



Review

# Transition-Metal Dichalcogenides in Electrochemical Batteries and Solar Cells

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**Abstract:** The advent of new nanomaterials has resulted in dramatic developments in the field of energy production and storage. Due to their unique structure and properties, transition metal dichalcogenides (TMDs) are the most promising from the list of materials recently introduced in the field. The amazing progress in the use TMDs for energy storage and production inspired us to review the recent research on TMD-based catalysts and electrode materials. In this report, we examine TMDs in a variety of electrochemical batteries and solar cells with special focus on MoS<sub>2</sub> as the most studied and used TMD material.

Keywords: transition metal dichalcogenides; electrochemical batteries; supercapacitors; solar cells



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## 1. Introduction

Over the past decade, transition metal dichalcogenides (TMDs) have attracted widespread scientific interest due to their special and often layer-tunable chemical, thermal, mechanical, electronic, magnetic, and optical properties [1–5]. TMDs have the chemical formula  $MX_2$ , where M is a transition metal and X is a chalcogen from the group VI-A. In general, TMDs with transition metals from 4B–7B groups have a layered structure, while TMDs with transition metals from 8B–10B groups, such as pyrite, have a non-layered structure. The atoms in the  $MX_2$  layered structure are in polytype mode, with one transition metal atom surrounded by six chalcogen atoms [6].

Following the discovery of strong photoluminescence in MoS<sub>2</sub> monolayers [7] and the demonstration of the first transistor [8], significant studies were performed on the application of TMDs in fields such as energy storage [9,10], energy production [11–13] electronics [14–16], and optoelectronics [17,18].

Molybdenite as the first structure of TMDs was determined in 1923 by Linus Pauling [19]. Over the following 40 years, around 40 TMDs with a layered structure became known [20]. However, the first attempt to produce ultrathin MoS<sub>2</sub> layers by use of adhesive tapes in 1963 was performed by Robert Frindt [21]. Later, in 1986 the production of monolayer MoS<sub>2</sub> suspensions was achieved [22]. The discovery of WS<sub>2</sub> nanotubes and nested particles [23], followed by the synthesis of MoS<sub>2</sub> nanotubes and fullerenes, further boosted this field [24]. After 2004, the rapid growth of graphene-based research sparked

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the interest in layered materials that, as graphene, could be exfoliated or produced in the monolayer form and led to the development of techniques well suited for new studies on two-dimensional (2D) materials.

The use of two-dimensional materials in various fields of science is of great interest [25,26]. The structural similarity to graphite and graphene is combined with specific and very special properties in TMDs. In general, the three-atom thick unit cell of TMDs is formed by a layer of transition metal atoms (e.g., Mo, W, Ta, etc.) sandwiched between two layers of chalcogen atoms (e.g., S, Se, Te). Due to the weak bonds between adjacent layers, TMDs can be thinned down to single layers [27,28]. The weak interlayer adhesion is caused by the presence of a gap or voids between the layers and is still a matter of debate in the scientific community [29,30].

A van der Waals gap between the layers, affecting the electronic and optical properties of the material, is supported by experimental evidence, such as scanning tunneling microscopy and X-ray diffraction studies, which have shown a clear separation between the layers in TMDs [30]. On the other hand, some studies have suggested that there may be voids rather than a gap between the layers in TMDs. These voids exist between the chalcogenides in the interfacial/interlayer region and are thought to be responsible for the weakening of the weak forces in the interlayer region. This view is also supported by experimental evidence, such as transmission electron microscopy studies, which have shown a lack of a clear separation between the layers in some TMDs [31].

It is important to note that the occurrence of a specific type of force between the layers depends on the nature of the transition metal chalcogenide system. The intercalation of foreign species, such as Li, can modify the interfacial geometry of TMDs and increase the distance between two consecutive monolayers, which in turn can tune the electronic and optical properties of the material. However, as noted in several studies, intercalation can also result in the loss of the pristine semiconducting properties of TMDs due to structural changes in the host materials [32]. While the debate over the presence of a gap or voids between the layers of metal dichalcogenide systems continues, it is clear that the interlayer region plays a critical role in determining the properties of these materials.

Most of the recent reviews cover the electronic and optical properties of TMDs and their heterostructures, as well as their optoelectronic applications [33–37]. The energy crisis and the need to use clean resources have urged researchers to pay attention to the design and construction of energy storage devices and energy production from renewable sources [38,39]. Among the mentioned devices, types of electrochemical batteries and solar cells have received more attention than others due to their extraordinary advantages. One of the most essential parts of batteries and solar cells is the material used in their electrode structure. Reviewing recent scientific sources confirms the attractiveness of introducing different materials for use in the electrode of these types of equipment. Oxides and sulfides of transition metals, as well as a combination of these materials, are attractive, inexpensive, and relatively stable options in catalysts, especially in energy [40–45]. Among the mentioned materials, TMDs are of great interest due to their very specific structure and extraordinary physical and chemical properties. In this review, the focus is on TMDs, mainly MoS<sub>2</sub>, used in the field of energy production and storage. This article describes the applications of these materials in electrochemical batteries and solar cells.

#### 2. Transition Metal Dichalcogenides in Electrochemical Batteries

TMDs have gained significant attention in recent years for electrochemical applications. Their layered structure gives rise to several interesting electrochemical properties that make TMDs attractive for use in a variety of applications, including energy storage, energy conversion, sensing, catalysis, and electronics. One of the key aspects of the electrochemical behavior of TMDs is their high electrochemical stability. TMDs exhibit minimal degradation and corrosion in electrochemical environments, making them useful for applications in energy storage and conversion devices. This high electrochemical stability is due to the strong covalent bonds between the atoms in the layers and the large energy band gap. This

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combination of properties makes TMDs resistant to degradation and corrosion in electrochemical environments. TMDs such as molybdenum disulfide (MoS<sub>2</sub>), tungsten disulfide (WS<sub>2</sub>), and titanium disulfide (TiS<sub>2</sub>), have been used as materials for electrochemical energy storage applications, such as batteries and supercapacitors, due to their high surface area, good electrical conductivity, and high stability. In batteries, TMDs can be used as cathode materials because they can store and release a large amount of lithium ions, leading to high specific capacity. For example, MoS<sub>2</sub> has a theoretical specific capacity of about 670 mAh  $\rm g^{-1}$ , which is higher than many other cathode materials used in lithium-ion batteries. Additionally, the layered structure of TMDs provides a large surface area for lithium-ion diffusion, which can improve battery performance [46].

Another important aspect of the electrochemical behavior of TMDs is their large surface area. TMDs have a large surface area compared to their volume, which enhances their electrochemical activity and makes them attractive for use in electrochemical sensing and catalysis applications. This large surface area is due to the layered structure of TMDs, which allows for a large number of atoms to be exposed at the surface. This large surface area increases the electrochemical activity of TMDs, making them attractive for use in applications where electrochemical reactions are important [46,47].

The start of the era of modern batteries, as energy storage devices, can be dated to the early 18th century when Alexander Volta introduced the voltaic pile [48]. Basically, batteries are composed of two electrodes separated by an electrolyte, with chemical energy stored in the active electrodes [49]. The chemical energy is directly converted to electrical energy through electrochemical redox reactions or charge-transfer reactions [50]. The way batteries work is that the reduction reactions that take place at the negative electrode produce electrons that through the external circuit reach the positive electrode where oxidation reactions take place [51].

There are different types of batteries, but they are generally divided into two main categories [52]: primary batteries, commonly known as non-rechargeable batteries, and secondary batteries which include rechargeable batteries. Primary batteries are used in small (portable) electrical devices and are generally non-rechargeable and recyclable after use. The most common are alkaline and zinc–carbon batteries with aqueous or non-aqueous electrolytes [53,54]. Secondary batteries are widely used in everyday life applications, like cell phones and electric vehicles, and include lithium-ion, sodium-ion, magnesium-ion, aluminum-ion, calcium-ion batteries with nickel-metal hydride, nickel–cadmium, and so on. These batteries are classified based on the material of their electrodes [55,56].

Higher efficiency, longer life, flexible design, and optimal energy density are the reasons for the high acceptance of lithium-ion batteries [57]. In commercialized lithium-ion batteries, graphite and Li<sup>+</sup> have been used as anode and cathode, respectively, which have drawn attention to new materials owing to their low performance [58].

Today, many materials are used to improve battery performance. Among them, TMDs are widely exploited, especially in rechargeable secondary batteries. TMDs are abundant in natural minerals and have a high charge–discharge, long life, and a high oper-ating temperature range.

One of the main energy storage mechanisms of TMDs is lithium-ion intercalation. In this process, lithium ions are inserted into the TMD crystal structure, causing a change in its electrical properties, and allowing for the storage of electrical energy. The reversible intercalation of lithium ions also provides a high capacity for energy storage, making TMDs attractive for use in lithium-ion batteries. Another mechanism is the capacitive energy storage that takes advantage of the high electrical conductivity of TMDs. In this process, charge is stored on the surface of the TMD material, creating a double-layer capacitor. The high surface-to-volume ratio of TMDs results in a high capacitance, allowing for efficient energy storage. TMDs also show promise as supercapacitor materials due to their high charge storage capacity and excellent stability. The energy storage mechanism in supercapacitors is based on the rapid transfer of charge between the TMD and the electrolyte, which enables rapid charging and discharging [59].

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In this review, we focus on the application of TMDs in batteries, such as lithium-ion and sodium-ion batteries, which are widely used in electric vehicles, energy storage bases, and smart networks. In this domain, the most used TMDs are MoS<sub>2</sub> and FeS<sub>2</sub>.

# 2.1. MoS<sub>2</sub>, MoS<sub>2</sub>-Metal Oxides, and Other TMDs in Lithium-Ion Batteries

TMDs have a layered crystal structure consisting of a transition metal layer sandwiched between two chalcogenide layers. The chalcogenide atoms can be sulfur (S), selenium (Se), or tellurium (Te), while the transition metal atoms can be molybdenum (Mo), tungsten (W), or other transition metals. The most commonly studied TMDs are  $MoS_2$ ,  $WS_2$ , and their derivatives.

The layered structure of such TMDs is similar to graphite, with strong covalent bonds within each layer and weak van der Waals forces between the layers. This results in a material that is mechanically flexible, optically transparent, and has high surface area. The electronic properties of TMDs are highly dependent on the number of layers. When TMDs are in their two-dimensional form (i.e., with one or few layers), they often exhibit a direct band gap, which makes them attractive for optoelectronic applications. As the number of layers increases, the band gap typically becomes indirect, leading to changes in the material's electronic properties.

To realize the full potential of TMD materials, a variety of synthetic techniques have been developed for their production. TMDs can be synthesized by mechanical exfoliation, chemical vapor deposition (CVD), solution-based methods, etc.

One of the most commonly used techniques for synthesizing TMDs is chemical vapor deposition (CVD). CVD involves the thermal decomposition of a precursor gas, which is introduced into a reaction chamber along with a substrate. The gas reacts on the substrate, forming a TMD material layer. CVD is a versatile technique that can be used to produce TMDs with different sizes, shapes, and properties [60].

Mechanical exfoliation is another technique that is used to produce TMDs. This technique involves peeling off individual layers of TMDs from a bulk crystal using scotch tape. The resulting flakes can be transferred to a substrate to form thin films. This technique is commonly used to produce high-quality TMDs for research purposes [61].

Hydrothermal and solvothermal synthesis are two other techniques used to synthesize TMDs. Hydrothermal synthesis involves reacting transition metal salts and chalcogenide salts in a high-pressure autoclave at elevated temperatures, while solvothermal synthesis involves reacting these salts in a solvent at high temperatures and pressures. Both of these techniques can produce TMDs with controlled sizes and shapes [62].

Electrochemical synthesis is a technique that involves electrochemically reducing transition metal ions and chalcogenide ions in a solution containing an electrolyte. This technique can produce TMDs with high purity and controlled morphology [63].

Ion exchange is another technique that is used to produce TMDs. This technique involves replacing the cations in a layered TMD material with other metal ions through ion exchange reactions. This technique can produce TMDs with unique properties, such as enhanced catalytic activity [64].

Thermal stability of TMDs is an important factor in their potential use in energy applications. The high thermal stability of TMDs makes them attractive for use in high-temperature environments and suggests that these materials have great potential for use in a range of energy-related applications. Thermal stability refers to a material's ability to resist degradation or decomposition when exposed to high temperatures. For TMDs, which are being considered for use in high-temperature environments, high thermal stability is particularly important. Fortunately, TMDs have been found to exhibit high thermal stability, with many materials able to withstand temperatures of up to 700–800 °C without significant degradation [65].

For example, WS<sub>2</sub> has a thermal stability of up to 900  $^{\circ}$ C in air and up to 1000  $^{\circ}$ C in vacuum. This high thermal stability makes WS<sub>2</sub> a promising material for high-temperature applications such as thermoelectrics, which involve converting waste heat into electricity.

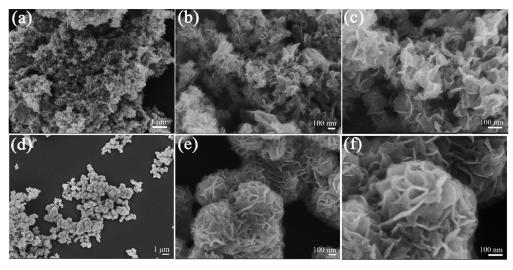
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Similarly,  $MoS_2$  has a thermal stability of up to 600–700 °C, making it suitable for use in catalysis and energy storage [66].

The high thermal stability of TMDs is due to the strong covalent bonding between the transition metal layer and the chalcogenide layers in the material. This bonding makes the material resistant to thermal degradation, allowing it to maintain its structural and electronic properties even at high temperatures [67]. Other TMDs, such as molybdenum diselenide (MoSe<sub>2</sub>) and tungsten diselenide (WSe<sub>2</sub>), have also been found to exhibit high thermal stability in the range of 500–700 °C. This suggests that TMDs have great potential for use in a range of high-temperature energy applications, including thermoelectrics, catalysis, and energy storage [68]. The measured room temperature thermal conductivity of TMDs such as MoS<sub>2</sub>, MoSe<sub>2</sub>, MoTe<sub>2</sub>, WS<sub>2</sub>, and WSe<sub>2</sub> is around  $\sim$ 154 W m<sup>-1</sup> K<sup>-1</sup>,  $\sim$ 70 W m<sup>-1</sup> K<sup>-1</sup>,  $\sim$ 262 W m<sup>-1</sup> K<sup>-1</sup>, and  $\sim$ 120 m<sup>-1</sup> K<sup>-1</sup>, respectively [69].

The pure  $MoS_2$  is prepared by many physical and chemical methods such as chemical vapor deposition, thermal decomposition, and gas phase synthesis.

Structurally,  $MoS_2$  comes in two forms, nanosheets and nanoflowers, both of which are layered structures used in rechargeable lithium-ion batteries (Figure 1). A comparison shows that for  $MoS_2$  nanosheets the reversible discharge capacity is 589 mAh  $g^{-1}$  at 100 mA  $g^{-1}$  after 80 cycles while for  $MoS_2$  nanoflower it is 883 mAh  $g^{-1}$  at 400 mA  $g^{-1}$  after 30 cycles, indicating the importance of the material morphology on the performance of the electrode. The discharge capacity depends also on the synthesis methods. A reversible discharge capacity of ~850 mAh  $g^{-1}$  at 100 mA  $g^{-1}$  after 50 cycles has been found with  $MoS_2$  produced by the hydrothermal method. These results demonstrate that  $MoS_2$  suits fast charge–discharge needs [70–73].



**Figure 1.** FESEM images of MoS<sub>2</sub> nanosheets (**a–c**) and MoS<sub>2</sub> nanoflowers (**d–f**) at different magnifications. Reprinted (adapted) with permission from Reference [74], Copyright (2019), American Chemical Society.

Accumulation of  $MoS_2$  plates during lithium-ion battery cycles is a major problem. Therefore, oxides with suitable reversible discharge capacity were used to facilitate the penetration of the electrolyte between the  $MoS_2$  plates and separate the plates to prevent them from accumulating on top of each other.  $TiO_2$  is known as a suitable anodic material for lithium-ion batteries with suitable properties such as low cost, easy access, and environmental friendliness. There is also a strong synergistic effect between  $MoS_2$  and  $TiO_2$ , which makes the transport of ions and electrons easier at the interface. For example, compounds such as hydrothermal plate  $MoS_2$ , coated with  $TiO_2$  tablet-shaped plates or  $TiO_2$  microsphere structure and  $MoS_2$  plates together have a high cyclic capacity of about 714–740 mAh  $g^{-1}$  after 150–200 cycles at 100 mA  $g^{-1}$ .  $TiO_2$  has also been used as an interfacial bonder to create a strong bond between  $MoS_2$  and other compounds, allowing

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electrons to be easily transported across the electrode. In addition to  $TiO_2$ , other metal oxides, such as  $Fe_3O_4$  and  $SnO_2$ , which have high capacities and are suitable for preparing electrode materials in the form of composites with  $MoS_2$ , have also been reported. These composites showed good reversible discharge capacity [75–80].

## 2.2. MoS<sub>2</sub>—Carbon Composites in Lithium-Ion Batteries

#### 2.2.1. MoS<sub>2</sub>-Graphene Hybrid

Graphene is a carbon material with a large specific surface area, excellent electron mobility, high thermal conductivity, and a carbon framework with  $\rm sp^2$  hybridization. Hence it is used as a matrix to support the anode or cathode in rechargeable lithium-ion batteries. There are different methods for preparing MoS2-graphene composite. These methods include hydrothermal liquid phase, chemical vapor deposition, and liquid phase with temperature correction. There are other easy fabrication methods such as heat-induced formation process that result in MoS2-graphene sheet composite with good efficiency of 552 mAh g $^{-1}$  at a current density of 10 A g $^{-1}$  after 7500 cycles [81]. The synergistic effect created from integration of MoS2 and graphene leads to enhanced lithium storage performance.

# 2.2.2. MoS<sub>2</sub>-Amorphous Carbon Hybrid

These types of carbon materials are very inexpensive and are used in rechargeable lithium-ion batteries to increase the distance between  $MoS_2$  plates and achieve the desired conductivity. Use of amorphous carbon in the combination of  $MoS_2$ –amorphous carbon showed relatively good results. So, this composite with suitable preparation methods can be used to form a uniform, inexpensive hybrid with optimal layout in rechargeable lithium-ion batteries [82–84].

#### 2.2.3. MoS<sub>2</sub>-Carbon Nanotube Hybrid

Carbon nanotubes are one of the carbon allotropes. Carbon nanotubes are divided into two categories: multi-walled and single-walled. Nanotubes have been noted for their excellent thermal conductivity, and good mechanical and electrical properties, and have a special place as a modifier in rechargeable lithium-ion batteries.

The combination of carbon nanotubes with  $MoS_2$  has been conducted in different ways and has provided good results [85–88]. The composition of  $MoS_2$  and single-walled carbon nanotube through the liquid-phase exfoliation of  $MoS_2$  showed a strong electrochemical capacitance of 992 mAh  $g^{-1}$  after 10 cycles with retention of 82% of the initial capacitance of 1117 mAh  $g^{-1}$  (Figure 2). In addition, the morphology of  $MoS_2$ –SWCNT thin film (Figure 3) retained structural impeccability after 100 cycles, while the  $MoS_2$  thin film without SWCNTs displays cracks, which indicate a mechanical degradation issue [89].

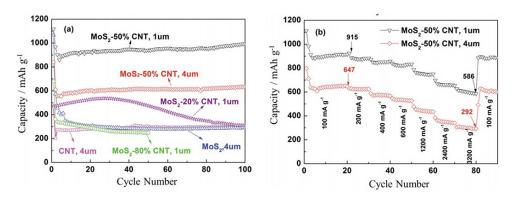


Figure 2. (a) Electrochemical performance of SWNT,  $MoS_2$ , and  $MoS_2$ –SWNT films between 0.01 and 3.0 V vs.  $Li/Li^+$  at a cycling rate of 100 mA  $g^{-1}$ , and (b) rate capacities of  $MoS_2$ –SWNT (50%) films with a thickness of 1  $\mu$ m and 4  $\mu$ m. Reprinted (adapted) with permission from Reference [89], Copyright (2013), Wiley.

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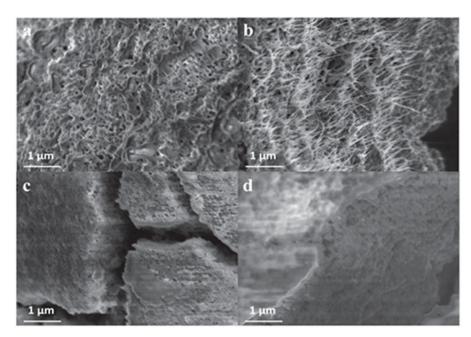


Figure 3. FESEM images of (a)  $MoS_2$ –SWNT composite film, (b) cross-section of  $MoS_2$ –SWNT composite film (c), pure  $MoS_2$  film electrode and (d) cross-section of pure  $MoS_2$  film electrode after cycling. Reprinted (adapted) with permission from Reference [89], Copyright (2013), Wiley.

# 2.2.4. MoS<sub>2</sub>-Polymer Hybrid

Conductive polymers are another material used in MoS<sub>2</sub>-based rechargeable batteries as molybdenum disulfide modifiers, for example combining a conductive polyaniline polymer with molybdenum disulfide. Of course, these conductive polymers have properties that limit their use in batteries based on TMDs. The limiting properties of these polymers are rigidity of the chains (which disrupts the movement of ions), high cost, insoluble, low permeability, and supply problems [90–92].

#### 2.3. Other Layered TMDs in Lithium-Ion Batteries

Other layered TMDs used in rechargeable lithium-ion batteries include  $WS_2$ ,  $TiS_2$ , and  $VS_2$ . due to their high theoretical capacity and good power density [93–96].

 $WS_2$  is a promising material to use for the anode in lithium-ion batteries. The interlayer distance in  $WS_2$  is about 0.63 nm, which is greater than graphite with an interlayer distance of 0.32 nm and provides more paths for intercalation of Li+. Liu et al. prepared a sample of mesoporous  $WS_2$  by a vacuum-assisted impregnation method. To achieve a high surface area (115 m<sup>2</sup> g<sup>-1</sup>) during a process, quantities of gases that were trapped in the internal pores were forced outwards. The high discharge capacity for mesoporous  $WS_2$  was obtained after 100 cycles with a discharge rate of 100 mA g<sup>-1</sup> that was equal to 805 mA h g<sup>-1</sup>. Additionally, during discharge current of 10 mA g<sup>-1</sup>, it showed a high rate capacity of 503 mA h g<sup>-1</sup> [97].

Bhandavat et al., with the help of acid exfoliation method and by chlorosulfonic acid, were able to convert bulk WS<sub>2</sub> into few-layer two-dimensional WS<sub>2</sub> sheets and achieve reversible capacity of 469 mA h  $\rm g^{-1}$  at a discharge rate of 25 mA  $\rm g^{-1}$  for lithium-ion batteries [98].

## 2.4. Layered TMDs in Rechargeable Sodium-Ion Batteries

Sodium-ion batteries are a more cost-effective method of energy storage used for large-scale energy storage applications. Applications for rechargeable sodium-ion batteries include energy storage bases, energy storage networks, and storage of renewable energy such as wind and solar power, in backup systems as a permanent power supply. Like lithium-ion batteries, TMDs such as MoS<sub>2</sub> and TiS<sub>2</sub> are widely used in sodium-ion batteries.

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As mentioned earlier,  $MoS_2$  and other TMDs have been used both in pure form and in composite form with carbon and non-carbon materials in sodium-ion batteries [99–102].

In general, the range of applications of  $MoS_2$  is reduced owing to inherent restrictions such as low electrical conductivity [99]. Hence, the applications of TMDs other than  $MoS_2$  to prepare TMD-based electrodes with excellent and reliable performance for use in batteries are highly regarded. A good example is  $MoSe_2$ , which has a narrower band gap and larger interlayer spacing compared to  $MoS_2$ , and these properties lead to better performance as anodes in sodium-ion batteries [103].

Similarly, another important member of the TMD family, which is MoTe<sub>2</sub>, with an interlayer distance of about 0.70 nm, can accommodate large-sized ions such as Na<sup>+</sup> (0.35 nm) [104]. Moreover, Te has better electronic conductivity (2  $\times$  10<sup>2</sup> S m<sup>-1</sup>) than S (5  $\times$  10<sup>-13</sup> S m<sup>-1</sup>) and Se (1  $\times$  10<sup>-3</sup> S m<sup>-1</sup>), making MoTe<sub>2</sub> a good anode material [104]. Indeed, MoTe<sub>2</sub> as the anode in sodium-ion batteries has exhibited acceptable performance stability [105].

Overall, TMDs appear as promising electrode materials for electrochemical energy storage in rechargeable batteries.

#### 2.5. Non-Layered TMDs in All Types of Rechargeable Batteries

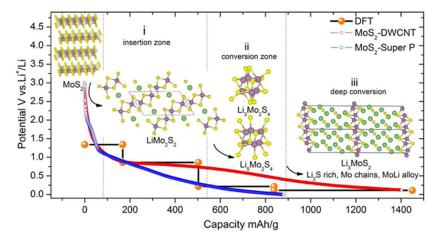
Another category of TMDs includes non-layered structures, which are used as electrode materials in lithium-ion batteries as well as rechargeable sodium-ion batteries. Non-layered TMDs, which are in the category of sulfide minerals, are used as electrode materials in a variety of rechargeable batteries due to their properties such as cost-effectiveness, high capacity, low toxicity, availability, and suitable frequency.

One of the most common sulfide minerals in the semiconductor category is pyrite (FeS<sub>2</sub>), which has been used both as pure and in composite with carbon fiber, graphene oxide, nanotubes, and other materials in rechargeable lithium-ion and sodium-ion batteries [101].

CoS<sub>2</sub> has a pyrite structure, and it can play the role of cathode in thermal batteries. It has also been used to replace pyrite in rechargeable batteries as an electrode material. This material is not yet widely used but has good potential for use in rechargeable batteries [106].

## 2.6. Storage Mechanisms

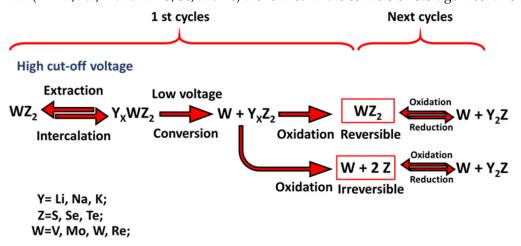
In rechargeable batteries, the storage mechanism of Li, Na, and K ions has been extensively studied using experimental methods and theoretical calculations (Figure 4). In most cases, alkali metals act likewise in charge storage owing to their similar physical properties [107-109]. Nevertheless, exacerbating the diffusion resistance and the increase of the volume of TMDs is possible based on the increase of the ionic radius from Li<sup>+</sup> ion to K<sup>+</sup> ion [110,111].



**Figure 4.** Li-ion transport mechanism of the MoS<sub>2</sub> structure in lithium-ion batteries. Reprinted (adapted) with permission from Reference [112], Copyright (2016), American Chemical Society.

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The cut-off voltage and the properties of TMDs are effective on the storage mechanism of rechargeable batteries (Figure 5). At high cut-off voltages of about 1 V, reversible intercalation and extraction reactions occur for all TMDs [113]. On the other hand, at low cut-off voltages of about 0.1 V, for some TMDs with Ti and Nb, reversible intercalation and extraction reactions can still be performed due to the interactions in the bond between groups (T = Ti, Nb) with (Z = S, Se, and Te), while the TMDs including (W = V, Mo, W, and Re) do not have the ability to act in these conditions [114]. After the first discharge process in rechargeable batteries,  $WZ_2$  (W = V, Mo, W, and Re) alone and a compound with formula Y2Z (Y = Li, Na, K and Z = S, Se, and Te) are formed in the conversion storage mechanism.



**Figure 5.** Schematic of electrochemical storage mechanisms for  $WZ_2$  in rechargeable batteries. Reprinted (adapted) with permission from Reference [115], Copyright (2020), Royal Society of Chemistry.

The dynamic properties of TMDs are effective on oxidation products [114]. After the initial charge, due to weak dynamic properties, (W = V, Mo, W, and Re) and (Z = S, Se, and Te) are irreversibly formed. In the next step, the remaining active material from the first charge step changes from the combined state between (W = V, Mo, W, and Re) and (Z = S, Se, and Te) to (Z = S, Se, and Te) only in the following cycles. The highest dynamic properties will cause complete reversible oxidation. Compound (W = V, Mo, W, and Re) and 2 (Z = S, Se, and Te) is formed during the first charge process and acts as an active material in the following cycles.

The intercalation mechanism of Li, Na, and K ions storage in (W = V, Mo, W, and Re) and 2 (S, Se, and Te) compounds has been established to be reversible, the intercalation mechanism of these compounds is similar to the assumed mechanism for graphite [116].

Until now, the intercalation mechanism of Li to the most famous member of the TMDs family, namely  $MoS_2$ , has been studied by many researchers. In a 2012 study, to confirm the reversible reactions of commercial  $MoS_2$  at a cutoff voltage of 0.8 V, and to discover its crystal structure, Fang et al. used in situ X-ray diffraction analysis [116]. Based on numerous research studies that were conducted on the intercalation mechanism of Li, Na, and K ions in the structure of TMDs, especially  $MoS_2$ , in general, the results of several analyses that were performed confirmed that the intercalation of Li ion in the structure is more than an emission of 4, also with the emission of ions within the  $MoS_2$  structural network, there undergoes a noticeable phase change. Alternatively, the results confirmed that the optimal cut-off voltage range to prevent the damage of  $MoS_2$  layered structure is in the range of 3 to 0.6 V [115].

# 3. Transition Metal Dichalcogenides in Solar Cells

All radiation in the electromagnetic spectrum, including sunlight, can be thought of in terms of photons, which carry certain amounts of energy that depend on the wavelength.

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To create an electron-hole pair, mainly the photons with energy greater than the band gap of a semiconductor contribute to the energy conversion process [117].

In semiconductors, which are the materials used in solar cells, there is a forbidden band or band gap that separates the valence and conduction bands. If the photon received by the semiconductor provides enough energy to the electron in the valence band to be transferred to the conduction band, a photovoltaic (PV) effect is possible. But if the band gap is large such that no electron can absorb light and make a transition to the conduction band, then this material will not have the necessary efficiency as a semiconductor. Therefore, materials with a large band gap have not been a good option in solar cells. Small-band gap materials are also not suitable, because the electron receives energy that transfers it to the higher levels of the conduction band. As there are lower energy levels in the conduction band, the electron scatters in transit to these levels, wasting energy in the form of heat. As a result, the most efficient conversion of light energy into electricity occurs when the received photon energy is very close to the band gap energy of the material in question [118].

Increasing the performance of a solar cell is possible by increasing the power conversion efficiency (PCE) of the cell, reducing production costs, and increasing the life of the cell. Therefore, researchers have studied different materials to obtain PV cells with better performance, cost-effectiveness, and safety [119].

Conventional silicon (Si) solar cells have a 95% share of the photovoltaic market due to their low-cost fabrication and reasonable PCE. What reduces the performance of ultra-thin and flexible Si solar cells is the fragile nature of Si, which makes it inefficient in applications such as aerospace, transportation, architecture, and wearable electronics because they require high power per weight [120].

A new generation of advanced solar cells such as Dye-Sensitized Solar Cells (DSSCs), Organic Solar Cells (OSCs), Quantum Dot Solar Cells (QDSC), and Perovskite Solar Cells (PSCs) has emerged. Researchers are currently attempting to replace silicon solar cells with a new generation of cells.

#### 3.1. Perovskite Solar Cells (PSCs)

Among the different types of solar cells, PSCs have attracted a lot of attention due to their high conversion efficiency, flexibility, low cost, and construction by the roll-to-roll method. Market conquest by a new generation of PSCs is predicted soon to take place [121]. Perovskite solar cells are a type of photovoltaic (PV) technology that uses perovskite material as the light-absorbing layer to convert sunlight into electricity. The perovskite layer is typically composed of a mixture of organic and inorganic materials, such as lead halides and organic compounds. One of the advantages of perovskite solar cells is their high light-to-electricity conversion efficiency, which has improved dramatically over the last decade, reaching over 25% in some cases. They are also relatively easy and cheap to produce, compared to other types of PV technologies. However, perovskite solar cells still face some challenges, such as stability and toxicity issues. Lead is a toxic element and its use in perovskite solar cells raises environmental and health concerns. In addition, the perovskite material is prone to degradation under harsh environmental conditions, such as high temperature and humidity. Despite these challenges, perovskite solar cells have the potential to become a game-changer in the renewable energy sector. Researchers are working to develop more stable and environmentally friendly perovskite materials, and to improve the manufacturing process and long-term stability of perovskite solar cells. Hence, scientists are evaluating different materials for making different types of solar cells. The use of ultra-thin nanometer-based light absorbers in solar cells reduces material consumption, charge emission length, and recombination. The use of two-dimensional materials such as graphene has also been useful in some solar cells [122].

# 3.2. TMD Solar Cells

Solar cells based on TMDs have attracted significant attention due to their unique optoelectronic properties, such as high absorption coefficients and efficient charge separa-

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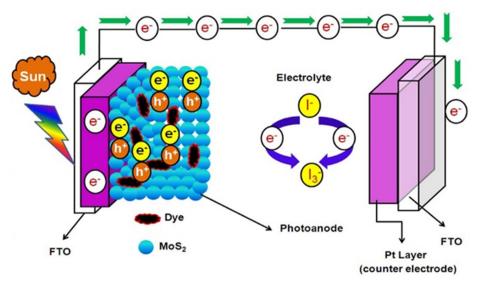
tion [123,124]. This has led to the development of TMD-based photovoltaic devices with improved performance compared to traditional silicon-based solar cells.

However, the development of TMD-based solar cells is still in its early stages, and there are several technical challenges that need to be addressed, such as improving the efficiency and stability of the devices. Nevertheless, the potential benefits of TMDs in the field of photovoltaics make them a promising area of research and development.

Semiconductor TMDs such as WS<sub>2</sub>, WSe<sub>2</sub>, MoS<sub>2</sub>, MoSe<sub>2</sub>, MoTe<sub>2</sub>, and TaS<sub>2</sub> have shown remarkable optical and mechanical properties with adjustable electronic properties and structural control capabilities. For instance, the MoX<sub>2</sub> (X = S, Se, Te) series, based on different levels of stacking between the layers, can be observed in several phases and has a crossover from an indirect band gap to a direct band gap [125–127]. In general, the band gap of TMD materials is in the range of 1 to 2 eV, which is in good compliance with the solar spectrum and is comparable with the band gap of Si (1.1 eV), GaAs (1.4 eV), and CdTe (1.5 eV) [128]. Most single-layer TMDs have a direct band gap and have potential as high-performance photovoltaic applications due to their high optical absorption coefficient and interesting electrochemical properties. Therefore, material engineering methods such as forming  $Mo_{1-x}W_xS_2$  alloys, applying mechanical stress, and controlling the band gap and ionization potential, can be very efficient for preparing suitable materials [129,130].

Researchers have studied the different functions of two-dimensional TMDs in different types of solar cells and introduced the most efficient ones such as  $MoS_2/RGO$  (reduced graphene oxide) nanocomposite,  $MoS_2$ -CuS composite, and  $MoS_2/FTO$  as materials with suitable performance in counter electrodes, with ITO/ $MoS_2/Au$ ,  $MoSe_2/GaN$ ,  $n-MoS_2/i-SiO_2/p-Si$  as an active layer, PEDOT:PSS/ $MoS_2/p$ erovskite/PCBM and  $n-MoS_2/Al2O_3/p-S$  as cavity transfer layer,  $MoS_2/P_3HT:PCBM/V_2O_5$ , and  $MS_2/P_3HT:PCBM/V_2O_5$  as the electron transfer layer [128].

It has been reported that  $MoS_2$  and  $WS_2$  as auxiliary electrode materials in dyesensitized solar cells (Figure 6) have increased electrocatalytic activity due to their high electron mobility, good stability, and low cost. These materials have also been found to improve the light harvesting efficiency of DSSCs and to increase their stability compared to other semiconductors that are commonly used in these devices, such as  $TiO_2$ .



**Figure 6.** The schematic of MoS<sub>2</sub>/graphene nanocomposite-based photoanode for dye-sensitized solar cells (DSSCs). Reprinted (adapted) with permission from Reference [131], Copyright (2020), Elsevier.

The use of TMDs as counter electrodes and light absorbers has increased power conversion efficiency [132]. It has also been shown that it is possible to increase the efficiency of the solar cell by using TMDs as the cavity carrier layer and by optimizing the layer thickness. In this regard, various combinations of these materials with graphene,

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graphene oxide, rGO,  $WS_2$ ,  $MoS_2$  as a hole transport layer (HTL), and counter electrode (CE) materials in PV cells have been investigated [133].

# 3.3. Schottky Junction of Solar Cells

A Schottky junction is a type of metal-semiconductor junction commonly used in solar cells. It is formed by metals, such as aluminum or platinum, coming into contact with a semiconductor material, such as silicon. The junction is named after Walter H. Schottky, who first described it in the late 1920s. The use of Schottky junctions in solar cells provides several advantages. For one, they enable low-resistance electrical contacts, which help to increase the efficiency of the solar cell. They can also provide a low reverse-bias leakage current, which helps to reduce power losses in the cell. In addition, Schottky junctions can be made with a wide range of metal materials, which makes them versatile for use in various types of solar cells.

#### 3.4. TMDs in Schottky Junction Solar Cells

In Schottky junction solar cells, two-dimensional TMDs are used in connection with metals or graphene to form interfaces for the separation of photoinduced charge carriers [122]. At the Schottky junction, a sudden potential difference occurs, the Schottky barrier, due to the difference in energy levels between the metal Fermi level and the conduction or the valence band of the TMD [134].

In  $Au/MoS_2$ -based Schottky junction solar cells for 220 nm  $MoS_2$  stacks, a power conversion efficiency equal to 1.8% was reported [135]. In another study, a  $WS_2$ /graphene-based solar cell in which  $WS_2$  with a thickness of 37 nm in contact with multilayer graphene was able to produce a PCE equal to 3.3% [136]. Monolayer TMDs absorb 5 to 10 percent of the sunlight which is higher than GaAs and Si absorbance of the same thickness (less than one nanometer) [137].

Using ultra-thin/two-dimensional  $MoS_2$  nanosheets as an effective hole extraction layer in organic solar cells, researchers achieved a relatively high PCE of 8.11% and by a graphene/ $MoS_2$ /n-Si based photovoltaic device gained a PCE of 11.1% [138].

The presence of non-covalent interaction and the absence of dangling bonds on the surfaces of two-dimensional TMD is the reason for their high resistance to reaction with other chemical species. In addition, layered TMD non-bonded surfaces enable the use of heterogeneous structures, without lattice matching limitations, so called van der Waals heterostructures, which offer many options for designing TMD photovoltaics. Therefore, nanosheets of different materials are expected to be used in organic solar cells to improve stability [139].

According to realistic balance models, in very thin TMD single-junction solar cells a PCE close to 27% can be obtained. Despite this prediction, the real PCE of TMD solar cells is usually no more than 2% which is mainly due to the damage done to the TMDs. Pinning of the Fermi level of the metals as well as doping by non-practical methods such as diffusion and ion implants and fabricating on a flexible substrate can cause contamination or damage to TMD heterojunctions and reduce performance.

One of the factors that significantly improved the performance of WSe $_2$  solar cells which led to a record of 5.1% PCE and a record of power of 4.4 W g $^{-1}$ , was the reduction or elimination of Fermi level pinning by adopting a metal transfer method. Additionally, putting a very thin intermediate layer of transparent graphene in the TMD-metal junction effectively improved the performance of the target cell.

The use of  $MoO_x$  coating with an anti-reflective effect for doping and a direct transfer method for fabrication as a clean and harmless procedure for the light-weight flexible polyimide substrates are other effective factors contributing to the record of PCE [140,141].

Moreover, the use of TMD-compatible doping methods to form a single p-n junction, such as surface charge transfer and fixed charge doping through metal oxides, plasma doping, or electrostatic doping, has significantly improved the performance of solar cells. The highest PCE is 2.8% in plasma-doped MoS<sub>2</sub> thin-film single-bonded TMDs and 6.3%

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in  $MoSe_2$  solar cells electrostatically doped. TMD-based solar cells have been reported to suffer from relatively low external quantum efficiency (EQE) and low open-circuit voltage due to the design and construction of the device that is not optimized [142].

The band energy of the WSe<sub>2</sub> is approximately 1.3 eV. Extremely high electron-hole mobility and high absorption coefficient ( $10^5$  cm<sup>-1</sup> at 780 nm compared to silicon  $10^3$  cm<sup>-1</sup>) make it one of the most promising TMD materials for high-performance photovoltaic applications.

Most work in TMD photovoltaics has been concentrated on the monolayer type due to the direct nature of the band gap, but monolayer TMDs can only absorb 10% of the incoming light. Although the absorption can be increased by using intensifiers, the use of multilayer WSe<sub>2</sub> has been very effective.

Jariwala et al. showed that a suitable photonic design for  $WSe_2$  multilayer shells and the use of a silverback reflector (Figure 7) can maximize absorption through Fabry–Perot effects to achieve an excellent absorption of over 90% with a thickness of approximately 15 nm [143].

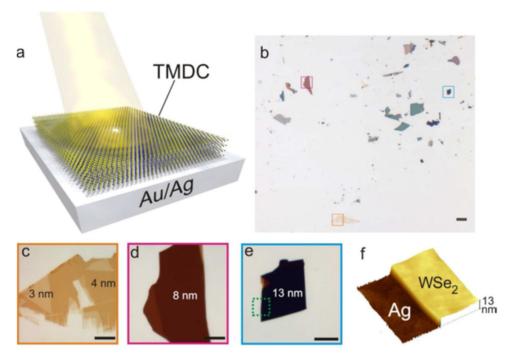
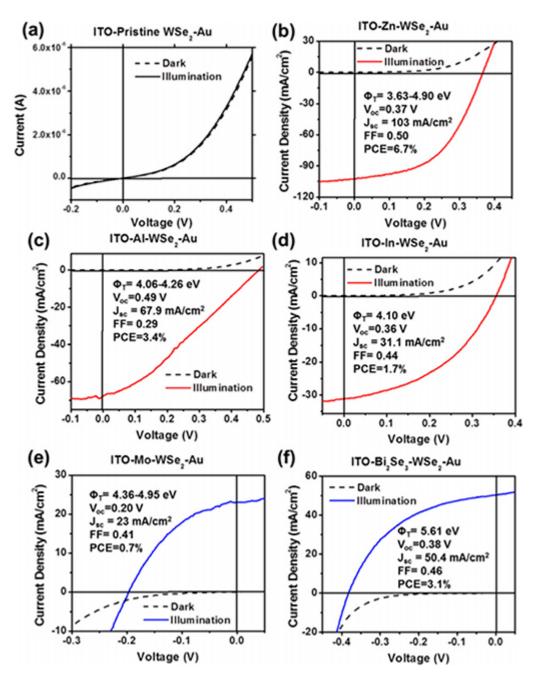


Figure 7. Absorbing TMD on metals: (a) Schematic diagram of a thin, multilayer TMDC film on an Au/Ag back reflecting substrate. (b) Low magnification optical micrograph of exfoliated WSe<sub>2</sub> flakes on template stripped Ag substrate. (Scale bar =  $50 \mu m$ ) (c-e) High magnification micrographs of yellow, red, and blue square regions on (b) respectively with increasing flake thickness from (c) to (e). The sharp blue shift in color and rising contrast with increasing thickness can be seen (Scale bar =  $10 \mu m$ ). (f) AFM topography of the flake region in (e). Reprinted (adapted) with permission from Reference [143], Copyright (2016), American Chemical Society.

Went et al. for investigation of the weakness of low open-circuit voltage in TMD-based solar cells, introduced a new metallization method for the junction line between the TMD absorber layer and metal, showing that gold transferred to the 16 nm  $WS_2$  layer could create a large Schottky barrier. This results in a PCE of 0.46% and an open-circuit voltage of 0.256 V at 1.5 AM [134].

Wi and colleagues also studied various metals in contact with  $WSe_2$  and showed that the high Schottky barrier between the zinc joints and the 100 nm layer of  $WSe_2$  (Figure 8) can increase the open-circuit voltage by 0.35 volts [144].

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**Figure 8.** Photovoltaic characteristics of representative WSe<sub>2</sub> PV devices coated with various top metals, including (b) Zn, (c) Al, (d) In, and (e) Mo, as well as (f) semimetal Bi<sub>2</sub>Se<sub>3</sub> nanocrystals. (a) Result from a control device with no top metal coating. Reprinted (adapted) with permission from Reference [144], Copyright (2015), AIP.

However, advances have been made in optimizing the design and fabrication of TMD-based solar cells, such as the WSe<sub>2</sub> Schottky junction solar cell, which has a power conversion efficiency of 0.6–0.8% at 532 nm. However, its value is still far from the expected value of 20–22%. To meet this expectation, these devices need to be designed and built using the most advanced electronic transmission, doping, and deactivation technologies [145].

Scientists have recently investigated the effects of  $Al_2O_3$  on activation on the properties of  $Pt/WSe_2$  vertical Schottky junction solar cells. Using  $WSe_2$  multilayer as an adsorbent layer with a thickness of 150–80 nm, as the optimal thickness for photovoltaic performance and by introducing a surface deactivation technique, they have shown that it improves short circuits due to the improvement in EQE. They have reported EQE of more than 50%

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at wavelengths below 600 nm, which depends on the deactivation of the trap modes, the effects of the anti-reflective coating, and the reorientation of the carrier due to doping.

They also showed that these deactivation effects of Al<sub>2</sub>O<sub>3</sub> improved the level of the optical current collector and suggested surface deactivation techniques as an easy way to adjust the performance of an electronic TMD optical device [146].

#### 4. Conclusions

Transition metal dichalcogenides are an attractive option for researchers as unrivaled semiconductors in various fields of nanoscience and electrochemistry. We examined some applications of these materials in the field of energy production and storage. We briefly mentioned the generalities of these materials and the recent studies on the mechanisms of energy production and storage along with the applications in energy storage and production equipment such as electrochemical batteries and solar cells. It has been shown that TMDs—despite some shortcomings and problems due to their high-speed performance, long-term cycles, and high capacity in lithium-ion and sodium-ion batteries and generally other rechargeable secondary batteries such as zinc-ion, magnesium-ion, aluminum-ion, calcium-ion, and potassium-ion—can be used as suitable electrode materials in the future. In addition, TMDs have shown potential for use in the field of solar energy conversion, specifically in the development of photovoltaic devices. The unique electronic and optical properties of TMDs, such as their two-dimensional structure, high absorption coefficients, and strong light-matter interactions, make them promising for use as active materials in solar cells. TMDs have been demonstrated to have a high light conversion efficiency and are able to absorb a wide range of light frequencies, making them an attractive material for the design of next-generation solar cells with improved performance. Furthermore, TMDs have the potential to significantly reduce the cost of solar cell production by enabling the use of low-cost and scalable fabrication methods. Therefore, the use of TMDs in solar cells has the potential to greatly improve the efficiency and affordability of solar energy conversion, making them an important area of research and development for the advancement of the renewable energy industry.

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#### References

- 1. Wang, H.; Yuan, H.; Sae Hong, S.; Li, Y.; Cui, Y. Physical and chemical tuning of two-dimensional transition metal dichalcogenides. *Chem. Soc. Rev.* **2015**, *44*, 2664–2680. [CrossRef] [PubMed]
- 2. Salarizadeh, P.; Rastgoo-Deylami, M.; Askari, M.B.; Hooshyari, K. A short review on transition metal chalcogenides/carbon nanocomposites for energy storage. *Nano Futur.* **2022**, *6*, 3. [CrossRef]
- 3. Manzeli, S.; Ovchinnikov, D.; Pasquier, D.; Yazyev, O.V.; Kis, A. 2D transition metal dichalcogenides. *Nat. Rev. Mater.* **2017**, 2, 17033. [CrossRef]
- 4. Grillo, A.; Faella, E.; Pelella, A.; Giubileo, F.; Ansari, L.; Gity, F.; Hurley, P.K.; McEvoy, N.; Di Bartolomeo, A. Coexistence of negative and positive photoconductivity in few-layer PtSe<sub>2</sub> field-effect transistors. *Adv. Funct. Mater.* **2021**, *31*, 2105722. [CrossRef]
- 5. Di Bartolomeo, A.; Pelella, A.; Liu, X.; Miao, F.; Passacantando, M.; Giubileo, F.; Grillo, A.; Iemmo, L.; Urban, F.; Liang, S. Pressure-Tunable Ambipolar Conduction and Hysteresis in Thin Palladium Diselenide Field Effect Transistors. *Adv. Funct. Mater.* **2019**, 29, 1902483. [CrossRef]
- 6. Freire, R.L.; de Lima, F.C.; Fazzio, A. Vacancy localization effects on MX 2 transition-metal dichalcogenides: A systematic ab initio study. *Phys. Rev. Mater.* **2022**, *6*, 084002. [CrossRef]

Micromachines 2023, 14, 691 16 of 21

7. Splendiani, A.; Sun, L.; Zhang, Y.; Li, T.; Kim, J.; Chim, C.Y.; Galli, G.; Wang, F. Emerging photoluminescence in monolayer MoS<sub>2</sub>. *Nano Lett.* **2010**, *10*, 1271–1275. [CrossRef]

- 8. Kozawa, D.; Pu, J.; Shimizu, R.; Kimura, S.; Chiu, M.-H.; Matsuki, K.; Wada, Y.; Sakanoue, T.; Iwasa, Y.; Li, L.-J.; et al. Photodetection in p–n junctions formed by electrolyte-gated transistors of two-dimensional crystals. *Appl. Phys. Lett.* **2016**, 109, 201107. [CrossRef]
- Salarizadeh, P.; Askari, M.B. MoS<sub>2</sub>–ReS<sub>2</sub>/rGO: A novel ternary hybrid nanostructure as a pseudocapacitive energy storage material. J. Alloy. Compd. 2021, 874, 159886. [CrossRef]
- 10. Askari, M.B.; Salarizadeh, P.; Seifi, M.; Rozati, S.M.; Beheshti-Marnani, A. Binary mixed molybdenum cobalt sulfide nanosheets decorated on rGO as a high-performance supercapacitor electrode. *Nanotechnology* **2020**, *31*, 275406. [CrossRef]
- 11. Salarizadeh, P.; Askari, M.B.; Di Bartolomeo, A. MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub>/Reduced graphene oxide nanostructure as an electrocatalyst for alcohol fuel cells. *ACS Appl. Nano Mater.* **2022**, *5*, 3361–3373. [CrossRef]
- 12. Askari, M.B.; Salarizadeh, P.; Seifi, M.; Rozati, S.M. Electrocatalytic properties of CoS<sub>2</sub>/MoS<sub>2</sub>/rGO as a non-noble dual metal electrocatalyst: The investigation of hydrogen evolution and methanol oxidation. *J. Phys. Chem. Solids* **2019**, *135*, 109103. [CrossRef]
- 13. Askari, M.B.; Beheshti-Marnani, A.; Seifi, M.; Rozati, S.M.; Salarizadeh, P. Fe<sub>3</sub>O<sub>4</sub>@ MoS<sub>2</sub>/RGO as an effective nano-electrocatalyst toward electrochemical hydrogen evolution reaction and methanol oxidation in two settings for fuel cell application. *J. Colloid Interface Sci.* **2019**, 537, 186–196. [CrossRef]
- 14. Urban, F.; Martucciello, N.; Peters, L.; McEvoy, N.; Di Bartolomeo, A. Environmental Effects on the Electrical Characteristics of Back-Gated WSe2 Field-Effect Transistors. *Nanomaterials* **2018**, *8*, 901. [CrossRef] [PubMed]
- 15. Di Bartolomeo, A.; Urban, F.; Passacantando, M.; McEvoy, N.; Peters, L.; Iemmo, L.; Luongo, G.; Romeo, F.; Giubileo, F. A WSe<sub>2</sub> vertical field emission transistor. *Nanoscale* **2019**, *11*, 1538–1548. [CrossRef] [PubMed]
- 16. Di Bartolomeo, A.; Pelella, A.; Urban, F.; Grillo, A.; Iemmo, L.; Passacantando, M.; Liu, X.; Giubileo, F. Field emission in ultrathin PdSe<sub>2</sub> back-gated transistors. *Adv. Electron. Mater.* **2020**, *6*, 2000094. [CrossRef]
- 17. Urban, F.; Gity, F.; Hurley, P.K.; McEvoy, N.; Di Bartolomeo, A. Isotropic conduction and negative photoconduction in ultrathin PtSe<sub>2</sub> films. *Appl. Phys. Lett.* **2020**, *117*, 193102. [CrossRef]
- 18. Tserkezis, C.; Fernandez-Dominguez, A.I.; Gonçalves, P.A.D.; Todisco, F.; Cox, J.D.; Busch, K.; Stenger, N.; Bozhevolnyi, S.I.; Mortensen, N.A.; Wolff, C. On the applicability of quantum-optical concepts in strong-coupling nanophotonics. *Rep. Prog. Phys.* **2020**, *83*, 082401. [CrossRef]
- 19. Dickinson, R.G.; Pauling, L. The crystal structure of molybdenite. J. Am. Chem. Soc. 1923, 45, 1466–1471. [CrossRef]
- 20. Wilson, J.; Yoffe, A. The transition metal dichalcogenides discussion and interpretation of the observed optical, electrical and structural properties. *Adv. Phys.* **1969**, *18*, 193–335. [CrossRef]
- 21. Frindt, R.F.; Yoffe, A.D. Physical properties of layer structures: Optical properties and photoconductivity of thin crystals of molybdenum disulphide. *Proc. R. Soc. Lond. Ser. A Math. Phys. Sci.* **1963**, 273, 69–83. [CrossRef]
- 22. Joensen, P.; Frindt, R.; Morrison, S.R. Single-layer MoS<sub>2</sub>. Mater. Res. Bull. 1986, 21, 457–461. [CrossRef]
- 23. Tenne, R.; Margulis, L.; Genut, M.; Hodes, G. Polyhedral and cylindrical structures of tungsten disulphide. *Nature* **1992**, *360*, 444–446. [CrossRef]
- 24. Feldman, Y.; Wasserman, E.; Srolovitz, D.J.; Tenne, R. High-rate, gas-phase growth of MoS<sub>2</sub> nested inorganic fullerenes and nanotubes. *Science* **1995**, 267, 222–225. [CrossRef] [PubMed]
- 25. Medeiros, P.V.; Gueorguiev, G.; Stafström, S. Bonding, charge rearrangement and interface dipoles of benzene, graphene, and PAH molecules on Au(1 1 1) and Cu(1 1 1). *Carbon* 2014, *81*, 620–628. [CrossRef]
- 26. Lundgren, C.; Kakanakova-Georgieva, A.; Gueorguiev, G.K. A perspective on thermal stability and mechanical properties of 2D Indium Bismide from ab initio molecular dynamics. *Nanotechnology* **2022**, *33*, 335706. [CrossRef]
- Salarizadeh, P.; Rastgoo-Deylami, M.; Askari, M.B. Electrochemical properties of Ni<sub>3</sub>S<sub>2</sub>@MoS<sub>2</sub>-rGO ternary nanocomposite as a promising cathode for Ni–Zn batteries and catalyst towards hydrogen evolution reaction. *Renew. Energy* 2022, 194, 152–162. [CrossRef]
- 28. Novoselov, K.S.; Jiang, D.; Schedin, F.; Booth, T.J.; Khotkevich, V.V.; Morozov, S.V.; Geim, A.K. Two-dimensional atomic crystals. Proc. Natl. Acad. Sci. USA 2005, 102, 10451–10453. [CrossRef]
- 29. Zhao, X.; Shi, Z.; Wang, X.; Zou, H.; Fu, Y.; Zhang, L. Band structure engineering through van der Waals heterostructing superlattices of two-dimensional transition metal dichalcogenides. *Infomat* **2020**, *3*, 201–211. [CrossRef]
- 30. Phalswal, P.; Khanna, P.K.; Rubahn, H.-G.; Mishra, Y.K. Nanostructured molybdenum dichalcogenides: A review. *Mater. Adv.* **2022**, *3*, 5672–5697. [CrossRef]
- 31. Varadwaj, P.R.; Varadwaj, A.; Marques, H.M.; Yamashita, K. Chalcogen Bonding in the Molecular Dimers of WCh<sub>2</sub> (Ch = S, Se, Te): On the Basic Understanding of the Local Interfacial and Interlayer Bonding Environment in 2D Layered Tungsten Dichalcogenides. *Int. J. Mol. Sci.* 2022, 23, 1263. [CrossRef] [PubMed]
- 32. Varadwaj, P.R.; Marques, H.M.; Varadwaj, A.; Yamashita, K. Chalcogen Chalcogen Bonding in Molybdenum Disulfide, Molybdenum Diselenide and Molybdenum Ditelluride Dimers as Prototypes for a Basic Understanding of the Local Interfacial Chemical Bonding Environment in 2D Layered Transition Metal Dichalcogenides. *Inorganics* 2022, 10, 11. [CrossRef]

Micromachines 2023, 14, 691 17 of 21

33. Iqbal, M.; Elahi, E.; Amin, A.; Hussain, G.; Aftab, S. Chemical doping of transition metal dichalcogenides (TMDCs) based field effect transistors: A review. *Superlattices Microstruct.* **2019**, *137*, 106350. [CrossRef]

- 34. Grillo, A.; Passacantando, M.; Zak, A.; Pelella, A.; Di Bartolomeo, A. WS<sub>2</sub> Nanotubes: Electrical Conduction and Field Emission under Electron Irradiation and Mechanical Stress. *Small* **2020**, *16*, 2002880. [CrossRef] [PubMed]
- 35. Grillo, A.; Di Bartolomeo, A. A Current–Voltage Model for Double Schottky Barrier Devices. *Adv. Electron. Mater.* **2021**, *7*, 2000979. [CrossRef]
- 36. Di Bartolomeo, A. Emerging 2D Materials and Their Van Der Waals Heterostructures. Nanomaterials 2020, 10, 579. [CrossRef]
- 37. Darsara, S.A.; Seifi, M.; Askari, M.B. One-step hydrothermal synthesis of MoS<sub>2</sub>/CdS nanocomposite and study of structural, photocatalytic, and optical properties of this nanocomposite. *Optik* **2018**, *169*, 249–256. [CrossRef]
- 38. Askari, M.B.; Salarizadeh, P. Binary nickel ferrite oxide (NiFe<sub>2</sub>O<sub>4</sub>) nanoparticles coated on reduced graphene oxide as stable and high-performance asymmetric supercapacitor electrode material. *Int. J. Hydrogen Energy* **2020**, 45, 27482–27491. [CrossRef]
- 39. Askari, M.B.; Rozati, S.M. Construction of Co<sub>3</sub>O<sub>4</sub>-Ni<sub>3</sub>S<sub>4</sub>-rGO ternary hybrid as an efficient nanoelectrocatalyst for methanol and ethanol oxidation in alkaline media. *J. Alloy. Compd.* **2021**, 900, 163408. [CrossRef]
- 40. Tajik, S.; Beitollahi, H.; Askari, M.B.; Di Bartolomeo, A. Screen-printed electrode surface modification with NiCo<sub>2</sub>O<sub>4</sub>/RGO nanocomposite for hydroxylamine detection. *Nanomaterials* **2021**, *11*, 3208. [CrossRef]
- 41. Salarizadeh, P.; Askari, M.B.; Askari, N.; Salarizadeh, N. Ternary transition metal chalcogenides decorated on rGO as an efficient nanocatalyst towards urea electro-oxidation reaction for biofuel cell application. *Mater. Chem. Phys.* **2019**, 239, 121958. [CrossRef]
- 42. Askari, M.B.; Salarizadeh, P.; Seifi, M.; Zadeh, M.H.R.; Di Bartolomeo, A. ZnFe<sub>2</sub>O<sub>4</sub> nanorods on reduced graphene oxide as advanced supercapacitor electrodes. *J. Alloy. Compd.* **2020**, *860*, 158497. [CrossRef]
- 43. Askari, M.B.; Salarizadeh, P.; Rozati, S.M.; Seifi, M. Two-dimensional transition metal chalcogenide composite/reduced graphene oxide hybrid materials for hydrogen evolution application. *Polyhedron* **2019**, *162*, 201–206. [CrossRef]
- 44. Askari, M.B.; Salarizadeh, P. Ultra-small ReS<sub>2</sub> nanoparticles hybridized with rGO as cathode and anode catalysts towards hydrogen evolution reaction and methanol electro-oxidation for DMFC in acidic and alkaline media. *Synth. Met.* **2019**, 256, 116131. [CrossRef]
- 45. Askari, M.B.; Rozati, S.M.; Salarizadeh, P.; Azizi, S. Reduced graphene oxide supported Co<sub>3</sub>O<sub>4</sub>–Ni<sub>3</sub>S<sub>4</sub> ternary nanohybrid for electrochemical energy storage. *Ceram. Int.* **2022**, *48*, 16123–16130. [CrossRef]
- 46. Shi, Z.; Huang, H.; Wang, C.; Huo, M.; Ho, S.-H.; Tsai, H.-S. Heterogeneous transition metal dichalcogenides/graphene composites applied to the metal-ion batteries. *Chem. Eng. J.* **2022**, 447, 137469. [CrossRef]
- 47. Gao, Y.; Deng, C.; Fan, Y.; Wang, S.; Huang, S.; Lu, S.; Tao, T. A liquid cathode/anode based solid-state lithium-sulfur battery. *Electrochim. Acta* **2022**, 421, 140456. [CrossRef]
- 48. Fabbrizzi, L. Strange Case of Signor Volta and Mister Nicholson: How Electrochemistry Developed as a Consequence of an Editorial Misconduct. *Angew. Chem. Int. Ed.* **2019**, *58*, 5810–5822. [CrossRef]
- 49. Dunn, B.; Kamath, H.; Tarascon, J.-M. Electrical Energy Storage for the Grid: A Battery of Choices. *Science* **2011**, *334*, 928–935. [CrossRef]
- 50. Winter, M.; Brodd, R.J. What are batteries, fuel cells, and supercapacitors? Chem. Rev. 2004, 104, 4245–4270. [CrossRef]
- 51. Li, G.; Guo, S.; Xiang, B.; Mei, S.; Zheng, Y.; Zhang, X.; Gao, B.; Chu, P.K.; Huo, K. Recent advances and perspectives of microsized alloying-type porous anode materials in high-performance Li- and Na-ion batteries. *Energy Mater.* **2022**, *2*, 200020. [CrossRef]
- 52. Gayathri, A.; Manimegalai, V.; Krishnakumar, P. Challenging Issues and Solutions on Battery Thermal Management for Electric Vehicles. *Electr. Electron. Devices Circuits Mater. Technol. Chall. Solut.* **2021**, *28*, 535–553. [CrossRef]
- 53. Zhang, Z.; Jin, H.; Zhu, J.; Li, W.; Zhang, C.; Zhao, J.; Luo, F.; Sun, Z.; Mu, S. 3D flower-like ZnFe-ZIF derived hierarchical Fe, N-Codoped carbon architecture for enhanced oxygen reduction in both alkaline and acidic media, and zinc-air battery performance. *Carbon* 2020, 161, 502–509. [CrossRef]
- 54. Kim, H.; Jang, Y.-C.; Hwang, Y.; Ko, Y.; Yun, H. End-of-life batteries management and material flow analysis in South Korea. *Front. Environ. Sci. Eng.* **2018**, 12, 3. [CrossRef]
- 55. Yuan, X.; Ma, F.; Zuo, L.; Wang, J.; Yu, N.; Chen, Y.; Zhu, Y.; Huang, Q.; Holze, R.; Wu, Y.; et al. Latest Advances in High-Voltage and High-Energy-Density Aqueous Rechargeable Batteries. *Electrochem. Energy Rev.* **2020**, *4*, 1–34. [CrossRef]
- 56. He, X.; Bresser, D.; Passerini, S.; Baakes, F.; Krewer, U.; Lopez, J.; Mallia, C.T.; Shao-Horn, Y.; Cekic-Laskovic, I.; Wiemers-Meyer, S.; et al. The passivity of lithium electrodes in liquid electrolytes for secondary batteries. *Nat. Rev. Mater.* **2021**, *6*, 1036–1052. [CrossRef]
- 57. Alias, N.; Mohamad, A.A. Advances of aqueous rechargeable lithium-ion battery: A review. *J. Power Sources* **2015**, 274, 237–251. [CrossRef]
- 58. Gao, Y.-P.; Wu, X.; Huang, K.-J.; Xing, L.-L.; Zhang, Y.-Y.; Liu, L. Two-dimensional transition metal diseleniums for energy storage application: A review of recent developments. *Crystengcomm* **2016**, *19*, 404–418. [CrossRef]
- 59. Li, C.-C.; Zhang, X.-S.; Zhu, Y.-H.; Zhang, Y.; Xin, S.; Wan, L.-J.; Guo, Y.-G. Modulating the lithiophilicity at electrode/electrolyte interface for high-energy Li-metal batteries. *Energy Mater.* **2022**, *1*, 100017. [CrossRef]
- 60. You, J.; Hossain, D.; Luo, Z. Synthesis of 2D transition metal dichalcogenides by chemical vapor deposition with controlled layer number and morphology. *Nano Converg.* **2018**, *5*, 26. [CrossRef]

Micromachines 2023, 14, 691 18 of 21

61. Budania, P.; Baine, P.; Montgomery, J.; McGeough, C.; Cafolla, T.; Modreanu, M.; McNeill, D.; Mitchell, N.; Hughes, G.; Hurley, P. Long-term stability of mechanically exfoliated MoS<sub>2</sub> flakes. *MRS Commun.* **2017**, 7, 813–818. [CrossRef]

- 62. Wang, Y.; Shen, C.; Niu, L.; Li, R.; Guo, H.; Shi, Y.; Li, C.; Liu, X.; Gong, Y. Hydrothermal synthesis of CuCo<sub>2</sub>O<sub>4</sub>/CuO nanowire arrays and RGO/Fe<sub>2</sub>O<sub>3</sub> composites for high-performance aqueous asymmetric supercapacitors. *J. Mater. Chem. A* **2016**, *4*, 9977–9985. [CrossRef]
- 63. Tajik, S.; Dourandish, Z.; Nejad, F.G.; Beitollahi, H.; Jahani, P.M.; Di Bartolomeo, A. Transition metal dichalcogenides: Synthesis and use in the development of electrochemical sensors and biosensors. *Biosens. Bioelectron.* **2022**, *216*, 114674. [CrossRef] [PubMed]
- 64. Yang, H.; Giri, A.; Moon, S.; Shin, S.; Myoung, J.-M.; Jeong, U. Highly Scalable Synthesis of MoS<sub>2</sub> Thin Films with Precise Thickness Control via Polymer-Assisted Deposition. *Chem. Mater.* **2017**, 29, 5772–5776. [CrossRef]
- 65. Cullen, C.P.; Hartwig, O.; Coileáin, C.Ó.; McManus, J.B.; Peters, L.; Ilhan, C.; McEvoy, N. Synthesis and thermal stability of TMD thin films: A comprehensive XPS and Raman study. *arXiv* **2021**, arXiv:2106.07366.
- 66. Perrozzi, F.; Emamjomeh, S.; Paolucci, V.; Taglieri, G.; Ottaviano, L.; Cantalini, C. Thermal stability of WS2 flakes and gas sensing properties of WS2/WO<sub>3</sub> composite to H<sub>2</sub>, NH<sub>3</sub> and NO<sub>2</sub>. Sens. Actuators B Chem. **2017**, 243, 812–822. [CrossRef]
- 67. Wang, X.; Fan, W.; Fan, Z.; Dai, W.; Zhu, K.; Hong, S.; Sun, Y.; Wu, J.; Liu, K. Substrate modified thermal stability of mono- and few-layer MoS<sub>2</sub>. *Nanoscale* **2018**, *10*, 3540–3546. [CrossRef]
- 68. Bollero, A.; Kaupmees, L.; Raadik, T.; Grossberg, M.; Fernandez, S. Thermal stability of sputtered Mo/polyimide films and formation of MoSe<sub>2</sub> and MoS<sub>2</sub> layers for application in flexible Cu (In, Ga)(Se, S) 2 based solar cells. *Thin Solid Film.* **2012**, 520, 4163–4168. [CrossRef]
- 69. Shen, J.; Han, D.; Zhang, B.; Cao, R.; Liu, Y.; Zheng, S.; Li, H.; Jiang, Y.; Xue, Y.; Xue, M. First-principles study on phonon transport properties of MoTe<sub>2</sub> and WTe<sub>2</sub> monolayers in different phases. *Phys. E Low-Dimens. Syst. Nanostruct.* **2023**, *145*, 115509. [CrossRef]
- 70. Zuo, X.; Chang, K.; Zhao, J.; Xie, Z.; Tang, H.; Li, B.; Chang, Z. Bubble-template-assisted synthesis of hollow fullerene-like MoS<sub>2</sub> nanocages as a lithium ion battery anode material. *J. Mater. Chem. A* **2015**, *4*, 51–58. [CrossRef]
- 71. Nandi, D.K.; Sen, U.K.; Choudhury, D.; Mitra, S.; Sarkar, S.K. Atomic Layer Deposited MoS<sub>2</sub> as a Carbon and Binder Free Anode in Li-ion Battery. *Electrochim. Acta* **2014**, *146*, 706–713. [CrossRef]
- 72. Liu, L.; Qiu, H.; Wang, J.; Xu, G.; Jiao, L. Atomic MoS<sub>2</sub> monolayers synthesized from a metal–organic complex by chemical vapor deposition. *Nanoscale* **2016**, *8*, 4486–4490. [CrossRef] [PubMed]
- 73. Jing, Y.; Ortiz-Quiles, E.O.; Cabrera, C.R.; Chen, Z.; Zhou, Z. Layer-by-Layer Hybrids of MoS<sub>2</sub> and Reduced Graphene Oxide for Lithium Ion Batteries. *Electrochim. Acta* **2014**, *147*, 392–400. [CrossRef]
- 74. Wei, S.; Zhou, R.; Wang, G. Enhanced Electrochemical Performance of Self-Assembled Nanoflowers of MoS<sub>2</sub> Nanosheets as Supercapacitor Electrode Materials. *ACS Omega* **2019**, *4*, 15780–15788. [CrossRef]
- 75. Zhuang, W.; Li, L.; Zhu, J.; An, R.; Lu, L.; Lu, X.; Wu, H. Ying, Facile synthesis of mesoporous MoS<sub>2</sub>-TiO<sub>2</sub> nanofibers for ultrastable lithium ion battery anodes. *ChemElectroChem* **2015**, *2*, 374–381. [CrossRef]
- 76. Zhou, Y.; Liu, Y.; Zhao, W.; Xie, F.; Xu, R.; Li, B.; Zhou, X.; Shen, H. Growth of vertically aligned MoS<sub>2</sub> nanosheets on a Ti substrate through a self-supported bonding interface for high-performance lithium-ion batteries: A general approach. *J. Mater. Chem. A* **2016**, *4*, 5932–5941. [CrossRef]
- 77. Zhang, D.A.; Wang, Q.; Wang, Q.; Sun, J.; Xing, L.L.; Xue, X.Y. High capacity and cyclability of hierarchical MoS<sub>2</sub>/SnO<sub>2</sub> nanocomposites as the cathode of lithium-sulfur battery. *Electrochim. Acta* **2015**, *173*, 476–482. [CrossRef]
- 78. Pan, Q.C.; Huang, Y.G.; Wang, H.Q.; Yang, G.H.; Wang, L.C.; Chen, J.; Zan, Y.-H.; Li, Q.-Y. MoS<sub>2</sub>/C nanosheets Encapsulated Sn@ SnOx nanoparticles as high-performance Lithium-iom battery anode material. *Electrochim. Acta* **2016**, 197, 50–57. [CrossRef]
- 79. Chen, Y.; Song, B.; Tang, X.; Lu, L.; Xue, J. Ultrasmall Fe<sub>3</sub>O<sub>4</sub> nanoparticle/MoS<sub>2</sub> nanosheet composites with superior performances for lithium ion batteries. *Small* **2014**, *10*, 1536–1543. [CrossRef]
- 80. Chen, Y.; Lu, J.; Wen, S.; Lu, L.; Xue, J. Synthesis of SnO<sub>2</sub>/MoS<sub>2</sub> composites with different component ratios and their applications as lithium ion battery anodes. *J. Mater. Chem. A* **2014**, 2, 17857–17866. [CrossRef]
- 81. Wang, R.; Xu, C.; Sun, J.; Liu, Y.; Gao, L.; Yao, H.; Lin, C. Heat-induced formation of porous and free-standing MoS<sub>2</sub>/GS hybrid electrodes for binder-free and ultralong-life lithium ion batteries. *Nano Energy* **2014**, *8*, 183–195. [CrossRef]
- 82. Chang, K.; Chen, W.; Ma, L.; Li, H.; Huang, F.; Xu, Z.; Zhang, Q.; Lee, J.-Y. Graphene-like MoS<sub>2</sub>/amorphous carbon composites with high capacity and excellent stability as anode materials for lithium ion batteries. *J. Mater. Chem.* **2011**, 21, 6251–6257. [CrossRef]
- 83. Das, S.K.; Mallavajula, R.; Jayaprakash, N.; Archer, L.A. Self-assembled MoS<sub>2</sub>–carbon nanostructures: Influence of nanostructuring and carbon on lithium battery performance. *J. Mater. Chem.* **2012**, 22, 12988–12992. [CrossRef]
- 84. Dinh, D.A.; Nguyen, T.L.; Cuong, T.V.; Hui, K.S.; Bui, T.H.; Wu, S.; Hui, K.N. Defect-Free MoS<sub>2</sub>-Flakes/Amorphous-Carbon Hybrid as an Advanced Anode for Lithium-Ion Batteries. *Energy Fuels* **2021**, *35*, 3459–3468. [CrossRef]
- 85. Miao, Y.-E.; Huang, Y.; Zhang, L.; Fan, W.; Lai, F.; Liu, T. Electrospun porous carbon nanofiber@MoS<sub>2</sub> core/sheath fiber membranes as highly flexible and binder-free anodes for lithium-ion batteries. *Nanoscale* **2015**, 7, 11093–11101. [CrossRef] [PubMed]
- 86. Nguyen, T.M.N.; Vuong, V.-D.; Phong, M.T.; Van Le, T. Fabrication of MoS<sub>2</sub> Nanoflakes Supported on Carbon Nanotubes for High Performance Anode in Lithium-Ion Batteries (LIBs). *J. Nanomater.* **2019**, 2019, 8364740. [CrossRef]

Micromachines 2023, 14, 691 19 of 21

87. Park, S.K.; Yu, S.H.; Woo, S.; Quan, B.; Lee, D.C.; Kim, M.K.; Sung, Y.-E.; Piao, Y. A simple l-cysteine-assisted method for the growth of MoS 2 nanosheets on carbon nanotubes for high-performance lithium ion batteries. *Dalton Trans.* **2013**, 42, 2399–2405. [CrossRef]

- 88. Zhao, C.; Kong, J.; Yao, X.; Tang, X.; Dong, Y.; Phua, S.L.; Lu, X. Thin MoS<sub>2</sub> Nanoflakes Encapsulated in Carbon Nanofibers as High-Performance Anodes for Lithium-Ion Batteries. *ACS Appl. Mater. Interfaces* **2014**, *6*, 6392–6398. [CrossRef]
- 89. Wang, J.Z.; Lu, L.; Lotya, M.; Coleman, J.N.; Chou, S.L.; Liu, H.K.; Minett, A.I.; Chen, J. Development of MoS<sub>2</sub>–CNT composite thin film from layered MoS<sub>2</sub> for lithium batteries. *Adv. Energy Mater.* **2013**, *3*, 798–805. [CrossRef]
- 90. Liu, H.; Zhang, F.; Li, W.; Zhang, X.; Lee, C.S.; Wang, W.; Tang, Y. Porous tremella-like MoS<sub>2</sub>/polyaniline hybrid composite with enhanced performance for lithium-ion battery anodes. *Electrochim. Acta* **2015**, *167*, 132–138. [CrossRef]
- 91. Raju, V.; Kumar, Y.V.N.; Jetti, V.R.; Basak, P. MoS<sub>2</sub>/Polythiophene Composite Cathode as a Potential Host for Rechargeable Aluminum Batteries: Deciphering the Impact of Processing on the Performance. *ACS Appl. Energy Mater.* **2021**, *4*, 9227–9239. [CrossRef]
- 92. Zhao, X.; Mai, Y.; Luo, H.; Tang, D.; Lee, B.; Huang, C.; Zhang, L. Nano-MoS<sub>2</sub>/poly (3,4-ethylenedioxythiophene): Poly(styrenesulfonate) composite prepared by a facial dip-coating process for Li-ion battery anode. *Appl. Surf. Sci.* **2013**, 288, 736–741. [CrossRef]
- 93. Che, G.; Jirage, K.B.; Fisher, E.R.; Martin, C.R.; Yoneyama, H. Chemical-Vapor Deposition-Based Template Synthesis of Microtubular TiS<sub>2</sub> Battery Electrodes. *J. Electrochem. Soc.* **1997**, 144, 4296. [CrossRef]
- 94. Chen, S.-Y.; Wang, Z.-X.; Fang, X.-P.; Zhao, H.-L.; Liu, X.-J.; Chen, L.-Q. Characterization of TiS<sub>2</sub> as an anode material for lithium ion batteries. *Acta Phys.-Chim. Sin.* **2011**, 27, 97–102. [CrossRef]
- 95. Huang, J.; Wang, X.; Li, J.; Cao, L.; Xu, Z.; Wei, H. WS<sub>2</sub>-Super P nanocomposites anode material with enhanced cycling stability for lithium ion batteries. *J. Alloy. Compd.* **2016**, *673*, 60–66. [CrossRef]
- 96. Su, D.; Dou, S.; Wang, G. WS<sub>2</sub>@graphene nanocomposites as anode materials for Na-ion batteries with enhanced electrochemical performances. *Chem. Commun.* **2014**, *50*, 4192–4195. [CrossRef]
- 97. Liu, H.; Su, D.; Wang, G.; Qiao, S.Z. An ordered mesoporous WS2 anode material with superior electrochemical performance for lithium ion batteries. *J. Mater. Chem.* **2012**, *22*, 17437–17440. [CrossRef]
- Bhandavat, R.; David, L.; Singh, G. Synthesis of Surface-Functionalized WS<sub>2</sub> Nanosheets and Performance as Li-Ion Battery Anodes. J. Phys. Chem. Lett. 2012, 3, 1523–1530. [CrossRef]
- 99. David, L.; Bhandavat, R.; Singh, G. MoS<sub>2</sub>/graphene composite paper for sodium-ion battery electrodes. *ACS Nano* **2014**, *8*, 1759–1770. [CrossRef]
- 100. Kim, C.S.; Jeong, Y.T.; Jeong, S.K. A Preliminary Study on Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> as a Novel Electrode Material for Calcium Ion Batteries. *Adv. Mater. Res.* **2015**, *1120*, 119–122. [CrossRef]
- 101. Kitajou, A.; Yamaguchi, J.; Hara, S.; Okada, S. Discharge/charge reaction mechanism of a pyrite-type FeS<sub>2</sub> cathode for sodium secondary batteries. *J. Power Sources* **2014**, 247, 391–395. [CrossRef]
- 102. Mori, T.; Orikasa, Y.; Nakanishi, K.; Kezheng, C.; Hattori, M.; Ohta, T.; Uchimoto, Y. Discharge/charge reaction mechanisms of FeS<sub>2</sub> cathode material for aluminum rechargeable batteries at 55 °C. *J. Power Sources* **2016**, *313*, 9–14. [CrossRef]
- 103. Zhang, Y.; Liu, Z.; Zhao, H.; Du, Y. MoSe<sub>2</sub> nanosheets grown on carbon cloth with superior electrochemical performance as flexible electrode for sodium ion batteries. *RSC Adv.* **2015**, *6*, 1440–1444. [CrossRef]
- 104. Panda, M.R.; Ghosh, A.; Kumar, A.; Muthuraj, D.; Sau, S.; Yu, W.; Zhang, Y.; Sinha, A.; Weyland, M.; Bao, Q.; et al. Blocks of molybdenum ditelluride: A high rate anode for sodium-ion battery and full cell prototype study. *Nano Energy* **2019**, *64*, 103951. [CrossRef]
- 105. Mukherjee, S.; Turnley, J.; Mansfield, E.; Holm, J.; Soares, D.; David, L.; Singh, G. Exfoliated transition metal dichalcogenide nanosheets for supercapacitor and sodium ion battery applications. *R. Soc. Open Sci.* **2019**, *6*, 190437. [CrossRef]
- 106. Liang, Y.; Yao, S.; Wang, Y.; Yu, H.; Majeed, A.; Shen, X.; Li, T.; Qin, S. Hybrid cathode composed of pyrite-structure CoS<sub>2</sub> hollow polyhedron and Ketjen black@sulfur materials propelling polysulfide conversion in lithium sulfur batteries. *Ceram. Int.* **2021**, 47, 27122–27131. [CrossRef]
- 107. Pramudita, J.C.; Sehrawat, D.; Goonetilleke, D.; Sharma, N. An Initial Review of the Status of Electrode Materials for Potassium-Ion Batteries. *Adv. Energy Mater.* **2017**, *7*, 1602911. [CrossRef]
- 108. Pan, H.; Hu, Y.-S.; Chen, L. Room-temperature stationary sodium-ion batteries for large-scale electric energy storage. *Energy Environ. Sci.* **2013**, *6*, 2338–2360. [CrossRef]
- 109. Hwang, J.-Y.; Myung, S.-T.; Sun, Y.-K. Recent Progress in Rechargeable Potassium Batteries. *Adv. Funct. Mater.* **2018**, *28*, 1802938. [CrossRef]
- 110. Mao, M.; Cui, C.; Wu, M.; Zhang, M.; Gao, T.; Fan, X.; Chen, J.; Wang, T.; Ma, J.; Wang, C. Flexible ReS<sub>2</sub> nanosheets/N-doped carbon nanofibers-based paper as a universal anode for alkali (Li, Na, K) ion battery. *Nano Energy* **2018**, *45*, 346–352. [CrossRef]
- 111. Li, Q.; Xu, Y.; Yao, Z.; Kang, J.; Liu, X.; Wolverton, C.; Hersam, M.C.; Wu, J.; Dravid, V.P. Revealing the Effects of Electrode Crystallographic Orientation on Battery Electrochemistry via the Anisotropic Lithiation and Sodiation of ReS. *ACS Nano* **2018**, 12, 7875–7882. [CrossRef] [PubMed]
- 112. George, C.; Morris, A.J.; Modarres, M.H.; De Volder, M. Structural Evolution of Electrochemically Lithiated MoS<sub>2</sub> Nanosheets and the Role of Carbon Additive in Li-Ion Batteries. *Chem. Mater.* **2016**, *28*, 7304–7310. [CrossRef] [PubMed]

Micromachines 2023, 14, 691 20 of 21

113. Peng, L.; Zhu, Y.; Chen, D.; Ruoff, R.S.; Yu, G. Two-Dimensional Materials for Beyond-Lithium-Ion Batteries. *Adv. Energy Mater.* **2016**, *6*, 1600025. [CrossRef]

- 114. Hao, S.; Shen, X.; Tian, M.; Yu, R.; Wang, Z.; Chen, L. Reversible conversion of MoS<sub>2</sub> upon sodium extraction. *Nano Energy* **2017**, 41, 217–224. [CrossRef]
- 115. Chen, B.; Chao, D.; Liu, E.; Jaroniec, M.; Zhao, N.; Qiao, S.-Z. Transition metal dichalcogenides for alkali metal ion batteries: Engineering strategies at the atomic level. *Energy Environ. Sci.* **2020**, *13*, 1096–1131. [CrossRef]
- 116. Fang, X.; Hua, C.; Guo, X.; Hu, Y.; Wang, Z.; Gao, X.; Wu, F.; Wang, J.; Chen, L. Lithium storage in commercial MoS<sub>2</sub> in different potential ranges. *Electrochimica Acta* **2012**, *81*, 155–160. [CrossRef]
- 117. Bisquert, J. The Physics of Solar Cells: Perovskites, Organics, and Photovoltaic Fundamentals; CRC Press: Boca Raton, FL, USA, 2017.
- 118. Suresh, B.V. Solid State Devices and Technology; Pearson Education: Noida, India, 2010.
- 119. Husain, A.A.; Hasan, W.Z.W.; Shafie, S.; Hamidon, M.N.; Pandey, S.S. A review of transparent solar photovoltaic technologies. *Renew. Sustain. Energy Rev.* **2018**, *94*, 779–791. [CrossRef]
- 120. Huang, S. Grand Challenges and Opportunities in Photovoltaic Materials and Devices. Front. Photon. 2021, 2, 651766. [CrossRef]
- 121. Giannouli, M. Current Status of Emerging PV Technologies: A Comparative Study of Dye-Sensitized, Organic, and Perovskite Solar Cells. *Int. J. Photoenergy* **2021**, 2021, 6692858. [CrossRef]
- 122. Di Bartolomeo, A. Graphene Schottky diodes: An experimental review of the rectifying graphene/semiconductor heterojunction. *Phys. Rep.* **2016**, *606*, 1–58. [CrossRef]
- 123. Cao, X.; Tan, C.; Zhang, X.; Zhao, W.; Zhang, H. Solution-Processed Two-Dimensional Metal Dichalcogenide-Based Nanomaterials for Energy Storage and Conversion. *Adv. Mater.* **2016**, *28*, 6167–6196. [CrossRef]
- 124. Tan, C.; Cao, X.; Wu, X.-J.; He, Q.; Yang, J.; Zhang, X.; Chen, J.; Zhao, W.; Han, S.; Nam, G.-H.; et al. Recent Advances in Ultrathin Two-Dimensional Nanomaterials. *Chem. Rev.* 2017, 117, 6225–6331. [CrossRef] [PubMed]
- 125. Zhu, W.; Low, T.; Wang, H.; Ye, P.-D.; Duan, X. Nanoscale electronic devices based on transition metal dichalcogenides. 2D Mater. 2019, 6, 032004. [CrossRef]
- 126. Kulichenko, M.; Boldyrev, A.I. σ-Aromaticity in the MoS<sub>2</sub> Monolayer. J. Phys. Chem. C 2020, 124, 6267–6273. [CrossRef]
- 127. Jung, Y.; Zhou, Y.; Cha, J.J. Intercalation in two-dimensional transition metal chalcogenides. *Inorg. Chem. Front.* **2016**, *3*, 452–463. [CrossRef]
- 128. Iqbal, M.Z.; Alam, S.; Faisal, M.M.; Khan, S. Recent advancement in the performance of solar cells by incorporating transition metal dichalcogenides as counter electrode and photoabsorber. *Int. J. Energy Res.* **2019**, *43*, 3058–3079. [CrossRef]
- 129. Coogan, Å.; Gun'ko, Y.K. Solution-based "bottom-up" synthesis of group VI transition metal dichalcogenides and their applications. *Mater. Adv.* **2021**, 2, 146–164. [CrossRef]
- 130. Lin, Y.; Torsi, R.; Geohegan, D.B.; Robinson, J.A.; Xiao, K. Controllable Thin-Film Approaches for Doping and Alloying Transition Metal Dichalcogenides Monolayers. *Adv. Sci.* **2021**, *8*, 2004249. [CrossRef]
- 131. Krishnamoorthy, D.; Prakasam, A. Preparation of MoS<sub>2</sub>/graphene nanocomposite-based photoanode for dye-sensitized solar cells (DSSCs). *Inorg. Chem. Commun.* **2020**, *118*, 108016. [CrossRef]
- 132. Rashidi, S.; Rashidi, S.; Heydari, R.K.; Esmaeili, S.; Tran, N.; Thangi, D.; Wei, W. WS<sub>2</sub> and MoS<sub>2</sub> counter electrode materials for dye-sensitized solar cells. *Prog. Photovolt. Res. Appl.* **2021**, 29, 238–261. [CrossRef]
- 133. Iqbal, M.Z.; Nabi, J.; Siddique, S.; Awan, H.T.A.; Haider, S.S.; Sulman, M. Role of graphene and transition metal dichalcogenides as hole transport layer and counter electrode in solar cells. *Int. J. Energy Res.* **2019**, *44*, 1464–1487. [CrossRef]
- 134. Went, C.M.; Wong, J.; Jahelka, P.R.; Kelzenberg, M.; Biswas, S.; Hunt, M.S.; Carbone, A.; Atwater, H.A. A new metal transfer process for van der Waals contacts to vertical Schottky-junction transition metal dichalcogenide photovoltaics. *Sci. Adv.* **2019**, *5*, eaax6061. [CrossRef] [PubMed]
- 135. Tsai, M.-L.; Su, S.-H.; Chang, J.-K.; Tsai, D.-S.; Chen, C.-H.; Wu, C.-I.; Li, L.-J.; Chen, L.-J.; He, J.-H. Monolayer MoS<sub>2</sub> heterojunction solar cells. *ACS Nano* **2014**, *8*, 8317–8322. [CrossRef]
- 136. Shanmugam, M.; Jacobs-Gedrim, R.; Song, E.S.; Yu, B. Two-dimensional layered semiconductor/graphene heterostructures for solar photovoltaic applications. *Nanoscale* **2014**, *6*, 12682–12689. [CrossRef] [PubMed]
- 137. Bernardi, M.; Palummo, M.; Grossman, J.C. Extraordinary Sunlight Absorption and One Nanometer Thick Photovoltaics Using Two-Dimensional Monolayer Materials. *Nano Lett.* **2013**, *13*, 3664–3670. [CrossRef] [PubMed]
- 138. Nawz, T.; Safdar, A.; Hussain, M.; Lee, D.S.; Siyar, M. Graphene to advanced MoS<sub>2</sub>: A review of structure, synthesis, and optoelectronic device application. *Crystals* **2020**, *10*, 902. [CrossRef]
- 139. Bernardi, M.; Ataca, C.; Palummo, M.; Grossman, J.C. Optical and Electronic Properties of Two-Dimensional Layered Materials. *Nanophotonics* **2017**, *6*, 479–493. [CrossRef]
- 140. Memaran, S.; Pradhan, N.R.; Lu, Z.; Rhodes, D.; Ludwig, J.; Zhou, Q.; Ogunsolu, O.; Ajayan, P.M.; Smirnov, D.; Fernández-Domínguez, A.I.; et al. Pronounced Photovoltaic Response from Multilayered Transition-Metal Dichalcogenides PN-Junctions. *Nano Lett.* **2015**, *15*, 7532–7538. [CrossRef]
- 141. Wi, S.; Kim, H.; Chen, M.; Nam, H.; Guo, L.J.; Meyhofer, E.; Liang, X. Enhancement of Photovoltaic Response in Multilayer MoS<sub>2</sub> Induced by Plasma Doping. *ACS Nano* **2014**, *8*, 5270–5281. [CrossRef] [PubMed]
- 142. Nazif, K.N.; Daus, A.; Hong, J.; Lee, N.; Vaziri, S.; Kumar, A.; Nitta, F.; Chen, M.E.; Kananian, S.; Islam, R.; et al. High-specific-power flexible transition metal dichalcogenide solar cells. *Nat. Commun.* **2021**, *12*, 1–39. [CrossRef]

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143. Jariwala, D.; Davoyan, A.R.; Tagliabue, G.; Sherrott, M.C.; Wong, J.; Atwater, H.A. Near-Unity Absorption in van der Waals Semiconductors for Ultrathin Optoelectronics. *Nano Lett.* **2016**, *16*, 5482–5487. [CrossRef] [PubMed]

- 144. Wi, S.; Chen, M.; Li, D.; Nam, H.; Meyhofer, E.; Liang, X. Photovoltaic response in pristine WSe<sub>2</sub> layers modulated by metal-induced surface-charge-transfer doping. *Appl. Phys. Lett.* **2015**, *107*, 062102. [CrossRef]
- 145. Akama, T.; Okita, W.; Nagai, R.; Li, C.; Kaneko, T.; Kato, T. Schottky solar cell using few-layered transition metal dichalcogenides toward large-scale fabrication of semitransparent and flexible power generator. *Sci. Rep.* **2017**, *7*, 11967. [CrossRef] [PubMed]
- 146. McVay, E.; Zubair, A.; Lin, Y.; Nourbakhsh, A.; Palacios, T. Impact of Al<sub>2</sub>O<sub>3</sub> passivation on the photovoltaic performance of vertical WSe<sub>2</sub> Schottky junction solar cells. *ACS Appl. Mater. Interfaces* **2020**, 12, 57987–57995. [CrossRef] [PubMed]

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