

Review

Recent Advances in the Catalyst Design and Mass Transport Control for the Electrochemical Reduction of Carbon Dioxide to Formate

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Received: 6 July 2020; Accepted: 29 July 2020; Published: 2 August 2020



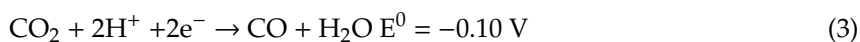
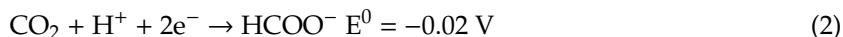
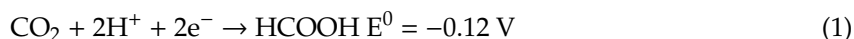
Abstract: Closing the carbon cycle by the electrochemical reduction of CO₂ to formic acid and other high-value chemicals is a promising strategy to mitigate rapid climate change. The main barriers to commercializing a CO₂ reduction reaction (CO₂RR) system for formate production are the chemical inertness, low aqueous solubility, and slow mass transport characteristics of CO₂, along with the low selectivity and high overpotential observed in formate production via CO₂ reduction. To address those problems, we first explain the possible reaction mechanisms of CO₂RRs to formate, and then we present and discuss several strategies to overcome the barriers to commercialization. The electronic structure of the catalyst can be tuned to favor a specific intermediate by adjusting the catalyst composition and tailoring the facets, edges, and corners of the catalyst to better expose the active sites, which has primarily led to increased catalytic activity and selectivity. Controlling the local pH, employing a high-pressure reactor, and using systems with three-phase boundaries can tune the mass transport properties of reactants at the catalyst surface. The reported electrocatalytic performances are summarized afterward to provide insight into which strategies have critical effects on the production of formate.

Keywords: electrochemical reduction; carbon dioxide; formate production

1. Introduction

The carbon dioxide concentration in the atmosphere has been rapidly increasing, which has become a major cause of global warming via the greenhouse effect [1]. The global CO₂ atmospheric concentration was approximately 409 ppm in 2014; the concentration had risen for about 120 ppm over the past 250 years and is likely to reach 1000 ppm by the end of this century [2]. To tackle this climate change issue, the Paris Agreement reached a commitment to keep the global average temperature increase to less than 2 °C above the preindustrial levels [3]. The CO₂ reduction reactions (CO₂RRs) have been proposed as a promising strategy to reduce the greenhouse gas emissions while storing the energy in a value-added chemical [4]. Electrochemical reduction of CO₂ can produce various types of product depending on the material of the electrode, solvent types, local pH on the catalysts surface, and CO₂ pressure [5,6]. Formic acid or formate is a promising CO₂RR product because of its wide applications, including leather tanning and animal feed markets [7]. Formic acid is also a good hydrogen storage material because of its high hydrogen density of 52 g of H₂ per liter of formic acid [8]. Furthermore, formic acid can be easily decomposed catalytically to CO₂ and H₂ at room temperature [9], and it can be utilized directly as the fuel for a direct formic acid fuel cell without pretreatment [10]. The electrochemical reaction equations and equilibrium potentials (V vs. reversible

hydrogen electrode (RHE)) for the reduction of CO₂ to formic acid/formate and the major competing reactions are presented as reactions (1)–(4) [11–13]:



The market size of formic acid was about 950 kilotons per year in 2014 and will grow to 1 megaton per year in 2030 [14]. Approximately, 90% of the existing production capacity generates formic acid by methyl formate hydrolysis [15,16]. According to a technoeconomic analysis by Verma et al. using the gross-margin model, for a Sn cathode catalyst with an assumption of 100% faradaic efficiency (FE) for formate production, a gross margin of 30% requires a minimum formate partial current density of 56 mA cm^{−2} and a catalyst durability of 4000 h at a cell potential of 4 V [17]. These technoeconomic reports suggest that practical implementation is closely tied to improvements in crucial electrocatalytic reaction parameters, such as FE, current density, and overpotential.

CO₂ is a stable molecule with an electrophilic carbon center and a linear molecular geometry. Therefore, the first electron transfer to form the CO₂[−] intermediate requires a very negative potential of −1.49 V vs. RHE [18]; this thermodynamic barrier makes CO₂RRs difficult. Hydrogen evolution is a major competing reaction, which lowers the FE_{HCOOH}. Notably, transition metals with relatively strong binding affinities for the hydrogen intermediate tend to show a marked tendency towards the hydrogen evolution reaction (HER) because its limiting potential is more positive than that of the CO₂RR to HCOOH [19]. Further, mass transport issues limit the partial current density of formate production [20]. The low aqueous solubility and slow mass transfer of CO₂ result in a low CO₂ concentration at the catalyst interface during electrolysis in conventional H-type cells [21], which restricts the partial current density of formate production to low levels.

A correlation has been reported between the binding energy of *HCOO, which is the main intermediate, and the activity of the catalyst in formate production [22]. Specifically, consistent with the Sabatier principle, a higher formate partial current density is achieved with a Sn catalyst with a moderate *HCOO binding affinity than with transition metals with much stronger or much weaker *HCOO binding affinities, including Ni, Cu, Pt, Ag, and Au [23]. The nature of the intermediate binding energy can be tailored by synthesizing a bimetallic or doped catalyst to change the electronic structure from that of the pristine catalyst [24–26]. Tuning the intermediate binding energy grants control over the selectivity and the overall reaction pathway as a result [27]. Exposing high-energy facets and increasing the number of edge, corner, and grain boundary (GB) sites have also been reported to improve catalyst activity [28–30]. These nanostructuring approaches ease the adsorption of the reaction intermediates and intensify the local electric field, which ultimately increase the formate production [28,31]. Meanwhile, the mass transport properties of CO₂ have been improved by raising the pressure of the reaction vessel to increase the CO₂ solubility and by employing three-phase-boundary reactors in which gaseous CO₂ is directly transported to the catalyst surface through a gas diffusion layer (GDL) [32,33].

In this review, we first explain the reaction mechanisms that have been reported for electrochemical formate production. We then proceed from this deeper understanding to ideas for optimizing important electrocatalytic performance parameters, such as FE, current density, and overpotential. The reaction mechanisms and electrocatalytic performances primarily depend on the types of the catalyst materials and the intrinsic natures of their intermediate binding energies. Accordingly, we review the strategies for modifying the binding energies and thereby the catalytic properties of the materials, which can be classified as tailoring the facets, edges, and corners of the catalyst to expose the active sites or adjusting the catalyst composition to tune its electronic properties. Next, the strategies for

improving CO₂ mass transport are outlined, which include local pH control, high-pressure reactors, and three-phase-boundary electrodes. Finally, the performances of the reported materials are compared at the end of this review to provide clear insight into which strategies have critical effects on formate production performance.

2. Reaction Pathways

The ability of a catalyst to adsorb and desorb specific reaction intermediates determines the main reaction pathway, along with the overpotential that is needed to overcome the energy barrier to that pathway. Therefore, a deep understanding of the CO₂RR mechanism will give researchers the necessary knowledge to design more efficient catalysts [34,35]. In this section, we will discuss four reported reaction pathways for HCOOH/HCOO[−] production: (1) CO₂ insertion, (2) an O-bound intermediate, (3) a C-bound intermediate, and (4) a bicarbonate intermediate.

The first probable mechanism for HCOOH production is CO₂ insertion into the metal-H bond and formation of the *HCOO intermediate (Figure 1a) [36]. HCOOH is subsequently obtained via *HCOO reduction. This mechanism occurs on Pd-based catalysts. The strong hydrogen binding affinity of Pd results in hydrogen absorption into the Pd lattice and the formation of β-phase PdH_x, if the Pd catalyst is held at potentials of <0 V vs. RHE [37]. The surface hydride will reduce the CO₂ and form the *HCOO intermediate. Formate has been produced by this reaction mechanism at a very low potential of 0 V vs. RHE in HCO₃[−] solutions [38,39].

The second reaction pathway, which involves an O-bound intermediate, starts with the first electron transfer to CO₂ and the formation of a weakly adsorbed CO₂[−] radical intermediate (Figure 1b). Formic acid is produced from the transfer of protons and electrons from the reaction between the CO₂[−] radical and proton donors like water, bicarbonate, and hydronium ions [36,40]. Li et al. reported that Sn foil follows this mechanism and can produce formate with an FE of 63.6% and a partial current density of 3.11 mA cm^{−2} at −1.01 V vs. RHE [41]. Metals such as In, Pb, Hg, Sn, and Bi are reported to follow the O-bound-intermediate pathway due to the easier formation of the *HCOO intermediate compared to *COOH [19,42]. This group produces formate with high selectivity overall, although Sn and In produce a small amount of CO.

In the C-bound-intermediate pathway, the CO₂[−] radical is formed and then bounded to the catalyst surface via the C atom (Figure 1c). The CO₂[−] radical next reacts with H⁺ to form an adsorbed *COOH. HCOO[−]/HCOOH can then be produced from *COOH. According to Sullivan et al., the *COOH form of the intermediate for HCOOH production is unstable and tends to either decompose into M-H + CO₂ or lose OH[−] through a nucleophilic attack to form the M-C = O⁺ intermediate [43]. However, appropriate metal catalysts can stabilize the *COOH intermediate by an isomerization process shown in (5)–(7) that forms a more stable *HCOO intermediate and ultimately releases HCOO[−]. The number of proposed alternative reactions indicates that reactions will occur in parallel in the majority of systems [40,44].



Cu- and Ru-based catalysts produce formate via the C-bound-intermediate pathway because of their basic natures and because the bond formed by electron transfer from CO₂ to unoccupied metal orbitals is stabilized by back-donation from the *d* orbitals of the metal atom. Those metals are also able to produce other products, such as alcohols or hydrocarbons, depending on the reaction conditions [40,45]. A theoretical report by Yoo et al. stated that Cu(211) can produce formic acid at low overpotentials via the *COOH pathway due to its higher (more positive) limiting potential compared to the *HCOO pathway. Cu-based electrocatalysts have been reported to produce formate as the main product when applied to the anionic membrane of a membrane electrode assembly (MEA)

electrolyzer [46]. The same report stated that applying high pressure resulted in formate becoming the main product of the reaction on the Cu catalyst.

The bicarbonate-intermediate pathway starts with a reaction between adsorbed $^*\text{OH}$ and CO_2 to form the adsorbed bicarbonate (CO_3H^*) species (Figure 1d) [47,48]. The CO_3H^* intermediate reacts with H^+ and receives an electron to form $^*\text{HCOO}$ and $^*\text{OH}$. HCOOH is released after HCOO^- has reacted with another H^+ and received another electron. This reaction pathway has been reported to occur at the Bi–Sn interface with an onset potential of -0.7 V vs. RHE in 0.5 M KHCO_3 . The FE_{HCOOH} reaches 96% at -1.1 V vs. RHE [47]. PdSnO_2 and SnO_x surfaces were also reported to produce formate via the bicarbonate-intermediate pathway [49,50]. This reaction mechanism is likely to occur on catalysts with relatively high $^*\text{OH}$ binding strengths.

The intermediate binding affinity of a catalyst is associated with its product selectivity, the reaction overpotentials, and the mechanism, which implies that catalytic performance can be enhanced by controlling the intermediate binding affinity. In the following sections, we will discuss several strategies in detail, including tailoring the facet, edge, and corner sites of the catalyst and controlling the catalyst composition.

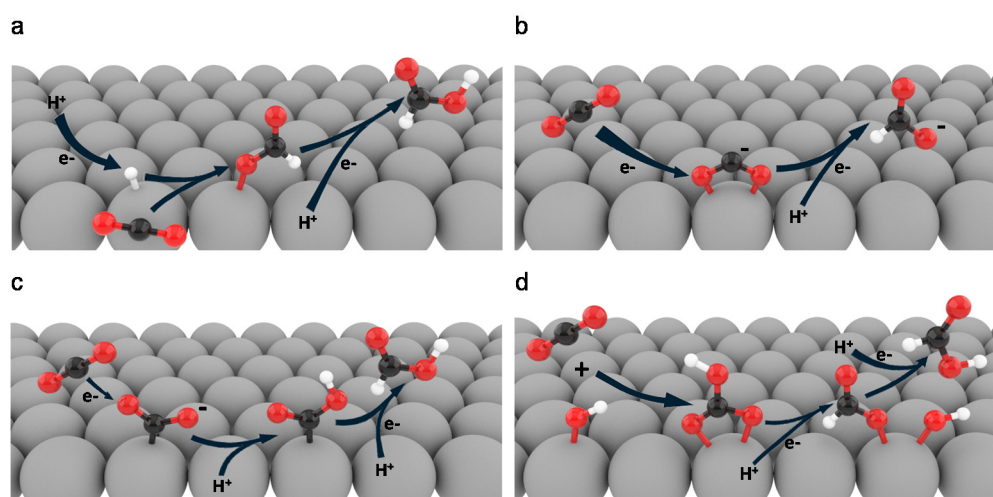


Figure 1. Reaction pathways for HCOOH (HCOO^-) production. (a) CO_2 insertion into a metal-H bond [36]; (b) CO_2 to HCOO^- via O-bound intermediate [36]; (c) CO_2 to HCOOH via C-bound intermediate [51,52]; and (d) HCOOH production from CO_3H^* [47]. Color codes: black, C; red, O; white, H; grey, metals.

3. Nanostructural Engineering

The geometrical features of a catalyst can greatly affect its CO_2 electroreduction performance. The exposed high-energy facets, edges, and corners will significantly enhance the catalytic activity towards CO_2 reduction [28,31,53,54]. The catalytic activity of a nanostructured catalyst is usually proportional to the number of under-coordinated sites [31]. Sites such as high-index facets have abundant under-coordinated atomic steps and exhibit enhanced catalytic activity compared to low-index nanoparticles [29,53]. Moreover, a large number of edge and corner sites enhances catalytic activity by facilitating a strong electric field at the catalyst surface to ease the adsorption of CO_2 [28,31].

Klinkova et al. reported the relationship between surface atom coordination and the catalytic performance of Pd nanoparticles (NPs) in CO_2 electroreduction to formate [53]. The authors first conducted density functional theory (DFT) calculations on Pd(111), Pd(110), Pd(100), and Pd(211) surfaces and a Pd_{19} cluster. As shown in Figure 2a, the authors found that the $\Delta G_{\text{formation}}$ of CO^* increased from -0.83 eV on Pd(111) to -0.65 eV on Pd(211), while the $\Delta G_{\text{formation}}$ of $^*\text{HCOO}$ decreased from 0.28 eV on Pd(111) to -0.09 eV on Pd(211). Therefore, the formation of $^*\text{HCOO}$ is made more favorable by incorporating higher-index facets. After obtaining these predictions from the DFT

calculations, the authors then synthesized {100} plane-enclosed nanocubes (NCs), {110} plane-enclosed rhombic dodecahedra (RDs), NPs with mixed low-index facets, and branched NPs (BNPs) enclosed by high-index facets. The synthesized NPs are shown in Figure 2c–g. Electrochemical characterization revealed that at the beginning of electrolysis, the total geometric current densities increased from 15 to 22 mA cm^{−2} at a −0.2 V overpotential. All geometric current densities decreased during the reaction due to CO poisoning of the catalyst surface. The measured rate of decrease of the current density was in agreement with the calculated results in which BNPs, which have the highest-index facets, have the lowest rate of decrease of the current density (Figure 2b).

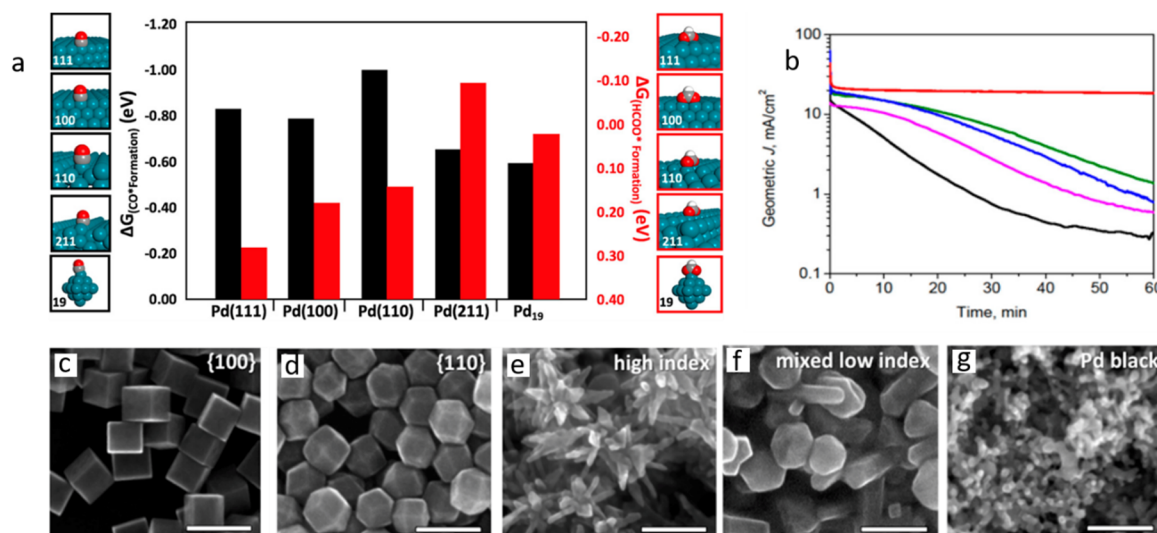


Figure 2. (a) Free energies of formation for *HCOO and CO^* intermediates on Pd(111), Pd(100), Pd(110), Pd(211), and Pd₁₉. The formation energy of the CO^* intermediate is reduced and the formation energy of the *HCOO intermediate is increased on higher-index surfaces; (b) geometric current densities of CO_2 electroreduction; scanning electron microscopy (SEM) images of Pd nanoparticles (NPs): (c) {100} plane-enclosed nanocubes (NCs), (d) {110} plane-enclosed rhombic dodecahedra (RDs), (e) branched nanoparticles (BNPs) enclosed by high-index facets, (f) NPs with mixed low-index facets, and (g) Pd black [53]. Copyright 2016, ACS.

Kim et al. reported the correlation between the number of edge and corner sites and the electrocatalytic performance of Bi NPs [31]. Sharper Bi nanoflakes had greater numbers of edge and corner sites. The authors used the electrodeposition pulse current (PC) method to synthesize nanostructured Bi nanoflakes. Figure 3a,b shows the samples fabricated with between 1 and 9 pulse cycles at 20 mA cm^{−2}. The samples at 1 and 3 pulse cycles showed FEs of 89% and 90%, respectively, and were not perfectly converted to Bi nanoflakes. The sample at 6 pulse cycles was completely converted to Bi nanoflakes and achieved an FE of over 100%. However, the sample at 9 pulse cycles showed a decrease in FE to 97% because Bi layers were covering the edge and corner sites, transforming the structure into a tripod shape. The prospects of electrostatic field intensification at the edges and corners were explored using a COMSOL Multiphysics simulation. A hexagonal Bi nanostructure was used as a representative three-dimensional structure with an edge length of 500 nm at various thicknesses (Figure 3c). Reducing the thickness from 500 to 10 nm enhanced the electrostatic electric field intensities at the edge and corner 3- and 5-fold, respectively. A two-dimensional Bi nanostructure was simulated to investigate the effect of sharpening the corners (Figure 3d); reducing the angle from 70 to 5° enhanced the electric field 2.5-fold. The electric field affects the overall reaction rate by increasing the local concentration of K^+ , which enhances CO_2 adsorption by lowering the free energy for the adsorption of intermediates [28,31].

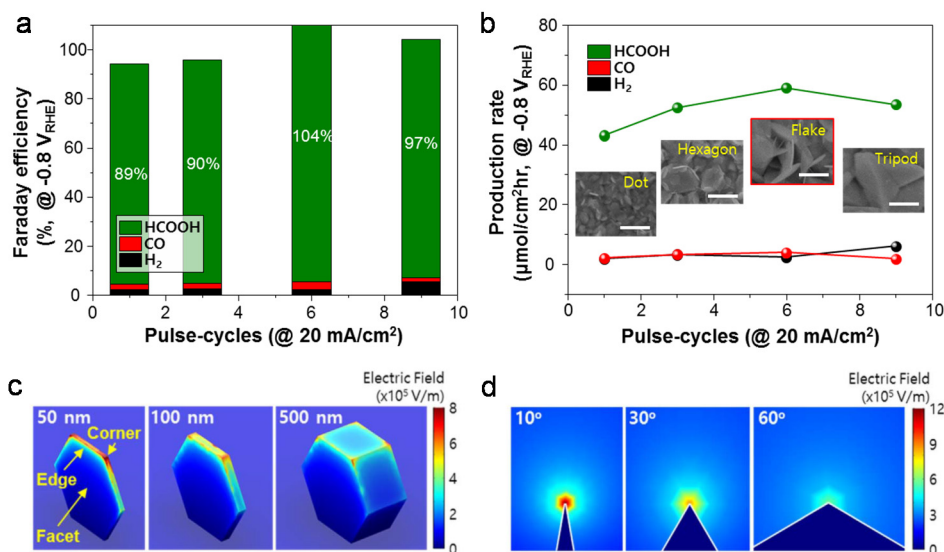


Figure 3. Performances of synthesized Bi nanostructures: (a) faradaic efficiency (FE) and (b) production rate of pulse-current-deposited Bi films obtained at -0.8 V vs. reversible hydrogen electrode (RHE) in CO_2 -purged 0.1 M KHCO_3 electrolyte. The insets in (b) are SEM images of pulse-current-deposited Bi films fabricated with different pulse cycles at 20 mA cm^{-2} . Simulated electric field distribution in (c) three-dimensional and (d) two-dimensional Bi nanostructures [31]. Copyright 2017, Elsevier.

Two-dimensional materials were reported to have higher selectivity and intrinsic activity than bulk materials because of the high exposure of the active facets. Additionally, these materials have high surface areas compared to bulk materials [55–58]. For example, Han et al. reported ultrathin Bi nanosheets (NSs; BiNS) with an exceptional FE of almost 100% and a larger partial current density than that of commercial Bi (Figure 4a,b) [56]. The ultrathin BiNS were prepared by in situ topotactic transformations of BiOI NSs. To support the study, DFT calculations were conducted via the computational hydrogen electrode methodology. These simulations were performed on the Bi(001) facet because it was the predominantly exposed facet of the BiNS (Figure 4c). The CO_2 reduction to formate was initiated by protonation of the C atom to form the $^*\text{HCOO}$ intermediate in a mildly endothermic reaction ($+0.49$ eV). The second proton-coupled electron transfer to HCOO^- was an exothermic reaction (-0.17 eV). Ultimately, HCOO^- was spontaneously released from the catalyst. Protonation of the O atom to form COOH^* , which was the intermediate for CO production, had a significantly higher energy barrier ($+1.16$ eV). The energy barrier for H adsorption into Bi(001) was also too high ($+0.95$ eV). The production of formate on Bi(001) was accordingly more favorable than the evolution of CO and H_2 .

Grain boundaries (GBs) enhance the activity of catalysts by providing highly active reaction sites [59]. Feng et al. pioneered the quantitative correlation analysis between the density of GBs and the catalytic activity of catalysts [60]. They found a linear correlation between the GB surface density and the specific activity for CO_2 RRs on vapor-deposited Au NPs on carbon nanotubes. Li et al. reported the catalytic performance of Bi NPs/ Bi_2O_3 NSs with abundant GBs. The catalyst, which was obtained by facile hydrothermal synthesis, was compared to low-GB-density Bi NSs/ Bi_2O_3 NSs. High-resolution transmission electron microscopy (HRTEM) images of the catalysts have shown the GB density difference between the two catalysts (Figure 5a,b). The Bi NPs/ Bi_2O_3 NSs exhibited a superior catalytic performance to that of Bi NSs/ Bi_2O_3 NSs (Figure 5c,d). The Bi NPs/ Bi_2O_3 NSs also showed a higher formate partial current density of 24.4 mA cm^{-2} , an FE_{HCOOH} of $>90\%$ over a broad potential range, and over 24 h of catalyst durability.

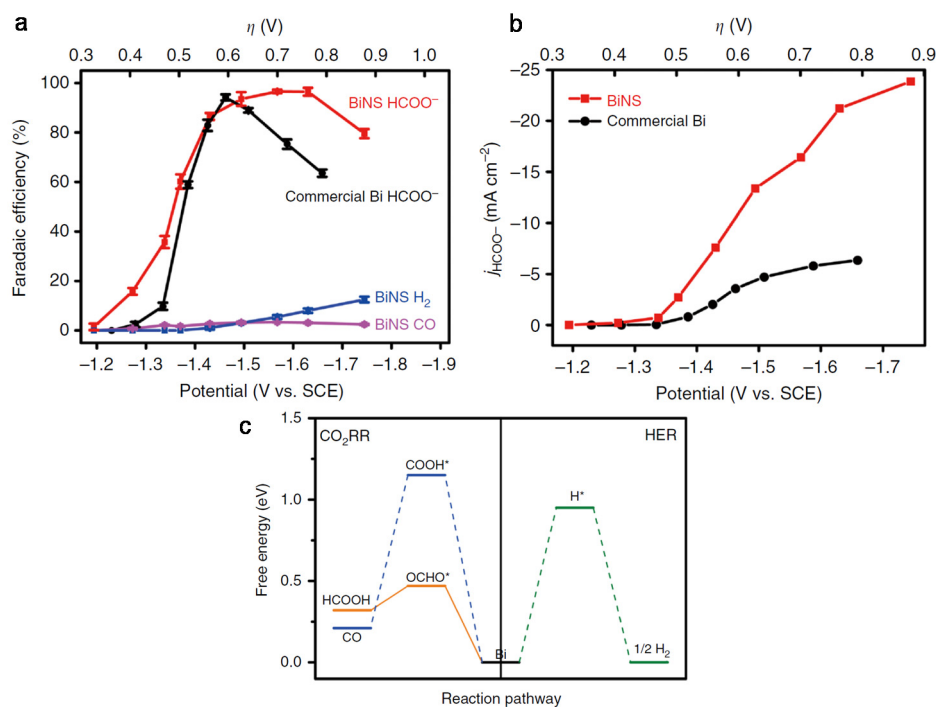


Figure 4. (a) Potential-dependent FEs of HCOO⁻, CO, and H₂ on Bi nanosheets (NSs; BiNS) in comparison with the FE of HCOO⁻ on commercial Bi nanopowder; (b) potential-dependent HCOO⁻ partial current density on BiNS and commercial Bi nanopowder; and (c) free-energy diagram for HCOO⁻, CO, and H₂ formation on the Bi(001) plane [56]. Copyright 2018, Nature.

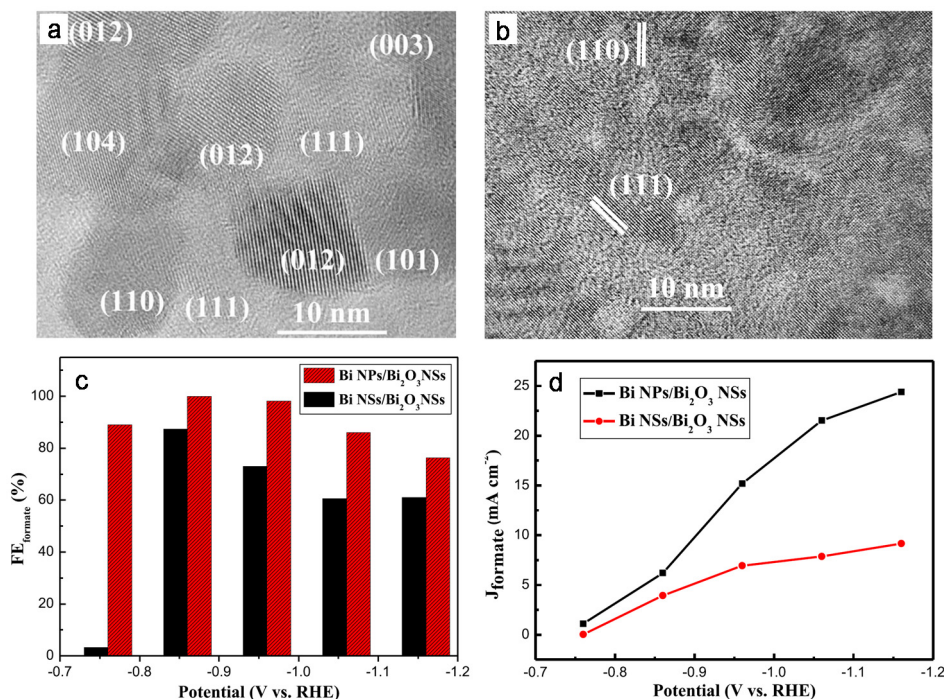


Figure 5. High-resolution transmission electron microscopy (HRTEM) images of (a) high-grain-boundary (GB)-density Bi NPs/Bi₂O₃ NSs and (b) low-GB-density Bi NSs/Bi₂O₃ NSs. (c) FE_{HCOOH} and (d) formate partial current density of Bi NPs/Bi₂O₃ NSs and Bi NSs/Bi₂O₃ NSs [61]. Copyright 2019, Elsevier.

4. Composition

4.1. Bimetallic Compounds

Controlling the composition of bimetallic compounds is an important strategy for tuning the electronic behavior of nanomaterials to enhance their selectivity and catalytic activity [24,62,63]. Bimetallic compounds can change the electronic structure of a catalyst, which affects the intermediate binding energy, which in turn controls the overall reaction pathway [27]. Many previously reported bimetallic compounds have shown enhanced catalytic activity compared to the corresponding pristine-metal-based catalysts, such as Ag–Sn [63], Cu–Bi [24,64], In–Sn [41], Sn–Cu [65,66], Pd–Ni [67], and Bi–Sn [47].

Wen et al. synthesized a Bi–Sn catalyst that had better selectivity for formate than a well-known Sn catalyst [47]. As-synthesized Bi–Sn on carbon fabric (CF) exhibited a dramatic current increase in CO₂-purged electrolyte (Figure 6a). At −1.14 V vs. RHE, Bi–Sn/CF had a superior FE_{HCOOH} of 94% ± 2% compared to Sn/CF, which had an FE_{HCOOH} of 78% ± 2% (Figure 6b). To support the experimental data, the authors conducted periodic DFT calculations on the reaction pathways to CO and HCOO[−] from the adsorption of CO₃H*. As shown in Figure 6c,d, the binding energy differences between the *COOH and *HCOO intermediates (ΔE_1) on Sn(101) and Bi–Sn(101) surfaces were 0.55 and 0.81 eV, respectively. The binding energy differences between the CO* and HCOOH* intermediates (ΔE_2) on Sn(101) and Bi–Sn(101) surfaces were 0.43 and 0.85 eV, respectively. The larger ΔE of the Bi–Sn catalyst indicates greater selectivity for formate, which agrees with the experimental results. In summary, tuning the composition of bimetallic compounds can help with the optimization of the intermediate binding energy of the catalysts to achieve higher catalytic selectivity and performance.

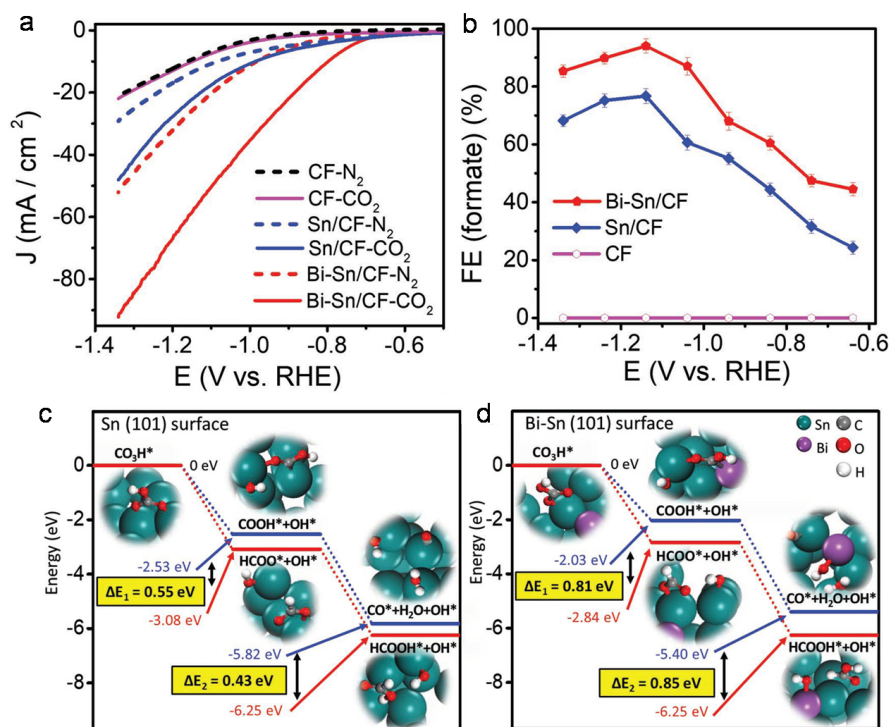


Figure 6. (a) CO₂ reduction reaction (CO₂RR) activities of prepared electrodes in N₂- (dotted line) or CO₂-purged (solid line) 0.5 M KHCO₃ electrolyte at a scan rate of 20 mV s^{−1}; (b) FE_{HCOOH} generated on electrodes at a series of potentials from −0.64 to −1.34 V vs. RHE; calculated reaction energy profiles for CO₂RRs to form CO (top) and HCOOH (bottom) on (c) Sn(101) surfaces and (d) Bi–Sn(101) surfaces. All energies are with reference to the energies of CO₃H adsorbed on Sn(101) or Bi–Sn(101) surfaces [47]. Copyright 2018, WILEY-VCH.

4.2. Doping Materials

Doping is another strategy to enhance the catalytic activities of electrocatalysts. The dopant affects the electronic properties of the host [66,68,69], which may allow the binding energy for a particular intermediate to be designed to facilitate the desired reaction pathway [70].

Pd-based catalysts are known to have almost zero overpotential to activate CO₂RRs to formate but also have a parallel CO pathway that competes with and may even deactivate formate production [37,71]. Bei Jiang et al. successfully increased the selectivity of a Pd-based catalyst for the production of formate with a high CO tolerance by doping the catalyst with boron [72]. The authors compared Pd–B/C to a control sample of Pd/C. The FE_{CO} increased as the potential became more negative on Pd-based catalysts. However, the FE_{CO} of Pd–B/C was much lower than that of Pd/C at any given potential (Figure 7a). The FE_{HCOOH} of Pd–B/C reached a maximum value of 70% at –0.5 V vs. RHE, but Pd/C had an FE_{HCOOH} of only 4.8% at the same potential. Furthermore, Pd/C had an FE_{HCOOH} close to zero at potentials lower than –0.7 V vs. RHE, whereas the Pd–B/C catalyst had an FE_{HCOOH} of 15–30% under those conditions. To gain a greater understanding of the promoting effect of B doping on formate production on Pd, the authors conducted DFT calculations on Pd(111) with and without B doping. Figure 7c,d present the Gibbs free energy diagrams for the intermediates of the CO and HCOOH pathways. The CO production pathway has a lower energy barrier than the HCOOH production pathway on the Pd(111) surface. In contrast, HCOOH production has the lower energy barrier on Pd(111)–4B, making it the more favorable pathway. The effects of B doping on the subsurface of the Pd lattice interstices include unique electronic behavior owing to partial electron transfer between Pd and B as well as a downshifted *d*-band center for the surface Pd. Thus, the new electronic structure leads to a more negative adsorption energy for *HCOO than for *COOH.

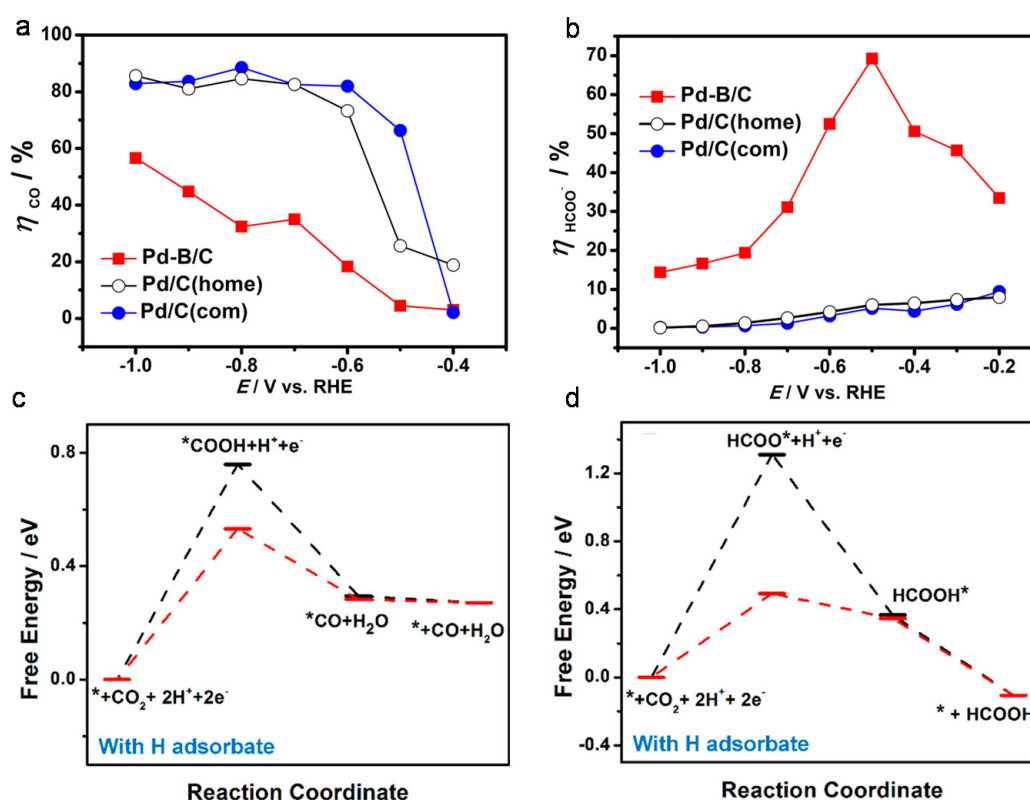


Figure 7. Potential-dependent FEs of: (a) CO and (b) formate; Gibbs free energy diagram for CO₂RRs on Pd(111) (black) and Pd(111)–4B (red) for the (c) CO pathway and (d) HCOOH pathway [72]. Copyright 2018, ACS.

A dopant can also affect the reaction rate by interacting with hydrated cations. For example, Ma et al. proposed that [73] the hydrated potassium cation ($K^+(H_2O)_n$) could form networks with the $S^{\delta-}$ anions of sulfur-doped indium in the double layer through non-covalent Coulomb interactions (Figure 8a). These interactions can enhance the dissociation of H_2O to form the adsorbed hydrogen intermediate ($*H$) that is responsible for the formation of the $*HCOO$ intermediate, which is the precursor of formic acid. The authors synthesized S2-In $_2$ O $_3$ -derived In/C (4.9 mol % S), along with a control sample of S0-In $_2$ O $_3$ -derived In/C (0 mol% S). The experimental results showed an enhancement of catalytic activity in the presence of S, with FE_{HCOOH} values of 89% and 93% and current densities of 37 and 84 mA cm $^{-2}$ (−0.98 V vs. RHE) for S0-In $_2$ O $_3$ -derived In/C and S2-In $_2$ O $_3$ -derived In/C, respectively (Figure 8b). The DFT calculation results indicated that the presence of S on the In surface decreased the Gibbs free energies for $*HCOO$ and $HCOOH^*$ in the $HCOOH$ pathway $*HCOO$ from 0.29 to −0.16 eV and from 0.67 to 0.10 eV, respectively (Figure 8c). The Gibbs free energy calculation for H^* formation showed that the H^* formation energy is much lower on the S sites of S-In (0.21 eV) than it is on either the In sites of S-In (0.69 eV) or on pure In (0.82 eV) (Figure 8d). These calculated results are in good agreement with the experimental results.

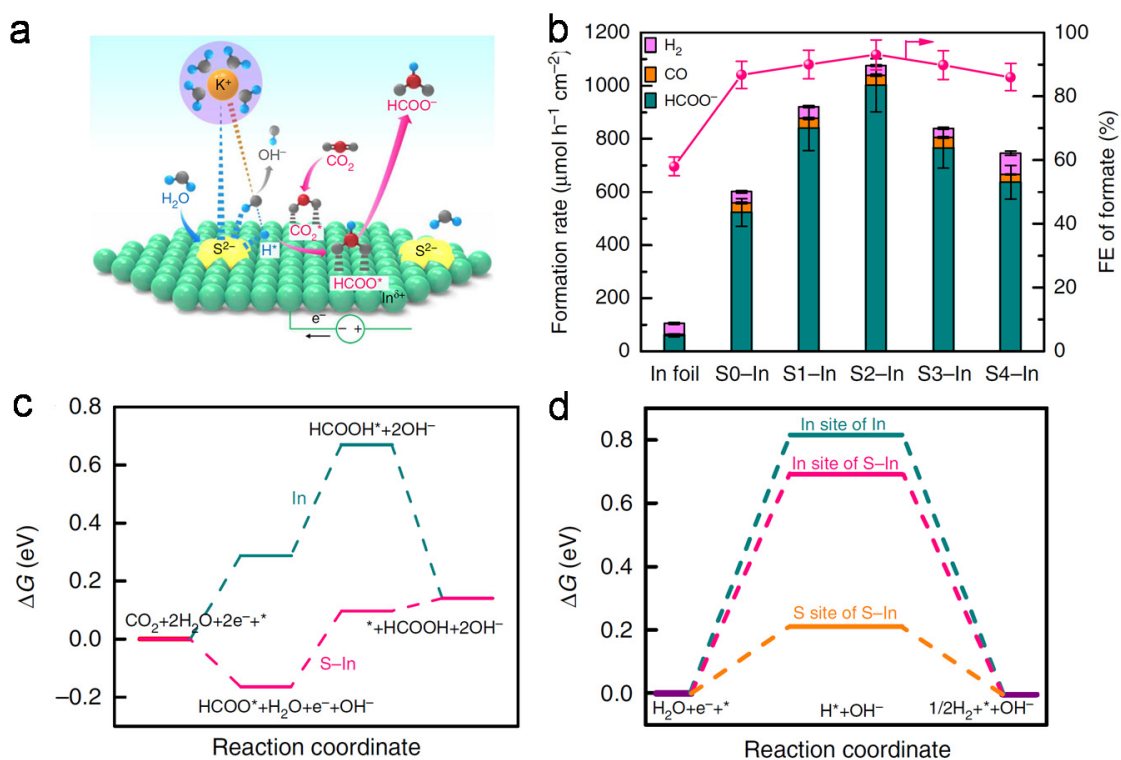


Figure 8. (a) Schematic illustration for the role of S^{2-} in promoting water dissociation and $*H$ formation for the reduction of CO_2 to formate. (b) Formation rates of H_2 , CO, and $HCOO^-$ and FE_{HCOOH} for In foil and S-In catalysts at −0.98 V vs. RHE. Density functional theory (DFT) calculation results for: (c) Gibbs free energy diagram for CO_2 RRs to $HCOOH$ on In(101) and S-In(101) surfaces; (d) Gibbs free energies for the formation of H^* on pure In(101) and on the In and S sites of S-In(101) surfaces. The free energies of (c) and (d) are shown relative to gaseous CO_2 and H_2 [73]. Copyright 2019, Nature.

5. Mass Transport

The catalytic properties of CO_2 RRs depend on the nature of the catalyst and the local concentrations of the reactants at the catalyst interface. The CO_2 RR to formate has been reported to be the first order with respect to the CO_2 concentration [37,51]. This means that the rate of formate production is limited by the CO_2 concentration but could be greatly enhanced by the mass transport control. Therefore, the mass transport control of CO_2 in CO_2 RRs is another crucial aspect of achieving good catalytic

performance. Further, substantial hydrogen evolution can proceed from various proton-donating species, such as bicarbonate and water, thereby lowering the CO₂RR selectivity [21]. Controlling the local pH at the catalyst surface can affect the chemical equilibria of proton-donating species and, therefore, the HER rate. In this section, we discuss mass transport control strategies that are relevant for CO₂RRs: (i) tuning the local pH at the catalyst–electrolyte interface; (ii) using high-pressure reactors; and (iii) employing three-phase-boundary electrodes and reactors.

5.1. Local pH Control

During electrolysis, the local pH near the electrode surface tends to be higher than that of the bulk region due to OH[−] production from both the CO₂RR and the HER [74,75]. The increase in local pH will change the concentrations of proton-donating species, which is beneficial for inhibiting the HER. However, a high local pH can also decrease the CO₂ concentration at the electrode surface because of the equilibria between CO₂, HCO₃[−], and CO₃^{2−}, which might lead to a low CO₂RR partial current density (Figure 9) [12,13,76,77]. Thus, the local pH value should be targeted to suppress hydrogen evolution while maintaining the CO₂ concentration near the electrode. Luo et al. reported the enhanced CO₂RR performance of porous-structured-Zn (P-Zn) owing to an enlarged surface area and a strengthened local pH effect [78]. Upon comparing Zn foil to P-Zn, the enlarged surface area of P-Zn led to a 10-fold higher total current density, and the local pH effect reduced the HER rate. The buffering capacity of the electrolyte also has an important role in controlling the local pH. Ma et al. evaluated the CO₂RR performance of a flat Ag electrocatalyst in 0.1 M K₂HPO₄, 0.1 M KHCO₃, and 0.1 M KClO₄ to identify the correlation between the electrolyte buffering capacity (K₂HPO₄ > KHCO₃ > KClO₄) and the local pH effect [79]. As the buffering capacity decreased, the FE for the CO₂RR increased because a lower buffering capacity allows the high local pH to be maintained, thereby suppressing the HER.

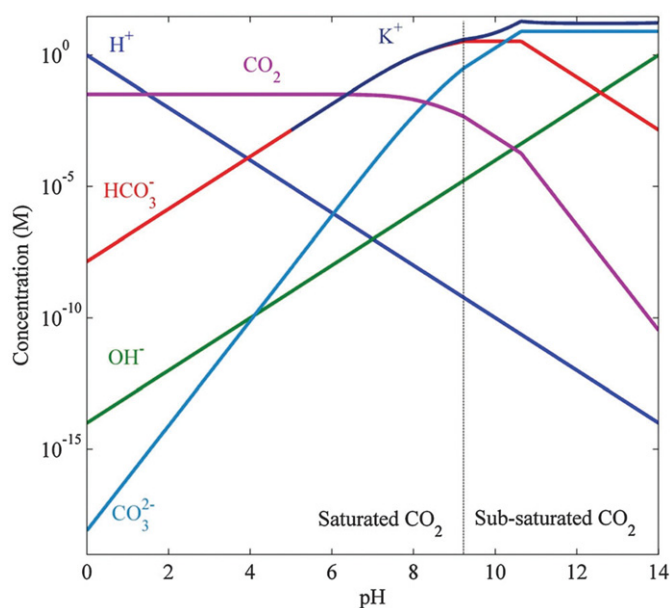


Figure 9. Concentrations of CO₂, H⁺, OH[−], HCO₃[−], CO₃^{2−}, and K⁺ as a function of pH of the KHCO₃/CO₃^{2−} electrolyte system at 25 °C and a total pressure of 1 atm [77]. Copyright 2015, RSC.

5.2. High-Pressure Reactor

According to Henry's law, increasing the partial pressure of the CO₂ can increase the concentration of the dissolved CO₂. One of the pioneering efforts by Todoroki et al. in this regard was an electrochemical reduction of CO₂ at a pressure of 60 atm, which achieved an FE_{HCOOH} of nearly 100% and a large current density of 200 mA cm⁻² [80]. Komatsu et al. reported that the FE_{HCOOH} rose significantly as the partial pressure of CO₂ increased on Cu catalysts [46]. These results indicated that an elevated CO₂ partial pressure would improve the rate of HCOOH formation by increasing the solubility of CO₂ in aqueous solutions. However, a recent report by Ramdin et al. found that CO₂ pressures of over 40 bar while using a bipolar membrane (BPM) had several disadvantages, including formate crossover through the BPM that decreases the FE_{HCOOH} by reoxidation of the as-synthesized formate on the anode. Another major disadvantage of this technique is the significant pH drop caused by the high concentration of CO₂ in the solution, which favors hydrogen evolution [81]. Thus, to obtain excellent performance in a high-pressure reactor, it is preferable to use an ion-exchange membrane with a low product crossover rate and to maintain the pH value.

5.3. Three-Phase Boundary

Despite the significant progress that has been made, there are still limitations for CO₂ mass transport that set the upper limit of current density at tens of milliamperes per square centimeter [82,83]. Therefore, utilizing a gas diffusion electrode (GDE), which was originally developed for fuel cell applications, for direct CO₂ delivery to the catalyst surface using water vapor as a carrier was proposed [84]. An article by Delafontaine et al. reported that in a conventional H-type cell, the concentration of CO₂ in aqueous solution was 0.038 M. Direct delivery of humidified CO₂ to the flow cell resulted in a modest increase in the relative saturated CO₂ concentration to 0.041 M. In the same article, the diffusion coefficient of aqueous CO₂ was reported to be as low as 0.0016 mm² s⁻¹ in CO₂-saturated 0.1 M KHCO₃ [85,86]. Delivering humidified gaseous CO₂ with a GDE dramatically increased the CO₂ diffusion coefficient 10,000-fold to 16 mm² s⁻¹ (Figure 10a) [86]. The ease with which gaseous CO₂ reached the catalyst resulted in a high CO₂ availability and a subsequent increase in the CO₂RR partial current density. As shown in Figure 10a, the GDE system is composed of a diffusion medium and a porous catalyst layer [87]. The diffusion medium is usually a hydrophobic carbon layer consisting of a macroporous carbon fiber layer and a microporous carbon powder layer. The diffusion medium serves many purposes, such as providing a porous medium through which CO₂ can diffuse to the catalyst layer, mechanically supporting the catalyst layer, and providing conductive pathways for the flow of electrons. Commercial diffusion media are usually treated with polytetrafluoroethylene (PTFE) to provide hydrophobicity. An ideal GDE remains hydrophobic throughout the process to prevent the electrolyte from leaking into the gas chamber. The catalyst is deposited on the microporous layer (MPL) to form the active catalytic site. The catalyst particles are usually mixed with a binder to hold the particles together and provide ionic conductivity.

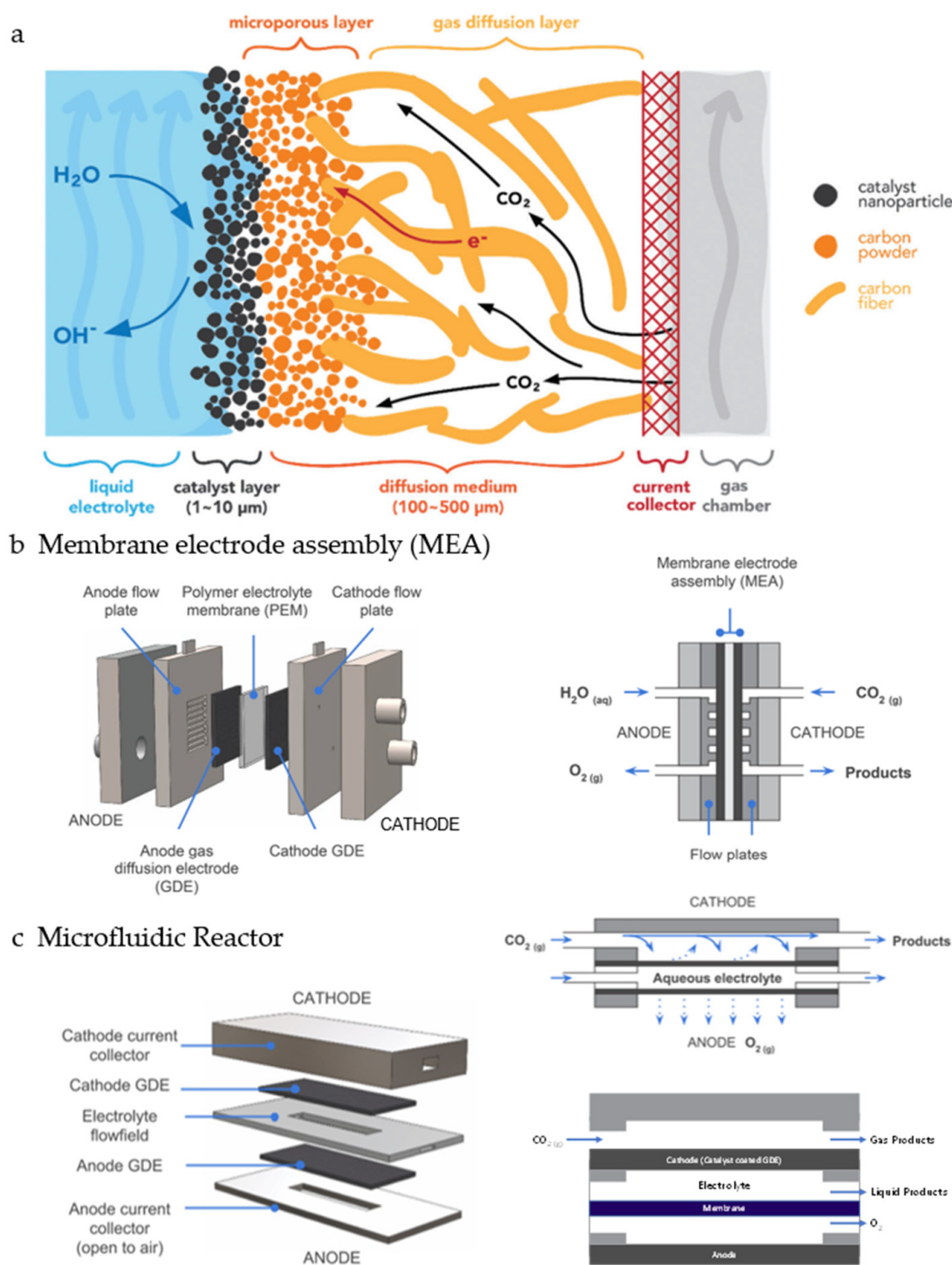


Figure 10. (a) Schematic of a gas diffusion electrode (GDE). [87] Copyright 2018, Royal Society of Chemistry. (b,c) Exploded (left) and cross-sectional (right) diagrams of two common flow cells for CO_2 RRs. (b) Membrane-based reactor containing a membrane electrode assembly (MEA) consisting of anode and cathode GDEs and (c) a microfluidic reactor with (top) and without (bottom) a membrane consisting of a liquid electrolyte flow channel for the anode and cathode GDE materials while $\text{CO}_2(\text{g})$ is supplied to the cathode side of the cell, where it diffuses to the electrocatalyst through the gas diffusion layer (GDL). [85,88] Copyright 2018, ACS. Copyright 2019, MDPI.

Del Castillo et al. reported the performance of an Sn/C-GDE using a membrane reactor [89]. Toray paper (TGP-H-90) was used as the carbonaceous support. The MPL was formed from Vulcan XC-72R and PTFE (40:60) that was air-brushed onto the carbonaceous support and sintered at 350 °C for 30 min. The catalyst ink consisted of Sn NPs and a Nafion solution (70:30) and was sprayed onto the MPL to achieve a Sn loading of 0.75 mg cm⁻². The Sn/C-GDE catalyst reached 70% FE at a current density of 150 mA cm⁻².

The MEA reactor is a common electrochemical flow cell for CO₂RRs (Figure 10a) [85]. In a traditional batch-type cell, the ions must diffuse in the bulk electrolyte to reach the membrane. This kind of system has a relatively high diffusion resistance, and the reactant availability in the counter electrode is accordingly limited. In contrast, in an MEA reactor, the membrane is sandwiched between the cathode and anode to directly transfer ions between the electrodes (e.g., H⁺ transfer between the anode and cathode). Consequently, the current density is markedly increased relative to traditional flow cells.

Another notable flow cell design is the microfluidic cell system proposed by the Kenis group (Figure 10b) [90,91]. This reactor has a thin channel of less than 1 mm through which the electrolyte flows between the anode and cathode [85]. CO₂ gas is supplied to the electrocatalyst through the GDL and reacts at the three-phase interface to form the product. A polymer electrolyte membrane (PEM) is not needed for a gaseous-product-type reactor. This reactor relies on the diffusion of the gaseous product to separate the oxidation and reduction products (Figure 10c, top), avoiding the use of high-cost membranes [85]. The microchannel enables laminar flow of the electrolyte, which eliminates the need for a membrane but still allows ionic transport between the electrodes. However, a PEM is still needed for liquid-product-type microfluidic cells to prevent product crossover (Figure 10c, bottom). Recently, Deng et al. reported the performance of a metal–organic framework (MOF)-derived carbon-nanorod-encapsulated bismuth oxide catalyst in both an H-type cell and a microfluidic cell system. Catalysis using an H-type cell resulted in an optimal FE_{HCOOH} of 92% and a partial current density of 7.5 mA cm⁻² at −0.9 V vs. RHE. As predicted, the microfluidic cell system exhibited a dramatic performance increase compared to the H-type cell, including a current density of 200 mA cm⁻² at −1.1 V vs. RHE. Overall, these results indicate that the barrier of low CO₂ solubility has been tackled by using GDL-based three-phase-boundary reactors.

6. Reported Performances

We have summarized the reported performances in terms of the strategies for enhancing catalytic performance to provide insight into the effects of these strategies (Figure 11) [10,47,53,61,72,92,93]. The results of the strategy of tailoring the exposed facets can be seen in the difference between Pd catalysts with low-index and high-index facets reported by Klinkova et al. The Pd BNPs with high-index facets have an almost identical FE_{HCOOH} and a 1.5-fold larger formate partial current density compared to the low-index-facet Pd NCs. High-density-GB Bi/Bi₂O₃ (HDGB) had an FE_{HCOOH} of almost 100%, whereas the FE_{HCOOH} of low-density-GB Bi/Bi₂O₃ (LDGB) was 87.5%. The formate partial current density of HDGB was enhanced almost 1.6-fold compared to that of LDGB. The effect of using bimetallic materials can be seen in the performances of Sn/CF and Bi–Sn/CF. The bimetallic Bi–Sn/CF had an almost 20% higher FE_{HCOOH} and a nearly 3-fold increase in formate partial current density compared to Sn/CF. The FE_{HCOOH} enhancement can be attributed to an interaction between Bi and Sn that modified the electronic structure of the catalyst in a way that increased the intermediate selectivity, but the improved formate partial current density was because of the nanostructuring effect where Bi–Sn/CF had greater active site exposure. The effect of dopants can be seen in the difference between the Pd/C and Pd–B/C catalysts. Boron insertion resulted in an enlargement within the lattice and a partial electron transfer between B and Pd. These phenomena altered the catalyst electronic structure, which increased the formate selectivity almost 15-fold to 69.3% on Pd–B/C. A two-dimensional SnO₂ material showed a higher formate selectivity of 85% compared to bulk Sn foil and had an FE_{HCOOH} of 63%. The current density of two-dimensional SnO₂ also increased to 14.75 mA cm⁻² from the

3.11 mA cm⁻² of bulk Sn foil at −1.01 V vs. RHE. Moreover, the application of SnO₂ NSs to GDEs successfully enhanced the formate partial current density from 14.75 mA cm⁻² on SnO₂ NSs to an incredibly high 471 mA cm⁻² on the SnO₂ GDE.

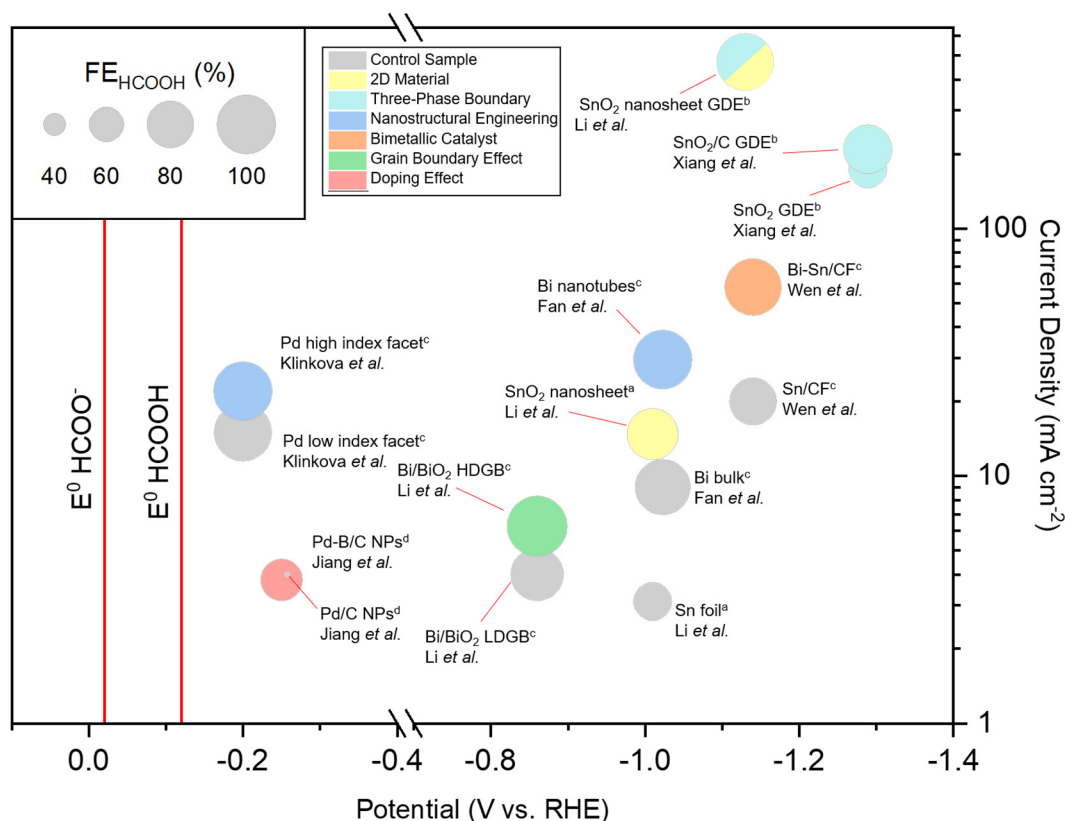


Figure 11. The summary of overpotentials, FE, and current densities previously reported for formic acid production. The electrolyte is indicated by the superscript letter after the catalyst name: a, 0.1 M NaHCO₃; b, 1 M KOH; c, 0.5 M KHCO₃; and d, 0.1 M KHCO₃. Bi/BiO₂ HDGB = high-density-GB Bi/BiO₂. Bi/BiO₂ LDGB = low-density-GB Bi/BiO₂.

Overall, optimizing the electronic structure of the electrocatalyst can enhance the formate selectivity, and the use of GDEs is the finest strategy for increasing the formate partial current density. However, achieving a low overpotential along with a high FE_{HCOOH} and a fast formate production rate remains a challenge. The reported Pd-based catalysts exhibited low overpotentials, but the stability issue raised by CO poisoning must be addressed for further optimization of catalytic performance. Furthermore, even though Bi- and Sn-based GDEs have resulted in higher current densities by orders of magnitude, new catalyst design strategies are required to reduce the overpotential to close to the equilibrium potential of formate production.

7. Conclusions

Researchers have made efforts to develop new catalysts and extrinsically enhance catalytic systems. However, the inert nature of CO₂, the low solubility of CO₂ in aqueous solutions, and the slow mass transport of CO₂ result in the CO₂RR system for formate production still having low selectivity, high overpotentials, and limited formate partial current density. Several strategies have been employed to address these challenges. Tailoring the binding affinity of a catalyst for a key intermediate was reported to enhance the selectivity and reduce the overpotential of the catalyst. This tailoring can be achieved in several ways, including exposing the high-energy facet, edge, and corner sites and controlling the composition of the catalyst. Attempts have been made to address the low CO₂ solubility and slow mass transport of CO₂ onto the catalyst surface by optimizing the local pH at the catalyst

surface, using a high-pressure reactor to increase the CO₂ solubility, and employing a GDE to directly transport gaseous CO₂ to the active sites. However, the high-pressure environment makes the product cross over through the ion transport membrane towards the anodic side of the reactor, which depresses formate production. The loss of hydrophobicity in the GDE during electrolysis also decreases the CO₂ concentration on the catalyst surface owing to the electrolyte flooding the gas diffusion medium. These issues must be addressed in future studies in addition to stabilizing the catalyst. Machine learning based on DFT calculations is expected to accelerate the identification of efficient catalysts satisfying the requirements of low overpotential with high selectivity and stability. We believe that the minimum requirements for technoeconomic feasibility of this pathway to formate production can be achieved via the combined approaches of catalyst development and broader system-level strategies.

Author Contributions: Writing—original draft preparation, M.A.; Writing—review and editing, C.W.L.; supervision, C.W.L. All authors have read and agreed to the published version of the manuscript.

Funding: This work was supported by the National Research Foundation of Korea (NRF) Grant funded by the Korean Government (MSIT; NRF-2020R1C1C1010963 and NRF-2016R1A5A1012966).

Acknowledgments: In this section you can acknowledge any support given that is not covered by the author contribution or funding sections. This may include administrative and technical support, or donations in kind (e.g., materials used for experiments).

Conflicts of Interest: The authors declare no conflict of interest.

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