

## Supporting Information

# Precise Construction and Growth of Submillimeter Two-Dimensional WSe<sub>2</sub> and MoSe<sub>2</sub> Monolayers

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**Table S1.** Current status of CVD method for growing two-dimensional materials in recent years

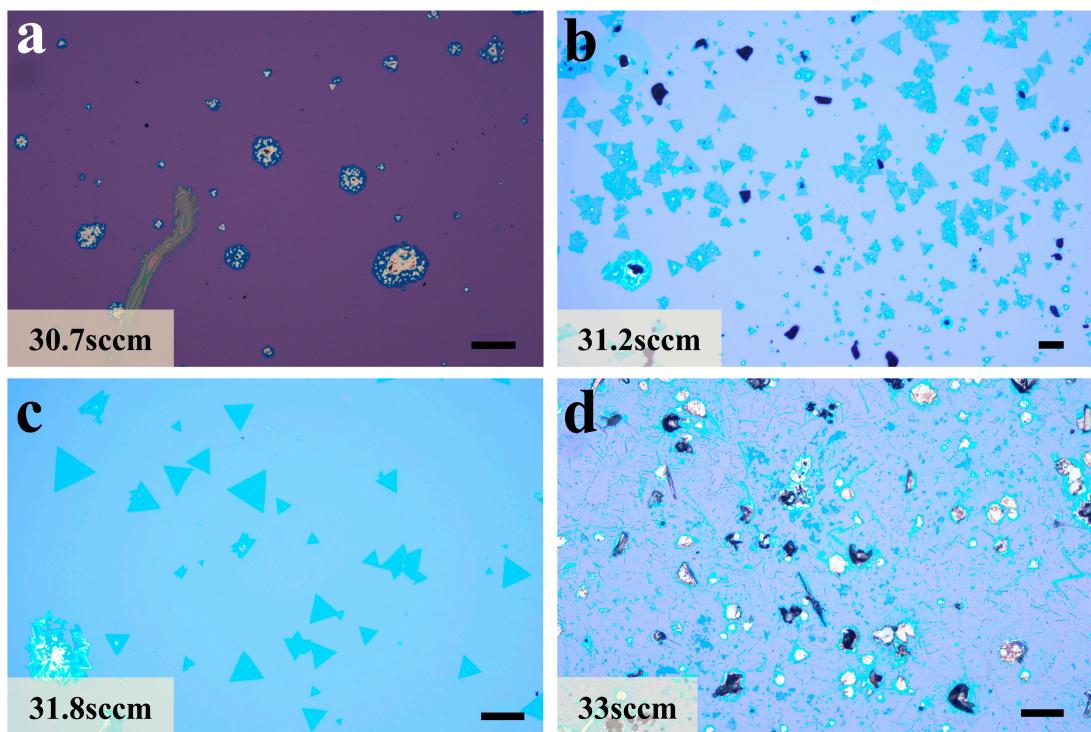
	Materials	Methods	Size/Thickness
Wang et al.[1]	TiSe <sub>2</sub>	the metal and non-metal precursors	5 × 10 <sup>5</sup> μm <sup>2</sup> film /4.5nm
Meng et al.[2]	CrTe <sub>2</sub>	Conventional growth methods	~80μm/1.2-47.9nm
Li et al.[3]	ReS <sub>2</sub>		10μm/No data
Chubarov et al.[4]	WS <sub>2</sub>	changing the growth substrate	Wafer-scale film/0.92nm
Kang et al.[5]	MoS <sub>2</sub> , WS <sub>2</sub>	the metal and non-metal precursors	Wafer-scale film/monolayer
Yu et al.[6]	WSe <sub>2(1-x)Te<sub>2x</sub></sub>		No data/monolayer
Liu et al.[7]	W <sub>x</sub> Mo <sub>1-x</sub> S <sub>2</sub>	tuning the elemental compositions of the precursors	20μm/0.94nm
Wu et al.[8]	Bi <sub>2</sub> O <sub>2</sub> Se		~200μm /6.7nm
Hong et al.[9]	MoSi <sub>2</sub> N <sub>4</sub>		~1cm/1.17nm
Wang et al.[10]	Graphene	space-confined growth	~3mm/~1.26nm
Xu et al.[11]			~0.6mm/No data
Zhou et al.[12]	47 different atomically thin TMCs		2μm-1mm/ monolayer
Cui et al.[13]	Cr <sub>2</sub> S <sub>3</sub>	assistance of molten salts	~60μm /1.84nm
Zhang et al.[14]	CrSe		0.15mm/2.50nm
Yang et al.[15]	MoS <sub>2</sub>		400μm/0.77nm
Zeng et al.[16]	PdSe <sub>2</sub>		Wafer-scale film /1.2-20 nm
Zhou et al.[17]	MoTe <sub>2</sub>	thermally assisted conversion	Wafer-scale film /3.1nm
Yu et al.[18]	MoS <sub>2</sub>		1mm/8nm

We show the parameter tuning process with the triangular WSe<sub>2</sub> as a representative. There is no such probing process in the already experienced MoSe<sub>2</sub> growth process. The optical microscope images shown are not the complete experimental process. They are only some representative experimental data selected to show the process of continuous sample refinement during parameter tuning.

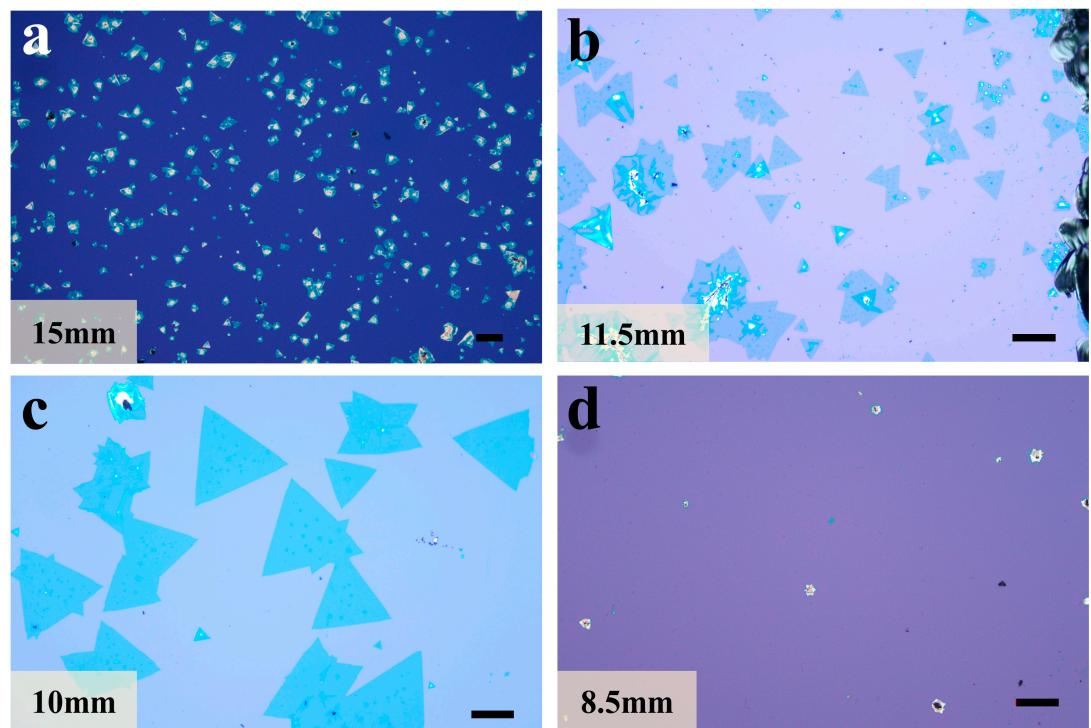
As shown in Figure S1a, when the flow rate of the H<sub>2</sub>/Ar was 30.7 sccm, the grown sample was WSe<sub>2</sub> in many bulk agglomerated states. The carrier gas flow rate is not sufficient to transport the sublimated selenium powder to the central heating zone of the furnace cavity to react with the precursor, and simultaneously, the intermediate state products are not effectively formed, and the nucleation points of WSe<sub>2</sub> are difficult to obtain, making the growth difficult. The carrier gas flow rate was increased to 31.2 sccm, as shown in Figure S1b. At this time, the number of nucleation points was sufficient and some small triangles started to be generated on the substrate. When the carrier gas flow rate is further increased to 31.8 sccm, as in Figure S1c, there is a significant increase in the size of the product on the substrate. When the reaction gas flow rate reaches 33 sccm, the triangles are no longer present on the substrate. Hydrogen promotes growth, but also reacts with WSe<sub>2</sub>, leading to corrosion (Figure S1d). The corrosion increases with increasing load.

At a distance of 15 mm, the transition phase WO<sub>3-x</sub> is far from the substrate, and the concentration reaching the substrate is not sufficient for effective growth of the film. Therefore, it tends to grow as bulk particles, as shown in Figure S2a. When the distance is adjusted to 11.5 mm (Figure S2b), triangular monolayer WSe<sub>2</sub> films can be clearly observed, and the size can also reach about 50 μm. This indicates that the concentration of the transition phase WO<sub>3-x</sub> increases and helps to form monolayer films, allowing the films to grow effectively. By decreasing the distance again to 10 mm, as shown in Figure S2c, it can be seen that the film size continues to increase, indicating that the distance at this point is more suitable for the growth of large-size films. After decreasing the distance again to 8.5 mm, as shown in Figure S2d, only a very small amount of unshaped particles are present on the substrate, indicating that the close distance between the intermediate phase products and the substrate may have caused a high

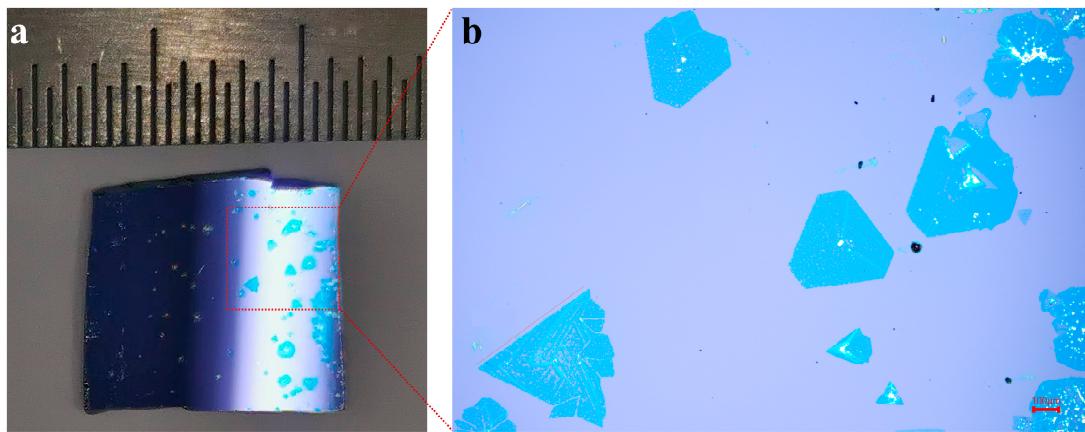
concentration of deposits, preventing nucleation points formation.



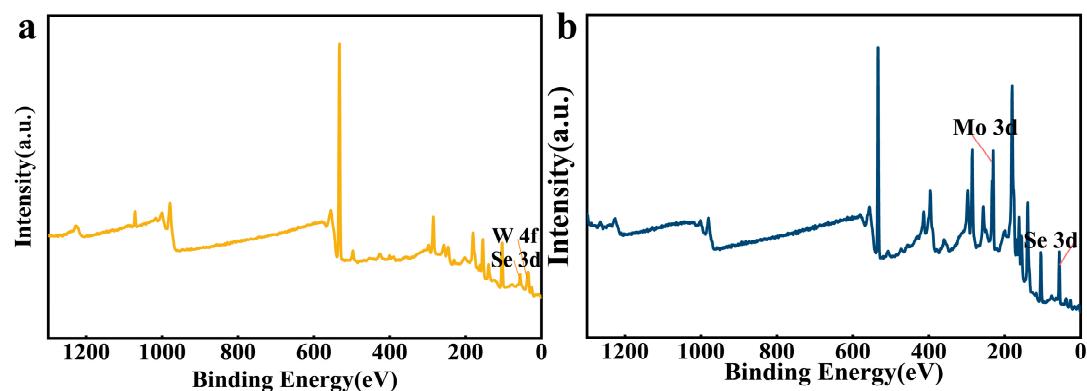
**Figure S1.** Growth of WSe<sub>2</sub> crystals at different carrier gas flow rates. (a,b) scale bar 50μm; (c,d) scale bar 100μm



**Figure S2.** Growth of WSe<sub>2</sub> at different distances of precursors from the substrate. Scale bar 50μm



**Figure S3.** Large-size WSe<sub>2</sub> grown on silicon substrate taken with a camera and WSe<sub>2</sub> images at 5x optical microscope magnification. In this figure, WSe<sub>2</sub> appears as a triangular or truncated triangular multilayer shape. The area circled in red in (a) corresponds to the area in (b). Comparison with a standard scale shows that the maximum size reaches about 1 mm.



**Figure S4.** Full spectrum of XPS. (a) is WSe<sub>2</sub>, and (b) is MoSe<sub>2</sub>. The figure does not show signals of other metals and halides, which indicates the high chemical purity of the XSe<sub>2</sub> (X=W, Mo) crystals.

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