

Article



The Resistive Switching Characteristics in ZrO₂ and Its Filamentary Conduction Behavior

Chun-Hung Lai¹, Hung-Wei Chen^{1,*} and Chih-Yi Liu²

- ¹ Department of Electronic Engineering, National United University, Miaoli 36063, Taiwan; brandon@nuu.edu.tw
- ² Department and Institute of Electronic Engineering, National Kaohsiung University of Applied Sciences, Kaohsiung 80778, Taiwan; cyliu@cc.kuas.edu.tw
- * Correspondence: hwchen@nuu.edu.tw; Tel.: +886-37-382528

Academic Editor: Teen-Hang Meen

Received: 29 May 2016; Accepted: 1 July 2016; Published: 8 July 2016

Abstract: This study investigated the conduction properties of sputtered ZrO_2 exhibiting reversible and stable resistance change. Similar current distributions in on/off conduction and set/reset switching were observed in top electrodes with a diameter of 150, 250, and 350 µm. The size independence of current magnitude implied the presence of an uneven filamentary path over the electrode area. Increased current compliance was imposed on the turn-on process, and the observed increase in on-state current and turn-off threshold was attributed to incremental filament diameter. Variations in current conduction and resistance switching were analyzed by monitoring sweeping bias limits in both positive and negative polarities. These experimental observations were interpreted based on the aspect ratio of channels comprising conductive and oxidized filament portions, thereby elucidating the characteristics of filamentary resistive switching.

Keywords: filament; resistance change; top electrode

1. Introduction

Resistive random access memory (RRAM), which is a crucial class of nonvolatile memory, is an alternative to commercial floating-gate flash memory [1]. RRAM is advantageous because of its simple device structure, easy operation, and high speed. Bistable low- and high-resistance states (LRS and HRS), can be switched reversibly by applying sweeping or pulse voltages with appropriate magnitude and polarity. The set and reset operations represent transitions to an LRS or HRS, which occur at the switching thresholds of (I_{SET}, V_{SET}) and (I_{RESET}, V_{RESET}), respectively. V_{SET} and V_{RESET} are of the same polarity for unipolar resistance switching (URS), whereas opposite polarities are required for bipolar resistive switching (BRS). Previous studies have observed URS and BRS in numerous binary and ternary metal oxides, and the coexistence of two switching modes has been observed in several materials, including TiO₂ [2], NiO [3], ZnO [4], ZrO₂ [5], HfO₂ [6], and Ta₂O₅ [7]. ZrO₂ resistive thin films exhibit long retention and superior endurance in the URS and BRS modes.

Both homogeneous and filamentary types of switching have been proposed in literature. The dependence of current conduction on the electrode area is a key difference between these two switching types. By filament model, the formation of conductive filaments (CFs) in LRS comes from the low-resistance channels between the top and bottom electrodes [8]. In previous studies, spot images have been directly observed by performing probe scanning, which provides strong support for this model [9,10]. An in-between layer was embedded to improve the switching performance by enhancing the confinement of the current path [11,12]. URS and BRS differ in resistance ratio and especially in the abrupt or gradual transition to the HRS [13,14], which implies a specific reset mechanism to be involved in the CF rupture process. The physics and characteristics of CFs provide guidelines

for controlling RRAM switching operations and enhancing conduction reliability. In this study, the conduction and switching dispersions in ZrO₂ devices were examined by applying various current compliances and sweeping voltage ranges. Influence of these measuring conditions was interpreted in perspective of CFs, as compared with those by preparing conditions, e.g., materials, structures, and atmosphere treatment.

2. Results and Discussion

The device under test is of the metal-oxide-metal structure, as described in Section 3 by Ti/ZrO₂/Pt. Figure 1 shows typical I–V sweeps for cells of three top electrode (TE) diameters. All of them exhibited transitions at $V_{\text{SET}} \approx +1.1$ V and $V_{\text{RESET}} \approx -1.2$ V after a forming operation at 6 V. Three crucial observations on the BRS polarity, current compliance, and current distribution are detailed as follows. First, a positive VSET polarity indicated that the reduction reaction to the LRS occurred in the ZrO₂ because of higher potentials at the TE. The Ti electrode functioned as an oxygen ion getter [15], and the positively charged oxygen vacancies remained, thereby forming a low-resistance path. A detailed possible mechanism was suggested by Jeong et al. in terms of electro-chemical reaction involving oxygen ions/vacancies [16]. Once the CF formed at V_{SET} , most of the applied current flowed through this path, and that is why I_{CC} was applied to prevent unlimited current rise. Further increases in voltage (i.e., if V_{POS} was higher than V_{SET}) and current compliance can create stronger paths or increase the CF diameter. By contrast, V_{RESET} denotes the threshold at which resistance values begin changing. A typical gradual transition to the HRS in the BRS mode manifested as a distinct negative resistance region, which is characteristic of a CF rupture induced by oxidation processes. The extent to which the voltage sweep V_{NEG} surpassed V_{RESET} determined the nature of the turn-off extent. The influence of I_{CC} and V_{POS} on LRS, as well as that of V_{NEG} on HRS, is discussed in subsequent paragraph.

Second, Figure 1 shows the current compliance as a horizontal line in the upper-right corner. This compliance level was applied to prevent current overrise, to control current levels after set-switching, and to monitor the switching mode. Previous studies have adopted current compliance at the sub-milliamp scale for small cell areas (e.g., $1 \mu m^2$) [13], and even at the nano-amp scale [17]. Proper adoption of down-size cell and compliance level provides a route to the low-power switching operation of ZrO₂ device. Lee et al. [4] reported a switching mode change from BRS to URS when an I_{CC} of up to 40 mA was applied to a ZnO semiconductor with a TE diameter of 150 μm . The decision to apply I_{CC} values higher than 10 mA in this study was based on the conduction current levels of I_{LRS} and I_{HRS}, as well as those obtained at V_{SET} and V_{POS}. Under relatively high compliance (e.g., up to 50 mA), the studied ZrO₂ devices exhibited reproducible and stable BRS properties. The increases in local temperature, known as the Joule heating effect, play a critical role in the electro-thermal process for URS. For the electro-chemical redox in BRS, this thermal impact is not discussed because of the complex nature of material thermal conductivity and filament distribution/density.

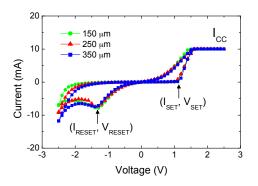


Figure 1. Typical room-temperature I–V curves in linear scale under $I_{CC} = 10$ mA with top electrode diameter of 150, 250, 350 µm.

As for the third point about the current distribution extracted from Figure 1, Figure 2 shows 100 successive sweeping cycles for conduction currents (I_{LRS} , I_{HRS}) and switching currents (I_{SET} , I_{RESET}). LRS and HRS conduction currents (I_{LRS} and I_{HRS}) were measured at -0.5 V, at which the I–V curve exhibited a linear relationship. The figures show that I–V curves are independent from TE size by fixing I_{CC} and (V_{POS}, V_{NEG}) . The size-insensitive property of I_{LRS} was attributed to the highly localized filament that formed in the ZrO₂ matrix [18], which correlated closely with the I_{RESET} threshold for CFs to rupture. Figure 2a,b shows that the current ratio between I_{LRS} and I_{HRS} was approximately two orders of magnitude by 2 mA over 20 $\mu A.$ Because of the considerably lower I_{HRS} typically observed near zero bias, a larger resistance change ratio exceeding 100 was anticipated for readout voltages less than 0.5 V. Under the condition of $V_{SET} \approx |V_{RESET}|$ and $|I_{RESET}| > I_{SET}$, as shown in Figure 2c,d, a larger reset power is required than that at set switching. Rohde [19] observed that the required power was the most dominant parameter for successful switching. The higher power required to initiate a reset transition indicates a more stable conduction in LRS, as compared with that in HRS. Supporting evidence in previous studies includes experimental findings of an improved thermal disturbance immunity facilitated by longer retention in LRS, and the substantially shorter pulse width required to induce a set-switching by an electric pulse [1]. In a compact model for a filament-type RRAM device, the contrasting operational properties were reported, e.g., a gradual resistance change in reset transition while an abrupt change in set switching [20].

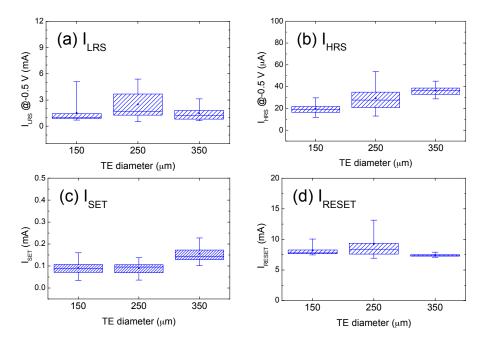


Figure 2. Distribution of critical current parameters extracted from Figure 1 after multiple measurements as a function of electrode areas. (a) I_{LRS} ; (b) I_{HRS} ; (c) I_{SET} ; (d) I_{RESET} .

Figure 3a shows I–V sweeps as a function of I_{CC} , and Figure 3b–d shows the extracted distributions of I_{LRS} and (I_{RESET} , V_{RESET}). As shown in Figure 3a, higher I_{CC} setting will induce extension on both sides of the I–V curve with clear linearity, indicating the development of an ohmic relationship followed by the CF conduction in LRS. Figure 3b shows the tendency for I_{LRS} to increase in conjunction with I_{CC} , which corresponds to the increasing slope of the I–V curve. The corresponding transition is distinctly more abrupt and the switching threshold (I_{RESET} , V_{RESET}) is higher, as evidenced by the negatively biased I–V curves shown in Figure 3a. Therefore, the distribution tends to increase with I_{CC} , as shown in Figure 3c,d. A more detailed explanation based on the filamentary mechanism is given below.

Voltage-induced ion migration and the resulting redox process drive the set transition in the BRS mode [20]. The CF formation is caused by interfacial oxygen ion movement toward the Ti,

and also by the resultant local oxygen vacancies accumulated in the ZrO₂ films [21]. Recent study revealed Ti ion contribution from the TiO₂ solid electrolyte [22], although Ti electrode migration was rarely reported, as compared with Ag and Cu [23,24]. Yoon el al. [10] declared that the non-uniform and uncontrolled formation of CFs over the cell area is caused by their random growth through the thermally assisted electrochemical reaction. Previous studies have ascribed the correlated increasing I_{LRS} with I_{CC} to the increase in filament diameter or effective area [25,26], although Guan et al. [18] stated that strong filaments with a more favorable percolation of elements are formed under higher I_{CC} values. Noh et al. [20] argued that the filament diameter is controlled by the bias setting of polarity, amplitude, and time. They derived CF resistance as a function of filament diameter based on ohmic conduction. Moreover, McWilliams et al. [27] used equations to demonstrate that I_{CC} correlates explicitly with resistance in the LRS and with I_{RESET}. Contrary to the field-induced set process, the reset transition is current driven [5]. Thus, higher I_{CC} leads to rising I_{LRS}, and an increasing I_{RESET} typically coincides with an increase in I_{LRS} [27,28]. Rahaman et al. [17] plotted the linear relationship of I_{CC}–I_{RESET} and I_{CC}–CF diameter. These arguments could be further verified by direct observation of the filaments [29,30].

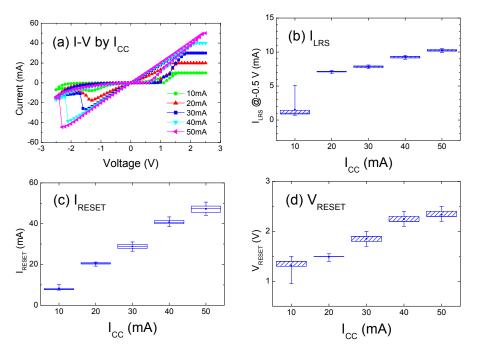


Figure 3. (a) Typical room-temperature I–V curves by top electrode (TE) diameter of 150 μ m under I_{CC} of 10 to 50 mA; (b) Distribution of extracted parameters after multiple measurements for I_{LRS}; (c) for I_{RESET}; and (d) for V_{RESET}.

The increasing I_{RESET} in Figure 3c is approximately equal to I_{CC} , and this rise surmounts that of V_{RESET} in Figure 3d. The resultant resistance decline with I_{CC} complies with the resistance decrease in Figure 3b, i.e., the resistance in LRS is related to the resistance at the reset threshold. Furthermore, Wang [31] observed a more abrupt reset process by increasing I_{CC} , which similarly occurred at higher V_{RESET} . They proposed a scheme of singly connected CFs formed under low I_{CC} , whereas net-like CFs generated at high I_{CC} . The reset process involves an oxidation reaction caused by oxygen ions to reenter into ZrO_2 ; subsequently, a re-oxidized portion forms near the interface. A ruptured filament of shorter length is then connected in series with an HRS region. Park et al. [32] applied a sweeping range to examine the gradual reset transition, and confirmed the existence of incompletely dissolved CFs by performing conductive atomic force microscopy. Long et al. [33] computed reset statistics and confirmed its dependence on I_{LRS} and, therefore, CF size, which was determined by the set or compliance setting in preceding operation. Liu et al. [34] controlled the CF growth process and

overcame the intrinsic multistep reset jiggles. The manipulation of set or reset conditions to obtain multiple distinguishable LRSs or HRSs provides a potential RRAM application for multibit storage. Varying the sweeping bias limits is an effective method, as discussed in the next two paragraphs.

Because V_{SET} is the threshold at which CFs are formed, it is a reasonable speculation that a sweep value higher than V_{SET} would enhance CF growth. The extent to which V_{POS} is higher than V_{SET} determines the filament diameter or strength. Compliance was removed to examine the V_{POS} effect alone. Figure 4a shows the I–V data where V_{POS} is 1.5, 2, and 2.5 V while –2.5 V is fixed. Similar to Figure 3, Figure 4b–d shows the extracted distributions of I_{LRS} and (I_{RESET}, V_{RESET}). The impact of V_{POS} on the conduction in LRS is similar to that exerted by I_{CC}, including an increase in I_{LRS} and in the steepness of the linear I–V curve, a larger power required at reset, and a more abrupt reset transition. Regarding the relatively small increase in V_{RESET} compared with that of I_{RESET} (see Figure 4c,d, the decrease in resistance at the reset threshold agrees with the decreasing resistance in LRS (see Figure 4b). The similarity between the effects of I_{CC} and V_{POS} was ascribed to the increase in CF diameter. Excess bias expels additional oxygen ions to the Ti reservoir and leaves an oxygen-deficient region with an enhanced cross-section. When the CFs ruptured under the same magnitude (i.e., -2.5 V) to have an identical re-oxidized length, the remaining CFs still exhibited lower resistance under the preceding V_{POS} values, which explains why the distribution in Figure 4e shows an increase in I_{HRS} as V_{POS} increases.

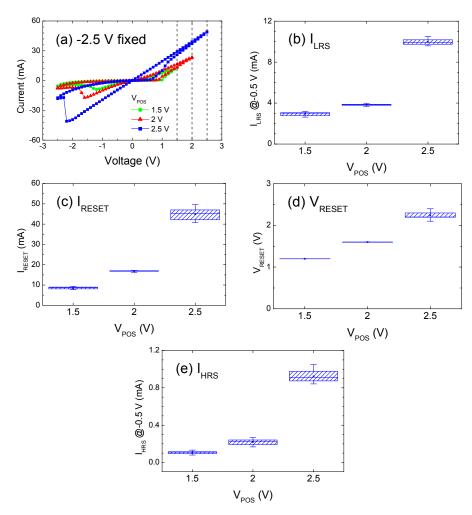


Figure 4. (a) Typical room-temperature I–V curves as a function of V_{POS} with no compliance setting; (b) Effect of V_{POS} on I_{LRS}; (c) on I_{RESET}; (d) on V_{RESET}; and (e) on I_{HRS}.

Figure 5a shows the I–V sweeps under various V_{NEG} settings. The dotted lines denote -1.5, -2, and -2.5 V, with I_{CC} maintained at +2.5 V. Contrary to the CFs growth as a result of the V_{POS} extent, V_{NEG} influenced the extension of the re-oxidized segment after reset operation. The decreasing I_{HRS} shown in Figure 5b is primarily attributed to the highly resistive re-oxidized portion, or could be ascribed to the shortened length of partially disconnected or narrower CFs. Numerous previous studies have reported that controlling the sweeping range in the reset polarity reduces the value of I_{HRS} [21,26,35–37]. Park et al. [35] successfully modulated I_{HRS} and therefore demonstrated the feasibility of multibit applications by adjusting the value of V_{NEG} . An engineered Schottky barrier height between Ir and TiO_X explains this effect. Yu et al. [36] ascribed the decrease in I_{HRS} to a large tunneling gap between the electrode and residual CFs, and proposed that the I_{HRS}– V_{NEG} mechanism is associated with the switching mode and electrode contact. In summary, the variation in the aspect ratio of the filament model presented in this study intuitively explains the difference between the values shown in Figures 4e and 5b; i.e., I_{HRS} increases in conjunction with the CF diameter, whereas I_{HRS} declines in conjunction with CF length.

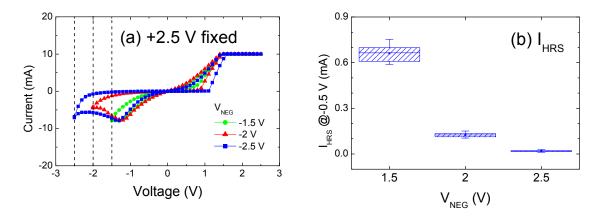


Figure 5. (a) Typical room-temperature I–V curves as a function of V_{NEG} under I_{CC} of 10 mA; (b) Effect of V_{NEG} on I_{HRS}.

3. Materials and Methods

Sputtered ZrO₂ (approximately 60-nm thick) was deposited on a Pt/Ti/SiO₂/Si substrate at 250 °C and 10 mTorr under an O₂: Ar gas mixture with a ratio of 6:12 and a flow rate of 18 cm³ per minute. Because of its oxygen getter properties, Ti was sputter-deposited as the top electrode (TE), and then patterned using a shadow mask with various diameters (150, 250, and 350 μ m) to analyze the size effect. The devices were tested in tri-layer structures comprising Ti (110 nm)/ZrO₂ (60 nm)/Pt. Current-voltage (I–V) curves were obtained using an HP4155A semiconductor parameter analyzer. Device switching between the LRS and HRS was achieved by applying a direct voltage sweep mode while the bottom electrode was grounded. Bias was applied to the TE during each sweep cycle in the range of V_{POS} to $-V_{NEG}$, which denote the positive and negative bias limits, respectively. The current compliance I_{CC} was set to 10–50 mA at the V_{SET} side to control the current level in the LRS. To differentiate between the dependence of specific parameters clearly, this study refers only to their magnitudes and ignores the corresponding signs. Setting V_{POS} (V_{NEG}) higher than V_{SET} (V_{RESET}) is necessary for successful switching. Ambient water vapor pressure is kept constant to exclude the effect of moisture on the switching characteristics [38,39]. Unless stated otherwise, the data for the I–V curves were measured at room temperature, with an I_{CC} of 10 mA, TE diameter of 150 μ m, and a sweeping range of ± 2.5 V.

4. Conclusions

The effect of current compliance and sweeping range on sputtered $Ti/ZrO_2/Pt$ was investigated by examining the distributions of the corresponding I–V curves. The area independence of conduction and switching currents indicates that the formation of CFs was localized over the electrode. The on-state current and corresponding turn-off threshold increased in conjunction with the current compliance. Similar results were observed when higher V_{POS} values were applied. The formation of CFs with increased diameters explains these experimental observations. By contrast, extending V_{NEG} resulted in shorter CFs and a re-oxidized portion, thereby decreasing the I_{HRS} .

Acknowledgments: The authors thank the Ministry of Science and Technology, R.O.C., Taiwan, for financially supporting this research under Contract No. of MOST 103-2221-E-239-031.

Author Contributions: Chun-Hung Lai conceived and designed the experiments, analyzed the data and wrote the paper. Chih-Yi Lin performed the experiments, and Hung-Wei Chen contributed materials/analysis tools.

Conflicts of Interest: The authors declare no conflict of interest.

References

- 1. Liu, C.Y.; Wu, P.H.; Wang, A.; Jang, W.Y.; Young, J.C.; Chiu, K.Y.; Tseng, T.Y. Bistable resistive switching of a sputter-deposited Cr-doped SrZrO₃ memory film. *IEEE Electron Device Lett.* **2005**, *26*, 351–353. [CrossRef]
- 2. Jeong, D.S.; Schroeder, H.; Waser, R. Coexistence of bipolar and unipolar resistive switching behaviors in a Pt/TiO₂/Pt stack. *Electrochem. Solid State Lett.* **2007**, *10*, G51–G53. [CrossRef]
- Goux, L.; Lisoni, J.G.; Jurczak, M.; Wouters, D.J.; Courtade, L.; Muller, Ch. Coexistence of the bipolar and unipolar resistive-switching modes in NiO cells made by thermal oxidation of Ni layers. *J. Appl. Phys.* 2010, 107. [CrossRef]
- 4. Lee, S.; Kim, H.; Park, J.; Yong, K. Coexistence of unipolar and bipolar resistive switching characteristics in ZnO thin films. *J. Appl. Phys.* **2010**, *108*. [CrossRef]
- 5. Lin, C.Y.; Wu, C.Y.; Wu, C.Y.; Tseng, T.Y. Modified resistive switching behavior of ZrO₂ memory films based on the interface layer formed by using Ti top electrode. *J. Appl. Phys.* **2007**, *102*. [CrossRef]
- Govoreanu, B.; Kar, G.S.; Chen, Y.; Paraschiv, V.; Kubicek, S.; Fantini, A.; Radu, I.P.; Goux, L.; Clima, S.; Degraeve, R.; et al. 10 × 10 nm² Hf/HfO_X crossbar resistive RAM with excellent performance, reliability and low-energy operation. *IEDM Tech. Dig.* 2011, 31.6.1–31.6.4.
- Lee, M.-J.; Lee, C. B.; Lee, D.; Lee, S.R.; Chang, M.; Hur, J. H.; Kim, Y.-B.; Kim, C.-J.; Seo, D.H.; Seo, S.; et al. A fast, high-endurance and scalable non-volatile memory device made from asymmetric Ta₂O_{5-X}/TaO_{2-X} bilayer structures. *Nat. Mater.* 2011, *10*, 625–630. [CrossRef] [PubMed]
- 8. Jung, K.; Seo, H.; Kim, Y.; Im, H.; Hong, J.; Park, J.W.; Lee, J.K. Temperature dependence of high- and low-resistance bistable states in polycrystalline NiO films. *Appl. Phys. Lett.* **2007**, *90*. [CrossRef]
- 9. Chen, X.; Wu, N.J.; Strozier, J.; Ignatiev, A. Synthesis and magnetic properties of Fe₃Pt nanowire arrays fabricated by electrodeposition. *Appl. Phys. Lett.* **2005**, *87*. [CrossRef]
- 10. Sun, J.; Liu, Q.; Xie, H.; Wu, X.; Xu, F. In situ observation of nickel as an xxidizable electrode material for the solid-electrolyte-based resistive random access memory. *Appl. Phys. Lett.* **2013**, *102*. [CrossRef]
- 11. Liu, C.Y.; Lin, X.J.; Wang, H.Y.; Lai, C.H. Improved resistive switching dispersion of NiO_X thin film by Cu-doping method. *Jap. J. Appl. Phys.* **2010**, *49*. [CrossRef]
- 12. Yoon, J.H.; Han, J.H.; Jung, J.S. Highly improved uniformity in the resistive switching parameters of TiO₂ thin films by inserting Ru nanodots. *Adv. Mater.* **2013**, *25*, 1987–1992. [CrossRef] [PubMed]
- Wang, W.; Fujita, S.; Wong, S.S. Reset mechanism of TiO_X resistance-change memory device. *IEEE Electron* Device Lett. 2009, 30, 733–735. [CrossRef]
- 14. Celano, U.; Goux, L.; Belmonte, A.; Giammaria, G.; Opsomer, K.; Detavernier, C.; Richard, O.; Bender, H.; Irrera, F.; Jurczak, M.; et al. Progressive vs. abrupt reset behavior in conductive bridging devices: A C-AFM tomography study. *IEDM Tech. Dig.* **2014**, 14.1.1–14.1.4.
- Clima, S.; Sankaran, K.; Chen, Y.Y.; Fantini, A.; Celano, U.; Belmonte, A.; Zhang, L.; Goux, L.; Govoreanu, B.; Degraeve, R.; et al. RRAMs based on anionic and cationic switching: A short overview. *Phys. Status Solid Rapid Res. Lett.* 2014, *8*, 501–511. [CrossRef]

- Jeong, D.S.; Schroeder, H.; Waser, R. Mechanism for bipolar switching in a Pt/TiO₂/Pt resistive switching cell. *Phys. Rev. B* 2009, 79. [CrossRef]
- Rahaman, S.Z.; Maikap, S.; Chen, W.S.; Lee, H.Y.; Chen, F.T.; Tien, T.C.; Tsai, M.J. Impact of TaO_X nanolayer at the GeSe_X/W interface on resistive switching memory performance and investigation of Cu nanofilament. *J. Appl. Phys.* 2012, 111. [CrossRef]
- Guan, W.; Long, S.; Liu, Q.; Liu, M.; Wang, W. Nonpolar nonvolatile resistive switching in Cu doped ZrO₂. *IEEE Electron Device Lett.* 2008, 29, 434–437. [CrossRef]
- 19. Rohde, C.; Choi, B.J.; Jeong, D.S.; Choi, S.; Zhao, J.S.; Hwang, C.S. Identification of a determining parameter for resistive switching of TiO₂ thin films. *Appl. Phys. Lett.* **2005**, *86*. [CrossRef]
- Noh, J.; Jo, M.; Kang, C.Y.; Gilmer, D.; Kirsch, P.; Lee, J.C.; Lee, B.H. Development of a semiempirical compact model for DC/AC cell operation of HfO_X-based ReRAMs. *IEEE Electron Device Lett.* 2013, 34, 1133–1135. [CrossRef]
- 21. Wang, S.Y.; Huang, C.W.; Lee, D.Y.; Tseng, T.Y.; Chang, T.C. Multilevel resistive switching in Ti/Cu_XO/Pt memory devices. *J. Appl. Phys.* **2010**, *108*. [CrossRef]
- Wedig, A.; Luebben, M.; Cho, D.Y.; Moors, M.; Skaja, K.; Rana, V.; Hasegawa, T.; Adepalli, K.K.; Yildiz, B.; Waser, R.; et al. Nanoscale cation motion in TaO_X, HfO_X and TiO_X memristive systems. *Nat. Nanotechnol.* 2016, 11, 67–74. [CrossRef] [PubMed]
- Gao, S.; Chen, C.; Zhai, Z.; Liu, H.Y.; Lin, Y.S.; Li, S.Z.; Lu, S.H.; Wang, G.Y.; Song, C.; Zeng, F.; et al. Resistive switching and conductance quantization in Ag/SiO₂/indium tin oxide resistive memories. *Appl. Phys. Lett.* 2014, 105. [CrossRef]
- Tsuruoka, T.; Valov, I.; Tappertzhofen, S.; Hurk, J.; Hasegawa, T.; Waser, R.; Aono, M. Redox reactions at Cu,Ag/Ta₂O₅ interfaces and the effects of Ta₂O₅ film density on the forming process in atomic switch structures. *Adv. Funct. Mater.* 2015, 25, 6374–6381. [CrossRef]
- Choi, S.J.; Kim, K.H.; Park, G.S.; Bae, H.J.; Yang, W.Y.; Cho, S. Multibit operation of Cu/Cu-GeTe/W resistive memory device controlled by pulse voltage magnitude and width. *IEEE Electron Device Lett.* 2011, 32, 375–377. [CrossRef]
- 26. Puglisi, F.M.; Larcher, L.; Bersuker, G.; Padovani, A.; Pavan, P. An empirical model for RRAM resistance in low- and high-resistance states. *IEEE Electron Device Lett.* **2013**, *34*, 387–389. [CrossRef]
- 27. McWilliams, C.R.; Celinska, J.; Araujo, C.A.; Xue, K.H. Device characterization of correlated electron random access memories. *J. Appl. Phys.* **2011**, *109*. [CrossRef]
- 28. Hu, W.; Chen, X.; Wu, G.; Lin, Y.; Qin, N.; Bao, D. Bipolar and tri-state unipolar resistive switching behaviors in Ag/ZnFe₂O₄/Pt memory devices. *Appl. Phys. Lett.* **2012**, *101*. [CrossRef]
- Kumar, S.; Graves, C.E.; Strachan, J.P.; Grafals, E.M.; Kilcoyne, A.L.D.; Tyliszczak, T.; Weker, J.N.; Nishi, Y.; Williams, R.S. Direct observation of localized radial oxygen migration in functioning tantalum oxide memristors. *Adv. Mater.* 2016, *28*, 2772–2776. [CrossRef] [PubMed]
- Celano, U.; Goux, L.; Degraeve, R.; Fantini, A.; Richard, O.; Bender, H.; Jurczak, M.; Vandervorst, W. Imaging the three-dimensional conductive channel in filamentary-based oxide resistive switching memory. *Nano Lett.* 2015, 15, 7970–7975. [CrossRef] [PubMed]
- 31. Wang, Z.Q.; Li, X.H.; Xu, H.Y.; Wang, W.; Yu, H.; Zhang, X.T.; Liu, Y.X.; Liu, Y.C. Effects of compliance currents on the formation and rupture of conducting filaments in unipolar resistive switching of CoO film. *J. Phys. D Appl. Phys.* **2010**, *43.* [CrossRef]
- 32. Park, J.; Jo, M.; Lee, J.; Jung, S.; Kim, S.; Lee, W.; Shin, J.; Hwang, H. Improved switching uniformity and speed in filament-type RRAM using lightning rod effect. *IEEE Electron Device Lett.* **2011**, *32*, 63–65. [CrossRef]
- 33. Long, S.; Lian, X.; Ye, T.; Cagli, C.; Perniola, L.; Miranda, E.; Liu, M.; Suñé, J. Cycle-to-cycle intrinsic reset satistics in HfO₂-based unipolar RRAM devices. *IEEE Electron Device Lett.* **2013**, *34*, 623–625. [CrossRef]
- 34. Liu, Q.; Long, S.; Lv, H.; Wang, W.; Niu, J.; Huo, Z.; Chen, J.; Liu, M. Controllable growth of nanoscale conductive filaments in solid-electrolyte-based ReRAM by using a metal nanocrystal covered bottom electrode. *ACS Nano* **2010**, *4*, 6162–6168. [CrossRef] [PubMed]
- Park, J.; Biju, K.P.; Jung, S.; Lee, W.; Lee, J.; Kim, S.; Park, S.; Shin, J.; Hwang, H. Multibit operation of TiO_X-based ReRAM by schottky barrier height engineering. *IEEE Electron Device Lett.* 2011, 32, 476–478. [CrossRef]
- 36. Yu, S.; Jeyasingh, R.; Wu, Y.; Wong, H.-S.P. AC conductance measurement and analysis of the conduction processes in HfO_X based resistive switching memory. *Appl. Phys. Lett.* **2011**, *99*. [CrossRef]

- Chen, M.C.; Chang, T.C.; Chiu, Y.C.; Chen, S.C.; Huang, S.Y.; Chang, K.C.; Tsai, T.M.; Yang, K.H.; Sze, S.M.; Tsai, M.J. The resistive switching characteristics in TaON films for nonvolatile memory applications. *Thin Solid Films* 2013, 528, 224–228. [CrossRef]
- 38. Messerschmitt, F.; Kubicek, M.; Rupp, J.L.M. How does moisture affect the physical property of memristance for anionic–electronic resistive switching memories? *Adv. Funct. Mater.* **2015**, *25*, 5117–5125. [CrossRef]
- Lübben, M.; Karakolis, P.; Ioannou-Sougleridis, V.; Normand, P.; Dimitrakis, P.; Valov, I. Graphene-modified interface controls transition from VCM to ECM switching modes in Ta/TaO_X based memristive devices. *Adv. Mat.* 2015, 27, 6202–6207. [CrossRef] [PubMed]



© 2016 by the authors; licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC-BY) license (http://creativecommons.org/licenses/by/4.0/).