

## Article

# First Principle Study on Structural, Thermoelectric, and Magnetic Properties of Cubic CdCrO<sub>3</sub> Perovskites: A Comprehensive Analysis

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**Abstract:** The primary objective of contemporary manufacturing is to produce items that are low-cost, environmentally friendly, and energy efficient. This study aimed to investigate compounds that fulfil these criteria, with a focus on CdCrO<sub>3</sub>. The full potential linearized augmented plane wave program (FP LAPW), as in Wien2K, was employed to examine the structural, electronic, thermodynamic, and transport characteristics of the material. Structural optimization was carried out using generalized gradient approximation (GGA), with lattice constants that were deemed satisfactory based on previous theoretical and experimental results. Calculations of the magnetic characteristics of CdCrO<sub>3</sub> show that the Cr atoms are principally responsible for magnetism. The quasi-harmonic Debye model allows for the identification of thermodynamic properties including trends, the relative Debye temperature, thermal expansion parameter, relative volume, and heat capacity at various pressures and temperatures. At constant volume, a heat capacity of 52 J/mol K was determined. The thermoelectric properties were examined using the Boltzmann transport offered by the BoltzTrap program. At room temperature, CdCrO<sub>3</sub> had a figure of merit (ZT) value that was almost equal to one, indicating that it may be used to make thermoelectric devices with the highest possible efficiency.

**Keywords:** perovskite; electronic property; thermodynamic property; figure of merit



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## 1. Introduction

Inorganic materials known as perovskite oxides have gained considerable attention recently due to their extraordinary properties and electrical, structural, optical, thermoelectric, and elastic behavior [1–3]. These materials possess different crystalline phases and physicochemical properties, including ferromagnetism, anti-ferromagnetism, ferroelectricity, ferro-elasticity, anti-ferroelectricity, and others. Ferromagnetic perovskite materials, in particular, have become more prominent in spintronics, a field that explores how to use electron spin to store, process, and transmit information [4,5]. The flexible crystal structure of perovskite oxides, which enables the development of several materials with distinctive electrical and other properties, is one of its most alluring features. To maintain the stability of the ABO<sub>3</sub> compound's structure, the standard cubic ABO<sub>3</sub> crystal structure is commonly utilized, where A and B represent two different cations, O represents an anion, and the radius of B is substantially lower than that of the A-cation. An oxygen atom normally occupies the unit cell's center in an ideal cubic ABO<sub>3</sub> structure, which also places a B atom in the center and another atom frequently at one of the cell's corners [6–8]. RE-TM-O<sub>3</sub> perovskites have recently become the subject of rare earth studies. Replacing some of the cations (B) or the TM with magnetic elements has been observed to result in the emergence of unexpected

magneto-electric and multiferroic characteristics in materials [9–11]. The interplay between electric and magnetic orders in these materials has sparked immense interest due to their potential for use in next-generation electronic devices, data storage, and spintronics [12,13]. These materials exhibit both ferroelectric and ferromagnetic orders that are coupled through strain or spin interactions, resulting in the emergence of novel properties. Furthermore, several researchers have investigated the magnetic and electrical properties of various perovskites using ab-initio methods. For instance, the suitability of the  $\text{PrCrO}_3$  compound for spintronics applications was confirmed based on its electronic and magnetic properties, as concluded by M. Yaseen et al. in 2020 [14]. Rashid et al. investigated the electrical and magnetic properties of the cubic  $\text{CeCrO}_3$  compound in the FM phase using density functional theory, specifically the mBJ approximation [15]. Their investigation revealed that the material had half-metallic, ferromagnetic, and total magnetic moments of 4.0 B. The characteristics of  $\text{CaCrO}_3$ ,  $\text{BaCrO}_3$ , and  $\text{SrCrO}_3$ , in particular the length of the Cr-O bond, have been the subject of recent study. Materials made of  $\text{RCrO}_3$  are incredibly adaptable and have outstanding physical studies. These characteristics include, among others, the magnetocaloric effect (MCE), exchange bias, and magnetization reversal. The peculiar magnetic characteristics of  $\text{RCrO}_3$  are a result of its intricate magnetic structure. Despite the scientific community's interest in ferromagnetism in transition metal oxides and their prospective applications, [16] there is currently no literature on the physical properties of  $\text{RCrO}_3$  ( $\text{R} = \text{Ca}, \text{Sr}, \text{Ba}, \text{Cd}, \text{etc.}$ ) for electrical or spintronic devices. Designing devices with the best implementation requires an understanding of the physical properties of materials. Stable phases of novel materials can be created using non-equilibrium progression conditions in molecular beam epitaxy (MBE) or any other physical vapor deposition techniques. This study aims to explore the structural, thermoelectric, and magnetic properties of the  $\text{CdCrO}_3$  cubic perovskite using density functional theory (DFT) [17] to evaluate its potential industrial applications. The FP-LAPW-GGA (PBE) method was used to analyze its features. Furthermore, the lack of available data on the material indicates that this study is the first of its kind to explore these properties, which may lead to future research in the field.

## 2. Computational Method

We performed a study on  $\text{CdCrO}_3$  to assess its structural, magnetic, elastic, and electronic properties using the DFT framework and the FP-LAPW [18] tool from WIEN2k [19]. In our simulations, we took the spin-orbit effect into account using semi-relativistic techniques. The magnetoelectronic properties were determined through DFT calculations, taking exchange and correlations into account. To achieve convergence in the basis set, we used a plane wave cut-off value of  $\text{RMT} \times K_{\text{max}} = 7$  [20,21]. To improve the wave functions inside the muffin-tin spheres, we expanded the radial eigenfunctions by spherical harmonics up to  $l_{\text{max}} = 10$ . Additionally, we raised the potential Fourier charge density of the interstitial region to  $G_{\text{max}} = 14$ . We applied the tetrahedron technique in the irreducible Brillouin zone with a k-grid of  $10 \times 10 \times 10$  and a charge convergence of  $10^{-5}$  as the convergence condition. We also investigated the thermoelectric transport features by employing the quasi-harmonic Debye model [22] and the BoltzTraP method [23] in order to calculate the thermodynamic parameters such as the heat capacity, thermal expansion, and Debye temperature at high pressures and temperatures. We found that a denser k-point grid is necessary for accurate predictions of transport parameters.

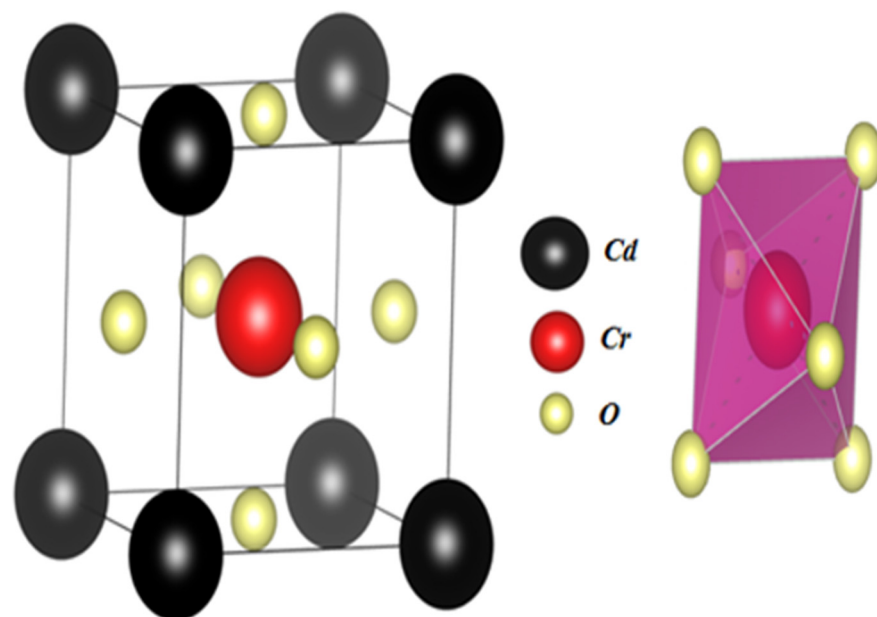
## 3. Results and Discussion

### 3.1. Structural Properties

For a description of geometric structure as mentioned earlier, we enhanced the topologies of the perovskite compounds under investigation while taking into account their cubic structure. Their fully relaxed structure is shown in Figure 1. The component atoms Cd and Cr are located at (0,0,0) and (1/2,1/2,1/2), respectively, whereas the halide ion O is located at (1/2,1/2,  $u$ ) ( $u = 0.249$ ). The examined compounds were discovered to have a Pm-3m space group in a cubic structure [24]. The optimal lattice parameters were determined

using PBE-GGA functional and the Birch Murnaghan equation of state [25] against volume (a.u) in the three phases of non-magnetic (NM), ferromagnetic (FM), and antiferromagnetic phase (AFM), as shown in Figure 2. The computed results of the structural optimization are shown in Table 1. The computed outcomes were found to be in accordance with earlier reports, suggesting the reliability of our computation. Moreover, it has been remarked that the FM phase has minimum energy, suggesting a ferromagnetic ground state for this perovskite. Furthermore, we estimated the tolerance factor ( $\tau$ ) [26,27] to support the cubic structural parameters. The tolerance factor in the range 0.96–1.04 suggests the cubic crystal structure. The computed  $\tau$  for the  $\text{CdCrO}_3$  was found to be 0.97, suggesting a cubic structure. Additionally, it was possible to determine the thermodynamic stability by taking into account the formation energy ( $E_f$ ), which was calculated as [28–30]:

$$E_f = \frac{1}{N} (E_{\text{CdCrO}_3} + E_{\text{Cd}} + E_{\text{Cr}} + 3E_{\text{O}}) \quad (1)$$

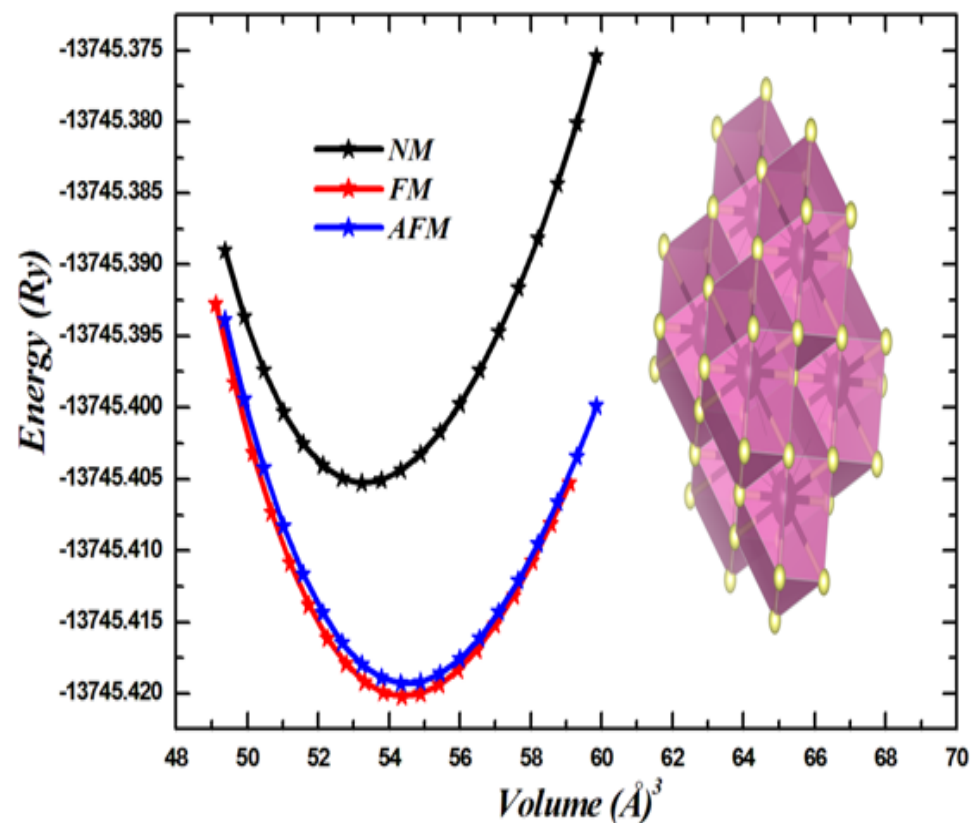


**Figure 1.** Computed crystal structure of cubic  $\text{CdCrO}_3$ .

**Table 1.** Calculated values of equilibrium lattice parameters  $a$  (Å), bulk modulus  $B$  (GPa), pressure derivative of bulk modulus ( $B'$ ), minimum energy ( $E$ ), and Formation energy ( $E_f$ ) of the cubic perovskite  $\text{CdCrO}_3$ .

Compound	States	$a$	$B$	$B'$	$E$ (Ry)	$\Delta E = E(\text{FM}) - E(\text{AFM})$	$E_f$ (eV)
$\text{CdCrO}_3$ (This study)	NM	3.76	200.88	4.65	−13,745.405290	852 mRy	−4.11
	FM	3.78	190.88	4.89	−13,745.420172		
	AFM	3.79	192.01	4.58	−13,745.419320		
$\text{NdCrO}_3$ [31] (other Study)	NM	3.74	226.45				
	FM	3.80	208.93				
	AFM	3.95	189.90				
	Experimental	3.890					

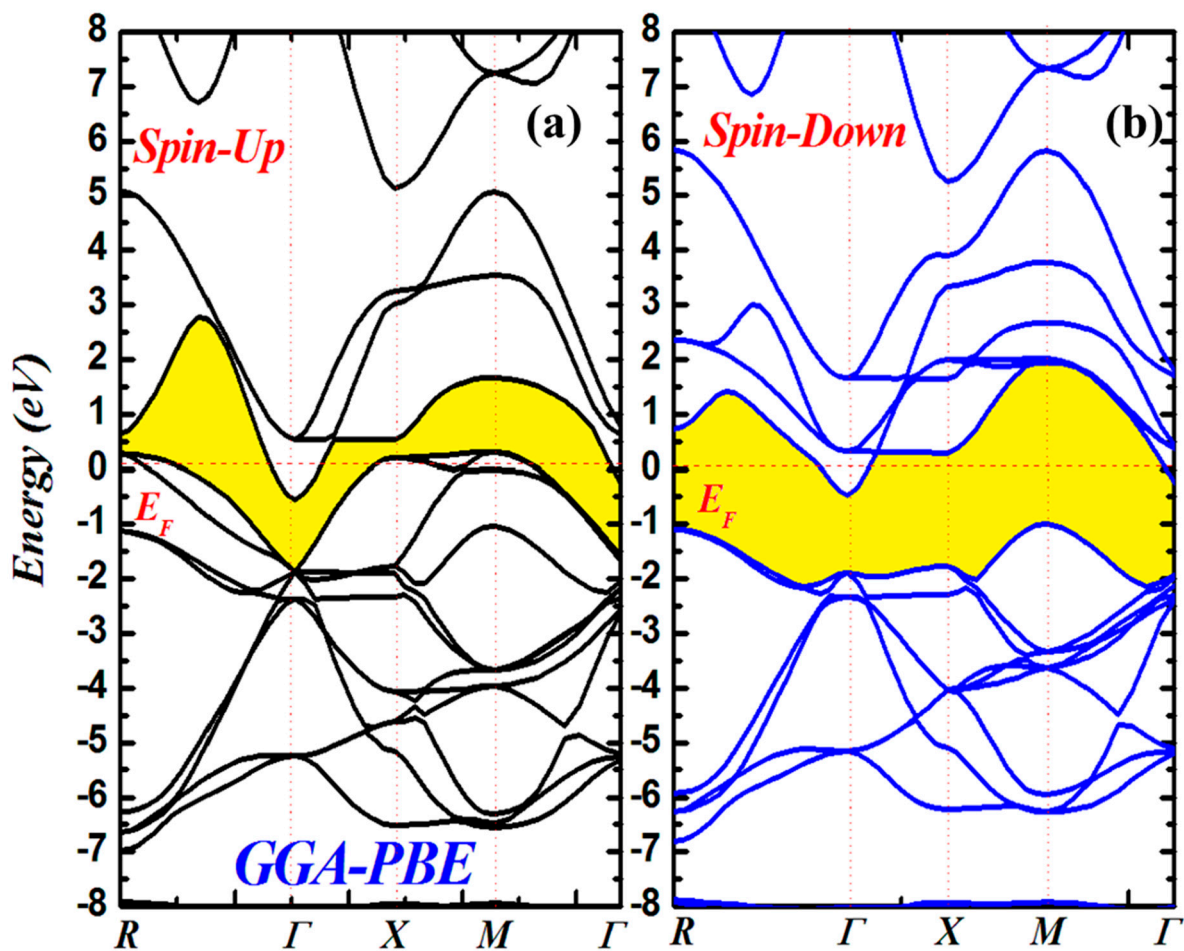
The computed formation energy was found to be negative, reflecting the thermodynamic stability of the investigated  $\text{CdCrO}_3$ . The above outcomes reflect the structural stability of the investigated  $\text{CdCrO}_3$ .



**Figure 2.** Total energy with the volume optimization curve of CdCrO<sub>3</sub> in its NM, FM, and AFM phases.

### 3.2. Electronic Properties

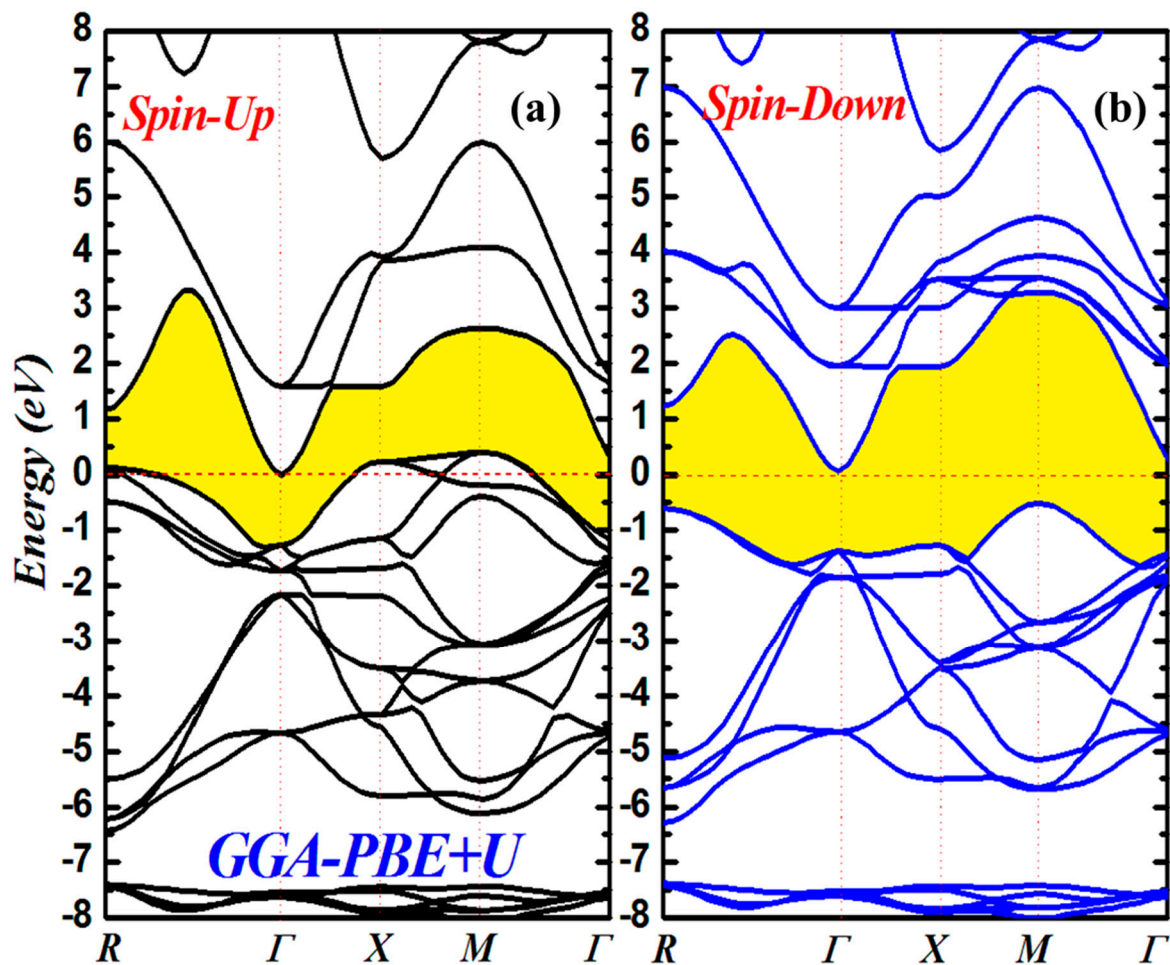
The study of electronic properties is an essential aspect of materials science and is crucial for developing new materials with specific properties for various applications. The electronic band structure of a material is a graphical representation of the allowed energy levels for electrons in a solid [32]. This is determined by the behavior of electrons in the valence and conduction bands and the size of the gap between these bands. The energy gap is a fundamental aspect of the electronic band structure, and its size determines whether a material is an insulator, semiconductor, or metal. The behavior of electrons in the valence and conduction bands and the size of the gap between these bands are critical factors that determine the electronic properties of a material, including its electrical conductivity, optical properties, and magnetic behavior. In this study, the electronic properties of the CdCrO<sub>3</sub> compound were investigated via the most stable structure in balance states and optimized lattice parameters. Figure 3a,b illustrates the electronic band structures for both spin-up and spin-down polarization with GGA. The results indicate that CdCrO<sub>3</sub> is a metallic compound using the GGA-PBE approach, which means that it has high electrical conductivity and behaves as a conductor of electricity. Furthermore, when the Coulomb repulsion terms (GGA+U) [33,34] are introduced, the metallic character is kept in the majority band structure, as shown in Figure 4a,b; however, for the minority band structure, this results in a semiconductor character, suggesting the half metallic nature of the CdCrO<sub>3</sub> compound. This is an important finding as it distinguishes CdCrO<sub>3</sub> from other perovskite compounds that are insulators or semiconductors. The absence of an energy gap in CdCrO<sub>3</sub> suggests that the compound has potential applications in electronic devices that require high electrical conductivity, such as interconnects in microprocessors and other electronic components.



**Figure 3.** Spin-polarized band structures of perovskite oxide  $\text{CdCrO}_3$  using GGA-PBE: (a) spin up and (b) spin down.

The total and partial (projected) densities of states (TDOSs and PDOSs, respectively) determined for  $\text{CdCrO}_3$  are shown in Figure 5. The Fermi level,  $E_F$ , is represented by the vertically broken line. We determined the PDOSs of Cd, Cr, and O in  $\text{CdCrO}_3$  to better understand how each atom contributes to the total TDOSs. The fact that the total DOSs at the Fermi level have non-zero values utilizing the GGA method is evidence that  $\text{CdCrO}_3$  should exhibit metallic conductivity for both spin channels [35,36]. The large contribution of the Cr 'd' states at the Fermi level is what causes this trend. In contrast, the Hubbard parameter has an impact on the Cr 'd' states, separating them in the majority spin and moving them towards higher energies, which results in the creation of a band gap in the minority spin. The GGA+U approximation revealed that the  $\text{CdCrO}_3$  perovskite is composed of half-metallic components. Additionally, the TDOSs curves of  $\text{CdCrO}_3$  show that the valence and conduction bands of this material primarily consist of Cr 'd' orbitals with O 'p'. Electrical stability is significantly influenced by the DOSs at  $E_F$ . The Fermi level's position and  $E_F$  strength affect the phase stability of intermetallic complexes. Those with lower  $E_F$  values are more stable when compared to those with higher  $E_F$  values.



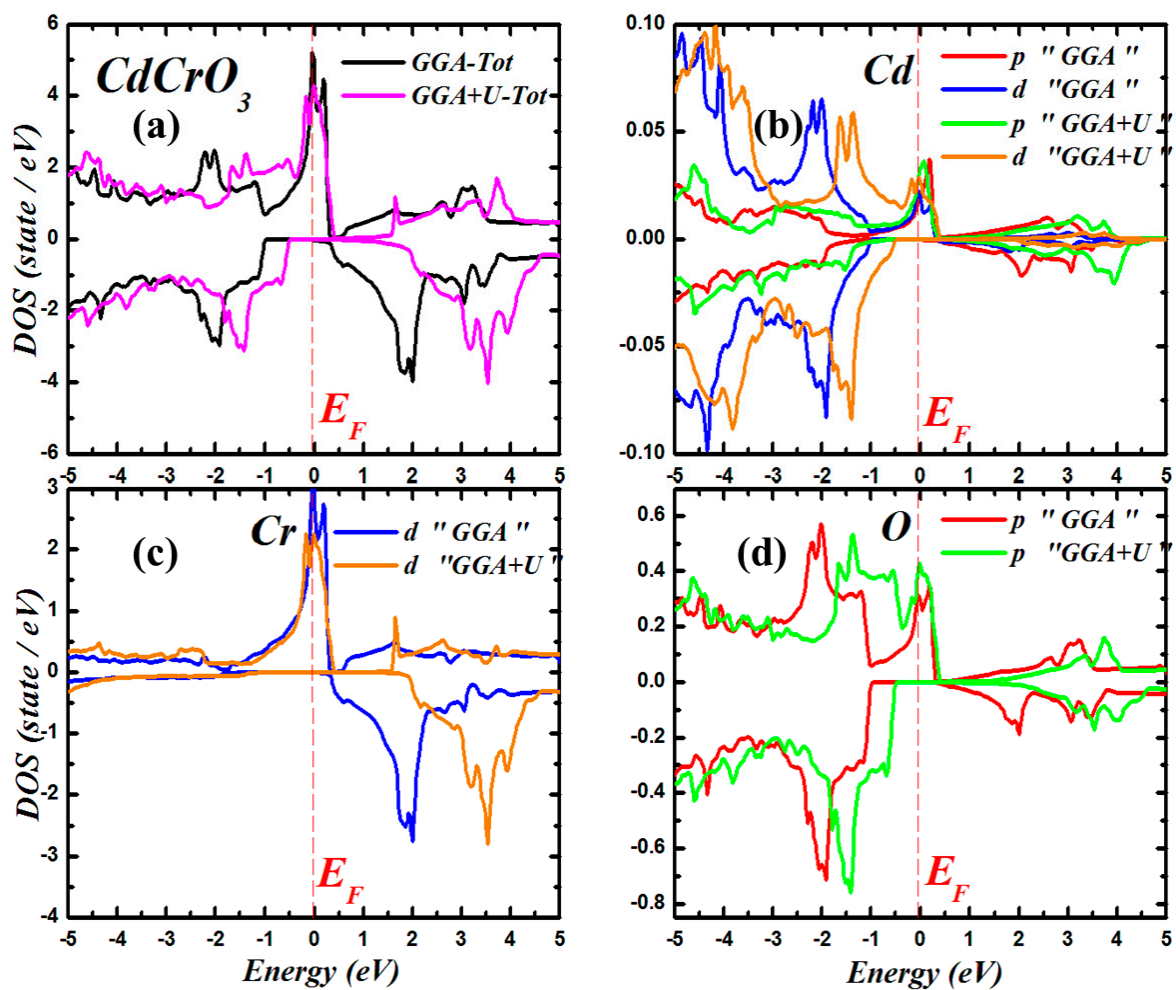


**Figure 4.** Spin-polarized band structures of perovskite oxide  $\text{CdCrO}_3$  using GGA-PBE+U: (a) spin up and (b) spin down.

### 3.3. Magnetic Properties

Magnetism is a fundamental property of materials that has significant applications in various fields, including electronics, energy, and medicine. To study the magnetic properties of materials, one of the crucial considerations is the magnetic moment. The magnetic moment reveals the components and frequency of the material's magnetic field. In this context, the magnetic moments of spins in the  $\text{CdCrO}_3$  compound were investigated.  $\text{CdCrO}_3$  is a complex compound that contains multiple elements, including cadmium (Cd), chromium (Cr), and oxygen (O). The total and partial magnetic moments with polarized spin in the interstitial sites and the muffin-tin spheres were estimated using the GGA and GGA+U models to investigate the magnetic properties of the compound. Table 2 displays the conclusions of these calculations. They are metallic because the total magnetic moments per cell unit were near to an integer Bohr magneton of  $2 \mu_B$ . It was interestingly found that the Cd and O atom magnetic moments were extremely modest and had no impact on the overall moment [37]. Nevertheless, they played a crucial role in the development of magnetism. The environment of these atoms also had an impact on the magnitude of the major band gap in the compound  $\text{CdCrO}_3$ . This means that when examining the magnetic properties of materials, it is important to take these atoms' magnetic moments into account because they are not insignificant. As can be seen in Table 2, the interstitial region was also discovered to have a significant magnetic moment. The result of a sizable charge transfer from Cd and Cr to O is this effect, which can be used to explain the magnetic moment in the interstitial area. This is explained by the fact that the electro-negativity of the O anion, which is 3.5 on Pauling's scale, is higher than that of the Cd and Cr, whose electro-negativity

values are both 1.69 higher than the O anion. The magnetic moment in the interstitial area is caused by a significant charge transfer from Cd and Cr to O as a result of the difference in electro-negativity values. Furthermore, the significant magnetic moment in the interstitial region can have important implications on the compound's properties [38]. For instance, the magnetic moment can affect the electronic and optical properties of the compound, which are essential for various applications. Furthermore, the stability and symmetry of the compound's structure can be affected by the magnetic moment in the interstitial area.



**Figure 5.** (a) Total and partial density of states of perovskite oxide  $\text{CdCrO}_3$  at the equilibrium lattice constant using GGA–PBE and GGA + U, (b) Cd, (c) Cr, and (d) O.

**Table 2.** Computed magnetic properties of  $\text{CdCrO}_3$ .

	$M_{\text{Tot}}$	$M_{\text{Cd}}$	$M_{\text{Cr}}$	$M_{\text{O}}$	$M_{\text{int}}$	$E_g$ (eV)	$T_C$ (K)
GGA	1.99	0.045	1.67	−0.004	0.292		89.46
Up						0	
Down						0	
GGA+U	1.99	0.036	1.91	−0.077	0.276		
Up						0	
Down						0.759	

### 3.4. Thermodynamic Properties

We calculated the thermodynamic characteristics of the  $\text{CdCrO}_3$  under investigation using Gibbs2 [22,39]. The compound under study had its thermal characteristics estimated for a temperature range of 0–800 K. The quasi-harmonic model was used to take pressure

variations between 0 and 15 GPa into account. For the investigated  $\text{CdCrO}_3$  compound, the bulk modulus is greatest at 210 GPa at 0 K, as observed in Figure 6a. According to this,  $B$  increases when pressure is applied at a specific temperature and decreases when the pressure is constant [40]. Additionally, the bulk modulus gradually decreases above 200 K and remains constant between 0 and 200 K. The entropy of  $\text{CdCrO}_3$  varies with temperature, as shown in Figure 6b. It has been noted that the curves are exponential in nature. It was discovered to be almost  $120.66 \text{ J} \cdot \text{mol}^{-1} \text{K}^{-1}$  for the examined  $\text{CdCrO}_3$  at zero pressure and 300 K. The vibrational properties of the compounds are influenced by the heat capacity. However, at low temperatures, it is directly proportional to  $T^3$ , while at intermediate temperatures, it depends on atomic vibration [41]. At high temperatures, the heat capacity at constant volume approaches the Dulong–Petit limit. The plot for  $C_v$  is shown in Figure 6c and contains two phases: the first phase, where  $T$  is below 300 K, shows a sharp increase in  $C_v$  with temperature, and the second phase, where  $T$  is over 300 K, shows a progressive increase in  $C_v$  with temperature. The  $C_v$  augments when it approaches the Dulong–Petit limit at greater temperatures during its final phase.  $\text{CdCrO}_3$  computed heat capacities of  $52 \text{ J/molK}$  at constant volume, as shown in Figure 6c. The  $C_p$  of  $\text{CdCrO}_3$  is depicted in Figure 6d for different pressures. As shown in Figure 6d, the  $C_p$  curves are initially sharply enhanced. Yet, the growth rate of the curves eventually slows down. At relatively low temperatures, only acoustic vibrations can cause vibrational excitations. It follows that variations in temperature and pressure have an equal impact on heat capacity. Through study of the thermal expansion coefficient  $\alpha(T)$ , the compounds' structural stability can be determined. The fluctuation of  $\alpha(T)$  with temperature and pressure is depicted in Figure 6e. Due to the use of a harmonic of the Debye model approximation [42], a strong rise is seen for both compounds up to 200 K. Nonetheless, the thermal expansion coefficient steadily decreases as the temperature rises, indicating that the  $\text{CdCrO}_3$  crystal has good volume invariance. For  $\text{CdCrO}_3$ , we determined the Debye temperature ( $\theta_D$ ), which is depicted in Figure 6f. We can see that at 0 K and 40 GPa pressure, the  $\theta_D$  of  $\text{CdCrO}_3$  is 575. With constant pressure and temperature, as depicted in Figure 6f, it linearly decreases. It was noted that the pressure effect increased the Debye  $D$  temperature. Extensive analysis of the thermodynamic properties will provide insight into the experimental efforts.

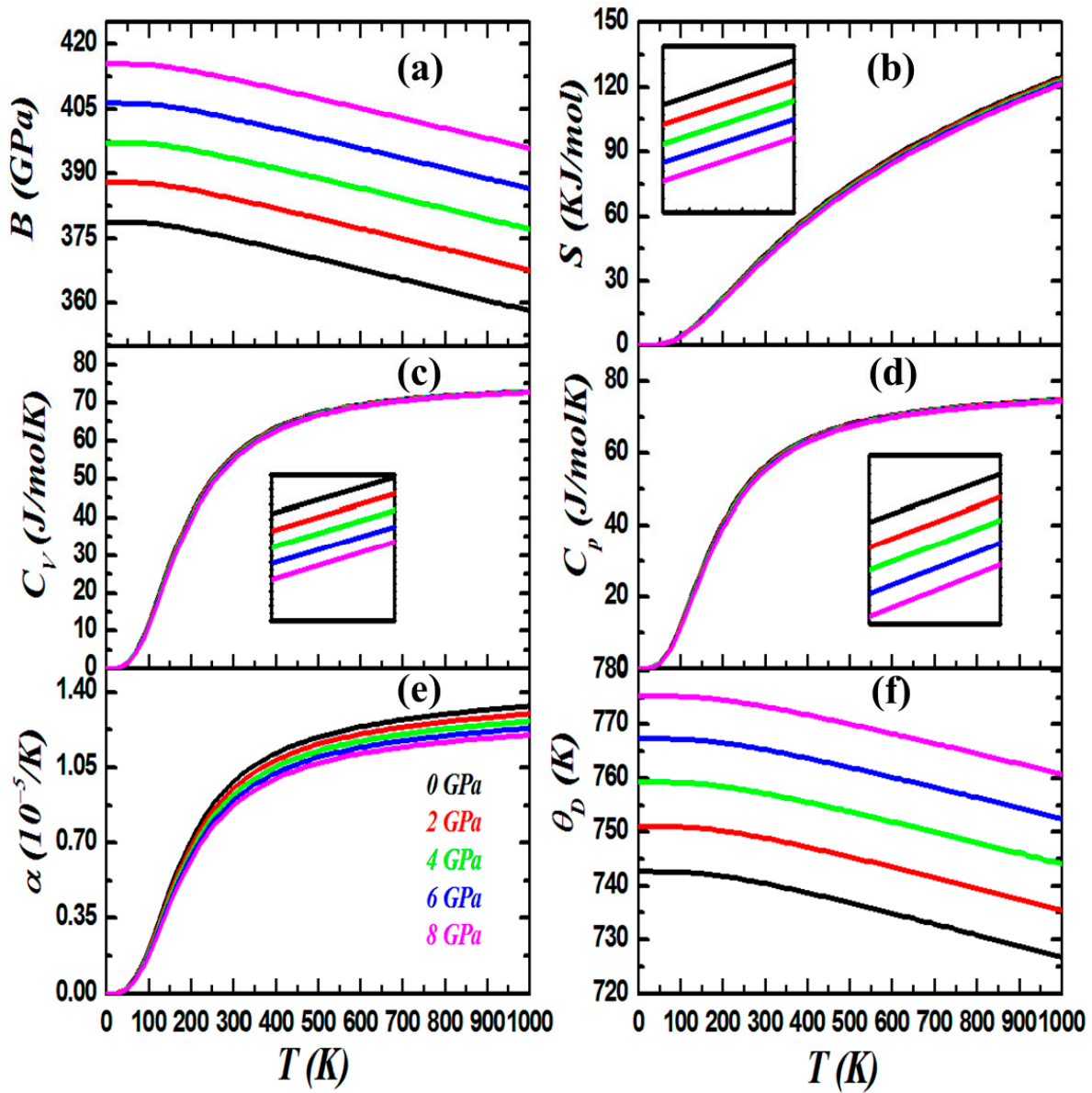
### 3.5. Thermoelectric Properties

The ability of thermoelectric materials to convert thermal energy into useful energy and vice versa has made them a subject of interest worldwide. In this section, we mainly focus on the analysis of several transport coefficients of  $\text{CdCrO}_3$  in both up- and down-spin states, such as electrical conductivity ( $\sigma/\tau$ ), Seebeck coefficient ( $S$ ), and the merit factor ( $ZT$ ). To calculate the thermoelectric parameters, we utilized the BoltzTraP code [23] and an approximation of charge carrier constant relaxation durations.

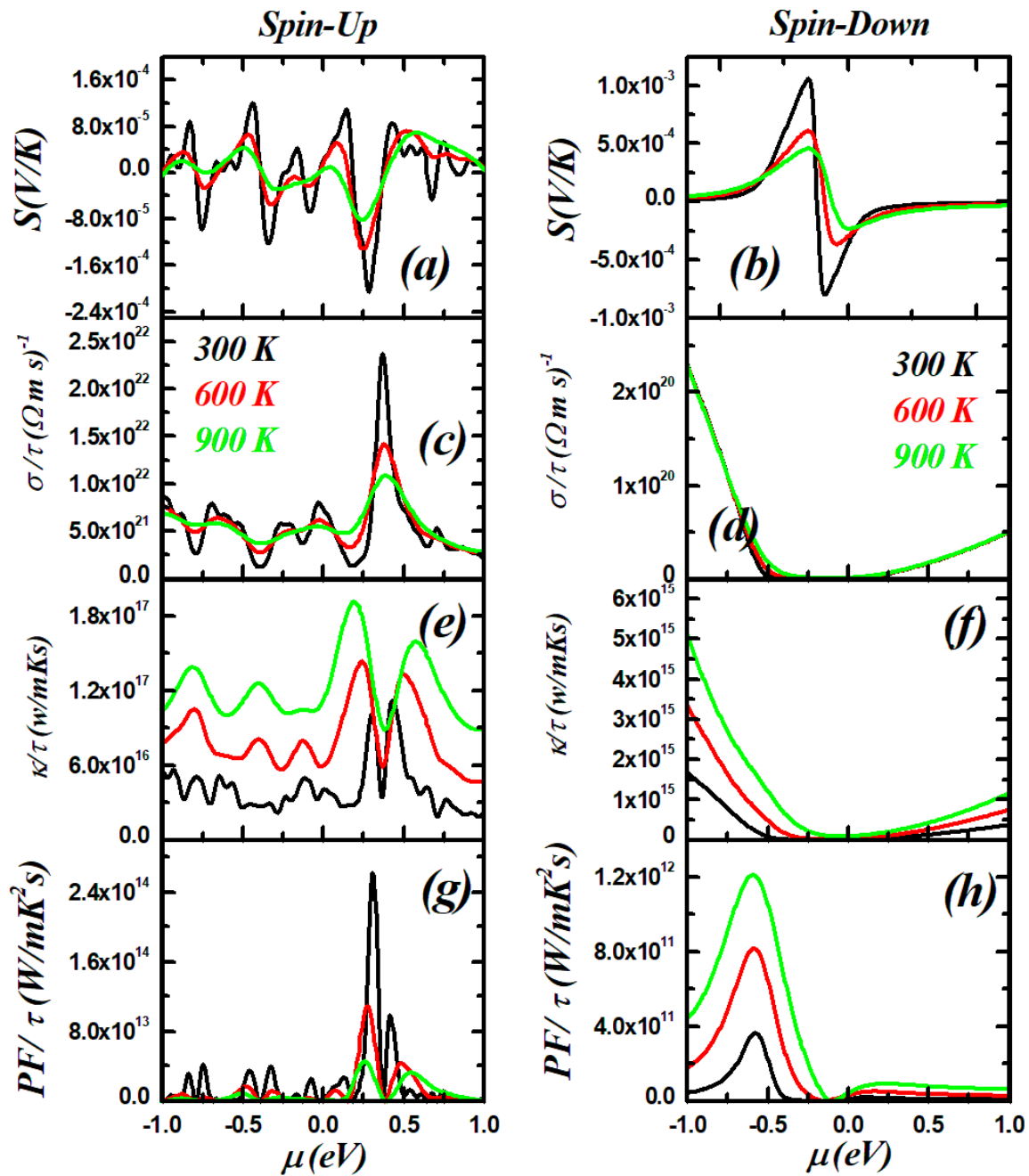
Figure 7 displays the chemical potential dependence of the Seebeck coefficient ( $S$ ), electrical conductivity ( $\sigma/\tau$ ), electronic thermal conductivity ( $\kappa_e/\tau$ ), and power factor ( $\text{PF}/\tau$ ) at different temperatures of 300 K, 600 K, and 900 K for  $\text{CdCrO}_3$ . It was observed that the  $S$  values were negative at the Fermi level ( $\mu_f = 0$ ), suggesting that the majority charge carriers are electrons for both spin channels (Figure 7a,b) and thus confirming the n-type conductivity of this  $\text{CdCrO}_3$  perovskites [43,44]. Moreover, it was also observed that the Seebeck coefficient value decreased with the increasing temperature. Furthermore, Figure 7c,d depicts the variation in electrical conductivity ( $\sigma/\tau$ ) of  $\text{CdCrO}_3$  with chemical potential. There is a difference in the electrical conductivity calculated for the spin-up state and that obtained for the spin-down state of  $\text{CdCrO}_3$ . For the spin-up state, we found that  $\sigma/\tau$  behaves as a metal, while for the spin-down state, it looks like a semiconductor. The same trend was observed in the case of the electronic thermal conductivity ( $\kappa_e/\tau$ ) (see Figure 7e,f) as  $\kappa_e$  is related to the electrical conductivity ( $\sigma$ ) through the Wiedemann–Franz [45,46] relation  $\kappa_e = L\sigma T$ , where  $L$  is the Lorenz number. Figure 7g,h represents the variation in the power factor ( $\text{PF}$ ) with the chemical potential for  $\text{CdCrO}_3$ . It was noted



that CdCrO<sub>3</sub> has a maximum value of PF for n-type in the majority spin whereas it has a maximum for p-type in the minority spin.



**Figure 6.** (a) Volume ( $V$ ), (b) entropy ( $S$ ), (c) heat capacity at constant volume ( $C_V$ ), (d) heat capacity at constant pressure ( $C_p$ ), (e) thermal expansion coefficient ( $\alpha$ ), and (f) Debye temperature ( $\theta_D$ ) versus the different temperatures for CdCrO<sub>3</sub>.



**Figure 7.** The Seebeck coefficient ( $S$ ) (a) Spin-Up, (b) Spin-Down, electrical conductivity ( $\sigma/\tau$ ), (c) Spin-Up, (d) Spin-Down, electronic thermal conductivity ( $\kappa_e/\tau$ ) (e) Spin-Up, (f) Spin-Down, power factor ( $S^2\sigma$ ) versus the chemical potential ( $\mu$ ) at different temperatures for  $\text{CdCrO}_3$  (g) Spin-Up, and (h) Spin-Down.

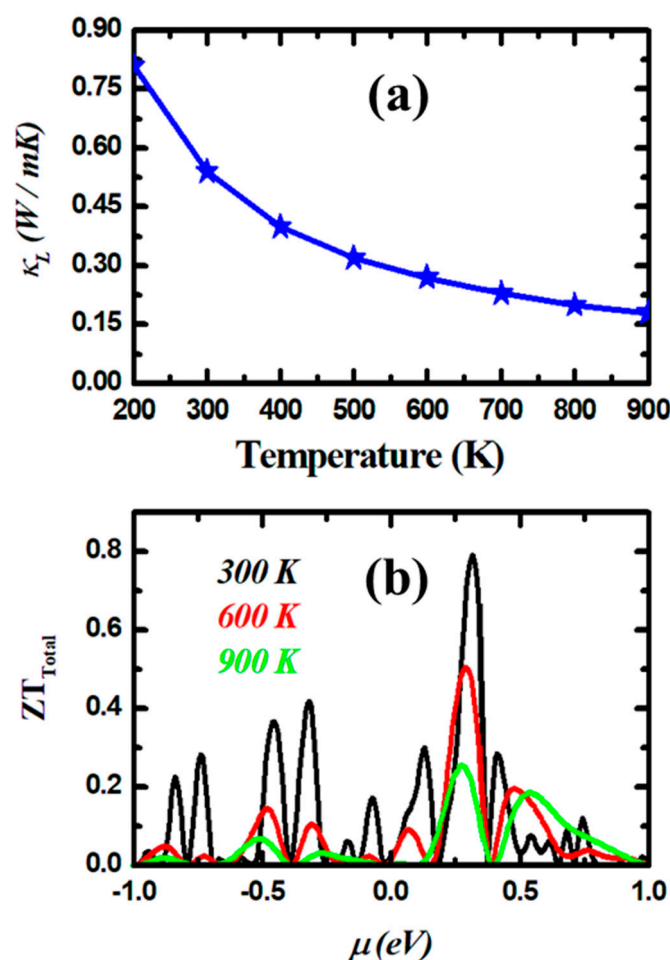
It is well known that the figure of merit [47] is given by:

$$ZT = \frac{S^2\sigma}{\kappa} T \quad (2)$$

where  $\kappa = \kappa_e + \kappa_l$  is the total thermal conductivity [48]. Therefore, one has to estimate the lattice thermal conductivity ( $\kappa_l$ ) using the following formula:

$$\kappa_l = A \cdot \frac{\overline{M}\theta_D^3\delta}{\gamma^2 T n^{2/3}} \quad (3)$$

where  $\bar{M}$  represents the average atomic mass of all constituent atoms,  $A$  is a constant calculated as  $A = \frac{2.43 \cdot 10^{-8}}{1 - \frac{0.514}{\gamma} + \frac{0.228}{\gamma^2}}$ ,  $n$  is the number of atoms in the primitive cell,  $\delta$  is the average atomic volume,  $T$  is the absolute temperature,  $\gamma$  is the Grüneisen parameter, and  $\theta_D$  is the Debye temperature [49]. The estimated  $\kappa_l$  for the CdCrO<sub>3</sub> compound is depicted in Figure 8a. From this figure, it can be well observed that the  $\kappa_l$  decreases with the temperature increase. Now, considering the total  $S$ ,  $\sigma$ , and  $\kappa$  and using a constant relaxation time  $\tau \approx 10^{-14}$  s, the total ZT of the CdCrO<sub>3</sub> material is evaluated, as shown in Figure 8b. At 300 K, the maximum ZT is marked to be 0.4 and 0.8 for p-type and n-type doping, respectively, suggesting the optimum thermoelectric response of the studied compound at room temperature. These ZT maximum values correspond to the carrier concentrations of about  $4.22 \times 10^{19} \text{ cm}^{-3}$  and  $2.31 \times 10^{22} \text{ cm}^{-3}$  for p-type and n-type doping, respectively. Moreover, by increasing the temperature, these ZT<sub>max</sub> values gradually decrease.



**Figure 8.** Computed lattice thermal conductivity ( $\kappa_L$ ) and figure of merit ( $ZT_{\text{Total}}$ ) for CdCrO<sub>3</sub>.

#### 4. Conclusions

In this study, the structural, elastic, vibrational, and thermodynamic characteristics of CdCrO<sub>3</sub> perovskite were examined using the FP-LAPW and quasi-harmonic Debye models. Using the generalised gradient approximation of Perdew–Burke and Ernzerh of exchange correlation (GGA-PBE), we were able to optimise the lattice parameters. The computed structural characteristics, such as the equilibrium lattice constant, bulk modulus, and pressure derivative, are consistent with previous studies. According to the overall density of states and the electronic energy band structure, both materials are half-metallic and have Cd-4d Cr ‘d’ electronic orbitals as their dominant conductivity. Through the quasi-harmonic Debye model, we also identified thermodynamic properties, such as the

relative Debye temperature, thermal expansion parameter, relative volume, heat capacity at different temperatures and pressures, and trends. In the study, 52 J/molK of heat capacity was discovered at constant volume. At high temperatures, the CdCrO<sub>3</sub> compound shows good volume invariance, according to the coefficient of thermal expansion. This study is a part of a larger theoretical undertaking that also includes our investigation into the characteristics of this compound and could inspire further studies in this area.

**Author Contributions:** Conceptualization, S.S., T.S., M.M.S. and K.K.M.; methodology, S.S., M.B., M.M.S. and K.K.M.; software, S.S., M.B., T.S., M.M.S. and K.K.M.; validation, K.K.M.; formal analysis, M.B. and T.S.; investigation, K.K.M.; writing—original draft preparation, S.S.; writing—review and editing, S.S., M.B. and K.K.M.; visualization, K.K.M. All persons who meet authorship criteria are listed as authors, and all authors certify that they have participated sufficiently in the work to take public responsibility for the content, including participation in the concept, design, analysis, writing, or revision of the manuscript. All authors have read and agreed to the published version of the manuscript.

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