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# Visible Light-Responsive CeO<sub>2</sub>/MoS<sub>2</sub> Composite for Photocatalytic Hydrogen Production

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**Abstract:** Semiconductor-based photocatalyst materials play an important role in solar hydrogen production. In the present work, we achieved the successful synthesis of a CeO<sub>2</sub>/MoS<sub>2</sub> composite using a facile hydrothermal method. For the preparation of the CeO<sub>2</sub>/MoS<sub>2</sub> composite, the hydrothermal process was carried out at a temperature of 120 °C for 24 h, and its performance in hydrogen production was tested. The CeO<sub>2</sub>/MoS<sub>2</sub> composite was characterized using XRD, XPS, Raman spectroscopy, SEM, and optical investigation. The optical study showed that after forming a composite with MoS<sub>2</sub>, the absorption edge of CeO<sub>2</sub> is shifted from the ultraviolet to the visible light region. Bandgap values decreased from 2.93 for CeO<sub>2</sub> to 2.34 eV for the CeO<sub>2</sub>/MoS<sub>2</sub> composite. In photocatalytic hydrogen production, Na<sub>2</sub>SO<sub>3</sub>–Na<sub>2</sub>S was used as a sacrificial agent. The CeO<sub>2</sub>/MoS<sub>2</sub> composite exhibited superior photocatalytic hydrogen production performance compared to CeO<sub>2</sub> and MoS<sub>2</sub>. The CeO<sub>2</sub>/MoS<sub>2</sub> composite achieved higher charge separation efficiency, faster charge transfer, more active sites available for redox reactions, and greater affinity towards the reactant ions due to such properties its hydrogen evolution rate has reached 112.5 μmol/h. The photostability of the CeO<sub>2</sub>/MoS<sub>2</sub> composite was tested in up to four cycles, with each cycle being four hours.



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**Keywords:** hydrogen production; hydrothermal method; composite

## 1. Introduction

The synthesis of visible light-responsive semiconductor photocatalyst materials to harvest hydrogen as a clean energy source by the photocatalytic method is an attractive, economical, and environmentally friendly process. It involves capturing light (Ultraviolet, visible, sunlight) by photocatalyst material and converting it into hydrogen [1,2]. Numerous methods have been adopted to generate hydrogen. These include steam reforming of hydrocarbons, electrolysis of water, fermentation, gasification of biomass, and photocatalytic hydrogen production. Compared to photocatalytic hydrogen production, steam reforming of hydrocarbons is an endothermic, reversible, and high-temperature reaction that generates CO<sub>2</sub>. Other methods have the same drawbacks. Photocatalytic hydrogen production is a green technology in which photo-created H<sub>2</sub> can easily be stored, and it is a potential approach to meet energy demands and is safe for the environment. It has gained more attention due to its easy operation, cost-effectiveness, efficiency, and pollution-free technique, and hydrogen energy can replace fossil fuels in the near future [3–5].

Different kinds of semiconductor materials have been used for photocatalytic hydrogen production under sunlight, UV, and visible light illumination. Among the semiconductors, TiO<sub>2</sub>, CuO, ZnO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>, CdS, AgX, V<sub>2</sub>O<sub>3</sub>, g-C<sub>3</sub>N<sub>4</sub>, and WO<sub>3</sub> are commonly used in photocatalytic hydrogen production [6]. CeO<sub>2</sub> has been studied as a catalyst material for decades. CeO<sub>2</sub> has similar properties to TiO<sub>2</sub> and ZnO. It is chemically stable, non-toxic, inexpensive, etc., and it is a substitute for TiO<sub>2</sub> and ZnO materials [7]. Its applications as a photocatalyst are limited due to its disadvantages such as its large bandgap, the ability to

absorb only a small portion of UV light, and more recombination of photogenerated charge carriers. Most studies proved that the photocatalytic activity of  $\text{CeO}_2$  can be enhanced using different strategies such as bandgap engineering, doping, heterojunction, composites, etc. Among the different strategies, the most effective method is making a composite of  $\text{CeO}_2$  with visible light response semiconductor materials that have a lower bandgap, such as transition metal dichalcogenides (TMDs), which cover the broader solar spectrum and promote charge generation and separation [8,9].

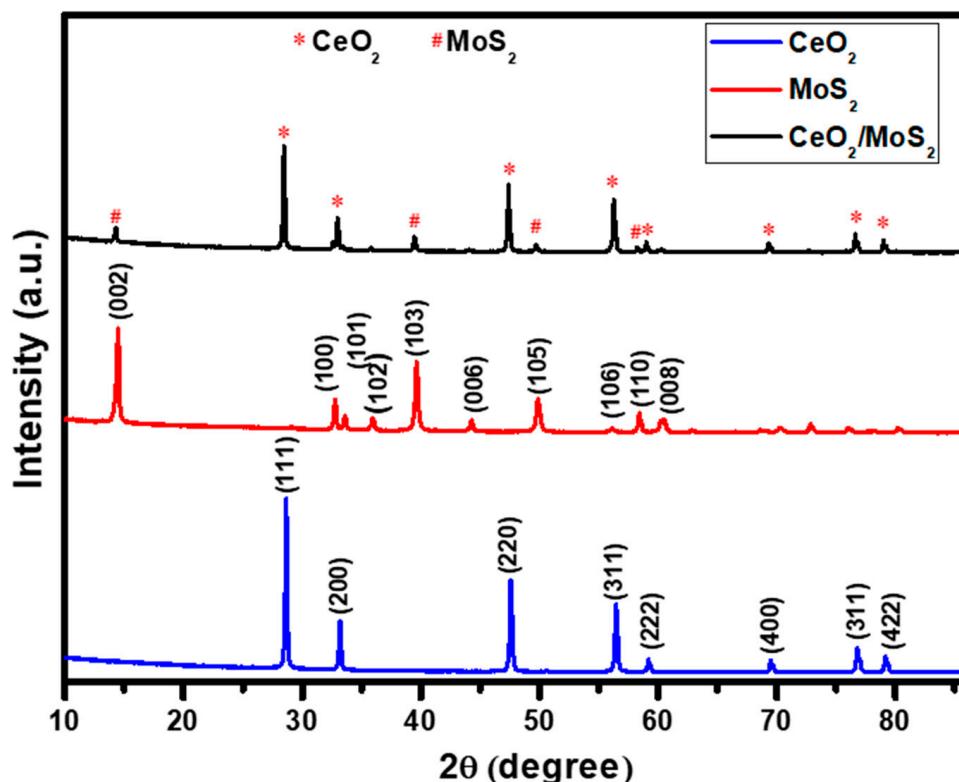
In the past few years, TMDs have received considerable attention because of their impressive structural, optical, electronic properties, and efficient co-catalytic supports, and suitable bandgaps. Thus, they are used in different fields such as photonics, energy, technology, electronics, physics, and chemistry. TMDs have layered structures similar to graphene.  $\text{MoS}_2$  is widely used as a visible light-active photocatalyst material in which Mo atoms are sandwiched between two layers of hexagonal, closely packed sulfur atoms. Such layered structures can be used in different applications such as photocatalysis, batteries, supercapacitors,  $\text{CO}_2$  reduction, water splitting, hydrogen production, etc. [10,11]. Furthermore,  $\text{MoS}_2$  is reported to be an alternative option for noble metals such as Pt, Ag, Au, etc. In addition, it is considered a suitable co-catalyst material for  $\text{H}_2$  generation using the photocatalytic and electrocatalytic methods. In many studies,  $\text{MoS}_2$  has been used as a co-catalyst or component material in the formation of a composite/heterojunction. A heterojunction provides a potential driving force, which is helpful for the generation and separation of photogenerated charge carriers, fast charge transfer, better interface contact, etc. [12,13].  $\text{MoS}_2$  possesses favorable electronic properties and thermal stability, a large surface area, and a narrow bandgap, responsible for visible light absorption due to its efficient charge transfer properties and for increasing the photocatalytic  $\text{H}_2$  evolution activity [14]. Gong et al. synthesized  $\text{MoS}_2/\text{CeO}_2$  material by a facile two-step wet chemistry process and studied the catalytic activity toward  $\text{H}_2$  production [15]. Li et al. prepared a  $\text{CeO}_2@\text{MoS}_2$  core-shell nanocomposite and used it to study supercapacitor applications (symmetric supercapacitors) [16]. Li et al. fabricated a ternary attapulgite– $\text{CeO}_2/\text{MoS}_2$  nanocomposite and studied the degradation of dibenzothiophene in gasoline under visible light irradiation [17]. Liu et al. prepared a  $\text{MoS}_2/\text{CeO}_2$  heterojunction and studied the photocatalytic degradation of methyl orange under visible light illumination [18]. From the above discussion, we can conclude that there are few reports available on  $\text{H}_2$  evolution using  $\text{MoS}_2/\text{CeO}_2$  photocatalyst.

In this work,  $\text{CeO}_2$  and  $\text{CeO}_2/\text{MoS}_2$  composite were prepared by the hydrothermal route and used to study photocatalytic hydrogen production. Prepared catalyst materials were characterized using different techniques, including X-ray diffraction (XRD), Raman spectroscopy, X-ray photoelectron spectroscopy (XPS), scanning electron microscopy (SEM), and UV–Vis spectroscopy, etc. A comparative study of  $\text{CeO}_2$ ,  $\text{MoS}_2$ , and the  $\text{CeO}_2/\text{MoS}_2$  composite was conducted to evaluate the photocatalytic hydrogen production activity. To evaluate the photostability of the  $\text{CeO}_2/\text{MoS}_2$  composite, recycling tests were performed with up to four cycles, each cycle being four hours. The presence of a sacrificial agent ( $\text{Na}_2\text{S}/\text{Na}_2\text{SO}_3$ ) stimulated  $\text{H}_2$  evolution by scavenging photogenerated holes.

## 2. Results and Discussion

The crystallographic structures of the prepared catalyst materials were investigated using the XRD technique. Figure 1 presents the XRD patterns of  $\text{CeO}_2$ ,  $\text{MoS}_2$ , and the  $\text{CeO}_2/\text{MoS}_2$  composite. The XRD pattern shows that all the prepared materials were well crystallized. For  $\text{CeO}_2$ , major diffraction peaks located at  $28.5^\circ$ ,  $33.1^\circ$ ,  $47.5^\circ$ , and  $56.3^\circ$  were well indexed as (111), (200), (220), and (311) crystal planes, respectively. The matching of standard and observed d values confirm the cubic crystal structure (Powder Diffraction File no. 34-0394) of  $\text{CeO}_2$  [19]. For  $\text{MoS}_2$ , major diffraction peaks located at  $14.44^\circ$ ,  $32.79^\circ$ , and  $39.66^\circ$  correspond to (002), (100), and (103), respectively. The matching of standard and observed d values using the Powder Diffraction File no. (PDF no. 77-1716) confirmed the hexagonal crystal structure of  $\text{MoS}_2$  [20]. In the case of the  $\text{CeO}_2/\text{MoS}_2$  composite,

MoS<sub>2</sub> and CeO<sub>2</sub> peaks are denoted by symbols # and \*, respectively. Major CeO<sub>2</sub> and MoS<sub>2</sub> diffraction peaks remain in the same positions in the CeO<sub>2</sub>/MoS<sub>2</sub> composite. XRD results confirm the formation of the CeO<sub>2</sub>/MoS<sub>2</sub> composite without any impurities. Additionally, prepared materials were characterized using Raman spectroscopy.



**Figure 1.** X-ray diffraction patterns of CeO<sub>2</sub>, MoS<sub>2</sub>, and CeO<sub>2</sub>/MoS<sub>2</sub> composite.

Raman spectroscopy is primarily used to characterize composite materials, and it also reveals information about types of bonding, symmetry, and structure in two-dimensional nanostructures [21]. Figure S1 presents the Raman spectrum of CeO<sub>2</sub>. From Figure S1, it is observed that the sharp and most intense characteristic peak was found at 442.48 cm<sup>-1</sup> (F2g band) which is associated with the symmetrical stretching mode of the oxygen atoms around each Ce cation [22]. Figure 2 presents Raman spectra of MoS<sub>2</sub> and the CeO<sub>2</sub>/MoS<sub>2</sub> composite. For MoS<sub>2</sub>, two major peaks positioned at 379.15 cm<sup>-1</sup> and 404.48 cm<sup>-1</sup> correspond to E1 2g and A1g modes respectively of pure MoS<sub>2</sub>. Out of these two modes, the E1 2g mode is associated with the in-plane opposite vibration of two S atoms, while the A1g mode is associated with the out-of-plane vibration of the S atoms in opposite directions with respect to the Mo atom. Peaks were observed at 820.55 and 993.22 cm<sup>-1</sup> due to the oxidation of MoS<sub>2</sub> by laser irradiation, leading to the formation of MoO<sub>3</sub>, which presents the vibrational energy of MoO<sub>3</sub> [23]. In the case of the CeO<sub>2</sub>/MoS<sub>2</sub> composite, CeO<sub>2</sub> peaks are denoted by the symbol \*. The E1 2g, A1g, and F2g modes correspond to the pure CeO<sub>2</sub> and MoS<sub>2</sub> samples. A small shift was detected in the CeO<sub>2</sub> peaks of the CeO<sub>2</sub>/MoS<sub>2</sub> composite due to an alteration or tension in the crystal lattice during the formation of the composite [24].

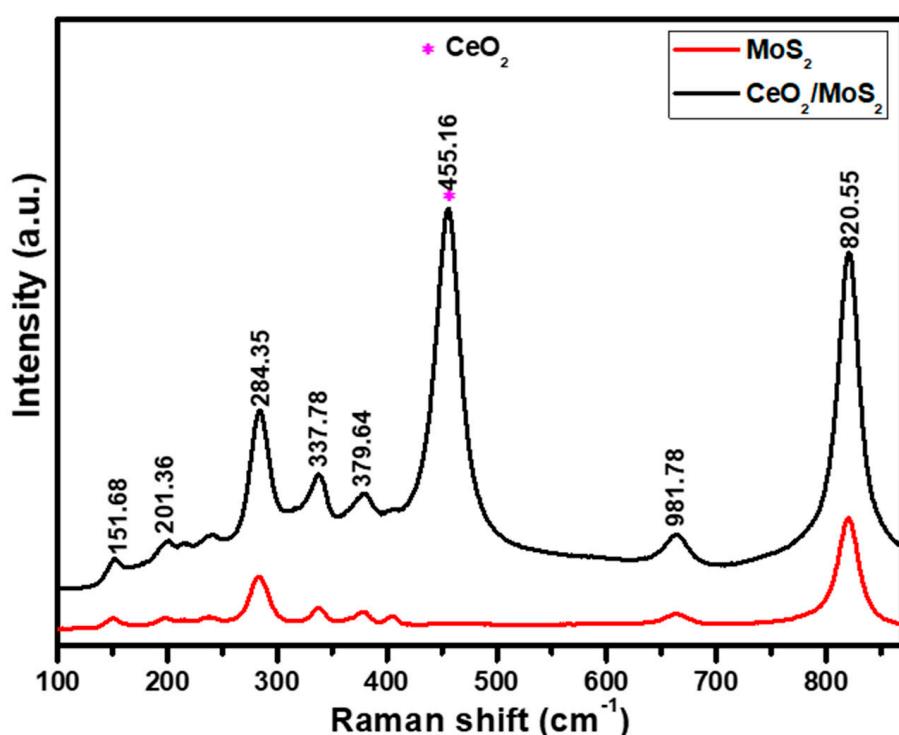
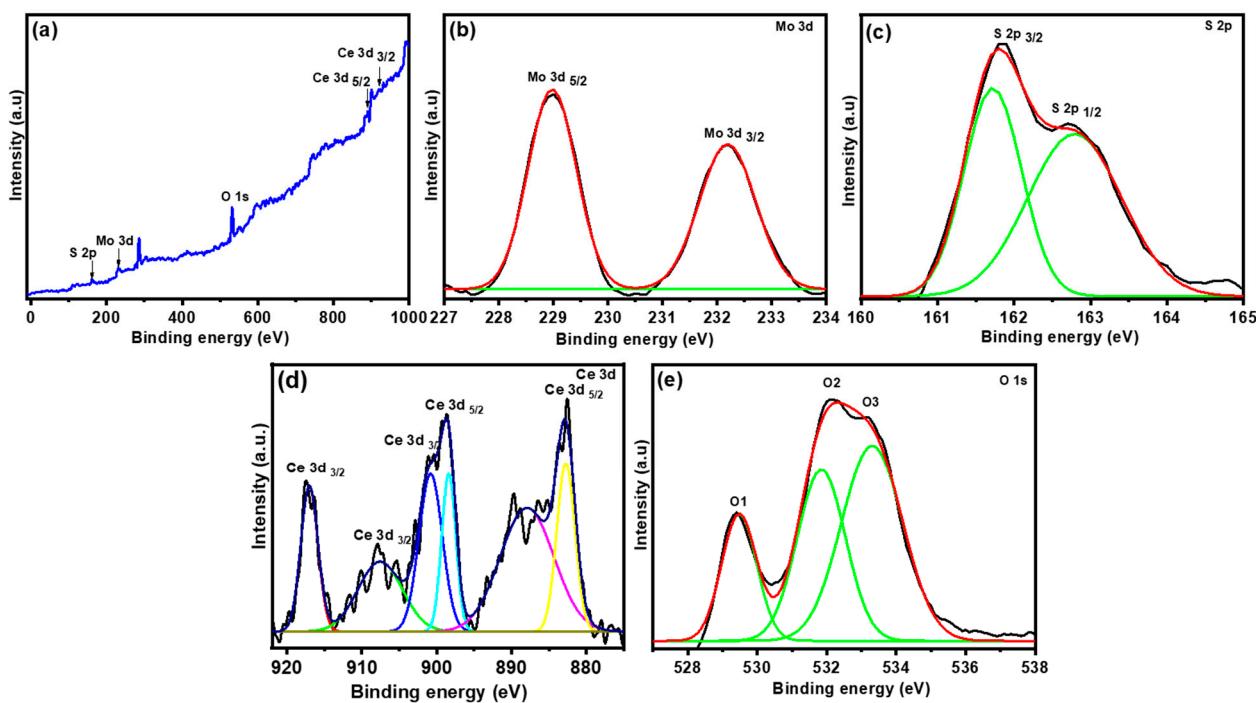


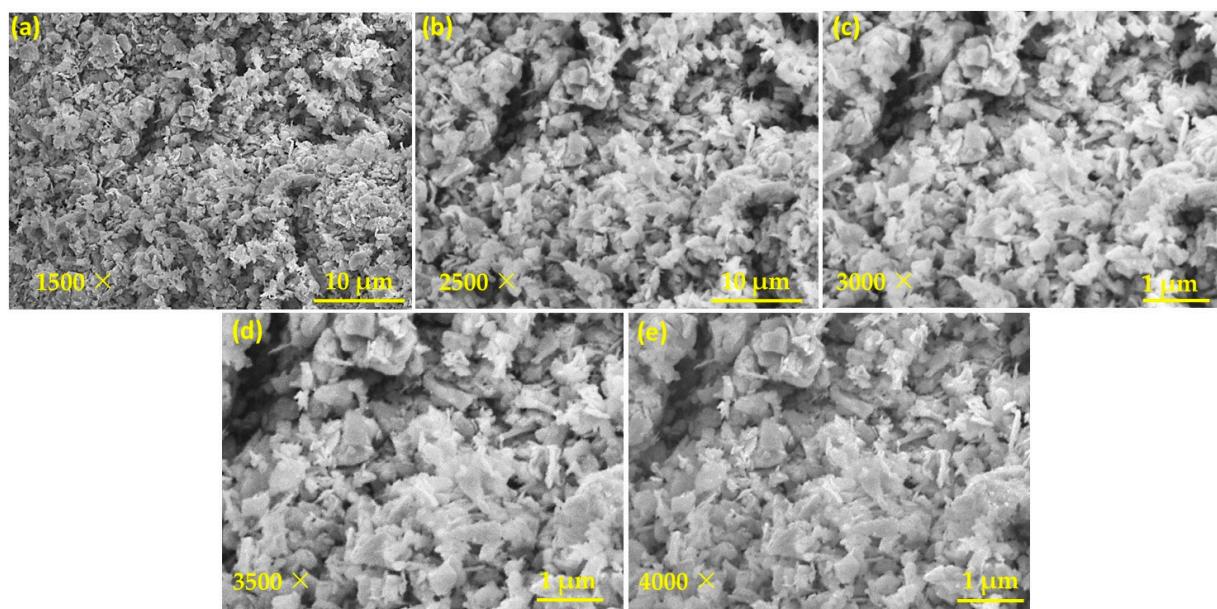
Figure 2. Raman spectra of  $\text{MoS}_2$  and  $\text{CeO}_2/\text{MoS}_2$  composite.

The  $\text{CeO}_2/\text{MoS}_2$  composite was further characterized by XPS. XPS study provides information about the chemical composition as well as the valence state of the  $\text{CeO}_2/\text{MoS}_2$  composite [25]. Figure 3a presents the XPS survey scan spectrum of the  $\text{CeO}_2/\text{MoS}_2$  composite. From the XPS survey scan spectrum, Mo 3d, S 2p, Ce 3d, and O 1s elements were detected. No other impurity elements were found. The Mo 3d high-resolution spectrum is presented in Figure 3b. The Mo 3d spectrum split into Mo 3d 5/2 and Mo 3d 3/2, which correspond to the binding energy of 229.05 and 232.20 eV, respectively, suggesting that Mo exists in a +4 oxidation state [26]. Figure 3c presents the S 2p spectrum, featuring two major bands with a binding energy of 161.85 and 162.79 eV, which correspond to S 2p 3/2 and S 2p 1/2, respectively [27]. Figure 3d shows the Ce (3d) XPS spectrum, which is deconvoluted into two sets. One set is in the range 880–900 eV, which is associated with Ce 3d 5/2, and the second one is in the range 900–920 eV, which corresponds to Ce 3d 3/2. Splitting of Ce 3d 5/2 and 3/2 reveals that Ce exists in the mixed valence state with oxidation states  $\text{Ce}^{3+}$  and  $\text{Ce}^{4+}$ . The main characteristic peaks of Ce at 900.05 and 882.02 eV belong to  $\text{Ce}^{3+}$  3d 3/2 and 3d 5/2, respectively. Along with this satellite, peaks observed at 907.1 eV correspond to  $\text{Ce}^{3+}$  3d 3/2, and those at 889.01 and 885.08 eV correspond to  $\text{Ce}^{3+}$  3d 5/2. In addition, three other satellite peaks were observed at 907.1 eV for  $\text{Ce}^{3+}$  3d 3/2 and at 888.7 and 885.1 eV for  $\text{Ce}^{3+}$  3d 5/2 [28]. The O 1s spectrum is shown in Figure 3e, and it is split into three major peaks with a binding energy of 529.40 eV (O1), 532.15 eV (O2), and 533.21 eV (O3), which correspond to the lattice, core level  $\text{O}_2$ , and S-O bonds, respectively [29].



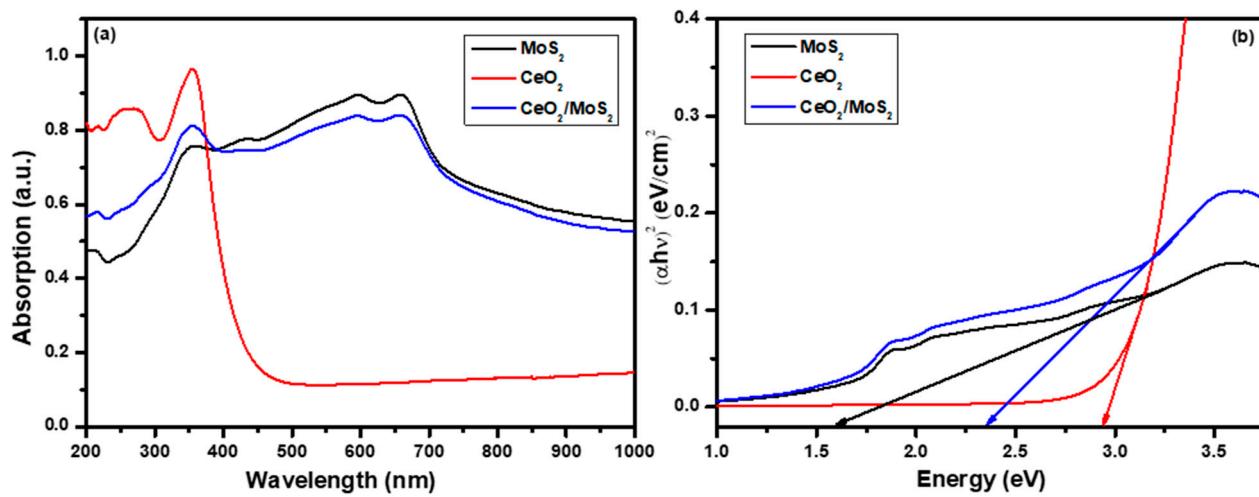
**Figure 3.** (a) X-ray photoelectron spectroscopy (XPS) spectrum of  $\text{CeO}_2/\text{MoS}_2$  composite, (b) Mo 3d spectrum, (c) S2 p spectrum, (d) Ce 3d spectrum, and (e) O 1s spectrum.

After structural characterization, a morphological study of the  $\text{CeO}_2/\text{MoS}_2$  composite catalyst was performed using scanning electron microscopy. Figure 4 shows the SEM images of the  $\text{CeO}_2/\text{MoS}_2$  composite at different magnifications. From the SEM images, it is observed that small flakes shaped  $\text{MoS}_2$  particles, and  $\text{CeO}_2$  particles have grains of different sizes and shapes (Figure 4a,b). Furthermore, a uniform distribution of both  $\text{CeO}_2$  and  $\text{MoS}_2$  in the particles is observed. At higher magnification ( $1 \mu\text{m}$ ), particles are clearly seen (Figure 4c–e).



**Figure 4.** SEM images of  $\text{CeO}_2/\text{MoS}_2$  composite at different magnifications (a)  $1500\times$ , (b)  $2500\times$ , (c)  $3000\times$ , (d)  $3500\times$ , and (e)  $4000\times$ .

An optical study is important while investigating the photocatalytic properties of MoS<sub>2</sub>, CeO<sub>2</sub>, and the CeO<sub>2</sub>/MoS<sub>2</sub> composite. Figure 5 shows the optical properties of prepared photocatalyst materials. Figure 5a presents the UV–Vis absorption spectra. CeO<sub>2</sub> shows the absorption edge at 460 nm and exhibits strong absorption in the ultraviolet region. For MoS<sub>2</sub>, the absorption edge is found at 875 nm, and it covers the visible region. The formation of the CeO<sub>2</sub>/MoS<sub>2</sub> composite caused the absorption edge of CeO<sub>2</sub> to shift towards the higher wavelength side due to the electronic interactions between the CeO<sub>2</sub> and MoS<sub>2</sub> grains through physical contact while making the composite [30]. The CeO<sub>2</sub>/MoS<sub>2</sub> composite covers both the visible and ultraviolet regions, which is beneficial for photocatalytic activity.



**Figure 5.** (a) Absorption spectra and (b) bandgap plots of MoS<sub>2</sub> and CeO<sub>2</sub>/MoS<sub>2</sub> composite.

Using the absorption data, the bandgap values were estimated. Figure 5b shows the plot of  $(\alpha h\nu)^2$  vs.  $h\nu$  for MoS<sub>2</sub>, CeO<sub>2</sub>, and the CeO<sub>2</sub>/MoS<sub>2</sub> composite. Using the following equation, bandgap energy, i.e., the energy difference between the conduction and valence bands, was calculated [31]:

$$\alpha h\nu = A(h\nu - E_g)^n \quad (1)$$

where A is a constant,  $E_g$  is the difference between the bottom of the conduction band and the top of the valence band,  $h\nu$  is the photon energy, and n is the order. Bandgap values were found to be 1.60, 2.93, and 2.34 eV for MoS<sub>2</sub>, CeO<sub>2</sub>, and the CeO<sub>2</sub>/MoS<sub>2</sub> composite respectively. Making the composite was beneficial for enhancing light absorption and reducing the bandgap energy of CeO<sub>2</sub>, suggesting a change in the electronic structure of CeO<sub>2</sub>. Along with this, it is essential to understand the band positions of CeO<sub>2</sub> and MoS<sub>2</sub>. The following equations were used to calculate the band positions [10],

$$E_{VB} = X - E_e + 0.5 E_g \quad (2)$$

$$E_{CB} = E_{VB} - E_g \quad (3)$$

where X represents the electronegativity of the catalyst material,  $E_{VB}$  is valence band potential,  $E_{CB}$  is conduction band potential,  $E_g$  is the bandgap energy, and  $E_e$  is the free electron energy (4.5 eV on the hydrogen scale). The electronegativity values for CeO<sub>2</sub> and MoS<sub>2</sub> were calculated as 5.56 eV and 5.32 eV, respectively [32]. Calculated values for CB potential for CeO<sub>2</sub> and MoS<sub>2</sub> were  $-0.41$  and  $0.02$  eV, and VB potential values were  $2.52$  and  $1.62$  eV, respectively. Table S1 (Supplementary Materials) presents the bandgap values and the CB and VB potentials (band positions) of MoS<sub>2</sub>, CeO<sub>2</sub>, and the CeO<sub>2</sub>/MoS<sub>2</sub> composite.

The specific surface area and pore size distribution are important parameters while studying photocatalytic activity. To study these surface properties, nitrogen adsorption and

desorption isotherm studies were performed for the CeO<sub>2</sub>/MoS<sub>2</sub> composite (Figure S2a,b). The CeO<sub>2</sub>/MoS<sub>2</sub> composite has H3 hysteresis-like loop and type IV adsorption and desorption, confirming its porous nature [33]. In Figure S2a, the calculated active surface area for the CeO<sub>2</sub>/MoS<sub>2</sub> composite is 48.72 m<sup>2</sup> g<sup>-1</sup>. The BJH model was used to study the pore size distribution of the CeO<sub>2</sub>/MoS<sub>2</sub> composite (Figure S2b). From the graph, the pore size was found to be ~6.5 nm. This study shows that the porous nature of the CeO<sub>2</sub>/MoS<sub>2</sub> composite, which is useful for redox reactions and thereby photocatalytic study [34].

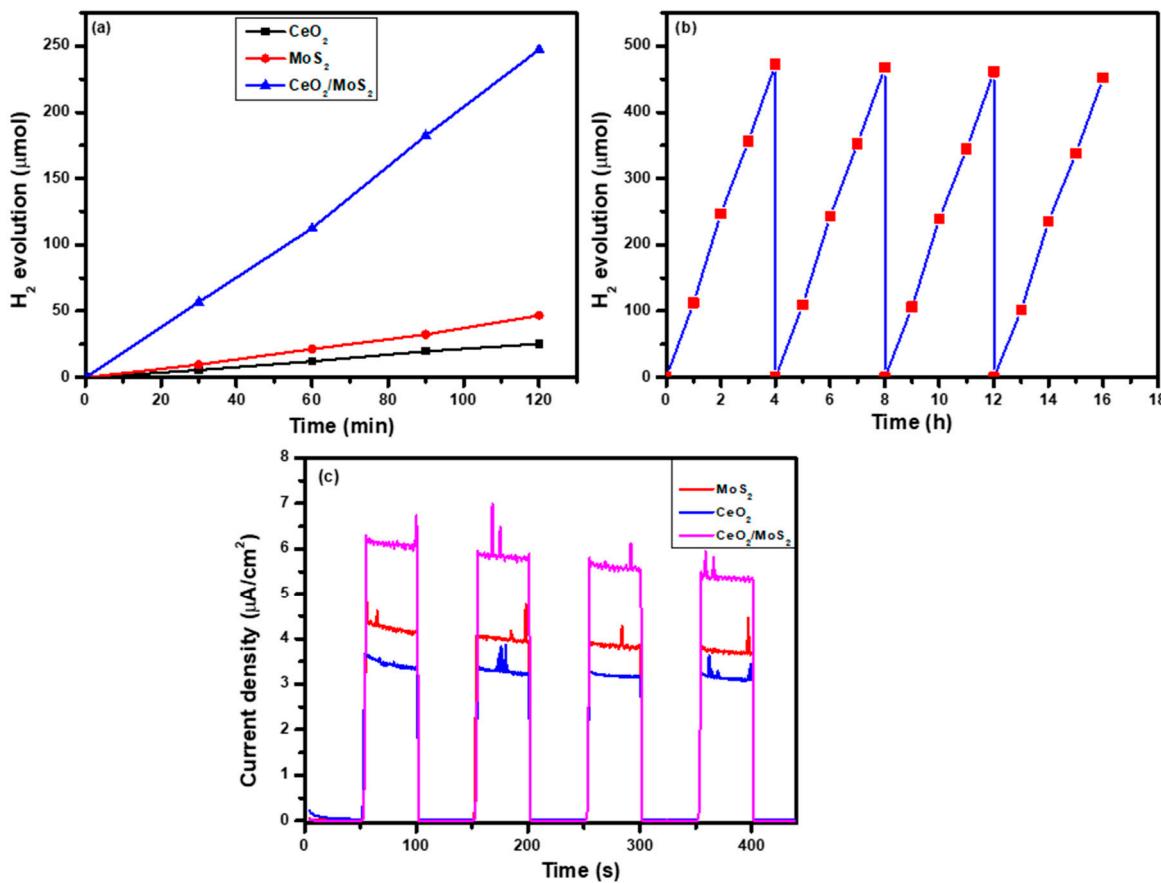
### 3. Photocatalytic Hydrogen Evolution Activity

The photocatalytic hydrogen evolution activity of MoS<sub>2</sub>, CeO<sub>2</sub>, and the CeO<sub>2</sub>/MoS<sub>2</sub> composite was measured under illumination. The yield of the hydrogen evolution rate of all the samples increased with the irradiation time increasing from 0 to 2 h. Figure 6a presents a plot of hydrogen evolution vs. time. For CeO<sub>2</sub>, MoS<sub>2</sub> and the CeO<sub>2</sub>/MoS<sub>2</sub> composite, the hydrogen production rates were 12.2, 21.5, and 112.5 μmol/h, respectively. The CeO<sub>2</sub>/MoS<sub>2</sub> composite exhibits superior photocatalytic activity towards hydrogen production as compared to CeO<sub>2</sub> and MoS<sub>2</sub>. Higher hydrogen production activity of the CeO<sub>2</sub>/MoS<sub>2</sub> composite sample corresponds to the effective charge generation, separation, and transfer, along with more active sites available for redox reactions and minimum charge recombination [10,35]. Active redox sites are situated at the edges and corners of the MoS<sub>2</sub> plane, helping to improve the hydrogen production rate of CeO<sub>2</sub>. Recycling tests of the CeO<sub>2</sub>/MoS<sub>2</sub> composite were performed to further study its stability. Figure 6b presents the recycling study of the CeO<sub>2</sub>/MoS<sub>2</sub> composite to observe its photostability. Recycling tests were performed with up to four cycles, each cycle being four hours. From the graph, it is observed that the CeO<sub>2</sub>/MoS<sub>2</sub> composite exhibits adequate photostability. For the first cycle, 472.5 μmol hydrogen evolution was detected. After the fourth cycle, this value was. A slight decrease in the hydrogen production rate was observed after the fourth cycle. The loss of the photocatalyst during the recovery process and the generation of the hydroxide layer on the catalyst surface are responsible for the decrease in H<sub>2</sub> production activity [36]. In addition, the photo-corrosion of photocatalysts is also responsible for the decrease in the H<sub>2</sub> production rate.

A possible mechanism of hydrogen production is proposed using the CeO<sub>2</sub>/MoS<sub>2</sub> composite. Under illumination, the CeO<sub>2</sub>/MoS<sub>2</sub> composite was easily excited, resulting in the generation of electrons and holes in the VB and CB respectively. The photo-generated electrons in the conduction band of CeO<sub>2</sub> can be transferred into the conduction band of MoS<sub>2</sub> because the conduction band potential of MoS<sub>2</sub> was more positive than that of CeO<sub>2</sub>. This type of composite can improve the lifetime of charge carriers, along with charge separation efficiency and the fast transfer of charge carriers from the CB of CeO<sub>2</sub> to the CB of MoS<sub>2</sub> for hydrogen evolution [37,38]. The CB electrons of MoS<sub>2</sub> exhibit strong reducibility and can easily reduce H<sup>+</sup> to H<sub>2</sub>, which agrees well with the results from the experiments of H<sub>2</sub> evolution. Furthermore, the role of MoS<sub>2</sub> as a co-catalyst is highly active for H<sub>2</sub> evolution as a result of the quantum confinement effect [39–41]. A comparison of photocatalytic hydrogen production activity using MoS<sub>2</sub>-based photocatalysts is presented in Table S2.

A photo-response study gives evidence regarding the highest current density and charge recombination process in the prepared materials. Figure 6c presents the transient chopping photocurrent responses of MoS<sub>2</sub>, CeO<sub>2</sub>, and the CeO<sub>2</sub>/MoS<sub>2</sub> composite deposited onto FTO-coated glass substrates, checked every 50 s of on/off cycles at an applied voltage of 0 V vs. Ag/AgCl. For all samples, under illumination, a swift photocurrent response was detected. The study revealed that the CeO<sub>2</sub>/MoS<sub>2</sub> composite exhibits a higher current density than pure MoS<sub>2</sub> and CeO<sub>2</sub>. The CeO<sub>2</sub>/MoS<sub>2</sub> composite sample achieved the highest current density of 6.15 μA/cm<sup>2</sup>, which is nearly double that of CeO<sub>2</sub>. The higher current density of the CeO<sub>2</sub>/MoS<sub>2</sub> composite sample corresponds to the appropriate band positions for effective charge transfer [42]. For CeO<sub>2</sub> and MoS<sub>2</sub>, the lowest current density values

of 3.40 and 4.35  $\mu\text{A}/\text{cm}^2$  were observed due to higher recombination of photo-generated charge carriers.



**Figure 6.** Photocatalytic  $\text{H}_2$  production by  $\text{CeO}_2$ ,  $\text{MoS}_2$ , and  $\text{CeO}_2/\text{MoS}_2$  composite: (a) recycling tests for photocatalytic  $\text{H}_2$  production; (b)  $\text{H}_2$  evolution in four consecutive 4 h cycles using  $\text{CeO}_2/\text{MoS}_2$  composite; (c) photocurrent density vs. time test.

#### 4. Experimental Details

**Materials:** Sodium hydroxide ( $\text{NaOH}$ , 99.8%), cerium hexahydrate nitrate ( $\text{Ce}(\text{NO}_3)_6 \cdot 6\text{H}_2\text{O}$ , 99%), and  $\text{MoS}_2$  powder (flakes, 98.8%) were all purchased from Sigma-Aldrich. Deionized water (D.W.) was used during the experiment. Analytical-grade (A.R.) chemical reagents were used in the experiment.

##### 4.1. Synthesis of Cerium Oxide

$\text{CeO}_2$  was prepared using the hydrothermal method. A certain amount of  $\text{Ce}(\text{NO}_3)_6 \cdot 6\text{H}_2\text{O}$  was dissolved in 40 mL D.W.  $\text{NaOH}$  (5 M) solution was added dropwise to the solution. Then, the reaction mixture was transferred into a 50 mL Teflon-lined autoclave. The hydrothermal process was carried out at 120  $^\circ\text{C}$  for 24 h. After the completion of the hydrothermal process, the formed product was filtered out, washed with D.W., and calcinated at 300  $^\circ\text{C}$ .

##### 4.2. Synthesis of $\text{CeO}_2/\text{MoS}_2$ Composite

To synthesize the  $\text{CeO}_2/\text{MoS}_2$  composite, the  $\text{CeO}_2$  solution was prepared as mentioned above. A certain amount of  $\text{MoS}_2$  powder was put into a distilled water-ethanol mixture. To achieve better dispersion of  $\text{MoS}_2$  flakes in the water-ethanol mixture, it was sonicated for 30 min. Finally, the two reaction solutions were mixed well. The reaction solution was kept in a 50 mL hydrothermal reactor at 120  $^\circ\text{C}$  for 24 h in an oven.

#### 4.3. Characterizations

Crystal structures of the prepared materials were analyzed using X-ray diffraction (XRD; CuK $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) from a Bruker D2 Phaser, Germany). Raman spectra were recorded in the range 100–875 cm $^{-1}$  using the NRS-5100 instrument of the Japan Spectroscopic Company (JASCO, Tokyo, Japan) with an excitation wavelength of 532 nm. Elemental composition was analyzed by X-ray photoelectron spectroscopy (XPS, Thermo Scientific, Waltham, MA, USA. K-Alpha set up using monochromatic Al K $\alpha$  X-ray source). The morphology of the prepared material was observed using the SEM instrument JSM-7600F of the Japan Electron Optics Laboratory (JEOL, Tokyo, Japan). Optical properties were studied using UV–Vis absorption spectroscopy (Shimadzu: UV-1800, Kyoto, Japan). The measurement of the BET-specific surface area of the photocatalysts was carried out using a nitrogen adsorption instrument at 77 K (Micrometrics, Norcross, GA, USA, ASAP 2020).

#### 4.4. Photocatalytic H<sub>2</sub> Production Study

A photocatalytic H<sub>2</sub> evolution study was conducted in a 100 mL sealed quartz batch reactor. A 150 W xenon arc lamp (>400 nm) was used as a light source. First, 50 mg of the catalyst was dispersed into a 100 mL aqueous solution containing 0.3 M Na<sub>2</sub>SO<sub>3</sub>/Na<sub>2</sub>S. Before light irradiation, N<sub>2</sub> gas was purged in a reaction mixture for 30 min through the reactor to remove the dissolved oxygen. After a particular interval of time, the quantity of gas was collected through a gas syringe. The collected gas sample was analyzed by gas chromatography (GC; GC-2014AT, Shimadzu, Japan) to observe the gas composition and quantity. Recycling (stability) tests were conducted using the same experimental conditions. In the recycling experiment, after each cycle, the catalyst powder was collected, washed with DI water, and reused. A total of four cycles were performed, with each cycle being 4 h.

### 5. Conclusions

Visible light-activated photocatalyst materials such as CeO<sub>2</sub>/MoS<sub>2</sub> composite can be successfully synthesized by the hydrothermal route. XRD, Raman spectroscopy, and XPS results confirmed the formation of a CeO<sub>2</sub>/MoS<sub>2</sub> composite without any impurities. An optical study revealed that the bandgap of CeO<sub>2</sub> reduced from 2.93 eV to 2.34 eV upon forming a composite with MoS<sub>2</sub>. The photocurrent response study showed that the CeO<sub>2</sub>/MoS<sub>2</sub> composite sample achieved the highest current density of 6.15  $\mu\text{A}/\text{cm}^2$ . The study of photocatalytic hydrogen production activity revealed that the CeO<sub>2</sub>/MoS<sub>2</sub> composite exhibits superior photocatalytic activity compared to pure MoS<sub>2</sub> and CeO<sub>2</sub>. For MoS<sub>2</sub>, CeO<sub>2</sub>, and the CeO<sub>2</sub>/MoS<sub>2</sub> composite, the hydrogen production rates were found to be 12.2, 21.5, and 112.5  $\mu\text{mol}/\text{h}$ , respectively. The enhanced photocatalytic activity of the CeO<sub>2</sub>/MoS<sub>2</sub> composite is attributed to its effective photo-response and charge generation, separation, and transfer, as well as its minimal recombination of photogenerated charge carriers. Thus, the CeO<sub>2</sub>/MoS<sub>2</sub> composite is a suitable photocatalytic material to support environmental remediation.

**Supplementary Materials:** The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/catal12101185/s1>, Figure S1. Raman spectrum of CeO<sub>2</sub>, Figure S2. (a) BET surface area and (b) pore size distribution curves of CeO<sub>2</sub>/MoS<sub>2</sub> composite, Table S1. Bandgap values of prepared photocatalyst materials with CB and VB potentials, Table S2. A comparison table showing photocatalytic hydrogen production activity using MoS<sub>2</sub>-based photo-catalysts. The references [40,43–45] are cited in the supplementary materials.

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**Data Availability Statement:** All data included in this study are available upon request by contact with the corresponding author.

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**Conflicts of Interest:** The authors declare no conflict of interest.

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