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A Numerical Study into the Effect of Machining on the Interaction between Surface Roughness and Surface Breaking Defects on the Durability of WAAM Ti-6Al-4V Parts

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Abstract: The airworthiness certification of military aircraft requires a durability analysis be performed using linear elastic fracture mechanics (LEFM). Furthermore, such analyses need to use a valid small crack growth equation. This paper focuses on the effect of rough surfaces and the effect of machining the surface on the durability of AM parts using LEFM and a valid small crack growth equation for the material. To this end, this paper analyses the effect of surface roughness on wire and arc additively manufactured (WAAM) Ti-6Al-4V titanium parts and the effect of machining on the durability of a part. The analysis reveals that the life of the component is a relatively strong function of the degree of surface roughness, and that the durability of a specimen is a strong function of the local radius of the curvature of the trough. It also appears that surfaces with tall narrow roughness will not overly benefit from partial machining of the surface.

Keywords: additive manufacturing; surface roughness; fatigue life; durability

1. Introduction

The 2019 memo by the Under Secretary, Acquisition and Sustainment [1] mandated the use of additive manufacturing (AM), and stated that AM parts could be used in both critical and non-critical applications. US Army Directive 2019-19 [2] subsequently stated that:

- (i) Advanced manufacturing can be used to address the readiness challenges posed by parts obsolescence, diminishing sources of supply, and sustained operations in austere environments.
- (ii) If employed to the maximum extent, advanced manufacturing could transform battlefield logistics through on-demand fabrication of parts close to the point of need, thus reducing the large number of parts stored and transported around the world.

The certification requirements for AM components are enunciated in United States Air Force (USAF) Structures Bulletin EZ-SB-19-01 [3], MIL-STD-1530D [4], and the United States (US) Joint Services Structural Guidelines JSSG2006 [5]. A proposed method for meeting the durability and damage tolerance (DADT) certification requirements is outlined in the recent keynote lecture at the 2021 EASA-FAA Industry Event [6]. This approach is built on the statement contained in MIL-STD-1530D [4] that analysis is the key to certification, and that the role of testing is merely to validate or correct the analysis.

In this context, is now well known that the fatigue life of additively manufactured (AM) parts can be a strong function of surface roughness [7–31]. Indeed, it has been



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Copyright: © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). suggested that the fatigue life of an AM part is driven by the size and nature of the defects at the root of the notches associated with the surface roughness [7,8,12,20,26–31].

When discussing the durability and damage tolerance (DADT) analyses needed to certify a part, MIL-STD-1530D states that the DADT analysis shall consider the disciplines of fracture mechanics as distinct from stress or strain life approaches. EZ-SB-19-01 [3] enunciates that both surface roughness and defect size be considered together. To this end, EZ-SB-19-01 requires the DADT analysis to utilize an "Equivalent Initial Damage Size (EIDS)" as per MIL-STD-1530D and JSSG2006. Here, EIDS is defined as per MIL-STD-1530D, viz: "an analytical characterization of the initial quality of the aircraft structure at the time of manufacture, modification or repair". Examples of this approach applied to the analysis of rough surfaces on the fatigue life of AM parts are given in [21]. However, as highlighted in [21], this analysis used the long crack da/dN versus ΔK curve rather than the corresponding short crack curve. (The authors of [21] explained that this was performed in order to simplify the analysis.) In this context, it is important point to note that, as explained in [32–34], when performing a durability analysis the analysis tools needed to perform a durability analysis differ from those needed to perform a damage tolerance assessment. This was first documented in the USAF study into cracking in an operational F-15 aircraft [33], where it was shown that, when the small crack da/dN versus ΔK curve was not used, the EIDS values were dependent on the load spectrum; see [32,33] for more details. Consequently, the durability analysis of AM parts requires the use of the associated small crack da/dN versus ΔK curve. Furthermore, as illustrated in Figure 4 in NASA HDBK-5010 [35], the durability analysis requires the use of the worst possible (i.e., upper bound) da/dN versus ΔK curve.

Fortunately, as noted in [25] and in [6,24,28,30,32,36–58], the growth of both long and small cracks in both conventionally manufactured and AM parts can be modelled using the Hartman–Schijve crack growth equation, viz

$$da/dN = D \left(\Delta\kappa\right)^p \tag{1}$$

where $\Delta \kappa$ is the Schwalbe crack driving force [52], viz:

$$\Delta \kappa = (\Delta K - \Delta K_{\text{thr}}) / (1 - K_{\text{max}} / \text{A})^{1/2}$$
⁽²⁾

Here, *a* is the crack length/depth, *N* is the number of cycles, *D* and *p* are material constants, *K* is the stress intensity factor, K_{max} and K_{min} are the maximum and minimum values of the stress intensity factor seen in the cycle, $\Delta K = (K_{max} - K_{min})$ is the range of the stress intensity factor that is seen in a cycle, ΔK_{thr} is the "effective fatigue threshold", and *A* is the cyclic fracture toughness. A related formulation is given in [59], where it is shown to be able to represent the growth of cracks that nucleate from near surface porosity. As explained in [32], the terms ΔK_{thr} and *A* are best interpreted as parameters that are chosen so as to fit the measured da/dN versus ΔK data.

A feature of this formulation is that, as noted in [47], for AM Ti-6Al-4V, when allowance is made for the effect of the different build processes and post built processing on the terms ΔK_{thr} and A, then each of the da/dN versus $\Delta \kappa$ curves in the 30 different tests analysed in [48], the 34 different tests in [49], and the 13 different tests in [50] all collapse onto the same master curve. As such, for both conventionally and AM Ti-6Al-4V the values of D (=2.79 × 10⁻¹⁰) and p (=2.12) are true materials constants, and remain the same for each of these seventy seven tests. The mean value of A given in [49] for as-built AM Ti-6Al-4V surfaces is 62 MPa \sqrt{m} .

Extensions of this observation, namely that when allowance is made for the effect of the different build processes and post built processing on the terms ΔK_{thr} and A then the da/dN versus $\Delta \kappa$ curves collapse to a single curve, to a wide range of conventionally manufactured, AM and cold spray additively manufactured (CSAM) materials are given in [53,54].

An additional feature of this formulation is that it has also been shown [6,25,27,28,30, 41,42,48,51,53] to account for:

- (a) The effect of different AM processes;
- (b) The effect of different build directions;
- (c) The residual stress fields induced by the different manufacturing process;
- (d) The variability in the crack growth rates associated with AM Ti-6Al-4V, thus enabling the worst case (mean- 3σ , where σ is the variance) crack growth curve required in [35] for a durability analysis to be determined; see [48].

As such, the focus of the present paper is to present a numerical study which uses the small crack growth equation for T-6Al-4V, i.e., Equation (1) with the threshold term set to a small value, to investigate the effect of as-built (rough) surfaces on the durability of an as-built WAAM Ti-6Al-4V specimen. Noting that surface machining has been proposed [16–18,22] to improve durability, an investigation into the effect machining rough surfaces is also presented.

In this context, it should also be noted that, as shown in [30,48,50], this formulation is known to be able to accurately compute the interaction between rough surfaces on the growth of small naturally occurring material discontinuities in conventionally built materials.

2. Materials and Methods

The majority of references quoted in this paper are taken from peer-reviewed journals. The conference proceedings referenced are fully refereed and can be obtained from the web. Similarly, the referenced texts are available via Google searches. The journal papers referenced are in journals that are either listed in SCOPUS or in the World of Science (WOS). The book chapters referenced are listed in SCOPUS. Seven of the references are available on various US Department of Defense DTIC websites, and one reference is available on a NASA website. Keywords that were used in these searchers were additive manufacturing, aircraft certification, durability, damage tolerance, Hartman–Schijve crack growth equation, and small cracks.

This paper uses finite element analysis to assess the effect of surface roughness and partial machining of the surface on the stress distribution of a WAAM built specimen. The surface roughness analysed were based on measurements obtained on WAAM samples built at Solvus Global in the US. The effect of these rough surfaces and partial machining on durability of the specimens was computed using the Hartman–Schijve crack growth equation, which as discussed above has been shown to be able to characterise the growth of both long and small cracks in AM Ti-6Al-4V. A key feature of this analysis is that we study the effect of both different surface roughness heights and different distances between the peaks in the surface roughness using a validated small crack equation for this material. This enables us to comment on the effect of tall narrow surface roughness's and hence make this paper potentially applicable to a greater range of WAAM built materials, i.e., WAAM built steels.

3. On the Interaction between Surface Roughness and Surface Breaking Defects on the Durability of WAAM Ti-6Al-4V Parts

The specimen geometry analysed in this study is the simple dogbone shape shown in Figure 1, where T is the height of the roughness and d is the separation between the peaks. The applied load is such that the working section sees a fatigue cycle with a (average section) maximum stress of 600 MPa and an *R* ratio of R = 0.1. (The *R* ratio in a fatigue test is defined as the minimum stress in a cycle divided by the maximum stress in the cycle.) In this preliminary study, specimens with both smooth (i.e., the baseline), and rough surfaces are analysed, as is the effect of machining the surface. The heights (T) of the surface roughness analysed were taken to be either 0.361, 0.142, or 0.071 mm; see Figures 2 and 3. These values were taken from measurements on representative WAAM specimens.



Figure 1. A schematic diagram of the baseline model, smooth surface, dimensions are in mm.



Figure 2. A schematic diagram of the as-built specimen with a wavy surface, dimensions are in mm.



Figure 3. A schematic diagram of specimen with machined surfaces, dimensions are in mm.

The analysis associated with the "machined surfaces" assumed that half the height (T) had been removed by the machining process. Schematic diagrams showing the geometry of the baseline (smooth surface) specimen, the rough surface specimen and the machined surface specimen geometries are shown in Figures 1–3. Nine different cases were analysed. These cases are outlined in Tables 1 and 2.

	Case	Surface Machined	T (mm)	Computed Life (Cycles)				
d (mm)				Semi-Circular Surface Crack	Reduction * in Fatigue Life (% diff)	Corner (Quadrant) Crack	Reduction * in Fatigue Life (% diff)	
1.57	1	No	0	16786	Baseline	16428	Baseline	
	2	Yes	0.071	9877	41.2	9610	41.5	
	3	No	0.142	7418	55.8	8018	51.2	
	4	No	0.1815	7112	57.6	6896	58.0	
	5	Yes	0.1815	5699	66.0	6162	62.5	
	6	No	0.363	2888	82.8	4971	67.7	
2.42	1	No	0	16786	Baseline	16428	Baseline	
	7	No	0.071	12585	25.0	13270	19.2	
	8	Yes	0.071	10464	37.7	10578	35.5	
	9	No	0.142	9765	41.8	10235	37.7	

Table 1. Summary of the fatigue life analyses for an initial crack depth of 0.03 mm.

* From the life of the baseline specimen.

	Fable 2. Summary	of the fat	tigue life anal	yses for an initial	crack depth of 0.254 mn
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		Surface Machined	T (mm)	Computed Life (Cycles)				
d (mm)	Case			Semi-Circular Surface Crack	Reduction in Fatigue Life (% diff) from the Baseline	Corner (Quadrant) Crack	Reduction in Fatigue Life (% diff) from the Baseline	
1.57	1	No	0	5875	Baseline	5639	Baseline	
	2	Yes	0.071	3275	44.3	3401	39.7	
	3	No	0.142	2746	53.3	3058	45.8	
	4	No	0.1815	2399	59.2	2186	61.2	
	5	Yes	0.1815	2186	62.8	2468	56.2	
	6	No	0.363	1135	80.7	1924	65.9	
2.42	1	No	0	5875	Baseline	5639	Baseline	
	7	No	0.071	4536	22.8	4701	16.6	
	8	Yes	0.071	3693	37.1	3883	31.1	
	9	No	0.142	3463	41.1	3730	33.9	

A durability analysis was then performed using Equation (1) with $D = 2.79 \times 10^{-10}$, p = 2.12, and A = 62 MPa \sqrt{m} . As noted above, this value of A represents the mean value given in [49] for as-built AM Ti-6Al-4V. The durability analysis requires the stress intensity factor solutions along the crack front to be determined as the crack grows. The approach adopted in this study to determine the stress intensity factor solutions is as outlined in [60]. This is a semi-analytical approach which has the advantage that it only uses the uncracked finite element model, and that the cracks are not explicitly modelled; see [60] for more details.

To be consistent with:

- (i) Appendix X3 of the fatigue test standard ASTM E647-15 [61], which questions the existence of a fatigue threshold for small (naturally occurring) cracks.
- (ii) The JSSG2006 [5] and NASA-HDBK-5010 [35] recommendations that the A-basis (mean-3σ) curves be used; see [48] for a more detailed discussion on this statement.
- (iii) The statement in the USAF Damage Tolerance Design Handbook [62] that the time to initiation is small.

(iv) The methodology used in [45] to predict the crack growth history of cracks that nucleated near surface cracking from material discontinuities in AM Ti-6Al-4V, and, as per [37,41–43,48], to allow for the fastest possible growth the durability analyses used $\Delta K_{\text{thr}} = 0.1$ MPa $\sqrt{\text{m}}$.

As noted above, the USAF Damage Tolerant Design Handbook [62] explains that more than 95% of the life of an airframe is consumed in crack growth so that the time consumed by crack initiation is minimal. Similar conclusions can be found in the USAF Durability Design Handbook [63], and the USAF approach to assessing the risk of failure [64]. These findings agree with those documented in [65], which discusses the Royal Australian Air Force (RAAF) experience with the RAAF F/A-18 fleet. Consequently, since it is accepted that AM material is generally more susceptible to cracking than conventionally manufactured aircraft quality materials and as explained above, the analysis performed in this section essentially enforces the requirement that time to crack initiation be minimal.

In this analysis, it was first assumed that the initial material discontinuity was either a small 0.03 mm deep surface breaking semi-circular crack, or a 0.03 mm quadrant crack located as per Figure 4. In each case, the crack is located at the notch root. The size of this initial defect was taken from [12,22]. The analysis was then repeat for the case when these initial discontinuities were 0.254 mm deep. In each case, for simplicity, the analyses assumed that the surface roughness was a simple wavy pattern; see Figures 3–5. The dimensions of this idealized wavy were taken from a typical WAAM specimen built by Solvus Global, for which the build parameters were not recorded. The uncracked finite element models corresponding to Cases 6 and 9 are shown in Figures 6 and 7. This analysis revealed that the stress concentration factors for Cases 6 and 9 were approximately 2.54 and 1.38, respectively.



Figure 4. A schematic diagram showing the locations of the cracks analysed in this study.



Figure 5. A photograph showing the "wavy" surface of a typical as-built billet.



Figure 6. The local stress field associated with a rough surface, Case 6.



Figure 7. The local stress field associated with a rough surface Case 9.

The effect of these various surface roughness on the computed fatigue lives for the case of a 0.03 mm semi-circular initial crack are given in Table 1 and Figures 8 and 9. (As shown in Tables 1 and 2, the effect of surface roughness on the fatigue life of a specimen containing a corner crack is very similar to that of a semi-elliptical centrally located crack.) Here, we see that Case 7, which has a roughness that has the smallest height and the largest radius of curvature, is least effected by surface roughness, and that Case 6, which has the greatest height and the smallest radius of curvature is the most effected by surface roughness. Indeed, examining Tables 1 and 2, and Figures 8 and 9, it appears that the local radius of curvature (associated with the surface roughness) is a key driver in reducing the fatigue life of the specimen.



Figure 8. The effect of surface roughness on the fatigue life from an initial semi-elliptical defect size of 0.03 mm. Case 1 is the baseline specimen with no surface roughness.



Figure 9. The effect of surface roughness on the fatigue life from an initial semi-elliptical defect size of 0.254 mm. Case 1 is the baseline specimen with no surface roughness.

4. Effect of Machining the Rough Surfaces

The paper by Seungjong Lee et al. [22] studied the effect of machining rough surfaces in Ti-6Al-4V specimens fabricated by a laser beam powder bed fusion (LBPF). It concluded that the fatigue lives of as-built and polished specimens, where only half of the height of the surface roughness was removed, were similar. However, the degree of surface roughness in these specimen tests was low. The question thus arises: Is this generally true, or does the result depend on the nature and the height/depth of the surface roughness? To investigate this question, we next studied the cases where the surfaces associated with the two prior studies were assumed to be machined such that 1/2 of the height was removed; see Figures 10 and 11.



Figure 10. The assumed machined surface profile for case when the original peak to trough height was 0.363 mm.



Figure 11. The assumed machined surface profile for case when the original peak to trough height was 0.141 mm.

The effect of machining on the stress distributions associated with Cases 5 and 8, which are the machined versions of Cases 6 and 9, are shown in Figures 12 and 13. Comparing Figures 6 and 12, and Figures 7 and 13, which compare the stress distribution in both un-machined and machined specimens, we see that whilst machining does indeed reduce the local stress concentration it does not remove it. The reason for this can be seen in Figure 14, which is associated with Case 6. Here, we see that the load path is such that very little load flows into the upper half (near peak) region that juts above the mean height of the surface. This phenomenon can be expected to occur whenever the height of the peak (T) divided by d, the separation between the peaks, is large.



Figure 12. The local stress field associated with Case 5, which is the "machined" version of Case 6.



Figure 13. The local stress field associated with the Case 8, which is the "machined" version of Case 9.



Figure 14. An expanded view of the stress field associated with Figure 6.

Comparing Figures 7 and 13, we see that the reduction in the stress field associated with Case 9 is very small. As shown in Table 1, this results in a small improvement in the fatigue life due to machining. On the other hand comparing Figures 6 and 12 we see that (in this instance) machining results in a slightly greater reduction in the local stress concentration and hence a greater increase in the fatigue life. In other words, it appears that surfaces with tall narrow roughness exhibit the largest reductions in fatigue life and that these cases do not overly benefit from partial machining of the surface.

The results of the durability analysis associated with these machined surfaces are also shown in Figures 8 and 9, and Tables 1 and 2. Here, we see that, as first reported in [22], removal of a half of the height T does not appear to make a significant difference to the fatigue life. As such, this study, when taken in conjunction with the experimental results presented in [22], suggests that partial machining may only result in a relatively small improvement in durability.

5. Conclusions

MIL-STD-1530D and USAF Structures Bulletin require that any durability analysis be performed using linear elastic fracture mechanics (LEFM). Furthermore, such analyses need to use a valid small crack growth equation. To the best of the authors' knowledge, this study is the first paper to numerically study the effect of rough surfaces and the effect of machining the surface on the durability of AM parts using LEFM and a valid small crack growth equation for the material. Indeed, a key feature of this analysis is that it uses the Hartman–Schijve crack growth equation, which has been shown to accurately model the growth of small cracks in both conventionally built and AM Ti-6Al-4V.

The primary conclusions that can be drawn from this preliminary analysis are:

- 1. Comparing the fatigue life-associated surfaces left in the unmachined state, the durability of an AM part appears to be a relatively strong function of the local radius of the curvature of the trough associated with the rough surfaces.
- 2. Surfaces with tall narrow roughness exhibit the largest reductions in fatigue life and these cases do not overly benefit from partial machining of the surface.
- 3. The size of the initial material discontinuities, porosity, lack of fusion, etc., associated with the AM process appears to strongly affect the fatigue life of the part.

It should be stressed that whilst this finding may be intuitively obvious, this is the first time that this observation has been supported by a detailed numerical study based on a crack growth equation that has been shown to hold for the growth of small naturally occurring cracks in AM Ti-6Al-4V.

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