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# Facet-Dependent Interfacial Charge Transfer in TiO<sub>2</sub>/Nitrogen-Doped Graphene Quantum Dots Heterojunctions for Visible-Light Driven Photocatalysis

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**Abstract:** Interfacial charge transfer is crucial in the efficient conversion of solar energy into fuels and electricity. In this paper, heterojunction composites were fabricated, comprised of anatase TiO<sub>2</sub> with different percentages of exposed {101} and {001} facets and nitrogen-doped quantum dots (NGQDs) to enhance the transfer efficiency of photo-excited charge carriers. The photocatalytic performances of all samples were evaluated for RhB degradation under visible light irradiation, and the hybrid containing TiO<sub>2</sub> with 56% {001} facets demonstrated the best photocatalytic activity. The excellent photoactivity of TiO<sub>2</sub>/NGQDs was owed to the synergistic effects of the following factors: (i) The unique chemical features of NGQDs endowed NGQDs with high electronic conductivities and provided its direct contact with the TiO<sub>2</sub> surface via forming Ti–O–C chemical bonds. (ii) The co-exposed {101} and {001} facets were beneficial for the separation and transfer of charge carriers in anatase TiO<sub>2</sub>. (iii) The donor-acceptor interaction between NGQDs and electron-rich {101} facets of TiO<sub>2</sub> could remarkably enhance the photocurrent, thus hindering the charge carriers recombination rate. Extensive characterization of their physicochemical properties further showed the synergistic effect of facet-manipulated electron-hole separation in TiO<sub>2</sub> and donor-acceptor interaction in graphene quantum dots (GQDs)/TiO<sub>2</sub> on photocatalytic activity.

**Keywords:** electron transfer; graphene quantum dots; heterojunction; photocatalysis; TiO<sub>2</sub>

## 1. Introduction

Anatase TiO<sub>2</sub> is generally considered a better photocatalyst than rutile, mainly due to its attributes of longer exciton diffusion length, higher electron mobility, and longer carrier life time [1,2]. The photocatalytic activity of anatase has been revealed to depend closely on the crystal surface [3,4]. Clear pictures have now shown that reduction and oxidation reactions would preferentially occur on {101} and {001} facets, respectively. Furthermore, it has been found that the {101} surface is attractive for electrons in aqueous solutions while excess electrons tend to strongly avoid the {001} surface via surface

science experiments and first-principles simulations [5]. The substantial electrons in anatase  $\text{TiO}_2$  generated via photoexcitation play an important part in many energy-related applications. However, due to intrinsic defects and the fast carrier recombination rate, electron trapping in anatase  $\text{TiO}_2$  are unavoidable, which hamper the overall photocatalytic activity [6,7].

To solve this problem, fabrication of heterojunctions modified anatase  $\text{TiO}_2$  has emerged as a promising method [8–10]. Angus and co-workers reported the development of a hetero-structured material by using pre-formed carbon nitride nanosheets (CNNS) composite with facet-controllable  $\text{TiO}_2$ . The materials possess an excellent  $\text{CO}_2$  adsorption capacity and charge transfer rate, thus leading to the improvement of the photocatalytic activity of  $\text{TiO}_2$  [11]. Luca Rimoldi et al. have then reported a method to combine  $\text{TiO}_2$  with  $\text{WO}_3$ . Due to the admirable properties of  $\text{WO}_3$ , the photocatalytic activity enhanced remarkably [12]. Through a series of experiments and calculations, Latterly Olowoyo et al. have also found that carbon nanotubes (CNTs) can strongly be attached to the {101} facet of  $\text{TiO}_2$ , since the atomic orbitals of anatase overlap with the orbitals of the CNTs [13].

As a novel class of quantum dots (QDs), graphene quantum dots (GQDs) have currently attracted intensive interest in fabricating new heterojunctions, due to their large surface areas, high electron mobilities, conductivity, and adjustable band gaps [14,15]. These properties ensure discrete electronic levels, which could allow for light-induced electron injection, efficient carrier transfer, and long-lived excited states [16–20]. GQDs have also been explored as the light absorber and heteroatoms-doped GQDs are expected to realize absorption in the visible region [21]. A variety of surface functional groups on GQDs could provide better covalent chemical linking between anatase and GQDs, facilitating charge separation and transfer behaviors [22–24]. Pan et al. have found that monodispersed amine-functionalized GQDs anatase  $\text{TiO}_2$  heterojunctions have an absorption range extended into the visible light region and a much lower carrier recombination rate. They attributed the improved performance to the proper energy position of GQDs/ $\text{TiO}_2$  [23]. Then, Yu and co-workers reported the decoration of GQDs on {001} faceted anatase  $\text{TiO}_2$  with an exposed percentage of 65%–75%. The experiments show a promotion of photocatalytic hydrogen evolution rate of the composites compared to bare anatase, which might originate from the higher-charge separation efficiency. [25] Zheng and co-workers also utilized  $\text{TiO}_2$  and sulfur, nitrogen co-doped GQDs (SN-GQDs) to develop an efficient photocatalyst for synthesizing  $\text{H}_2\text{O}_2$ . They testified that SN-GQDs induced visible light absorption, promoted charge transfer, and provided active sites for ·OOH formation [26]. Recently, Prezhdo et al. built several models of donor-acceptor interaction between GQDs and  $\text{TiO}_2$  via stacking and covalent bonding, respectively, to provide guidance for subsequent photocatalysis application [16].

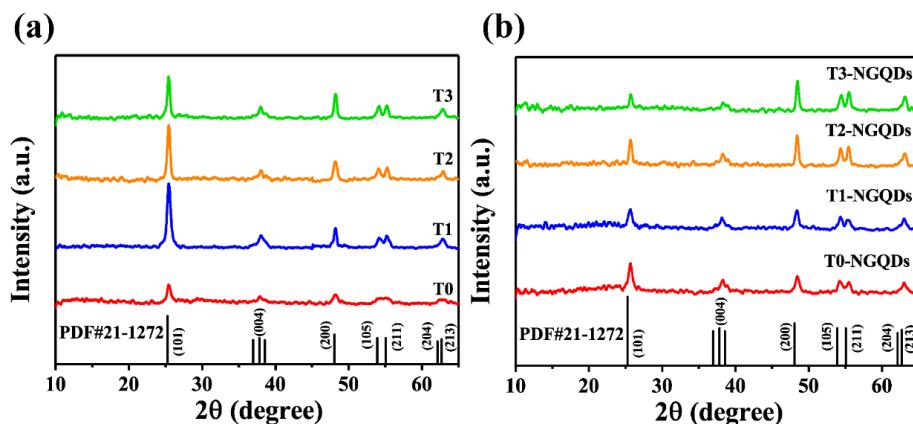
These findings provide strong evidence and also motivation for deeply understanding the models of electron-hole separation dynamics at heterojunction interfaces [3,21,27,28]. There has still been no adequate investigation, considering the influence of facet-dependent photogenerated charge-carrier separation in anatase  $\text{TiO}_2$ , on the photocatalytic activity of GQDs/ $\text{TiO}_2$ . Since both theoretical and experimental studies had shown that {101} crystal facets are electron-rich while {001} crystal facets are hole-rich in anatase  $\text{TiO}_2$ , it would be desirable to elucidate the different donor-acceptor interaction between NGQDs and  $\text{TiO}_2$  with specific facet composition, thus providing explicit guidance in constructing heterojunction structures with superior performance.

Herein, we have designed a heterojunction composite via depositing nitrogen-doped GQDs on anatase  $\text{TiO}_2$  with different exposure percentages of {001} and {101} facets, combining the advantages of facet and interfacial modification to maximize the driving force promoting charge carrier transfer. The visible light-driven dye degradation performances on anatase  $\text{TiO}_2$  with {101}, {001}, and {001}–{101} facets, and their corresponding heterojunctions with nitrogen-doped GQDs (NGQDs) have been systematically studied. To further understand the electron transfer mechanism, the relationships of the morphology, chemical states, optical, and electrical properties with the photocatalytic activity were intensively analyzed.

## 2. Results and Discussion

### 2.1. Structural Characterization

The crystalline phases of different samples before and after decoration with NGQDs were firstly identified. Figure 1 shows the X-ray diffraction (XRD) patterns of anatase  $\text{TiO}_2$  with different exposed facets. The diffraction peaks appeared around  $2\theta$  values of 25.3, 38.6, 48.0, 53.9, and 62.1, assigned to the (101), (112), (200), (105), and (213) crystal planes of anatase titania, respectively [27]. For bare anatase  $\text{TiO}_2$ , with the increase of Hydrofluoric acid (HF) volume, the {004} diffraction peak was broadened, implying the thickness of the  $\text{TiO}_2$  along the {001} direction was decreased. Meanwhile, the intensity of the {200} diffraction peaks was enhanced, indicative of the increasing side length of the nanoparticles along the {100} direction. According to the two peaks, the percentage of the exposed {001} facet could be estimated, based on the calculation method reported in the literature. Shown in Table 1, the percentage was increased when increasing the HF volume [29]. The calculation method was described in the supporting information, as shown in Figure S1. By comparison, the XRD patterns of  $\text{TiO}_2/\text{NGQDs}$  exhibited similar but much lower diffraction peaks. The XRD pattern of NGQDs showed a wide weak diffraction peak centered at  $26.8^\circ$ , assigned to the {002} facet [30]. Though the diffraction peak of NGQDs was not observed in the composites, which might be due to the relatively low diffraction intensity of NGQDs, the peak attributed to the {101} facet of anatase decreased obviously. The possible reason is that the {002} crystal orientation in NGQDs influenced the {101} surface of  $\text{TiO}_2$ , thus leading to decrease of the peak [31].



**Figure 1.** XRD patterns of different samples (a) without and (b) with nitrogen-doped quantum dots (NGQDs) decoration.

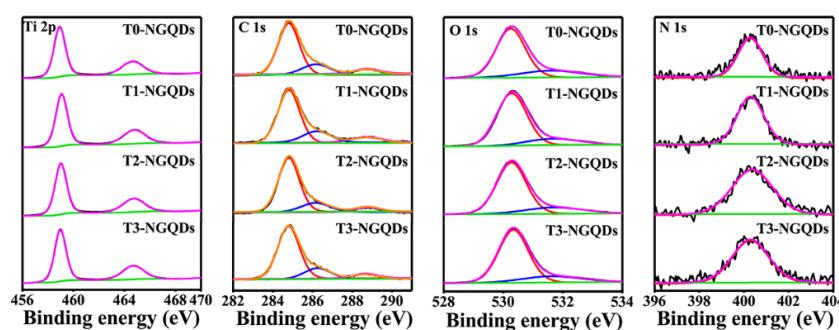
**Table 1.** Structural information of different anatase titania samples.

Samples	Average thickness (nm)	Average length (nm)	Percentage of {001}
T0	8.9	8.1	12%
T1	8.5	13.8	56%
T2	6.1	21.6	71%
T3	4.3	25.5	85%

In Raman spectroscopy, all samples show similar peaks centered at 144, 394, 514, and  $636\text{ cm}^{-1}$  shown in Figure S2. When increasing the addition amount of HF, the intensity of the Eg peak at  $144\text{ cm}^{-1}$  decreased simultaneously. The Eg peak is mainly attributed to the symmetric stretching vibration of O–Ti–O  $\text{TiO}_2$  [29]. A higher percentage of exposed {001} facets generally represented fewer amount of symmetric O–Ti–O stretching vibration modes, thus leading to the decreasing intensity of Eg peak in the Raman spectra. Thus, it could be concluded from the Raman spectra that the exposure percentage of {001} facets increased with the increase of HF volume. Shown in the inset of Figure

S2b, two characteristic peaks of D and G band appeared at  $1351\text{ cm}^{-1}$  and  $1590\text{ cm}^{-1}$ , respectively, confirming the presence of graphite-like structure in the composites.

X-ray photoelectron spectroscopy (XPS) spectra were measured to study the bonding conditions in the heterojunctions, shown in Figure 2. The Ti 2p spectra showed two peaks with the binding energies of 458.9 eV and 464.6 eV, which are assigned to  $\text{Ti } 2p^{3/2}$  and  $2p^{1/2}$  spin-orbital splitting photoelectrons, respectively. The splitting values indicated  $\text{Ti}^{4+}$  chemical states in these samples, while no  $\text{Ti}^{3+}$  forms were observed [32]. The C 1s spectra could be then fitted into three Gaussian peaks (288.8 eV, 286.1 eV, and 284.8 eV). The peak at 288.8 eV was assigned to the  $\text{sp}^2$  hybridized carbon in the skeleton of NGQDs and also some carbon contaminants from the ambience. The other two peaks corresponded to the oxygenated carbon, representative of carboxyl carbon (288.8 eV) and hydroxyl carbon (286.1 eV) functional groups, respectively [33–36]. No Ti–C bond related peak (282 eV) was observed, implying that the NGQDs were probably anchored to the surface of  $\text{TiO}_2$  via Ti–O–C bonds. In some reported studies, functional groups including C–O and COOH were evaluated to identify the existence of Ti–O–C bonding [32,37]. These groups are not that stable and might be converted to the epoxy group during the composite formation process.



**Figure 2.** XPS spectra of different  $\text{TiO}_2/\text{NGQDs}$  composites.

The evidence of Ti–O–C bonding formation was further provided in the O 1s XPS spectra. It was fitted into two symmetric peaks. The peaks at 530.3 eV is ascribed to the oxygen in crystal lattice ( $\text{Ti}=\text{O}=\text{Ti}$ ) and the other peak at 531.6 eV is believed to result from the Ti–O–C bonding, based on previously reported cases [25,32,38]. These results indicated the composite formation of  $\text{TiO}_2/\text{NGQDs}$  through the C–O–Ti bonds. The C–O–Ti bonds are capable of mediating the coupling between NGQDs and  $\text{TiO}_2$ , which could promote the interfacial electron transfer. The N 1s spectra revealed a peak centered at 400.3 eV, which could be assigned to the pyrrolic N (400.5 eV). The nitrogen atoms are mainly introduced by the NGQDs, demonstrating the successful decoration of NGQDs on the surface of  $\text{TiO}_2$  [30]. Furthermore, the percentages of different bonds according to the fitting results of the XPS high-resolution spectra were calculated, shown in Table 2 and Table S1. Similar percentages implied similar bonding and chemical composition in different composites.

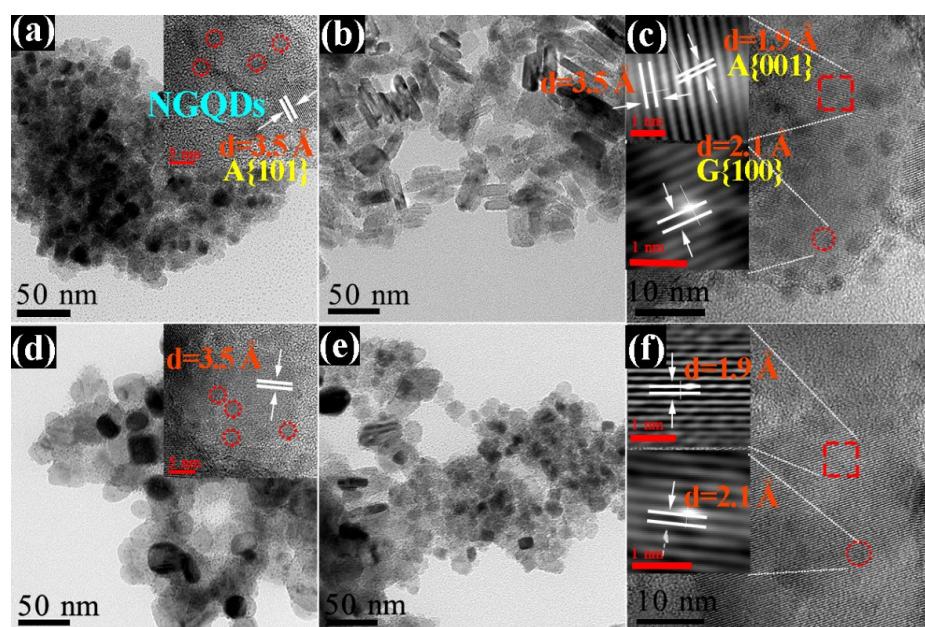
**Table 2.** Percentages of different bonds according to the fitting results of the XPS high-resolution spectra calculated from Figure 2.

Bond	T0-NGQDs	T1-NGQDs	T2-NGQDs	T3-NGQDs
% of Ti 2p	26.57	27.63	27.23	27.69
% of C 1s	21.18	19.4	20.65	18.99
O=C=O/Ti–O–C 288.8 eV	7.91	8.03	7.84	7.83
C–O 286.1 eV	19.53	20.30	19.05	19.26
C=C 284.8 eV	72.56	71.07	73.40	73.34
% of O 1s	51.69	52.33	51.53	52.85
Ti–O–Ti 530.3 eV	77.80	80.97	80.84	80.42
Ti–O–C 531.6 eV	22.20	19.03	19.16	19.58
% of N 1s	0.55	0.64	0.58	0.47

## 2.2. Morphology Characterization

The morphology of anatase  $\text{TiO}_2$  with and without NGQDs decoration were characterized by Transmission Electron Microscope (TEM) and High Resolution Transmission Electron Microscope (HR-TEM). The morphology and crystal facets of  $\text{TiO}_2$  remained similar before and after NGQDs modification, as can be seen from Figure 3 and Figure S3. Figure S3 shows that sample T0 is mostly composed of nanoparticles with a truncated octahedral bipyramidal. After adding HF into the reaction system, it could be found that the anatase  $\text{TiO}_2$  mostly consists of nanoplates and the plate size increases with the increase of HF volume, which is consistent with the XRD calculation results. Figure S4 shows the TEM image of NGQDs, which has an average size of  $\sim 2.4$  nm. Clear lattice fringes demonstrative of its well-crystalline structure, and the autocorrelated HRTEM lattice image (inset in Figure S4b) show a  $0.21$  nm lattice fringe assigned to the  $\{100\}$  plane of GQDs [39,40].

The formation of NGQDs/ $\text{TiO}_2$  heterojunctions could be obviously observed in the TEM images, shown in Figure 3, of which the NGQDs were uniformly decorated on both  $\{001\}$  and  $\{101\}$  facets of  $\text{TiO}_2$ . The NGQDs are shown with red circles. No selective deposition of NGQDs on a specific facet of anatase  $\text{TiO}_2$  was found. The autocorrelated HRTEM lattice image (inset in Figure 3c,f) both show  $0.21$  nm lattice fringes assigned to the  $\{100\}$  plane of GQDs. Meanwhile, the autocorrelated HRTEM lattice image (inset in Figure 3c,f) also show clear lattice fringes of  $0.35$  nm and  $0.19$  nm, which could be assigned to the  $\{101\}$  and  $\{001\}$  facet of  $\text{TiO}_2$ , respectively. The HRTEM lattice images of all composites show that the lattices of both NGQDs and  $\text{TiO}_2$  are simultaneously recognized, revealing good attachment of NGQDs over the  $\text{TiO}_2$  surface.



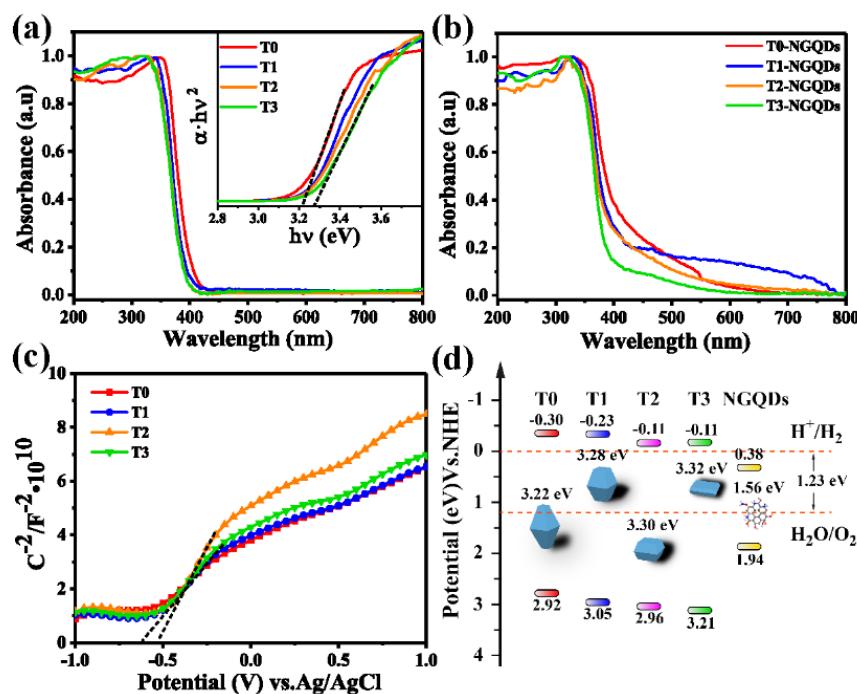
**Figure 3.** TEM images of (a) T0-NGQDs, (b,c) T1-NGQDs, (d) T2-NGQDs, and (e–f) T3-NGQDs. The inset in (a) is the HR-TEM image of T0-NGQDs. The inset in (c) is the autocorrelated HRTEM lattice images recorded from the corresponding selected areas. The inset in (d) is the HRTEM image of T2-NGQDs. The inset in (f) is the autocorrelated HRTEM lattice images recorded from the corresponding selected areas.

## 2.3. Optical and Electrical Properties

The optical properties of all samples were investigated via the Ultraviolet-Visible (UV-Vis) diffuse reflection spectroscopy (UV-DRS). All anatase  $\text{TiO}_2$  demonstrate an absorption threshold near  $400$  nm in the ultraviolet region. The band gaps of the anatase could be obtained based on the Kubelka–Munk rule, seen from the inset in Figure 4a. The band gaps of T0, T1, T2, and T3 are  $3.22$ ,  $3.28$ ,  $3.30$ , and  $3.32$  eV, respectively. Apparently, with the increase of the  $\{001\}$  facet percentage, the light absorption

edge was slightly blue shifted [21]. The UV-vis spectrum of NGQDs is shown in Figure S4d, and the band gap was approximately 1.56 eV, according to our previous work [30,41]. After decorating NGQDs, it was found that there was an increasing visible light absorption for all composites. Generally, the resultant extended light absorption was due to the existence of Ti–O–C chemical bonds between GQDs and TiO<sub>2</sub> [32,37]. The interaction could improve the interfacial carrier transfer rate, which is beneficial to photocatalysis under visible light irradiation. Meanwhile, the presence of the energy gap of NGQDs further ensured the long-lived excited states and absorbance of solar photons in the broad solar spectrum.

The electrochemical Mott–Schottky experiments of the anatase were then measured (Figure 4c). The plots present a positive slope and the flat band potential values were recalculated vs. NHE (Normal Hydrogen Electrode). Combined with the band gaps of anatase TiO<sub>2</sub> and NGQDs, we proposed electronic band structures for all composites, shown in Figure 4d. It was hypothesized that the downshift of conduction band (CB) band level of the anatase might make the electrons less reductive and also weaken the dynamics of the electron transfer rate between TiO<sub>2</sub> and NGQDs [42–44].

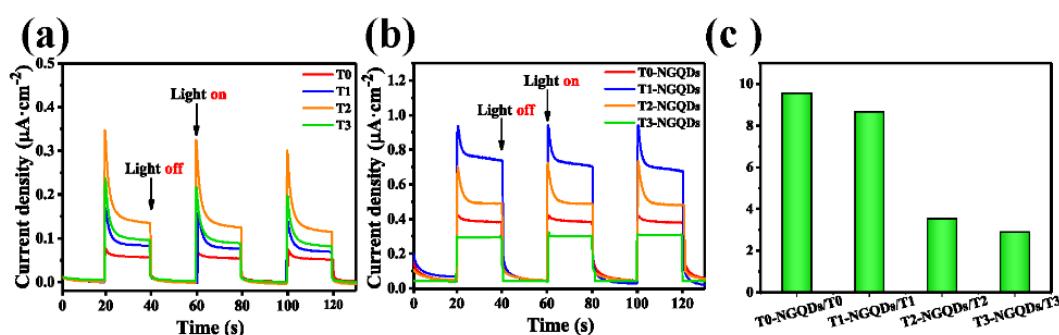


**Figure 4.** UV-vis diffuse reflection spectra (UV-DRS) of (a) bare anatase TiO<sub>2</sub>, (b) TiO<sub>2</sub>/NGQDs composites, (c) Mott–Schottky plots of bare anatase TiO<sub>2</sub>, and (d) band structure diagram of different samples and NGQDs. The inset in (a) is the Tauc plot of the corresponding bare anatase TiO<sub>2</sub>.

Moreover, the photogenerated charge carrier separation and transfer rate was determined using photocurrent responses, shown in Figure 5. Fast and uniform photocurrents with good reproducibility was demonstrated, indicative of relatively reversible photo-responses. Via three on-off cycles under visible light irradiation (>420 nm), it was found that the photocurrent density of bare anatase TiO<sub>2</sub> decreased in the order of T2, T3, T1, and T0. This suggests that defects (oxygen vacancies) are possibly formed on the surface of pristine anatase TiO<sub>2</sub> with small size under light illumination in our work, which would introduce defect energy levels in the band gap and lead to electron-hole separation under visible light irradiation [25]. Meanwhile, the synergistic effect of {001} and {101} facets would also affect the charge carrier separation efficiency, resulting in the difference of photocurrent.

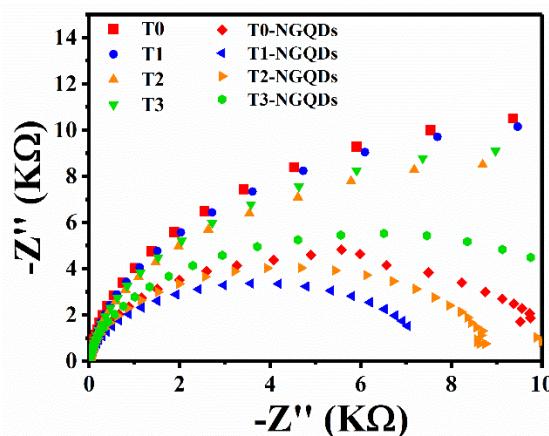
After depositing NGQDs, their corresponding photocurrent responses were all enhanced obviously. The photocurrent density of TiO<sub>2</sub>/NGQDs composites decreased in the order of T1-NGQDs, T2-NGQDs, T0-NGQDs, and T3-NGQDs. The photocurrent density of sample T1-NGQDs was the highest, about

three times larger than that of sample T3-NGQDs. The photocurrent enhancement could be attributed to the promoted separation rate of photogenerated charge carriers, owing to the introduction of NGQDs. The unique chemical features of NGQD endow it with superior carrier mobility and excellent electronic conductivity [45,46]. Moreover, the extended  $\pi$ -electron systems of NGQD provide its sufficient contact with the surface of titania, and the formation of Ti–O–C bonding could also facilitate the donor-acceptor interaction [16]. These factors contributed to the apparent enhancement of photocurrent density. We calculated multiple times the photocurrents of the anatase  $\text{TiO}_2$  with and without NGQDs decoration, shown in Figure 5c. Interestingly, the times decreased with the decrease of exposed {101} facet percentage in  $\text{TiO}_2$ . The phenomenon indicated a noticeable improvement of electron-hole separation efficiency between the NGQDs and the electron-rich {101} facet, compared to that between the NGQDs and the hole-rich {001} facet. The difference demonstrated that efficient electron transfer existed in the interfacial interaction between NGQDs and  $\text{TiO}_2$  with high {101} facet exposure.



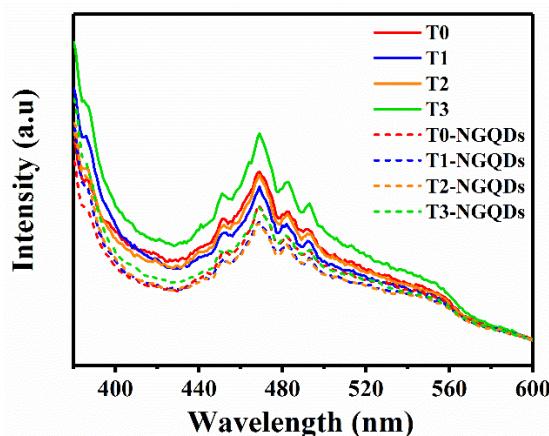
**Figure 5.** Periodic on-off photocurrent output of (a) bare anatase  $\text{TiO}_2$ , and (b)  $\text{TiO}_2$ /NGQDs composites. (c) The multiple times of the photocurrents of the anatase  $\text{TiO}_2$  with and without NGQDs decoration.

Electrochemical impedance spectroscopy (EIS) measurements were utilized to investigate the mechanism of photocurrent improvement. The semicircle diameter generally indicates the carrier transfer resistance. In Figure 6, all Nyquist plots of the composites presented as semicircle, the corresponding arc radius of EIS Nyquist plots all decreased compared to that of bare anatase  $\text{TiO}_2$ , which is consistent with the photocurrent output. Among all photocatalysts, sample T1-NGQDs exhibited the smallest semicircle while sample T0 the largest. This result is demonstrative of more effective carrier separation and transfer process in the heterojunctions [47,48]. Thus, it could be concluded that the percentage of exposed {001} facet indeed had different effects on the electronic properties of the composites, which might be attributed to the different interfacial interaction between NGQDs and  $\text{TiO}_2$  with different facet compositions.



**Figure 6.** Electrochemical impedance spectroscopy (EIS) Nyquist plots of different  $\text{TiO}_2$  samples without and with NGQDs decoration.

Photoluminescence (PL) emission spectrum could help to directly understand the carrier behaviors and observe the radiative recombination of charge carriers [49,50]. Generally, PL emission signals are caused by the photo-induced carrier recombination process. Lower intensity is relevant to better photocatalytic performance. All peaks shape similarly in Figure 7. After decorating NGQDs, the composites exhibited a slight decrease in the PL intensity, compared to the bare anatase, which was probably due to the efficient electron transfer from the CB band of anatase to NGQDs. Thus, the trapping and recombination of charge carriers could be hindered.



**Figure 7.** PL spectra of different  $\text{TiO}_2$  samples without and with NGQDs decoration.

#### 2.4. Photocatalytic Performance

The photocatalytic performances of all samples were evaluated for RhB photodegradation under visible light irradiation after achieving absorption-desorption balance in the dark, as seen from Figure 8. Obviously, the photocatalytic degradation efficiency of the  $\text{TiO}_2/\text{NGQDs}$  heterojunctions was greatly enhanced compared with that of the bare anatase  $\text{TiO}_2$ . Among the composites, approximately 96% of the dye was photo-degraded by the T1-NGQDs within 3 h. The degradation process followed the pseudo-first-order kinetics:

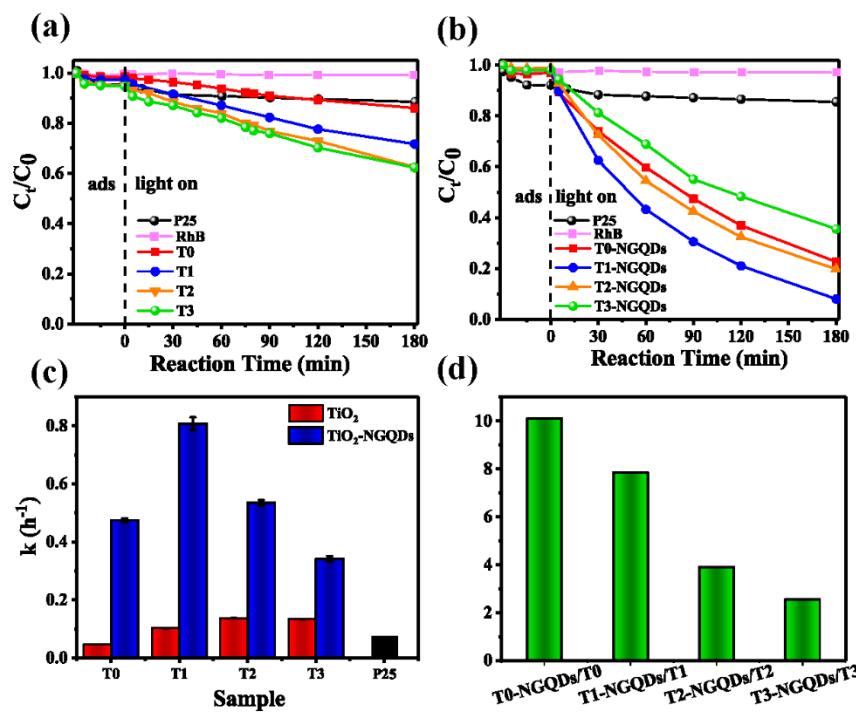
$$-\ln\left(\frac{c}{c_0}\right) = kt \quad (1)$$

where  $k$  is equal to the corresponding slope of the fitting line, representing the rate constant indicative of the photocatalytic efficiency. To prove the efficiency of sample T1-NGQDs, the photocatalytic degradation activity of other three organic pollutants including methylene blue (MB), methyl orange (MO), and phenol (Phe) were compared. It demonstrated that the T1-NGQDs exhibited high photocatalytic activity for degrading MB dye as well as other common organic species (including phenol, colorless aqueous solution), shown in Figure S6.

The calculated  $k$  values of different samples are shown in Figure 8c (the red column corresponds to the bare anatase  $\text{TiO}_2$  and the blue column to the composites). Before decorating NGQDs, sample T2 and T3 exhibited nearly the same constant rate, higher than that of T0 and T1. RhB could be easily absorbed on the titania surface with reactive {001} facet exposure, leading to subsequent dye self-photosensitization process decomposing RhB under visible light irradiation [51–53].

After decoration of NGQDs, it was found that the values were much higher, and the calculated  $k$  of T1-NGQDs reached the highest, about  $0.8 \text{ h}^{-1}$ . The multiple times of the  $k$  values of the anatase  $\text{TiO}_2$  with and without NGQDs decoration were then calculated, shown in Figure 8d. Notably, it was shown that the times decreased with the increase of the exposed {001} facet percentage in  $\text{TiO}_2$ , which is consistent with the photocurrent variation in Figure 5c. The same law illustrates the photocatalytic degradation performance depends greatly on the interfacial charge carrier separation and transfer rate, demonstrative of the significant donor-acceptor interaction between NGQDs and {101} facets. At the same time, it was supposed that the different roles of {001} and {101} facets in separating the

photogenerated electron-hole are unignorable, since the hybrid containing  $\text{TiO}_2$  with 56% {001} facets exhibited a higher reaction rate value than that with 12% {001} facets.



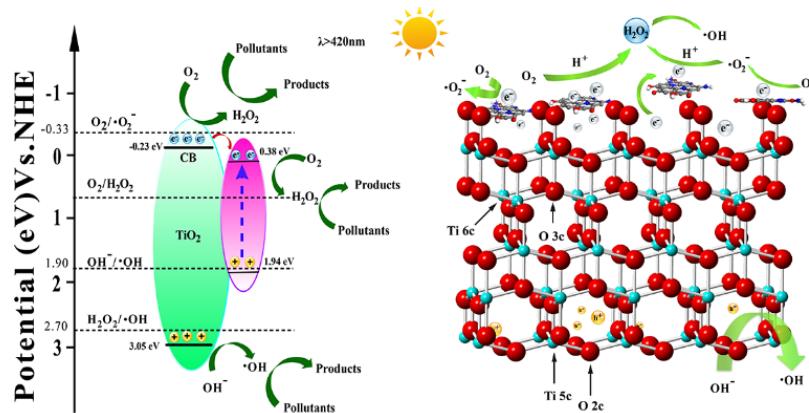
**Figure 8.** The photocatalytic degradation of RhB for different samples under visible light irradiation. Change of the relative concentration ( $C_t/C_0$ ) of RhB in (a) bare anatase  $\text{TiO}_2$ , (b)  $\text{TiO}_2/\text{NGQDs}$  composites as a function of irradiation time up to 180 min. (c) Plot of  $k$  ( $\ln(C_0/C_t)$ ) values for RhB degradation in different samples. (d) The times of the  $k$  values of the anatase with and without NGQDs decoration.

## 2.5. Photocatalytic Mechanism

The schematic representation of the electron-hole separation and transfer in  $\text{TiO}_2/\text{NGQDs}$  heterojunction composites during the photocatalytic reaction is shown in Scheme 1. Under visible light irradiation, both NGQDs and anatase  $\text{TiO}_2$  of avoidable intrinsic defects were capable to generate photo-excited electrons. Obviously, the narrow energy gaps for NGQDs allow for rich hot electrons to produce when the excitation wavelength is larger than 420 nm. According to the measured band potential values, the energy levels in Figure 4d further demonstrate that the band configuration contributes to electron injections from the CB level of  $\text{TiO}_2$  to the LUMO of NGQDs. The donor-acceptor interaction was greatly promoted due to the full contact and C–O–Ti formation between NGQDs and  $\text{TiO}_2$ . The appropriate band alignments explain for the reason why the photocatalytic activities were all enhanced after depositing NGQDs on the surface of  $\text{TiO}_2$ , compared to that of bare  $\text{TiO}_2$ .

On the other hand, for bare  $\text{TiO}_2$ , it was found that the anatase  $\text{TiO}_2$  with the higher percentage of exposed {001} facet owned a much better photocatalytic performance which is due to the high activity of the {001} facet. However, the reaction rate value ( $k$ ) achieved the highest for T1-NGQDs rather than T3-NGQDs. The different variation trend of  $k$  value in bare  $\text{TiO}_2$  and  $\text{TiO}_2/\text{NGQDs}$  heterojunctions imply the possible influence of facets in the interfacial electron transfer process. It revealed that the photo-excited electron and holes behave differently in  $\text{TiO}_2$ , where electrons could be easily trapped in the {101} facet while holes tend to run to the {001} facet. Since NGQDs are a good electron transport medium, its deposition on the titania surface with more percentages of exposed {101} facets could result in a higher electron transfer efficiency, which is consistent with the increasing multiple times of photocurrent and also  $k$  values with and without NGQDs decoration. Simultaneously, the different roles of {101} and {001} facets in anatase  $\text{TiO}_2$  on charge carrier separation is not negligible, considering that the  $k$  value of T1-NGQDs is higher than that of T0-NGQDs. By comparing the degradation activity

of  $\text{TiO}_2$  before and after NGQDs decoration, it could be concluded that there is a synergistic effect of facet-manipulated electron-hole separation in  $\text{TiO}_2$  and donor-acceptor interaction in GQDs/ $\text{TiO}_2$  on the visible light driven photocatalytic performance.



**Scheme 1.** Schematic illustration of the proposed band alignment and interfacial electron transfer process for the  $\text{TiO}_2$ /NGQDs heterojunction composites under visible light irradiation ( $\lambda > 420 \text{ nm}$ ).

### 3. Materials and Methods

#### 3.1. Synthesis of Anatase $\text{TiO}_2$

The anatase  $\text{TiO}_2$  with different exposed percentages of {001} and {101} facets were prepared via the traditional hydrothermal method. Typically, different volumes (0.2, 0.4, and 0.8 mL) of hydrofluoric acid ( $\geq 40.0\%$ , Sinopharm Chemical Reagent Co., Ltd, Shanghai, China) were added into the mixture of tetrabutyl titanate (5 mL;  $\geq 99.0\%$ , Aladdin, Shanghai, China) and ethanol (8 mL). After stirring for 30 min, the mixture was then transferred into a 50 mL Teflon-lined autoclave and heated at  $180^\circ\text{C}$  for 24 h. After that, the products were collected by centrifugation, followed by being rinsed several times with absolute ethyl alcohol and dried at  $60^\circ\text{C}$  overnight. The obtained anatase  $\text{TiO}_2$  co-exposed with {001} and {101} facets were named as T1, T2, and T3 respectively. To prepare  $\text{TiO}_2$  with the {101} dominating plane, the same procedure was conducted but with 0.4 mL of  $\text{H}_2\text{O}$ , which was named T0.

#### 3.2. Synthesis of NGQDs

The NGQDs were prepared according to a one-step hydrothermal process reported by Sun and co-workers [54]. In a typical run, 1.44 g of urea (AR, Aladdin, Shanghai, China) and 1.68 g of citric acid (GR, Aladdin, Shanghai, China) were dissolved in 40 mL of deionized water (DI water). The solution was transferred into a 50 mL Teflon-lined stainless autoclave and heated at  $180^\circ\text{C}$  for 8 h. The final product was centrifuged several times at 10,000 rpm for 5 min with absolute alcohol. The obtained NGQDs precipitate was dried at  $80^\circ\text{C}$  for 1 h to obtain the NGQDs powders.

#### 3.3. Synthesis of $\text{TiO}_2$ /NGQDs Heterojunction Composites

An ultrasonic-hydrothermal method was used to prepare the  $\text{TiO}_2$ /NGQDs heterojunction photocatalysts. A certain amount of  $\text{TiO}_2$  and GQDs at a low doping level (1.0 wt%) was added into 40 mL of DI water. The suspension solution was placed in an ultrasonic bath for 30 min and then transferred into a 50 mL Teflon-lined stainless autoclave by heating at  $120^\circ\text{C}$  for another 2 h. After centrifuging several times with DI water and being dried at  $60^\circ\text{C}$  overnight, the final powders were named as T0-NGQDs, T1-NGQDs, T2-NGQDs, and T3-NGQDs, respectively.

### 3.4. Characterization

The XRD patterns of all samples were recorded with a PAN analytical X'Pert Pro MPD diffractometer (Pananalytical, Holland) using Cu-K $\alpha$  radiation ( $\lambda = 0.1541$  nm), and the data was collected from  $20^\circ$  to  $80^\circ$  ( $2\theta$ ). UV-vis diffuse reflectance spectroscopy (DRS) were taken at room temperature measured using BaSO<sub>4</sub> as the reference on a UV-3150 spectrophotometer (Shimadzu, Kyoto, Japan). The photoluminescence (PL) spectroscopy was performed using a RF-5301pc fluorescence spectroscopy (Shimadzu, Kyoto, Japan). Band gap energies were calculated by analysis of the Tauc-plots resulting from Kubelka–Munk transformation of absorption spectra. High resolution transmission electron microscope (HRTEM) were conducted by a Phillips/FEI Tecnai F20 S-TWIN TEM (Hillsborough, OR, USA) instrument operating at 200 kv. X-ray photoelectron spectroscopy (XPS) measurements were carried out on an ESCALAB 250 Xi (Thermo Scientific, MA, USA) using non-monochromatized Mg-K $\alpha$  X-ray as the excitation source. The binding energies for the samples were calibrated by setting the measured binding energy of C 1s to 284.60 eV. The Raman spectra were measured on a LabRAM HR Evolution (Horiba, Tokyo, Japan) at room temperature using the 532 nm line of an argon ion laser as the excitation source.

### 3.5. Photocatalytic Performance

The photocatalytic performance of the as-synthesized photocatalysts were examined under a 300 W Xe lamp (PLS-SXE 300/300 UV, Perfect Light, Shanghai, China) equipped with a 420 nm cut-off filter as the visible light irradiation source. A total of 15 mg of catalysts were added into 50 mL of a  $10.0 \text{ mg}\cdot\text{L}^{-1}$  solution of different dyes: Rhodamin B (RhB), methylene blue (MB), methyl orange (MO), and phenol (Phe) (Sinopharm Chemical Reagent Co. Ltd., Shanghai, China). The molar concentration was 0.026, 0.031, 0.030, and 0.11 mmol/L, respectively. Before irradiation, the suspension was stirred in the dark to ensure the adsorption-desorption equilibrium of RhB on the surface of the photocatalyst.

### 3.6. Photoelectrochemical Measurements

The photocurrent measurements, electrochemical impedance spectroscopy (EIS) and Mott–Schottky experiments were conducted on an electrochemical analyzer (CHI 660C work station, CHI, Shanghai, China). The employed standard three-electrode configuration included a platinum plate (as the counter electrode), an Ag/AgCl electrode (as the reference electrode), and a working electrode. The working electrodes were prepared as follows: 40 mg of powders and 5 mg of Mg(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O ( $\geq 98\%$ ; Alfa Aesar, Shanghai, China) were dispersed in 100 mL of isopropanol. The suspension was ultrasound for 1 h. A clean SnO<sub>2</sub> transparent conductive glass doped with fluorine, FTO (as cathode) facing the stainless-steel anode was then immersed into this suspension. The distance between the two electrodes was fixed at about 5 cm. The Mg<sup>2+</sup> adsorbed samples suspension was loaded in a quartz vessel as the electrolyte, and the electrophoresis process was performed at 60 V for 120 s. After the electrophoretic deposition (EPD) process, the prepared electrodes were washed by ethanol and deionized water several times and dried at room temperature. A 350 W xenon lamp with a cut-off filter ( $\lambda > 420$  nm) was used as a light source and placed 20 cm away from the working electrode. The working electrode was immersed in 0.1 M Na<sub>2</sub>SO<sub>4</sub> aqueous solution. The EIS measurements were performed over a range from 0.01 to 1000 Hz at 0.2 V, and the amplitude of the applied potential in each case was 5 mV.

## 4. Conclusions

In summary, nitrogen-doped graphene quantum dots were successfully deposited onto the surface of anatase TiO<sub>2</sub> with different percentages of exposed {001} facets to form TiO<sub>2</sub>/GQDs heterojunction composites. The photocatalytic performances of the hybrid containing TiO<sub>2</sub> with 56% {001} facets exhibited the highest reaction rate value under visible light irradiation. The successful decoration of NGQDs on the TiO<sub>2</sub> surface extended the light absorption edge into the visible light region. Due

to Ti–O–C formation and high electron conductivity of NGQDs, the photocurrent responses of the composites were all enhanced obviously, compared to that of the bare samples. Meanwhile, the different roles of {101} and {001} facets in anatase  $\text{TiO}_2$  on charge carrier separation was also not negligible. The improved photocatalytic activity was due to the synergistic effect of facet-manipulated electron-hole separation in  $\text{TiO}_2$  and the remarkable donor-acceptor interaction between NGQDs and the electron-rich {101} facet. The existence of the {101} facet contributed to the interfacial electron transfer that played a vital role in improving the photocatalytic activity of the NGQDs/ $\text{TiO}_2$  heterojunctions. Furthermore, the existence of both facets in anatase assisted the further enhancement of photocatalytic performance. This work provided a new clue to improve the interfacial charge transfer in faceted semiconductor related heterojunctions.

**Supplementary Materials:** The following are available online at <http://www.mdpi.com/2073-4344/9/4/345/s1>, Figure S1: (a) Slab model of anatase  $\text{TiO}_2$  single crystal. (b) Equilibrium model of anatase  $\text{TiO}_2$  single crystal. (Calculation method of the percentage of {001} facets). Figure S2: (a) Raman spectra of different  $\text{TiO}_2$  samples without and with NGQDs decoration. (b) Raman spectra of T1 before and after decoration of NGQDs. The inset in Figure S2 b is the enlargement of  $1200\text{--}1700\text{ cm}^{-1}$  of T1-NGQDs. Figure S3: TEM images of (a) T0, (b) T1, (c) T2, and (d) T3. Figure S4: (a) TEM image, (b) HRTEM image, (c) AFM image, (d) UV-vis spectra and PL spectra of the GQDs (the excitation wavelength is 365 nm), (e) Raman spectra, and (f) XRD pattern of NGQDs. The inset in (a) is the size distribution of NGQDs. The inset in (b) is the autocorrelated HRTEM lattice images recorded. Figure S5: HRTEM images of the anatase  $\text{TiO}_2$  decorated with NGQDs. Figure S6: (a) The photocatalytic degradation of different pollutants for T1-NGQDs, (b) plot of k values for different pollutants degradation in T1-NGQDs.

**Author Contributions:** The manuscript was written through contributions of all authors. N.-Q.O., H.-J.L. designed the experiments; N.-Q.O., B.-W.L., B.-J.G. and X.S. performed the experiments and analyzed the data; N.-Q.O. and H.-J.L. wrote the paper; D.-J.Q. revised the paper. X.W., Y.J. and J.Y. acted as supervisor.

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**Conflicts of Interest:** The authors declare no conflicts of interest.

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