

Article



# An Alumina-Supported Ni-La-Based Catalyst for Producing Synthetic Natural Gas

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Academic Editor: Rajendra S. Ghadwal

Received: 21 September 2016; Accepted: 25 October 2016; Published: 31 October 2016

**Abstract:** LaNi<sub>5</sub>, known for its hydrogen storage capability, was adapted to the form of a metal oxide-supported ( $\gamma$ -Al<sub>2</sub>O<sub>3</sub>) catalyst and its performance for the Sabatier reaction assessed. The 20 wt % La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> particles were prepared via solution combustion synthesis (SCS) and exhibited good catalytic activity, achieving a CO<sub>2</sub> conversion of 75% with a high CH<sub>4</sub> selectivity (98%) at 1 atm and 300 °C. Characteristics of the La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst were identified at various stages of the catalytic process (as-prepared, activated, and post-reaction) and in-situ DRIFTS was used to probe the reaction mechanism. The as-prepared catalyst contained amorphous surface La–Ni spinels with particle sizes <6 nm. The reduction process altered the catalyst make-up where, despite the reducing conditions, Ni<sup>2+</sup>-based particles with diameters between 4 and 20 nm decorated with LaO<sub>x</sub> moieties were produced. However, the post-reaction catalyst had particle sizes of 4–9 nm and comprised metallic Ni, with the LaO<sub>x</sub> decoration reverting to a form akin to the as-prepared catalyst. DRIFTS analysis indicated that formates and adsorbed CO species were present on the catalyst surface during the reaction, implying the reaction proceeded via a H<sub>2</sub>-assisted and sequential CO<sub>2</sub> dissociation to C and O. These were then rapidly hydrogenated into CH<sub>4</sub> and H<sub>2</sub>O.

Keywords: Sabatier reaction; lanthanum nickel; catalysis

# 1. Introduction

Fossil fuels such as oil, coal, and natural gas have and continue to be the primary means employed to meet society's energy needs. A key downside in relying on these carbon-rich compounds has been the accompanying increase in atmospheric CO<sub>2</sub> levels, from ~290 ppm in pre-industrial days to ~395 ppm in 2013, which is seen as the major culprit behind climate change [1,2]. Growing awareness of the impact anthropogenic CO<sub>2</sub> emissions have on global warming and the detrimental consequences further increases may bring has resulted in a need for reducing/capping these emissions [1,3,4]. The magnitude of the CO<sub>2</sub> emissions calls for the application of a portfolio of strategies and technologies to approach the reduction/capping task, from which converting the otherwise wasteful CO<sub>2</sub> into added value products including synthetic fuels, chemicals, and raw materials is particularly interesting assuming that the needed energy comes from renewables.

As a mitigation strategy, using  $CO_2$  as a feedstock is based on the possibility of storing this compound in manufactured products and therefore preventing it from being released into the atmosphere. Functionality, abundance, renewable nature, and environmental friendliness are some of the features that make  $CO_2$  an attractive  $C_1$  building block for synthesis products such as paraffin,

dimethyl ether, methanol, higher alcohols, and formic acid [3,4]. Nowadays, CO<sub>2</sub> is employed in the production of urea, salicylic acid, and polycarbonates at an industrial scale [5], consuming altogether ~10 Mt of CO<sub>2</sub>  $y^{-1}$  as of 2007 [6]. However small this rate may seem compared to the ~35 BMt of CO<sub>2</sub>  $y^{-1}$  emitted in 2013, it only accounts for those few applications which have commercial status and hence represents a minute portion of its real mitigating potential. One application which could be deployed at a scale that would significantly contribute to the reduction of CO<sub>2</sub> emissions is the production of synthetic fuels through direct hydrogenation [3]. Holding great potential among the direct conversion processes is the Sabatier reaction (1):

$$\mathrm{CO}_2 + 4\mathrm{H}_2 \to \mathrm{CH}_4 + \mathrm{H}_2\mathrm{O} \tag{1}$$

The main advantage behind the Sabatier reaction is its favourable thermodynamics and theoretical capability to reach high conversions at ambient conditions [7]. However, the stability of the fully oxidised CO<sub>2</sub> molecule and the eight-electron process its reduction entails set considerable kinetic limitations that hinder the performance of the process at mild operating conditions [3,4,7]. Thus, using a highly efficient catalyst is required to fully exploit the potential of the Sabatier reaction. Most of the research conducted in this field has dealt with the use of Ni-based catalysts [3,4,7–11], although other transition (e.g., Fe and Co) [9,12] and noble (e.g., Rh, Pt, Ru and Pd) [12–16] metal based systems have also proven active.

While more commonly associated with H<sub>2</sub> storage applications, bulk LaNi<sub>5</sub> has exhibited promising activity/selectivity for the Sabatier reaction [17–19]. For instance Ando et al. achieved CO<sub>2</sub> conversions of 92% and methane selectivity of 98% (achieved after a 5 h induction period) for bulk LaNi<sub>5</sub> (prepared by arc-melting) at 250 °C with a high reactants pressure of 5 MPa [18]. The bulk LaNi<sub>5</sub> was far superior to a mixed Ni<sub>5</sub>LaO<sub>x</sub> oxide, prepared by co-precipitation, which exhibited 8% CO<sub>2</sub> conversion after 5 h under the same reaction conditions.

The work reported here takes LaNi<sub>5</sub> and adapts it to a metal oxide-supported ( $\gamma$ -Al<sub>2</sub>O<sub>3</sub>) catalyst system where its performance for the Sabatier reaction is assessed. Changes in the La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst structure invoked by the initial activation step and the ensuing reaction process were examined in detail. In-situ DRIFTS analyses were performed to establish the nature of the surface species generated over the course of the reaction and provide insights into the Sabatier mechanism.

## 2. Results and Discussion

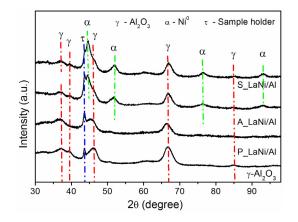
#### 2.1. As-Prepared Catalyst (P\_LaNi/Al)

ICP–OES gave the La-Ni loading and La/Ni molar ratio in P\_LaNi/Al to be 18.3 and 4.5 wt %, respectively. The XRD pattern of P\_LaNi/Al, as shown in Figure 1, exhibited reflections similar to those found in the pristine support, which corroborated the presence of poorly crystalline  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (JCPDS 086-1410) and revealed a signal (centred at 20 = 43.6°) corresponding to the stainless-steel (JCPDS 033-0397) sample holders used in the analyses. No additional features were observed in the XRD pattern of the P\_LaNi/Al sample, where the absence of La- and Ni-related reflections suggested that these components were present either as amorphous phases or as small crystalline structures indiscernible by this technique.

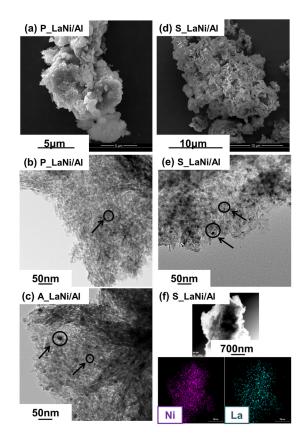
Considering the aforementioned, La was assumed to be present as highly-dispersed, amorphous  $La_2O_3$  at the surface of the P\_LaNi/Al. In contrast, nickel in P\_LaNi/Al was presumed to have formed Ni-Al surface spinels since the occurrence of such species in Ni/Al systems is widely known [19–24]. According to Li et al. the absence of XRD lines corresponding to these surface spinels results from their short-range three-dimensional order, provided that Ni ions only diffused into the first few outer layers of the Al<sub>2</sub>O<sub>3</sub> support during their 450 °C calcination stage [22].

SEM imaging of P\_LaNi/Al (Figure 2a) revealed an irregular surface comprising superimposed slab-like features and sub-micrometric aggregates (ca. 200–600 nm) which were evenly distributed across the micron-sized catalyst particles. TEM analysis of P\_LaNi/Al (Figure 2b) revealed the presence

of small, evenly distributed particles with a diameter ranging from 1 to 6 nm (Table 1), which were discernible from the prevailing  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support by their darker tone. These particles likely correspond to La- and Ni-containing deposits, but their exact make-up could not be defined.



**Figure 1.** XRD patterns of the pristine  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and the La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst as-prepared via SCS (solution combustion synthesis) (P\_LaNi/Al), reduced under H<sub>2</sub> at 700 °C (A\_LaNi/Al), and after 3 h of CO<sub>2</sub> hydrogenation at 400 °C (S\_LaNi/Al).

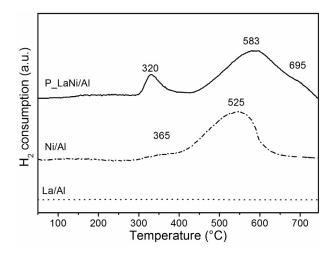


**Figure 2.** (a) SEM; and (b) TEM images of the La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst as-prepared via SCS (P\_LaNi/Al); (c) reduced under H<sub>2</sub> at 700 °C (A\_LaNi/Al); and (d) SEM; (e) TEM images and (f) associated elemental mapping after 3 h of CO<sub>2</sub> hydrogenation at 400 °C (S\_LaNi/Al). Black arrows indicate La- and/or Ni particles.

Materials	Particle Size (nm)	Ni Crystallite Size (nm)		
	TEM	XRD	H <sub>2</sub> Pulse (% Disp.)	
P_LaNi/Al	1–6	-	-	
A_LaNi/Al	4-20	5	37 (2.58)	
S_LaNi/Al	4–9	6	-	
Ni/Al	-	-	41 (2.36)	

**Table 1.** Average particle and Ni crystallite sizes (including degree of Ni dispersion) as estimated from TEM, XRD and pulsed H<sub>2</sub> chemisorption for the P\_LaNi/Al, A\_LaNi/Al, and S\_LaNi/Al materials.

The H<sub>2</sub>-TPR profile of P\_LaNi/Al (Figure 3) revealed the presence of three reduction bands within two major reduction zones at temperatures across the range of 290–750 °C. The low-temperature zone between 290 and 425 °C contained a symmetrical band centred at 320 °C attributed to the reduction of amorphous NiO on the catalyst surface [22,25–27] since no nitrates from the SCS were expected to remain after calcination at 500 °C [28]. The peak at 583 °C was ascribed to octahedrally coordinated Ni<sup>2+</sup> species (i.e., Ni<sup>2+</sup><sub>OC</sub>) embedded in the crystalline structure of the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support [24,27,28], which concurs with the formation of surface Ni-Al spinels. The small shoulder at 695 °C was assigned to the reduction of Ni<sup>2+</sup> species, either tetrahedrally coordinated (i.e., Ni<sup>2+</sup><sub>T</sub>) in the crystalline structure of the support as per previous reports [22,27], or in La–Ni–Al containing species derived from the strong interaction between La and Ni ions [26]. The former interpretation stems from the widely accepted perception that Ni<sup>2+</sup><sub>T</sub> species are more difficult to reduce than Ni<sup>2+</sup><sub>OC</sub> ones [23,27], while the latter coincides with experimental observations from studies dealing with the promotion effects of La on Ni-based catalysts [26].

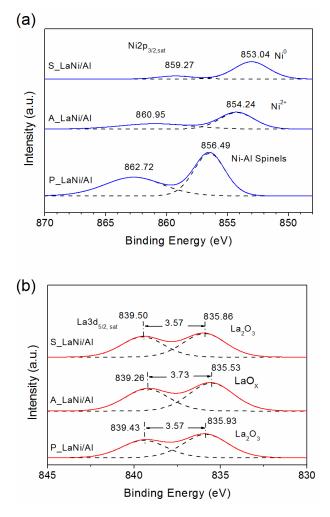


**Figure 3.** H<sub>2</sub>-TPR profiles of La deposited on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (La/Al), Ni deposited on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (Ni/Al), and the as-prepared La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (P\_LaNi/Al).

Additional H<sub>2</sub>-TPR analyses of Ni/Al (12.5 wt %) and La/Al (6.4 wt %) were conducted to assist with interpretation of the P\_LaNi/Al profile (Figure 3). No reducible phases were observed in the La/Al sample, agreeing with the known irreducible nature of La oxides below 1000 °C [26,29]. However, the Ni/Al sample exhibited two reduction peaks including a broad shoulder-like feature at 365 °C and a peak centred at 583 °C. Similar to P\_LaNi/Al, the low-temperature feature was attributed to surface NiO while the higher temperature peak was attributed to Ni<sup>2</sup><sub>OC</sub> species. The absence of a higher temperature shoulder, in conjunction with the absence of La in Ni/Al, infers that the 695 °C shoulder in the P\_LaNi/Al H<sub>2</sub>-TPR profile originates from the reduction of La–Ni–Al species and not from Ni<sup>2</sup><sub>T</sub> species. Additionally, the more prominent bulk NiO peak exhibited by the P\_LaNi/Al sample indicates that La promoted the formation of the bulk-like NiO on the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support.

The XPS spectrum of the P\_LaNi/Al sample was examined at the overlapping Ni2p/La3d, C1s, Al2p, and O1s regions to establish the surface chemical characteristics of the as-synthesised material. Analysis of the C1s region of the P\_LaNi/Al spectra gave a C/Al atomic ratio of 0.11 (Supplementary Materials, Figure S1; Table S1), indicating that no substantial residual carbon remained on the catalyst surface after synthesis.

Peaks relating to the Ni and La species were extracted from the de-convoluted spectra (Supplementary Materials, Figure S2) and are shown in Figure 4a,b, respectively, with values for the relative amounts of each species provided in Table 2. Within Figure 4a the peaks at 856.49 and 862.72 eV for P\_LaNi/Al correspond to the primary and shake-up lines of octahedrally coordinated Ni (Ni<sup>2+</sup><sub>OC</sub>) in nickel-alumina spinels, respectively [21,23,27,30]. According to Poncelet et al., enhanced reducibility in Ni/Al observed at high Ni loadings is associated with octahedral nickel (Ni<sup>2+</sup><sub>OC</sub>) species which resemble NiO [23]. These species would account for the low temperature reduction peak in the P\_LaNi/Al TPR profile, provided that no bulk NiO signals (i.e., ca. 854.24 eV) [26] were observed from the XPS spectru. No signals attributable to Ni<sup>2+</sup><sub>T</sub> (i.e., at 858.1 eV) [23] were observed in the P\_LaNi/Al XPS spectra, corroborating the H<sub>2</sub>-TPR findings which suggested suppression of these species in the presence of La. The signals observed at 835.86 and 839.43 eV are attributed to the main and satellite peaks of La<sup>3+</sup> from La<sub>2</sub>O<sub>3</sub> dispersed on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> [18,26,31].



**Figure 4.** Deconvoluted XPS spectra for (**a**) Ni and (**b**) La species on the surface of the P\_LaNi/Al, A\_LaNi/Al, and S\_LaNi/Al materials.

Materials	Ni2p <sub>3/2</sub> (Sat)			La3p <sub>5/2</sub> (Sat)		
	BE (eV)	Species	Ni/Al	BE (eV)	Species	La/Al
P_LaNi/Al	856.49 862.72	Ni-Al surface spinels	0.081	835.86 839.43	La <sub>2</sub> O <sub>3</sub>	0.022
A_LaNi/Al	854.24 860.95	Ni <sup>2+</sup>	0.028	835.53 839.26	LaO <sub>x</sub>	0.022
S_LaNi/Al	853.04 859.27	Metallic Ni	0.029	835.93 839.50	La <sub>2</sub> O <sub>3</sub>	0.020

**Table 2.** Binding energies (BE) of core electrons, surface species and relative atomic ratios for Ni and La in the P\_LaNi/Al, A\_LaNi/Al, and S\_LaNi/Al materials.

The Al2p and O1s regions of the P\_LaNi/Al XPS spectrum (Supplementary Materials, Figure S3) exhibited single peaks centred at 75.17 and 531.75 eV, respectively, which corresponded to the reported binding energies for  $Al_2O_3$  [26,32]. These results diverge from the 74.0 eV (Al2p) and 830.8 eV (O1s) expected for bulk NiAl<sub>2</sub>O<sub>4</sub> [33], implying the support surface remained essentially as alumina.

## 2.2. Reduced Catalyst (A\_LaNi/Al)

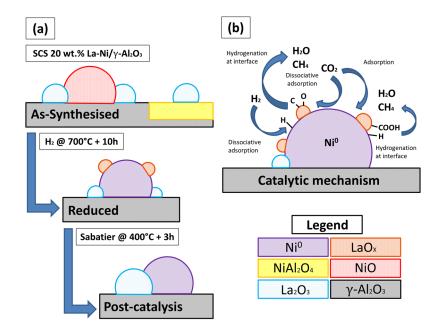
The XRD pattern of the reduced catalyst (Figure 1) exhibited diffraction lines corresponding to the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support and the stainless-steel sample holder used for the analysis. Typical metallic nickel (JCPDS 001–1260) diffraction peaks were also observed with the Ni crystallite size estimated to be of 5 nm from the Scherrer equation (Table 1). No distinct LaNi<sub>5</sub> diffraction lines were observed for the A\_LaNi/Al material as LaNi<sub>5</sub> formation from the corresponding metallic oxides has been reported to occur at temperatures >1000 °C [29,34,35].

SEM imaging of the A\_LaNi/Al revealed a morphology akin to that of P\_LaNi/Al. Clearly distinguishable nanoparticles were observable by TEM (Figure 2c), which were uniformly scattered across the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support and had diameters ranging between 4 and 20 nm. The majority of the deposits were observed to have diameters at the lower end of the size range, coinciding with the average Ni crystallite size determined from XRD. EDS elemental mapping (Supplementary Materials Figure S4) revealed even La and Ni dispersions across the catalyst surface, which suggests that the deposits comprise both Ni and La.

From the H<sub>2</sub>-TPR profiles (Figure 3), the conditions used to activate the catalyst (700 °C under pure H<sub>2</sub>) were expected to reduce all the nickel species present on Ni/Al and the majority of the nickel species on P\_LaNi/Al. However, a minor portion of the nickel in the form of the La–Ni–Al component on P\_LaNi/Al may have remained in an oxidised state. The average crystallite size of Ni and dispersion from the pulsed H<sub>2</sub> adsorption experiments for the reduced catalysts are given in Table 1. Both the A\_LaNi/Al and Ni/Al samples exhibited a relatively low degree of Ni dispersion (<2.6%) as can occur for catalysts with a high Ni content [26]. A slightly higher Ni dispersion and concomitant lower crystallite size were found for the A\_LaNi/Al sample compared to the Ni/Al sample, providing additional evidence of the La having some impact in the interaction between the Ni and the support.

Considerable disagreement was found between the average crystallite size estimated via pulsed  $H_2$  chemisorption and the value derived from XRD (Table 1). The wide variation in average crystallite size cannot be solely attributed to uncertainties within the estimation techniques. A possible explanation for the difference may be via the occurrence of a Strong Metal Support Interaction-like (SMSI-like) effect between the dispersed La and Ni particles. That is, partially reduced LaO<sub>x</sub> species, originating from the well-dispersed La<sub>2</sub>O<sub>3</sub> on the support, may begin to decorate the surface of metallic Ni crystallites during the reduction at 700 °C, hindering H<sub>2</sub> adsorption at the blocked surface Ni sites and distorting the dispersion calculations. A depiction of the proposed SMSI-like effect is shown in Scheme 1a. An analogous SMSI-like effect was proposed by Zhang et al. who reported similar

discrepancies in Ni crystallite size estimation for a Ni/La<sub>2</sub>O<sub>3</sub> catalyst reduced under H<sub>2</sub> at 500 °C [36]. Sánchez-Sánchez et al. also hinted at the decoration of metallic particles by dispersed species upon reducing a La-promoted Ni/Al catalyst at 700 °C [26].



**Scheme 1.** Representation of the: (a) evolution of the La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst structure driven by SMSI-like (Strong Metal Support Interaction-like) effects promoted by La<sub>2</sub>O<sub>3</sub>; and (b) proposed catalytic reaction mechanism for CO<sub>2</sub> hydrogenation into CH<sub>4</sub>.

Deconvoluted peaks for the Ni2p and La3d region of the A\_LaNi/Al XPS spectrum are depicted in Figure 4a,b, respectively, with related information summarised in Table 2. Both the main and satellite Ni2p peaks were present at 854.24 and 860.95 eV, indicating the presence Ni<sup>2+</sup> despite the catalyst undergoing a prior reduction step [21,26,27,30]. The presence of Ni<sup>2+</sup> may derive from either: (i) La–O–Ni species at the interface between Ni crystallites and the LaO<sub>x</sub> decoration (potentially formed during the reduction process); or (ii) surface NiO formed during re-oxidation of the metallic Ni during sample transfer to the XPS equipment.

The La3d signal for A\_LaNi/Al appeared at 835.53 eV, representing a small shift towards lower binding energies (compared to 835.9 eV for P\_LaNi/Al) but remaining within the range acceptable for surface La<sub>2</sub>O<sub>3</sub>. The 0.33 eV shift towards lower binding energies supports the notion of LaO<sub>x</sub> decorating the Ni deposits as a 0.2 eV shift was reported by Fleisch et al. in a Pd/La<sub>2</sub>O<sub>3</sub> system exhibiting SMSI effects [37]. Additionally, the increase in the La3d<sub>5/2</sub> satellite-split between the P\_LaNi/Al and A\_LaNi/Al samples (i.e., 3.57 to 3.73 eV) may endorse the proposed SMSI-like effect as they observed an analogous increase (i.e., from 3.3 to 3.8 eV) after reduction at 500 °C [37].

As shown in Table 2, a decrease in relative Ni/Al atomic ratio between the A\_LaNi/Al and P\_LaNi/Al samples indicates a loss in exposed Ni during the reduction at 700 °C. In contrast, no variation was seen for the relative La/Al ratio between the two materials signifying that La remained exposed during the reduction step. This loss of exposed Ni and the retention of exposed La further support the perception that La is decorating the Ni deposits during the reduction process.

## 2.3. Spent Catalyst (S\_LaNi/Al)

The XRD pattern of the S\_LaNi/Al (Figure 1) exhibited the reflections associated with the catalyst alumina support and the stainless steel sample holders. Metallic Ni reflections were also observed in the S\_LaNi/Al XRD pattern, with a small relative increase in signal intensity being discernible

compared to the A\_LaNi/Al counterpart. The increase in Ni diffraction peaks intensity may correspond to further reduction of remaining NiO during the catalytic process.

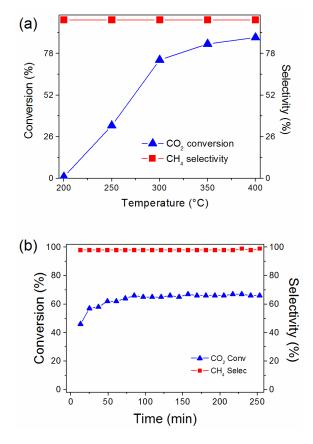
SEM imaging of the S\_LaNi/Al sample (Figure 2d) revealed a surface mainly composed of aggregates (150–600 nm) and flake-like components evenly distributed across the catalyst particles. TEM micrographs (Figure 2e) revealed the presence of clearly discernible La/Ni particles with diameters between 4 and 9 nm (Table 1), uniformly distributed across the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support. None of the larger Ni deposit diameters (in the order of 20 nm) observed for A\_LaNi/Al (Figure 2c) were readily observed for S\_LaNi/Al suggesting a redistribution of the larger Ni deposits occurred during the reaction. EDS mapping (Figure 2f) confirmed that both La and Ni remained well-dispersed across the catalyst particle following reaction. No carbonaceous deposits were observed by TEM/SEM and associated EDS elemental mapping.

The impact of the Sabatier reaction on the Ni and La surface species, from XPS, is provided in Figure 4a,b, respectively, with associated details provided in Table 2. The Ni2p primary and satellite peaks for S\_LaNi/Al occurred at 853.04 and 859.27 eV, respectively, which represented a shift towards lower binding energies compared to A\_LaNi/Al and revealed the presence of surface metallic Ni [21,26]. No peaks corresponding to Ni<sup>2+</sup> species were found for S\_LaNi/Al, indicating that in-situ reduction of these species occurred during the course of the reaction. Abelló et al. reported a similar in-situ reduction in their study of a mixed-oxide  $NiO/Al_2O_3$  catalyst for the Sabatier reaction [38]. They concluded that the Ni<sup>2+</sup> that was reduced during reaction corresponded to NiO which had remained unreduced after the H<sub>2</sub> activation stage owing a certain level of sintering. In the case of  $S_LaNi/Al$ , though, the Ni<sup>2+</sup> reduced in-situ is suspected to either comes from the NiO surface layer on the deposits and/or the possible La–O–Ni species observed in A\_LaNi/Al. Reduction of surface NiO is supported by the small increase in intensity observed by XRD (Figure 1). The alternate hypothesis of La–O–Ni reduction is supported by a shift in the La3d peaks to 835.86 and 839.50 eV (Figure 4b), reverting to values similar to those observed for the P\_LaNi/Al. This includes the satellite split which returned to a value of 3.57 eV. Accordingly, these results indicate that La is present as La<sub>2</sub>O<sub>3</sub> after the reaction and provide evidence towards the disintegration of the proposed  $LaO_x$  decorations during reaction in agreement with the re-dispersion of the larger Ni particles observed by TEM (Figure 2b,c,e). A small decrease in the La/Al atomic ratio is also noted for S\_LaNi/Al when compared to A\_LaNi/Al (0.022 to 0.020, Table 2) and this also corroborates with the hypothesis of a disintegration of the  $LaO_x$ decorations, as previously hypothesized. No major changes were observed in the binding energies of the S\_LaNi/Al XPS C1s components compared to those observed in P\_LaNi/Al and A\_LaNi/Al, (Supplementary Materials, Table S1). A depiction of the SMSI-like effect breakup leading to the re-dispersion of La and Ni at the surface of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is shown in Scheme 1a.

## 2.4. Catalyst Performance

Performance of the A\_LaNi/Al catalyst for the Sabatier reaction over the range 200–400 °C is shown in Figure 5a in terms of CO<sub>2</sub> conversion and CH<sub>4</sub> selectivity. CO<sub>2</sub> conversion increased from <3% at 200 °C to ~80% at 350 °C. Selectivity towards CH<sub>4</sub> remained constant across the entire temperature range at values >98%. CO was the only secondary product detected from the hydrogenation reaction. Stability tests on A\_LaNi/Al were conducted at 275 °C (Figure 5b) whereby, once stabilised, the CO<sub>2</sub> conversion remained steady at 65% with 98% selectivity towards CH<sub>4</sub> (pressure = 1 atm, H<sub>2</sub>:CO<sub>2</sub> ratio = 4:1, GHSV = 7.2 × 10<sup>4</sup> mL·h<sup>-1</sup>·g<sup>-1</sup> cat., 20 wt % Ni loading). By comparison: (i) Ando et al. reported a 56% CO<sub>2</sub> conversion with 98% CH<sub>4</sub> selectivity at 300 °C for a neat LaNi<sub>5</sub> catalyst prepared by arc melting (pressure = 1 atm, H<sub>2</sub>:CO<sub>2</sub> ratio = 4:1, WHSV = 3000 mL·h<sup>-1</sup>·g<sup>-1</sup> cat., 15 wt % Ni loading) [39]; and (iii) Abelló et al. reported a 8% CO<sub>2</sub> conversion with 95% CH<sub>4</sub> selectivity at 300 °C for a Ni<sub>6</sub>Al (reduced) catalyst prepared by co-precipitation (pressure = 5 atm, H<sub>2</sub>:CO<sub>2</sub> ratio = 4:1, WHSV = 26.88 × 10<sup>5</sup> mL·h<sup>-1</sup>·g<sup>-1</sup> cat., 69.1 wt % Ni loading) [38]. The stability profile also showed

that an initial induction period (~90 min) was needed prior to the catalyst reaching a stable and maximum activity. Although no direct evidence is available at this time, the induction period may account for the transformation of the LaO<sub>x</sub>-decorated NiO into the reduced Ni/La<sub>2</sub>O<sub>3</sub> form, which could be responsible for the higher catalyst activity. Irrespective of this and to our best knowledge, the P\_LaNi/Al catalyst shows good performances as compared to other Ni-Al-based catalysts at a moderate temperature and ambient pressure.



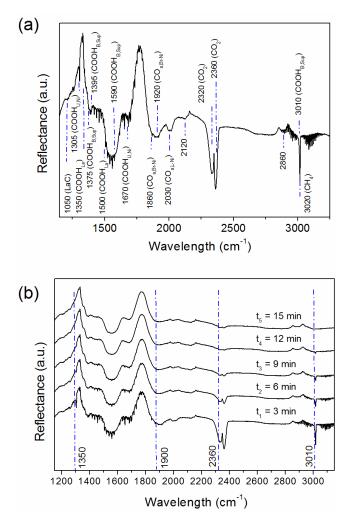
**Figure 5.** CO<sub>2</sub> conversion and CH<sub>4</sub> selectivity for the Sabatier reaction of A\_LaNi/Al at: (a) temperatures ranging between 200 and 400 °C; and (b) at a constant temperature of 275 °C for 5 h.

The performance of the La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst prepared via SCS is postulated to derive from a combination of the SMSI-like effect and the physical-chemical promotional impact of La itself. According to the former, the formation of LaO<sub>x</sub> decorations provided the catalytic system with La–O–Ni at the interface with the metallic Ni crystallites, which have been proposed to be highly active for the Sabatier reaction [40]. On the other hand, the promoting effects of La would be responsible for quicker/improved CO<sub>2</sub> dissociative adsorption due to its higher basicity [17,41–43] and better dispersion/stabilisation of the active Ni phase during preparation and reaction [27,44,45]. Interestingly, the post-reaction state of Ni and La in the catalyst differed to the activated.

#### 2.5. Reaction Routes

An example of in-situ DRIFTS analysis profile for the La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst immediately following a H<sub>2</sub> pulse is provided in Figure 6a. The spectrum displays multiple bands between 1050 and 3010 cm<sup>-1</sup>. The different species found at the surface of the catalyst during reaction include: La<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> (1050 cm<sup>-1</sup>) [46], unidentate formates on the Ni surface (1305 and 1607 cm<sup>-1</sup>) [47], formates on the La<sub>2</sub>O<sub>3</sub> surface (1350 and 1500 cm<sup>-1</sup>) [40], bidentate formates on the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support (1375, 1395, 1590, and 3010 cm<sup>-1</sup>) [47], adsorbed CO on Ni in bridge configuration (1860 and 1910 cm<sup>-1</sup>) [47,48],

adsorbed CO on Ni in linear configuration (2030 cm<sup>-1</sup>) [47,48], and adsorbed/gaseous CO<sub>2</sub> (2320 and 2360 cm<sup>-1</sup>). Conversion of CO<sub>2</sub> into CH<sub>4</sub> during reaction was corroborated by the presence of the characteristic multi-peak bands observed at ca. 3100 and 1400 cm<sup>-1</sup> with a strong peak at 3020 cm<sup>-1</sup> corresponding to the Q-band of gaseous CH<sub>4</sub>, while similar signals between 1200 and 2200 cm<sup>-1</sup> indicated the presence of water.



**Figure 6.** In-situ DRIFTS spectra of the A\_LaNi/Al catalyst immediately following pulsed H<sub>2</sub> injection at 400 °C and 0.1 MPa over pre-adsorbed CO<sub>2</sub>: (**a**) detailed account of the surface species present. LaC corresponds to La<sub>2</sub>O<sub>2</sub>CO<sub>3</sub>, COOH<sub>U,Ni</sub> to unidentate formate on Ni, COOH<sub>La</sub> to formate on La<sub>2</sub>O<sub>3</sub>, COOH<sub>B,Sup</sub> to bidentate formate on the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support, CO<sub>a,Br-Ni</sub> to adsorbed CO<sub>2</sub> on Ni in a bridged configuration, and CO<sub>a,L-Ni</sub> to adsorbed CO<sub>2</sub> on Ni in a linear configuration; and (**b**) evolution of surface species as a function of time. The spectra correspond to the raw data and the "negative" features, e.g., at 15 min, are related to the slight differences between the diamond powder background and the spectrum of the catalyst diluted in diamond powder.

The evolution of surface species during the Sabatier reaction (400 °C and 0.1 MPa) is displayed by the consecutive in-situ DRIFTS spectra in Figure 6b. The monitored reflectance bands were those corresponding to unidentate formates over Ni, adsorbed CO on Ni in bridged configuration, adsorbed/gaseous CO<sub>2</sub>, and bidentate formates on the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support. Results showed that gaseous/adsorbed CO<sub>2</sub> bands appeared right after the injection of H<sub>2</sub>, indicating that H<sub>2</sub> preferentially adsorbed onto the catalyst and released part of the pre-adsorbed CO<sub>2</sub>. Bands corresponding to water, methane, and formates on both Ni and the support were detected early within the reaction process (i.e., at  $t_1 = 3$  min). As reaction time elapsed, the higher-amplitude signal corresponding to CO adsorbed on Ni in bridge configuration (1860 cm<sup>-1</sup>) and the formates on the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support (3010 cm<sup>-1</sup>) gradually disappeared. The Ni-bound formates disappeared quicker than those bound to the support, and were indiscernible after 9 min ( $t_3$ ). The bands corresponding to the Ni adsorbed linear CO (2030 cm<sup>-1</sup>) and the lower-amplitude bridged CO (1910 cm<sup>-1</sup>), which were apparent 3 min after the first H<sub>2</sub> pulse ( $t_1$ ), were not present after 6 min ( $t_2$ ). After 15 min ( $t_5$ ), some CO<sub>2</sub>-related species remained adsorbed on the catalyst since the band at 2360 cm<sup>-1</sup> was still present. After a second H<sub>2</sub> injection, the whole cycle was repeated and the presence of CO<sub>2</sub>, remaining since the first H<sub>2</sub> injection, was corroborated.

On studying the hydrogenation of  $CO_2$  into  $CH_4$  over Ni/Al using FTIR and TPR, Fujita et al. identified different types of CO adsorbed on Ni in bridged position, proposing that the higher-amplitude band (i.e., 1860 cm<sup>-1</sup>) corresponded to a strongly bound CO species while the lower-amplitude band (i.e., 1910 cm<sup>-1</sup>) was related to weakly bound CO [47]. Additionally, formates present on the support and adsorbed CO species were presumed to undergo hydrogenation into CH<sub>4</sub> through a mechanism which involved further dissociation into adsorbed carbon, allowing for the amounts of adsorbed carbon and linear CO species to be negligible during reaction.

The results from in-situ DRIFTS appear to corroborate the presence and hydrogenation of formates and adsorbed CO strongly bound in a bridged configuration. Based on the findings by Fujita et al. [47] and accounting for the possible effects of  $LaO_x$  decorations on the catalyst as found in this work, a mechanism is proposed for the Sabatier reaction as follows: (a) dissociative adsorption of H<sub>2</sub> into H<sub>atomic</sub> on Ni sites; (b) adsorption and hydrogen assisted cleavage of CO<sub>2</sub> on Ni sites; (c) dissociative adsorption of CO<sub>2</sub> onto  $LaO_x$  decorations; (d) cleavage of strongly adsorbed CO into adsorbed O and C; (e) hydrogenation of  $LaO_x$ -adsorbed CO over highly active La-O-Ni at the interface with the Ni crystallites; and (f) instantaneous hydrogenation of adsorbed C and O from adsorbed CO and formates into CH<sub>4</sub> and H<sub>2</sub>O, respectively. The final step is consistent with the negligible carbon deposition on the S\_LaNi/Al sample from XPS and TEM/SEM. A depiction of the previous reaction steps and the catalytic system is provided in Scheme 1b, where Ni- and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-bound formates are represented since in-situ DRIFTS revealed their presence.

## 3. Materials and Methods

#### 3.1. Catalyst Preparation

The Ni-La/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst was prepared via solution combustion synthesis (SCS) using the corresponding metallic nitrates as oxidisers and glycine as the fuel/complexing agent [40]. A nickel nitrate (Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O, Ajax Finechem Pty. Ltd., Sydney, Australia), lanthanum nitrate (La(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O, Merck KGaA, Sydney, Australia), and glycine (C<sub>2</sub>H<sub>5</sub>O<sub>2</sub>N, Ajax Finechem Pty. Ltd., Sydney, Australia) solution was prepared at a fixed 1:5:6 molar ratio using deionised water as the solvent and keeping the overall combusting mixture under 500 mg. Once the nitrates and glycine had been dissolved, sufficient  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (particle size 45–53 µm, Saint-Gobain NorPro, Japan) was added to the solution to give a nominal 20 wt % Ni-La loading on the final catalyst. The resulting slurry was stirred for 30 min and allowed to settle for a further 30 min after which it was transferred into a ceramic crucible and placed in a temperature controlled (±1 °C, FP21 Shimade Co., Ltd., Tokyo, Japan) muffle furnace. The SCS mixture was heated in the presence of air at 5 °C·min<sup>-1</sup> from room temperature to 500 °C, calcined for 1 h, and then allowed to cool to room temperature. Control samples comprising 6.4 wt % La on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (La/Al) and 12.5 wt % Ni on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (Ni/Al) were also prepared using SCS.

#### 3.2. Catalyst Characterisation

The La-Ni/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst was characterised following preparation (P\_LaNi/Al), activation (A\_LaNi/Al) and post-reaction (i.e., spent following the Sabatier reaction at 400 °C for 3 h, S\_LaNi/Al). P\_LaNi/Al composition was determined by ICP-OES on a Perkin Elmer OPTIMA 7300 (Perkin Elmer, Sydney, Australia). Prior to analysis a small amount of the material was digested in concentrated

nitric acid (70 vol % HNO<sub>3</sub>, Ajax Chem, Sydney, Australia) for 8 h at 80 °C and then diluted with de-ionised water.

Crystallinity of P\_LaNi/Al, A\_LaNi/Al and S\_LaNi/Al was studied by X-ray Diffraction (XRD), using a Philips PANalytical X'pert Multipurpose XRD System (Phillips, Sydney, Australia) equipped with a monochromated Cu-K $\alpha$  source ( $\lambda = 1.5418$  Å), and operating at 45 kV and 40 mA.

Hydrogen temperature programmed reduction (H<sub>2</sub>-TPR) analyses were performed on a Micromeritics Autochem II 2920 (Micromeritics, Sydney, Australia) to assess the reducibility of P\_LaNi/Al catalyst as well as the control materials. Materials were pre-treated by flowing argon through the bed at 120 °C for 30 min and then cooled to 50 °C. Then, 10 vol % H<sub>2</sub>/Ar was passed over the catalyst bed while the temperature was increased at 5 °C·min<sup>-1</sup> from 50 to 800 °C.

Pulsed H<sub>2</sub> chemisorption assays were conducted to estimate the Ni dispersion and crystallite size for P\_LaNi/Al and Ni/Al using a Micromeritics Autochem II 2920. The materials were initially pre-treated by flowing argon through the bed at 120 °C for 30 min after which they were cooled to room temperature. Pure hydrogen was passed through the sample bed as the temperature was increased at 10 °C·min<sup>-1</sup> up to 700 °C where it was held for 5 h. The system was subsequently flushed with argon for 2 h at 700 °C and cooled to room temperature. Consecutive volumetric pulses (0.4664 mL) of 10 vol % H<sub>2</sub>/Ar were then dosed to the sample every 1.5 min. Ni dispersion and crystallite size were estimated assuming an atomic ratio of 1 between adsorbed atomic hydrogen and surface Ni at room temperature [20,36,49] and by following the procedure proposed by Bartholomew et al. [50].

Scanning Electron Microscopy (SEM) analyses were carried out using a FEI Nova Nano-SEM 230 FESEM instrument (FEI, Sydney, Australia), while Transmission Electron Microscopy (TEM) and energy-dispersive X-ray spectroscopy (EDS) studies were performed on a Philips CM200 FEG TEM (Phillips, Sydney, Australia) equipped with an interfaced EDAX system (Phillips, Sydney, Australia) and operated at 200 keV.

Chemical characteristics and relative atomic compositions of the P\_LaNi/Al, A\_LaNi/Al and S\_LaNi/Al catalyst surfaces were studied by XPS using a Thermo ESCALAB250i XPS instrument (Thermo Scientific, Sydney, Australia) equipped with a monochromated Al-K $\alpha$  X-ray source and powered at 13 kV/12 mA. Adventitious hydrocarbon, C1s (285.0 eV), was used as the binding energy (BE) reference. Air sensitive samples were prepared in a glove box (LC Technology, Salisbury, MD, USA) under high purity argon immediately prior to XPS analysis to minimise the potential of oxidation.

#### 3.3. Catalytic Assessment

Catalytic activity tests for the Sabatier reaction were conducted in a continuous-flow fixed-bed reactor (6 mm ID quartz tube, (CGB, Sydney, Australia)) equipped with both temperature and mass flow controllers (H<sub>2</sub> and N<sub>2</sub>: Brooks Sydney, Australia; CO<sub>2</sub>: Aalborg, Sydney, Australia). Prior to each catalytic test, 0.05 g of the catalyst was reduced in-situ for 3 h using 50% hydrogen in nitrogen at 60 mL·min<sup>-1</sup> and 700 °C. A stoichiometric mixture of H<sub>2</sub>/CO<sub>2</sub> (i.e., 4:1 molar ratio) was fed into the system at a  $7.2 \times 10^4$  mL·h<sup>-1</sup>·g<sup>-1</sup> cat. GHSV, with hydrogen and carbon dioxide (Coregas, Sydney, Australia) flowing at 40 and 10 mL·min<sup>-1</sup>·g<sup>-1</sup> cat. STP, respectively. Reaction temperatures were varied between 150 and 450 °C. The product gas was analysed with an on-line gas chromatograph (Shimadzu GC-2010 Plus, Shimadzu, Sydney, Australia) equipped with a TCD and Agilent CarboPlot P7 column (Agilent, Sydney, Australia), and an FID and methaniser, and used argon (Coregas, Sydney, Australia) as the carrier gas. The reaction was then stopped, the reactor cooled under flowing H<sub>2</sub>/CO<sub>2</sub>, and the spent catalyst samples collected/stored in sealed Ar-filled vials for further analysis. Catalyst stability was also assessed by running the reaction under the conditions described above at 275 °C for 5 h.

The evolution of surface species during CO<sub>2</sub> hydrogenation was followed using in-situ Diffuse Reflectance Infrared Fourier Transform Spectroscopy (DRIFTS) (Harrick, New York, NY, USA). Prior to measurements, the catalyst was activated in-situ as per conditions above. Spectra were collected using a Brüker VERTEX 70v FTIR spectrometer (Bruker, Sydney, Australia) and equipped with a commercial in-situ DRIFTS cell (Harrick's Scientific, Ithaca, NY, USA). For each analysis, approximately 20 mg of the activated catalyst (A\_LaNi/Al) was diluted with synthetic mono-crystalline diamond powder (~1  $\mu$ m, Sigma-Aldrich, Sydney, Australia). Operating conditions within the sample cell were set at 400 °C and 0.1 MPa. Prior to analysis, the cell was flushed with argon for 30 min at 150 °C. Carbon dioxide was then fed for 30 min into the reaction chamber at 400 °C and allowed to adsorb onto the activated catalyst. The pre-adsorbed CO<sub>2</sub> was reacted with H<sub>2</sub> by dosing ~20 mL injections every 3 min and the evolution of surface species during reaction was followed.

# 4. Conclusions

A LaNi/γ-Al<sub>2</sub>O<sub>3</sub> catalyst, synthesised using SCS, exhibited good activity for the Sabatier reaction at atmospheric pressure, reaching CO<sub>2</sub> conversions of ~80% at temperatures of 350 °C and beyond and a CH<sub>4</sub> selectivity consistently >98%. Characterisation of the LaNi/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst at various stages of the reaction process (i.e., as-prepared, activated and post-reaction) demonstrated the La and Ni existed in different forms at each stage. La and Ni in the as-prepared catalyst were observed to be present as amorphous surface La-Ni spinels. The activation process did not reduce the Ni as anticipated, rather it appeared to promote the formation of oxidised Ni particles (i.e.,  $Ni^{2+}$ ) decorated with  $LaO_x$  moieties. The reaction conditions led to the in-situ reduction of the Ni (to give metallic Ni deposits) with the LaO<sub>x</sub> species returning to an oxidation state (La<sub>2</sub>O<sub>3</sub>) similar to that observed for the as-prepared catalyst. At the interface between  $LaO_x$  decorations and the Ni crystallites, La–O–Ni species may be present which could potentially be accountable for the catalyst activity, although the change in catalyst characteristics during the reaction in conjunction with the stable activity of the catalyst imply that the activated catalyst may be an intermediary state of the ultimately active material. In-situ DRIFTS analyses revealed the presence of adsorbed formates and adsorbed CO species during reaction at 400 °C. The results suggested that the reaction advanced through a CO-based mechanism involving CO<sub>2</sub> hydrogenation, although the pathway involving the hydrogenation of formates cannot be excluded. The findings demonstrate the capacity for hydrogen storage based materials, such as LaNi alloys, to behave as active catalysts for the Sabatier reaction in this instance, and potentially for other hydrogenation reactions in general, when loaded on a metal oxide support.

**Supplementary Materials:** The following are available online at www.mdpi.com/2073-4344/6/11/170/s1, Figure S1: X-ray Photoelectron (XPS) spectra and curve fitting of P\_LaNi/Al, A\_LaNi/Al and S\_LaNi/Al at the C1s region, Figure S2: X-ray photoelectron spectra and curve fitting of P\_LaNi/Al, A\_LaNi/Al and S\_LaNi/Al at the Ni2p and La3d region, Figure S3: X-ray photoelectron spectra of the P\_LaNi/Al at the (a) O1s and (b) Al2p regions, Figure S4. TEM images and (f) associated elemental mapping of A\_LaNi/Al, Table S1: Carbon (C1s) binding energies (BEs) of core electrons, surface species and relative Ci/Al atomic ratio for the P\_LaNi/Al, A\_LaNi/Al.

**Acknowledgments:** The authors would also like to acknowledge the use of facilities within the UNSW Mark Wainwright Analytical Centre and Bill Bin Gong for assistance with the XPS analyses, also located within the UNSW Mark Wainwright Analytical Centre.

**Author Contributions:** Daniel E. Rivero-Mendoza carried out all the experimental work which was conceived and designed with Jason Scott and Kondo-François Aguey-Zinsou. Jessica N. G. Stanley, conducted the stability test. All contributed to the writing of the manuscript.

Conflicts of Interest: The authors declare no conflict of interest.

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