

Article

A Link between Neutron and Ion Irradiation Hardening for Stainless Austenitic and Ferritic-Martensitic Steels

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Abstract: Radiation hardening is studied for stainless austenitic and ferritic-martensitic chromium steels after ion and neutron irradiation at various temperatures. Austenitic and ferritic-martensitic steels irradiated up to 30 dpa in various nuclear reactors and ion accelerators are studied at various temperatures. A change in Vickers microhardness is used as the radiation hardening parameter. A methodology is developed that allows one to determine the ion irradiation parameters, which ensure the radiation hardening of ferritic-martensitic and austenitic steels, as close as possible to the radiation hardening of the same steels under neutron irradiation. A transferability function is introduced to connect the irradiation temperatures for ion and neutron irradiation that provides the same radiation hardening. On the basis of the obtained experimental data, after ion and neutron irradiation the transferability functions are determined for the investigated austenitic and ferritic-martensitic steels, which connect the temperatures for ion and neutron irradiation and provide the same radiation hardening at a given damage dose.

Keywords: radiation-induced hardening; neutron and ion irradiation; ferritic-martensitic steels; austenitic steels; Vickers microhardness



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1. Introduction

One of the promising directions of radiation materials science is the modeling of the neutron irradiation effects in materials by ion irradiation in accelerators. The application of ion irradiation instead of neutron irradiation can significantly reduce the time for irradiation and the cost. Moreover, further investigation of the ion-irradiated material is easier than of the neutron-irradiated material.

At present, a large number of studies is being carried out on the use of ion accelerators for modeling radiation damage in structural materials under neutron irradiation.

The methodology of conducting simulation tests is presented in detail in the monographs [1,2]. A new wave of research started in the frame of the IAEA program “Accelerator simulation and theoretical modeling of radiation effects in structural materials” [3]. In addition, the simulation testing program “Simulation of Neutron Damage for High Dose Exposure of Advanced Reactor Materials” was launched in the USA [4]. The current state of simulation tests is presented in [5–9]. The standard for accelerated testing developed by ASTM [10] has already undergone several revisions. Simulation experiments on ion beams and subsequent studies are being conducted in a significant number of research centers. In Russia, such studies are conducted mainly in the Institute for Physics and Power Engineering, National Research Center “Kurchatov Institute”—Central Research Institute of Structural Materials “Prometey” and NRC “Kurchatov Institute”—Institute for Theoretical and Experimental Physics.

The application of ion irradiation in accelerators allows one to assess the degradation of the studied material at high values of the damage dose (100 dpa and above). The duration of irradiation is reduced from several years in the reactor to several hours in the ion accelerator due to the much higher damage dose rate in accelerator compared to reactor

irradiation. For example, it takes about 5 years to obtain the damage dose of 100 dpa in material with a BOR-60 fast neutron research reactor. The same dose can be provided in a few hours when the material is irradiated in the ion accelerator.

Another undoubted advantage of ion irradiation compared to neutron irradiation is the absence of induced activity in the irradiated material due to the lack of interaction of heavy ions with the nuclei of chemical elements of the material and so the absence of nuclear reactions.

At the same time, this advantage turns into a disadvantage from the viewpoint of modeling material damage under irradiation. Under neutron irradiation of a material, nuclear reactions of the $n \rightarrow p$ or/and $n \rightarrow \alpha$ type usually occur that lead to an increase in the hydrogen (H) and helium (He) concentration in the material. As known, He has a strong influence on the formation of radiation defects—in particular, vacancy voids—and also leads to a decrease in the strength of grain boundaries [11].

The above nuclear reactions do not occur under ion irradiation. Therefore, for the necessary H and He concentration to be created in the irradiated layer, light He and H ions should be used in addition to heavy ions under irradiation in the ion accelerator.

Another significant disadvantage of ion irradiation is the very small depth of the irradiated layer. This depth is the penetration depth of the heavy ion in the material under irradiation. For typical energies used in accelerators, the irradiated layer depth usually does not exceed 2–3 microns.

The depth of the irradiated layer under neutron irradiation is usually tens of millimeters. Therefore, various standard and special specimens may be machined and tested to obtain the performance properties such as tensile strength, impact strength and fracture toughness for the neutron-irradiated metal. The minimum size of the net sections of these specimens exceeds (3 ÷ 5) mm. It is clear that such specimens cannot be used to study metal after ion irradiation, because the minimum size of the specimens exceeds the irradiated layer depth by more than 1000 times.

Precisely for this reason, the small-scale material evaluation methods should be used to assess the ion irradiation effect on microstructural changes and mechanical properties. Many sufficient methods are currently available that may be used for studying the microstructure evolution. It is another situation entirely for the assessment of the mechanical properties of ion-irradiated thin layers, which are important characteristics for a material that has undergone neutron irradiation.

One of the consequences of neutron irradiation, which limits the lifetime of irradiated reactor components, is radiation embrittlement. In steels with a BCC lattice, including ferritic-martensitic steels (FMS), radiation embrittlement may occur by hardening and non-hardening mechanisms [12–17]. The hardening mechanism is connected to a radiation-induced increase in the yield strength (i.e., with radiation hardening). The non-hardening mechanism is connected with the formation of intragranular interphase or/and intergranular segregations. This mechanism does not lead to an increase in the yield strength.

The estimation of the radiation embrittlement of polycrystalline metals with a BCC lattice, including FMS, can be performed using the following empirical dependence [16,18–22]:

$$\Delta T_{tr} = k_{\sigma} \cdot \Delta \sigma_{\gamma}, \quad (1)$$

where $\Delta \sigma_{\gamma}$ is the radiation hardening in terms of the increment of material yield strength, ΔT_{tr} is the brittle-to-ductile transition temperature shift and k_{σ} is a numerical coefficient dependent on the test method used (impact strength or fracture toughness tests) and the phosphorus content in the material.

When impact strength specimens are tested, the ΔT_{tr} value is determined as the shift of the Charpy impact transition curve [23]. When fracture toughness specimens are tested, the ΔT_{tr} value is determined as the reference temperature shift ΔT_0 in the Master Curve method [24,25] or the reference temperature shift ΔT_{100} in the Unified Curve method [26].

Dependence (1) may describe the radiation embrittlement of FMS by both hardening and non-hardening mechanisms. The non-hardening mechanism is taken into account by

changing the value of the coefficient k_σ depending on the phosphorus content. FMS with an impurity content below the threshold (required for the formation of segregation) are embrittled only by the hardening mechanism [27]. For such materials, the coefficient k_σ does not depend on the content of impurities.

It is known that the change in the yield strength is linearly related to the change in its microhardness H_V [28–31]

$$\Delta\sigma_Y = a_v \cdot \Delta H_V, \quad (2)$$

where a_v is a proportionality coefficient.

Based on this, the radiation hardening under ion irradiation and the associated radiation embrittlement of FMS can be determined from the measurement of microhardness. Measuring the microhardness of the ion-irradiated layer H_V^i and the unirradiated material H_V^0 makes it possible to determine the radiation-induced change in microhardness as $\Delta H_V^i = H_V^i - H_V^0$.

Austenitic steels also undergo significant radiation hardening under neutron irradiation [2,32–36]. Despite the fact that the embrittlement mechanism of austenitic steels is fundamentally different from the embrittlement of FMS, radiation hardening also has a noticeable effect on the embrittlement of austenitic steels under irradiation when radiation swelling is absent [2,33,37]. In addition, radiation hardening has a strong effect on the corrosion cracking resistance of irradiated austenitic steels in a water environment, which is the coolant of the first circuit of WWER and PWR type reactors [34,38–41]. Moreover radiation hardening may affect the properties at high temperatures when helium embrittlement occurs with radiation hardening in a grain [42,43].

Thus, the change in microhardness under ion irradiation can be used as a characteristic controlling the embrittlement of materials by various mechanisms. Modeling the effect of neutron irradiation by ion irradiation can be considered adequate from the viewpoint of modelling the embrittlement mechanisms controlled by radiation hardening if such an ion irradiation regime is found for which the following condition is met:

$$\Delta H_V^i = \Delta H_V^n, \quad (3)$$

where ΔH_V^i and ΔH_V^n are the radiation-induced change in microhardness under ion and neutron irradiations, respectively.

The question is now how to provide radiation hardening for ion-irradiated material that is identical to that realized under neutron irradiation.

At present, despite the high intensity of research conducted in the direction of developing the methodology of simulation tests, there are still no recommendations on the choice of ion irradiation conditions that provide the above requirement.

Some recommendations on the choice of ion irradiation condition are given in [5] from the viewpoint of the identity of the evolution of microstructures (vacancy void, dislocation loops and radiation-induced phases). However, the analysis in [5] was performed only for one pair of neutron and ion irradiation temperatures. In common cases, the temperature shift between neutron and ion irradiation may depend on the temperature of neutron irradiation. Moreover, in paper [5], the difference in the sizes and densities of microstructure parameters between neutron and ion irradiation is about double. For this case the difference in radiation hardening after neutron and ion irradiation may be significant.

Thus, the objectives of this article are as follows:

- The development of a methodology for determining the ion irradiation parameters, which ensure the radiation hardening of ferritic-martensitic and austenitic steels, as close as possible to the radiation hardening of the same steels under neutron irradiation;
- The determination of microhardness and radiation hardening for ferritic-martensitic and austenitic steels after ion and neutron irradiation in various states;
- The development of some transferability function which connects the temperatures under neutron and ion irradiation for ferritic-martensitic and austenitic steels and that provides the same hardening at a given damage dose.

2. Methodology for Determining the Ion Irradiation Parameters and Development of the Transferability Function from Neutron to Ion Irradiation

The radiation embrittlement of ferrite-martensitic steels is much more strongly associated with radiation hardening than the embrittlement of austenitic steels. Therefore, when developing a methodology for determining the ion irradiation parameters for modeling hardening, it is advisable, first of all, to focus on the processes occurring in ferrite-martensitic steels.

The embrittlement of ferritic-martensitic steels under irradiation occurs by two mechanisms: hardening and non-hardening [12–17]. Embrittlement by the hardening mechanism occurs due to the radiation hardening of the material, which leads to an increase in the acting stresses in the sample or structural element. The increase in stresses, in turn, facilitates the nucleation and growth of microcracks in the material. The non-hardening mechanism of embrittlement is connected with a weakening of the interphase or grain boundaries in the material due to the segregation of impurity elements (mainly phosphorus). Such segregation also facilitates the nucleation of intragranular cleavage microcracks and intergranular microcracks, as well as the growth of intergranular microcracks. The non-hardening embrittlement mechanism does not lead to the radiation hardening of the material.

The non-hardening mechanism in FMS, connected with phosphorus segregation, does not practically depend on the damage dose rate, and the degree of embrittlement of FMS (with the same phosphorus content) is determined practically only by the accumulated damage dose [15]. Therefore, the non-hardening mechanism of embrittlement of FMS under ion irradiation may be modeled without taking into account the difference in the damage dose rates between ion and neutron irradiation. Hence, for FMS, the same degree of embrittlement for a non-hardening mechanism will be obtained when the ion irradiation damage dose equals the neutron irradiation damage dose.

Let us consider approaches to modeling the hardening mechanism of embrittlement by the ion irradiation for ferritic-martensitic and austenitic steels.

In cases where the hardening is mainly due to the formation of point defects and dislocation loops, as well as thermally activated processes of their annihilation, an increase in the dose rate leads to an increase in the radiation hardening degree under irradiation of the same dose.

At a constant damage dose rate, an increase in the irradiation temperature increases the annihilation of radiation defects and leads to a decrease in radiation hardening [16,44–46].

In connection with the above, to compensate for the difference in the dose rates under neutron irradiation in the reactor and ion irradiation in the accelerator when modeling the hardening embrittlement mechanism, the temperature of ion irradiation should, in general, exceed the temperature of neutron irradiation. It should be noted that ion irradiation leads to a temperature shift not only for radiation hardening, but also for radiation swelling, as well as the segregation of alloying and impurity elements.

Therefore, for modeling the hardening embrittlement mechanism, the difference in the dose rates under neutron irradiation in the reactor and ion irradiation in the accelerator may be compensated, in general, by increasing the temperature of ion irradiation.

The most adequate method for determining the hardening of a material under ion irradiation is the microhardness measurement.

Let a value $\delta T_{\text{irr}}^{\text{n-i}}$ be

$$\delta T_{\text{irr}}^{\text{n-i}} = T_{\text{irr}}^{\text{n}} - T_{\text{irr}}^{\text{i}}, \quad (4)$$

where $T_{\text{irr}}^{\text{n}}$ and $T_{\text{irr}}^{\text{i}}$ are temperatures of the neutron irradiation in reactor and the ion irradiation in the accelerator for which Condition (3) is met. As can be seen, the $\delta T_{\text{irr}}^{\text{n-i}}$ value describes the shift of the ion irradiation temperature relative to the neutron one.

In general, the value $\delta T_{\text{irr}}^{\text{n-i}}$ depends on $T_{\text{irr}}^{\text{n}}$. Let us call this dependence the transferability function from neutron irradiation to ion irradiation: $\delta T_{\text{irr}}^{\text{n-i}} = \varphi(T_{\text{irr}}^{\text{n}})$. The transferability function $\varphi(T_{\text{irr}}^{\text{n}})$ is monotonically decreasing, since the relaxation processes connected with

the annihilation of radiation-induced defects occurs faster with an increase in the irradiation temperature and the effect of the dose rate on the material hardening decreases.

Thus, for the radiation hardening to be modeled by ion irradiation, the following ion irradiation parameters should be used.

1. The target damage dose for ion irradiation D^i should be equal to the dose D^n for modeled neutron irradiation. Then, the non-hardening embrittlement mechanism of FMS is automatically modeled.

2. The ion irradiation temperature T_{irr}^i should be set using the transferability function $T_{\text{irr}}^i = T_{\text{irr}}^n + \varphi(T_{\text{irr}}^n)$. Then, the hardening embrittlement mechanism of FMS is modeled.

Since the damage dose distribution over the depth of the ion-irradiated layer is extremely heterogeneous, a separate task is to choose the distance from the ion-irradiated surface for which the target dose $D^i = D^n$ should be obtained. As an example, Figure 1 shows the dependence of the relative damage dose D^i / D_{max}^i on the relative depth of the ion-irradiated layer for 12Cr-Ni-Mo-V-Nb steel. The calculation was carried out with the SRIM2008 software package for the case of irradiation with Fe^{3+} ions with an energy of 11.5 MeV in Institute for Physics and Power Engineering (JSC “SSC RF-FEI”). Figure 1 also shows a calculated relative concentration $C^{\text{Fe}} / C_{\text{max}}^{\text{Fe}}$ of the Fe^{3+} ions over the depth of the sample.

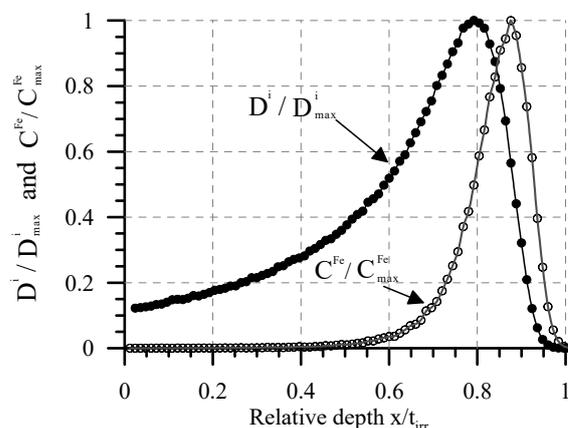


Figure 1. The depth-variation in the relative damage dose and the relative concentration of implanted Fe ions for the irradiated layer of the sample from the FMS 12Cr-Ni-Mo-V-Nb irradiated with Fe^{3+} ions with an energy of 11.5 MeV.

It should be noted that, when irradiating austenitic steels with Ni^{3+} or Ni^{4+} ions of the same energy, the dependences of the relative damage dose and the concentration of injected heavy ions are practically the same as those shown in Figure 1.

In Figure 1, the designations are used as follows: D_{max}^i and $C_{\text{max}}^{\text{Fe}}$ are the maximum values of the damage dose D^i and the concentration of injected Fe^{3+} ions; and t_{irr} is the depth of the ion-irradiated layer.

Let us denote x^* as the distance from the irradiated surface to the layer for which the condition $D^i = D^n$ should be satisfied.

As follows from the presented dose distribution, with a decrease in the x^* value, the value D^i / D_{max}^i also decreases. Based on this, the maximum dose of ion irradiation D_{max}^i should be increased when the x^* value decreases to fulfill the condition $D^i = D^n$. In other words, the degree of ion irradiation of the sample depends on the choice of the x^* value.

To estimate the x^* value when modeling radiation hardening by ion irradiation, let us divide the whole irradiated layer on three test zones adjoined to each other. Every test zone is the area where the irradiation effect on material is investigated for the average calculated dose in the area.

The damage doses on the left (closest to the surface) and right (farthest from the surface) boundaries of the zones are denoted, respectively, as D_{j-1}^i and D_j^i , where j is the number of the test zone from 1 to 3.

The left boundary of the first zone closest to the surface should be located a distance of 0.2–0.3 microns from the surface. This limitation is due to the fact that the surface distorts radiation-stimulated processes.

As can be seen from Figure 1, the location of the maximum damage dose and the concentration of injected ions is extremely close. It is clear that the injected ions change the microstructure of the investigated material, and the test zones should be located closer to the surface than the maximum concentration of injected ions. In order to avoid the influence of injected heavy ions on the microstructure, it is recommended to place the right border of the last test zone (in our case, the third) no deeper than the distance from the surface corresponding to 20% of the maximum implanted heavy ions concentration.

Based on this, let us set the location of the right border of the third zone. According to the calculations shown in Figure 1, the depth where the concentration of injected heavy ions is 20% of the maximum corresponds to approximately $D_3^i \approx 0.8D_{\max}^i$.

As can be seen from Figure 1, the damage dose is $D_0^i \approx 0.12D_{\max}^i$ for the left boundary of the first test zone when the maximum dose is located at a depth of ≈ 2 microns.

Assume that the ratio of damage doses D_j^i/D_{j-1}^i on the borders of each zone should be the same. Then, the ratio of the maximum dose (on the right border of the third zone) to the minimum (on the left border of the first zone) is equal to

$$\frac{D_3^i}{D_0^i} = \left(\frac{D_j^i}{D_{j-1}^i} \right)^N, \quad (5)$$

where N is a number of test zones.

Based on (5), the ratio of damage doses on the boundaries of each test zone for $N = 3$ is equal to

$$\frac{D_j^i}{D_{j-1}^i} = \sqrt[3]{\frac{D_3^i}{D_0^i}} = 1.88 \quad (6)$$

Let us choose the middle of the third test zone as the x^* value. Then, the condition $D^i = D^n$ can be written as

$$\frac{D_3^i + D_2^i}{2} = D^n. \quad (7)$$

The choice of the third test zone, in which the target dose $D^i = D^n$ is set, allows one to set a smaller value of D_{\max}^i for ion irradiation compared to the choice of the first or second test zones. At the same time, it is obvious that the choice of one or another test zone for setting the parameters of ion irradiation should allow for modeling the radiation hardening of the material in terms of microhardness.

Approaches to measuring the microhardness of the irradiated layer should be proposed that take into account the strong gradient of the damage dose over the depth of the irradiated layer and some features of the material hardening under ion irradiation.

Radiation hardening in terms of the yield strength $\Delta\sigma_Y$ increment is known to increase monotonically with an increase in the damage dose D . The dependence $\Delta\sigma_Y(D)$ tends to saturate when a dose of $15 \div 30$ dpa is reached for FMS [47–50] and a dose of 10 dpa for austenitic steels [3,33,35,36]. In other words, with a further increase in the dose, hardening does not practically increase.

Hence, for these doses, the hardening of the ion-irradiated layer is homogeneous if the minimum damage dose (dose on the sample surface) is greater than D_{sat} , where D_{sat} is the dose corresponding to the saturation of the dependence $\Delta\sigma_Y(D)$. For other cases, the variation in $\Delta\sigma_Y$ over the irradiated layer depth is at least partially similar to the dose variation.

Denote the radiation hardening in each zone as $\Delta\sigma_Y^{(j)}$, where $j = 1 \dots 3$ is the number of the test zone. As a common case, $\Delta\sigma_Y^{(3)} \geq \Delta\sigma_Y^{(2)} \geq \Delta\sigma_Y^{(1)}$. The average hardening for all three zones is denoted as $\Delta\bar{\sigma}_Y$. As shown above, for $D_0 \geq D_{\text{sat}}$, the equality $\Delta\sigma_Y^{(1)} = \Delta\sigma_Y^{(2)} = \Delta\sigma_Y^{(3)} = \Delta\bar{\sigma}_Y$ is practically met, and for $D_0 < D_{\text{sat}}$, the value $\Delta\sigma_Y^{(3)}$ is greater than $\Delta\bar{\sigma}_Y$.

As noted above, an increase in radiation hardening $\Delta\sigma_Y$ occurs with an increase in the dose rate. Therefore, when the condition $D^i = D^n$ is met, one should expect $\Delta\sigma_Y^{(3)} > \Delta\sigma_Y^n$, where $\Delta\sigma_Y^n$ is the radiation hardening of the material after neutron irradiation. Considering that, in common cases, the inequalities $\Delta\sigma_Y^{(3)} \geq \Delta\bar{\sigma}_Y$ and $\Delta\sigma_Y^{(3)} > \Delta\sigma_Y^n$ are fulfilled, the value $\Delta\bar{\sigma}_Y$ can be either less or more than $\Delta\sigma_Y^n$, depending on the influence of the dose rate on material hardening. If the influence is strong, then $\Delta\bar{\sigma}_Y > \Delta\sigma_Y^n$. If the effect of the dose rate is weak and $D_0 \geq D_{\text{sat}}$, then $\Delta\bar{\sigma}_Y = \Delta\sigma_Y^n$, and for $D_0 < D_{\text{sat}}$, it should be expected that $\Delta\bar{\sigma}_Y < \Delta\sigma_Y^n$.

When measuring the hardness, the material is consistently plastically deformed under the indentation process, starting from the surface. As is well known [51–54], the depth of the plastic deformation area exceeds the depth of indentation. In this regard, it should be taken into account that the microhardness of the irradiated layer, depending on the indentation depth, is affected by the substrate (the material of the sample located deeper than the irradiated layer). For a homogeneous material (unirradiated or irradiated by neutrons), the value H_V does not practically depend on the depth of indentation.

Since the unirradiated material usually has a lower σ_Y value compared with the irradiated layer, the dependence H_V^i on x/t_{irr} has a non-monotonic character with a maximum. The microhardness dependence on the indentation depth for the ion-irradiated layer is schematically shown in Figure 2. The increasing part of the dependence H_V^i on x/t_{irr} is due to an increase in the damage dose, and the decreasing part is connected with the influence of a soft substrate. It should be noted that the depth of the location of the microhardness maximum in the irradiated layer does not coincide with the location of the damage dose maximum and is much closer to the surface.

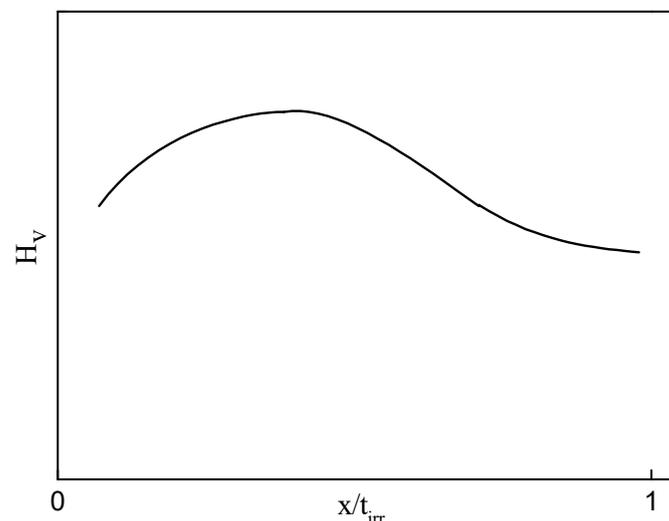


Figure 2. General view of the dependence H_V^i on x/t_{irr} for ion-irradiated layer.

In connection with the above, to assess the microhardness of the ion-irradiated layer, it is advisable to take the maximum value H_V^i determined in the irradiated zone for different indentation depths. If this value corresponds to an indentation depth less than half the thickness of the ion-irradiated layer, then it corresponds to the value of $\Delta\bar{\sigma}_Y$ to a certain extent.

This is consistent with the recommendations of the international standard for measuring the microhardness of a thin coating [55]. According to [55], when determining

the hardness of the hardened layer on a soft substrate, the maximum value in the area of indentation depths less than half of the layer thickness is recommended for use as the microhardness of the hardened layer.

Then, the radiation hardening after ion irradiation in terms of changes in microhardness can be written as

$$\Delta H_V^i = \max(H_V^i) - H_V^0, \quad (8)$$

where H_V^0 is the value of H_V for unirradiated material.

Since the value of $\max(H_V^i)$ depends on the microhardness of each test zone in the ion-irradiated layer, the value ΔH_V^i determines the radiation hardening $\Delta \bar{\sigma}_Y$ in the irradiated layer. As shown above, the value of $\Delta \bar{\sigma}_Y$ under ion irradiation can be either greater or less $\Delta \sigma_Y^n$, which depends on the sensitivity of the hardening to the damage dose rate. Thus, the algorithm may be recommended for constructing the dependence $\delta T_{irr}^{i-n} = \varphi(T_{irr}^n)$ as follows.

1. The dependence $\Delta H_V^i(T_{irr}^n)$ is constructed for the material irradiated by neutrons for the dose D^n or for doses $D^n > D_{sat}$, as Figure 3 illustrates.

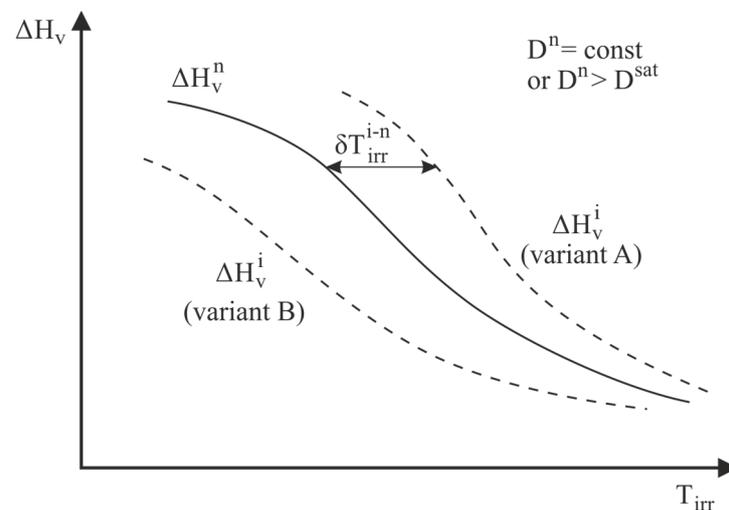


Figure 3. Common view of the dependence of ΔH_V on T_{irr} .

2. The investigated material is irradiated in an ion accelerator at various temperatures T_{irr}^i until the target damage dose is reached in the third test zone $\frac{D_3^i + D_2^i}{2} = D^n$ or $\frac{D_3^i + D_2^i}{2} > D_{sat}$.
3. The microhardness of the irradiated layer is measured for the material after ion irradiation at various T_{irr}^i and the dependence $\Delta H_V^i(T_{irr}^i)$ is determined.
4. If the dependence $\Delta H_V^i(T_{irr}^i)$ corresponds to variant A in Figure 3, then the transferability function $\delta T_{irr}^{i-n} = \varphi(T_{irr}^n)$ is determined.
5. If the dependence $\Delta H_V^i(T_{irr}^i)$ corresponds to variant B in Figure 3, ion irradiation should be repeated, choosing another test zone in which the target dose is set closer to the surface (second or first). This stage provides an increase in the maximum dose of ion irradiation D_{max}^i and an increase in the ΔH_V^i value.

Variant B corresponds to the case when $D_0 \ll D_{sat}$ and radiation hardening is practically invariant to the dose rate.

6. The test zone in which the target dose is set varies until the dependence $\Delta H_V^i(T_{irr}^i)$ begins to correspond to variant A.

3. Investigated Materials and Specimens

3.1. Investigated Materials

Two grades of stainless FMS were taken for research, 12Cr-Ni-Mo-V-Nb (denoted hereafter as F1 material) and EP-823 (denoted hereafter as F2 material), and two grades of stainless chromium–nickel austenitic steels, 18Cr-10Ni-Ti (denoted hereafter as A1 material) and 16Cr-20Ni-2.5Mo-Ti (denoted hereafter as A2 material). The chemical compositions of F1 and F2 materials according to the technical specifications are presented in Table 1. The actual chemical compositions of A1 and A2 materials are shown in Table 2.

Table 1. Chemical composition of 12Cr-Ni-Mo-V-Nb steel (material F1) and EP-823 steel (material F2) according to the technical specifications.

Material	Mass Fraction of Chemical Elements, %							
	C	Si	Mn	Cr	Ni	Mo	S	P
F1	0.08	0.29	0.70	12.1	1.06	0.94	0.003	0.013
F2	0.16	1.18	0.65	10.7	0.62	0.76	0.002	0.015
	Nb	V	W	Ti	Al	B	N	
F1	0.11	0.20	-	-	0.029	0.03	0.059	
F2	0.33	0.33	0.54	0.01	0.02	0.003	0.04	

Table 2. The actual chemical composition of 18Cr-10Ni-Ti steel (material A1) and 16Cr-20Ni-2.5Mo-Ti steel (material A2).

Material	Mass Fraction of Chemical Elements, %									
	C	Si	Mn	S	P	Cr	Ni	Co	Mo	
A1	0.071	0.52	1.71	0.002	0.028	17.5	10.2	0.021	0.01	
A2	0.065	0.48	1.65	0.004	0.026	15.3	20.2	0.027	2.59	
	Ti	Al	W	V	Nb	Cu	N	O	H	
A1	0.53	0.110	-	0.025	-	0.018	0.0059	0.0012	0.00050	
A2	0.63	0.146	-	-	-	0.040	0.0081	0.0042	0.00041	

The presented steels were studied in the initial condition and after ion and neutron irradiation at different temperatures.

3.2. Specimens for Ion Irradiation

Special disk specimens 12 mm in diameter and 2 mm in thickness were developed and manufactured for ion irradiation. A sketch of the disk specimen is shown in Figure 4. A cylindrical hole with a diameter of 1 mm and length of 2–3 mm was drilled into the disc-shaped samples for thermocouple installation to control the irradiation temperature (see Figure 4).

Disk specimens were made using the electro spark technique, which excludes the strain hardening of the surface. The surface intended for irradiation was subjected to sequential grinding and polishing with a finishing pass with “soft” abrasives based on colloidal silica with a dispersion of 0.3–0.5 µm.

The absence of strain hardening was checked by scanning electron microscopy by the diffraction of backscattered electrons by a visualization of the so-called Kikuchi lines [56,57].

Disk specimens of F1 material were cut via cylindrical forging. Disk specimens of F2 material were cut from sheet metal with a thickness of 2 mm. Disk specimens of A1 and A2 materials were cut from small forgings of 500 kg in weight which were annealed over a temperature range 1040–1060 °C with subsequent air cooling.

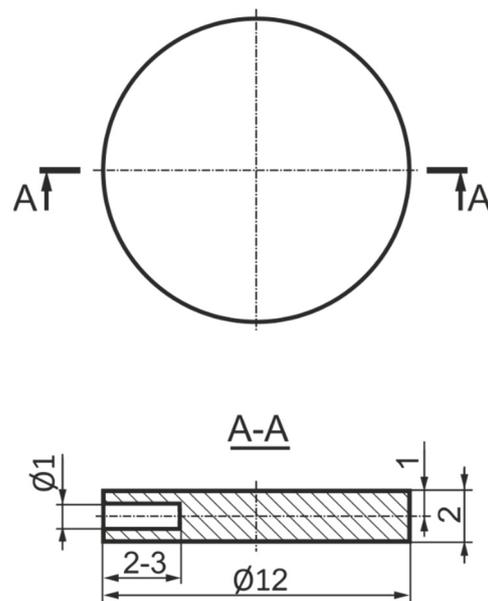


Figure 4. Sketch of disk specimen for ion irradiation (unit: mm).

3.3. Specimens for Neutron Irradiation

Specimens from F1, A1 and A2 materials, intended for study of the effects of neutron irradiation, were made from the same forgings that were used for specimens for ion irradiation. The measurement of the microhardness of these materials after neutron irradiation was carried out on microsections made from the heads of cylindrical tensile specimens irradiated in the reactor and tested at room temperature.

The measurement of the microhardness of F2 material after neutron irradiation was carried out on the metal shells of experimental fuel elements. To determine the microhardness of the metal fuel element shells in the initial state, they were subjected to tempering for 2 h at 720 °C.

Additionally, the measurement of the microhardness for the irradiated A1 material was carried out on the metal cut from the baffle-former-barrel assembly or, briefly, from the baffle of the decommissioned WWER-440 from Unit 3 of the Novovoronezh nuclear power plant. This reactor was in service for 45 years and was decommissioned in 2016 [58].

4. Neutron and Ion Irradiation Conditions

4.1. Neutron Irradiation

4.1.1. Ferritic-Martensitic Steels

Specimens of the F1 material were irradiated in a research fast reactor with sodium coolant BOR-60 at temperatures $T_{irr}^n = 390$ °C up to damage doses $D^n = 10.3$ and 11.6 dpa and $T_{irr}^n = 550$ °C up to $D^n = 14.6$ dpa.

To simulate $T_{irr} = 450$ °C and 500 °C, annealing of the F1 material irradiated at $T_{irr}^n = 390$ °C to $D^n = 11.6$ dpa was carried out. One sample was annealed at 450 °C, another sample at 500 °C. During annealing at $T_{ann} > T_{irr}^n$, some part of the radiation defects, such as point defects and dislocation loops, formed at neutron irradiation dissociates. The non-dissociable part of radiation defects corresponds to the radiation defects formed at $T_{irr}^n = T_{ann}$. The possibility of neutron irradiation simulation at temperatures exceeding the actual irradiation temperature by annealing was demonstrated in [16,59].

Such an approach was verified as follows.

The radiation hardening of the F2 material, irradiated at $T_{irr}^n = 390$ °C to $D^n = 11.6$ dpa and annealed at $T_{ann} = 500$ °C for 10 h, was compared with the radiation hardening of 12% Cr steel with a similar chemical composition irradiated at $T_{irr}^n = 490$ °C to $D^n = 13$ dpa in a BN-350 fast reactor [50]. The radiation hardening of irradiated 12% Cr steel was

$\Delta\sigma_Y^n = 50$ MPa [50]. This specified value may be recalculated in terms of changes in microhardness using the following equation [28]:

$$\Delta\sigma_Y = 0.306 \cdot \Delta H_V, \text{MPa} \quad (9)$$

Equation (9) provides for 12% Cr steel $\Delta H_V^n = 163$ MPa. This estimation coincides with $\Delta H_V^n = 160$ MPa, represented hereafter for F1 material irradiated at $T_{irr}^n = 390$ °C and annealed at $T_{ann} = 500$ °C.

This allows us to conclude that the simulation of higher neutron irradiation temperatures by thermal annealing is adequate, at least from the viewpoint of modeling radiation hardening.

Specimens from F2 materials were cut from the shells of experimental fuel elements irradiated in a BN-600 reactor with damage doses from 14 dpa to 33 dpa. Disk specimens cut from different zones on the shell height provided the data for various irradiation temperatures ranging from 380 °C to 580 °C.

The damage doses and irradiation temperatures for F1 and F2 materials are given in Table 3.

Table 3. Condition of neutron irradiation for F1 and F2 materials.

Material	Irradiation Temperature T_{irr}^n , °C	Damage Dose, D^n , dpa
F1	390	10.3
	390	11.6
	450 *	11.6
	500 *	11.6
	550	14.6
F2	380	14
	390	22
	410	33
	425	33
	580	22

*—irradiation temperature modeled by annealing.

4.1.2. Austenitic Steels

Specimens from A1 and A2 materials were irradiated under various conditions in reactors, such as the research fast neutron reactor BOR-60, the research pressure water reactor SM-3 (both reactors are located in NRC RIAR, Dimitrovgrad, Russia) and the power pressure water reactor WWER-440, over temperatures ranging from $T_{irr}^n = 60$ °C to $T_{irr}^n = 500$ °C up to damage doses from $D^n = 10.2$ dpa to $D^n = 33.7$ dpa. Irradiation condition, designated in Table 4 as (SM-3 + BOR-60,) means a two-stage irradiation for which the first stage is irradiation in the side reflector of SM-3 at $T_{irr}^n \approx 320$ °C up to $D^n \approx 3$ dpa and the second stage is additional irradiation in a BOR-60 reactor under various conditions.

For studying the radiation-induced variation in microhardness over a wider temperature range, the procedure described above for FM steels was also used for austenitic A1 and A2 steels. Post-irradiated annealing of irradiated A1 and A2 steels was carried out over temperatures ranging from 400 to 600 °C.

Table 4. Conditions of neutron irradiation and post-irradiation annealing for A1 and A2 materials.

Material	Reactor	Irradiation Temperature T_{irr}^n , °C	Damage Dose, D^n , dpa
A1	SM-3	60	10.2
	WVER-440	280	15.7
	WVER-440	280	33.7
	WVER-440	400 (280) **	33.7
	WVER-440	450 (280) **	33.7
	WVER-440	500 (280) **	33.7
	WVER-440	550 (280) **	33.7
	WVER-440	600 (280) **	33.7
	BOR-60	330	10.8
	BOR-60	500 (330) **	10.8
	BOR-60	500	29.0
	BOR-60	550 (500) **	29.0
	BOR-60	600 (500) **	29.0
	SM-3 + BOR-60 *	500	11.3
SM-3 + BOR-60 *	550 (500) **	11.3	
A2	SM-3	60	12.1
	BOR-60	330	10.8
	BOR-60	500	29.0
	BOR-60	550 (500) **	29.0
	SM-3 + BOR-60 *	500	11.3
	SM-3 + BOR-60 *	550 (500) **	11.3

* two-stage irradiation was carried out: the first-stage irradiation was carried out in the side reflector of SM-3 at $T_{irr}^n \approx 320$ °C up to $D^n \approx 3$ dpa; in the second stage, additional irradiation was carried out in a BOR-60 reactor under the conditions indicated in the table. ** specimens were subjected to post-irradiated annealing: irradiation temperature is given in brackets and annealing temperature—without brackets.

The stacking fault energy for austenitic steels is lower than for ferrite-martensitic steels. Therefore, diffusion processes in austenitic steels are slower than in ferrite-martensitic steels and, as a result, the equilibrium state under annealing takes a longer time to be achieved. Based on this, the duration of post-irradiation annealing of austenitic steels was doubled in comparison with ferrite-martensitic steels and was 20 h.

Radiation hardening, in terms of yield strength increment for austenitic steels, has a maximum in the temperature range of ~300–450 °C [35,36]. Therefore, specimens after irradiation in a SM-3 reactor at 60 °C were not used for post-irradiation annealing to model various temperatures of neutron irradiation.

The conditions of neutron irradiation and post-irradiation annealing for A1 and A2 materials are represented in Table 4.

4.2. Ion Irradiation

4.2.1. Ferritic-Martensitic Steels

Disc-shaped samples from the studied F1 and F2 materials were irradiated in the ion accelerator 3MV Tandatron (at “Institute for Physics and Power Engineering”) by a continuous beam of heavy ions with Fe^{3+} or Fe^{4+} ions in combination with He ions.

The main condition of irradiation was cyclic irradiation, in which the time of heavy ion irradiation required to set the target dose was divided into five identical cycles, between which the samples were irradiated with He ions. To assess the effect of He on radiation

hardening for the F2 material, a continuous irradiation condition with Fe^{3+} or Fe^{4+} ions without injecting He was also used. The target dose D^i corresponding to the average dose in the third test zone was set at a depth of $1.4 \mu\text{m}$ of the irradiated layer.

The values of target doses for the studied materials, F1 and F2, were selected based on the parameters of neutron irradiation of these materials given in Table 3.

The energy of heavy Fe^{3+} or Fe^{4+} ions under irradiation in all conditions was 11.5 MeV. The depth of the location of the maximum damage dose is $\sim 2 \mu\text{m}$ for this ion energy. Figure 5 shows the damage dose variation on the irradiated layer depth for $D^i = 13 \text{ dpa}$ and 30 dpa under the ion irradiation of the studied ferrite-martensitic steels.

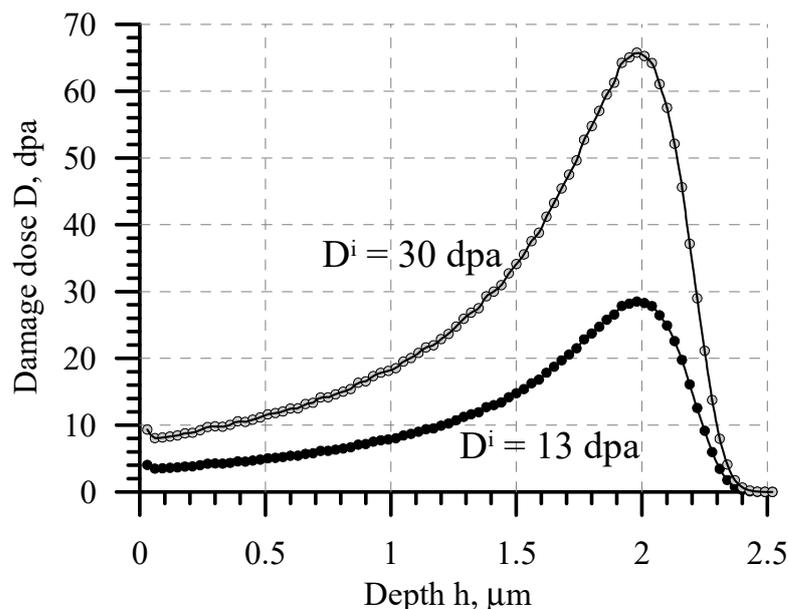


Figure 5. The depth-variation in the damage dose for the irradiated layer for F1 and F2 materials irradiated with Fe^{3+} and Fe^{4+} ions up to two target doses of D^i .

The energy of He ions was varied under irradiation to ensure the same specific concentration of injected He $\eta = C_{\text{He}}/D$ over the depth range of the irradiated layer from $1.1 \mu\text{m}$ to $1.8 \mu\text{m}$ (here, C_{He} is the atomic concentration of helium).

The value of the specific concentration η was chosen corresponding to the actual He concentration under the operation of a fast reactor due to the nuclear reactions of fast-spectrum neutrons with the studied FMS. According to the performed estimates, for the spectrum of the core of the BOR-60 reactor, $\eta = 0.2 \text{ appm/dpa}$.

For studying the He effect on radiation hardening, some samples from the F2 material were irradiated with a higher value of $\eta = 4 \text{ appm/dpa}$.

The irradiation temperature was controlled by a thermocouple installed directly in the sample and maintained by an infrared heater installed behind the target with the sample.

The ion beam after the diaphragm created a spot with a diameter of 6 mm on the irradiated sample surface with a decrease in the damage dose along the edges of the spot relative to the target dose of no more than 10%.

Table 5 shows the ion irradiation conditions for samples from F1 and F2 materials.

Table 5. The ion irradiation conditions for samples from F1 and F2 materials.

Material	Identification Sample Marking	Irradiation Temperature, T_{irr}^i , °C	Damage Dose, D^i , dpa	Specific He Concentration, η , appm/dpa
F1	X55	350	13	0.2
	X56	400	13	0.2
	X57	450	13	0.2
	X58	500	13	0.2
	X59	550	13	0.2
	X60	600	13	0.2
F2	P30	350	15	0
	P31	380	14	0.2
	P62	400	15	0
	P59	450	14	0.2
	P29	400	30	0
	P49	400	30	0.2
	P21	400	30	4.0
	P28	450	30	0
	P27	450	30	4.0
	P60	450	30	0.2
	P61	500	30	0
	P22	500	30	0.2
	P26	500	30	4.0
	P18	600	30	0.2

4.2.2. Austenitic Steels

The ion irradiation of disk samples from austenitic steels was carried out in the ion accelerator 3MV Tandetron (at the “Institute for Physics and Power Engineering”) with a continuous beam of heavy Ni^{3+} or Ni^{4+} ions in combination with He ions.

Two conditions of irradiation were applied, as well as for ferrite-martensitic steels:

- Continuous irradiation with Ni^{3+} or Ni^{4+} ions without injection of He;
- Cyclic irradiation, in which the time of heavy ion irradiation required to set the target dose was divided into five identical cycles, between which the samples were irradiated with He^+ ions.

The energy of heavy Ni^{3+} or Ni^{4+} ions in all conditions was 11.5 MeV.

The parameters of irradiation with He ions were similar to those for irradiation of ferrite-martensitic steels (F1 and F2 materials) to ensure the same η value over the depth range of the irradiated layer from 1.1 μm to 1.8 μm .

The value of η was assigned equal to the actual specific He concentration for the thermal spectrum of the WWER. This type of reactor leads to a higher specific He concentration in austenitic steels.

The actual specific He concentration in 18Cr-10Ni-Ti steel is 5–7 appm/dpa for the spectrum of neutrons typical for the Internals of WWER type reactors [60]; therefore, the value of η was chosen to be 7 appm/dpa.

The target dose D^i , corresponding to the average dose in the third test zone, was set to a depth of 1.4 μm of the irradiated layer.

The values of target doses for the studied A1 and A2 materials were selected based on the parameters of neutron irradiation of these materials given in Table 4. The dose of $D^i = 30$ dpa was chosen as the first target dose, and $D^i = 13$ dpa was chosen as the second.

Figure 6 shows the variation in the damage dose over the depth of the ion-irradiated layer for both target doses.

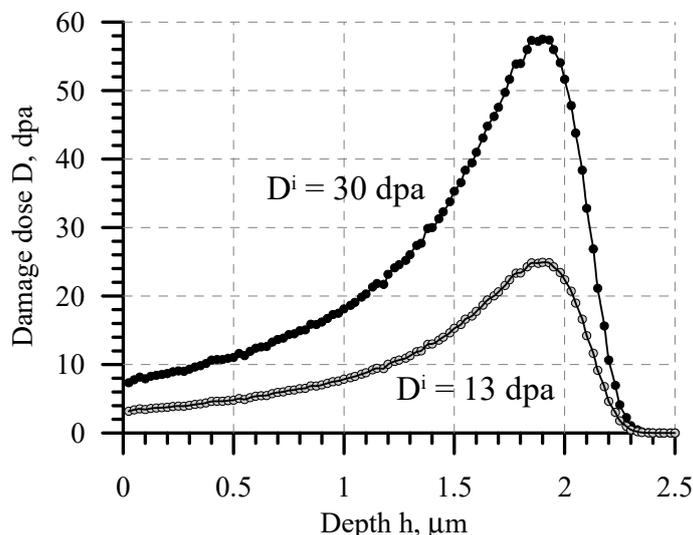


Figure 6. The depth-variation in the damage dose for the irradiated layer for A1 and A2 materials irradiated with Ni^{3+} and Ni^{4+} ions up to two target doses of D^i .

Table 6 shows the ion irradiation conditions for the studied austenitic steels.

Table 6. The ion irradiation conditions for A1 and A2 materials.

Material	Irradiation Condition		
	The Specific He Concentration, η , appm/dpa	The Target Damage Dose D^i *, dpa	Irradiation Temperature T_{irr}^i , °C
A1	0	13	300, 400, 500, 550, 650
	0	30	400, 500
	7	13	300, 400, 500
	7	30	300, 400, 500, 550, 650
A2	0	13	300, 400, 500
	0	30	400, 500

*— D^i is the damage dose for a depth of 1.4 microns.

5. Microhardness Measurement Procedure

The microhardness was measured by microhardness tester CPX 25-099 from “CSM-instruments” using the Vickers pyramid indentation method, with a given strain rate equal to 0.1 s^{-1} up to a given load. The given load was withstood for 10 s in accordance with GOST R 8.748-2011 [61].

The microhardness tester CPX 25-099 was equipped with a remote control for measuring the microhardness of irradiated samples in the hot cell of the Laboratory of Radiation Materials Science of CRISM “Prometey”.

The microhardness value was determined as the ratio of the load to the contact area of the indenter with the material. The contact area of the indenter was calculated based on the measured projection area of the indent on the sample surface.

Thin sections from samples irradiated in reactors were made on a Saphir 520 machine from QATM with a remote control, which was also installed in a hot cell.

The microhardness of neutron-irradiated samples was measured at two loads, $P = 0.5$ N and 1 N. For the studied steels, the indentation depth, h , at these loads was $h = 2.5\text{--}3.8$ μm . The error in measurement of the microhardness did not exceed 30 MPa. The microhardness value was determined as the average value determined for loads of 0.5 N and 1 N.

The microhardness of ion-irradiated samples was determined when using the same equipment as for samples after neutron irradiation. The microhardness of ion-irradiated samples was measured at a greater number of loads due to the shallow depth of the ion-irradiated layer and the inhomogeneity of the damage dose at its depth. The load level itself was chosen to be lower than for samples after neutron irradiation. The following series of loads was taken to determine the microhardness after ion irradiation: $P = 0.03, 0.04, 0.05, 0.06, 0.07, 0.08, 0.1, 0.2, 0.5$ N and 1 N. These loads correspond to the range of depth from $h = 0.4$ to $h = 5.5$ μm . For $P = 0.03\text{--}0.05$ N, at least five measurements were performed; for $P = 0.06\text{--}1$ N—at least three. The root-mean-square error of measuring the microhardness value for $P = 0.03 \div 0.05$ N was ± 120 MPa; for $P = 0.06 \div 1.0$ N, the error was ± 60 MPa.

The microhardness measurement results were processed in accordance with the standards for the microhardness measurement of a thin solid layer on a soft substrate [55]. The standard [55] recommends taking the thin solid layer microhardness as the maximum value of microhardness measured over a depth range of less than half a thin layer. Hence, for the irradiated layer thickness of ~ 2 microns, the microhardness should be measured for the depth, $h < 1$ μm .

When determining radiation hardening after neutron and ion irradiation, the microhardness of the unirradiated samples was determined at $P = 1.0$ N, which corresponds to the indentation depths, $h = 2.5\text{--}3.0$ μm .

The exception is the determination of the hardening of ferrite-martensitic steels (F1 and F2 materials) after ion irradiation. In this case, the microhardness of the irradiated and unirradiated metal was determined for the same sample. The microhardness of the unirradiated metal was determined outside the ion irradiation spot at $P = 0.5$ N, which corresponds to $h = 2.5\text{--}3.0$ μm .

Figure 7 shows a photograph of indents at $P = 0.03$ and 0.05 N. As seen, the shape of the indent is close to square, indicating the negligible influence of the jumper at the top of the Vickers pyramid used at minimal loads.

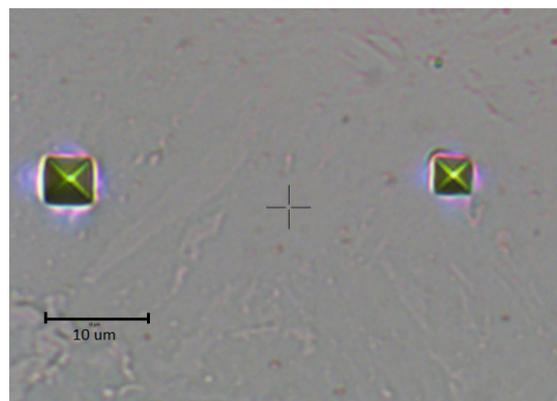


Figure 7. Photo of indents at $P = 0.03$ N (on the right) and 0.05 N (on the left).

6. Experimental Results

6.1. Microhardness of Ferritic-Martensitic Steels

6.1.1. After Neutron Irradiation

The microhardness values of samples from F1 and F2 materials are represented in Tables 7 and 8, respectively, as follows: H_V^0 is the microhardness in the initial (unirradiated)

state and H_V^n is the same in the irradiated state, with the difference $\Delta H_V^n = H_V^n - H_V^0$ being a measure of the radiation hardening (see Equation (2)). Hereafter, this short difference is called the radiation hardening.

Table 7. Neutron irradiation parameters, microhardness in the initial H_V^0 and irradiated H_V^n states and radiation hardening ΔH_V^n for F1 material.

Irradiation Temperature T_{irr}^n , °C	Damage Dose D^n , dpa	H_V^n , MPa	H_V^0 , MPa	ΔH_V^n , MPa
390	10.3	3650	2370	1280
390	11.6	3780	2370	1410
450 *	11.6	3430	2370	1060
500 *	11.6	2530	2370	160
550	14.6	2380	2370	10

*—the irradiation temperature has been modeled by annealing of metal irradiated at 390 °C up to 11.6 dpa at the specified temperature over 10 h.

Table 8. Neutron irradiation parameters, microhardness in the initial H_V^0 and irradiated H_V^n states and radiation hardening ΔH_V^n for F2 material.

Irradiation Temperature T_{irr}^n , °C	Damage Dose D^n , dpa	H_V^n , MPa	H_V^0 , MPa	ΔH_V^n , MPa
380	14	4220	2400	1820
390	22	4120	2400	1720
410	33	4050	2400	1650
425	33	3930	2400	1530
495 *	11 *	-	-	490 *
580	22	2696	2400	296

*—the data from paper [50] recalculated using the Dependence (9).

In addition to the original data, the results from paper [50] have been added to Table 8. It should be noted that, in paper [50], the radiation hardening is presented in terms of changes in yield strength $\Delta\sigma_V^n$. For comparison, the changes in the microhardness in the data from [50] were recalculated using the correlation Dependence (9) presented above. These data have been used due to the absence of original data for the F2 material at T_{irr}^n over the range from 425 to 580 °C.

An analysis of the data presented in Tables 7 and 8 allows us to draw the following conclusions. The data obtained at $T_{irr}^n = 380\text{--}390$ °C show that D_{sat} for F2 material does not exceed at least 14 dpa, since there is no increase in hardening with an increase in the damage dose. For F1 material, such an unambiguous conclusion cannot be made as the increase in hardening with an increase in dose at $T_{irr}^n = 390$ °C can be explained both by the scatter of experimental data and by the fact that $D_{sat} > 10.3$ dpa.

For both FMS, the radiation hardening monotonically decreases with increasing T_{irr}^n over the range of the investigated temperatures. Moreover, for F1 material, an almost complete annealing of radiation hardening can already be observed at $T_{irr}^n = 550$ °C, while, for F2 material at $T_{irr}^n = 580$ °C, some radiation hardening is still observed.

6.1.2. After Ion Irradiation

The typical dependences of microhardness on the indentation depth, h , are shown in Figure 8 for the irradiated and unirradiated regions of samples from F1 and F2 materials after ion irradiation.

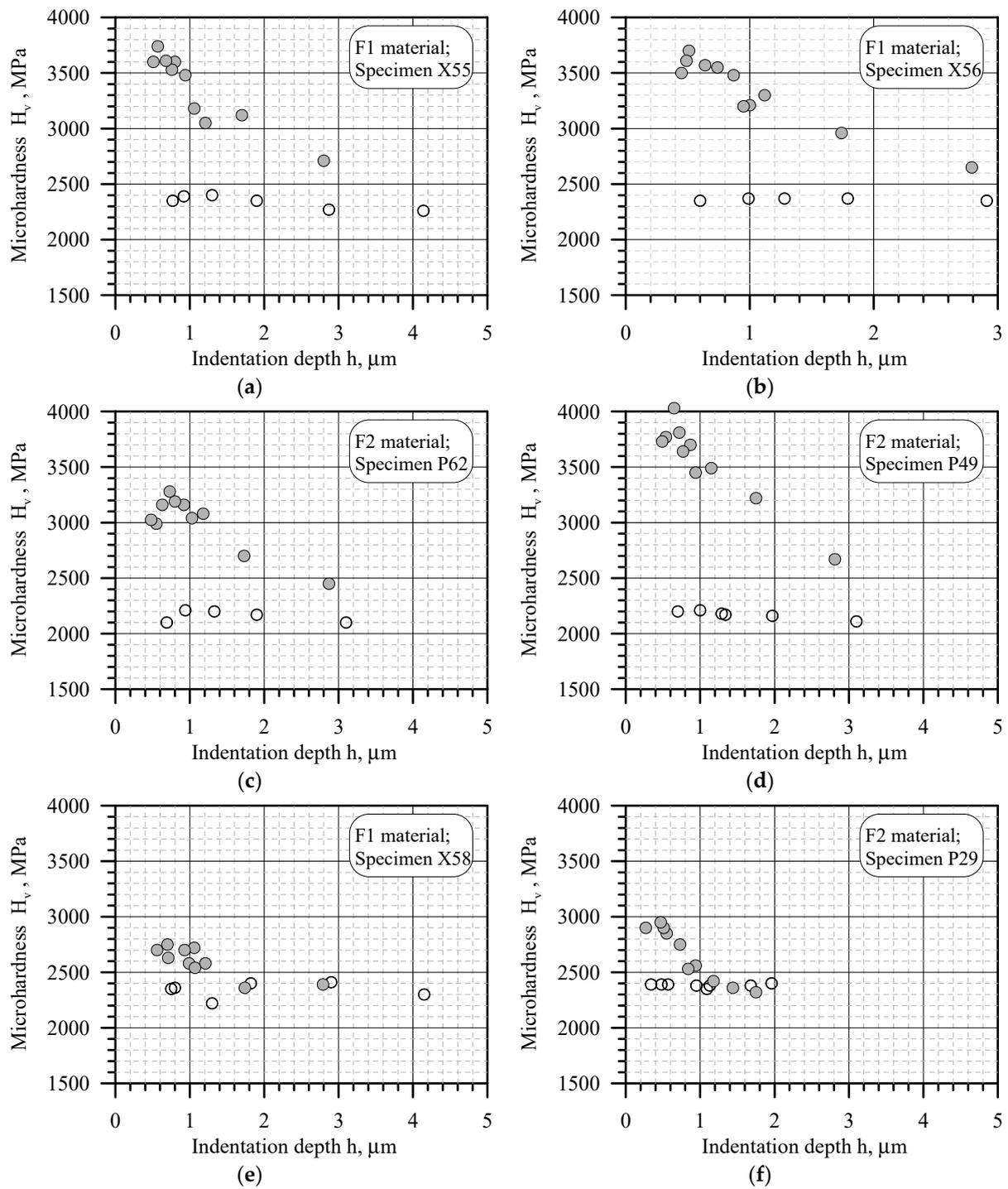


Figure 8. The dependence of microhardness on the indentation depth, h , for samples from F1 and F2 materials after the combined Fe^{4+} and He ions irradiation in various states (the legends for each sample are given in Tables 9 and 10). ●—ion irradiated region of sample, ○—unirradiated region of sample: (a) specimen X55, (b) specimen X56, (c) specimen P62, (d) specimen P49, (e) specimen X58, (f) specimen (P29).

Table 9. Ion irradiation parameters, microhardness in the initial H_V^0 and irradiated H_V^i state and radiation hardening ΔH_V^i for F1 material.

Sample	Irradiation Parameters			Indentation Depth h for H_V^i , μm	Microhardness		Radiation Hardening ΔH_V^i , MPa
	T_{irr}^i , $^\circ\text{C}$	D^i , dpa	η , appm/dpa		Irradiated Region H_V^i , MPa	Unirradiated Region H_V^0 , MPa	
X55	350	13	0.2	0.57	3740	2350	1390
X56	400	13	0.2	0.51	3700	2350	1350
X57	450	13	0.2	0.78	3550	2350	1200
X58	500	13	0.2	0.56	2700	2350	350
X59	550	13	0.2	0.80	2507	2270	237
X60	600	13	0.2	0.80	2240	2200	40

Table 10. Ion irradiation parameters, microhardness in the initial H_V^0 and irradiated H_V^i state and radiation hardening ΔH_V^i for F2 material.

Sample	Irradiation Parameters			Indentation Depth h for H_V^i , μm	Microhardness		Radiation Hardening ΔH_V^i , MPa
	T_{irr}^i , $^\circ\text{C}$	D^i , dpa	η , appm/dpa		Irradiated Region H_V^i , MPa	Unirradiated Region H_V^0 , MPa	
P30	350	15	0	0.34	3350	2360	990
P31	380	14	0.2	0.37	4000	2360	1640
P62	400	15	0	0.70	3276	2100	1176
P59	450	14	0.2	0.47	3843	2360	1483
P29	400	30	0	0.43	4214	2360	1854
P49	400	30	0.2	0.65	4019	2100	1919
P21	400	30	4.0	0.33	4278	2360	1918
P28	450	30	0	0.47	3670	2360	1310
P27	450	30	4.0	0.54	4050	2360	1690
P60	450	30	0.2	0.60	3552	2100	1452
P61	500	30	0	0.60	2460	2100	360
P22	500	30	0.2	0.64	2966	2166	866
P26	500	30	4.0	0.47	2953	2360	607
P18	600	30	0.2	0.60	2253	2100	153

As can be seen from Figure 8, the microhardness of the unirradiated regions does not practically depend on the indentation depth. This indicates the invariance of the Vickers microhardness value from the indentation depth, h , when measuring the microhardness of a homogeneous material with a constant strain rate.

The microhardness of the initial (unirradiated) material H_V^0 was determined outside the ion irradiation spot at $P = 0.5$ N, corresponding to $h = 2.5\text{--}3.0$ μm .

The microhardness of samples from F1 and F2 materials in the initial state H_V^0 , after ion irradiation H_V^i and the radiation hardening $\Delta H_V^i = H_V^i - H_V^0$ are shown in Tables 9 and 10, respectively.

The results presented in these tables show that, for all the samples studied, the maximum value of the dependence $H_V(h)$ was observed at $h = 0.3\text{--}0.8$ μm . This makes it possible to use the standard approaches for measuring the hardness of a thin-layer coating [55], as described earlier in Section 6 of this article. According to [55], the maximum value of microhardness over an indentation depth range of $0.3\text{--}0.8$ μm was taken as the microhardness of the ion-irradiated layer, ΔH_V^i .

As can be seen from the data presented in Table 9, the maximum radiation hardening for F1 material after ion irradiation is observed over the $T_{\text{irr}}^{\text{i}}$ range from 350 to 400 °C and reaches a value of $\Delta H_{\text{v}}^{\text{i}} \approx 1350\text{--}1390$ MPa. At higher irradiation temperatures, the radiation hardening decreases, reaching a minimum value of 40 MPa at $T_{\text{irr}}^{\text{i}} = 600$ °C.

A similar trend is observed for F2 material. Based on the data obtained for samples irradiated up to 14–15 dpa, radiation hardening does not change at $T_{\text{irr}}^{\text{i}} = 350\text{--}400$ °C and begins to decrease at higher temperatures (see the data for samples irradiated up to 30 dpa).

It can be noted that the maximum values of radiation hardening for F1 and F2 materials after ion and neutron irradiation are close (see Tables 7–10).

Comparing the data in Table 10, for samples irradiated with and without implantation of He, the following can be noted. For all temperatures at two levels of the target dose D^{i} , irradiation with only heavy ions and without He ions leads to weaker radiation hardening than after combined irradiation with heavy and light He ions. An increase in the helium content from 0.2 to 4 appm/dpa does not practically affect the radiation hardening (see samples P49, P21, P22 and P26 in Table 10).

Figure 9 shows the temperature dependences of radiation hardening for the F2 material after ion irradiation at various doses with heavy ions with and without implantation of He.

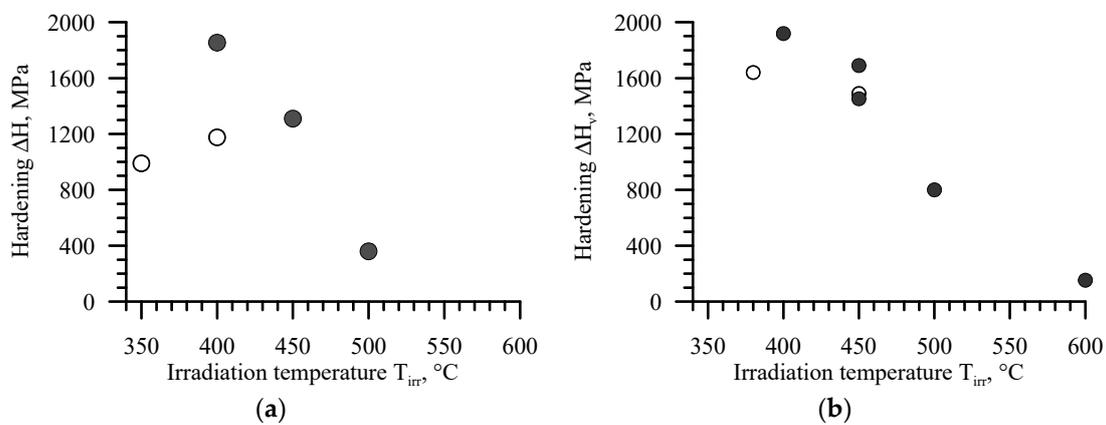


Figure 9. The dependencies $\Delta H_{\text{v}}^{\text{i}}(T_{\text{irr}}^{\text{i}})$ for F2 material after ion irradiation with heavy ions: Fe only (a) and after the combined irradiation with heavy ions Fe and He with $\eta = 0.2$ appm/dpa (b). ○— $D^{\text{i}} = 14\text{--}15$ dpa, ●— $D^{\text{i}} = 30$ dpa.

It can be seen from the presented data that combined irradiation with heavy ions and light He ions leads to an earlier saturation of the dose dependence of radiation hardening, compared with irradiation with heavy ions only. At the same time, irradiation without He leads to a more intensive decrease in radiation hardening with an increase in $T_{\text{irr}}^{\text{i}}$ in comparison to irradiation with He. This effect is apparently due to the stabilization of dislocation loops (one of the main sources of FMS hardening) by implanted helium that leads to the less intense dissociation of loops with the growth of $T_{\text{irr}}^{\text{i}}$ in comparison to irradiation with heavy ions only.

6.2. Microhardness for Austenitic Steels

6.2.1. After Neutron Irradiation

The microhardness of samples from materials A1 and A2 in the initial (unirradiated) state H_{v}^0 , irradiated condition H_{v}^{n} and the radiation hardening $\Delta H_{\text{v}}^{\text{n}} = H_{\text{v}}^{\text{n}} - H_{\text{v}}^0$ are shown in Table 11.

Table 11. Neutron irradiation parameters, microhardness in the initial H_V^0 and irradiated H_V^n states and the radiation hardening ΔH_V^n for A1 and A2 materials.

Materials	Reactor	T_{irr}^n , °C	D, dpa	H_V^n , MPa	H_V^0 , MPa	ΔH_V^n , MPa
A1	SM-3	60	10.2	2700	1420	1280
	WWER-440	280	15.7	3700	1420	2280
	WWER-440	280	33.7	3550	1420	2130
	WWER-440	400 (280)	33.7	3540	1420	2120
	WWER-440	450 (280)	33.7	3650	1420	2230
	WWER-440	500 (280)	33.7	3410	1420	1990
	WWER-440	550 (280)	33.7	3060	1420	1640
	WWER-440	600 (280)	33.7	2740	1420	1320
	BOR-60	330	10.8	3640	1420	2220
	BOR-60	500 (330)	10.8	3390	1420	1970
	BOR-60	500	29.0	3290	1420	1870
	BOR-60	550 (500)	29.0	2940	1420	1520
	BOR-60	600 (500)	29.0	2760	1420	1340
	SM-3 + BOR-60	500	11.3	3310	1420	1890
	SM-3 + BOR-60	550 (500)	11.3	3010	1420	1590
A2	SM-3	60	12.1	2780	1450	1330
	BOR-60	330	10.8	3520	1450	2070
	BOR-60	500	29.0	3330	1450	1880
	BOR-60	550 (500)	29.0	2950	1450	1500
	SM-3 + BOR-60	500	11.3	3180	1450	1730
	SM-3 + BOR-60	550 (500)	11.3	2980	1450	1530

To expand the range of neutron irradiation temperatures for the austenitic steels under consideration, additional data from [62] were also used. Reference [62] presents the study results of the effect of post-irradiation annealing on the microstructure and hardening of HC-18Cr-10Ni-Ti steel with a carbon content of 0.12% after irradiation in a BOR-60 reactor up to doses over 100 dpa over the T_{irr}^n range from 330 to 360 °C [62]. The radiation swelling of these samples did not exceed 1%. The mechanical properties of the irradiated HC-18Cr-10Ni-Ti steel were studied in the irradiation condition and after post-irradiation annealing for 20 h at temperatures $T_{ann} = 600, 700, 800$ and 900 °C. The test temperature was equal to 290 °C.

In reference [62], it was shown that, for $T_{ann} > 800$ °C, an almost complete recovery of hardening was observed. Therefore, in this work, we only used the data for $T_{ann} \leq 800$ °C.

Table 12 shows the values of radiation hardening in terms of $\Delta\sigma_\gamma$ for HC-18Cr-10Ni-Ti steel after neutron irradiation and post-irradiation annealing at $T_{ann} = 600\text{--}800$ °C [62], and the calculated values of ΔH_V^n . The ΔH_V^n value was calculated using the equation obtained in [28] for austenitic steels:

$$\Delta H_V = 3.03 \cdot \Delta\sigma_\gamma, \text{MPa} \quad (10)$$

Table 12. Radiation hardening in terms of $\Delta\sigma_Y$ and ΔH_V^n for HC-18Cr-10Ni-Ti steel after neutron irradiation at $T_{irr}^n = 330\text{--}360$ °C up to $D = 130\text{--}145$ dpa and post-irradiation annealing [62].

Temperature of Post-Irradiation Annealing, °C	$\Delta\sigma_Y$, MPa	ΔH_V^n , MPa
600	439	1330
700	104	315
800	4	12

It was assumed that the value ΔH_V^n , calculated from the data on the yield strength at 290 °C, is equal to that determined at a temperature of 20 °C.

Figure 10 shows the dependences of radiation hardening ΔH_V^n on T_{irr}^n for the austenitic steels under investigation. Some of the irradiation temperatures were modeled by the post-irradiation annealing of neutron-irradiated samples at higher temperatures.

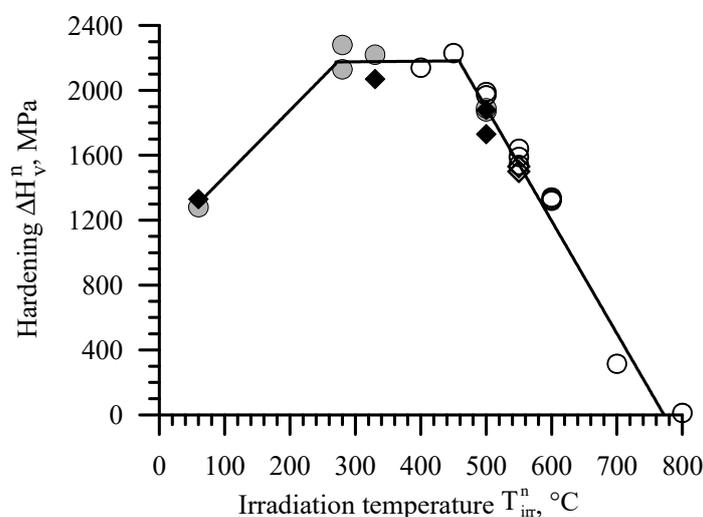


Figure 10. The dependence $\Delta H_V^n(T_{irr}^n)$ for A1 and A2 materials and for HC-18Cr-10Ni-Ti steel irradiated in reactors of various types with doses of 10.2–33.7 dpa: ●—A1 material after neutron irradiation; ○—A1 material and HC-18Cr-10Ni-Ti steel after post-irradiation annealing; ◆—A2 material after neutron irradiation; ◇—A1 material after post-irradiation annealing.

It can be seen that the presented dependencies are close for both investigated steels, regardless of the irradiation dose.

The dependence $\Delta H_V^n(T_{irr}^n)$ has a maximum over the T_{irr}^n range from 280 to 450 °C, which is a plateau when radiation swelling is absent. The obtained result fully corresponds to the data presented in [35], where the effect of T_{irr}^n on the yield strength increase under neutron irradiation is studied.

This type of the temperature dependence can be explained as follows. Dislocation loops and radiation-induced phases are the main contributors to the radiation hardening of austenitic steels, with the contribution of dislocation loops being dominant.

At low irradiation temperatures (up to 300 °C), at which there is no phase formation, the radiation hardening is determined only by dislocation loops. The concentration of dislocation loops corresponding to the saturation level (the equilibrium concentration of loops at a given temperature) increases with the irradiation temperature increase, reaching a maximum at ~300 °C [63]. Therefore, the radiation hardening increases with an increase in the irradiation temperature at doses corresponding the saturation of the dislocation loops concentration. A further increase in the irradiation temperature from ~300 °C to ~450 °C leads to a decrease in the saturation level of the dislocation loops concentration and a decrease in their contribution to hardening. On the other hand, T_{irr}^n increase leads to phase formation and an increase in their concentration that leads to the further hardening

of the material. These two processes compensate for each other, so that, over a range from $T_{irr}^n = 280$ to $T_{irr}^n = 450$ °C, radiation hardening practically does not change.

With a further increase in the irradiation temperature, thermally activated processes of the annihilation of radiation defects lead to a decrease in the concentration of dislocation loops. In addition, the size of the radiation-induced phases increases and their concentration drops drastically. This leads to a decrease in the radiation hardening induced by both phases and dislocation loops (see Figure 10).

6.2.2. After Ion Irradiation

Figures 11 and 12 show the typical dependences of the microhardness H_v on the indentation depth, h , for A1 and A2 materials, respectively, after ion irradiation. The results are given for different irradiation temperatures and damage doses.

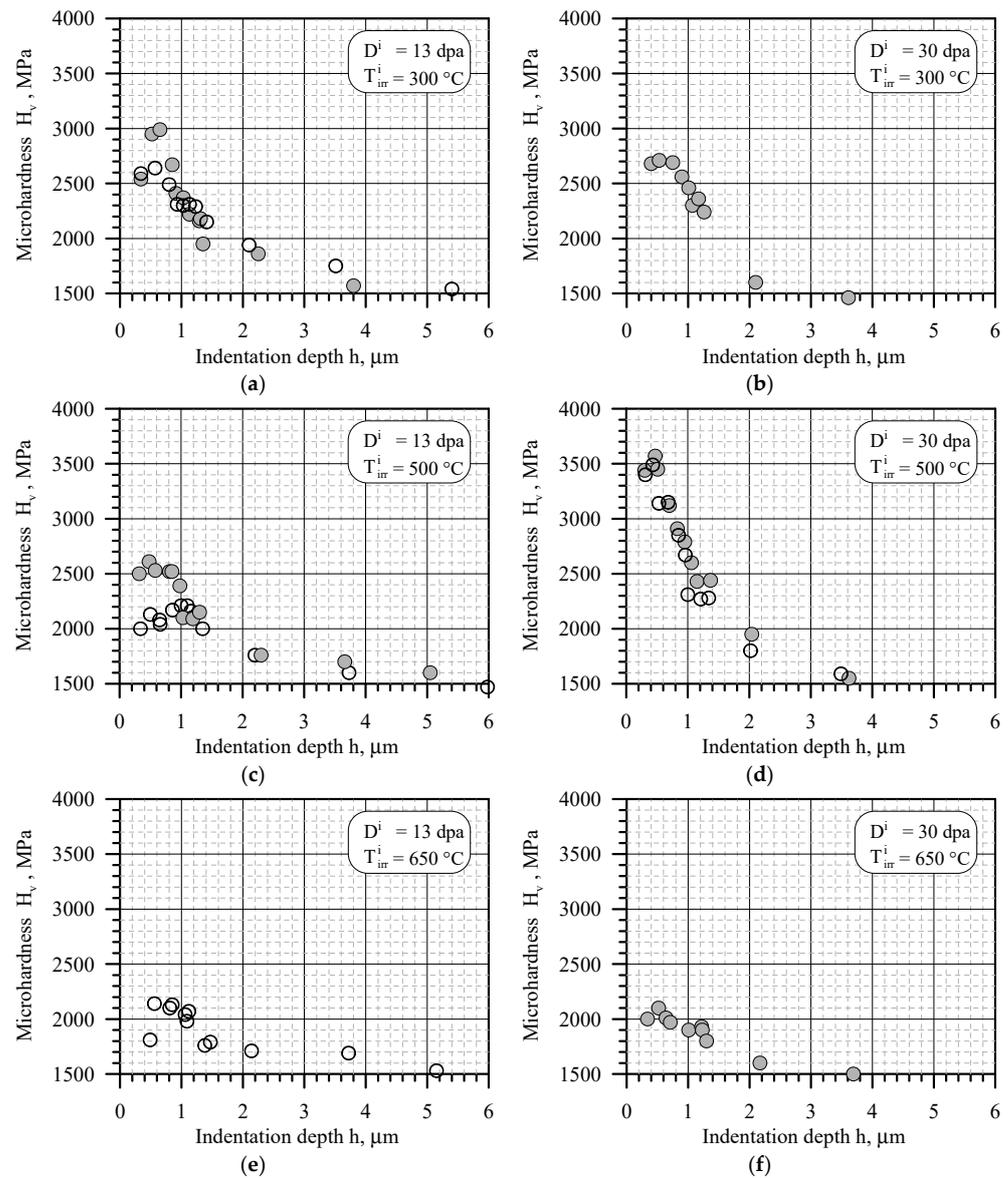


Figure 11. The dependences of microhardness on the indentation depth, h , for samples from A1 material after ion irradiation at different temperatures T_{irr}^i and target doses D^i : ○—irradiation without implantation of He; ●—irradiation with implantation of He to $\eta = 7$ appm/dpa. (a) $D^i = 13$ dpa, $T_{irr}^i = 300$ °C; (b) $D^i = 30$ dpa, $T_{irr}^i = 300$ °C; (c) $D^i = 13$ dpa, $T_{irr}^i = 500$ °C; (d) $D^i = 30$ dpa, $T_{irr}^i = 500$ °C; (e) $D^i = 13$ dpa, $T_{irr}^i = 650$ °C; (f) $D^i = 30$ dpa, $T_{irr}^i = 650$ °C.

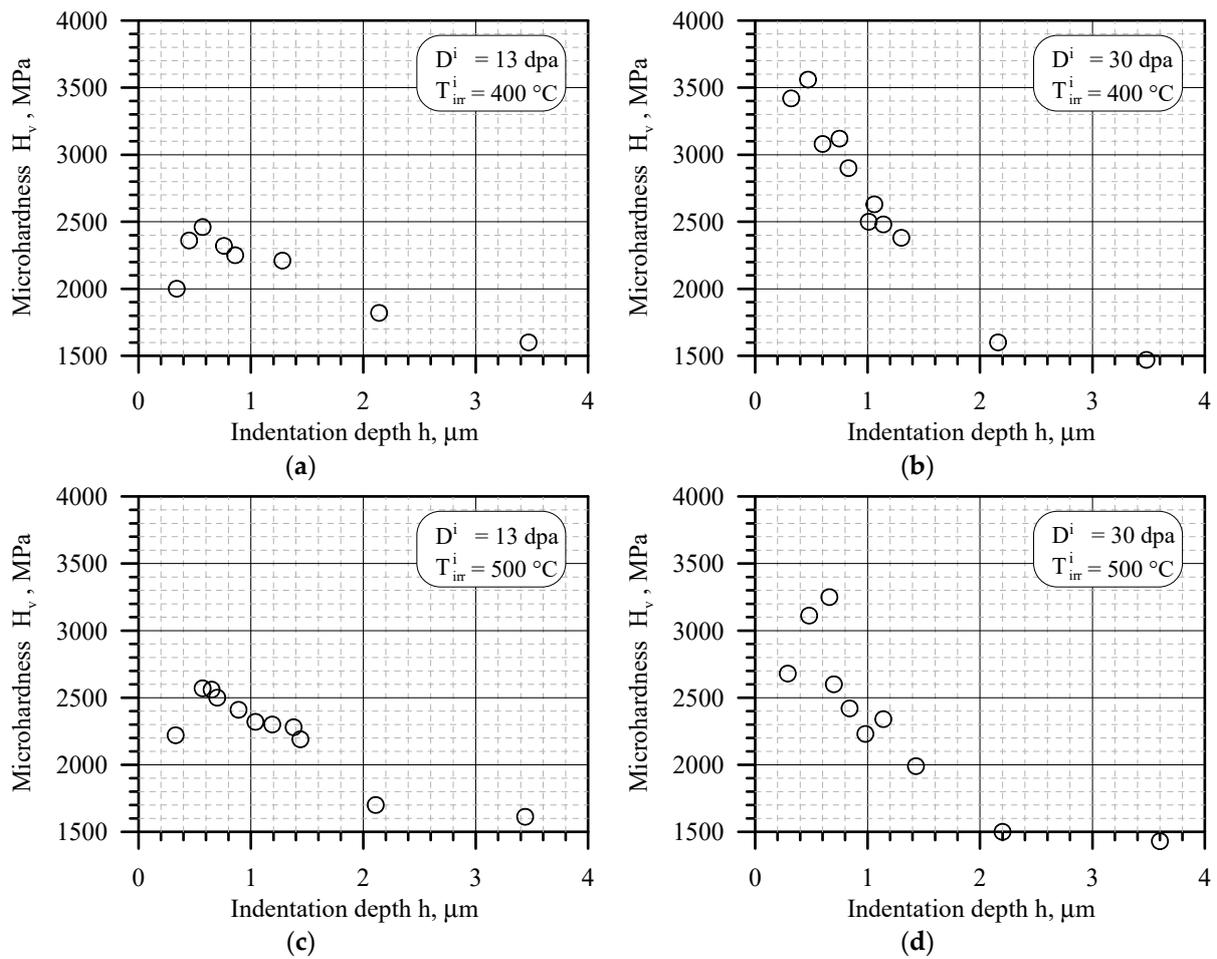


Figure 12. The dependences of microhardness on the indentation depth, h , for samples from A2 material after ion irradiation at different temperatures T^i_{irr} and target doses D^i without implantation of He. (a) $D^i = 13$ dpa, $T^i_{\text{irr}} = 400$ °C; (b) $D^i = 30$ dpa, $T^i_{\text{irr}} = 400$ °C; (c) $D^i = 13$ dpa, $T^i_{\text{irr}} = 500$ °C; (d) $D^i = 30$ dpa, $T^i_{\text{irr}} = 500$ °C.

The processing of microhardness measurements of all studied austenitic steels after ion irradiation showed that the dependence $H_v(h)$ sees a maximum at $h = 0.5$ – 1.0 μm .

According to the methodology presented above, the microhardness after ion irradiation was determined as the maximum value at an indentation depth up to 1 micron.

Tables 13 and 14 present the results of measuring the microhardness of the studied austenitic steels after ion irradiation and the calculated radiation hardening $\Delta H^i_v = H^i_v - H^0_v$. Here, the values H^0_v presented in Table 11 were used as the microhardness in the initial state.

Figure 13 shows the dependencies of radiation hardening ΔH^i_v on the temperature of ion irradiation T^i_{irr} for samples of A1 and A2 materials.

From the presented data, it can be concluded that, for ion irradiation with $D^i = 30$ dpa, the dependence $\Delta H^i_v(T^i_{\text{irr}})$ is similar to the dependence $\Delta H^n_v(T^n_{\text{irr}})$ for neutron irradiation. The maximum radiation hardening is observed at $T^i_{\text{irr}} = 400$ °C and this coincides with the hardening under neutron irradiation.

Table 13. Ion irradiation parameters, microhardness in the irradiated condition H_V^i and radiation hardening ΔH_V^i for A1 material.

Sample	Irradiation Parameters			Indentation Depth h for H_V^i , μm	Microhardness after Ion Irradiation H_V^i , MPa	Radiation Hardening ΔH_V^i , MPa
	T_{irr}^i , $^{\circ}\text{C}$	D^i , dpa	η , appm/dpa			
S38	300	13	0	0.52	2648	1228
S39	400	13	0	0.44	2395	975
S18	400	30	0	0.45	3685	2265
S40	500	13	0	0.98	2208	788
S19	500	30	0	0.44	3490	2070
S41	550	13	0	0.84	2121	701
S42	650	13	0	0.57	2140	720
S35	300	13	7	0.65	2984	1564
S23	300	30	7	0.51	2711	1291
S36	400	13	7	0.70	2471	1051
S43	400	13	7	0.57	2580	1160
S20	400	30	7	0.40	3700	2280
S37	500	13	7	0.47	2616	1196
S21	500	30	7	0.47	3568	2148
S44	550	30	7	0.43	2740	1320
S22	650	30	7	0.51	2100	680

Table 14. Ion irradiation parameters, microhardness in the irradiated condition H_V^i and radiation hardening ΔH_V^i for A2 material.

Sample	Irradiation Parameters			Indentation Depth h for H_V^i , μm	Microhardness after Ion Irradiation H_V^i , MPa	Radiation Hardening ΔH_V^i , MPa
	T_{irr}^i , $^{\circ}\text{C}$	D^i , dpa	η , appm/dpa			
2.95	300	13	0	0.58	2629	1179
2.96	400	13	0	0.58	2457	1007
2.98	500	13	0	0.58	2562	1112
2.93	400	30	0	0.46	3563	2113
2.94	500	30	0	0.65	3239	1789

The radiation hardening at $T_{\text{irr}}^i = 400\text{--}500$ $^{\circ}\text{C}$ is significantly lower (by 1000–1200 MPa) for $D^i = 13$ dpa than for $D^i = 30$ dpa. The dependence $\Delta H_V^i(T_{\text{irr}}^i)$ decreases with increasing temperature, without having a maximum in the temperature range $T_{\text{irr}}^i = 400\text{--}500$ $^{\circ}\text{C}$.

Estimating the effect of implanted He on the radiation hardening, the following can be noted. As can be seen from Figure 13a, for $D^i = 30$ dpa, He has practically no effect for the temperatures of maximum hardening. For $D^i = 13$ dpa, the implantation of He leads to a stronger hardening compared to irradiation without He.

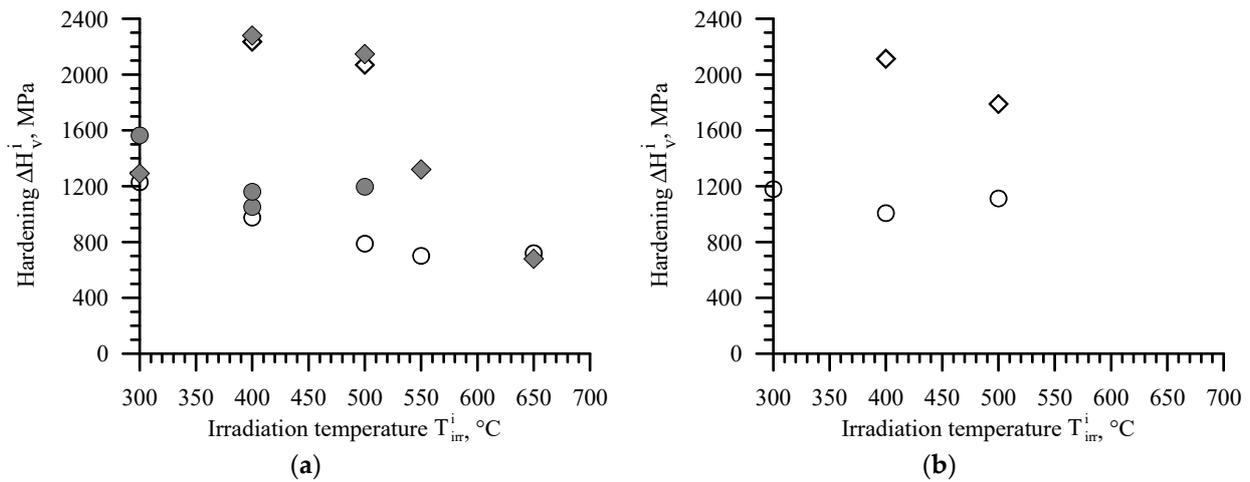


Figure 13. The dependencies of radiation hardening ΔH_v^i on the temperature of ion irradiation T_{irr}^i for samples of A1 (a) and A2 (b) materials: \circ — $D^i = 13$ dpa, without He implantation; \bullet — $D^i = 13$ dpa with He implantation to $\eta = 7$ appm/dpa; \diamond — $D^i = 30$ dpa, without He implantation; \blacklozenge — $D^i = 30$ dpa with He implantation to $\eta = 7$ appm/dpa.

7. Construction of the Transferability Functions

7.1. Ferritic-Martensitic Steels

Figures 14 and 15 show the dependences of radiation hardening on the irradiation temperature for F1 and F2 materials after neutron (see Tables 7 and 8) and ion irradiation (see Tables 9 and 10). For ion irradiation, the data were used only for regimes with implantation of He.

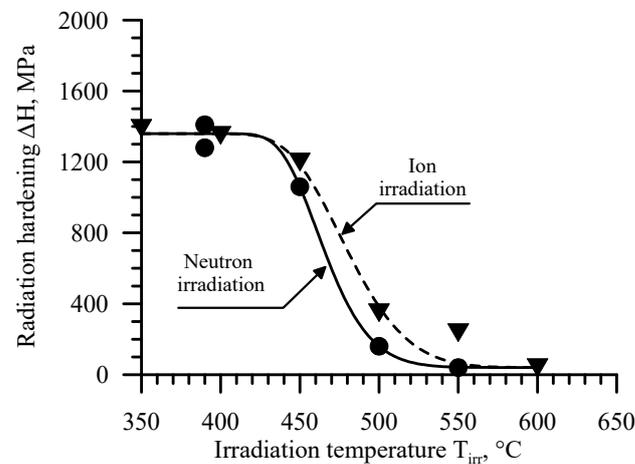


Figure 14. The dependences $\Delta H_v(T_{irr})$ for F1 material after neutron and ion irradiation: \bullet —neutron irradiation up to $D^n = 10$ –14 dpa; \blacktriangledown —ion irradiation up to $D^i = 13$ dpa with implantation of He to $\eta = 0.2$ appm/dpa; curves—approximation by Dependence (11).

It can be seen that the dependences $\Delta H_v(T_{irr})$ for FMS for both types of irradiation monotonically decrease. The maximum value of ΔH_v is observed at $T_{irr} = 350$ – 400 °C, and this maximum value for ion and neutron irradiation is close for both steels. At higher irradiation temperatures, the condition $\Delta H_v^n < \Delta H_v^i$ is fulfilled, which corresponds to variant “A” presented in Figure 3. This means that the change in microhardness after ion irradiation corresponds to the target dose in the third test zone.

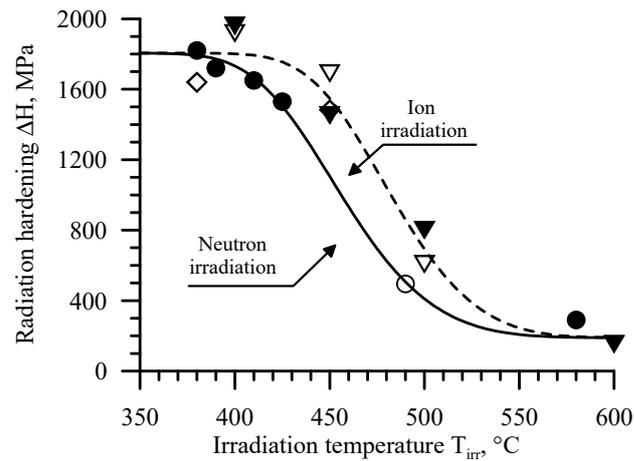


Figure 15. The dependences $\Delta H_v(T_{irr})$ for F2 material after neutron and ion irradiation: ●—neutron irradiation up to $D^n = 14\text{--}33$ dpa; ○—neutron irradiation up to $D^n = 11$ dpa [50]; ▼—ion irradiation up to $D^i = 30$ dpa with implantation of He to $\eta = 0.2$ appm/dpa; ▽—ion irradiation up to $D^i = 30$ dpa with implantation of He to $\eta = 4$ appm/dpa; ◇—ion irradiation up to $D^i = 14$ dpa with implantation of He to $\eta = 0.2$ appm/dpa; curves—approximation by Dependence (11).

To construct the transferability function, the temperature dependences of the radiation hardening of F1 and F2 materials for both types of irradiation were approximated using the following function:

$$\Delta H_v = \Delta H_v^{\max} \cdot \left(1 - \exp \left[- \left(\frac{A - T_{irr}}{B} \right)^\omega \right] \right) + \Delta H_v^{\min}, \quad (11)$$

where ΔH_v^{\max} and ΔH_v^{\min} are the maximum and minimum values of the radiation hardening over the considered irradiation temperature range; A, B and ω are some coefficients.

An analysis of the obtained data shows, that for both steels, the values ΔH_v^{\max} and ΔH_v^{\min} are close for ion and neutron irradiation. Therefore, when approximating experimental data, these values were taken to be the same for both irradiation types.

The coefficients in Equation (11) for the investigated steels were determined using the least squares method. As a result, the following dependencies were obtained.

For F1 material:

- After neutron irradiation:

$$\Delta H_v^n = 1320 \cdot \left(1 - \exp \left[- \left(\frac{630 - T_{irr}}{171.8} \right)^{8.43} \right] \right) + 40, \text{ MPa} \quad (12)$$

- After ion irradiation:

$$\Delta H_v^i = 1320 \cdot \left(1 - \exp \left[- \left(\frac{630 - T_{irr}}{159.7} \right)^{5.82} \right] \right) + 40, \text{ MPa}. \quad (13)$$

For F2 material:

- After neutron irradiation:

$$\Delta H_v^n = 1615 \cdot \left(1 - \exp \left[- \left(\frac{630 - T_{irr}}{186.1} \right)^{5.35} \right] \right) + 190, \text{ MPa}; \quad (14)$$

- After ion irradiation:

$$\Delta H_V^i = 1615 \cdot \left(1 - \exp \left[- \left(\frac{630 - T_{\text{irr}}}{159.0} \right)^{4.9} \right] \right) + 190, \text{ MPa}; \quad (15)$$

The obtained equations are valid for $T_{\text{irr}} \leq 630$ °C.

The obtained equations are also presented in Figures 14 and 15.

The data in Figures 14 and 15 allow one to determine the transferability function $\delta T_{\text{irr}}^{i-n} = \varphi(T_{\text{irr}}^n)$, which describes the shift of the ion irradiation temperature relative to the neutron one.

As can be seen from Figure 14, the transferability function $\delta T_{\text{irr}}^{i-n} = \varphi(T_{\text{irr}}^n)$ for F1 material is non-monotonic with maximum value of about 27 °C. It is appropriate to use the following conservative approach to describe the transferability function. For F1 material, the value $\delta T_{\text{irr}}^{i-n}$ in the irradiation temperature range from 350 to 600 °C can be assumed to be constant and equal to 10 °C. This estimation provides adequate modeling for the radiation hardening ΔH_V^n at $T_{\text{irr}}^n \leq 450$ °C, while the error in ΔH_V^n at $T_{\text{irr}}^n > 450$ °C is small.

For F2 material (see Figure 15), a monotonous decrease in $\delta T_{\text{irr}}^{i-n}$ is observed with irradiation temperature increase. It should be noted that the difference between ΔH_V^i and ΔH_V^n at $T_{\text{irr}} = 490$ °C, as observed in Figure 15, should be less, since the microhardness under neutron irradiation corresponds to $D^n = 11$ dpa, which, apparently, may be slightly lower than D_{sat} . At the same time, since D_{sat} , for this steel, does not exceed 14 dpa (see above), the error in the estimation of $\delta T_{\text{irr}}^{i-n}$ at $T_{\text{irr}} = 490$ °C is not significant.

By solving Equations (9) and (10), the transferability function $\delta T_{\text{irr}}^{i-n} = \varphi(T_{\text{irr}}^n)$ was obtained as a function of the T_{irr}^n value normalized to 100 °C in the temperature range 390–620 °C. This function, as shown in Figure 16, can be approximated using the following power dependence:

$$\delta T_{\text{irr}}^{i-n} = 32.6 - 3.2 \cdot 10^{-3} \cdot \left(\frac{T_{\text{irr}}^n}{100 \text{ °C}} \right)^5, \text{ °C} \quad (16)$$

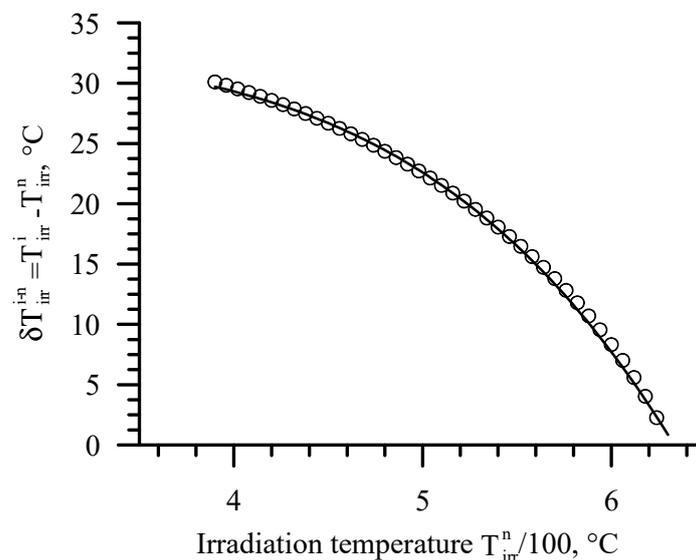


Figure 16. The transferability function $\delta T_{\text{irr}}^{i-n} = \varphi(T_{\text{irr}}^n)$ for F2 material: ○—the results of the joint solution of Equations (14) and (15); the curve is the Dependence (16).

For neutron irradiation temperatures $T_{\text{irr}}^n < 390$ °C, it is allowable to take $\delta T_{\text{irr}}^{i-n} = 0$ for F2 material.

Figures 14 and 15 solely include data with implantation of He. It is interesting to consider the data without implantation of He.

Figure 17 shows a comparison of the dependences $\Delta H_v(T_{irr})$ for F2 material after neutron and ion irradiation to $D^i = 30$ dpa without implantation of He. As seen, the radiation hardening for both types of irradiation is close.

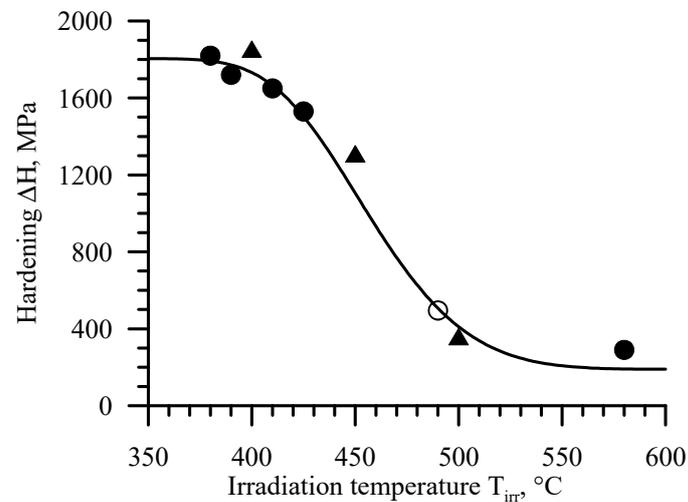


Figure 17. The dependences $\Delta H_v(T_{irr})$ for F2 material after neutron and ion irradiation without implantation of He: ●—neutron irradiation to $D^n = 14-33$ dpa; ○—neutron irradiation to $D^n = 11$ dpa [50]; ▲—ion irradiation to $D^i = 30$ dpa without implantation of He; curve—approximation by Equation (11).

Apparently, in the absence of He, the relaxation processes increase that cause the annihilation of dislocation loops. Consequently, the dynamic equilibrium between the generation and annihilation of dislocation loops shifts towards a decrease in their density. As a result, in the absence of He, the equilibrium concentration of dislocation loops at a given irradiation temperature becomes smaller, relative to the combined ion irradiation with implantation of He. Therefore, in the absence of He, the hardening also decreases.

The coincidence of radiation hardening after neutron irradiation and ion irradiation without He implantation seems to be the result of compensation of two processes: an increase in hardening due to an increase in dose rate and a decrease in hardening due to ion irradiation without He implantation, which is generated under neutron irradiation.

It is clear that, in the general case, such compensation may not occur and ion irradiation without implantation of He may lead to the underestimation of hardening. At the same time, the obtained experimental data show that the possible underestimation of hardening is insignificant at sufficiently high values of the target damage dose ($D^i = 30$ dpa). For a more accurate assessment of the effect of implanted He on the radiation hardening of FMS under ion irradiation, additional studies are required.

Based on the results obtained, in general, for the conservative modeling of radiation hardening of FMS under neutron irradiation, ion irradiation with implantation of He is required and $\delta T_{irr}^{i-n} = 0$ should be taken.

7.2. Austenitic Steels

The results obtained for austenitic steels after ion irradiation, and their comparison with the results after neutron irradiation, show the following.

For the target dose $D^i = 30$ dpa in the third test zone, the maximum value of radiation hardening after ion irradiation $\max(\Delta H_v^i)$ and after neutron irradiation $\max(\Delta H_v^n)$ are equal over the temperature range from 300 to 650°. This equality allows us to construct a transferability function $\varphi(T_{irr}^n)$ for $D^i = 30$ dpa, since it is possible to find such a shift of the ion irradiation temperature relative to the neutron irradiation temperature when Condition (3) is met.

For the target dose $D^i = 13$ dpa in the third test zone, the inequality $\max(\Delta H_V^i) < \max(\Delta H_V^n)$ is satisfied; therefore, any correction of T_{irr}^i does not provide the fulfillment of Condition (3). Therefore, according to the presented above methodology, it is necessary to change the test zone with the target dose from the third to the second or first.

Thus, the choice of test zone with the target dose for austenitic steels differs from that for ferrite-martensitic steels.

Let us consider a possible reason for this difference.

In the second section, it was shown that the choosing of the third test zone as a zone with target dose for ion irradiation can lead to the fulfillment of the inequality $\Delta \bar{\sigma}_Y = \Delta \sigma_Y^n$ for the same neutron dose in the case where the dose on the surface of the ion-irradiated sample $D_0 < D_{sat}$ and the radiation hardening of the material is weakly sensitive to the dose rate.

In the case where $D_0 \geq D_{sat}$, regardless of the sensitivity of hardening to the dose rate, the condition $\Delta \bar{\sigma}_Y = \Delta \sigma_Y^n$ is met.

As can be seen from the data presented in Figure 10, the shape of the temperature dependence of the radiation hardening, and the values of hardening after neutron irradiation, are close for the studied austenitic steels, regardless of the damage dose. This result indicates that $D_{sat} \leq 10.8$ dpa, regardless of the irradiation temperature (see Table 11). This D_{sat} value is also confirmed by other studies of irradiated chromium–nickel austenitic steels [3,33,35,36].

Figure 6 shows that, for samples from A1 and A2 materials irradiated to $D_i = 30$ dpa in the third test zone, $D_0 \approx D_{sat}$. At the same time, for these samples, $\max(\Delta H_V^i) = \max(\Delta H_V^n)$ or, in other words, $\Delta \bar{\sigma}_Y = \Delta \sigma_Y^n$.

For samples irradiated to $D_i = 13$ dpa, $D_0 < D_{sat}$ (see Figure 6). For these samples, the inequality $\Delta \bar{\sigma}_Y < \Delta \sigma_Y^n$ is met as $\max(\Delta H_V^i) < \max(\Delta H_V^n)$. This means that the sensitivity of radiation hardening to the dose rate for the studied austenitic steels is weak.

The choice of the second test zone as a target for A1 and A2 materials leads to the following.

It follows from the definition of the test zones given in Section 2 that the average damage dose in the second zone is 1.88 times less than the dose in the third zone used as the target for ion irradiation. Then, when $D^i = 13$ dpa in the second test zone, the ion irradiation dose should be $D^i = 24.4$ dpa in the third test zone at a depth of $1.4 \mu\text{m}$.

As seen, the change in the test zone from the third to the second with $D^i = 13$ dpa is, in fact, close to the ion irradiation with $D^i = 30$ dpa in the third test zone. At the same time, the distribution of the damage dose over the irradiated layer depth shows that the value of $D_{sat} = 10.8$ dpa is achieved at a depth of $\approx 0.5 \mu\text{m}$ (see Figure 6). In this case, the condition $D \geq D_{sat}$ is met for almost the entire ion-irradiated layer.

Thus, it may be concluded that, for austenitic steels after neutron irradiation up to doses $D^n \geq D_{sat}$, it is possible to find a transferability function if the condition $D \geq D_{sat}$ is met under ion irradiation in almost the entire irradiated layer.

Apparently, for $D_0 < D_{sat}$, the transferability function can be used when setting the target dose in the second test zone.

In connection with the above, all the data on the temperature dependence of radiation hardening of A1 and A2 materials and HC-18Cr-10Ni-Ti steel, including data after post-irradiation annealing, were used to determine the transferability function for austenitic steels. Data on the radiation hardening of A1 and A2 materials after ion irradiation were used only for regimes when a saturation of radiation hardening is achieved, namely, after ion irradiation to $D^i = 30$ dpa at all irradiation temperatures T_{irr}^i .

Figure 18 shows a comparison of the temperature dependences of radiation hardening after ion and neutron irradiation for A1 and A2 materials and HC-18Cr-10Ni-Ti steel.

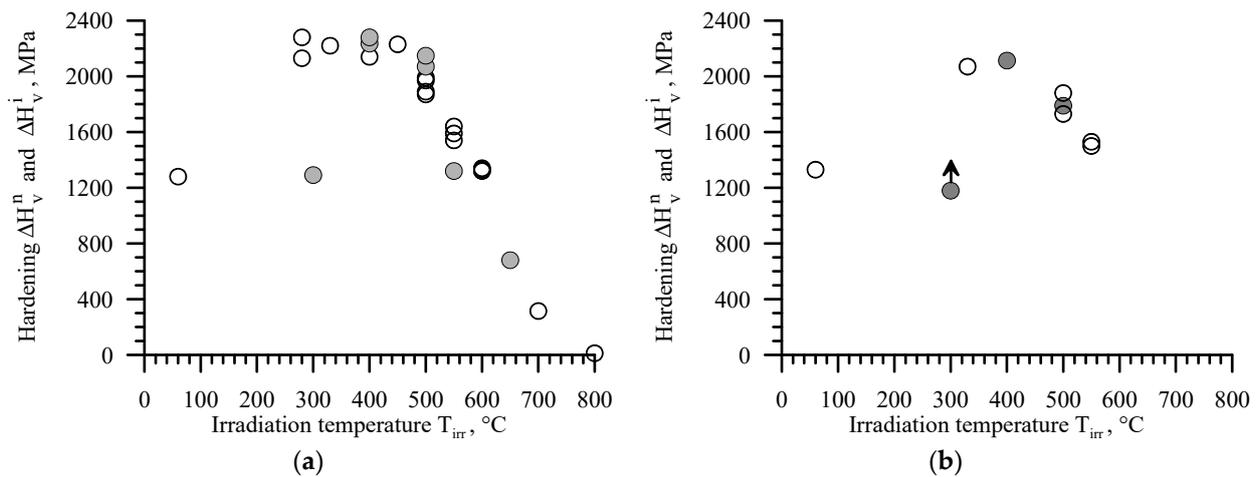


Figure 18. Comparison of the dependences of radiation hardening ΔH_V^n (○) and ΔH_V^i (●) on the irradiation temperature for A1 material and HC-18Cr-10Ni-Ti steel (a) and for A2 material (b) for the irradiation conditions when saturation of radiation hardening is achieved; point with arrow demonstrates similar trends between materials A1 and A2, corresponding to 13 dpa of ion irradiation. Realistic hardening for point with arrow is higher than measured value.

As can be seen from Figure 18, for the studied austenitic steels, the level of maximum radiation hardening after ion and neutron irradiation coincides. The temperature dependence after the maximum also coincides for ion and neutron irradiation for the studied steels. In other words, in this temperature range, the transferability function $\varphi(T_{irr}^n) = 0$. At $T_{irr}^n \leq 300$ °C, the temperature of ion irradiation should be increased compared to neutron irradiation to obtain the same radiation hardening.

Based on the experimental data presented in Figure 18, the following transferability function $\varphi(T_{irr}^n)$ was constructed:

$$\varphi(T_{irr}^n) = \begin{cases} -0.56 \cdot T_{irr}^n + 276.8, & \text{for } 60 \leq T_{irr}^n \leq 280 \text{ °C} \\ -T_{irr}^n + 400, & \text{for } 280 < T_{irr}^n \leq 400 \text{ °C} \\ 0, & \text{for } T_{irr}^n > 280 \text{ °C} \end{cases} \quad (17)$$

The obtained function is presented in Figure 19. It should be noted that the obtained function is applicable when radiation swelling is less than 1% and only for doses D equal to or above D_{sat} , corresponding to the saturation of radiation hardening.

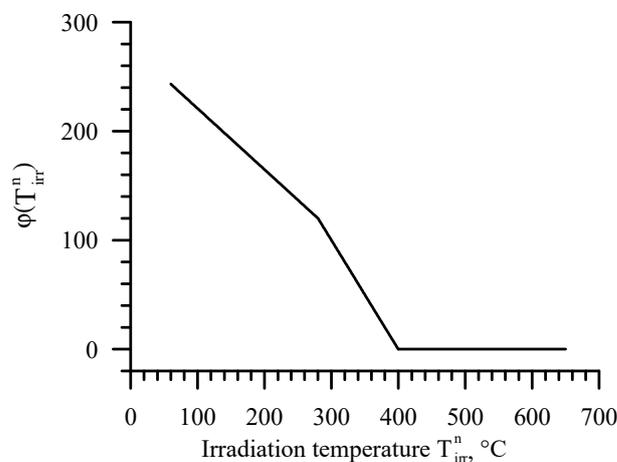


Figure 19. The transferability function $\varphi(T_{irr}^n)$ for austenitic steels provided the fulfillment of Condition (3).

In common cases, in order to adequately simulate hardening under neutron irradiation, it is recommended to perform ion irradiation with the implantation of He. At the same time, the necessity of He implantation under ion irradiation cannot be fully confirmed based on the results obtained. On the one hand, the effect of He on the radiation hardening of austenitic steels is practically absent at ion irradiation temperatures corresponding to the maximal hardening for doses corresponding to the saturation of radiation hardening (in the presented case, at $D^i = 30$ dpa). On the other hand, it is shown for FMS that the absence of He implantation can lead to a significant decrease in hardening.

8. Conclusions

1. A methodology for the ion irradiation and determination of microhardness and radiation hardening of an ion-irradiated layer is proposed, which allows one to obtain the radiation hardening of ferritic-martensitic and austenitic steels, as close as possible to the radiation hardening under neutron irradiation.

2. For ferritic-martensitic steels (FMS) of 12Cr-Ni-Mo-V-Nb grade (F1 material) and EP-823 grade (F2 material), and for austenitic steels of 18Cr-10Ni-Ti grade (A1 material) and 16Cr-20Ni-2.5Mo-Ti grade (A2 material), microhardness values were measured in the initial state and after neutron and ion irradiation at different temperatures. The dependences of radiation hardening, in terms of changes in microhardness ΔH_v on the irradiation temperature, are constructed. It is shown that the dependence of ΔH_v on the irradiation temperature for austenitic steels has a maximum.

3. The concept of a target zone in an ion-irradiated layer is formulated, in which the value of the damage dose should be equal to the damage dose under neutron irradiation. It is shown that, in order to obtain the same radiation hardening under neutron and ion irradiation, in general, the ion irradiation temperature must exceed the neutron irradiation temperature by a value of δT_{irr}^{i-n} , depending on the neutron irradiation temperature. Moreover, an equality of damage doses in the target zone and under neutron irradiation should be ensured. The dependence δT_{irr}^{i-n} on the neutron irradiation temperature T_{irr}^n is called the transferability function $\delta T_{irr}^{i-n} = \varphi(T_{irr}^n)$.

4. The effect of ion irradiation with the implantation of helium on the radiation hardening of FMS and austenitic steels is determined. It is shown that combined ion irradiation with helium implantation leads to a more intense radiation hardening with an increase in the damage dose and to a decrease in the dose D_{sat} at which the hardening saturation occurs.

5. The transferability functions for FMS and austenitic steels are determined on the basis of the obtained experimental data. These functions determine the connection between neutron and ion irradiation temperatures, for which the same radiation hardening is provided.

6. It is shown that, for austenitic steels, in contrast to ferrite-martensitic, the damage dose rate has a weak effect on radiation hardening. As a result, for austenitic steels, the transferability function can be obtained only for damage doses corresponding to the saturation of radiation hardening and such a dose should be provided practically over the entire depth of the ion-irradiated layer.

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