



Article Phytic Acid-Silica System for Imparting Fire Retardancy in Wood Composites

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Abstract: Fire-retardant (FR) treated wood-based panels, used commonly in furniture and construction, need to meet stringent fire safety regulations. This study presents a novel treatment for imparting fire resistance to wood composites by applying separate solutions of phytic acid and sodium silicate onto wood particles before the hot pressing at 160 °C. A scanning electron microscopy-energy dispersive X-ray spectroscopy (SEM-EDX) analysis revealed that phytic acid and sodium silicate were uniformly distributed throughout the wood particles, and the formation of silica gel resulted in the aggregation of elemental silicon. Fourier-transform infrared spectroscopy (FTIR) displayed that phytic acid caused the thermal degradation of hemicelluloses, which led to a brownish outer appearance of the FR-treated composites. Fire performance was assessed using both limiting oxygen index (LOI) and a cone calorimeter. These techniques showed a higher LOI value and a significant reduction in heat-release rate (HRR), total heat release (THR), smoke-production rate (SPR), and total smoke production (TSP). In addition, cone calorimeter and thermogravimetric (TGA) analyses consistently showed increased char residue in treated wood composites. Moreover, internal bond strength (IB) and modulus of elasticity (MOE) of the wood composite were not significantly changed compared with those of the untreated composite. Surprisingly, in the FR-treated composite, the 24 h-thickness swelling, and the water uptake were slightly decreased. Consequently, this new treatment has the potential to increase the fire retardancy of wood composites, such as particleboard, without deteriorating the key mechanical properties.

Keywords: inorganic-organic hybrid system; sodium phytate; fire retardant; particleboard

1. Introduction

Wood-based panels such as particleboard, fibreboard, oriented strand board (OSB), and plywood are very common products for furniture and construction applications [1]. These panels are typically produced by bonding wood particles, fibres, strands, or veneers, by using water-borne, formaldehyde-based adhesives such as urea-formaldehyde (UF), melamine-urea-formaldehyde (MUF), or phenol-formaldehyde (PF) prepolymers under high pressure and heat [2]. However, due to the inherently low fire resistance of wood, fire retardant additives (FRs) are incorporated into the panels for use in public buildings where fire safety is of paramount importance [3]. Conventional FRs such as, for instance, ammonium phosphate salts, for instance, ammonium dihydrogen phosphate, ammonium polyphosphates (APP), and urea phosphate, among others, have been used in the industry



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Copyright: © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). to increase the fire retardancy of wood-based panels [4,5]. These FRs are well-known for their affordability, effectiveness, and low toxicity. Phosphate-containing compounds are particularly effective in advancing the fire resistance of wooden materials. This is due to their ability to promote the formation of char in the OH-abundant polymers and to condense as a polyphosphate to protect the inner part of the material from catching fire. In addition, they act as radical scavengers by terminating the combustion. Moreover, the ammonium component has a synergetic action with phosphate-containing compounds, as they are decomposed into N₂ or NH₃ to reduce the contact with the combustible gas O_2 .

Recently, phytic acid (C₆H₁₈O₂₄P₆), a six-fold dihydrogen phosphate ester of inositol, was explored for its prospective to upgrade the fire retardancy of lignocellulosic materials [6–11]. This phosphate-rich compound is mainly sourced from nuts, oilseeds, and cereals, and it is comprised of six phosphate groups attached to a six-carbon ring, unlike conventional phosphates obtained from minerals. However, due to its high acidity, it can cause deterioration in the mechanical properties of materials at elevated processing temperatures. To address this issue, the reaction of phytic acid with urea was suggested in order to form ammonium phytate, which reduces the acidity and provides a synergistic fire retardancy [11,12]. While ammonium phosphate-based fire retardants have been proved to substantially decrease heat release, they possess limited efficacy in suppressing smoke production. To overcome this limitation, inorganic compounds such as CuSO₄, FeSO₄, MgSO₄, $ZnSO_4$, SiO₂, Al(OH)₃, and zinc borate have been proposed in combination with phytic acid-based FRs [5,13–15]. This combined approach has been shown to further suppress smoke production and enhance fire retardancy in wood-based materials. For example, the combination of nano-SiO₂ with phytic acid has been shown to have a smoke suppression effect on solid wood [15].

Inorganic silica (SiO₂) is commonly synthesised from a sodium silicate (Na₂SiO₃, water glass) precursor solution through the sol-gel method, which requires an acid (e.g., H_2SO_4 or $Al_2(SO_4)_3)$ [16,17]. The reduction in pH causes the formation of silicic acid, which further condenses/polymerizes as a solid SiO₂ gel. The size of the SiO₂ gel particle can be controlled by varying the temperature, pH, and other parameters (e.g., particle concentration, particle charge, and surface tension) of the precursor solution [16,18]. The incorporation of SiO₂ into wood has been utilized for protection against ultraviolet (UV) light and fire [19–22]. This is achieved by impregnating the wood structure with sodium silicate solution, followed by drying. The deposition of SiO_2 in the wood matrix relies upon the reaction of sodium silicate with carbonic acid (CO₂ from air dissolved in water) to initiate the sol-gel process. However, using sodium silicate for treating wood comes with some disadvantages. The high alkali solution can potentially cause mechanical property deterioration at elevated temperatures, a fact that makes the modification of solid wood with sodium silicate a highly complicated approach [19]. Furthermore, the high viscosity of the various solutions may limit the penetration depth. However, in the wood-panel production process, it may be easier to incorporate the sodium silicate solution. In fact, spraying sodium silicate solution onto wood particles as a pre-treatment can be easily realised.

Herein, a new approach for upgrading the fire retardancy of wood composites is proposed. This process involves pre-treating wood particles by sequentially spraying them with aqueous solutions of phytic acid and sodium silicate. The acidic phytic acid acts as a catalyst for the sodium silicate to form thermally stable silica gels. The sodium ions from the sodium silicate and phytic acid are combined to form sodium phytate, which is expected to provide synergistic fire retarding properties by improving the smoke suppression of the system. This process also neutralizes the highly acidic phytic acid and the highly alkaline sodium silicate, which alleviates the detrimental effects on the adhesive curing during the hot-pressing. The chemical functionalities of the inorganic-organic hybrid wood composites produced in this work were examined by Fourier transform infrared spectroscopy (FTIR). The elemental distribution was carried out by using scanning electron microscopy–energy dispersive X-ray analysis (SEM-EDX). Additionally, the thermal behaviour was investigated by thermogravimetric analysis (TGA), while the fire performance was examined by limiting the oxygen index (LOI) and cone calorimeter. Furthermore, the main mechanical properties (e.g., IB, MOR, MOE) and moisture properties (e.g., swelling, water uptake) of the composites produced in the work were investigated in detail.

2. Materials and Methods

2.1. Materials

Industrially prepared Norway spruce (*Picea abies*) particles, with a particulate size in the range of approx. 0.25–2.00 mm and a moisture content (MC) of approx. 4%, were obtained from the company Byggelit AB (Östersund, Sweden). A 50 wt% of aqueous phytic acid solution was purchased from TCI (Tokyo, Japan). Technical-grade aqueous sodium silicate solution (i.e., 27.6% SiO₂ and 8.5% Na₂O) was acquired from VWR (Stockholm, Sweden). The analytical-grade ammonium nitrate (NH₄NO₃) was purchased from Merck KGaA (Darmstadt, Germany). The melamine-urea-formaldehyde (MUF) resin powder (Prefere 4639) was provided by the industry Dynea AS (Lillestrøm, Norway). The MUF resin powder was completely dissolved in deionized (DI) water prior to usage. All chemicals were used as received without any dissolution.

2.2. Composite Fabrication

The wood particles were initially sprayed with a 20 wt% solution of phytic acid, followed by a 20 wt% solution of sodium silicate. The solid content of phytic acid, corresponding to the dry weight of the wood particles, varied at 0%, 6%, 9%, and 12%, while the corresponding solid content of sodium silicate was 0%, 3%, 4.5%, and 6%. The specimens are hence forth denoted as unmodified, 6-FR, 9-FR, and 12-FR wood composites, respectively. The mass ratio of the phytic acid and sodium silicate (2:1) was predetermined based on the pH of the mixture, and it was close to the pH value of the 50 wt% MUF adhesive with 1 wt% NH₄NO₃ catalyst solution (i.e., pH 6.5). The wet particles were then oven-dried at 60 °C for 24 h so as to reduce the MC of wood. A total of 12 wt% solid content of MUF adhesive (corresponding to the dry weight of particles) was used for all wood composites.

The MUF adhesive sprayed wood particles were pre-pressed by hand at room temperature in a mould measuring 210 mm \times 210 mm, before undergoing the hot-pressing (Höfer Presstechnik GmbH, Taiskirchen im Innkreis, Austria) at 160 °C for 13 min. The mat density of the wood particles before the hot-pressing was approx. 300 kg/m³. A three-step process of reducing pressure from 3.3 MPa to 2.4 MPa, and finally to 1.4 MPa, was applied. Each step lasted for 4.3 min. Four replicates for each treatment were manufactured, having a target thickness of 10 mm and a target panel density of 900 kg/m³. Following the hotpressing, the wood composites were cooled down to room temperature, surface sanded, trimmed, and climatized at 20 °C and 65% relative humidity before further analysis. The illustration of the wood-composite manufacturing is shown in Figure 1a. The mechanism of silica formation using sodium silicate as a precursor is shown in Figure 1b.



Figure 1. (a) Schematic of the wood-composite manufacturing, and (b) the mechanism of silica formation using sodium silicate as a precursor.

2.3. Characterization

The infrared spectra were recorded using a PerkinElmer FTIR Frontier spectrometer (Waltham, MA, USA) equipped with a UATR Diamond/ZnSe ATR (single reflection). The surface of the specimens was collected using a knife before being analysed over the wavenumber range of 4000–650 cm^{-1} with four scans at a resolution of 4 cm^{-1} . Three replicates were used. The pH of the wood composites was measured by a Metrohm pH meter 744 (Herisau, Switzerland). This value was determined by soaking 0.15 g of wood composites powder in 6 g of DI water. Scanning electron microscopy– energy dispersive X-ray (SEM-EDX, JeolJSM-IT300LV, Tokyo, Japan) was used to analyse the morphology and the elemental distribution of the specimens under low-vacuum mode at 100 Pa through secondary electron (SE) mode. The acceleration voltage of the electron beam was set at 10 kV for the SEM, and 15 kV for EDX, respectively. The spectrometer was operated using Oxford Instruments ZAtec V3.1 software (Buckinghamshire, UK). A surface sample of the specimen was collected using a knife and placed on conductive carbon tape without coating. Scanning times of about 300 sec were used for the EDX mapping. A thermogravimetric analysis (TGA) was carried out using a PerkinElmer TGA 4000 (Waltham, MA, USA). About 5 mg of specimen was collected with a knife and loaded in an alumina crucible before being heated from 30 °C to 800 °C at a constant heat rate of 10 °C min⁻¹ under a purse gas N_2 at a flow rate of 20 mL min⁻¹. The first-order derivative curve of the TGA (DTG curve) was smoothed using a 20-point smooth algorithm through Metter Toledo STARe Evaluation software V16.20. A limiting oxygen index (LOI, Fire Testing Technology Ltd., East Grinstead, UK) was performed (according to the ISO 4589 standard) [23] on five replicates with dimensions of $10 \times 10 \times 150$ mm. The cone calorimeter test was performed by following the ISO 5660-1 standard [24] under a heat flux of 50 kW/m² on five replicates with dimensions of $100 \times 10 \times 100$ mm. The modulus of elasticity (MOE) and modulus of rupture (MOR) were determined on five specimens with dimensions of $180 \times 10 \times 10$ mm through the use of a by four-point bending test described in the EN 408 standard [25] using an MTS Systems Corporation Criterion Model 43 universal testing machine (Créteil, France). The internal bond strength (IB) was determined according to the EN319 standard [26]. The thickness swelling of the specimens after 2 h and 24 h of soaking in distilled water was determined according to the EN317 standard [27]. The water uptake during the thickness swelling test was also evaluated.

3. Results and Discussion

3.1. Chemical Functionalities, Appearance, and Elemental Distribution

The chemical functionalities of the specimens analyzed by FTIR, are presented in Figure 2a. The spectrum of the untreated wood is attributed to the wood polymeric constituents of cellulose, hemicelluloses, and lignin [28]. After spraying the MUF adhesive and following the hot-pressing, the produced wood composite ('unmodified') revealed the band shifting from 1738 cm⁻¹ to 1727 cm⁻¹, which is due to the autocatalytic thermal degradation of principal hemicellulose, galactoglucomannan, during the hot-pressing, by cleaving the O-acetyl bond with the formation of acetic acid [29]. The band originally at 808 cm⁻¹, corresponding to the C–H out-of-plane bending in lignin, was shifted to the band at 812 cm⁻¹. This shift was due to the overlapping with the band attributed to the triazole rings from the melamine of the MUF adhesive [28,30].

The introduction of FRs changed the chemical functionalities of the wood composites. The band originally located at 1727 cm^{-1} of the unmodified wood composite was further shifted to the lower band at 1722 cm^{-1} along with the increasing concentration of the FR additives. This is probably related to the acidity of the phytic acid acting as a catalyst, promoting the thermal degradation of hemicelluloses by cleaving the O–acetyl bonds [8,9]. This is supported by the pH values of the unmodified, 6-FR, 9-FR, and 12-FR wood composites, which measured 4.84, 2.95, 2.75, and 2.72, respectively. Additionally, the protonation of acetic acid could also lead to the band shifting. The thermal degradation can also cause the brownish colour of the treated specimens, as shown in Figure 2b. The

band at 1657 cm⁻¹ assigned to the conjugated C=O stretching was shifted to a lower band and was more pronounced at 1638 cm⁻¹. This is likely attributed to peak overlapping with the vibration stretching of O–P–O from phytic acid at 1638 cm⁻¹ [31]. The band at about 1000 cm⁻¹ is related to the vibration stretching of the ether bond C–O–C of the wood components cellulose, hemicelluloses, and lignin, which is overlapped with the Si–O–Si network of SiO₂ from sodium silicate [32]. The band at 812 cm⁻¹ contributed to the MUF adhesive [28]. Finally, the other bands of the FR-modified wood composite are similar to those of the unmodified wood composite.



Figure 2. Chemical functionalities and appearance of the wood composites: (**a**) FTIR spectra and (**b**) appearance of the unmodified, 6-FR, 9-FR, and 12-FR wood composites.

The SEM-EDX analysis of the 12-FR specimens is presented in Figure 3a–h. The SEM image in Figure 3a revealed a much less porous anatomical appearance of the wood, as seen in the lumina, pits, bordered pits, and resin canals, due to the compression of the wood particles during hot-pressing. The mechanical grinding for making wood particles could also lead to damage in the particles, resulting in fewer anatomical features. Moreover, the filling of the MUF adhesive to the porous structure of wood may have contributed to the smoother surface [33].



Figure 3. SEM-EDX analysis: (**a**) SEM image (**b**) EDX spectrum, and (**c**–**h**) the elemental mapping of elemental C, O, N, Na, Si, and P, respectively. The 12-FR wood composite was used for this investigation.

The EDX spectrum of the SEM image in Figure 3b shows the main composition of the 12-FR specimen, including C, N, O, F, Na, Al, Si, and P. The presence of elemental Al is due to the sample holder, while the presence of elemental F is a result of the microtome blade contamination during the sample preparation. The elemental C and O are mainly

derived from cellulose, hemicelluloses, and lignin, as well as from the MUF resin and phytic acid [28]. The elemental N is mainly attributed to the nitrogenous components of the MUF adhesive, such as urea and melamine. The elemental P is attributed to the phytic acid. The elemental Na and Si mostly originated from the sodium silicate.

Characteristically, the weight ratio of Na:Si over the mapping image in the EDX spectra was 0.67, which is close to the theoretical value of the sodium silicate applied (0.69). It was unusual to note an uneven distribution of elemental Si by EDX mapping, which shows as white spots in the SEM image and elemental Si mapping (Figure 3g). This segregation is likely the result of the condensation of the silica gel [20].

These findings suggest that the aqueous solutions of sodium silicate, phytic acid, and MUF adhesive were evenly sprayed during the FR wood composite manufacturing, while the formation of silica gelled to a higher contrast in specific areas of the SEM image.

3.2. Thermal Behaviour

The thermal behaviour of the wood composites produced with different concentrations of FR additives determined by TGA is presented in Figure 4. The unmodified wood composite exhibited three stages of thermal degradation. The first stage, beginning at 30 °C and ending at around 120 °C, is related to the removal of the bound water. The following stage, between 250 °C and 420 °C, with a maximum decomposition temperature at 364 °C, is related to the thermal decomposition of wood components, namely hemicelluloses and cellulose [28]. At this stage, volatile degradation products such as levoglucosan, furfural, acetic acid, CO₂, CO, and H₂O are produced [34,35]. The main thermal decomposition of MUF adhesive also occurs at this stage, involving the cleavage of side groups in the MUF polymeric network, with the release of small molecules such as CO₂, NH₃, methanol, formaldehyde, and amines [36,37]. Temperatures above 420 °C, up to approximately 700 °C, are related to the thermal decomposition of the most thermally stable constituent, which is lignin, as well as to the decomposition of the condensed cyameluric (i.e., heptazine) structures of the MUF adhesive [28,38]. The mass residue of the unmodified wood composite at 800 °C is approximately 2.6 wt% of its initial mass.



Figure 4. TGA and DTG curves of the unmodified 6-FR, 9-FR, and 12-FR wood composites.

The addition of FRs in the wood structure had a significant impact on the thermal behaviour of the composites, as evidenced by the lower main decomposition temperature and the increase in residue. For all FR wood composites, the main decomposition temperature was evidently shifted to a lower temperature of around 310 °C. This is mainly due to the influence of phytic acid with the promotion of dense char at a lower temperature to prevent further thermal degradation of the material [8,9]. Furthermore, the condensation

of phosphate groups led to the formation of polyphosphate, which acts as a protective insulation layer [39]. Silica gel is supposed to act as an incombustible inert material, thereby increasing the final char residue [40]. As the concentration of FRs increases, the amount of residue at 800 °C also increases. Noticeably, the 12-FR specimen exhibited the highest char residue, at 26% of its initial mass at 800 °C.

3.3. Fire Performance

The fire performance of the wood composite examined by the limiting oxygen index (LOI) is presented in Figure 5a. A higher LOI value indicates a higher level of O_2 concentration required to sustain combustion, which simply suggests a higher fire resistance [5]. The result showed that the 12-FR wood composite had the highest LOI value, which corresponded to the highest amount of FR added to the specimen.



Figure 5. Fire performance according to (**a**) limiting oxygen index (LOI), and (**b**–**f**) the cone calorimeter results of the heat release rate (HRR), total heat release (THR), smoke-production rate (SPR), total smoke production (TSP), and mass loss of the unmodified, 6-FR, 9-FR, and 12-FR wood composites produced, respectively.

The fire performance of the wood composite was further evaluated using a cone calorimeter in order to analyse its heat and smoke release behaviour, as depicted in Figure 5b–f and Table 1. It is worthy of note that the heat-release rate (HRR) is the most critical parameter for identifying the fire retarding properties of a material. The unmodified wood composite exhibited typical heat-release behaviour, with two HRR peaks, as shown in Figure 5b. The first peak occurs during surface oxidation, releasing heat, while the valley before the second peak results from surface charring. The second peak is a result of the overheating of the specimen in its entire thickness, which is typical for thermally thick materials placed on an insulating material [41]. The subsequent reduction of HRR arises out of the glowing of the char after consuming volatile products [28,42].

Table 1. Time to ignition (TTI), first peak of heat release rate (pHRR₁), second peak of heat release rate (pHRR₂), total heat release (THR), maximum average rate of heat emission (MARHE), effective heat of combustion (EHC), total smoke production (TSP), residue mass, and time to flashover values for the unmodified, 6-FR, 9-FR, and 12-FR wood composites. The values in parentheses are the standard deviations.

Specimen	Unmodified	6-FR	9-FR	12-FR
TTI (s)	24 (2)	19 (3)	21 (4)	25 (2)
$pHRR_1 (kW/m^2)$	200 (12)	168 (14)	157 (4)	135 (2)
$pHRR_2 (kW/m^2)$	314 (17)	193 (26)	222 (11)	203 (35)
THR (MJ/m^2)	146 (11)	101 (14)	102 (5)	91 (6)
TSP (m ²)	421 (44)	88 (23)	124 (15)	208 (169)
Residue mass (%)	8.39 (1.03)	15.95 (1.47)	21.10 (1.25)	30.11 (4.55)
EHC (MJ/kg)	15.03 (0.98)	12.48 (1.62)	12.17 (0.43)	11.19 (1.03)
MARHE (kW/m ²)	185 (7)	118 (6)	116 (1)	91 (8)
Time to flashover (s)	433 (41)	566 (85)	740 (133)	1616 (648)

The addition of FRs substantially reduced the heat release of the wood composites, resulting in lower peaks of HRR and total heat release (THR). The phosphate group from the phytic acid is particularly effective in reducing the heat release of lignocellulosic materials, since it acts as a radical scavenger, promotes char residue, and forms protective insulation layers. During combustion, the production of P radicals can scavenge the O and H radicals that are produced, thus terminating the combustion behaviour. Protons from phytic acid can accelerate the dehydration of the hydroxyl groups from wood particles and promote char residue. Simultaneously, the release of water molecules causes a decrease of concentration, or partial pressure, of oxygen in the combustion zone. The dehydration of phosphate with the formation of polyphosphate can create protective glassy layers that prevent further combustion. Silica gel acts as a physical barrier, suppressing smouldering and flaming combustion by preventing flammable products from escaping and preventing the matrix from contacting the O₂ [39].

Regarding the smoke production, Figure 5d,e shows that the FRs can greatly suppress the smoke production. However, it should be noted that the total smoke production(TSP) increased proportionally to the concentration of FRs, primarily due to the limited smoke suppression ability of phosphate-based fire retardants such as phytic acid on lignocellulosic materials [5,43]. To achieve better smoke suppression, a combination with inorganic compounds, such as the synergetic effect of phytic acid and sodium ions, is suggested, which has been shown to be more efficient than phytic acid alone [44]. This result implies that the amount of sodium silicate can be further increased for better smoke suppression. Additionally, based on the results of FTIR and the brown colour of the FR wood composites, the FR wood composites were rather more acidic than the unmodified specimen during the hot-pressing. This result actually confirms the potential of increasing the amount of alkaline sodium silicate for the optimization of smoke suppression. However, it is important to consider the influence of pH on the hardening of the MUF adhesive. The mass residue of the specimens was found to be proportional to the amount of FR additives; this alone confirms the promotion of char residue, as observed in the TGA analysis.

Furthermore, Table 1 includes the results of other key fire performance parameters, such as time to ignition (TTI), effective heat of combustion (EHC), maximum average rate of heat emission (MARHE), and prediction of time to flashover. The TTI measures the time required for the material to ignite. This FR treatment had a negligible effect on TTI, possibly due to the similar density between the groups [45]. The EHC represents the amount of heat released per unit mass of material burned. The MARHE is a single value estimation of the fire performance originally defined in the EN 45545-2 standard [46,47] for comparing the fire hazard of different materials. A lower value of MARHE indicates a lower fire hazard. The prediction of time to flashover estimates the early stages of a fire from the ignition source up to the flashover [48], and this calculation is based on the THR and TTI of the cone calorimeter by the method developed by [49]. A higher value means a lower fire growth hazard of the material. The improvement in these values of the FR wood composites suggests that the treatment effectively enhances the fire retardancy. Overall, the LOI and cone calorimeter results support the conclusion that the fire retarding properties of wood composites were significantly enhanced by combining the phytic acid and sodium silicate as FRs. The mechanism of phytic acid and sodium silicate treatment can be concluded as involving char formation, the creation of inert layers by polyphosphate and silica gel, and radical scavenging. Additionally, the smoke production was suppressed, which is likely related to the use of sodium silicate.

3.4. Mechanical and Physical Properties

The impact of FRs on the mechanical and physical properties of the treated wood composite, was assessed through the four-point bending test, internal bond strength (IB), water uptake, and thickness swelling. The results are presented in Figure 6. The MOE and MOR (Figure 6a,b indicates that the FRs do not significantly influence the MOE, while the MOR seemed to be slightly reduced. The MOR is determined by the maximum load that the material can withstand before fracture. The reduction of MOR might be due to the extensive thermal degradation of the hemicelluloses caused by phytic acid during hot-pressing [50], thus affirming the observation of the FTIR analysis.



Figure 6. Mechanical and physical properties of the composite: (**a**) modulus of elasticity (MOE), (**b**) modulus of rupture (MOR), (**c**) internal bond strength (IB), (**d**) 2 h and 24 h water uptake, and (**e**) 2 h and 24 h thickness swelling.

The results of IB strength and water uptake/thickness swelling after 2 h and 24 h are presented in Figure 6c–e. The IB is a fundamental indicator of the adhesive bonding in wood panels [4]. The average IB for all testing specimens was estimated to be about 1.3 MPa–1.6 MPa. The addition of FRs did not have any considerable impact on the IB strength.

The thickness swelling test was conducted to evaluate the specimen's tolerance in a humid environment. The results showed that the unmodified wood composite had a water uptake of about 50% and a thickness swelling of about 20% after 24 h of water soaking. Uncommonly, the incorporation of FRs reduced the water uptake, while the extent of thickness swelling in the water was maintained; this result is positive. It should be noted that conventional ammonium phosphate salt-based retardants generally deteriorate the mechanical properties, e.g., inferior IB and thickness swelling. This is because the bonding between the wood particles and the adhesive is partially replaced by salt, and the acidic ammonium phosphate salt affects the adhesive polymerisation [4,51]. Finally, the results show a negligible influence on the mechanical properties of FR-treated composites and a slight improvement in reducing the water uptake. This is likely due to the reinforcement of the wood composite with the polymeric network of silica gel, as well as with the voids filled by the silica gel [20].

4. Conclusions

The research study demonstrated that the combination of phytic acid and sodium silicate can effectively enhance the fire retardancy of a wood composite (e.g., particleboard), while maintaining its mechanical properties. An FTIR analysis showed changes in the chemical functionalities due to the thermal degradation of hemicelluloses and the integration of phytic acid and sodium silicate into the wood. The thermal degradation resulted in a browner appearance. A SEM-EDX analysis revealed a uniform solution spraying of MUF adhesive, phytic acid, and sodium silicate, while the formation of the SiO_2 gel caused the segregation of elemental Si signals. LOI and cone calorimeter tests showed the upgraded fire resistance of the produced FR-treated wood composite resulting from the higher LOI value, the reduced heat release, the suppressed smoke production, and the increased char residue. The introduction of 12% FR additives significantly improved the overall fireretardant properties of wood composites by reducing the release of both heat and smoke and simultaneously increasing the time to flashover. This feature of the investigated FR is exceptional, because most other FRs reduce HRR, but also concurrently increase smoke production. The TGA exhibited a lowered main decomposition temperature and increased char residue. The mechanical and physical properties of the modified composite were examined through a four-point bending test, an internal bond strength test, and thickness swelling tests, which indicated a negligible alteration in the mechanical properties, while the moisture properties were slightly improved. To conclude, this promising approach should be investigated further at the pilot-scale level, as it presents a prospective approach for the FR treatment of wood-based composites, which could increase the fire safety of the construction materials.

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