



# Article Viscosity Estimation of TiO<sub>2</sub>-Bearing Blast Furnace Slag with High Al<sub>2</sub>O<sub>3</sub> at 1500 °C

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**Abstract:** Slag compositions are significant for the viscosity of blast furnace slag. An improved Urbain model (IUM) was proposed by introducing  $R_5$  (( $X(CaO) + X(MgO) + 2X(TiO_2)$ )/( $2X(SiO_2) + 3X(Al_2O_3)$ )) and N ( $X(MgO)/3X(Al_2O_3)$ ) as the model parameters. By comparing IUM with other models, the model parameters of  $R_5$  and N are more reasonable and suitable for TiO\_2-bearing blast furnace slag, and IUM for predicting viscosity has a higher precision, and its relative error is only 10%. The viscosity isolines of the CaO–SiO\_2–15%Al\_2O\_3–MgO–2.5% TiO\_2 system were plotted, and the results show that the viscosity center of the slag is between  $R_{w2}$  ( $w(CaO)/w(SiO_2)$ ) = 0.77–1.39 and  $N_w$  ( $w(MgO)/w(Al_2O_3)$ ) = 0–1.37, the value of the viscosity center is 0.3 Pa·s, the viscosity increases gradually from the center to the outside, and the viscosity of the slag gradually decreases with the increase in  $N_w$  and  $R_{w2}$ . Furthermore, FTIR (Fourier Transform Infrared Spectroscopy) analysis was carried out in order to understand the mechanism between the slag structure and viscosity. With the increase in  $N_w$  and  $R_{w2}$ , the peak values of the symmetrical stretching vibration of non-bridging oxygen in the Si–O tetrahedral structure of slag decrease, and the slag structures depolymerize, which leads to the decrease in the viscosity of the slag.

**Keywords:** TiO<sub>2</sub>-bearing blast furnace slag; viscosity estimation; improved Urbain model (IUM); viscosity isoline; FTIR analysis

# 1. Introduction

In the process of blast furnace (BF) ironmaking, the viscosity of blast furnace slag plays an important role in blast furnace smelting, so it is very meaningful to investigate the relationship of viscosity with slag composition [1,2]. In recent years, many iron and steel enterprises have had to adopt oversea ores (from Australia), with the lack of high-grade ores in China, in order to reduce the production cost. The main difference between Australian ore and Chinese ore is that the content of  $Al_2O_3$  in Australian ore is higher, so the usage of Australian ore in large amounts will inevitably lead to an increase in the  $Al_2O_3$  content in the slag. In addition, the erosion of the blast furnace body is deteriorated due to the increase in smelting strength, and the quality of the hearth and bottom of the furnace has become the main factor affecting the life of the first generation of blast furnaces [3,4]. In order to protect the hearth and bottom of the furnace, alkaline low titanium slag is often used, and this leads to a small amount of TiO<sub>2</sub> in the slag [5]. Based on the abovementioned background, it is significant to investigate the effects of the composition of slag on the viscosity.

At present, the method for measuring the viscosity of blast furnace slag is commonly conducted by the rotating column method in a laboratory. Although this method can obtain the viscosity of BF slag accurately, it takes time and effort, along with a high economic cost.



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**Copyright:** © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). Therefore, developing a viscosity prediction model of slag containing  $TiO_2$  and high  $Al_2O_3$  not only obtains the viscosity of blast furnace slag efficiently but also saves time spent on viscosity measurement and reduces the cost. The traditional idea is that the content of  $Al_2O_3$  is more than 16%, which is very bad for blast furnace smelting.

The existing viscosity prediction models mainly include an empirical model and a structural model. The empirical model is based on a large number of viscosity experimental data to optimize the model and obtain the empirical equation to improve the accuracy of the model, and the model is mainly divided into the Urbain model [6], modified Urbain model [7–10], NPL model [11], Zhao model [12], etc. The structure model is mainly based on the slag structure, takes into account the deep internal structure of silicate melt, and includes the KTH model [13], Iida model [14], and Zhang model [15]. The structural model is characterized by the relationship between the slag viscosity and deep internal structure, so the calculation is relatively cumbersome and usually needs to be carried out with the help of specific software. Compared with other models [16-18], the Urbain model is simple and belongs to the empirical model, whose accuracy can be improved by using a large amount of data fitting. Previous research shows that the improved Urbain model (IUM) is suitable for the CaO–SiO<sub>2</sub>–Al<sub>2</sub>O<sub>3</sub>–MgO quaternary slag system by modifying the Urbain model parameters [19], and its relative error is 24%, which is better than that of most of the models. Therefore, an improved Urbain model (IUM) for CaO– SiO<sub>2</sub>–Al<sub>2</sub>O<sub>3</sub>–MgO–TiO<sub>2</sub> is proposed by introducing two of the modified model parameters,  $R_5 ((X(CaO) + X(MgO) + 2X(TiO_2))/(2X(SiO_2) + 3X(Al_2O_3)))$  and  $N (X(MgO)/3X(Al_2O_3))$ , for predicting the viscosity of the slag in this paper. In addition, the effect of slag structure on viscosity was also explored.

### 2. Improved Urbain Model (IUM Model)

### 2.1. Definition of the Model Parameters

The Urbain model is based on the theory of liquid dynamics, and its basic equation is the Weymann-Frenkel equation.

$$\eta = ATexp^{\frac{1000B}{T}} \tag{1}$$

where  $\eta$  is the viscosity (poise), A is the pre-exponential factor (poise/K), T is the absolute temperature (K), B is the viscous activation energy (J/mol), and A and B are functions of slag composition.

Taking logarithms on both sides of the Weymann-Frenkel equation, Equation (2) is obtained as follows:

$$\ln\frac{\eta}{T} = \ln A + \frac{1000}{T}B \tag{2}$$

In metallurgical production, the weight ratio of main basic oxides to acid oxides in slags is often expressed by either  $R_{w2}$  (the weight ratio of  $w(CaO)/w(SiO_2)$ ) or  $R_w$ (the weight ratio of  $(\Sigma w(basic oxide)/\Sigma w(acid oxide))$ . Presently,  $N_w$  (the weight ratio of  $w(MgO)/w(Al_2O_3)$ ) has been gaining more attention due to the increasing use of iron ores with high  $Al_2O_3$  content in recent years [17–19]. Therefore, an improved Urbain model (IUM) is proposed by introducing two of the model parameters,  $R_5$  and N, denoted, respectively, as Equations (3) and (4), which are more reasonable and suitable than not only the original model parameters (X = the mole fraction sum of acid oxide and  $\alpha =$  the ratio of the mole fraction sum of basic oxide to the mole fraction sums of basic oxide and amphoteric oxide) in the Urbain model but also  $R_2$  (the ratio of  $X(CaO)/(2X(SiO_2))$ ) and N, because R and N are significant for the metallurgical slag, and the two model parameters of  $R_5$  and N should be dependent on each other, but  $R_2$  is independent of N. In this paper, w in  $R_w$ ,  $R_{wi}$ , and  $N_w$  is the weight percent of the composition in the slag, and i in  $R_{wi}$  and  $R_i$  is the number of slag compositions used in the parameters.

$$R_5 = \frac{X(\text{CaO}) + X(\text{MgO}) + 2X(\text{TiO}_2)}{2X(\text{SiO}_2) + 3X(\text{Al}_2\text{O}_3)}$$
(3)

$$N = \frac{X(MgO)}{3X(Al_2O_3)}$$
(4)

where  $R_5$  is denoted as five components' basicity: CaO, MgO, TiO<sub>2</sub>, SiO<sub>2</sub>, and Al<sub>2</sub>O<sub>3</sub>, *N* is the mole ratio of MgO to Al<sub>2</sub>O<sub>3</sub>, and *X*(*i*) is the mole fraction of component *i*, such as CaO, MgO, Al<sub>2</sub>O<sub>3</sub>, SiO<sub>2</sub>, and TiO<sub>2</sub>.

The viscous activation energy *B* can be expressed as follows:

$$B = B_0 + B_1 R_5 + B_2 R_5^2 + B_3 R_5^3 \tag{5}$$

$$B_i = a_i + b_i N + c_i N^2 (i = 0, 1, 2, 3)$$
(6)

where  $B_i$ ,  $a_i$ ,  $b_i$ , and  $c_i$  are parameters of the model.

According to Equations (5) and (6), the expression of viscous activation energy B can be obtained as Equation (7).

$$B = a_0 + b_0 \times N + c_0 \times N^2 + (a_1 + b_1 \times N + c_1 \times N^2) \times R_5 + (a_2 + b_2 \times N + c_2 \times N^2) \times R_5^2 + (a_3 + b_3 \times N + c_3 \times N^2) \times R_5^3$$
(7)

The flow sheet for establishing the model is shown in Figure 1. The major steps in the flow sheet include: (1) Determining *m* (1000/*T*) and *n* (ln(( $\eta \times 10$ )/*T*)); (2) Determining *B* (*A*) by using the least square method with MATLAB R2014a software; (3) Model validation by model error analysis for model optimization. Mills et al. [20] said that the model can be considered as high reliability when the relative error ( $\Delta$ ) of the model is less than 35%. Thus,  $\Delta < 35\%$  was set as the decision condition for modeling, as shown in Figure 1, i.e., the database and representation data used for the model will be updated until  $\Delta < 35\%$ .

 $R_5$  indicates that TiO<sub>2</sub> plays a basic oxide role in the all kinds of TiO<sub>2</sub>-bearing blast furnace slag with high Al<sub>2</sub>O<sub>3</sub>. Zhang et al. [21] used molecular dynamics simulation and FTIR (Fourier Transform Infrared Spectroscopy) spectroscopy to study the effect of the change in the ratio of CaO/TiO<sub>2</sub> on the structure of the CaO–SiO<sub>2</sub>–15%Al<sub>2</sub>O<sub>3</sub>–MgO–TiO<sub>2</sub> quinary slag system at 1500 °C. The results showed that the substitution of CaO by TiO<sub>2</sub> could only lead to a slight change in the degree of polymerization, indicating that TiO<sub>2</sub> has similar effects to CaO and that TiO<sub>2</sub> is a basic oxide.

# 2.2. Determination of m and n

A total of 57 groups of different viscosity data [22–24] were selected, as shown in Table 1. The pre-exponential factor *A* and viscous activation energy *B*, corresponding to each group of viscosity data, can be obtained by fitting ln  $((\eta \times 10)/T)$  and 1000/T linearly. Taking a group of data (1# in Table 1) as an example, ln  $((\eta \times 10)/T)$  and 1000/T are fitted linearly. The results are shown in Figure 2. As seen in Figure 2, the fitting degree is high (to 0.9988), which indicates that there is a good linear relationship between ln  $((\eta \times 10)/T)$  and 1000/T.



Figure 1. Flow sheet for establishing the model.



Figure 2. Relationship between viscosity and temperature.

No	Cher	vical Compositions, Mass% Measured Vise		red Viscosit	osity, Pa·s *		
110	CaO	SiO <sub>2</sub>	MgO	$Al_2O_3$	TiO <sub>2</sub>	1450 °C	1500 °C
1# **	42.23	36.72	5.85	13.00	2.20	0.516	0.393
2#	40.68	35.37	6.75	15.00	2.20	0.538	0.409
3#	39.13	34.02	7.65	17.00	2.20	0.556	0.437
4#	41.88	36.42	6.50	13.00	2.20	0.475	0.309
5#	40.28	35.02	7.50	15.00	2.20	0.492	0.346
6#	38.67	33.63	8.50	17.00	2.20	0.528	0.389
7#	41.53	36.12	7.15	13.00	2.20	0.489	0.334
8#	39.88	34.67	8.25	15.00	2.20	0.453	0.335
9#	38.22	33.23	9.35	17.00	2.20	0.485	0.363
10#	43.06	35.89	5.85	13.00	2.20	0.484	0.339
11#	41.48	34.57	6.75	15.00	2.20	0.513	0.351
12#	39.90	33.25	7.65	17.00	2.20	0.530	0.372
13#	42.71	35.59	6.50	13.00	2.20	0.456	0.320
14# 15#	41.07	34.23	7.50	15.00	2.20	0.447	0.311
15#	12 35	35.30	7 15	12.00	2.20	0.409	0.302
10# 17#	42.55	33.80	8 25	15.00	2.20	0.412	0.207
18#	38.97	32.48	935	17.00	2.20	0.421	0.294
19#	43.86	35.09	5.85	13.00	2.20	0.425	0.300
20#	42.25	33.80	6.75	15.00	2.20	0.441	0.331
21#	40.64	32.51	7.65	17.00	2.20	0.488	0.339
22#	43.50	34.80	6.50	13.00	2.20	0.380	0.275
23#	41.83	33.47	7.50	15.00	2.20	0.410	0.317
24#	40.17	32.13	8.50	17.00	2.20	0.439	0.326
25#	43.14	34.51	7.15	13.00	2.20	0.319	0.220
26#	41.42	33.13	8.25	15.00	2.20	0.347	0.248
27#	39.69	31.76	9.35	17.00	2.20	0.380	0.273
28#	41.25	35.25	8.50	14.00	1.00	0.435	0.284
29# 20#	40.92	34.98	8.50	14.00	1.60	0.320	0.242
30# 31#	40.00	34.70	8.50	14.00	2.20	0.308	0.240
32#	39.50	33.82	8.50	16.50	1.00	0.422	0.203
33#	39.25	33.55	8.50	16.50	2 20	0.321	0.245
34#	39.25	33.55	8.50	16.50	2.20	0.321	0.245
35#	38.71	33.09	9.50	16.50	2.20	0.34	0.218
36#	38.17	32.63	10.50	16.50	2.20	0.327	0.202
37#	35.08	36.92	8.00	19.00	1.00	0.57	0.37
38#	36.88	35.12	8.00	19.00	1.00	0.54	0.35
39#	34.10	35.90	10.00	19.00	1.00	0.61	0.39
40#	35.85	34.15	10.00	19.00	1.00	0.59	0.38
41#	32.15	33.85	8.00	25.00	1.00	0.57	0.36
42#	33.80	32.20	8.00	25.00	1.00	0.55	0.34
45# 44#	31.10 22.78	32.82 21.22	10.00	25.00	1.00	0.59	0.38
44# 45#	32.70	31.22 26.41	8.00	23.00	2.00	0.57	0.30
45# 46#	36.37	34.63	8.00	19.00	2.00	0.53	0.33
47#	33.62	35.38	10.00	19.00	2.00	0.57	0.35
48#	35.34	33.66	10.00	19.00	2.00	0.55	0.33
49#	31.67	33.33	8.00	25.00	2.00	0.52	0.32
50#	33.29	31.71	8.00	25.00	2.00	0.51	0.3
51#	30.69	32.31	10.00	25.00	2.00	0.56	0.37
52#	32.27	30.73	10.00	25.00	2.00	0.57	0.35
53#	29.20	38.30	2.42	27.80	1.71	1.51	0.98
54#	34.50	35.40	2.11	25.80	1.86	0.94	0.63
55# 56#	38.00	32.00	2.14	25.20	2.28	0.61	0.41
50# 57#	32.00 39.40	37.00	4.70	23.30 21.20	1.02 2.17	0.85	0.57
51#	J7.40	52.00	<b>±.0</b> 0	Z1.ZU	4.17	0.41	0.27

Table 1. Viscosity data used in this study.

\*: Nos. 1–27: Experimental data measured by our research group; Nos. 28–57: Experimental data from the literature [22–24]. \*\*: Viscosity is 0.49, 0.47, 0.44, and 0.41 at 1460 °C, 1470 °C, 1480 °C, and 1490 °C, respectively.

The function relationship between the pre-exponential factor A and viscous activation energy B can be determined by linear fitting the obtained pre-exponential factor A and viscous activation energy B, which were obtained from 27 groups of viscosity data in Table 1, and the others were used for model validation. The fitting results are shown in Figure 3. It can be seen in Figure 3 that there is a linear relationship between  $-\ln A$  and B, the fitting degree is as high as 0.99192, and the relationship between the pre-exponential factor *A* and viscous activation energy *B* can be obtained, as shown in Equation (8).



 $-\ln A = 0.59B + 5.65 \tag{8}$ 

Figure 3. The functional relationship between *A* and *B*.

The calculated data of this model are close to those of Zhao's model [12] (0.501 and 7.681) and different from those of the original Urbain model [6] (0.29 and 11.57).

# 2.3. Determination of Viscous Activation Energy B

According to Equation (7), the viscous activation energy *B* can be calculated. Because there are 12 unknown parameters  $(a_i, b_i, c_i)$  in Equation (7), 12 independent equations (12 groups of viscosity data) are needed to solve the equation. In order to reduce the prediction error of the model, 57 equations are constructed by using 57 groups of viscosity data. The equations are solved by using the least square method with MATLAB software. The solution results of 12 unknown parameters of the model are shown in Table 2.

Table 2. Parameters	in	Equation	(7)	•
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i	a <sub>i</sub>	$b_i$	c <sub>i</sub>
0	265.57	-5846.71	-3353.78
1	-1202.13	38,717.73	565.08
2	1480.97	-78,892.50	26,151.32
3	-129.95	50,232.33	-28,517.60

Based on the data in Table 2, the functional relationship between the viscous activation energy *B* and model parameters (slag composition) can be obtained; Equation (7) can be rearranged, as shown in Equation (9).

$$B = 265.5775 - 5846.71 \times N - 3353.78 \times N^{2} + (-1202.13 + 38717.73 \times N + 565.0804 \times N^{2}) \times R_{5} + (1480.97 - 78892.5 \times N + 26151.32 \times N^{2}) \times R_{5}^{2} + (-129.951 + 50232.33 \times N - 28517.6 \times N^{2}) \times R_{5}^{3}$$

$$(9)$$

For the slag with a given composition,  $R_5$  and N are known. Combined with Equations (8) and (9), the pre-exponential factor A and viscous activation energy B of the slag with a different composition are obtained. Therefore, the viscosity of the slag can be predicted by Equation (1), using B and A obtained from Equations (8) and (9).

# 2.4. Model Validation

For the titanium-bearing blast furnace slag system, the comparison between the predicted viscosity and the measured viscosity is shown in Figure 4. It can be seen in Figure 4 that the viscosity predicted by IUM is in good agreement with the measured viscosity.



Figure 4. Comparison of predicted viscosity and measured viscosity.

In order to understand the reliability of the model, the relative error ( $\Delta$ ) of the model is determined by Equation (10), and the quantitative calculation is carried out.

$$\Delta = \frac{1}{n} \times \sum_{i=1}^{n} \left| \frac{\eta_{\text{Calc},i} - \eta_{\text{Exp},i}}{\eta_{\text{Exp},i}} \right| \times 100\%$$
(10)

where *n* is the number of data,  $\eta_{Calc}$  is the predicted viscosity, and  $\eta_{Exp}$  is the measured viscosity.

By combining Equations (1), (8), and (9), the viscosity of the slag can be predicted, and the  $\Delta$  of IUM using Equation (10) is about 10% and far less than 35% of the decision condition, as shown in Figure 1. Thus, it is considered that the prediction effect of IUM is good and that IUM is reliable and of high accuracy.

# 3. Results and Discussion

3.1. Viscosity Isoline of the CaO–SiO<sub>2</sub>–15%Al<sub>2</sub>O<sub>3</sub>–MgO–2.5%TiO<sub>2</sub> Slag System at 1500 °C

It is well known that slag viscosity plays an important role in blast furnace smelting. Therefore, the viscosity isolines of the CaO–SiO<sub>2</sub>–15%Al<sub>2</sub>O<sub>3</sub>–MgO–2.5%TiO<sub>2</sub> slag system at 1500 °C were plotted based on the calculation data obtained from IUM, and the results are shown in Figure 5. It can be seen in Figure 5 that the center viscosity calculated by IUM is 0.3 Pa·s, and its composition range is between  $R_{w2} = 0.77-1.39$  and  $N_w = 0-1.37$ . The viscosity isolines diffuse, and the viscosity increases from the center to the outside gradually. Figure 5 also shows that the viscosity is between 0.3 and 0.4 Pa·s when  $N_w$  is 0.45–0.55 and  $R_{w2} = 1.15-1.25$  (the shaded area in Figure 5), in which the slag composition conforms to the practical slag composition for blast furnace smelting. Thus, it shows that the viscosity isoline based on IUM is accurate and that IUM is reliable and will play a certain guiding role in optimizing the slag system for blast furnace smelting.



Figure 5. Viscosity isoline of the CaO–SiO<sub>2</sub>–15%Al<sub>2</sub>O<sub>3</sub>–MgO–2.5%TiO<sub>2</sub> slag system at 1500 °C.

## 3.2. Effect of $R_{w2}$ and $N_w$ on Viscosity

In the process of establishing the model, the model parameters ( $R_5$  and N) are defined with the mole fraction of each component, while the mass fraction is commonly used to express the composition of the slag in the practical production process. Therefore, the influences of  $R_{w2}$  (the ratio of  $w(CaO)/w(SiO_2)$ ) and  $N_w$  (the ratio of  $w(MgO)/w(Al_2O_3)$ ), instead of  $R_5$  and N, on slag viscosity are discussed according to the data obtained from IUM.

According to IUM, the viscosity of slag for a given composition at different temperatures can be calculated. When  $w(Al_2O_3)$  is 15% and  $w(TiO_2)$  is 2.2%, the viscosity under different  $R_{w2}$  and  $N_w$  conditions is calculated, and the influences of  $R_{w2}$  and  $N_w$  on slag viscosity are investigated, respectively. The results are shown in Figure 6.



Figure 6. Effect of compositions on viscosity at 1500 °C.

It can be seen in Figure 6a that the slag viscosity presents a downward trend when  $R_{w2}$  increases. It is considered that the absolute content of CaO in the slag increases with the increase in  $R_{w2}$ , CaO provides more (O<sup>2–</sup>) into the slag, which leads to the depolymerization of the complex silicate and aluminate compounds in the slag, simple structure compounds increase, and viscosity decreases. Figure 6b shows that the slag viscosity gradually decreases with the increase in  $N_w$ . Similar to the principle of increasing  $R_{w2}$ , the increase in  $N_w$  means there is more MgO in the slag, which is a basic oxide, can provide (O<sup>2–</sup>), and leads to the depolymerization of complex compounds and the decrease in the viscosity of the slag. As a summary, the calculated data obtained from IUM show

that, with the increase in  $R_{w2}$  and  $N_w$ , the slag viscosity gradually decreases, which is consistent with the change trend of the experimental data.

## 3.3. Effect of Slag Structure on Viscosity

In order to clear the effect of the slag structure on the viscosity, the slag quenched at 1500 °C was subjected to XRD analysis (X-ray Diffraction Spectroscopy, ultima IV, Rigaku, Japan) with Cu– $K_{\alpha}$ , at five degrees per minute. The XRD pattern is shown in Figure 7. However, there is no characteristic peak in the XRD pattern of the quenched slag, which indicates that the slag structure is amorphous and complex, and the slag maintains the glassy structure of elevated temperature during the quenching process. Therefore, in order to explore the influence of the slag structure on viscosity, the quenched amorphous slag samples were analyzed by FTIR (Nicolet iS10, Thermo Fisher Scientific, Shanghai, China).



Figure 7. XRD pattern of the quenched slag (Sample 17#).

The FTIR spectrums of the quenched amorphous slag are shown in Figure 8. Generally, the spectrum peak in the range of 800–1200 cm<sup>-1</sup> represents the symmetric stretching vibration of non-bridged oxygen in the Si–O tetrahedral structure, while the spectrum peak in the range of 400–600 cm<sup>-1</sup> represents the Al–O tetrahedral structure [25,26]. It can be seen in Figure 8 that the peak values in the range of 800–1200 cm<sup>-1</sup> and 400–600 cm<sup>-1</sup> gradually decrease with the increase in  $R_{w2}$  (Figure 8a) and  $N_w$  (Figure 8b), and this means that the peak values (trough depth) of symmetric stretching vibration of bridging oxygen in Si–O and Al–O tetrahedral structures in the slag were depolymerized into simple compounds, and the viscosity decreased. Therefore, the influence of the slag structure on the viscosity of the slag is due to the change in the composition of the slag.



(a) Effect of  $R_{w2}$  on the slag structure

(**b**) Effect of  $N_w$  on the slag structure

Figure 8. Effect of the compositions on the slag structure by FTIR.

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# 4. Conclusions

In this paper, an improved Urbain model (IUM) for CaO–SiO<sub>2</sub>–Al<sub>2</sub>O<sub>3</sub>–MgO–TiO<sub>2</sub> was proposed by introducing two of the modified model parameters,  $R_5$  ((X(CaO) + X(MgO) + 2X(TiO<sub>2</sub>))/(2X(SiO<sub>2</sub>) + 3X(Al<sub>2</sub>O<sub>3</sub>))) and N (X(MgO)/3X(Al<sub>2</sub>O<sub>3</sub>))). Based on IUM, the viscosity isolines were plotted, and the effects of the slag structure on the viscosity were discussed. The main achievements are summarized as follows.

- (1) The relative error of IUM is 10%, and this shows that IUM is reliable and is of high accuracy.
- (2) The viscosity isolines of the CaO–SiO<sub>2</sub>–15%Al<sub>2</sub>O<sub>3</sub>–MgO–2.5%TiO<sub>2</sub> melts were plotted using the data obtained from IUM. The viscosity center of the slag was located between  $R_{w2} = 0.77$ –1.39 and  $N_w = 0$ –1.37, and the viscosity value was 0.3 Pa·s, which gradually increased from the center to the outside.
- (3) With the increase in  $N_w$  and  $R_{w2}$ , the peak values of symmetric stretching vibration of bridging oxygen in the Si–O tetrahedral structure decreased, the complex Si–O tetrahedron and Al–O tetrahedron in the slag were depolymerized into simple compounds, and the viscosity of the slag decreased.

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