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Chemical Bond Formation between Vertically Aligned Carbon Nanotubes and Metal Substrates at Low Temperatures

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Abstract: The exceptional physical properties of carbon nanotubes (CNTs) have the potential to transform materials science and various industrial applications. However, to exploit their unique properties in carbon-based electronics, CNTs regularly need to be chemically interfaced with metals. Although CNTs can be directly synthesized on metal substrates, this process typically requires temperatures above 350 °C, which is not compatible for many applications. Additionally, the CNTs employed here were highly densified, making them suitable as interconnecting materials for electronic applications. This paper reports a method for the chemical bonding of vertically aligned CNTs onto metal substrates that avoids the need for high temperatures and can be performed at temperatures as low as 80 °C. Open-ended CNTs were directly bonded onto Cu and Pt substrates that had been functionalized using diazonium radical reactive species, thus allowing bond formation with the open-ended CNTs. Careful control during grafting of the organic species onto the metal substrates resulted in functional group uniformity, as demonstrated by FT-IR analysis. Scanning electron microscopy images confirmed the formation of direct connections between the vertically aligned CNTs and the metal substrates. Furthermore, electrochemical characterization and application as a sensor revealed the nature of the bonding between the CNTs and the metal substrates.

Keywords: carbon nanotubes; metal-carbon interface; bond formation

1. Introduction

Carbon nanotubes (CNTs) are macromolecules whose discovery, arguably attributable to Professor Sumio Iijima [1,2], has provided heretofore unimagined potential for engineering applications. CNTs have garnered immense research interest because of their unique structure and physical properties [3–5]. At the nanoscale level, they exhibit very high strength and electrical and thermal conductivities [6–8]. Single-walled CNTs have been shown to have a Young's modulus of greater than 1 TPa [9], with an electrical resistivity as low as $3 \times 10^{-7} \Omega$ m [10] and a thermal conductivity as high as 3000 Wm K⁻¹ [11,12]. Furthermore, CNTs have been reported to have a large ampacity compared with metals, suggesting their untapped potential in electronics [13]. In addition, the heat dissipation capabilities of CNT arrays as thermal interfaces have been demonstrated [14]. Several researchers have attempted to prepare CNT/Cu composites with varying degrees of success [15–17], but in order to take advantage of CNTs' physical properties, significant efforts have been devoted to growing CNTs on metal substrates in order to achieve chemical bonding [18–20].

Chemical vapor deposition (CVD) has been adopted as the most effective and suitable method for synthesizing vertically aligned CNTs on metals, but traditional CVD requires temperatures above 650 °C to produce high-quality CNTs. It has been reported that high temperatures negatively affect the lifetime of the catalyst nanoparticles by promoting catalyst ripening, carbide formation, alloying, and coarsening [21,22]. Both the essential necessity of an Al_2O_3 support during synthesis and the negative effect of its dielectric nature



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Copyright: © 2021 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). on limiting the electron transport process have been demonstrated [23]. High-density CNT arrays that can support interconnections have been developed [24–26]. However, the creative approaches required to synthesize CNTs directly on metal substrates, including Cu, Al, Ti, Ta, and stainless steel, demonstrate the challenges involved in growing high-quality CNTs [18,26–28]. In addition, experimental metal alloy combinations for interfacing through traditional soldering have been reported [29,30]. Although synthesis of CNTs at temperatures as low as 350 °C have been accomplished, increasing their density has shown itself to be a challenge [31–33]. In general, interfacing metals with organic molecules or solvents is a major challenge in material development, especially for CNTs because of issues such as the nanotube dimensions, orientation, and wettability. In most efforts to grow CNTs directly on metal substrates, the CNT quality has been compromised by attempts to reduce the temperature and oxide catalyst support thickness [27,28]. Other interfacing approaches such as the self-assembly of monolayers [14,34], soldering [29,30], colloidal metallic pastes [35], and electrodeposition have also been employed, but they all have shown limitations.

Despite almost three decades of intense CNT research, most synthesis methods have failed to meet expectations due to challenges associated with CNT synthesis, controlling CNT type, chirality and diameter [36–39]. Unsatisfactory results have been achieved for several electrical and thermal applications based on current CNT assemblies. One reason for this poor performance includes the ability to assemble CNTs into macroscopic fibers and films with similar properties [40,41], as well as the assumption that traditional macroscopic contact interfacing methods are sufficient to connect most carbon nanomaterials to metal surfaces. High thermal and electrical interface resistances have been reported between CNTs and their contacts, owing to weak adhesion of the nanoscale paths [14,34,42,43]. Interfaces between dissimilar materials often control phonon and electron transport, especially at the nanoscale level where the importance of the interfaces relative to bulk material properties is substantially increased [42,44,45]. CNTs, like most nanomaterials, require a metal support that can efficiently harvest electrons and thus take advantage of their physical properties [21,46]. Notably, short covalently bonded organic molecules at the interface can dramatically increase adhesion between CNT arrays and noble metal contacts, resulting in an approximately sixfold reduction in the thermal interface resistance [14].

Because of their geometrical shape, the conducting properties of CNTs depend on their orientations in assemblies, with true physical properties being lost at the interfaces between the anisotropic nanotubes and metal substrates [47–49]. The electrical conductance of a CNT is closely related to its orientation and interface contact [50]. Based on extensive work with highly ordered pyrolytic graphite, the basal and edge plane reactivities [51] and the electrochemical behavior of CNTs have been shown to differ [52,53]. Various reports have indicated that the edge planes of CNTs have higher electron transfer rates than the basal planes [51,53,54]. Furthermore, the intrinsic metallic or semiconducting properties of CNTs, as well as their single- or multi-walled nature, can lead to discrepancies in the measured electron transfer rates. Theoretical studies on simulating the interface between individual CNTs and metals have found great resistance at the metal–CNT interfaces with simple contact [55–57]. More recently, systematic methods to control, engineer, and study open-ended CNTs have been developed, and the applications of CNT tip reactions have led to the development of highly sensitive sensors [35,58].

This paper reports a method for chemically joining open-ended CNTs to metal substrates (Cu or Pt). For this purpose, high-density CNTs (HD-CNTs) with functionalized open ends were positioned orthogonally to a metal substrate. The metal substrate was also functionalized with suitable organic linkers to promote a chemical junction between the carboxylic and amine groups at the ends of the CNTs and metal substrates, respectively. Successful cross-linking between the CNTs and the linkers on the metal surface was demonstrated via conductivity measurements and electrochemical characterization.

2. Materials and Methods

2.1. Materials

CNTs were synthesized as vertically aligned arrays that would allow assembly into fibers or threads (Figure 1A). Due to its self-assembly capabilities, this type of vertically aligned CNT is also called spinnable or drawable. The CNTs, which had lengths of approximately 400 μ m, were vertically aligned in a forest format. The catalyst film thickness, precursors, and CNT synthesis parameters have been published by our group elsewhere [41]. The as-synthesized CNT arrays were completely detached from the substrate during synthesis to eliminate catalyst contamination, and the CNT fibers with a 70- μ m diameter were spun and subsequently densified with acetone before embedding in a polymer.



Figure 1. Characterization and fabrication of HD-CNT. (**A**) SEM image of a CNT fiber. (**B**) CNT fiber encapsulation within a polymer and subsequent sectioning. (**C**) SEM image of a cross-section of the resulting HD-CNT.

Lithium trifluoromethanesulfonate (CF₃SO₃Li, 95%) was obtained from Ambeed Inc., sodium nitrite (NaNO₂, 99.4%) was obtained from Chem-Impex International Inc., and hexaammineruthenium(III) chloride ([Ru(NH₃)₆]Cl₃, 98%) was obtained from Acros Organics. Sodium acetate (NaOAc·3H₂O, 99.0%) and acetic acid (CH₃COOH, 99.0%) were purchased from Fisher Scientific. Ethylenediamine (99%), nitric acid (HNO₃, 95%), hydrochloric acid (HCl, 37%), potassium chloride (KCl, 99%), acetone (99.7%), ethanol (99.8%), potassium ferricyanide (K₃[Fe(CN)₆], 99%), lead standard (1000 mg L⁻¹), and acetonitrile (99%) were purchased from Electron Microscopy Sciences and mixed following the manufacturer's instructions.

2.2. Carbon Nanotube Fiber Cross-Section Preparation and Bonding to Metal

To position enough CNTs orthogonal to the substrates, we employed aligned CNTs assembled within a fiber section. Embedding or encapsulating sections of aligned CNTs within polymers allowed the fabrication of HD-CNT assemblies, which was accomplished using a clear polymer (Embed-812) [58]. Ultramicrotoming of the polymer-embedded CNT fibers was performed to create open-ended HD-CNTs within a 40- μ m thick film (Figure 1B), where the CNTs passed through the film and had an open end on both sides of the film. A scanning electron microscopy (SEM) image of the HD-CNT cross-section for a single CNT fiber is shown in Figure 1C. This process gave access to the open ends of the CNTs while limiting the reactions at their cross-sections, as published by our group elsewhere [35,58]. Additionally, embedding in a polymer maintained the CNTs in a vertical position and allowed convenient manipulation, including applying pressure to create intimate contact with metal surfaces for bonding. The open ends of the fiber cross-sections were treated with HNO₃ (2 M) at 70 °C for 24 h for additional functionalization with the carboxylic groups.

2.3. Modification of the Metal Surface

Cu and Pt metal sheets were polished with 600-grit silicon carbide paper followed by crystal polishing paper, sonicated in acetone for 30 min, and then washed with deionized water and isopropyl alcohol before the electrochemical–chemical reaction. For CNT–Cu bonding, it was important to minimize Cu oxidation; therefore, acetic acid treatment was employed. The 4-aminophenyl diazonium cations generated in situ by the reaction of *p*-phenylenediamine and NaNO₂ in an acidic medium were used as the precursor, which were known to reduce to a highly reactive radical capable of bonding to metal surfaces (Figure 2A) [59–61]. For CNT-Pt bonding, ethylenediamine was electrochemically grafted on the Pt surface in acetonitrile as a solvent similar to that reported by Segut et al. (Figure 2B) [62]. Herlem et al. has used a similar electrochemical grafting reaction to modify metal electrodes with ethylenediamine [63,64]. Cyclic voltammetry at a scan rate of 50 mV s⁻¹ was used to identify the oxidative grafting of ethylenediamine on a Pt sheet electrode in acetonitrile with an Ag pseudo-reference electrode (Figure S1).



Figure 2. Proposed mechanism for the chemical bond formation between metals and open-ended CNTs. (**A**) In situ generation and attachment of an amine functional group to a Cu surface and subsequent bonding to a carboxylic acid functional group at the open end of a CNT. (**B**) Electrochemical attachment of ethylenediamine to a Pt surface and subsequent bonding of the amine end to the carboxylic acid at the open ends of the CNTs.

Subsequently, the amine-grafted metal surface and carboxylic-functionalized HD-CNT cross-section within the film were clamped together and heated to ~80 °C to promote a reaction between the surface functional groups [14].

2.4. CNTs Bonded to Metal as a Working Electrode

Electrochemical characterization and electrical conductivity measurements of the chemically bonded CNTs to Cu metal were performed on electrodes assembled as shown in Figure S2. A single HD-CNT within a film (\sim 70 µm in diameter and \sim 40 µm thick) with open-ended CNTs was connected to a metal and employed as a working electrode. To investigate the stability and electrochemical behavior of the fabricated electrodes, cyclic voltammetry measurements of [Ru(NH₃)₆]Cl₃ with 0.5 M KCl as a supporting electrolyte were performed in the potential window from 0.1 to -0.5 V. Stripping voltammetry is an effective technique for heavy metal detection with various kinds of electrodes. Therefore, stripping analysis was performed for Pb^{2+} detection in the potential window of -1.2-0 V, with CNT-Cu chemically bonded electrodes as the working electrode. Commercially available $Pb(NO_3)_2$ in a 2% HNO₃ solution was used to prepare the Pb^{2+} solutions in a 0.1 M acetate buffer (pH 4.3) at concentrations of 20–50 ppb, and calibration curves were constructed based on the current intensities recorded for the Pb²⁺ solutions. The following optimized square wave anodic stripping voltammetry (SWASV) parameters were used: frequency, 15 Hz; amplitude, 0.025 V; deposition potential, -1.2 V; potential step, 0.004 V; and deposition time, 120 s. The stripping peaks were used to quantify the Pb concentration.

2.5. Instrumentation

SEM images were recorded using an FEI Apreo scanning electron microscope at acceleration voltages of 5–15 kV to visualize the CNT cross-sections on the metal surfaces. An EDAX elemental analysis detector coupled with the same SEM instrument was used to analyze the atomic compositions of the samples. The Raman spectra of HD-CNTs attached to the metal surface were collected using a Renishaw inVia Raman microscope with a HeNe laser as the excitation source (633 nm). FT-IR spectra were collected using a Nicolet 6700 FT-IR instrument to identify the surface functionalization. All electrochemical analyses were performed using a PalmSens3 potentiostat/galvanostat. A three-electrode system was used with a Ag/AgCl reference electrode and a Pt wire counter electrode unless otherwise specified.

3. Results and Discussion

Freshly microtomed HD-CNTs from the same fiber were functionalized at their open ends. For this purpose, positioning CNTs while maintaining their orientations and protecting their side walls within the assembly was required. We accomplished this by assembling CNTs into macroscopic fibers, where the CNTs had linear orientations and maintain their pristine nature of individual CNTs while their side walls were encapsulated within a polymer [40,41]. Aside from protecting the side walls, the encapsulating polymer prepared as film facilitated handling of the microscale HD-CNTs without ever making contact with their open ends. The CNTs in Figure 2A,B illustrate the carboxylic functionalized ends that were employed to react with amino functional groups at the Cu and Pt surfaces. In general, covalent bond formation between CNTs and other molecules requires functionalization of the CNT surfaces. A variety of methods for CNT covalent functionalization have been reported, including free radical addition, oxidation, carboxyl-based coupling, fluorination, and addition or substitution reactions. These functionalization reactions have mainly been used to enhance CNT dispersion and compatibility [65,66]. Typically, functionalization occurs at the edges of five- or seven-membered ring defects in the basal planes of the CNTs or the tip of the CNTs [66,67]. Unfortunately, most of those approaches for functionalization do not allow the positioning of functional groups to be controlled. With the current method, access to all carbon atoms at the open-ended CNTs was made easier, and their complete functionalization would be ideal. To date, quantification of the degree of functionalization has been limited to theoretical studies, where the optimum number of functional groups at the open ends has been determined [68].

The thin films obtained by microtoming that held the HD-CNTs had two important functions: (1) holding the CNTs of a uniform length packed in a cylindrical structure so that the majority of the continuous CNTs pass through the film and have open ends on both sides of the film and (2) covering the sidewalls of the CNTs so that only the open ends of the CNTs are exposed, which allows for chemical modification of the CNT tips, conserving the pristine CNT structure. It was assumed that the number of open-ended CNTs in the HD-CNT film samples were similar because multiple samples could be prepared from the same CNT fiber using microtoming. This is a unique advantage of this approach, because it allows the number of CNTs assembled within a particular fiber to be controlled by limiting the width of the CNT array employed for fiber assembly, as demonstrated previously [40,69]. Wet chemistry functionalization is preferential at the open ends of CNTs [70,71]. We selected carboxylic functionalization at the open ends of the CNTs, because carboxylic acid functionalization is a well-established single-step functionalization reaction for CNTs which gives a reactive end for a second molecule attachment [72]. To achieve a high carboxylic density on the CNT open ends, the functionalization time with HNO_3 (2 M) was increased to 24 h. The Raman spectra were recorded for CNT crosssections chemically attached to metal surfaces (Figure S3). The peak at \sim 1585 cm⁻¹ was attributed to the G band, which originated from the in-plane tangential stretching of the C–C bonds in CNTs, whereas the peak at ~1334 cm^{-1} was attributed to the D band. The ratio between these bands (I_G/I_D) indicated a high degree of functionalization. However, as the assemblies contained multi-walled CNTs with a relatively high D band intensity, quantitative information about functionalization could not be obtained by comparing only the (I_G/I_D) ratio. However, EDAX analysis consistently supported an increase in the oxygen content, although by no more than 2%.

Aside from CNT open-end functionalization, suitable functional groups at the metal surface are needed in order to chemically link CNTs to metal surfaces. Metal surface functionalization was accomplished using organic radical metal reactions, also known as grafting. To realize bond formation between a carboxylic functionalized CNT tip and a metal, the metal surface was functionalized with the amine groups (Figure 2A,B). Amine functionalization of the Cu surface was achieved using a spontaneous reaction between a p-aminobenzenediazonium cation and Cu metal, which left the chemically bonded aminophenyl group on the Cu surface in a similar manner to that reported by Chamoulaud et al. [60]. In contrast, the Pt surface was electrografted by short ethylamine groups with ethylenediamine as described in the experimental section. Then, to promote bond formation between the CNTs and the organic groups grafted on the metal surfaces, functionalized open-ended CNTs were pressed against the metal surfaces using small magnetic discs during the reaction while the temperature was increased. The electrografted organics on metals acted as linkers to join the open-ended CNTs. This type of metal functionalization using reactive organic molecules is a topic of intense research. Several metals, including stainless steel, Ni, Au, and polycrystalline Cu, have been functionalized using aryl diazonium cations (R-N₂⁺). Anthracene, anthraquinone, and hydroquinone have been covalently bonded to metal surfaces, presumably through the formation of carbides and nitrides [73].

As shown by the reaction mechanism in Figure 2A, upon reduction, the diazonium salts generated strong radical species that could bond to metal and carbon surfaces [74]. p-Phenylenediamine reacted with NaNO₂ and HCl to generate the *p*-aminobenzenediazonium cation in situ as described by Lyskawa et al., which was spontaneously grafted onto the Cu surface to generate aminophenyl groups [75]. Spontaneous grafting will occur if the surface of the substrate is sufficiently reduced to convert the diazonium salt to a radical that can react with the same surface. In addition, there is the potential to be applied to promote a reaction between *p*-aminobenzenediazonium cations and metals such as Pt and Au [76]. The grafted aminophenyl groups on the Cu surface reacted with the carboxylic groups on the CNT open ends, which were obtained by CNT oxidation. Although the amine-carboxylic coupling reactions employed in this work were aimed at covalent bond formation between functional groups at the metal surface and open-end CNTs, the nature of the resulting bonding was not possible to determine. As a result of those challenges, "chemical bond" is used throughout the text instead of covalent bonding. The expected amide formation resulting from amine-carboxylic coupling is localized between macroand micro-structures, where the access is limited.

Covalent bonding of ethylenediamine on the Pt surface was accomplished via electrografting (Figure 2B). The highly reactive ethylenediamine radical is known to attack metal surfaces, leaving an amine functional group available for subsequent reactions. Similar bonding has been reported by Adenier et al., and a mechanism of bond formation between metals and organic moieties has been reported [73]. Upon the electrochemical oxidation of primary amines using Pt metal as a working electrode, bond formation and the growth of a continuous film was observed. Unfortunately, due to the polymerization of ethylenediamine on the working electrode surface, more than a monolayer buildup was observed, which resulted in an electrical passivation of the metal surface. Although further investigations to identify the thickness of the interface between Pt and the CNTs are pending, bonding of ethylenediamine has been confirmed by several other researchers.

FT-IR spectral analysis on the functionalized Cu supported the organic moiety grafting to the Cu surface. Spectra were recorded before and after grafting on the Cu surface to identify the functional groups attached to the metal surface. Furthermore, the spectra after sonication of the samples to determine the bonding strength between grafted organic groups and the metal surface were collected (Figure 3). The FT-IR spectrum of the aminophenyl-grafted Cu surface exhibited a weak broad peak at ~3400 cm⁻¹, which was attributed to N-H stretching [77]. A similar broad peak has been observed in the FT-IR spectra of polymerized diamines, whereas crystalline diamines exhibit sharp peaks in this region [77,78]. Clear peaks were observed in the 1350–1610 cm⁻¹ region, specifically at 1496, 1511, and 1608 cm^{-1} , which were attributed to C=C bond stretching in the aromatic rings [60,77,78]. The band observed at 1259 cm⁻¹ was attributed to C-NH₂ stretching, whereas that at 1178 cm⁻¹ was attributed to CH bending. The peak at 831 cm⁻¹ corresponded to CH out-of-plane bending, and that at 630 cm⁻¹ could correspond to ring deformation [78]. The observation of peaks around 2300 cm^{-1} indicated the presence of trace CO_2 in the atmosphere. No peak corresponding to the N \equiv N stretching mode of diazonium at 2280 cm⁻¹ was observed, which supports the proposed metal–organic reaction [60]. The peak intensities decreased as the reaction temperature decreased from 65 to $25 \,^{\circ}$ C on the FTIR peaks in Figure 3, which suggests that this grafting reaction is dependent on the temperature. Alternatively, this behavior could be due to multilayer formation, owing to the reactions between in situ generated diazonium molecules. For comparison purposes, the FTIR of pure 4-phenylenediamine was recorded (Figure S4), where the main features discussed in Figure 3 are also visible. The covalent nature of Cu-C bonds has been reported by McCreery et al., as has multilayer formation on Cu substrates via diazonium salt-Cu metal reactions [61]. However, in our work, strong peak intensities were observed after sonication, which suggests the formation of strong bonds between the metal surface and the organic groups. Furthermore, an EDAX analysis (not shown) indicated that the atomic percentage of C increased upon grafting of the organic groups to the metal surface.



Figure 3. FTIR spectra of organic grafted Cu surface. Normalized FTIR spectra of (**a**) polished Cu surface, (**b**) aminophenyl group attached to Cu surface at 25 °C, (**c**) aminophenyl group attached to Cu surface at 25 °C after sonication, (**d**) aminophenyl group attached to Cu surface at 65 °C, and (**e**) aminophenyl group attached to Cu surface at 65 °C after sonication.

As shown in Figure 4A, vertically standing CNTs bonded to Cu metal were obtained from promoting the bond formation between the aminophenyl groups grafted on the metal

surface and the oxidized CNT open ends. Afterward, the embedding polymer around the HD-CNTs was physically removed using forceps while the CNTs remained anchored on the metal surface. The forceful removal of the polymer appeared to tear the CNTs at the outer surface of the HD-CNTs, but the number of removed CNTs was minimal considering the large number of them on the cylindrical structure that remained standing (Figure 4A). The three-dimensional nature of the assembly was clearly observed, and the interface between the CNTs and the Cu metal surface is visible in Figure 4B. Similar attempts with physisorbed embedded CNTs typically removed both the polymer and the CNTs at once, suggesting that the chemical bonding to the metal substrate in the HD-CNT samples was extremely strong for the cross-linked HD-CNTs compared with simple adhesion. The interface between the CNTs and the Cu substrate is shown at a higher magnification in Figure 4B, and the connections joining the CNT bundles to the Cu surface are clearly visible. Figure 4B also displays the surface of the Cu, where a rough coating-like surface is visible, suggesting that more than a monolayer of aminophenyl was present after grafting, also supporting the FT-IR spectrum in Figure 3. Ethylenediamine and *p*-phenylenediamine were both employed to functionalize the metal surface, but due to its electron-rich and electrically conductive nature, *p*-phenylenediamine is preferred.



Figure 4. SEM images of HD-CNTs bonded to a solid Cu and Pt surfaces after removing the encapsulating polymer. (**A**) Low-magnification and (**B**) high-magnification images showing the three-dimensional nature of the CNTs assembled on the Cu surface. (**C**) Low magnification and (**D**) high magnification of CNTs connected to the metal Pt surface.

Similarly, a cylindrical structure of vertically oriented CNTs remained attached to the flat Pt surface after removing the encapsulating polymer (Figure 4C), which suggests that the interaction between the CNTs and Pt was also strong. The junction between the Pt

surface and the CNTs is displayed in Figure 4D. Despite the roughness of the Pt metal surface, the pressure applied to the HD-CNTs during the reaction between the amine and carboxylic groups was sufficient to connect the CNTs to the Pt substrate via linker molecules.

Additionally, after sonication of the Cu metal substrate with bonded HD-CNTs, some CNT fragments remained on the metal substrate, as observed by SEM (Figure S5). These CNT fragments were located in the area where the HD-CNTs were originally bonded to the Cu substrate, suggesting that the degree of bonding between CNTs was extremely strong, but not all CNTs were chemically bonded to the Cu. The presence of CNTs on the metal surface after sonication was also confirmed by Raman spectroscopy, as the G and D bands typical of CNTs were observed. However, the fact that only small bundles of CNTs remained after sonication suggests non-uniformity among the CNTs bonding to the Cu surface. This was possible due to the roughness of the Cu substrate and lack of atomically flat HD-CNT cross-sections. In a similar experiment with a Pt substrate, it was difficult to locate clean CNTs due to some coating that occurred after sonication.

Successful metal–CNT bonding was also confirmed using electrochemical methods. CNT–Cu bonded electrodes were prepared in which only the HD-CNT cross-section was exposed to the electrolyte. Typically, $Ru(NH_3)_6^{2+/3+}$ is a good indicator for carbon electrode surfaces and electrolyte interactions. The absence of sharp oxidation or reduction peaks in the background cyclic voltammogram (CV) confirmed the inert nature of the tailored electrode surface for oxidation or reduction in the presence of a supporting electrolyte (Figure 5), which is characteristic of CNT cross-sectional electrodes. This observation suggests that electron transfer at the working electrode can only occur through the CNT cross-sections available to the solution. In addition, the interface between the Cu metal and HD-CNTs must be electrically conductive to allow electron transfer between them. Furthermore, the observation of characteristic oxidation and reduction peaks in the cyclic voltammogram of the ruthenium redox couple confirmed the electrically conductive nature of the CNT–Cu bonded electrode, attributable to a stable electrically conductive joint between the CNT cross-section and the metal substrate (Figure 5).



Figure 5. Electrochemical characterization of CNTs bonded to metal surfaces. Cyclic voltammograms of CNTs bonded to Cu as the working electrode: red and black lines = background response in 0.5 M KCl aqueous solution; pink and blue lines (pink barely visible under the blue) = response for 2 mM $Ru(NH_3)_6^{2+/3+}$ in 0.5 M aqueous KCl solution. The pink line corresponding to 2 mM $Ru(NH_3)_6^{2+/3+}$ in 0.5 M aqueous KCl has been replotted as an inset to make it visible.

As a benchmark, the electrochemical performance of freshly microtomed HD-CNTs connected to a metal surface using colloidal Ag paste was compared with that of CNTs co-

valently bonded to the metal surface. In addition, a physiadsorbed HD-CNT cross-section to Cu metal was also characterized, but the results were significantly inconsistent. The covalently bonded to Cu and Pt and Ag paste-connected CNTs displayed very similar CV characteristics, suggesting good electrical contact between the CNTs and metals. The contact effectiveness with the metal surface was evaluated using cyclic voltammetry and the electroactive surface area, as determined using the Randles–Sevcik equation [79], which was similar to the geometrical surface area. To determine the heterogeneous electron transfer rates (k° , cm s⁻¹), cyclic voltammetry experiments were performed in 2 mM of $Ru(NH_3)_6^{2+/3+}$ with 0.5 M KCl as a supporting electrolyte in distilled water at scan rates of 10–20 mV s⁻¹. As can be seen in Figure 5, the covalently bonded HD-CNTs displayed a sigmoidal steady state limiting current with a magnitude of ~17 nA. These are typical characteristics of hemispherical diffusion at a reduced diameter of microelectrodes. The steady state behavior of both redox species at a scan rate of 10 mV s^{-1} was determined in a similar manner to our previous work, in which CNTs were connected with Ag paint [58]. The peak current response increased as the scan rate increased, further confirming that radial diffusion occurred at the electrode-electrolyte interface [58]. In addition, the electrode response was evaluated at increasing potentials. The electrodes generated reproducible cyclic voltammetry responses in the potential range from +1 V to -1.25 V. Furthermore, an $E_{1/4} - E_{3/4}$ wave potential difference of 59 mV was observed for the open-ended CNTs connected through covalent bonds to the metal.

Theoretical studies and a few experimental reports have shown the enormous potential of open-ended CNTs connected to metals [50,55,58,80]. Electrical resistivity of 9 \times 10⁻⁷ Ω m was reported for CNTs connected to Au, which is two orders of magnitude higher than that achieved using simple contact, conducting almost as much as Cu might [46]. To evaluate the impact of covalent bonding HD-CNT to metal, the electrical resistances of the covalently bonded CNT cross-section electrodes were determined using a sensitive picoamp meter. The covalently bonded CNT cross-sections were compared to connections made using colloidal Ag paint and simple pressure contact (physiadsorbed HD-CNTs). In all cases, sealing with epoxy was applied in a similar manner to all covalently bonded HD-CNTs. The average resistance measured for covalently bonded CNTs was 305 Ω , whereas that for colloidal Ag-connected CNTs was 151 Ω , and the pressure contact samples gave inconsistent measurements throughout five different samples. The difference, although minimal, was attributed to the nanoscale roughness of the metal surfaces and required optimization in order to guarantee that all dangling carbons at the circumference of the open ends were bonded. Although our results are not close to the theoretical published results, to the best of our knowledge, this is the first attempt to covalently bond open-ended HD-CNTs to metal substrates.

Finally, the sensing capabilities of the CNTs covalently bonded to metal surfaces were demonstrated by detecting Pb^{2+} ions in an aqueous solution (Figure 6). The fabricated CNT–Cu bonded electrodes were able to detect low concentrations of Pb^{2+} , and a linear relationship was observed between the Pb^{2+} concentration and the peak current response ($R^2 = 0.9973$). These results are extremely important because they confirm the possibility of using CNTs bonded to a Cu metal surface via wet chemistry functionalization to develop electrochemical sensors for heavy metals. The observed behavior suggests that effective electrical contacts can be created at low temperatures, providing an alternative to surface melting using noncontact induction heating, which has previously been employed to achieve CNTs bonded to Au metal for microelectronics [46].



Figure 6. Sensing capability of CNTs bonded to metal surfaces for Pb²⁺ detection. The peak current response from the SWASV analysis was extracted to construct a calibration curve using CNTs bonded to Cu as the working electrode.

The scope of this work was limited to Cu and Pt surfaces, but more metals should be explored for comparison purposes. In particular, owing to their broad potential industrial applications, CNTs bonded to Cu and Al are expected to be most relevant. Metals that form more bonds during metal grafting should maximize the number of bonds formed with dangling functional groups on open-ended CNTs. According to theoretical studies and experimental reports, N- and C-based linkers on metal substrates should be advantageous, and such molecules have already been shown to form nitrates and carbides [73,81]. For N–Cu bonding, we selected oxidative electrografting for amine functionalization, preferentially with primary amines [82]. In addition, we selected spinnable CNTs to prepare HD-CNT assemblies because of their simplicity. Furthermore, assembling fibers from spinnable CNTs is among the cleanest methods because catalyst particles can be removed during synthesis [41] and no further processing is required, which minimizes potential contamination.

4. Conclusions

Chemical bond formation between CNTs and metal supports at low temperatures provide ideal contact for efficient electron transfer between metals and CNTs, thus opening a path for CNT application in electronics. This approach provided control of both the CNT orientation and the chemistry at the CNT–metal interface. Due to the high density and vertical arrangement of the CNTs, the bonded CNTs are suitable to work as interconnects. The successful covalent bonding of HD-CNT assemblies to Cu and Pt was confirmed, and high-resolution SEM revealed the nature of their connections. Furthermore, electrochemical characterization demonstrated the conductivity and sensing capabilities of the new metal–CNT interfaces. Appropriate functionalization at the opposite ends of the CNTs will allow the development of highly sensitive electrochemical sensors, realizing efficient charge transfer to the solution and tissue and producing high-energy-density and high-powerdensity energy storage devices.

Supplementary Materials: The following are available online at https://www.mdpi.com/article/ 10.3390/app11209529/s1, Figure S1: Cyclic voltammograms recorded on a standard Pt electrode, Figure S2: Schematic showing the process used to fabricate the CNT bonded to metal electrode, Figure S3: Raman spectra of Cu surface after CNT attachment, Figure S4: FTIR spectra of pure 4-phenylenediamine, Figure S5: High magnified SEM images of CNT attached Cu metal surface after sonication. **Author Contributions:** Conceptualization, N.T.A.; methodology, N.T.A. and C.P.N.; validation, formal analysis, N.T.A., C.P.N., A.H., C.K.R.; investigation and data curation, C.P.N., A.H., C.K.R. and C.E.R.; writing—original draft preparation, C.P.N. and N.T.A.; writing—review and editing, C.P.N., A.H., C.K.R., C.E.R. and N.T.A.; supervision, project administration, and funding acquisition, N.T.A. All authors have read and agreed to the published version of the manuscript.

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