collapsed Cr metal mask layer during the BOE process and no surface patterning for surface structure fabrication. This is because the  $SiC_xF_y$  layer, therefore, we inserted an ultra-violet ozone (UVO) surface cleaning treatment, which is simple and required a no-vacuum process. This UVO treatment was performed before the Cr layer deposition. The UVO process provides not only effective cleaning on organic contamination, but also a non-aggressive etching environment to prevent possible surface damage inducing degraded solar cell performance. The effect of UVO surface treatment is clearly shown in Figure 5f. For the sample, a 10-min UVO treatment was performed before Cr layer evaporation, and as expected, the surface revealed very clean SS removal after BOE (10:1) etch, resulting in well-defined surface patterns for SWSS fabrication.



**Figure 5.** (**a**–**f**) Schematic illustration of ultra-violet ozone (UVO) treatment effect on SS removal after RIE. (Green-line indicates deposited Cr layer, and red-line for  $SiC_xF_y$  passivating layer formed during RIE process, scale-bar = 1 µm).

#### 3.4. SWSS Fabrication: Inverted Pyramid

For the first demonstration of SWSS from our designed lithography process, we decided to fabricate an inverted pyramid Si surface structure which is well known for its highly effective light trapping when it is built in an optimal subwavelength scale [7]. In this paper, we successfully demonstrated the effectiveness of our lithography process and the results are shown in Figure 6. With our process, we fabricated a ~1.1  $\mu$ m scale inverted pyramid with ~1.57  $\mu$ m period in a hexagonal array. (Note: Optically, this is not an optimal SWSS scale and density, but is intended to show the capability to control the size and density of fabricated SWSS.) To provide a stable masking layer on the Si surface for the KOH etching process, Cr metal layer has been chosen since it provides a reasonably robust material nature in both BOE and KOH solutions. In addition, Cr is a relatively cheap material that meets our intention to develop a low cost lithography process in this article. For Cr mask layer deposition, an e-beam evaporation system has been used and deposited a 50 nm thick Cr layer. The sample with Cr layer was, then, dipped in BOE (10:1) solution for SS layer etch. The SS removed Si sample was subsequently immersed in KOH (1%) solution for anisotropic etching of Si inverted

pyramid structure. The KOH solution was prepared after Si saturation and 4% IPA addition to provide moderate Si etching rate and enhanced etching uniformity for fabricated structures. After 2-min KOH etching at 85 °C, well dimension-defined Si inverted pyramids were produced as shown in Figure 6b–d. In Figure 6a, the diameter of patterned circle was ~1.1  $\mu$ m, and the lateral dimension of fabricated Si inverted pyramid is also approximately 1.1  $\mu$ m, as shown in Figure 6b. Furthermore, we also successfully confirmed that our SWSS fabrication process using a solvent-controlled SS lithography technique can produce highly uniform Si inverted pyramid over a large surface area in a desired dimension and density as shown in Figure 6c–e. In the study from Mavrokefalos et al., the optimized dimension of SWSS can produce a greatly enhanced absorption close to the Lambertian limit in the visible and near IR regions [7]. Therefore, future dimension and period optimization for SWSS will be able to produce broadband absorption enhancement for thin c-Si solar cell applications.



**Figure 6.** SEM images of (**a**) SS removed Cr mask layer deposited Si surface, (**b**) 1% KOH etched Si surface with Cr mask layer, (**c**) inverted Si pyramid without Cr mask layer, (**d**) cross-sectional image of inverted Si pyramid, and (**e**) fabricated inverted pyramids over large surface area to show its uniformity (Inverted pyramid has ~10% size variation likely caused by an initial SS size variation of 10%.). Note: (**a**–**d**) scale-bar = 1  $\mu$ m, and (**e**) scale-bar = 10  $\mu$ m.

#### 4. Conclusions

In this paper, a novel SS based lithography technique using a solvent-controlled spin-coating method has successfully demonstrated to deposit a uniform SS monolayer on large Si surfaces. Based on our SS deposition method, we designed a SWSS fabrication process targeting 1.1  $\mu$ m inverted Si pyramid surface structures with 1.57  $\mu$ m period in a hexagonal array. We successfully demonstrated that our process can provide a promising approach to fabricate SWSS in controlled-scale even on a relatively large Si surface areas. Furthermore, from the results, we showed that our process offers great flexibility in varying the target size and density of the SWSS by using a simple RIE process and different size SS. Therefore, our process potentially provides a more realistic approach for surface

texturing of thin c-Si PV cells with broadband absorption improvement. Currently, we are working on optical modeling to determine the optimal size of SWSS for enhanced light absorption for thin c-Si solar cell application, and the results will be reported shortly.

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