

Article

Properties in Langmuir Monolayers and Langmuir-Blodgett Films of a Block Copolymer Based on *N*-Isopropylacrylamide and 2,2,3,3-Tetrafluoropropyl Methacrylate

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Abstract: The amphiphilic block copolymer poly(*N*-isopropylacrylamide)-Ge(C₆F₅)₂-poly(2,2,3,3-tetrafluoropropyl methacrylate) was prepared by the reaction of chain transfer to *bis*-(pentafluorophenyl)germane during the polymerization of *N*-isopropylacrylamide and the subsequent postpolymerization of isolated functional polymers in 2,2,3,3-tetrafluoropropyl methacrylate. The conversion of the block copolymer was 68% and the molecular weight of the sample was 490,000 g/mol. The colloidal chemical properties of Langmuir monolayers and Langmuir-Blodgett films of synthesized block copolymer have been studied. For comparison, a functional polymer, namely, poly-*N*-isopropylacrylamide with terminal -Ge(C₆F₅)₂H group, was synthesized and studied. The concentrations of spreading solutions were selected and the effect of subphase acidity on the formation of monolayers of macromolecules of the block copolymer was studied. It was found that regardless of the acidity of the subphase, high pressure of fracture of films are characteristic of monolayers of collapse pressures $\pi_{\max} = (48-61)$ mN/m. The morphology of the Langmuir-Blodgett films of functional polymer exhibit isolated elongated micelles with high densities in the form of "octopus" on the periphery of which there are terminal hydrophobic groups. For the Langmuir-Blodgett film of block copolymer, a comb-like structure is observed with characteristic protrusions.

Keywords: block copolymer; synthesis; poly-*N*-isopropylacrylamide; Langmuir monolayer; Langmuir-Blodgett films



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1. Introduction

Currently, poly-*N*-isopropylacrylamide (PNIPAAm) remains the most studied polymer, demonstrating a phase transition from an insoluble to a soluble state at a certain temperature. It exhibits the so-called lower critical dissolution temperature (LCST) behavior [1,2]. Below LCST, PNIPAAm is hydrophilic, and its copolymerization with hydrophobic or hydrophilic comonomers changes its properties, lowering or increasing the LCST of its aqueous solutions, respectively. At present, the copolymerization of the NIPAAm monomer makes it possible to obtain copolymers of various structures with different phase transition temperatures, which have already found applications in medicine [3–5].

Piskin E. et al. [6–9] described the synthesis and determined the characteristics of PNIPAAm-SCH₂COOH samples with a terminal carboxyl group, differing in molar mass (MM) and LCST. The possibility of synthesis of block copolymers of PNIPAAm with polyethyleneimine (PEI) ($M_n = 2000$ g/mol) was also demonstrated. Block copolymers were prepared by the reaction of the terminal carboxylic acid groups of PNIPAAm with PEI, using 1-ethyl-3-(3-(dimethylamino)propyl)carbodiimide (EDAC) in an aqueous solution. The

viscosity MM of block copolymers, obtained using the well-known constants $K_{\eta} = 0.0003$ and $\alpha = 0.64$ in the Mark–Kuhn–Houwink equation, ranged from 23,000 to 57,000 g/mol. An increase in the MM of the PNIPAAm block led to a change in the LCST from 43.1 to 34.9 °C at pH = 4.0 and from 42.0 to 33.4 °C at pH = 7.4 [10].

Amphiphilic diblock copolymers of PNIPAAm with polystyrene (PS) and poly-tert-butyl methacrylate (PBMA), differing in the MM of the blocks, were obtained by RAFT polymerization using 4-cyanopentanoic acid dithiobenzoate as a chain transfer agent [11]. In water, copolymers with hydrophobic blocks formed micelles for a long time. It has been established that the reason of the formation of micellar structures is the dehydration of PNIPAAm chains, which is characteristic of all studied copolymers in a wide temperature range. In addition, it was found that the surface of the PS and PBMA of the micelle cores is covered with hydrophobic groups in such a way that they prevent the compression of the hydrophilic shell of the PNIPAAm core. The micelles were colloidally stable and did not precipitate from water at high temperatures.

A thermo- and pH-responsive copolymer of *N*-isopropylacrylamide with maleic acid was studied by the light scattering and turbidimetry [12,13]. Aqueous solutions with pH values from 1.8 to 10.9 and in the concentration range from 0.001 to 0.015 g/cm³ were investigated. At all pH values and concentrations, phase separation was observed at temperatures $T > 33$ °C. The temperatures of the onset of the phase separation and the width of this interval increased with decrease in copolymer concentration and increase in pH. Three types of species were found in solutions, namely, macromolecular unimers, micellar structures, and loose aggregates. At heating, their size and fraction in solutions were changed.

Copolymers of PNIPAAm with fluorine-containing comonomers are of particular interest. The need to create methods for the synthesis of block and graft copolymers based on fluorine-containing compounds is primarily due to the surface properties of these materials, which are characterized by low values of surface energy [14,15]. In addition, the presence of fluorinated groups with a strong cohesive potential in the composition of block copolymers leads to an increased propensity of macromolecules for self-organization in solution [16–20]. Diphility of components in copolymers based on NIPAAm and fluorinated comonomers is the cause of self-organization on the intra- and supramolecular levels, the nature of which is determined by competing interactions of various types. Thus, modification of PNIPAAm by fluorinated comonomers is a promising way of obtaining new copolymers with unique properties.

A fluorine-containing grafted copolymer of polyvinylidene fluoride (PVDF) with PNIPAAm was synthesized [21]. PVDF has many unique properties such as chemical resistance, heat resistance, and good mechanical properties. However, the application of PVDF-based materials is limited due to their hydrophobic nature, while the polymer PNIPAAm, on the contrary, shows hydrophilic properties. PVDF-graft-PNIPAAm copolymers were obtained by the free radical polymerization with transportation of PVDF atom as a macroinitiator and CuCl/tris-(2-(dimethylamino)ethyl) as an amine catalyst. The chemical structure of the grafted copolymers was characterized by IR- and ¹H NMR spectroscopy. Thermosensitive membranes were obtained from PVDF-graft-PNIPAAm by phase inversion. The influence of temperature on the release of pure water and bovine serum albumin during membrane development was also studied. It was shown that grafted PNIPAAm chains, as a rule, enrich the surface of membranes or membrane pores during their formation. The pore diameter and porosity of the copolymer membranes were greater than those of pure PVDF membranes. In addition, PVDF-PNIPAAm membranes may exhibit thermosensitivity in the analysis of aquatic environments and bovine serum albumin.

Linear-dendritic block copolymers based on *N*-isopropylacrylamide and perfluorinated polyphenylgermane with different MM of hydrophilic blocks were obtained by chain transfer reaction on *bis*-(pentafluorophenyl)germane by radical polymerization of *N*-isopropylacrylamide and subsequent activated polycondensation [22]. The structure of the polymers was confirmed by IR- and ¹H NMR spectroscopy. The obtained linear-dendritic

copolymers were studied in solutions at the water–air interface and in Langmuir-Blodgett (LB) films. It was found that the resulting macromolecules are capable of forming stable monolayer films at the water-air interface, regardless of the subphase acidity. For solutions of PNIPAAm containing a *bis*-pentafluorophenylgermanium group at the end of the chain, no LCST behavior was found. In addition, the introduction of pentafluorophenylgermanium groups into PNIPAAm significantly changes its properties at various interface boundaries.

Previously, using one of the water-soluble monomers of *N*-vinylpyrrolidone, we proved the possibility of synthesis of linear amphiphilic diblock copolymers of poly-*N*-vinylpyrrolidone-2,2,3,3-tetrafluoropropyl methacrylate by the reaction of chain transfer to *bis*-pentafluorophenylgermane during the polymerization of *N*-vinylpyrrolidone and subsequent post-polymerization of selected functional polymers in 2,2,3,3-tetrafluoropropyl methacrylate medium [23]. Properties of copolymers with different MM of hydrophilic block (M_w from 22,000 to 57,000 g/mol) have been studied at various interphase boundaries [24]. It was shown that the behavior of macromolecules of amphiphilic block copolymers in solutions and on the boundary of phases is determined by the formation of hydrogen bonds between the molecules of the solvent and the hydrophobic fluorinated fragment. Moreover, fluorine atoms of the perfluorinated phenyl rings participate in the formation of hydrogen bonds at the germanium atom. It was demonstrated that the processing of porous polymer membranes based on different dimethacrylic esters by the solutions of amphiphilic block copolymers of poly-*N*-vinylpyrrolidone-2,2,3,3-tetrafluoropropyl methacrylate in methanol significantly changes their sorption and surface properties [25]. This is what made it possible to test these polymers as modifying components in the production of porous polymer membranes for diesel fuel filtration. It was established that the use of a filter element based on the obtained hydrophobic materials reduces the water content in diesel fuel from 0.54% to 0.27%.

The use of chain transfer reactions during radical polymerization for the synthesis of block copolymers makes it possible to obtain polymers containing blocks of different nature in two stages. The advantage of this method, for example, over RAFT-polymerization, is the ease of synthesis, and the absence of special agents (chain transfer agents).

Interest in amphiphilic copolymers on based PNIPAAm is due not only to their LCST behavior, but also the possibility of the formation of various self-assembly structures in solutions and at interfaces [26,27]. In recent years, the formation of stable Langmuir monolayers of a block copolymer based on PNIPAAm has been studied actively [28,29]. Determining the potential application of new copolymers based on *N*-isopropylacrylamide requires a comprehensive study of properties for the purpose of their directed regulation. This will allow optimizing the choice of polymer materials for solving specific practical problems.

The aim of this work is the study of the behavior of amphiphilic block copolymer of poly-*N*-isopropylacrylamide and poly-2,2,3,3-tetrafluoropropyl methacrylate at the water–air interface (in Langmuir monolayers) and the surface properties of Langmuir-Blodgett films. An important task is the choice of optimal conditions for the formation of monolayers (concentration of the spreading solution and pH of the subphase) and analysis of the stability of multilayer films using surface pressure isotherms obtained under compression-tension conditions.

2. Materials and Methods

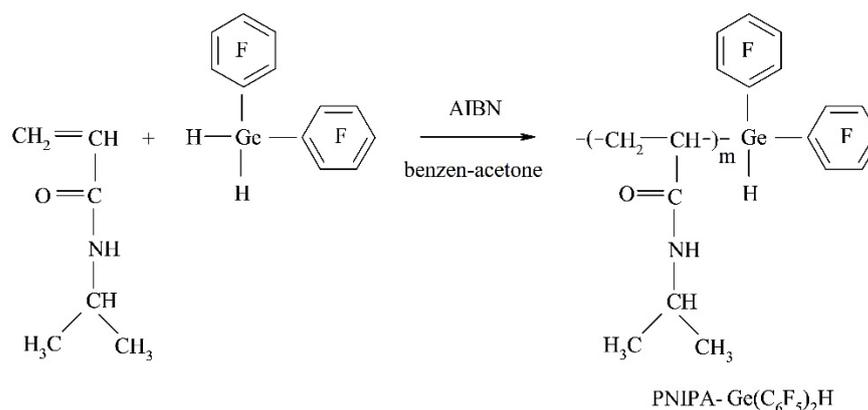
2.1. Materials

The monomer *N*-isopropylacrylamide (Aldrich, 97%, Moscow, Russia) was recrystallized twice from hexane, the crystals were dried in vacuum at room temperature. Then, 2,2,3,3-tetrafluoropropyl methacrylate (FMA) (P&M-Invest, Ltd., 99%, Moscow, Russia) was distilled under reduced pressure ($T = 48$ °C). Methanol-*d* (Aldrich, 99%, Moscow, Russia) was used without pretreatment. The used solvents (chloroform, methanol, benzene, acetone, hexane, ethanol, and methylene iodide) were purified in accordance with

standard procedures [30]. 2,2'-azobisisobutyronitrile recrystallized twice from ethanol, *bis*-(pentafluorophenyl)germane used without pre-cleaning.

2.2. Synthesis of Functional Polymers PNIPAAm-Ge(C₆F₅)₂H

To obtain a functional PNIPAAm polymer containing a *bis*-(pentafluorophenyl)germanium group at the end of the chain, a radical polymerization of the NIPAAm was conducted in the presence of 2,2'-azobisisobutyronitrile (AIBN) initiator (0.009 mol/L) in a mixture of solvents and benzene-acetone (1:1) with *bis*-(pentafluorophenyl)germane (Ge(C₆F₅)₂H₂) ([Ge(C₆F₅)₂H₂] = 0.02 mol/L, *t* = 24 h, *T* = 60 °C, [NIPAAm] = 1.3 mol/L) in sealed dilatometer ampules [31] (Scheme 1).

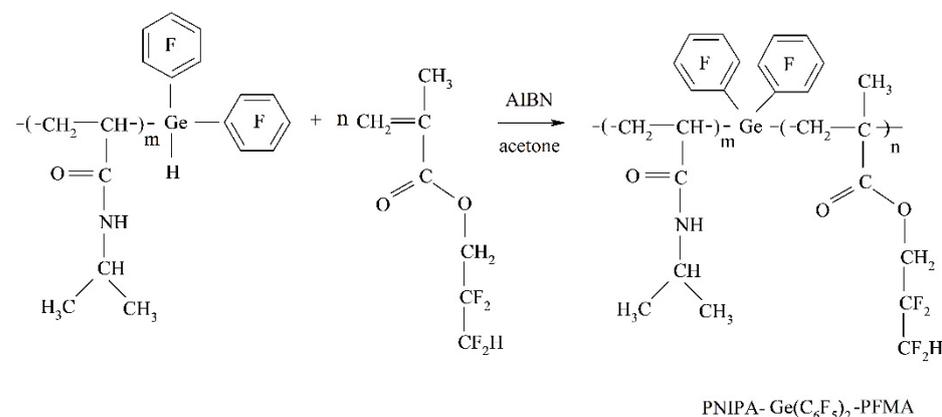


Scheme 1. The reaction of chain transfer to Ge(C₆F₅)₂H₂ during the polymerization of NIPAAm.

The reaction mixtures were degassed in vacuum by triple refreezing, and then soldered and polymerized to 10% conversion. The resulting polymer PNIPAAm-Ge(C₆F₅)₂H was purified from the monomer by tripple reprecipitation using a solvent system: acetone-hexane (1:9 by volume) precipitant, and then dried under reduced pressure and room temperature to constant weight.

2.3. Synthesis of Block Copolymers PNIPAAm-Ge(C₆F₅)₂-PFMA

Postpolymerization of PNIPAAm-Ge(C₆F₅)₂H in the FMA medium was carried out up to 12% conversion at *T* = 60 °C, PNIPAAm + FMA = 4% + 96%, [AIBN] = 0.005 mol/L, acetone:FMA = 2:1, [FMA] = 2.08 mol/L [23] (Scheme 2). The postpolymerization product was purified by tripple reprecipitation using the solvent system: methanol-hexane precipitator and dried under reduced pressure and room temperature until constant weight.



Scheme 2. The postpolymerization of PNIPAAm-Ge(C₆F₅)₂H in the FMA.

During the polymerization reaction of functional PNIPAAm-Ge(C₆F₅)₂H in the FMA, not all macromolecules with the active *bis*-(pentafluorophenyl)germanium group partici-

pate in the chain transfer reaction. Therefore, the obtained product, which contained, in addition to the PNIPAAm–Ge(C₆F₅)₂H, a block copolymer PNIPAAm–Ge(C₆F₅)₂–PFMA, homopolymers of PNIPAAm and PFMA. To separate them, a hot extraction method was used in the Soxhlet apparatus in specially selected solvents. PFMA was separated from the homopolymer in chloroform, and from PNIPAAm–Ge(C₆F₅)₂H in ethanol.

2.4. Polymer Characterization

The IR spectra of the polymers were obtained in KBr tablets on the infrared spectrometer “InfracumFT-801”. Analysis of the structure of the compounds obtained was carried out on ¹H and ¹³C using the Bruker Avance III 400 NMR spectrometer at *T* = 25 °C (methanol-*d* was used as the solvent). The spectra were processed using the MestReNova software.

The content of the hydrophilic block in the block copolymer PNIPAAm–Ge(C₆F₅)₂–PFMA was determined using automatic apparatus Kjeldahl Vilitex AKV-20 (Moscow, Russia) designed to measure nitrogen-containing compounds. To accelerate the decomposition reaction, we used a mixed catalyst (10 g HgSO₄, 7.1 g MgSO₄ and 0.44 g Se). Polymer samples were decomposed at *T* = 200 °C for 1 h, distillation time 9 min.

2.5. Properties of Solution Polymers

The molar masses, hydrodynamic, and conformational characteristics of all samples were determined in solutions in chloroform (density $\rho_0 = 1.486$ g/cm³, dynamic viscosity $\eta_0 = 0.57$ cP, and refractive index $n_0 = 1.443$), methanol ($\rho_0 = 0.79$ g/cm³, $\eta_0 = 0.54$ cP, and $n_0 = 1.326$), chloroform/methanol mixture (9:1) ($\rho_0 = 1.460$ g/cm³, $\eta_0 = 0.56$ cP, and $n_0 = 1.432$) and THF ($\rho_0 = 0.890$ g/cm³, $\eta_0 = 0.46$ cP, and $n_0 = 1.407$) by the methods of light scattering, velocity sedimentation, and GPC triple analysis. All measurements were performed at 21 °C. Weight-average molar masses M_w of polymers investigated were measured by static light scattering using Photocor goniometer (Photocor Instruments Inc., Moscow, Russia). For all samples over the studied concentration range, asymmetry of light scattering intensity was not observed, and the M_w values were found by the Debye method [32] using the formula

$$\frac{cH}{I_{90}} = \frac{1}{M_w} + 2A_2c, \quad (1)$$

where *H* is optical constant:

$$H = \frac{4\pi^2 n_0^2 (dn/dc)^2}{N_A \lambda^4} \quad (2)$$

Here, *c* is the solution concentration, *I*₉₀ is the excessive intensity of scattered light at an angle of 90°, *A*₂ is second virial coefficient, *dn/dc* is the refractive index increment, and *N*_A is Avogadro’s number. The refractive index increment *dn/dc* was measured using a RA-620 refractometer (KEM, Kyoto, Japan). For chloroform solution of PNIPAAm–Ge(C₆F₅)₂H, the second virial coefficient has high positive value; that is, chloroform is a thermodynamically good solvent for this polymer. MM of PFMA was determined in methanol which was a thermodynamically good solvent for this polymer.

The translation diffusion constant *D*₀ of investigated samples was estimated by dynamic light scattering using the same apparatus as for static light scattering investigation. The correlation function of scattered light intensity was derived with the aid of Photocor-FC correlator with 288 channels (Photocor Instruments Inc., Moscow, Russia). The data were treated by the cumulant method and the Tikhonov regularization procedure. One peak on the intensity distribution has been detected for PNIPAAm–Ge(C₆F₅)₂H in chloroform, PFMA in methanol, and PNIPAAm–Ge(C₆F₅)₂–PFMA in THF. Over the studied concentration range, the hydrodynamic radii *R*_{h-D} of samples macromolecules were independent of concentration *c*. The hydrodynamic radius *R*_{h-D} was calculated using Stoker’s equation:

$$R_{h-D} = kT/6\pi \eta_0 D_0 \quad (3)$$

where *k* is the Boltzmann constant and *T* is the absolute temperature.

A Shimadzu Prominence series GPC system equipped with a refractive index (RI) detector, and a Styragel HR 4E (Waters Associates, Milford, MA, USA) column (7.8 × 300 mm packed with 5 μm particles) was used. The column was calibrated with narrow molecular weight polystyrene standards (purchased from Waters Associates). THF stabilized with 2,6-*tert*-butyl-4-methylphenol (BHT) was used as the mobile phase, at a flow rate of 0.5 mL/min at 40 °C. Three methods for estimation of MM were applied. Firstly, MM were obtained by the refractometric detection (polystyrene standards). Secondly, a combination of refractometric and viscometric detectors was used (universal calibration). Thirdly, a combination of a refractometric and viscometric detector with a light scattering detector (the so-called “triple” detection) was used also.

In a manner similar to [14] velocity sedimentation was studied in chloroform on a MOM-3180 analytical ultracentrifuge (Budapest, Hungary). The rotor rotation speed was 45,000 rpm. The sedimentation boundary was formed artificially by stratifying the solvent on the solution and was recorded with a Philpot–Svensson refractometric optical system. Sedimentation diagrams obtained for all solutions had a unimodal pattern. The sedimentation coefficient s was calculated from the rate of movement of the sedimentation boundary. The concentration dependences of s are satisfactorily described by the Gralen relationship $s^{-1} = s_0^{-1}(1 + k_s c)$, where k_s is the concentration sedimentation coefficient. Extrapolation of s^{-1} to zero concentration made it possible to determine the values of the sedimentation constant $s_0 = 6.1$ Sv and the parameter $k_s = 42$ cm³/g.

The density of solvent and solution were measured using a densimeter DA-640 (KEM). The partial specific volume \bar{v} was determined from the slope of the concentration dependences of the difference between the densities of the solution and the solvent. The sedimentation–diffusion molar mass M_{sD} was calculated using the Svedberg equation:

$$M_{sD} = s_0 RT / (1 - \bar{v} \rho_0) D_0, \quad (4)$$

where R is universal gas constant, s_0 is the sedimentation coefficient at $c = 0$, ρ_0 is density of solvent, D_0 is diffusion constant, T is the absolute temperature, and \bar{v} is partial specific volume.

2.6. Properties of Monolayers Polymers

Surface pressure isotherms $\pi = f(A)$ were obtained in air using a KSVMMini (Finland) apparatus by the Wilhelmy plate method. Solutions of polymers (1 g/L) in a mixture of solvents chloroform/methanol mixture (9:1) were applied to the surface of the subphase with a 50 μL micro syringe in 2 μL portions in a checker board pattern, to evenly distribute the substance over the entire bath surface. The monolayer was held for 30 min to evaporate the solvent from the surface of the subphase, and then compressed at a rate of 10 mm/min. The rate of compression of the monolayers in all experiments was the same and amounted to 10 mm/min. All studies were performed at constant temperature of subphase (21 ± 1) °C. Deionized water with a resistivity of 15 MΩ cm (Spectrum Osmosis water treatment system, Dzerzhinsk, Russia) and 0.1 N HCl solutions were used as a subphase. The area attributable to 1 mg of copolymers in the monolayer (A_0) was determined graphically by extrapolating the descending portion of the isotherm $\pi = f(A)$ on the horizontal axis to $\pi = 0$. For the reliability of the result, the surface pressure isotherms for each of the polymers were removed 6 times, and reproducibility was 100%.

The condition in the hysteresis experiment (compression–expansion cycles) was similar to that in the isotherm experiment: after reaching the maximum value surface pressure, the barriers were stopped and monolayer has been expanded with the same speed (the compression rate of the monolayer was 10 mm/min, and the expansion rate was 3 mm/min).

2.7. Properties of Langmuir–Blodgett Films Polymers

The monolayer films were transferred using silicon wafers (Telecom-STV JSC, polished, with a specific volume resistance of more than 1 Ω cm, crystallographic orientation 100, Moscow, Russia) as substrates. Before use, the wafers were incubated for 20 min in a

chromium mixture, washed further with a flowing, then deionized water and dried at $T = 80\text{ }^{\circ}\text{C}$ for 1 h. The transfer of monomolecular films from the water–air interface was carried out using the LB technology by KSV minitrough. The transfer conditions were determined from the surface pressure isotherms of the corresponding polymers. For the functional polymer, the films were transferred at a constant surface pressure of 12 mN/m (10 μL), and for the block copolymer at 25 mN/m (30 μL) ($T = 21\text{ }^{\circ}\text{C}$ and $\text{pH} = 7$). During the deposition, the substrate was dipped into the water (before the spread of the solution) and withdrawn vertically through the monolayer at the speed of 5 mm/min. The transfer coefficient was 0.92–0.98.

Wetting angles θ of LB films were determined under the conditions of leakage by the “sitting drop” method using a setup consisting of a microscope with a light source, a lifting table for a plate and a computer with the “CoolingTech” software. The kinetics of wetting was studied for all films of polymers.

For this purpose, 2 μL of test liquid was applied to the films with a micro syringe, after which, at certain time intervals, we determined the chord of profile l and drop height h . The measurements were performed until equilibrium was reached (~ 30 min). The contact angle of wetting θ was calculated in accordance with equation:

$$\theta = 2 \arctg \left(\frac{2h}{l} \right) \quad (5)$$

Gibbs surface energy of LB films γ_s , its polar γ_s^p and dispersion components γ_s^d were determined using the Rukenstein’s approach [33–35] (wetting liquids—water and diiodomethane).

The resulting LB films were analyzed by atomic force microscopy (AFM) using Solver P47 microscopes (NT-MDT, Moscow, Russia). The scanning was performed in the tapping mode.

3. Results and Discussion

3.1. Synthesis and Characterization of Polymers

The relative constant of chain transfer on *bis*-(pentafluorophenyl)germane during radical polymerization of NIPAAm equal to 2.2 determined previously using Mayo method indicates that if it is possible to use this polymer as a basis for production of block copolymers [22]. To obtain an amphiphilic linear block copolymer, a chain transfer reaction was used at the initial stage to obtain a functional polymer PNIPAAm–Ge(C₆F₅)₂H (Scheme 1). As a result of its further postpolymerization with 2,2,3,3-tetrafluoropropyl methacrylate (Scheme 2), the PNIPAAm–Ge(C₆F₅)₂-PFMA block copolymer was synthesized with a yield of 68.2% and a hydrophilic block fraction of 14.1%. Characteristics of the block copolymer are presented in Table 1.

Table 1. Molar masses and hydrodynamic characteristics for functional polymer PNIPAAm–Ge(C₆F₅)₂H, homopolymer PFMA and block copolymer PNIPAAm–Ge(C₆F₅)₂-PFMA.

Solvent	MM $\times 10^{-3}$, g/mol	R_h , nm	dn/dc , cm ³ /g
	PNIPAAm–Ge(C ₆ F ₅) ₂ H		
Chloroform	52 ^a / 44 ^b	4.2	0.060
Chloroform/Methanol (9/1)	43	6.2	0.075
	PFMA		
Methanol	200 ^a	14	0.080
	PNIPAAm–Ge(C ₆ F ₅) ₂ -PFMA		
Chloroform/Methanol (9/1)	-	40	0.040
THF	490 ^c	10/100	0.010

^a MM was determined by light scattering. ^b M was determined by velocity sedimentation. ^c MM was determined by GPC ($PDI = 2.1$).

Figure 1 shows the IR spectrum of the homopolymer PNIPAAm, the functional polymer PNIPAAm-Ge(C₆F₅)₂H and the block copolymer PNIPAAm-Ge(C₆F₅)₂-PPMA. It was shown that the IR spectrum of the functional polymer in comparison with the homopolymer has an absorption band corresponding to the group -CF (1080.5 cm⁻¹) and -C₆F₅ (960 cm⁻¹), which indicates the presence of -GeH(C₆F₅)₂ groups in the macromolecules of the functional polymer. The presence in the IR spectrum of the block copolymer of the absorption bands of -C(O) (1743 cm⁻¹) and -CF₂- (833 cm⁻¹) groups, characteristic of PFMA units, indicates the formation of a block copolymer.

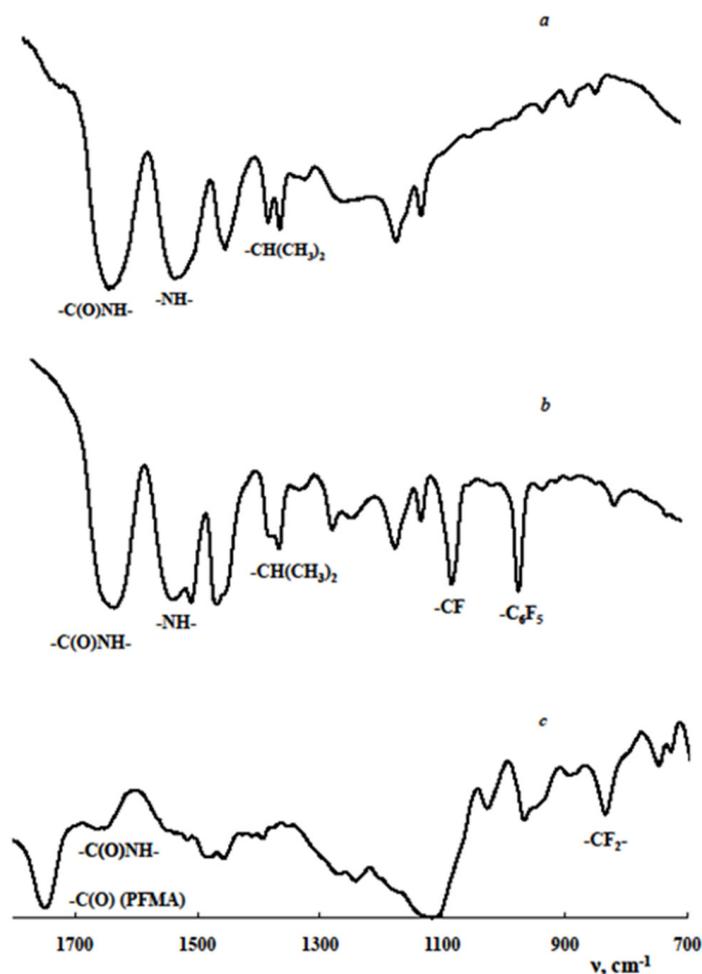


Figure 1. Fragments IR spectrum of (a) PNIPAAm, (b) the functional polymer PNIPAAm-Ge(C₆F₅)₂H and (c) the linear block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA.

Figure 2 shows ¹H NMR spectrum of a block copolymer. The spectrum of the block copolymer, compared with the functional polymer, indicates the transfer of chemical shifts of the protons of the -CH and -CH₃ groups to a higher region (the signal 3.96 ppm has shifted to 4.42 and the signal 0.87 ppm shifted to 0.97, respectively). The chemical shifts of the protons of the -O-CH₂-CF₂- (4.59 ppm) and -CF₂H (6.20 ppm) groups also appear in the spectrum (Table 2). The chemical shifts 3.31 and 4.87 (impurity signal) correspond to solvent CD₃OD. Table 2 also presents the data of ¹³C NMR analysis, according to which the chemical shifts of the carbon atoms of the -CH and -C(O) groups moved to the higher region (44.82 and 175.72 ppm, respectively), and the chemical shift of the carbon atom of the -CH₃ group to the lower region (18.11 ppm).

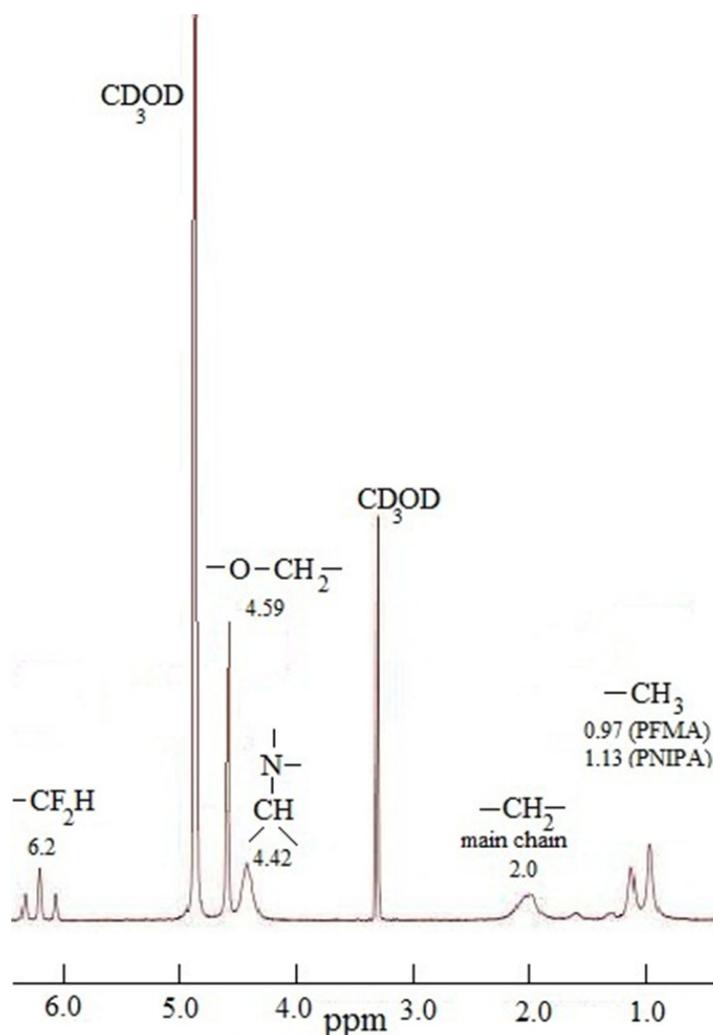


Figure 2. ^1H NMR spectrum of block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA in CD₃OD.

Table 2. Assignment of ^1H and ^{13}C NMR signals of functional polymer PNIPAAm-Ge(C₆F₅)₂H and block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA.

Polymer	Chemical Shifts of Protons (ppm)				
	-N-CH<	-CH ₂ - Main Chain	-CH ₃	-O-CH ₂ -CF ₂ -	-CF ₂ H
PNIPAAm-Ge(C ₆ F ₅) ₂ H	3.96	1.15–1.7	0.87, 1.15	–	–
PNIPAAm-Ge(C ₆ F ₅) ₂ -PFMA	4.42	1.15–1.7	0.97, 1.13	4.59	6.20
Polymer	Chemical shifts of carbon atom (ppm)				
	-C(O)-	-NH-CH<	-CH<	-CH ₃	
PNIPAAm-Ge(C ₆ F ₅) ₂ H	174.75	47.60	41.11	21.29	
PNIPAAm-Ge(C ₆ F ₅) ₂ -PFMA	175.72	47.60	44.82	18.11	

The resulting block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA is insoluble in chloroform; therefore, to study the Langmuir monolayers and to obtain LB films, a mixture of solvents chloroform/methanol was used. However, the presence of methanol in the solvent mixture can have a significant effect on the self-organization of macromolecules at the interphase boundaries, because applying the solution spread on the water–air phase boundary causes chloroform to evaporate, while methanol can either go into the subphase

or retain in the loops of macromolecules, as we have previously shown in the case of a block copolymer poly-*N*-vinylpyrrolidone-2,2,3,3-tetrafluoropropyl methacrylate [24].

Molar masses of PNIPAAm-Ge(C₆F₅)₂H determined by static light scattering and velocity sedimentation methods in chloroform and mixed solvent were close (Table 1). Taking into account the high experimental error in determining small values of hydrodynamic radii (up to 20%), it can be assumed that the difference in the R_h values determined in chloroform and in a mixed solvent is insignificant.

The PFMA sample has a higher molecular weight. Accordingly, a higher value of the hydrodynamic radius was obtained for PFMA. The relatively low value of the refractive index increment for solution PNIPAAm-Ge(C₆F₅)₂-PFMA in chloroform/methanol mixture did not allow reliable determination of MM of this sample by light scattering. Moreover, the compositional heterogeneity of the PNIPAAm-Ge(C₆F₅)₂-PFMA sample should be taken into account. This usually results in an incorrect value of MM at a low value of dn/dc [36–38]. Therefore, MM of block-copolymers was determined by chromatographic method in THF solution. The obtained value of MM is relatively high. It can be assumed that supramolecular structure is formed in THF. This assumption is confirmed by dynamic light scattering data: two modes were detected in solution of PNIPAAm-Ge(C₆F₅)₂-PFMA in THF. Radii of particles responsible for these modes differ ten times. The fast mode reflects the diffusion of macromolecules, while the slow mode reflects aggregate diffusion. As for solutions in a mixed solvent, a high value of R_h was obtained for them, which may indicate the formation of supramolecular structures.

3.2. Langmuir Monolayers of Block Copolymers

The concentrations of spreading solutions were selected and the effect of subphase acidity on the formation of monolayers of macromolecules of the PNIPAAm-Ge(C₆F₅)₂-PFMA block copolymer was studied.

The obtained data show that regardless of the acidity of the subphase, at the studied volumes of the spreading solutions (30–60 μ L), high pressure fracture films are characteristic of monolayers collapse pressures (π_{max}) = 48–61 mN/m. At small volumes of spreading solutions $V_{s.sol.} = 20 \mu$ L, the breakdown pressure of a monolayer film is much higher at the acid subphase ($\pi_{max} = 33$ mN/m) compared to water ($\pi_{max} = 13$ mN/m).

As an example, Figure 3 shows the block copolymer surface pressure isotherms obtained at $V_{s.sol.} = 30 \mu$ L (pH = 7.0 and pH = 1.3). It can be seen that the monolayers undergo several phase transitions, like a three-dimensional gas through “gaseous” (I), “liquid” (II) and “solid” (III) states. When pH = 7.0, these monolayers have anomalous sizes of macromolecules: $A_0(\text{pH} = 7.0) = 0.6 \text{ m}^2/\text{mg}$, $A_0(\text{pH} = 1.3) = 0.4 \text{ m}^2/\text{mg}$ (Table 3). This could be due to the fact that the acidic subphase triggers the intermolecular interactions between macromolecule-H₂O.

Table 3. Characteristics of Langmuir monolayers of block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA at pH = 7.0 and pH = 1.3 obtained from isotherms of surface pressure $\pi = f(A)$ spread with chloroform/methanol.

$V_{s.sol.}^a$ (μ L)	A_0^b (m^2/mg)	π_{max}^c (mN/m) pH = 7.0	$\beta = -\frac{d\pi}{dA} \cdot 10^{-14}^d$ (N/m^3)
20	0.4	13	5.7
30	0.6	47	3.0
50	0.3	52	5.6
60	0.3	47	2.3
		pH = 1.3	
20	0.4	33	7.8
30	0.4	61	5.1
50	0.4	48	4.4
60	0.3	53	5.6

^a The volume of spreading solution. ^b The area attributable to 1 mg of polymers in the monolayer. ^c The monolayer collapse. ^d The stiffness coefficient of the monolayer.

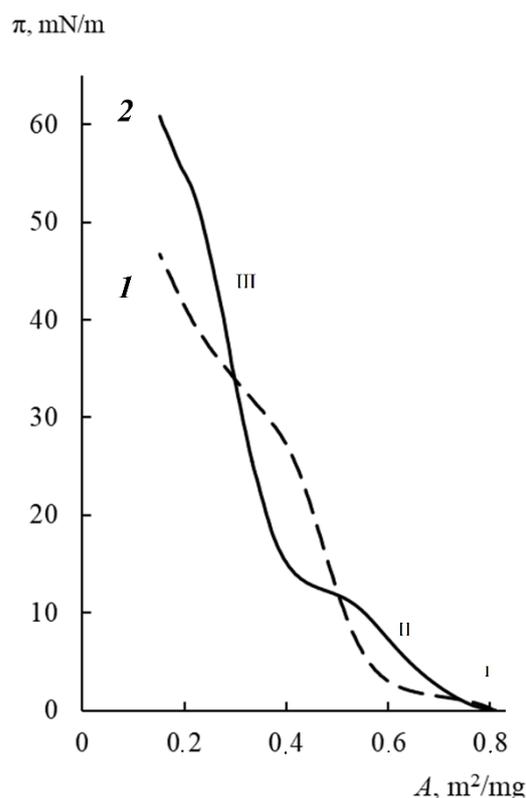


Figure 3. Isotherms of surface pressure π of block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA on the subphase at pH = 7.0 (1) and pH = 1.3 (2) spread with chloroform/methanol (volume of spreading solution 30 μ L, $T = 21$ °C, A —molecular area).

At large volumes of the spreading solution (for example, $V_{s,sol.} = 50$ μ L) π_{max} (pH = 1.3) $<$ π_{max} (pH = 7.0): with pH = 7.0 $A_0 = 0.3$ m²/mg, and at pH = 1.3 $A_0 = 0.4$ m²/mg. At the subphase with a pH = 7.0, the breakdown pressure of the monolayer reaches its maximum ($\pi_{max} = 52$ mN/m) compared to other $V_{s,sol.}$; in the case of pH = 1.3 $\pi_{max} = 48$ mN/m. Apparently, at such a polymer concentration at the water–air interface, intermolecular interactions begin to predominate over intramolecular interactions, which increases the binding forces between macromolecules in the monolayer and leads to high monolayer collapse pressures.

Apparently, this is due to the ionization of the PNIPAAm blocks, which leads to the unfolding of the hydrophilic block of the macromolecule. At the acidic subphase, intermolecular interactions between macromolecules of the block copolymer predominate over intramolecular interactions, which are confirmed by an increase of the stiffness coefficient of the monolayer β . For example, the value of β for $V_{s,sol.} = 20$ μ L at pH = 1.3 is equal to 7.8×10^{14} N/m³, and in the case of pH = 7.0 it is 5.7×10^{14} N/m³ (Table 3).

To determine the stability of monolayers, surface pressure isotherms were obtained for the block copolymer in the compression–expansion condition (30 μ L). Figure 4 shows that at pH = 7.0, the “pseudoplateaus” region corresponding to the formation of a dense monolayer coincides practically in the compression and expansion cycles. However, this is not observed at pH = 1.3: the $\pi = f(A)$ curve shifts down along the ordinate axis upon stretching. This behavior confirms the presence of significant intermolecular interactions in the case of the acidic subphase. It allowed us to transfer films using the LB technology ($n = 1$) from the subphase at pH = 7.0. The expansion rate of the monolayer was lower than the compression rate.

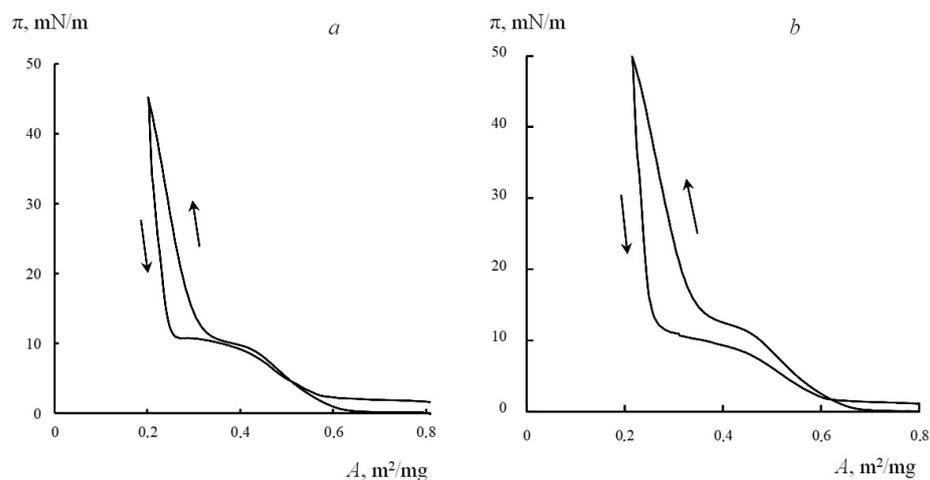


Figure 4. Compression–expansion curves of block copolymer PNIPAAm–Ge(C₆F₅)₂–PFMA monolayers spread with chloroform/methanol at various pH = 7.0 (a) and 1.3 (b) (volume of spreading solution 30 μ L, $T = 21$ °C).

To adequately analyze the behavior of macromolecules of the amphiphilic block copolymer, the PFMA homopolymer and the functional polymer PNIPAAm–Ge(C₆F₅)₂H (at pH = 1.3 and 7.0) in a chloroform/methanol mixture were studied (Figure 5). It has been established that when chloroform is used as a solvent, PFMA is characterized by low surface pressure values $\pi_{\max} = 14$ mN/m and $A_0 = 0.2$ m²/mg ($V_{s.sol.} = 30$ μ L). In addition, in this case, the surface pressure isotherms obtained under compression conditions manifest the degeneration of the “pseudoplateaus” region, which is the region of formation of a dense monomolecular layer. When chloroform/methanol is used at pH = 7.0 ($V_{s.sol.} = (20–60)$ μ L), all monolayer films of PFMA are characterized by high values of $\pi_{\max} = (45–60)$ mN/m and the presence of three phase states of the monolayer (“gaseous” (I), “liquid” (II), “solid” (III) Figure 5). The latter circumstance also indicates the effect of methanol on the behavior of macromolecules at the water–air interface [24].

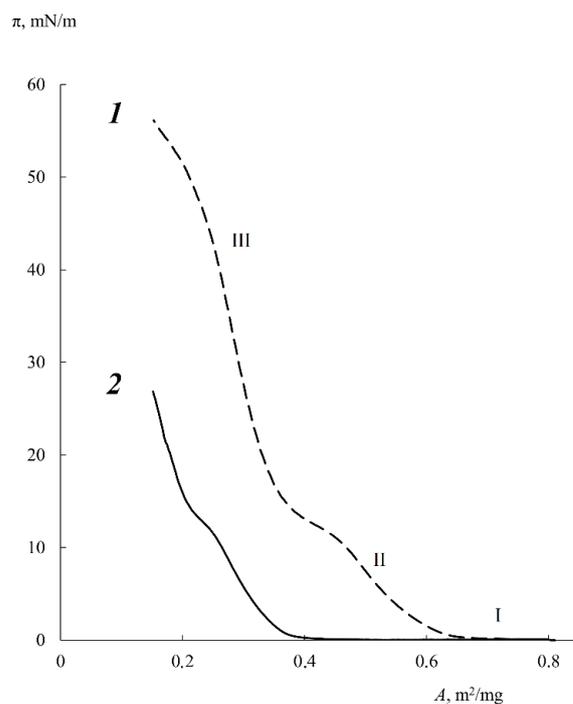


Figure 5. Isotherms of surface pressure π of homopolymer PFMA at pH = 7.0 (1) and 1.3 (2) spread with chloroform/methanol (volume of spreading solution 30 μ L, $T = 21$ °C).

The surface pressure isotherms of the functional polymer PNIPAAm-Ge(C₆F₅)₂H using a mixture of solvents chloroform/methanol on subphases of different acidity and at different volumes of spreading solutions were obtained also. The curves were compared with similar curves obtained when chloroform was used as a solvent [22]. It was found that in the case of the functional polymer PNIPAAm-Ge(C₆F₅)₂H the presence of methanol in the mixture of the spreading solution does not have a significant effect on the behavior of macromolecules at the water–air interface, that is, only PFMA units participate in the formation of hydrogen bonds.

3.3. Surface Properties of LB Films

To determine the Gibbs surface energy of LB films of the block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA and functional polymer PNIPAAm-Ge(C₆F₅)₂H, the Rukenstein's approach was used. The surface Gibbs of films (γ_s) and its polar (γ_s^p) and dispersion (γ_s^d) components were assessed by the Good–Kaelble–Dan–Fowkes equation [33], and water and methylene iodide used as test liquids.

Calculations were performed according to equations

$$\begin{cases} \gamma_{\text{CH}_2\text{I}_2} \cdot (1 + \cos \theta_{\text{CH}_2\text{I}_2}) = 2 \left(\gamma_{\text{CH}_2\text{I}_2}^d \cdot \gamma_s^d \right)^{1/2} + 2 \left(\gamma_{\text{CH}_2\text{I}_2}^p \cdot \gamma_s^p \right)^{1/2} \\ \gamma_{\text{H}_2\text{O}} \cdot (1 + \cos \theta_{\text{H}_2\text{O}}) = 2 \left(\gamma_{\text{H}_2\text{O}}^d \cdot \gamma_s^d \right)^{1/2} + 2 \left(\gamma_{\text{H}_2\text{O}}^p \cdot \gamma_s^p \right)^{1/2}, \end{cases} \quad (6)$$

where $\theta_{\text{CH}_2\text{I}_2}$ and $\theta_{\text{H}_2\text{O}}$ are the advancing contact angles of methylene iodide and water on the surface of the films. Table 4 shows the results of the wetting of the LB films of the functional polymer and the block copolymer transferred at pH = 7.0.

Table 4. The equilibrium contact angle (θ) and energy characteristics of LB films of functional polymer PNIPAAm-Ge(C₆F₅)₂H and block copolymer PNIPAAm-Ge(C₆F₅)₂-PFMA.

Polymer	$\theta (\pm 1), ^\circ$		$\Delta\theta, ^\circ$		Gibbs Surface Energy (mJ/m ²)		
	H ₂ O	CH ₂ I ₂	H ₂ O	CH ₂ I ₂	$\gamma_s^p (\pm 1.5)^a$	$\gamma_s^d (\pm 1.5)^b$	γ_s^c
PNIPAAm-Ge(C ₆ F ₅) ₂ H	34	48	6	1	34.3	28.2	62.5
PNIPAAm-Ge(C ₆ F ₅) ₂ -PFMA	64	61	20	1	17.1	23.9	41.0

^a The polar component of Gibbs surface energy of the LB film. ^b The dispersion component of Gibbs surface energy of the LB film. ^c The total Gibbs surface energy of the LB film.

Table 4 demonstrates that the values of the total Gibbs surface energy of the LB films for the functional and block copolymer are different. In the case of LB film of PNIPAAm-Ge(C₆F₅)₂-PFMA total Gibbs surface energy of the film $\gamma_s = 41.0$ mJ/m², which characterizes it as more hydrophobic than the functional polymer film (62.5 mJ/m²). Value γ_s of this LB film is close to the total Gibbs surface energy of the LB film of pure PFMA ($\gamma_s = 33.8$ mJ/m²) [24], that is, in the surface layer of the film predominantly the PFMA blocks remain oriented. For LB film of PNIPAAm-Ge(C₆F₅)₂-PFMA dispersion component (23.9 mJ/m²) prevails over the polar component (17.1 mJ/m²), and in the case of a functional polymer, on the contrary, the polar component (34.3 mJ/m²) is higher than dispersion component (28.2 mJ/m²). Therefore, in the case of a block copolymer, dispersion interactions between groups appear in the surface layer of the film -CH(CH₃)₂ and -CF₂CF₂H, which reduces the wettability of these films with a polar liquid (H₂O).

The excess of the polar component over the dispersion component in the case of the LB film of functional polymer is most likely due to the predominance of interactions between the water molecules and the polar groups of the -C(O)NH chain over other types of interactions. Practically the same wetting angles of the block copolymer film with water and diiodomethane ($\theta_{\text{H}_2\text{O}} = 64^\circ$, $\theta_{\text{CH}_2\text{I}_2} = 61^\circ$) indicate that the nonpolar contact fluid interacts with the surface in almost the same way as the polar one, i.e., in the surface layer

of the film, both the hydrophilic units of NIPAAm ($-C(O)NH$) and the hydrophobic units of PFMA ($-CF_2CF_2H$) are oriented.

The study of the kinetics of wetting showed that for the LB film of PNIPAAm- $Ge(C_6F_5)_2H$ when droplets of test liquids are applied, the change in the wetting angle is $\Delta\theta_{H_2O} = 6^\circ$, and $\Delta\theta_{CH_2I_2} = 1^\circ$ (30 min). For the LB film of the block copolymer, the change in wetting angles is $\Delta\theta_{H_2O} = 20^\circ$ and $\Delta\theta_{CH_2I_2} = 1^\circ$.

Thus, a comprehensive study of the properties of the obtained polymers based on *N*-isopropylacrylamide at various interphase boundaries showed that the modification of this polymer with a fluorinated linear block is much more efficient than with branched perfluorinated polyphenylenegermane [22], which illustrates the decrease in the surface energy of the LB film (to 41.0 mJ/m^2) in the case of an amphiphilic block copolymer. Low surface energy of the LB film and the hydrophobicity of the block copolymer PNIPAAm- $Ge(C_6F_5)_2$ -PFMA can be used, for example, to build medical devices with a thromboresistant surface (biosensors, artificial endoprostheses, heart valves and other implants for which the components forming thrombi are unacceptable) [39].

It can be assumed that in the LB film of an amphiphilic block copolymer both links of different nature have contact with the silicon wafer and, based on the above, their orientation in the monolayer can be represented as follows [31].

The AFM images of LB films of functional polymer and block copolymer spread with chloroform/methanol (as an example, transferred at $\text{pH} = 7$) are shown in Figure 6. From Figure 6a, it can be seen that for LB film of functional polymer PNIPAAm- $Ge(C_6F_5)_2H$ exhibit isolated elongated micelles with high densities in the form of “octopus” on the periphery of which there are terminal hydrophobic groups $-Ge(C_6F_5)_2H$, and the links of the hydrophilic polymer PNIPA are turned inward. When transferring the film LB to a silicon wafer, the entire surface is not covered with a monolayer. For LB film of block copolymer PNIPAAm- $Ge(C_6F_5)_2$ -PFMA (Figure 6b), a comb-like structure is observed with characteristic protrusions that are formed by hydrophobic groups (PFMA and $-Ge(C_6F_5)_2H$, for example, see Scheme 3).

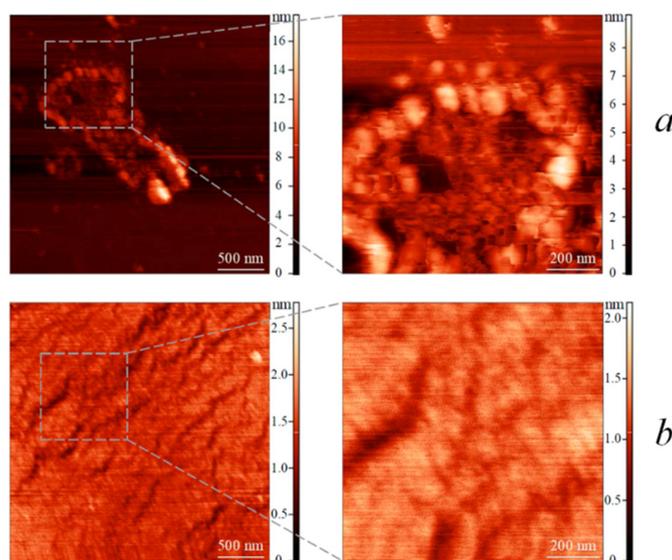


Figure 6. AFM images of the LB films of functional polymer (a) and block copolymer (b) at transfer pressure of 12 mN/m ($10 \mu\text{L}$) and 25 mN/m ($30 \mu\text{L}$), respectively ($T = 21 \text{ }^\circ\text{C}$, $\text{pH} = 7.0$).

Similar processes of surface micelle formation were discovered by the authors of the work [40] for amphiphilic block copolymers poly(*N*isopropylacrylamide)-*block*-poly[oligo (ethylene glycol) acrylate]. The structure that was first established by the authors for the studied systems differs from the typical core-corona structure consisting of a hydrophobic block core and a hydrophilic block crown in amphiphilic block copolymer systems.

Similar processes of self-organization in Langmuir monolayers and Langmuir-Blodgett films were found for almost all previously obtained amphiphilic block copolymers of various structures [41–47].



Scheme 3. Orientation macromolecules of block copolymer in monolayer and LB film (at pH = 7.0).

4. Conclusions

For the first time, a new amphiphilic block copolymer based on *N*-isopropylacrylamide and 2,2,3,3-tetrafluoropropyl methacrylate was obtained by a double successive chain transfer reaction to *bis*-(pentafluorophenyl)germane groups. The structure of the polymer was confirmed by IR and NMR spectroscopy. The molar masses, hydrodynamic, and conformational characteristics of all polymers were determined in solutions by the methods of light scattering, velocity sedimentation, and GPC triple analysis.

The effect of the acidity of the subphase on the behavior of macromolecules in the Langmuir monolayers of the amphiphilic block copolymer was studied. At pH = 1.3, the formation of monolayers with high surface pressures ($\pi_{\max} = 33\text{--}61$ mN/m) and film stiffness coefficient ($\beta = (4.4\text{--}7.8) \times 10^{14}$ N/m³) was found, which indicates ionization hydrophilic unit of the block copolymer macromolecule on the acidic subphase. The influence of methanol on the self-organization of fluorinated units at the water–air interface, which is capable of forming hydrogen bonds with macromolecules, has been revealed. Results provide insight into the two-dimensional self-assembly of amphiphilic block copolymers consisting of molecular-level amphiphilic block components.

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References

- Roy, B.; Brooks, W.L.A.; Sumerlin, B.S. New directions in thermoresponsive polymers. *Chem. Soc. Rev.* **2013**, *42*, 7214–7243. [[CrossRef](#)]
- Dong, Q.; Luo, C.; Li, N.; Chi, J.; Zhang, Q. Temperature and recognition dual responsive poly(*N*-isopropylacrylamide) and poly(*N,N*-dimethylacrylamide) with adamantly side group. *Materials* **2018**, *11*, 473. [[CrossRef](#)] [[PubMed](#)]

3. Lanza-laco, S.; Armelin, E. Poly(*N*-isopropylacrylamide) and copolymers: A review on recent progresses in biomedical applications. *Gels* **2017**, *3*, 36. [[CrossRef](#)] [[PubMed](#)]
4. Tu, Y.L.; Wang, C.C.; Chen, C.Y. Preparation of shell crosslinked nanoencapsulate for drug carriers by using poly(*N*-isopropylacrylamide)-co-poly(L-lysine) grafted copolymer. *J. Polym. Res.* **2018**, *25*, 134. [[CrossRef](#)]
5. Pawar, K.; Kutcherlapati, S.N.R.; Yeole, N.; Jana, T.; Hundi-wale, D. Synthesis of poly(*N*-isopropylacrylamide-*b*-*N*-vinylcarbazole) copolymers via RAFT polymerization and its stimuli responsive morphology in aqueous media. *J. Polym. Res.* **2018**, *25*, 91. [[CrossRef](#)]
6. Zareie, M.H.; Dincer, S.; Piskin, E. Observation of phase transition of thermoresponsive poly(NIPA)-PEI block copolymers. *J. Coll. Inter. Sci.* **2002**, *251*, 424–428. [[CrossRef](#)]
7. Dincer, S.; Tuncce, A.; Piskin, E. A potential gene delivery vector: *N*-isopropylacrylamide-ethyleneimine block copolymers. *Macromol. Chem. Phys.* **2002**, *203*, 1460–1465. [[CrossRef](#)]
8. Piskin, E. Molecularly designed water soluble, intelligent, nanosize polymeric carriers. *Int. J. Pharm.* **2004**, *227*, 105–118. [[CrossRef](#)]
9. Piskin, E.; Dincer, S.; Turk, M. Gene delivery: Intelligent but just at the beginning. *J. Biomater. Sci. Polym. Ed.* **2004**, *15*, 1181–1202. [[CrossRef](#)] [[PubMed](#)]
10. Arotcarena, D.M.; Heise, B.; Ishaya, S.; Laschewsky, A. Switching the inside and the outside of aggregates of water-soluble block copolymers with double thermoresponsivity. *J. Am. Chem. Soc.* **2002**, *124*, 3787–3793. [[CrossRef](#)] [[PubMed](#)]
11. Nuopponen, M.; Ojala, J.; Tenhu, H. Aggregation behavior of well defined amphiphilic diblock copolymers with poly(*N*-isopropylacrylamide) and hydrophobic blocks. *Polymer* **2004**, *45*, 3643–3650. [[CrossRef](#)]
12. Tarabukina, E.B.; Simonova, M.A.; Bucatariu, S.; Fundueanu, G.; Filippov, A.P. Behavior of thermo- and pH-responsive copolymer of *N*-isopropylacrylamide and maleic acid in aqueous solutions. *Int. J. Polym. Anal. Charact.* **2016**, *21*, 11–17. [[CrossRef](#)]
13. Simonova, M.A.; Tarabukina, E.B.; Filippov, A.P.; Constantin, M.; Popescu, I. Effect of Concentration on the properties of heat- and PH-sensitive copolymers of poly(*N*-isopropylacrylamide) with maleic acid in aqueous solutions. *Fiber. Chem.* **2015**, *47*, 152–155. [[CrossRef](#)]
14. Filippov, A.P.; Zamyshlyayeva, O.G.; Tarabukina, E.B.; Simonova, M.A.; Kozlov, A.V.; Semchikov, Y.D. Structural and conformational properties of hyperbranched copolymers based on perfluorinated germanium hydrides. *Polym. Sci. Ser. A.* **2012**, *54*, 319–329. [[CrossRef](#)]
15. Pitois, C.; Vestberg, R.; Rodlert, M. Fluorinated dendritic polymers and dendrimers for waveguide applications. *Opt. Mater.* **2002**, *21*, 499–506. [[CrossRef](#)]
16. Gasilova, E.R.; Koblyakova, M.A.; Filippov, A.P.; Zakharova, O.G.; Zaitsev, S.D.; Yu, D. Semchikov. Hydrodynamics and light scattering in solutions of a hyperbranched perfluorinated polyphenylenegermane–poly(methyl methacrylate) block copolymer. *Polym. Sci. Ser. A.* **2006**, *48*, 989. [[CrossRef](#)]
17. Zakharova, O.G.; Tarasova, E.V.; Simonova, M.A.; Semchikov, Y.D.; Filippov, A.P. Synthesis and structural and conformational properties of hybrid polymers of styrene with perfluorinated compounds of germanium. *Polym. Sci. Ser. A.* **2009**, *51*, 512–517. [[CrossRef](#)]
18. Tarabukina, E.; Kozlov, A.; Simonova, M.; Zamyshlyayeva, O.; Semchikov, Y. Hydrodynamic and molecular properties of hyperbranched copolymers formed by pentafluorophenylgermane hydrides. *Int. J. Polym. Anal. Charact.* **2011**, *16*, 369–376. [[CrossRef](#)]
19. Simonova, M.A.; Zamyshlyayeva, O.G.; Simonova, A.A.; Tarasova, E.V.; Filippov, A.P. Model and hybrid polystyrenes containing trispentafluorophenylgermanium and groups. *Int. J. Polym. Anal. Charact.* **2009**, *14*, 454–467.
20. Simonova, M.A.; Zamyshlyayeva, O.G.; Simonova, A.A.; Filippov, A.P. Conformation of the linear-dendritic block copolymers of hyperbranched polyphenylenegermane and linear poly(methylmethacrylate). *Int. J. Polym. Anal. Charact.* **2015**, *20*, 223–230. [[CrossRef](#)]
21. Xian-Kai, L.; Xia, F.; Li, C.; Yi-Ping, Z. Characterization of temperature-sensitive membranes prepared from poly(vinylidene-fluoride)-graft-poly(*N*-isopropylacrylamide) copolymers obtained by atom transfer radical polymerization. *Front Mater Sci. China* **2010**, *4*, 345–352.
22. Zamyshlyayeva, O.G.; Smirnov, E.A.; Zakharycheva, N.S. Linear-dendritic block copolymers based on *N*-isopropylacrylamide and perfluorinated polyphenylenegermane: Synthesis and properties at various interfacial boundaries. *Polym. Sci. B* **2017**, *59*, 708–717.
23. Zamyshlyayeva, O.G.; Deniskina, I.V.; Filippov, A.P.; Semchikov, Y.D. Synthesis and self-organization of amphiphilic poly(*N*-vinylpyrrolidone-2,2,3,3-tetrafluoropropyl methacrylate) diblock copolymers in solution and bulk. *Polym. Sci. A* **2011**, *53*, 691–697. [[CrossRef](#)]
24. Zamyshlyayeva, O.G.; Lapteva, O.S.; Baten'kin, M.A.; Semchikov, Y.D.; Mel'nikova, N.B. Self-organization and aggregation of amphiphilic block copolymers of *N*-vinylpyrrolidone-block-2,2,3,3-tetrafluoropropyl methacrylate at interfaces. *Rus. Chem. Bull.* **2014**, *63*, 1823–1836. [[CrossRef](#)]
25. Kovylin, R.S.; Baten'kin, M.A.; Kulikova, T.I.; Ermolaeva, C.G.; Zamyshlyayeva, O.G.; Chesnokov, S.A. Amphiphilic fluorinated block-copolymer coating for the preparation of hydrophobic porous materials. *J. Polym. Res.* **2018**, *25*, 208. [[CrossRef](#)]
26. Tritschler, U.; Pearce, S.; Gwyther, J.; Whittell, G.R.; Manners, I. Functional nanoparticles from the solution self-assembly of block copolymers. *Macromolecules* **2018**, *50*, 3439–3463. [[CrossRef](#)]

27. Atanase, L.I.; Riess, G. Self-Assembly of block and graft copolymers in organic solvents: An overview of recent advances. *Polymers* **2018**, *10*, 62. [[CrossRef](#)] [[PubMed](#)]
28. Matsuoka, H.; Uda, K. Nanostructure of poly(N-isopropylacrylamide) brush at the air/water interface and its responsivity to temperature and salt. *Langmuir* **2016**, *32*, 8383–8391. [[CrossRef](#)] [[PubMed](#)]
29. Sakuma, M.; Kumashiro, Y.; Nakayama, M.; Tanaka, N.; Haraguchi, Y.; Umemura, K.; Shimizu, T.; Yamato, M.; Okano, T. Preparation of thermoresponsive nanostructured surfaces for tissue engineering. *J. Visual. Exper.* **2016**, *109*, 1–8. [[CrossRef](#)]
30. Armarego, W.L.F.; Chai, C.L.L. *Purification of Laboratory Chemicals*; Elsevier: Amsterdam, The Netherlands, 2012.
31. Fujishige, S. Intrinsic viscosity-molecular weight relationships for poly(N-isopropylacrylamide) solutions. *Polym. J.* **1987**, *19*, 297–300. [[CrossRef](#)]
32. Tsvetkov, V.N. *Rigid-Chain Polymers*, 1st ed.; Plenum Press: New York, NY, USA, 1989.
33. Jaycock, M.J.; Parfitt, G.D. *Chemistry of Interfaces*; John Wiley and Sons: New York, NY, USA, 1981.
34. Kaelble, D.H. *Physical Chemistry of Adhesion*; Wiley Interscience: New York, NY, USA, 1971.
35. Rukenstein, E.; Lee, S.H. Estimation of the equilibrium surface free energy components of restructuring solid surfaces. *J. Coll. Inter. Sci.* **1987**, *120*, 153–161. [[CrossRef](#)]
36. Bushuk, W.; Benoit, H. Light scattering studies of copolymers: I. effect of heterogeneity of chain composition on the molecular weight. *Canad. J. Chem.* **1958**, *36*, 1616–1626. [[CrossRef](#)]
37. Kratochvil, P. *Classical Light Scattering from Polymer Solution*, 1st ed.; Elsevier: Amsterdam, The Netherlands, 1987; pp. 1–346.
38. Schärftl, W. *Light Scattering from Polymer Solutions and Nanoparticle Dispersions*, 1st ed.; Springer: Berlin, Germany, 2007; pp. 1–187.
39. Scholz, C. *Polymer for Biomedicine. Synthesis, Characterization, and Applications*; John Wiley & Sons Ltd: New York, NY, USA, 2017.
40. Pan, W.; Chen, H.; Wen, G.; Giaouzi, D.; Pispas, S.; Zuo, J. Surface micelle structures and monolayer compression moduli of double hydrophilic block copolymer. *J. Phys. Chem. C* **2020**, *124*, 17150–17157. [[CrossRef](#)]
41. Perepichka, I.I.; Lu, Q.; Badia, A.; Geraldine, C.B. Understanding and controlling morphology formation in Langmuir-Blodgett block copolymer films using PS-P4VP and PS-P4VP/PDP. *Langmuir* **2013**, *29*, 4502–4519. [[CrossRef](#)] [[PubMed](#)]
42. Peetla, C.; Graf, K.; Kressler, J. Langmuir monolayer and Langmuir-Blodgett films of amphiphilic triblock copolymers with water-soluble middle block. *Colloid Polym. Sci.* **2006**, *285*, 27–37.
43. Robert, B.; Moffitt, C.M.G. Novel two-dimensional “ring and chain” morphologies in Langmuir-Blodgett monolayers of PS-b-PEO block copolymers: Effect of spreading solution concentration on self-assembly at the air-water interface. *Langmuir* **2005**, *21*, 5453–5460. [[CrossRef](#)]
44. Rodriguez-Parada, J.M.; Kaku, M.; Sogah, D.Y. Monolayers and Langmuir-Blodgett films of poly(N-acylthylenimines) with hydrocarbon and fluorocarbon side chains. *Macromolecules* **1994**, *27*, 1571–1577. [[CrossRef](#)]
45. Choi, M.; Chung, B.; Chun, B.; Chang, T. Surface micelle formation of polystyrene-b-poly(2-vinylpyridine) diblock copolymer at air-water interface. *Macromol. Res.* **2004**, *12*, 127–133.
46. Claro, P.C.S.; Coustet, M.E.; Diaz, C.; Maza, E.; Cortizo, M.S.; Requejo, F.G.; Pietrasanta, L.I.; Ceolín, M.; Azzaroni, O. Self-assembly of PBzMA-b-PDMAEMA diblock copolymer films at the air-water interface and deposition on solid substrates via Langmuir-Blodgett transfer. *Soft Matter* **2013**, *9*, 10891–10898. [[CrossRef](#)]
47. Hood, J.; Van Gordon, K.; Thomson, P.; Coleman, B.R.; Burns, F.; Moffitt, M.G. Structural hierarchy in blends of amphiphilic block copolymers self-assembled at the air-water interface. *J. Colloid Interface Sci.* **2019**, *556*, 392–440. [[CrossRef](#)]