



# Article Characterization and Performance Evaluation of Metakaolin-Based Geopolymer Foams Obtained by Adding Palm Olein as the Foam Stabilizer

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**Abstract:** Geopolymer foams with different pore structures can be used in construction, water treatment, and heavy metal adsorption. The preparation of high porosity geopolymer foams using vegetable oil as a foam stabilizer is a feasible and cost-effective route. In this study, metakaolinbased geopolymer foams with hierarchical pore structures were fabricated by adding  $H_2O_2$  as the foaming agent with palm olein as the foam stabilizer. The effects of  $H_2O_2$  and palm olein content on the chemical features and pore structure of geopolymer foams were evaluated. Water absorption, thermal conductivity, and mechanical behaviors of geopolymer foams were also investigated. The results indicate that fatty acid salt surfactants were generated in situ in the geopolymer matrix due to the addition of palm olein. Geopolymer foams with  $H_2O_2$  and palm olein addition possess a homogeneously concentrated macropore distribution. Palm olein exhibits a refining effect on intrinsic pores formed by geopolymerization. In addition, using appropriate amounts of palm olein and  $H_2O_2$ , geopolymer foams can achieve higher open porosity and better pore connectivity, resulting in the improvement of water absorption and thermal insulation capacity.

Keywords: geopolymer foams; foam stabilizer; pore structure; water absorption; thermal conductivity

# 1. Introduction

Geopolymers are inorganic polymers with zeolite-like structures synthesized by the alkali activation of alumina silicate precursors, such as metakaolin (MK), fly ash, red mud, and several industrial byproducts. It is a material that possesses excellent mechanical properties [1], chemical and high-temperature resistance [2] and is available for various fields. The low carbon emissions in raw material production and preparation process means geopolymers have the potential to replace traditional high-energy-consuming materials. With a proper structural design, such as lightweighting, geopolymers can become versatile, eco-friendly materials with a low carbon footprint throughout their life cycle.

Geopolymer foams, also called lightweight or porous geopolymers, are new porous materials fabricated by introducing pores into the geopolymer matrix. Due to their characteristics as both geopolymer and porous materials, geopolymer foams have been widely investigated by researchers recently. The properties of geopolymer foams are significantly influenced by their pore characteristics, such as porosity, pore types (closed or open), connectivity, and pore size distribution [3–6]. Generally, geopolymer foams with higher porosity possess lower compressive strength, thermal conductivity, and bulk density [5]. However, the effect of different pore types on the performance of geopolymer foams is inconsistent. Compared with open pores, fine and uniform closed pores are more beneficial for improving mechanical and thermal insulation properties [3]. With elevated open



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**Copyright:** © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). porosity, the connectivity of isolated pores in the structure increases, forming irregular large open and connected pores [4]. Geopolymer foams having connected pore networks own high specific surface area and volume capacity, which make them hold significant advantages in terms of water and sound absorption, although the mechanical properties are slightly inferior [5]. In addition to the above factors, the pore size distribution significantly affects the performance of geopolymer foams [6]. Samples with wider average pore size show poorer compressive strength but may exhibit superior thermal insulation capacity if they have higher porosity. As previously described, geopolymer foams with lower pore connectivity generally have superior strength and low thermal conductivity, making them suitable for construction [7,8], thermal barrier [9–11], and sound insulation [12,13]. With improved pore connectivity, the porosity and specific surface area of geopolymer foams increase accordingly. It means that geopolymer foams can be more widely used in wastewater treatment (adsorption of heavy metals [14–16] and organic pollutants dissolved in water [17]) and the preparation of functionalized composite materials [18,19]. Therefore, the regulation and optimization of the pore structure in geopolymer foams have become one of the top research hotspots in this field.

Geopolymer foams with various pore structures can be prepared by changing the preparation method and adjusting raw materials and additives [20,21]. The main fabrication methods of geopolymer foams can be divided into four categories: (i) Direct foaming, (ii) Sacrificial templates, (iii) Additive manufacturing, and (iv) Other methods. The direct foaming method is one of the most common methods used in the preparation of geopolymer foams. Several foaming agents, Al powder [22–24], Si powder [25], and H<sub>2</sub>O<sub>2</sub> [26,27] are most commonly selected to introduce gases to the geopolymer matrix. Among them, H<sub>2</sub>O<sub>2</sub> is the most desirable for its economical, efficient, and impurity-free nature. It decomposes and releases oxygen in the matrix to develop pore structures. In order to prevent the escape of gases from the matrix and excessive merging of air bubbles, foaming agents are usually used in conjunction with foam stabilizers.

Foam stabilizers can elevate the stability of foams by strengthening the molecular structure of the pore walls and reducing the surface tension of the mixture [3]. Some surfactants or stabilizing agents, such as sodium dodecyl sulfate (SDS) [28,29], sodium lauryl ether sulfate (SLES) [5], tween80 [30], and vegetable oils [17,27,31], are often used as foaming stabilizers to prepare geopolymer foams in combination with foaming agents [5,17,27–31]. Compared to those prepared with other foam stabilizers, geopolymer foams that are fabricated using vegetable oils have higher porosity [32,33]. With a low cost and pollution-free production process, vegetable oil is a desirable option for foam stabilizers to achieve ecofriendly geopolymer foams. However, the types and proportions of fatty acids in vegetable oils significantly affect the pore structure of geopolymer foams. Vegetable oils high in palmitic and oleic acids are preferable for preparing geopolymer foams with high porosity and concentrated pore size distribution, such as palm olein and olive oil [27]. Palm olein is the liquid fraction of palm oil after fractionation and crystals removal [34]. It mainly contains palmitic acid (38–43%), oleic (39–45%), and linoleic acids (10–13%). The composition of palm olein makes it a more suitable foam stabilizer for preparing geopolymer foams with high open porosity than conventional vegetable oils [27].

The pore characteristics of geopolymer foams, such as pore size distribution, porosity, and connectivity, are obviously influenced by the amount of foaming agent and stabilizer used. It also further affects the macroscopic properties of geopolymer foams. The image analysis method has been employed to characterize the pore structure of geopolymer foams in published studies. However, limited by the capacity of this technique, it is difficult to obtain information about small pores, such as those smaller than 1  $\mu$ m. Meanwhile, quantitative characterization of the connectivity of macropores is vital for establishing the relationship between microstructure and macroscopic properties, which are rarely reported in the existing literature. In this study, palm olein was introduced as a new foam stabilizer for geopolymer foams preparation coupled with H<sub>2</sub>O<sub>2</sub> as the foaming agent. The chemical features of geopolymer foams with different H<sub>2</sub>O<sub>2</sub> and palm olein

content were investigated. In order to gain a comprehensive understanding of the effects of  $H_2O_2$  and palm olein on different types of pores in geopolymer foams, an integrated characterization was performed by combining mercury intrusion porosimetry (MIP), image analysis, and geometric calculations. In parallel, we will attempt a quantitative description of the pore connectivity of geopolymer foams using a simplified geometric model, which was first proposed. Furthermore, the influences of  $H_2O_2$  and palm olein on the water absorption, thermal conductivity, and mechanical behaviors of geopolymer foams were also examined. We hope that the work performed in the paper will provide a feasible way for the preparation and application of low carbon footprint geopolymer foams.

# 2. Materials and Methods

# 2.1. Materials

# 2.1.1. Precursor

Metakaolin (Calcined kaolin, SP33, BASF Corporation, Florham Park, NJ, USA), with a bulk density of 0.445 g/cm<sup>3</sup>, was selected as the precursor material to prepare geopolymer foams. The chemical composition was analyzed by an X-ray fluorescence spectrometer (XRF, Panalytical Axios, Almelo, The Netherlands), as listed in Table 1. Figure 1 shows the morphology of metakaolin under SEM. As can be seen from the image, most of the particles of metakaolin are smaller than 10  $\mu$ m. The particle size distribution of metakaolin was analyzed by a laser particle size analyzer (BT-2001, BETTER, Dandong, China), as shown in Figure 2.

Table 1. Chemical composition of metakaolin (wt.%).

Oxide	SiO <sub>2</sub>	$Al_2O_3$	TiO <sub>2</sub>	Fe <sub>2</sub> O <sub>3</sub>	Na <sub>2</sub> O	Cr <sub>2</sub> O <sub>3</sub>	L.O.I. *
МК	53.15	42.28	3.58	0.56	0.39	0.03	0.08



\* L.O.I.: Loss of ignition at 950 °C.

Figure 1. Microscopic morphology of metakaolin.

# 2.1.2. Alkali Activator

The alkali activator was prepared by mixing aqueous sodium hydroxide with liquid sodium silicate. Sodium hydroxide (Analytical reagent, NaOH  $\geq$  96.0%, Dalu, Tianjin, China) is mainly used to reduce the modulus of the activator and make its viscosity suitable for mixing. The aqueous sodium silicate (molar ratio of SiO<sub>2</sub>/Na<sub>2</sub>O = 2.2, H<sub>2</sub>O content = 55.61%) used in this experiment was purchased from Hebei Julide Co., Ltd. (Langfang, China).



Figure 2. Particle size distribution of metakaolin.

## 2.1.3. Foaming Agent

 $H_2O_2$  solution (Analytical reagent, purity 30 wt.%, Keshi, China) was used as the foaming agent.  $H_2O_2$  will slowly and spontaneously decompose into water and oxygen at room temperature, according to Equation (1). The decomposition process will be more violent at elevated temperatures and in alkaline solutions.

$$H_2O_2_{(\text{liquid})} \to H_2O_{(\text{liquid})} + O_2_{(\text{gas})}$$
(1)

# 2.1.4. Foam Stabilizer

Palm olein (Food grade, Julong, Tianjin, China) was chosen as the foam stabilizer to prepare geopolymer foams with  $H_2O_2$ . Distinct from palm stearin, which has a high melting point [27], the melting point of palm olein is below 20 °C [34], which makes it feasible to prepare geopolymer foam with palm olein by a one-step process at room temperature without additional heat treatment. The fatty acids in palm olein are saponified with an alkali solution in the matrix. Fatty acid salts, as the surfactants, are generated in situ by the saponification.

# 2.2. Fabrication of Geopolymer Foams

Prior to the preparation of geopolymer slurry, the alkali activator, with molar ratio  $SiO_2/Na_2O/H_2O = 1.2/1.0/15.0$ , was obtained by thoroughly mixing liquid sodium silicate and a 6.8 M NaOH solution (prepared by dissolving NaOH particles in tap water). The alkali activator should be sealed and placed at room temperature for more than 24 h before use.

Figure 3 presents a schematic diagram of the preparation process of geopolymer foams. The geopolymer slurry was prepared by mechanical mixing of metakaolin and the alkali activator at 1000 rpm for 3 min. H<sub>2</sub>O<sub>2</sub> and palm olein were then added to the slurry and stirred for 2 min at 1000 rpm according to the mix proportion listed in Table 2 Adding the slurry to one-third of the cylindrical plastic mold ( $\phi$ 35.0 × 85.0 mm) leaves enough space for volume expansion.

After filling the slurry, the molds were sealed with special caps and transferred to a 60 °C oven for 24 h. The samples were taken out from the dry oven and demolded after the temperature of the samples equilibrated to room temperature one day later. Cylindrical samples were cut into discs suitable for characterization using a precision cutter (SYJ-200, KeJing, Shenyang, China) with 10  $\mu$ m precision. The prepared specimens were dried to constant weight in a blast drying oven at 60 °C before testing.



Figure 3. Fabrication process of geopolymer foams.

Table 2.	Mix pro	oportions	of the sam	ples (g).
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Туре	Sample	Metakaolin	Sodium Silicate	NaOH	Water	$H_2O_2$	Palm Olein
Matrix	H0P0	100.0	100.73	15.33	54.61	0	0
Geopolymer - foams	H2P0 H3P0	100.0 100.0	100.73 100.73	15.33 15.33	54.61 54.61	2 3	0 0
	H2P6 H3P6 H3P10	100.0 100.0 100.0	100.73 100.73 100.73	15.33 15.33 15.33	54.61 54.61 54.61	2 3 3	6 6 10

The synthesized geopolymer foams samples were named according to  $H_2O_2$  and palm olein amounts. The letters H and P represent  $H_2O_2$  and palm olein, respectively. The number following each letter represents the percentage by mass of that additive to the metakaolin. Matrix samples (H0P0) without blowing agent and foam stabilizer were also prepared in the same proportions as a reference.

# 2.3. Characterization

## 2.3.1. Chemical Features

Fourier-transform infrared (FTIR) (Nicolet iS10, Thermo Fisher, Waltham, MA, USA) was employed to determine the functional groups and chemical bonds in geopolymer foams. The wavenumber range was 4000–400 cm<sup>-1</sup> with a resolution of 2 cm<sup>-1</sup>. The phases compositions of geopolymer foams were evaluated using X-ray diffraction (XRD) (PANalytical Xpert3, MalvernPanalytical, Malvern, UK) with CuK $\alpha$  X-ray (wavelength equal to 1.54 angstrom) generated at 40 kV and 40 mA in the 2 theta range from 10 to 80. Step size and scan step time were set to 0.013° and 30 s, respectively.

# 2.3.2. Morphology

Three imaging methods were used to characterize the morphologies of geopolymer foams. A high-resolution camera (IMX498, Sony, Tokyo, Japan) was employed to capture the full-size images of geopolymer foams. The millimeter-scale and micron-scale topography were characterized by Optical-digital microscope (DSX500, Olympus, Tokyo, Japan) and SEM (VEGA3 XMU, TESCAN, Brno, Czech Republic). The samples were fixed on a sample stage and sputtered with a 10 nm gold coating to improve conductivity prior to SEM observation. Secondary electron images of geopolymer foams were obtained by magnifying 500 times with 5 kV accelerating voltage.

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## 2.3.3. Pore Size Distribution

A MIP apparatus (AutoPore IV 9500, Micromeritics, Norcross, GA, USA) was adopted to evaluate the pore structure of geopolymer foams and the matrix sample. The pressure analysis was performed for pressure from 0.52 to 33,000 Psia, which means the measurable pore size distribution ranges from 0.005 to 350  $\mu$ m. The mercury parameters used in this experiment, including contact angle and surface tension, were 130° and 485 dynes/cm, respectively. The samples were cut into approximately 4 mm × 4 mm × 4 mm pieces and vacuum dried for 4 h before testing. In this study, a powder penetrometer with a stem volume of 1.31 mL is preferred for geopolymer foams characterization.

## 2.3.4. Porosity

Geopolymer foams contain pores ranging from nanometers to millimeters [35,36]. The total open porosity ( $O_{total}$ ) of each sample was calculated by the following Equation (2):

$$O_{total} = \left(1 - \frac{\rho_{bulk}}{\rho_{skeletal}}\right) \times 100\%$$
<sup>(2)</sup>

where  $\rho_{bulk}$  is the bulk density calculated by the geometric method (ratio of weight to volume). A caliper with an accuracy of  $\pm 0.02$  mm was used to measure the dimensional data of samples.  $\rho_{skeletal}$  is the skeletal density deduced from data collected by MIP and directly reported in the result files.

## 2.3.5. Pore Shape

*Roundness* is used to evaluate the irregularity of pores [4,37]. It can be calculated by Equation (3):

$$Roundness = \frac{P^2}{4\pi A} \tag{3}$$

where *A* and *P* are the area and diameter of the pore, respectively. These two parameters can be measured by ImagePro Plus software (v6.0, Media Cybernetics, Maryland, USA) from the binarized cross-sectional image of each sample. Equation (3) is the equation adopted by the ImagePro Plus software to calculate the *roundness*. An alternative method of calculating *roundness* used by some researchers [38] is given in Equation (4), whose values are reciprocal to that of Equation (3).

$$Roundness^1 = \frac{4\pi A}{P^2} \tag{4}$$

In this study, the two-dimensional pore data were acquired by ImagePro Plus software. For the convenience of data processing, Equation (3) was chosen for *roundness* calculation.

# 2.3.6. Water absorption and thermal conductivity

The water absorption of geopolymer foams was conducted according to ASTM C642. Three specimens with a diameter of 35 mm and a thickness of 10 mm were tested for each sample. An average value of these test results was taken. The water absorption can be calculated by Equation (5).

Water absprption (%) = 
$$\frac{m_{saturated} - m_{dry}}{m_{dry}} \times 100\%$$
 (5)

where  $m_{saturated}$  is the mass of water-saturated specimen (g),  $m_{dry}$  is the mass of dry specimen (g).

The thermal conductivity of geopolymer foams at room temperature was determined using a hot wire thermal conductivity analyzer (TC3000E, XIATECH, Xi'an, China). Six specimens of the same size as the water absorption test were used to measure the thermal conductivity of geopolymer foams. The hot wire probe should be in good contact with the samples during the measurement. Thermal conductivity testing three times for each set of specimens to ensure reproducibility and accuracy. The compressive strength of matrix and geopolymer foams was determined three days after sample fabrication using a universal mechanical testing machine (YAW-300D, HRJ, Jinan, China) with a loading speed of 2400 N/s. The final value of compressive strength is the average result of 3 specimens with dimensions of  $\phi$ 35 × 35 mm<sup>3</sup>.

## 3. Results and Discussions

# 3.1. Chemical Features

The infrared (IR) spectra of matrix and geopolymer foams with different  $H_2O_2$  and palm olein content are presented in Figure 4.



Figure 4. IR spectra of geopolymer foams with different H<sub>2</sub>O<sub>2</sub> and palm olein content.

For all samples, the absorption band around  $1012 \text{ cm}^{-1}$  corresponds to the symmetrical vibration of Si (Al)-O-Si bonds is observed, while the absorption peak corresponding to the in-plain bending vibration of Si-O bond appears at 441 cm<sup>-1</sup> [39]. The absorption bands located at 700 and 586 cm<sup>-1</sup> are related to the bending vibration of Si-O-Si and Si-O-Al bonds. This illustrates the formation of aluminosilicate structures in the matrix and geopolymer foams. Typical absorption bands corresponding to the stretching and bending vibration of H-O-H and O-H groups from free water and bound water are observed around 3445 and 1651 cm<sup>-1</sup> [40], respectively. Three new absorption peaks around 2923, 2852, and 1564 cm<sup>-1</sup> appear in the IR spectra of geopolymer foams with palm olein addition, corresponding to the stretching vibration of -CH<sub>2</sub>- and -COO<sup>-</sup> [41]. This reveals the existence of long-chain fatty acid salts in geopolymer foams.

As shown in Figure 4, the IR spectra of geopolymer foams prepared with hydrogen peroxide only (H2P0 or H3P0) is similar to that of the matrix (H0P0), which indicates that the use of hydrogen peroxide has no significant effect on the chemical structure of geopolymer foams. However, fatty acid salts were found in geopolymer foams with palm olein addition, which were produced by the saponification between palm olein and alkali solution [42]. These fatty acid salts are highly effective surfactants and can have a significant impact on the formation process of geopolymer foams. Meanwhile, it can be found that the intensity of the characteristic absorption peaks of fatty acid salts grows with the increase in palm olein doping. This indicates that the added palm olein can adequately react with the alkali solution in the matrix.

Figure 5 shows the XRD patterns of matrix and geopolymer foams with different  $H_2O_2$  and palm olein content. The diffraction peaks of all of the geopolymer foams are consistent with raw precursors. Some diffraction peaks of anatase (TiO<sub>2</sub>) and kaolinite are detected. The diffraction band located at 20–25° is attributed to the amorphous phase of MK [43,44].

After geopolymerization, a new diffraction hump in the range of  $25-30^{\circ}$  was detected in the matrix and geopolymer foams. In comparison with MK, a noticeable shift of the diffraction hump center was observed, which is consistent with the literature results [43–46]. The hump centered at ~28° can be attributed to the amorphous aluminosilicate gel, the typical binder phase in metakaolin-based geopolymer [43,45]. The existence of this amorphous hump coupled with the shift of the hump center demonstrated the occurrence of geopolymerization during geopolymer preparation. Typical characteristic peaks of amorphous structures were observed in all of the samples. This indicates that the use of H<sub>2</sub>O<sub>2</sub> and palm olein does not significantly affect the alkali activation process of the precursor [40].



Figure 5. XRD patterns of matrix and geopolymer foams with different H<sub>2</sub>O<sub>2</sub> and palm olein content.

# 3.2. Morphology

Figures 6 and 7 present the optical and SEM images of the geopolymer matrix and geopolymer foams at different scales, of which (1) is the overall picture of each sample captured by the camera, (2) and (3) are the microtopographies obtained by optical-digital microscope and SEM, respectively.



Figure 6. Morphology of matrix and geopolymer foams: (A1-A3) H0P0, (B1-B3) H2P0, (C1-C3) H3P0.



Figure 7. Morphology of geopolymer foams: (A1-A3) H2P6, (B1-B3) H3P6, (C1-C3) H3P10).

With the addition of  $H_2O_2$ , a large number of pores were introduced into the matrix, resulting in a porous structure (Figure 6(B1,C1)). From the microscopic images, it can be noticed that these cells produced by the foaming agent are mainly closed cells. Comparing Figure 6(B2,C2), the average pore size of these samples increased significantly with the increase in  $H_2O_2$  dosage. Meanwhile, many pores with a pore size of more than 350 µm were formed in these samples. The structural information of these pores could not be obtained by MIP.

Figure 7 illustrates the effect of palm oil on the morphology of the geopolymer foam prepared by the addition of  $H_2O_2$ . As can be seen from the optical images, the addition of palm oil resulted in a highly homogeneous pore structure in the geopolymer foams. The use of palm olein significantly influenced the pore morphology and pore size distribution of geopolymer foams. Microscopic images show that there is a considerable number of open pores with diameters of tens to hundreds of microns in the pore walls of the geopolymer foams is mainly caused by the saponification between palm oil and alkali activator. The surfactant, sodium fatty acid, generated in situ during the reaction was adsorbed on the gas-liquid interface created due to gas release. The foam walls were enhanced, and the surface tension of the mixture was reduced by the surfactant [3,32], which prevent drainage and coalescence of the foams from occurring.

By comparing Figures 6 and 7, some cracks can be found in the matrix and in the geopolymer foams only containing  $H_2O_2$ , but not in those with palm olein addition. These cracks in the matrix and geopolymer foams shown in A3–C3 of Figure 6 are attributed to the high capillary pressures between dry and wet parts of the micropore network [47]. Due to the drop in ambient humidity, the geopolymer loses water rapidly, and thus cracks are initiated in the matrix [48]. With the increase in open porosity, the temperature and humidity in different parts of geopolymer foams can be equilibrated quickly, which effectively inhibits cracking. This is why no cracks are found on the microscopic images in Figure 7.

#### 3.3. Pore size Distribution

MIP was employed to characterize pores in the  $0.005-350 \mu m$  range in geopolymer foams. The pore size distribution of these pores cannot be precisely described by image



measurements. Figure 8 presents the pore size distribution in geopolymer foams with different  $H_2O_2$  and palm olein contents.

**Figure 8.** Pore size distribution of matrix and geopolymer foams: (**A**,**B**) H0P0, H2P0, and H3P0; (**C**,**D**) H2P6, H3P6, and H3P10.

The pore size distribution data indicated that the geopolymer foams consist of two kinds of pores. The intrinsic pores derived from the geopolymerization ranged from 0.005 to 1  $\mu$ m [49]. These macropores ranging from 1 to 350  $\mu$ m are due to the decomposition of H<sub>2</sub>O<sub>2</sub>, since they do not present in the matrix.

The macropore distribution of geopolymer foams made with  $H_2O_2$  only (H2P0 and H3P0) was scattered and irregular (Figure 8A). It is evident from the cumulative distribution curve that the volume of macropores increases with the growth of  $H_2O_2$  dosage. For H2P0 and H3P0 samples, the volume of macropores increases from 0.333 mL/g to 0.492 mL/g (Figure 8B and Table 3, calculated from MIP data). The foaming agent also affects the structure of the intrinsic pores. For geopolymer foams with 3 wt.%  $H_2O_2$  (H3P0), the intrinsic pore size was distributed between 0.01 and 0.05  $\mu$ m, while for matrix, the size of the intrinsic pores was mainly between 0.01 and 0.03  $\mu$ m. The expansion of the distribution range may result from the decomposition of  $H_2O_2$  in the intrinsic pores. The critical pore entry diameter of H3P0 samples is 0.032  $\mu$ m which is larger than 0.026  $\mu$ m in the matrix, proving that  $H_2O_2$  has the effect of increasing the size of the intrinsic pores in geopolymer foams.

For all of the geopolymer foams with palm olein addition, the pore size distribution was more regular than those without using the foam stabilizer. Both intrinsic pores and macropores show concentrated distribution. For macropores ranging from 1 to 350  $\mu$ m, the critical pore entry diameter of H2P6 and H3P6 samples are 60.41  $\mu$ m and 145.17  $\mu$ m, respectively. It is revealed that the concentration of H<sub>2</sub>O<sub>2</sub> dominates the critical pore entry diameter. In addition, the cumulative pore volume was significantly increased after the addition of palm olein. As shown in Table 3, for geopolymer foams with 2 and 3 wt.% H<sub>2</sub>O<sub>2</sub>, the cumulative pore volumes of macropores are 0.973 mL/g and 1.185 mL/g, which are three and two times higher than samples without palm olein (H2P0 and H3P0). The effect

of palm olein content on macropores was also carried out. The macropores distribution is barely changed except for a slight increase in the peak, which means there will be an improvement in open porosity.

Sample	Pore Volume 0.005–1 μm (mL/g)	Pore Volume 1–350 μm (mL/g)	Total Pore Volume (mL/g)	Total Pore Area (m²/g)	Average Pore Diameter (nm)
H0P0	0.377	0.014	0.391	60.563	25.824
H2P0	0.397	0.333	0.729	53.731	54.300
H3P0	0.379	0.492	0.871	51.420	67.771
H2P6	0.381	0.973	1.354	58.370	92.815
H3P6	0.397	1.185	1.581	53.158	118.981
H3P10	0.413	1.242	1.655	49.421	133.951

Table 3. Pore volume, pore area, and average pore diameter of geopolymer foams from MIP.

Another attractive finding is that the critical pore entry diameter of the intrinsic pores in geopolymer foams was decreased with palm olein addition, even smaller than that in the matrix. For H2P6 and H3P6 samples, the critical pore entry diameter in the range from 0.005 to 1  $\mu$ m are 0.023  $\mu$ m and 0.029  $\mu$ m (Figure 8C), while for H2P0, H3P0, and the matrix (H0P0), the critical pore entry diameters are 0.026  $\mu$ m, 0.029  $\mu$ m, and 0.026  $\mu$ m (Figure 8A), respectively.

This reduction in intrinsic pores was attributed to the sodium fatty acid generated in situ by saponification. Sodium fatty acid is an anionic surfactant with low critical micelle concentration (CMC). It can quickly form micelles at low concentrations. Sodium fatty acid molecules will be adsorbed on the pore surface to improve foam stability [42]. The intrinsic pore diameter will decrease when surfactant molecules generated in situ are adsorbed on the pore surface.

The pore size distribution and total pore volume changes will affect the total pore area, which is directly related to the adsorption capacity of geopolymer foams. The cumulative pore area data of matrix and geopolymer foams measured by MIP are presented in Figure 9. As shown in the curve, the total pore area of the samples is almost all derived from the intrinsic pores. These macropores have little to no effect on the cumulative pore area, though most of the pore volume was contributed by them. With 6 wt.% palm olein addition, a slight increase in the cumulative pore area of the H2P6 and H3P6 samples. It indicates that the reduction in intrinsic pores will be beneficial for improving the pore adsorption capacity.



**Figure 9.** Cumulative pore area of matrix and geopolymer foams: (**A**) H0P0, H2P0, and H3P0; (**B**) H2P6, H3P6, and H3P10.

However, when the palm olein content increased from 6 to 10 wt.% (H3P6 and H3P10), the total pore area of geopolymer foams with 3 wt.%  $H_2O_2$  decreased from 53.158% to 49.421% (Figure 9 and Table 3). It can be seen from Figures 8C and 9B that the difference in cumulative pore area is mainly derived from the pores ranging from 0.005 to 0.01  $\mu$ m. These pores contributed a pore area of 5 m<sup>2</sup>/g in the H3P6 sample, while the H3P10 sample was almost free of pores in the same range. These pores disappeared in geopolymer foams with high palm olein content may be filled by surfactant micelles, so they cannot be detected by MIP.

## 3.4. Porosity

The skeletal density and bulk density of samples obtained by MIP and geometric method are listed in Table 4. The open porosities of geopolymer foams calculated by Equation (1) are shown in Figure 10, along with the porosity determined by MIP.

Table 4. Skeletal and bulk densities of geopolymer foams.

Sample	Skeletal Density by MIP (g/cm <sup>3</sup> )	Bulk Density at 0.52 Psia by MIP (g/cm <sup>3</sup> )	Bulk Density by Geometric Calculation (g/cm <sup>3</sup> )
H0P0	2.248	1.197	$1.118\pm0.010$
H2P0	2.007	0.815	$0.580\pm0.006$
H3P0	1.841	0.707	$0.478 \pm 0.024$
H2P6	1.726	0.504	$0.482\pm0.002$
H3P6	1.702	0.515	$0.370\pm0.002$
H3P10	1.807	0.469	$0.355\pm0.001$



Figure 10. Open porosity of matrix and geopolymer foams.

The porosity of all of the geopolymer foams exceeded 70% with or without the addition of the foam stabilizer. For samples with 3 wt.%  $H_2O_2$ , the open porosity increased from 74.01% to 79.50%, using 6 wt.% palm olein. This proves that the use of palm olein can promote the opening of pores. However, for the sample with 2 wt.%  $H_2O_2$  doping, there was only a slight increase in open porosity after using palm olein, which was inconsistent with the results measured by the MIP. It might be due to the fact that the use of palm olein improved the homogeneity of pores. Pores in these samples with diameters that exceed the upper limit of MIP cannot be detected. When adding the foam stabilizer, the pore size distribution of geopolymer foams was adjusted and concentrated so that most of the pores are less than 350  $\mu$ m in size and can be detected. This also reveals that palm olein can optimize the pore size distribution and improve the open porosity in geopolymer foams. Comparing H3P6 and H3P10 samples, it can be found that increasing the amount of foam stabilizer can also slightly increase the open porosity. A linear regression analysis was employed to describe the effect of open porosity on the bulk density of geopolymer foams. Regression lines in Figure 10 showed a distinct negative correlation between the bulk density and open porosity of geopolymer foams, which is consistent with the finding in literature [5].

More accurate macropore porosity can be obtained by combining MIP with geometric calculations. Table 5 presents the data extracted from the MIP and geometric calculations for the graded porosity analysis (Graded by diameter (d)). The graded porosity of macropores and intrinsic pores of the geopolymer foams is provided in Figure 11.

**Open Porosity (%)** Sample  $d < 1 \,\mu m$  $1 \,\mu\mathrm{m} \leq d < 350 \,\mu\mathrm{m}$  $d \geq$  350  $\mu$ m  $d \geq 1 \ \mu m$ 45.071 1.709 H0P0 3.492 5.201 H2P0 32.316 27.102 11.694 38.796 H3P0 26.783 34.808 12.420 47.228 H2P6 19.638 50.103 1.910 52.013 H3P6 18.581 55.493 5.429 60.922 H3P10 18.684 56.246 5.427 61.673



Figure 11. Graded open porosity of matrix and geopolymer foams.

The porosity of the macropores was further improved using the foam stabilizer. For samples with  $H_2O_2$  dosage of 2 and 3 wt.%, the porosity of the macropores increased from 38.80 and 47.23% to 52.01 and 60.92% with the addition of 6 wt.% palm oil, respectively. The intrinsic porosity in the same samples was also reduced accordingly. These results reveal the effect of palm oil on foam enhancement. The foam film was enhanced by the in situ generated surfactant, preventing drainage and consolidation from occurring. As a result, more of the gas generated by  $H_2O_2$  can be trapped in the matrix and form pores.

Table 5. Graded open porosity of geopolymer foams.

For geopolymer foams with  $H_2O_2$  dosage, increasing the palm content from 6% to 10% has a negligible effect on the macroporosity.

# 3.5. Pore Shape Evaluation

Cross-sectional images of porous materials are often used for pore structural analysis, such as pore distribution frequency [24], porosity [50], and the shape of pores [4]. The cross-sectional binarized images of the matrix and geopolymer foams involved in this study are presented in Figure 12.



Figure 12. Cross-sectional binarized images of matrix and geopolymer foams.

As can be seen in the binarized cross-sectional images, there are heteromorphic wormlike pore structures in the geopolymer foams in which palm olein was used. Observing the microscopic morphology of geopolymer foams, these heterogeneous macropores are composed of many uniform fine pores that have not fully merged. *Roundness* is a parameter used to assess the extent to which pores or particles deviate from a mathematically perfect circle. It can be used to evaluate the effect of palm olein on the shape of macropores in geopolymer foams [37,51].

The pore *roundness* can be deduced from Equation (1). The perimeter, area, and minimum diameter of each identifiable pore were captured by software for calculation and correlation analysis. Figure 13 gives a cloud plot of the *roundness* distribution of pores against the minimum diameters in geopolymer foams. A color cross-sectional view of the sample is provided in the upper right corner of each image, where each pore was filled using the color corresponding to the *roundness*.



Figure 13. Cloud plots of pore roundness distribution of geopolymer foams.

As shown in Figure 13B,C, in the samples using only  $H_2O_2$ , the *roundness* of almost all of the pores was below 5. The *roundness* of pores with a diameter greater than 1000  $\mu$ m is practically located in the range from 1 to 2. The cloud map of pore distribution shows that the shape of pores with a *roundness* of less than 2 is mostly circular or sub-circular. It means that these pores are isolated and not interconnected to others. The distribution frequency and area fraction of pores with different *roundness* are illustrated in Figure 14.



Figure 14. Pore distribution frequency and area fraction of geopolymer foams.

The trend of the pore *roundness* distribution is the same as in Figure 13. For the H2P0 and H3P0 samples, about three-fourths of the pores have *roundness* between 1 and 2. It can also be found that the area fraction is consistent with the distribution frequency of pores, which indicates that the *roundness* distribution has similar regularity in all pore size ranges. The macropores in geopolymer foams using  $H_2O_2$  only are nearly closed pores with a

low interconnectivity, which is consistent with the presented microscopic morphology in Figure 6.

Figure 13D–F shows that many pores with *roundness* higher than 5 were found in the geopolymer foams containing palm olein. These pores correspond to the heterogeneous pores on the color cloud map. Moreover, the shape of these pores becomes more irregular as the *roundness* increases. As presented in Figure 14E, for H2P6 samples, the proportion of pores with *roundness* between 5 and 10 is 3.92%, which is 7 times as high as that of geopolymer foams with  $H_2O_2$  only. In parallel, the proportion of pores with *roundness* greater than 20 were observed. In addition, the pore diameter distribution in geopolymer foams was significantly narrowed after using palm olein, as shown in Figure 13. As with the H3P6 sample, the pore diameter distribution ranges from 0 to 1000  $\mu$ m, one-third of the geopolymer foams without palm olein addition.

It can be observed from Figure 14 that pores with large *roundness* (larger than 5) contribute more to the total area, despite their low frequency of distribution. This indicates that the macropores with larger *roundness* are composed of more individual fine pores that are not fully integrated. In other words, the *roundness* of a heteromorphic pore is positively related to the number of individual pores it contains.

#### 3.6. Connectivity of Pores

As shown by binarized and microscopic images, the geopolymer foams with the addition of palm olein contained many heteromorphic worm-like pores. These irregular pores, formed by interconnected individual fine pores, significantly affect the macropore connectivity of geopolymer foams. As mentioned in the previous section, the *roundness* of these pores is correlated positively with the number of individual pores. This makes it possible to use *roundness* to evaluate the local connectivity of these irregular pores.

Here, we define the local connectivity ( $\kappa_{Local}$ ) as the number (*n*) of internally connected pores that constitute an irregular pore. A simplified interconnected pore model was proposed to derive the relationship between each irregular pore's *roundness* and local connectivity, as shown in Figure 15. The model is based on 2 assumptions: (1) all individual pores are identical spheres; (2) except for the pores located at the ends, the remaining pores are only connected to other two pores through single points. It can be summarized that the model is only related to the number of pores involved and is applicable to various shapes of pores that satisfy the assumptions.



Figure 15. A simplified interconnected pore models of geopolymer foams.

As shown in Figure 15, for an irregular pore consisting of *n* internally connected pores, the calculated pore *roundness* is also approximately equal to *n*. According to the definition of local connectivity in this study, the local connectivity of the corresponding irregular pore is *n*. It means that the pore *roundness* can be used to evaluate the local connectivity quantitatively. Based on a simplified linear connectivity model, a perfect linear relationship exists between pore *roundness* and local connectivity.

The local connectivity of all of the pores determines the global connectivity of geopolymer foams. As shown in Figure 13, the connectivity in geopolymer foams with different  $H_2O_2$  and palm olein contents is significantly different.

Standard deviation is often used to evaluate the degree of dispersion of a data set. Here, an attempt to evaluate the collated connectivity of geopolymer foams was processed by calculating the standard deviation of the local connectivity of all pores in cross-sectional images.

In this study, the global connectivity ( $\kappa_{Global}$ ) of geopolymer foams was defined as the standard deviation ( $\sigma$  ( $\kappa$ )) of the *roundness* of all identifiable pores on the two-dimensional images. It can be calculated by Equation (6):

$$\kappa_{Global} = \sigma(\kappa) = \sqrt{\frac{\sum_{i=1}^{n} (\kappa_i - \kappa_{avg})^2}{n}}$$
(6)

where  $\kappa_i$  is the local connectivity of the *i*th pores,  $\kappa_{avg}$  is the average of the local connectivity of all pores, and *n* is the total number of pores.

The global connectivity and the total number of pores for statistical analysis are listed in Table 6. The global connectivity of geopolymer foams showed a significant increase after using palm olein. For samples H2P6 and H3P6, the global connectivity is twice and three times higher than these without palm olein. It indicates that palm olein can improve the pore connectivity of geopolymer foams with the cooperation of  $H_2O_2$ . However, the global connectivity of geopolymer foams was not significantly improved by increasing the amount of palm olein with the same  $H_2O_2$  dosage. The result suggests there is an optimal doping of palm olein for a defined geopolymer foams system and that overdosing does not further affect the pore connectivity of geopolymer foams.

Table 6. Global connectivity and the total number of pores of geopolymer foams.

Sample	п	$\kappa_{Global}$
H2P0	1929	0.754
H3P0	1357	0.752
H2P6	2247	1.383
H3P6	1608	2.128
H3P10	1546	2.141

The open porosity of porous materials is often used to describe the pore connectivity in porous structures qualitatively. The histograms of connectivity and macropore porosity for different samples shown in Figure 16 illustrate a positive correlation between these two parameters. A simple linear regression analysis was performed to verify the validity of this quantitative connectivity evaluation method. The regression line, fitting equation, and R-squared value presented in Figure 17 reveal that the values derived by this method can represent the connectivity of the pores in geopolymer foams.

However, the simplified model was built based on the two assumptions previously mentioned. For geopolymer foams with a wider pore size distribution and more complex connectivity, the model needs to be modified accordingly.

#### 3.7. Water Absorption

Figure 18 shows the water absorption of the geopolymer foams with different  $H_2O_2$  and palm oil contents. It can be seen that the water absorption of the geopolymer foams increases significantly with the amount of  $H_2O_2$ . The water absorption of the sample with

the addition of 3 wt.%  $H_2O_2$  was 141.925%, which was 2 times higher than that of the matrix. The water absorption of sample H3P6 has reached 221.169%. This indicates that the use of palm olein can further improve the absorption capacity of geopolymer foams. A similar trend was observed in the samples with low  $H_2O_2$  incorporation.



Figure 16. Global connectivity and macropore porosity of geopolymer foams.



Figure 17. Regression analysis of pore connectivity of geopolymer foams.

The water absorption of geopolymer foams is mainly influenced by the open pore structure. The effects of open porosity on the water absorption of geopolymer foams with or without palm olein addition were discussed and presented in the accompanying figure of Figure 18. It is evident that the open porosity of geopolymer foams and their water absorption is significantly positively correlated regardless of the addition of palm olein, which means that the water absorption of geopolymer foams can be improved by increasing the open porosity. It is worth mentioning that the regression line slope was higher for geopolymer foams with palm olein addition compared to those with  $H_2O_2$  only. This indicates that the former has better water absorption, which also validates the positive

effect of palm olein on increasing the open porosity of geopolymer foams. In summary, geopolymer foams with higher open porosity have better water absorption capacity. This makes geopolymer foams have the potential to be used in moisture conditioning and composite carriers.



Figure 18. Water absorption of matrix and geopolymer foams.

## 3.8. Thermal Conductivity

The effect of  $H_2O_2$  and palm olein doping on the thermal conductivity of geopolymer foams is shown in Figure 19. It can be found that the thermal conductivity of samples gradually decreases with the increase in  $H_2O_2$  dosages. As the  $H_2O_2$  content increased from 2 to 3 wt.%, the thermal conductivity of geopolymer foams reduced from 0.127 to 0.103 W/ (m·K), which is only two-fifths of that for the matrix. This is because the foaming agent introduces plenty of closed pores into the matrix [37].



Figure 19. Thermal conductivity of matrix and geopolymer foams.

With the use of palm olein, geopolymer foams with lower thermal conductivity were obtained. For the H3P6 sample, thermal conductivity as low as 0.081 W/(m·K) was measured. This indicates that the thermal insulation properties of geopolymer foams can be improved with the addition of palm olein. The thermal conductivity of geopolymer foams did not change obviously by further increasing the amount of palm olein, which is attributed to the fact that excessive palm olein dosage can no longer significantly affect the pore structure. It can be concluded that geopolymer foams with different thermal insulation properties can be obtained by adjusting the amounts of H<sub>2</sub>O<sub>2</sub> and palm olein. Regression analysis showed that the thermal conductivity of geopolymer foams was highly linear correlated with their bulk density ( $R^2 = 0.995$ ). Combining the results presented in Figure 10, we can infer that there is also a negative correlation between open porosity and thermal conductivity in geopolymer foams.

## 3.9. Compressive Strength

The effects of  $H_2O_2$  and palm olein on the compressive strength of geopolymer foams are presented in Figure 20. With  $H_2O_2$  content increasing, the compressive strength of geopolymer foams declined regardless of whether palm olein was added. However, palm olein positively affects the strength development of geopolymer foams with low  $H_2O_2$ doping. For samples with 2 wt.%  $H_2O_2$ , the compressive strength increased by 75% from 1.534 to 2.685 MPa after adding 6 wt.% palm olein. This enhancement effect should be attributed to the modulation of pore distribution in geopolymer foams by palm olein. As shown in Figure 8, the H2P6 sample possessed a more homogeneous pore structure than H2P0, which is beneficial for compressive strength.



Figure 20. Compressive strength of matrix and geopolymer foams.

All of the regression analyses indicate that the various properties of geopolymer foams are closely related to the pore characteristics. As mentioned before, the open porosity significantly affects the water absorption, mechanical behaviors, and thermal conductivity of geopolymer foams. This study confirms the positive role of palm olein in the optimization and regulation of pore structure. With the synergistic effect of  $H_2O_2$  and palm olein, geopolymer foams featuring lightweight, low thermal conductivity, and good mechanical properties can be prepared. Compared with lightweight geopolymers reported in the literature [6,46,52], the geopolymer foams fabricated in this study (H2P6) possess comparable low thermal conductivity, allowing them to be used for building thermal insulation.

Moreover, geopolymer foams with high open porosity show favorable capabilities in heavy metal adsorption [53,54], liquid phase separation [55], and wastewater purification [56]. Therefore, the geopolymer foams prepared by this research route have significant potential in all the fields mentioned above.

Many pieces of literature reported an exponential correlation between porosity and compressive strength of geopolymer foams [4,5]. In this study, a non-linear regression analysis was performed to discuss the relations between these two parameters of geopolymer foams. The regression curve and correlation coefficient in Figure 20 illustrate an exponential correlation between the compressive strength of geopolymer foams and their open porosity, which means that geopolymer foams with adequate mechanical properties can be obtained by modulating the porosity.

## 4. Conclusions

Geopolymer foams with high porosity and hierarchical pore structure were prepared using the direct foaming method with  $H_2O_2$  as the foaming agent and palm olein as the foaming stabilizer. The chemical features and pore structure of geopolymer foams with and without palm olein addition were investigated by experiments and calculations. Meanwhile, the water absorption, thermal insulation, and mechanical properties of geopolymer foams were evaluated. The main findings obtained in this study are as follows.

Fatty acid salt surfactant was generated in situ in the geopolymer foams after adding palm olein. However, the addition of  $H_2O_2$  and palm olein did not affect the phase composition of the amorphous structure in the geopolymer matrix.

The macropores in geopolymer foams introduced by  $H_2O_2$  were regulated by the surfactant generated in situ via saponification of palm olein with alkali solution. A uniformly concentrated macropore distribution was obtained in geopolymer foams using palm olein. The novel foam stabilizer also affects the structure of the intrinsic pores formed by geopolymerization. With palm olein addition, the intrinsic pores in geopolymer foams were fine-tuned, which resulted in an elevated total pore area.

The addition of palm olein has the effect of increasing the open porosity of the geopolymer foams. However, the impacts on macropores and intrinsic pores are different. The open porosity of macropores in geopolymer foams significantly increases after palm olein use, while the opposite effect was observed for intrinsic pores.

The roundness analysis indicates that irregular worm-like pores were formed in geopolymer foams with palm olein addition. These irregular pores are composed of interconnected, not fully consolidated individual fine pores. A simplified interconnected pore model based on the statistical analysis of pore roundness was proposed to quantify the macropore connectivity of geopolymer foams. The results demonstrated that the macropore connectivity of geopolymer foams was significantly improved using palm olein. The validity of the connectivity evaluation method based on the model was verified by correlation analysis.

The water absorption and thermal insulation capacity of geopolymer foams were significantly affected by the porosity and pore connectivity due to the addition of  $H_2O_2$  and palm olein. By adding 3 wt.%  $H_2O_2$  and 6 wt.% palm olein, geopolymer foams with water absorption of 221.169% and thermal conductivity of 0.081 W/(m·K) can be obtained. The compressive strength of geopolymer foams reduces with  $H_2O_2$  content increasing, while palm olein positively affects the strength development of geopolymer foams with low  $H_2O_2$  doping. Regression analysis showed strong correlations between the above properties and the pore characteristics of geopolymer foams.

From the results of this study, it can be concluded that palm olein is an efficient foam stabilizer that can be used to prepare eco-friendly geopolymer foams with different pore structures. However, the mechanism of pore structure evolution of geopolymer foams fabricated using palm olein still needs to be systematically investigated. Through further modification of the pore structure, geopolymer foams can be used in a wide range of applications such as adsorption, filtration, and reaction carriers.

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