

## Article

# Oxygen-Vacancy-Rich Fe@Fe<sub>3</sub>O<sub>4</sub> Boosting Fenton Chemistry

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**Abstract:** Iron-based materials are widely applied in Fenton chemistry, and they have promising prospects in the processing of wastewater. The composition complexity and rich chemistry of iron and/or oxides, however, hamper the precise understanding of the active sites and the working mechanism, which still remain highly controversial. Herein, iron oxides of four different model systems are designed through a conventional precipitation method plus H<sub>2</sub> reduction treatment. These systems feature Fe@Fe<sub>3</sub>O<sub>4</sub> with abundant oxygen vacancy, Fe<sup>0</sup> and Fe<sub>3</sub>O<sub>4</sub> particles with interface structures, and Fe<sub>3</sub>O<sub>4</sub>-dominated nanoparticles of different sizes. These materials are applied in the decomposition of methyl orange as a model reaction to assess the Fenton chemistry. The Fe@Fe<sub>3</sub>O<sub>4</sub> with core-shell structures exhibits significantly higher decomposition activity than the other Fe<sub>3</sub>O<sub>4</sub>-rich nanoparticles. A thin Fe<sub>3</sub>O<sub>4</sub> layer formed by auto-oxidation of iron particles when exposed to air can boost the activity as compared with the Fe<sup>0</sup> and Fe<sub>3</sub>O<sub>4</sub> particles with interface structures but poor oxygen vacancy. The unique hetero-structure with the co-existence of both metallic iron and oxygen vacancy displays excellent redox propensity, which might account for the superior Fenton activity. This finding provides a new perspective to understand and design highly efficient iron-based Fenton catalysts.

**Keywords:** advanced oxidation process; core-shell structure; fenton chemistry; Fe@Fe<sub>3</sub>O<sub>4</sub> interface; methyl orange decomposition; oxygen vacancy



**Citation:** Zheng, R.; Tan, R.; Lv, Y.; Mou, X.; Qian, J.; Lin, R.; Fang, P.; Kan, W. Oxygen-Vacancy-Rich Fe@Fe<sub>3</sub>O<sub>4</sub> Boosting Fenton Chemistry. *Catalysts* **2023**, *13*, 1057. <https://doi.org/10.3390/catal13071057>

Academic Editor: Enric Brillas

Received: 25 May 2023

Revised: 19 June 2023

Accepted: 20 June 2023

Published: 30 June 2023



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## 1. Introduction

The widespread pollution of wastewater due to the discharge of high-concentration chemical solutions containing diverse organic compounds from the industry has posed a formidable challenge to a sustainable society [1–4]. The conventional approaches to wastewater management relying on biodegradation or physiochemical methods (such as the combination of chlorination and adsorption) are still not optimal for handling industrial wastewater. In this context, advanced oxidation processes (AOPs) are catching great attention as one of the most promising alternative solutions [5–8]. Compared with the traditional biodegradation method, which might be subjected to biomass poisoning, AOPs are a non-selective technology that depends on the oxidation of organic compounds with the in situ generated radical species with a strong oxidizing ability. Therefore, it is extremely flexible in handling the diversity of organic pollutants from the different chemical sectors.

As one of the most cost-effective AOP technologies, the Fenton process has drawn great interest in wastewater treatment [5,6]. The classic Fenton chemistry is initiated by

the formation of hydroxyl radicals ( $\text{HO}^*$ ) through the activation of  $\text{H}_2\text{O}_2$  by  $\text{Fe}^{2+}$  ions. The as-formed  $\text{HO}^*$ , together with other radicals with strong oxidizing potentials, can then convert the organic pollutants into smaller molecules or  $\text{CO}_2$  [6]. To further enhance the oxidation efficiency, similar Fenton-like processes, such as photo-Fenton [9–11], electro-Fenton [12], and sono-Fenton [13], have also been studied by different researchers. Despite the difference in various techniques, the development of highly active Fenton catalysts is of optimal importance. The classic Fenton reaction was performed by soluble iron salts as homogeneous catalysts conducted at an optimal pH of 3.0. In addition to the requirement of neutralization before the discharge, the precipitation and subsequent disposal of iron hydroxides are the major problems [8]. To overcome this disadvantage, different heterogeneous Fenton systems have been proposed, which can be roughly classified into two categories: (i) the iron-containing catalysts, including iron minerals, clay-based catalysts, and other iron-containing catalysts; and (ii) non-ferrous catalysts based on cerium [14], chromium [15], cobalt [16], copper [17], manganese [18], ruthenium [19], and polyoxometalates [20]. Despite extensive explorations, iron-based Fenton systems possess several advantages, such as the high reactivity of  $\text{Fe}^{2+}/\text{Fe}^{3+}$  [21], the abundance and low cost of the metal, and low toxicity and environmental compatibility, rendering them highly appealing.

Iron minerals, including magnetite ( $\text{Fe}_3\text{O}_4$ ) [22–24], hematite ( $\alpha\text{-Fe}_2\text{O}_3$ ) [25], maghemite ( $\gamma\text{-Fe}_2\text{O}_3$ ) [26], goethite ( $\alpha\text{-FeOOH}$ ) [27], akaganèite ( $\beta\text{-FeOOH}$ ) [28,29], lepidocrocite ( $\gamma\text{-FeOOH}$ ), etc., are among the most extensively studied Fenton systems. In addition, zero-valence iron (ZVI) [30–32] and hybridized structures (e.g.,  $\text{Fe@Fe}_2\text{O}_3$  [33–36] and  $\text{Fe}^{2+}/\text{Fe@Fe}_2\text{O}_3$  [37,38]) are also widely explored. Despite the different compositions and structures of these catalytic materials, it is generally accepted that  $\text{Fe}^{2+}$  species are crucial for the generation of  $\text{HO}^*$  [5–8]. However, highly controversial questions remain regarding the nature of the active sites for metallic iron and oxides. For instance, Yuan et al. [31] employed ZVI in methyl orange decomposition and found an activity drop in consecutive runs. Analysis of the recovered catalyst confirmed the formation of oxide layers that was suggested to be responsible for the activity degrading. On the contrary, recent studies showed that by combining ZVI with iron oxides such as  $\text{Fe@Fe}_2\text{O}_3$  core-shell structures are able to achieve excellent Fenton activity [33]. It has been proposed that  $\text{Fe}^0$  can transfer two electrons to  $\text{O}_2$  to generate  $\text{H}_2\text{O}_2$ , which further reacts with generated  $\text{Fe}^{2+}$  to produce  $\text{HO}^*$  under acidic to neutral pH conditions [36]. The above opposing opinions highlight the necessity of a more in-depth understanding of the  $\text{Fe}^0$  and iron oxide interfaces. Furthermore, ZVI is known to be susceptible to auto-oxidation when exposed to air. This means the outermost surfaces of the bulk metallic iron are likely covered by iron oxide layers. The nature of these layers and the impact on the Fenton activity are still not well understood.

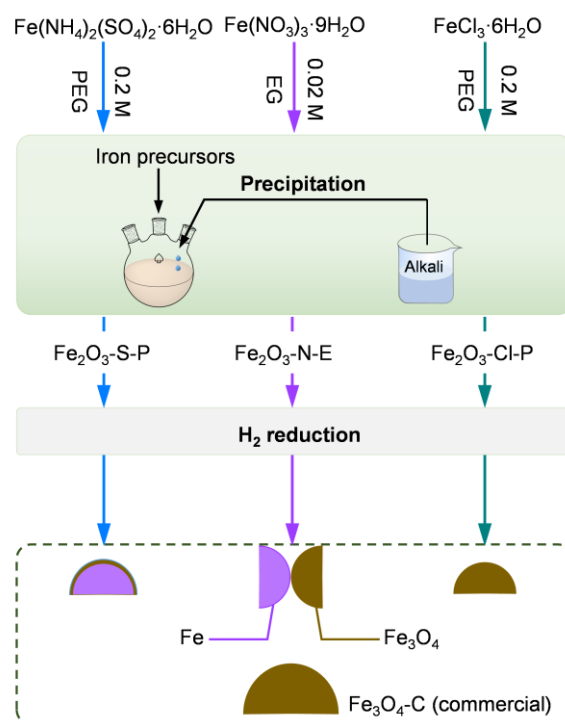
In this contribution, we synthesized different iron oxide-based model systems through a conventional precipitation method in conjunction with  $\text{H}_2$  reduction treatment. These systems showed very different structural features: (1)  $\text{Fe}^0$ -rich cores covered with  $\text{Fe}_3\text{O}_4$  layers ( $\text{Fe@Fe}_3\text{O}_4$ ), (2)  $\text{Fe}^0$  and  $\text{Fe}_3\text{O}_4$  nanoparticles with interface structures, and (3)  $\text{Fe}_3\text{O}_4$ -dominated nanoparticles of different sizes. These materials were then applied as the Fenton catalysts in the decomposition of methyl orange, which was selected as the model reaction to assess the catalytic activity in the removal of azo-dye pollutants from aqueous solutions [39]. Our catalytic results showed that the first two systems with the co-existence of both  $\text{Fe}^0$  and  $\text{Fe}_3\text{O}_4$  phases were significantly more active than the  $\text{Fe}_3\text{O}_4$ -dominated nanoparticles of different sizes, and the  $\text{Fe@Fe}_3\text{O}_4$  with core-shell structures exhibited more than twice higher decomposition activity than the  $\text{Fe}^0$  and  $\text{Fe}_3\text{O}_4$  nanoparticles with interface structures. We also demonstrated that a thin  $\text{Fe}_3\text{O}_4$  layer up to tens of nanometers was formed by the auto-oxidation of iron particles when exposed to air, leading to similar core-shell structures of the  $\text{Fe@Fe}_3\text{O}_4$  model system and, consequently, comparable Fenton activity. Furthermore, our results also indicated that the redox propensity might be a more critical activity descriptor than the contents of  $\text{Fe}^{2+}$  and oxygen vacancy in the Fenton chemistry, and the co-existence of both zero-valent iron and rich oxygen vacancy can boost the Fenton activity.

## 2. Results and Discussion

### 2.1. Synthesis and Characterization of the Iron Oxides

#### 2.1.1. Synthesis of the Iron Oxides

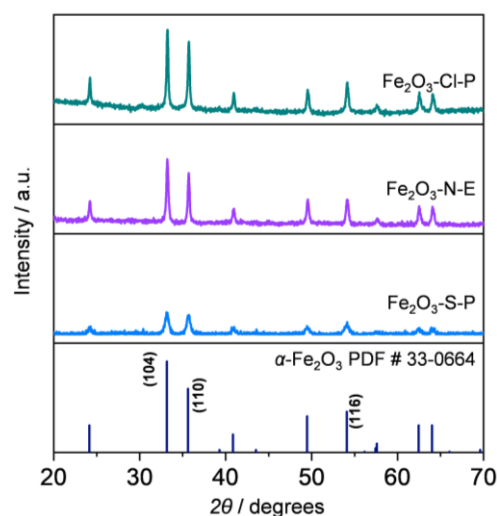
Different iron oxides were prepared by the precipitation methods by varying the iron precursors and the surfactants (Figure 1). The resultant precipitates derived from  $\text{Fe}(\text{NH}_4)_2(\text{SO}_4)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ , and  $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$  were denoted as  $\text{Fe}_2\text{O}_3\text{-S-P}$ ,  $\text{Fe}_2\text{O}_3\text{-N-E}$ , and  $\text{Fe}_2\text{O}_3\text{-Cl-P}$ , respectively. These samples were further subject to  $\text{H}_2$  reduction before the application in MO decomposition. In addition, a commercial sample denoted as  $\text{Fe}_3\text{O}_4\text{-C}$  was purchased as the reference.



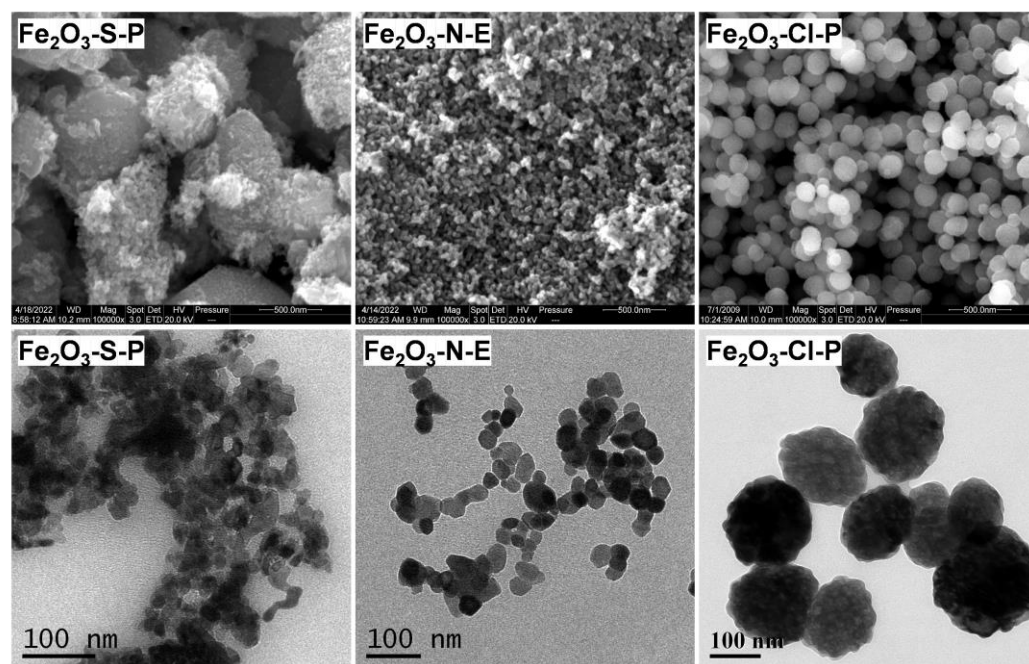
**Figure 1.** Schematic illustration of the synthetic procedures of the iron oxides, accompanied by the sample codes and structural characteristics.

#### 2.1.2. Compositional and Structural Analyses of the Iron Oxides

The compositions of the as-prepared iron oxides were first checked by powder X-ray diffraction patterns (PXRD, Figure 2). The representative diffraction lines corresponding to  $\alpha\text{-Fe}_2\text{O}_3$  (PDF # 33-0664) can be found for all the samples, confirming the successful synthesis of the targeted materials. In addition, these materials showed clear differences in the diffraction intensities with the order of  $\text{Fe}_2\text{O}_3\text{-Cl-P} \sim \text{Fe}_2\text{O}_3\text{-N-E} \gg \text{Fe}_2\text{O}_3\text{-S-P}$ . The crystallite sizes of  $\text{Fe}_2\text{O}_3$  were then estimated by using the Scherrer equation and the (104) facet. The sizes of  $\text{Fe}_2\text{O}_3\text{-Cl-P}$ ,  $\text{Fe}_2\text{O}_3\text{-N-E}$ , and  $\text{Fe}_2\text{O}_3\text{-S-P}$  were determined to be 44, 33, and 15 nm, respectively. The morphologies of as-derived  $\text{Fe}_2\text{O}_3$  were studied by combined electron microscopic techniques (Figure 3). Scanning electron microscopy (SEM) revealed the formation of condensed aggregates for  $\text{Fe}_2\text{O}_3\text{-S-P}$  and  $\text{Fe}_2\text{O}_3\text{-N-E}$  and the more homogeneous nanospheres for  $\text{Fe}_2\text{O}_3\text{-Cl-P}$ , which were likely resulting from the combined use of different iron precursors and surfactants. In line with the SEM observation, transmission electron microscopy (TEM) also confirmed the relatively smaller sizes of the irregular nanoparticles for  $\text{Fe}_2\text{O}_3\text{-S-P}$  and  $\text{Fe}_2\text{O}_3\text{-N-E}$ .



**Figure 2.** The PXRD patterns of the as-prepared iron oxides. Vertical lines are the reference standard.

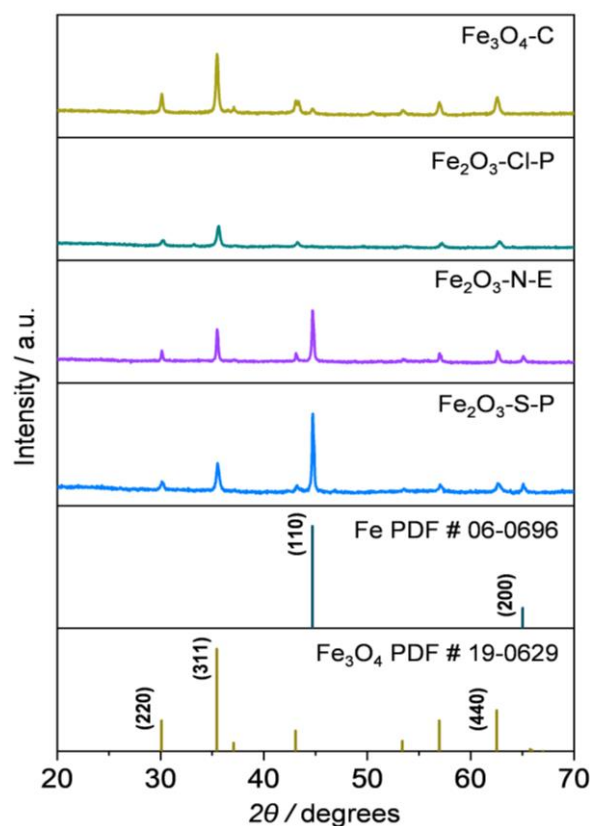


**Figure 3.** The SEM (top) and TEM (bottom) images of the as-prepared iron oxides.

### 2.1.3. Compositional and Structural Analyses of the Reduced Iron Oxides

The PXRD patterns of the  $\text{Fe}_2\text{O}_3$  and the commercial  $\text{Fe}_3\text{O}_4$ -C after  $\text{H}_2$  reduction at  $350^\circ\text{C}$  were presented in Figure 4. The results can be grouped into two classes. Both  $\text{Fe}_2\text{O}_3$ -N-E and  $\text{Fe}_2\text{O}_3$ -S-P showed the predominant diffraction lines corresponding to the (110) facet of metallic iron ( $\text{Fe}^0$  PDF # 06-0696) with the coexistence of relatively weak diffractions related to the (311) and (220) facets of  $\text{Fe}_3\text{O}_4$  (PDF # 19-0629). On the contrary, all the typical diffraction patterns of  $\text{Fe}_3\text{O}_4$  were found for  $\text{Fe}_2\text{O}_3$ -Cl-P and  $\text{Fe}_3\text{O}_4$ -C, but with quite different intensities. The above results suggested the different redox abilities of these samples, which will be studied in detail in the next sections. Furthermore, it means  $\text{Fe}_3\text{O}_4$  was probably the predominant phase with the coexisted  $\text{Fe}^0$  in  $\text{Fe}_2\text{O}_3$ -S-P and  $\text{Fe}_2\text{O}_3$ -N-E, while  $\text{Fe}_3\text{O}_4$  was the main phase in  $\text{Fe}_2\text{O}_3$ -Cl-P and  $\text{Fe}_3\text{O}_4$ -C. Estimation of the crystallite sizes of  $\text{Fe}^0$  based on the (110) facet revealed comparable values between  $\text{Fe}_2\text{O}_3$ -S-P and  $\text{Fe}_2\text{O}_3$ -N-E (57 vs. 52 nm), while the sizes of  $\text{Fe}_3\text{O}_4$  based on the (311)

facet were determined to be 24, 53, 36, and 56 nm, respectively, for  $\text{Fe}_2\text{O}_3\text{-S-P}$ ,  $\text{Fe}_2\text{O}_3\text{-N-E}$ ,  $\text{Fe}_2\text{O}_3\text{-Cl-P}$  and  $\text{Fe}_3\text{O}_4\text{-C}$ .



**Figure 4.** The PXRD patterns of the iron oxides after 350 °C reduction. Vertical lines are the reference standards.

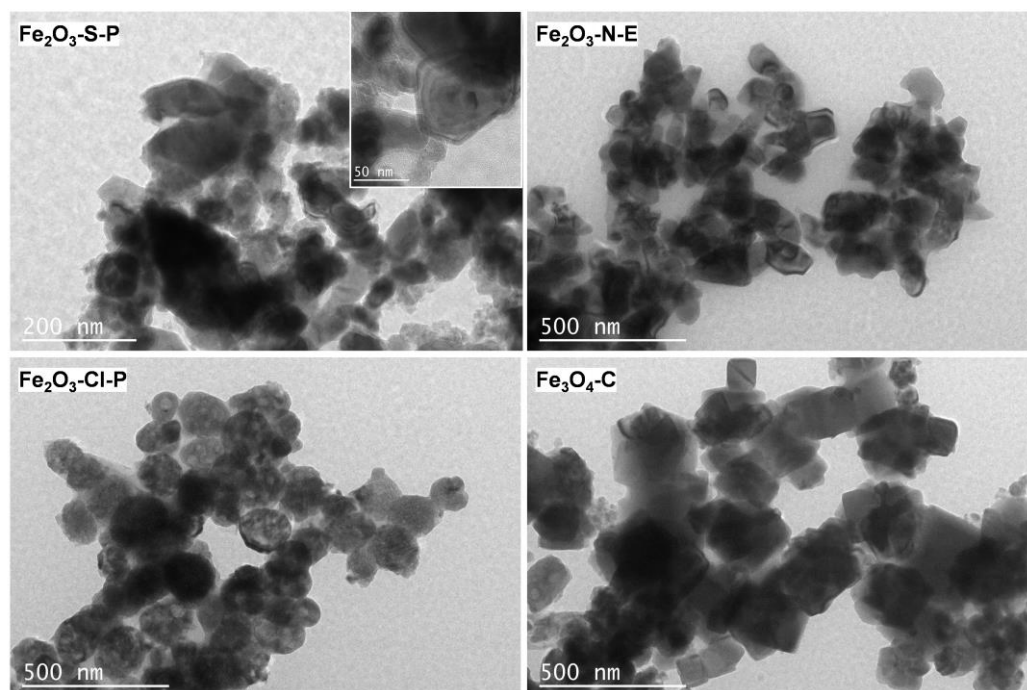
The morphologies of the reduced samples were assessed by TEM (Figure 5).  $\text{Fe}_2\text{O}_3\text{-S-P}$  and  $\text{Fe}_2\text{O}_3\text{-N-E}$  showed irregular-shaped particles, while  $\text{Fe}_2\text{O}_3\text{-Cl-P}$  can still preserve the sphere-like morphology. As compared with un-reduced iron oxides, the particle sizes became much larger, hinting at the severe aggregation during the reduction treatments. Additionally,  $\text{Fe}_3\text{O}_4\text{-C}$  also showed big particles of irregular shapes. To further assess the structural differences, high-resolution TEM (HRTEM) analyses of the reduced samples were conducted (inset in Figure 5 and S1). Among all these materials, a clear core-shell structure was evidenced only in  $\text{Fe}_2\text{O}_3\text{-S-P}$ . Detailed analyses of this material revealed the continuous lattice fringes of Fe(111) with a distance of 2.0268 Å, and in the periphery, the lattice fringes of  $\text{Fe}_3\text{O}_4$ (111) with a distance of 2.5320 Å were found (Figure 6). These observations, therefore, ambiguously pointed to the formation of  $\text{Fe}^0$  as the core covered with a thin shell of  $\text{Fe}_3\text{O}_4$ . As in the case of  $\text{Fe}_2\text{O}_3\text{-S-P}$ , the lattice fringes of both Fe(110) and  $\text{Fe}_3\text{O}_4$ (220) facets were observed in  $\text{Fe}_2\text{O}_3\text{-N-E}$ . These two facets were also in close proximity, suggesting the presence of Fe- $\text{Fe}_3\text{O}_4$  interfaces. In contrast, lattice fringe analyses of  $\text{Fe}_2\text{O}_3\text{-Cl-P}$  and  $\text{Fe}_3\text{O}_4\text{-C}$  only found the presence of the  $\text{Fe}_3\text{O}_4$  phase. In general, the TEM observations were highly consistent with the PXRD results.

#### 2.1.4. Redox Properties of the Iron Oxides

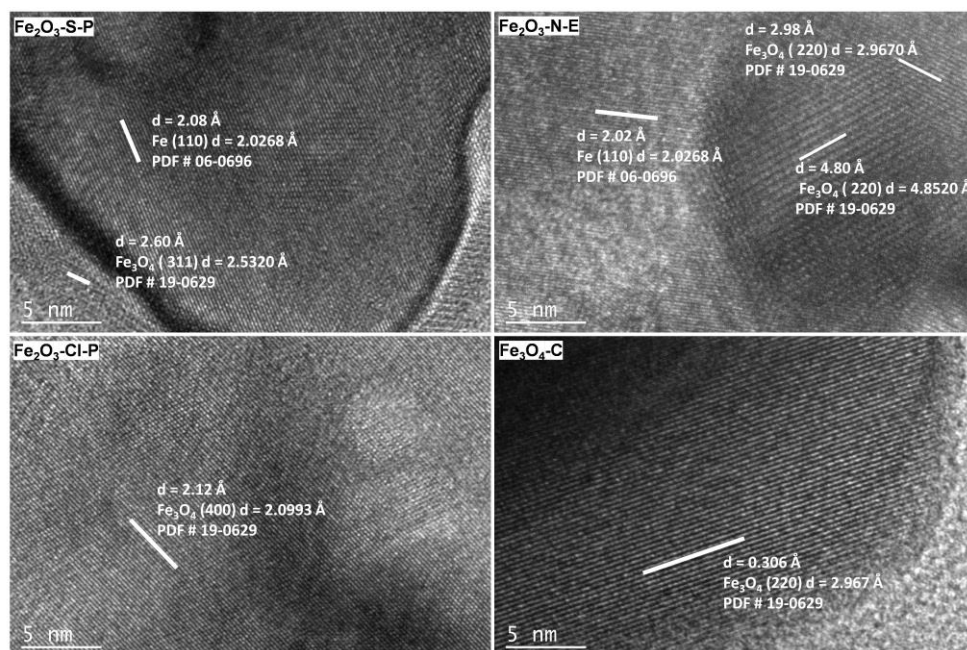
The redox properties of a catalytic material play an important role in the Fenton chemistry. Therefore, different techniques were applied to assess the redox properties of the as-derived iron oxides. The phase transitions of different iron compositions in flowing  $\text{H}_2$  from room temperature to 700 °C were followed by in situ PXRD (Figure 7). Both  $\text{Fe}_2\text{O}_3\text{-S-P}$  and  $\text{Fe}_2\text{O}_3\text{-N-E}$  showed very similar phase transition patterns. Namely, the  $\text{Fe}_3\text{O}_4$  phase in these two samples vanished at much lower temperatures as compared



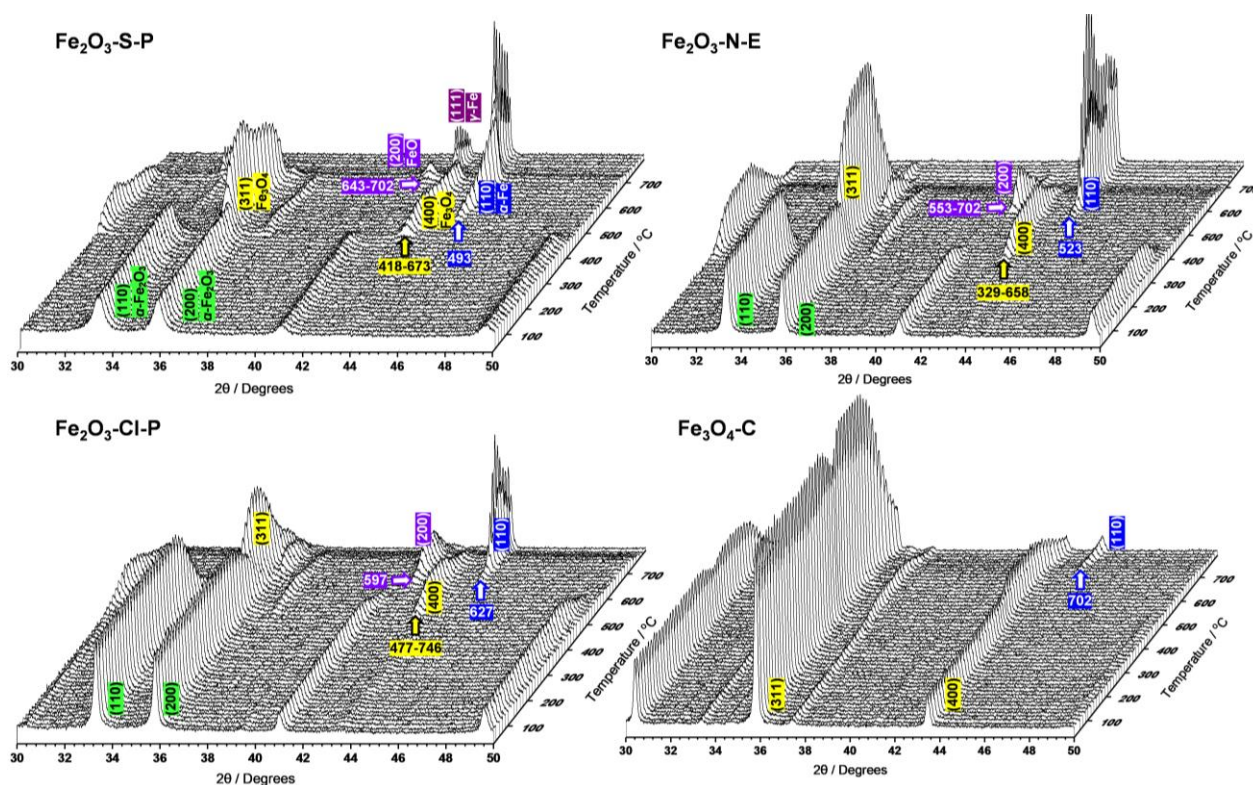
with  $\text{Fe}_2\text{O}_3\text{-Cl-P}$  (658–673 vs. 746 °C). Meanwhile, the appearance of  $\text{Fe}^0$  diffractions emerged at significantly lower temperatures (493–523 vs. 627 °C). Differing from the three as-prepared  $\text{Fe}_2\text{O}_3$  materials, which showed the reduction to different degrees, the in situ PXRD patterns of  $\text{Fe}_3\text{O}_4\text{-C}$  indicated that this material was extremely difficult to reduce. The  $\text{Fe}_3\text{O}_4$  phase predominated with only very weak diffractions related to  $\text{Fe}^0$ . Overall, these results suggested the very different reduction propensities of the materials with the order of  $\text{Fe}_2\text{O}_3\text{-S-P} \sim \text{Fe}_2\text{O}_3\text{-N-E} > \text{Fe}_2\text{O}_3\text{-Cl-P} \gg \text{Fe}_3\text{O}_4\text{-C}$ .



**Figure 5.** The TEM images of the iron oxides after 350 °C reduction. The inset in top-left shows the core-shell structures in  $\text{Fe}_2\text{O}_3\text{-S-P}$ .



**Figure 6.** The HRTEM images of the iron oxides after 350 °C reduction.



**Figure 7.** The in situ PXRD patterns of the iron oxides in flowing  $H_2$ .

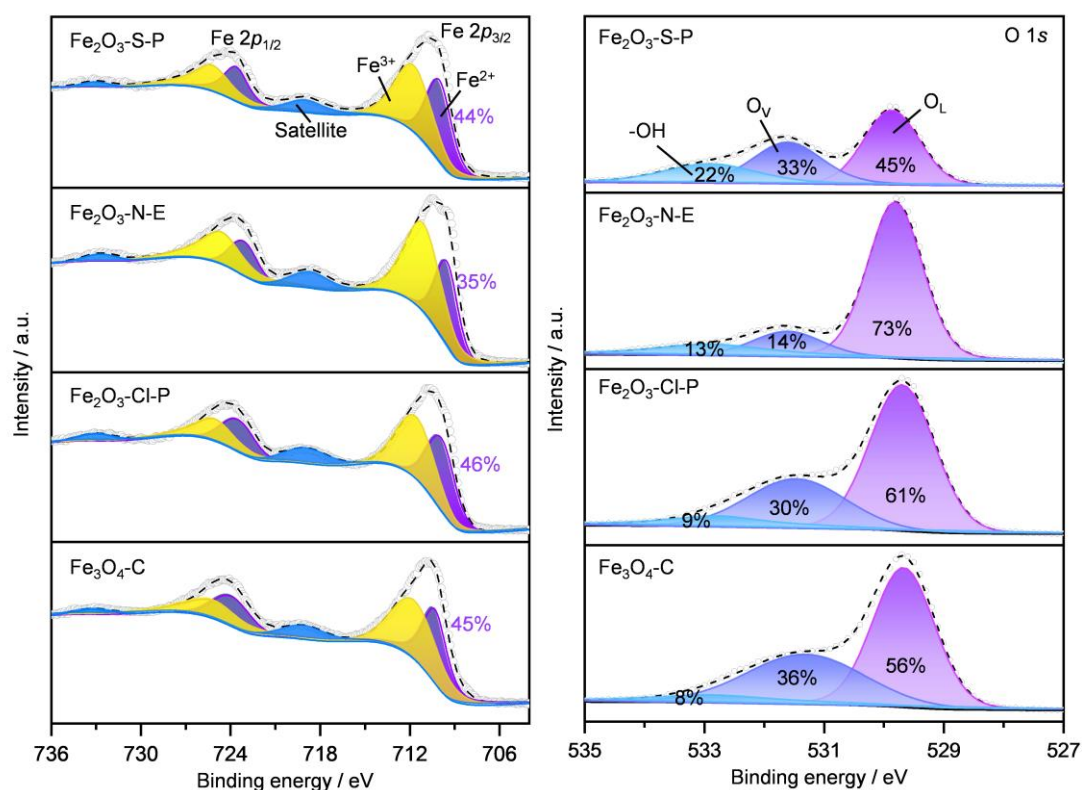
### 2.1.5. Surface Iron Species and Oxygen Vacancy

The different surface iron and oxygen species of the reduced iron oxides were studied by X-ray photoelectron spectroscopy (XPS, Figure 8), as they were suggested to have important influences on the redox and Fenton chemistry. The Fe 2p spectra of all the samples showed a doublet at 736–706 eV with a satellite around 718 eV. The spectra were analyzed by peak fitting with  $Fe^{2+}$  and  $Fe^{3+}$  species with known peak positions from the literature results. It was determined that the shares of  $Fe^{2+}$  were comparable for  $Fe_2O_3$ -S-P,  $Fe_2O_3$ -Cl-P, and  $Fe_3O_4$ -C (44–46%) and were much higher than that of  $Fe_2O_3$ -N-E (35%). The O 1s of these reduced samples were also analyzed by fitting the spectra. It was revealed that  $Fe_2O_3$ -S-P,  $Fe_2O_3$ -Cl-P, and  $Fe_3O_4$ -C possessed similar shares of oxygen vacancy ( $O_v$ , 30–36%), while it was the lowest for  $Fe_2O_3$ -N-E (14%).

### 2.2. Performance in MO Decomposition

The catalytic performance of the derived iron oxides was evaluated in the decomposition of methyl orange that was chosen as a model reaction to assess the Fenton chemistry. All the materials were reduced in  $H_2$  at 350 °C and stored in air at room temperature for more than 24 h prior to the tests. At first, the activity was evaluated at a lower MO concentration of 100  $\mu g/mL$  (Figure 9a). The conversion steadily increased for  $Fe_2O_3$ -Cl-P and  $Fe_3O_4$ -C. While full conversion was already achieved at 60 min for  $Fe_2O_3$ -C, the conversion only slowly increased and stayed at about 22% at 120 min for  $Fe_2O_3$ -Cl-P. In stark contrast, full conversion was readily achieved upon the addition of  $H_2O_2$  for both  $Fe_2O_3$ -S-P and  $Fe_2O_3$ -N-E. To further discriminate the activity difference, the decomposition activity was further evaluated at a higher MO concentration of 500  $\mu g/mL$  (Figure 9b). Under such conditions, full conversion was still readily reached at 30 min for  $Fe_2O_3$ -S-P. In contrast, the conversions at 120 min were only 80% and 35%, respectively, for  $Fe_2O_3$ -N-E and  $Fe_3O_4$ -C. To quantitatively compare the Fenton activity of the four samples, the initial reaction rates were calculated by considering the linear parts of the conversion-time profiles (Figure 9c).  $Fe_2O_3$ -S-P exhibited the highest rate of 370  $mg_{MO} g_{cat}^{-1} min^{-1}$ , which is 2 times higher than that of  $Fe_2O_3$ -N-E and significantly outperforms  $Fe_2O_3$ -Cl-P and

$\text{Fe}_3\text{O}_4\text{-C}$  by 1–2 orders of magnitude. In order to confirm that the reaction proceeded in a heterogeneous manner, we conducted another test by replacing the iron oxides with trace  $\text{Fe}(\text{NO}_3)_3$  so as to mimic the leaching of iron species into the solution. It turned out that the MO decomposition occurred roughly slowly in the first 30 min, and a low conversion of ca. 26% was reached for 120 min (Figure S2). The extremely slow kinetics, as compared with the highly active  $\text{Fe}_2\text{O}_3\text{-S-P}$  and  $\text{Fe}_2\text{O}_3\text{-N-E}$ , thus suggested the heterogeneous nature of the decomposition reaction on the latter. The cycling performance of  $\text{Fe}_2\text{O}_3\text{-S-P}$  was attempted. To fully recover the solid catalyst after use, the catalyst amount and the volume of wastewater were scaled up by four folds. Under such conditions, similar kinetics were achieved in the two consecutive cycles, although full removal of MO cannot be reached in 120 min (Figure S3). The relatively lower activity in the scale-up tests might be caused by the mass transfer effect. In addition, the dissolved iron in the solution after the first cycle was determined to be ca. 48 ppm by inductively coupled plasma optical emission spectroscopy (ICP-OES).



**Figure 8.** The Fe 2p and O 1s XPS profiles of the iron oxides after 350 °C reduction. O<sub>v</sub>—oxygen vacancy; O<sub>L</sub>—lattice oxygen.

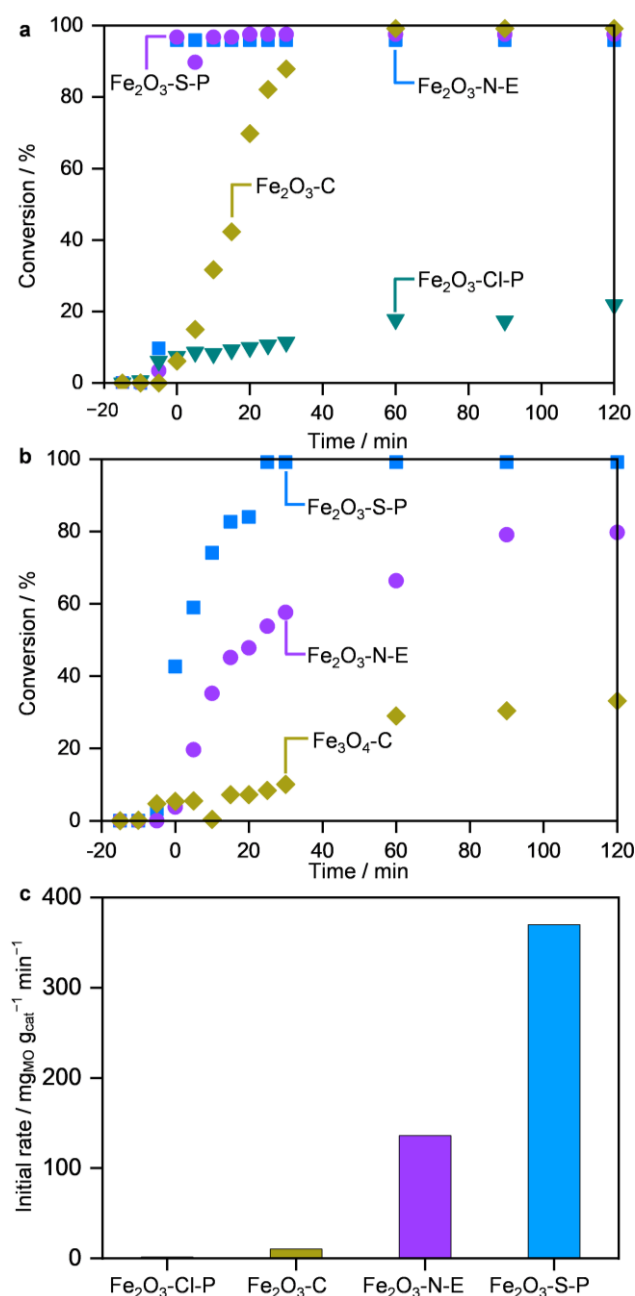
## 2.3. Origins of the Divergent Catalytic Performance

### 2.3.1. Structure–Performance Relationships

The Fenton activity of iron-based materials can be influenced by many factors, which makes it difficult to discriminate the key activity descriptors. Among the suspected properties, the  $\text{Fe}^{2+}$  species, the oxygen vacancy, and redox properties might play important roles in influencing the activity. Therefore, tentative correlations were made between the initial reaction rates with the shares of  $\text{Fe}^{2+}$  and O<sub>v</sub> derived from the XPS fitting and the  $T_{\text{Fe}}$  (the temperature corresponding to the appearance of the  $\text{Fe}^0$  phase from in situ PXRD, Figure 10). We noted that  $T_{\text{Fe}}$  was adopted to roughly reflect the redox properties of the iron oxides. The results demonstrated that there was no clear dependence of the initial activity on either the shares of  $\text{Fe}^{2+}$  or O<sub>v</sub>, whereas increasing activity as a function of



decreasing  $T_{Fe}$  was evidenced. Therefore, it seems that the redox propensity might be a reasonable activity descriptor in the Fenton chemistry.

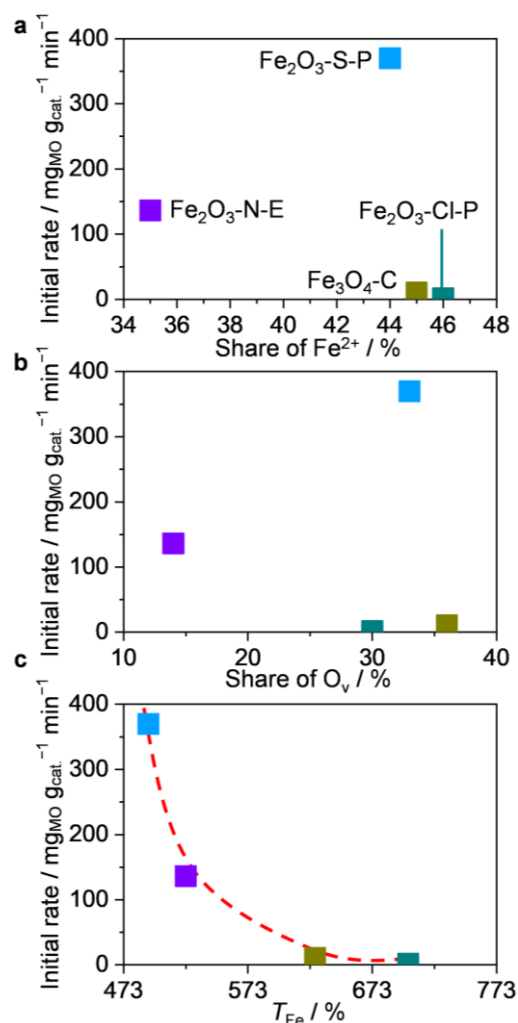


**Figure 9.** (a,b) The MO decomposition activity as a function of time on different iron oxides. (c) Comparison of the initial activity of different iron oxides. Reaction conditions:  $w_{cat} = 5$  mg (350 °C reduced); pH = 3,  $V = 100$  mL,  $C_{H_2O_2} = 20$  mM; (a)  $C_{MO} = 100 \mu g/mL$ , (b)  $C_{MO} = 500 \mu g/mL$ . The first three data points before the addition of  $H_2O_2$  correspond to the freshly prepared solution, the solutions after the adjustment in pH, and after 30 min stirring, respectively.

### 2.3.2. Importance of the Fe@Fe<sub>3</sub>O<sub>4</sub> Interfaces

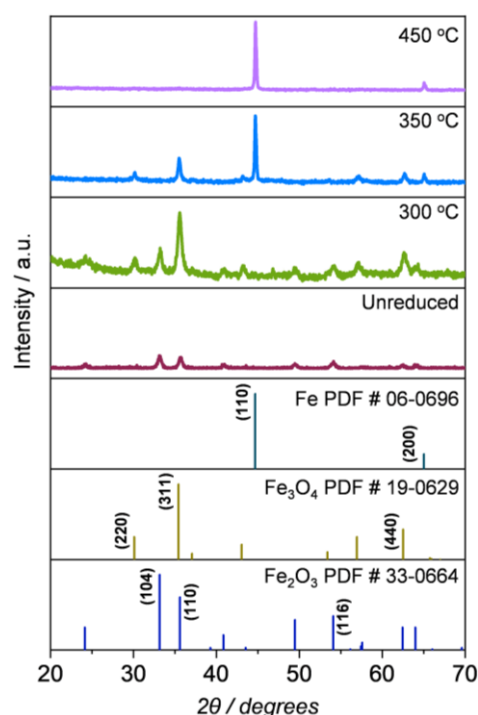
The detailed characterization of the iron oxides revealed that both  $Fe^0$  and  $Fe_3O_4$  were formed on  $Fe_2O_3$ -S-P and  $Fe_2O_3$ -N-E, while only  $Fe_3O_4$  existed on  $Fe_2O_3$ -Cl-P and  $Fe_3O_4$ -C. Thus, it is reasonable to speculate that the presence of  $Fe^0$  was crucial to the high Fenton activity. To further discriminate the MO decomposition performance of the different compositions of iron species,  $Fe_2O_3$ -S-P reduced in  $H_2$  at varying temperatures from

300 to 450 °C was conducted. The PXRD results confirmed that  $\text{Fe}_2\text{O}_3$  was the major phase after 300 °C reduction (Figure 11). After reduction at 350 °C,  $\text{Fe}^0$  was the major phase with the coexistence of  $\text{Fe}_3\text{O}_4$ . After further increasing the temperature to 450 °C,  $\text{Fe}^0$  was the only phase.

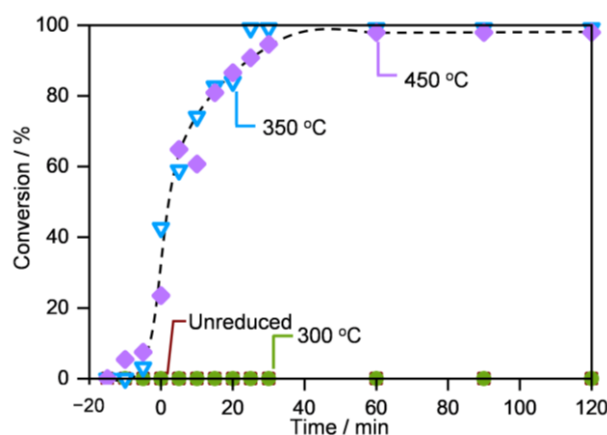


**Figure 10.** The initial reaction rates of different iron oxides as a function of (a) surface  $\text{Fe}^{2+}$  ratio, (b) oxygen vacancy, and (c)  $T_{\text{Fe}}$  (the temperature corresponding to the appearance of  $\text{Fe}^0$  phase from in situ PXRD).  $T_{\text{Fe}}$  can be viewed as an indicator of the redox properties of the respective catalysts.

Having derived different compositions of the iron-based materials from  $\text{Fe}_2\text{O}_3\text{-S-P}$ , their activity in MO decomposition was evaluated (Figure 12). The results showed that the samples reduced at 350 and 450 °C displayed basically the same activity, whereas no appreciable conversion can be observed for the unreduced sample and after 300 °C reduction. Combined with the activity evaluation, it can be concluded that  $\text{Fe}^0$  or  $\text{Fe}^0/\text{Fe}_3\text{O}_4$  were responsible for the catalytic activity, but both  $\text{Fe}_3\text{O}_4$  and  $\text{Fe}_2\text{O}_3$ , when used alone, were completely inactive in Fenton chemistry.



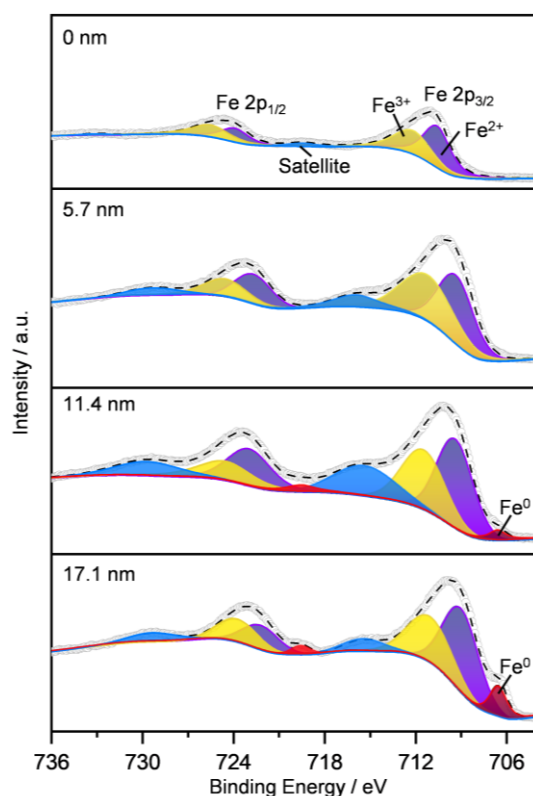
**Figure 11.** The PXRD patterns of  $\text{Fe}_2\text{O}_3$ -S-P after reduction in  $\text{H}_2$  at 300–450 °C. Vertical lines are the reference standards.



**Figure 12.** The MO decomposition activity as a function of time on  $\text{Fe}_2\text{O}_3$ -S-P subject to  $\text{H}_2$  reduction at different temperatures. Reaction conditions:  $w_{\text{cat}} = 5$  mg,  $\text{pH} = 3$ ,  $V = 100$  mL,  $C_{\text{H}_2\text{O}_2} = 20$  mM,  $C_{\text{MO}} = 500$   $\mu\text{g/mL}$ . The first three data points before the addition of  $\text{H}_2\text{O}_2$  correspond to the freshly prepared solution, the solutions after the adjustment in pH, and after 30 min stirring, respectively.

Although the iron oxide in  $\text{Fe}_2\text{O}_3$ -S-P was completely converted into  $\text{Fe}^0$  after 450 °C reduction based on PXRD, the exposure to air should easily lead to the formation of a thin oxide layer. It means for this sample, both  $\text{Fe}^0$  and  $\text{Fe}_3\text{O}_4$  might co-exist, like in the case of the sample reduced at 350 °C. To prove the speculation, XPS with depth profile analysis was performed on  $\text{Fe}_2\text{O}_3$ -S-P after 450 °C and exposure to air (Figure 13). The fitting of the  $\text{Fe } 2p$  XPS spectra evidently revealed the exclusive presence of both  $\text{Fe}^{3+}$  and  $\text{Fe}^{2+}$  species at the outer surfaces, while the contribution of  $\text{Fe}^0$  emerged up to a depth of 11.4 nm, and the share became larger, subject to further sputtering. The results thus confirmed the formation of an iron oxide layer of ca. 11–17 nm thickness owing to the auto-oxidation of metallic iron. In addition, the previous TEM and PXRD results of  $\text{Fe}_2\text{O}_3$ -S-P after 350 °C reduction also indicated the presence of the  $\text{Fe}@\text{Fe}_3\text{O}_4$  core-shell structure and a thickness of  $\text{Fe}_3\text{O}_4$  layer of <24 nm. Therefore, the similar activity of  $\text{Fe}_2\text{O}_3$ -S-P subject to

350–400 °C reduction can be attributed to the presence of similar Fe@Fe<sub>3</sub>O<sub>4</sub> core-shell structures. On the other hand, Fe<sub>2</sub>O<sub>3</sub>-N-E with both Fe<sup>0</sup> and Fe<sub>3</sub>O<sub>4</sub> as the major phases without the apparent core-shell structures exhibited relatively lower activity as compared to Fe<sub>2</sub>O<sub>3</sub>-S-P. Albeit an interface between both phases was confirmed by HRTEM, no core-shell structures can be observed, plus the Fe<sub>3</sub>O<sub>4</sub> particle sizes (ca. 54 nm) were much larger as compared with the layer thickness in Fe<sub>2</sub>O<sub>3</sub>-S-P. These factors might lead to a reduced number of interface structures and, thus, lower activity.



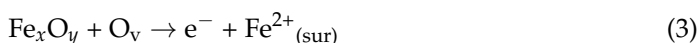
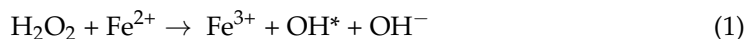
**Figure 13.** The Fe 2p XPS depth profiles of Fe<sub>2</sub>O<sub>3</sub>-S-P after 450 °C reduction. All the samples were stored in air at room temperature for more than 24 h.

### 2.3.3. Discussions on the Origin of Superior Activity

The formation of radicals such as OH<sup>\*</sup> and HO<sub>2</sub><sup>\*</sup> was generally acknowledged to be responsible for the MO decomposition in the Fenton chemistry. Indeed, the reaction was immediately jeopardized upon the addition of a radical trapper such as dimethyl sulfoxide (Figure S4), demonstrating the radical-induced nature of this reaction. Both Fe<sub>2</sub>O<sub>3</sub>-Cl-P and Fe<sub>3</sub>O<sub>4</sub>-C showed comparable rich surface Fe<sup>2+</sup> species and O<sub>v</sub> but without the presence of ZVI. In contrast, Fe<sub>2</sub>O<sub>3</sub>-N-E containing ZVI but with a lower number of O<sub>v</sub> also displayed inferior activity. These results hinted that the superior Fenton activity of Fe<sub>2</sub>O<sub>3</sub>-S-P with the Fe@Fe<sub>3</sub>O<sub>4</sub> core-shell structures might be attributed to the abundance of both Fe<sup>0</sup> and O<sub>v</sub>. The rate-limiting step in the heterogeneous Fenton reaction is viewed as the formation of Fe<sup>2+</sup> species that are responsible for generating highly active OH<sup>\*</sup> through Equation (1). Previous studies have proposed that Fe<sup>2+</sup> species might be produced by the oxidation of ZVI by H<sub>2</sub>O<sub>2</sub> or molecular oxygen in acidic conditions [40,41]. In addition, the studies on the Fe@Fe<sub>2</sub>O<sub>3</sub> systems also suggested the excellent reducing properties of metallic iron to achieve a high Fe<sup>2+</sup>/Fe<sup>3+</sup> redox cycling rate, wherein ZVI can directly inject two electrons to activate molecular oxygen [42,43]. On the other hand, the rich number of O<sub>v</sub> has been demonstrated to be beneficial to the generation of OH<sup>\*</sup> [44–47]. Li et al. [46] confirmed that H<sub>2</sub>O<sub>2</sub> adsorbed on the electron-rich location readily underwent dissociation induced by O<sub>v</sub> (Equation (2)). Chen et al. [44] reported the accelerated electron transfer



process and threefold yield of  $\text{OH}^*$  for the  $\text{O}_v$ -rich micro ZVI interface. The synergistic cooperation catalysis between ZVI and iron oxides was proposed in Equation (3). Following this equation, the surface-bonded  $\text{Fe}^{2+}$  species were generated between the iron oxides and  $\text{O}_v$ . The surface-bonded  $\text{Fe}^{2+}$  species were suggested to be more reactive than ferrous ions in activating  $\text{H}_2\text{O}_2$ , thus avoiding the rate-limiting step in the classic Fenton reaction.



### 3. Materials and Methods

#### 3.1. Synthesis of the Iron Oxides

All the reagents were obtained from commercial suppliers and used without further purification.  $\text{Fe}(\text{NH}_4)_2(\text{SO}_4)_2 \cdot 6\text{H}_2\text{O}$  (99.5%),  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ , and  $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$  were purchased from Aladdin. Polyethylene glycol 400 (PEG-400), ethylene glycol (EG,  $\geq 99.5\%$ ), and sodium carbonate ( $\text{Na}_2\text{CO}_3$ ,  $\geq 99.8\%$ ) were purchased from Sinopharm. Ammonium hydroxide solution ( $\text{NH}_3 \cdot \text{H}_2\text{O}$ , 25–28%) and  $\text{Fe}_3\text{O}_4\text{-C}$  (99.5%) were purchased from Macklin.

The other three iron oxides were prepared by a precipitation method. A solution containing  $\text{Fe}(\text{NH}_4)_2(\text{SO}_4)_2 \cdot 6\text{H}_2\text{O}$  (0.02 mol/L) and polyethylene glycol 400 (200 mL) was gradually heated to 120 °C under mechanical stirring and maintained at 120 °C for 1 h. Then a  $\text{Na}_2\text{CO}_3$  aqueous solution (200 mL, 0.2 mol/L) was added gradually dropwise through a syringe pump at a rate of 5.5 mL/min. The mixture was aged at 120 °C for 1 h. The precipitate was washed with water and ethanol, dried at 120 °C for 12 h, and calcined at 500 °C for 5 h. The solids obtained, which were denoted as  $\text{Fe}_2\text{O}_3\text{-S-P}$ ,  $\text{Fe}_2\text{O}_3\text{-N-E}$ , and  $\text{Fe}_2\text{O}_3\text{-Cl-P}$ , were obtained following similar precipitation methods by using  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  and  $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$  as the precursors, EG and PEG-400 as the solvents, and sodium carbonate concentrations of 0.02 and 0.2 M, respectively.

#### 3.2. Material Characterizations

The composition and crystalline structures of the catalysts were characterized by powder X-ray diffraction (PXRD) with an x'Pert3 Panalytical X-ray diffractometer (PANalytical, Almelo, The Netherlands) using  $\text{Cu K}_\alpha$  radiation ( $\lambda = 0.154178$  nm). The tube voltage and current were 40 kV and 40 mA, respectively. In situ PXRD patterns were recorded with Rigaku SmartLab-9 kW D/teX Ultra250 diffractometer using a  $\text{Cu K}_\alpha$  radiation source operated at 40 kV and 200 mA. The powder samples were spread into a high-temperature chamber, and the diffraction patterns were recorded at  $2\theta$  of 30–90° with a scanning rate of  $10^\circ \text{ min}^{-1}$  at room temperature. Then, the sample was heated to 823 K successively at a rate of  $10^\circ \text{ C min}^{-1}$  under 10%  $\text{H}_2/\text{He}$  flow of  $30 \text{ mL min}^{-1}$ .

Scanning electron microscopy (SEM) was conducted on a Quanta 200F instrument operating at 10 kV and 50 pA. The powder sample was dispersed in dry form onto fresh carbon paint deposited on an aluminum holder.

Transmission electron microscopy (TEM) images were taken on an FEI Tecnai G2F 20 microscopy operating at 120 kV. The specimen was prepared by ultrasonically dispersing the powder sample in ethanol, depositing droplets of the suspensions onto carbon-coated copper or gold grids, and drying in air.

X-ray photoelectron spectroscopy (XPS) analysis of the materials was performed on a Thermo ESCALAB 250Xi spectrometer using a 15 kV  $\text{Al K}_\alpha$  X-ray source as a radiation source. The binding energy was calibrated using the C 1s peak (284.6 eV) as the reference. For the depth profile analysis, the samples were sputtered with Ar ions.

The iron content in the solution after the catalytic test was measured by inductively coupled plasma optical emission spectroscopy (ICP-OES) on a Perkin-Elmer Optima 7300DV.

### 3.3. Activity Evaluation of the Iron-Based Materials in Decomposition of Methyl Orange

The activity evaluation of the developed iron-based materials in the decomposition of methyl orange (MO) was performed in a 250 mL three-necked round flask equipped with a magnetic stirrer and a temperature thermometer. The temperature was controlled by a pre-heated water bath. For a typical reaction, 100 mL of aqueous MO solutions of 100 or 500  $\mu\text{g/mL}$  were charged into the flask reactor. The pH of the solution was then adjusted to 3.0 by diluted HCl solution. After that, a certain amount of the solid catalysts was added, and the suspension was stirred for 30 min to reach the adsorption–desorption equilibrium. Finally, the reaction was initiated upon the addition of a diluted  $\text{H}_2\text{O}_2$  solution, and the concentration of MO was continuously monitored by taking a small portion of the liquid samples for analysis on a UV–Vis spectrometer (TU-1810, Beijing Puxi General Motors Instrument Co.) at  $\lambda_{\text{max}} = 464 \text{ nm}$  (see Figure S5 for more details).

## 4. Conclusions

We have developed iron oxides of four different model systems through the conventional precipitation method by combining them with subsequent  $\text{H}_2$  reduction treatment. These materials possess very different compositions and structural features, including the  $\text{Fe@Fe}_3\text{O}_4$  core–shell structures with rich oxygen vacancy ( $\text{Fe}_2\text{O}_3\text{–S–P}$ ), the mixtures of  $\text{Fe}^0$  and  $\text{Fe}_3\text{O}_4$  particles with interface structures but low oxygen vacancy ( $\text{Fe}_2\text{O}_3\text{–N–E}$ ), and  $\text{Fe}_3\text{O}_4$ -dominated nanoparticles of different sizes and rich in oxygen vacancy ( $\text{Fe}_2\text{O}_3\text{–Cl–P}$  and  $\text{Fe}_3\text{O}_4\text{–C}$ ). These materials, when used as Fenton catalysts in the methyl orange decomposition, displayed significantly different activity with the order of  $\text{Fe}_2\text{O}_3\text{–S–P} > \text{Fe}_2\text{O}_3\text{–N–E} > \text{Fe}_3\text{O}_4\text{–C} > \text{Fe}_2\text{O}_3\text{–Cl–P}$ . The passivating layer of metallic iron of up to ca. 24 nm can lead to similar  $\text{Fe@Fe}_3\text{O}_4$  core–shell structures with excellent activity. This study further demonstrated that the numbers of  $\text{Fe}^{2+}$  or oxygen vacancy alone might not be the only activity descriptors for the iron-based materials, while the formation of metallic iron coupled with the rich oxygen vacancy can significantly boost the Fenton activity, thus highlighting the necessity of more in-depth analyses of the role of the hybrid  $\text{Fe@Fe}_x\text{O}_y$ .

**Supplementary Materials:** The following supporting information can be downloaded at <https://www.mdpi.com/article/10.3390/catal13071057/s1>, Figure S1: The additional TEM images of the iron oxides after 350 °C reduction. Figure S2: The MO decomposition activity as a function of time in the presence of an  $\text{Fe}^{3+}$  solution. Reaction conditions:  $C_{\text{MO}} = 100 \mu\text{g/mL}$ ,  $\text{pH} = 3$ , 2.6 mg  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ , and  $C_{\text{H}_2\text{O}_2} = 20 \text{ mM}$ . Figure S3: The kinetic profiles of MO decomposition on  $\text{Fe}_2\text{O}_3\text{–S–P}$  in two consecutive runs. Reaction conditions:  $w_{\text{cat}} = 20 \text{ mg}$ ;  $\text{pH} = 3$ ,  $V = 400 \text{ mL}$ ,  $C_{\text{H}_2\text{O}_2} = 80 \text{ mM}$ ,  $C_{\text{MO}} = 500 \mu\text{g/mL}$ . The catalyst after the use was recovered by centrifugation. Figure S4: The MO decomposition activity as a function of time on  $\text{Fe}_2\text{O}_3\text{–N–E}$ . DMSO was added at 5 min, and the conversion did not increase hereafter, thus demonstrating the radical induction nature of the reaction. Figure S5: (a) The absorbance of MO solutions of different concentrations as a function of the wavelength, and (b) the linear relationship between the MO concentration and the absorbance intensity at 464 nm.

**Author Contributions:** Conceptualization, X.M. and W.K.; investigation, R.Z., J.Q. and R.T.; XRD and XPS, Y.L.; TEM, R.L.; writing—review and editing, X.M., P.F. and W.K. All authors have read and agreed to the published version of the manuscript.

**Funding:** This research received no external funding.

**Data Availability Statement:** All data are available from the authors upon request.

**Acknowledgments:** The authors would like to thank Deiyu Chen from Shiyanjia Lab ([www.shiyanjia.com](http://www.shiyanjia.com), accessed on 1 May 2023) for TEM measurements.

**Conflicts of Interest:** The authors declare no conflict of interest.

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