



Article

Characterisation of the Chemical Composition and Structural Features of Novel Antimicrobial Nanoparticles

Yuen-Ki Cheong ¹ , Jesus Calvo-Castro ², Lena Ciric ³, Mohan Edirisinghe ⁴, Elaine Cloutman-Green ⁵, Upulitha Eranka Illangakoon ⁴, Qiang Kang ⁶, Suntharavathanan Mahalingam ⁴, Rupy Kaur Matharu ⁴, Rory M. Wilson ⁷ and Guogang Ren ^{1,*}

¹ School of Engineering and Technology, University of Hertfordshire, Hatfield AL10 9AB, UK; y.cheong2@herts.ac.uk

² School of Life and Medical Sciences, University of Hertfordshire, Hatfield AL10 9AB, UK; j.calvo-castro@herts.ac.uk

³ Department of Civil, Environmental and Geomatics Engineering, University College London, London WC1E 7JE, UK; l.ciric@ucl.ac.uk

⁴ Department of Mechanical Engineering, University College London, London WC1E 7JE, UK; m.edirisinghe@ucl.ac.uk (M.E.); upulitha.illangakoon.12@ucl.ac.uk (U.E.I.); suntharavathanan.mahalingam@ucl.ac.uk (S.M.); rupy.matharu.15@ucl.ac.uk (R.K.M.)

⁵ Department of Microbiology, Virology, and Infection Prevention Control, Great Ormond Street Hospital NHS Foundation Trust, London WC1N 3JH, UK; elaine.cloutman-green@gosh.nhs.uk

⁶ Institute of Metal Research, Chinese Academy of Science, 72 Wenhua Road, Shenyang 110016, China; qkang@imr.ac.cn

⁷ Materials Research Institute, Queen Mary University of London, London E1 4NS, UK; r.m.wilson@qmul.ac.uk

* Correspondence: g.g.ren@herts.ac.uk; Tel.: +44-1707-28-1075

Received: 28 March 2017; Accepted: 14 June 2017; Published: 23 June 2017

Abstract: Three antimicrobial nanoparticle types (AMNP0, AMNP1, and AMNP2) produced using the TesimaTM thermal plasma technology were investigated and their compositions were determined using a combination of analytical methods. Scanning electron micrographs provided the morphology of these particles with observed sizes ranging from 10 to 50 nm, whilst FTIR spectra confirmed the absence of polar bonds and organic impurities, and strong Raman active vibrational bands at ca. 1604 and 1311 cm⁻¹ ascribed to C–C vibrational motions were observed. Carbon signals that resonated at δ_C 126 ppm in the solid state NMR spectra confirmed that sp² hybridised carbons were present in high concentration in two of the nanoparticle types (AMNP1 and AMNP2). X-ray powder diffraction suggested that AMNP0 contains single phase Tungsten carbide (WC) in a high state of purity and multiple phases of WC/WC_{1-x} were identified in both AMNP1 and AMNP2. Finally, X-ray photoelectron spectral (XPS) analyses revealed and quantified the elemental ratios in these composite formulations.

Keywords: antimicrobial; antiviral; antibacterial; nanoparticles; tungsten; carbide; Raman; XRD; SS-NMR; XPS

1. Introduction

Nanoparticles (NP) have been extensively investigated in biomedical applications ranging from biomaterials, diagnostics, to therapeutic treatments for cancers and other related diseases [1–10]. Research into the use of engineered bulk size materials (i.e., stainless steels) [11–13], micro- [14,15], and nano- [1,16] sized materials (i.e., metal and metal oxides) against a spectrum of bacteria and viruses

have been a popular field, especially since the SARS (Severe acute respiratory syndrome) outbreak in 2003. The continuation of growing numbers of infectious diseases in hospitals, [17] propagation of pathogens, and their resistance against conventional antibiotics [18] have significantly raised global concern. Therefore using nanomaterials as antibacterials complementary to antibiotics is highly promising and is gaining a large interest, as they may fill the gaps where antibiotics frequently fail [19].

The activity of a widening spectrum of engineered metal/metal-oxide nanoparticles which can counter a specific range of oral pathogens associated with *peri-implantitis* has been investigated [20]. The findings assisted in the development of novel and innovative antimicrobial agents in an era of ever-increasing antimicrobial resistance. The potential of using nanoparticles as antimicrobial agents raised concerns about the possibility of nanoparticle toxicity on the central nervous system (CNS) [21–23]. In vivo investigation using hippocampus cells in young rats suggested an interactive connection between nano ZnO/Ag/CuO and the effectiveness of learning ability and the flexibility of cognition [21,24–26]. Nevertheless, incremental nano-dosages increased the excitability in rat CA1 pyramidal neurons and confirmed a safety threshold of 0.05 wt % of injection suspension, which has provided an insight regarding the relative toxicity concern over the use of nanoparticles in biomedical engineering [27–29]. These findings proved the novelty of antimicrobial nanoparticles (AMNP) in a cost-effective fashion for inhibiting microbial growth, which may bring major changes in policy and new regulations on the wider uses of nanoparticles in biomedical healthcare [7,16,30].

AMNP derivatives were synthesized in a previous work with a stable efficacy (99.99% killing rates) in order to counteract a range of super bugs (i.e., *E. coli*, MRSA) and a range of viruses (SARS, H5N1, and Noroviruses) [30,31]. As well as the accumulative biological results obtained from AMNP powder suspensions, AMNP doped polymer fibres produced using pressurized gyration [32] were recently found to inhibit the growth of *P. aeruginosa*, a Gram-negative bacterial species that is commonly found in hospitals [33].

Although, AMNP nanoparticles exhibit antimicrobial functions, their chemical and particle characterisations are not documented. Furthermore, how they interact with microbes has not been directly investigated. Most known antibacterial nanomaterials interact electrostatically with the bacterial membrane causing membrane disruption [19,34]. Consequently, free radicals are produced and instigate secondary membrane damage causing protein malfunction and DNA destruction [3]. Other antimicrobial nanomaterials such as nitric oxide matrices involve photoactivation with RNS (Reactive oxygen and nitrogen species) [35] or polycationic nanomaterials induce signal secretion to promote programmed cell death [36].

In this paper, we report chemical analyses obtained from these AMNP series (AMNP0, AMNP1, and AMNP2) using a range of techniques including FTIR/Raman spectroscopy, Solid state Carbon-13 Nuclear Magnetic Resonance spectroscopy, Powder X-ray Diffraction and X-ray Photoelectron Spectroscopy [37]. Through the chemical investigation, we found that C, W, Ag, Cu, and O are the main elements contained in these AMNP formulations, and we also identified several phases present in these series (i.e., WC, WC_{1-x}, CuO). By understanding the lattice structures and chemistry of these antimicrobial nanoparticle composites, we hope to identify biological surface-interactions between these ultra-small particles and the microbes. This will help in studying the reaction mechanisms involved in target bacterium or viruses, and will also help to design and create custom-made antimicrobial formulations in the future.

2. Results and Discussion

2.1. Particle Characteristics

The surface morphologies of the AMNP derivatives were studied using SEM. Figure 1a–c show the representative SEM images of AMNP0, AMNP1, and AMNP2, respectively. As shown in Figure 1a, the SEM image of the examined AMNP0 particles indicates that the sample was quite uniformly distributed. Although a large differentiation in their particle sizes were measured (50–500 nm), we

were able to observe the apparent hexagonal units' presence in these particles, which were later confirmed to be the $P\bar{6}m2$ tungsten carbide in our XRD study. In contrast, both SEM images of AMNP1 (Figure 1b) and AMNP2 (Figure 1c) showed a significant decrease in their particle sizes (10–30 nm). Both AMNP1 and AMNP2 were found to be heavily agglomerated and highly charged due to the nonconductive nature (i.e., non-metallic) of the samples. As a consequence, capturing good SEM images at higher resolution using accelerating voltage was difficult.

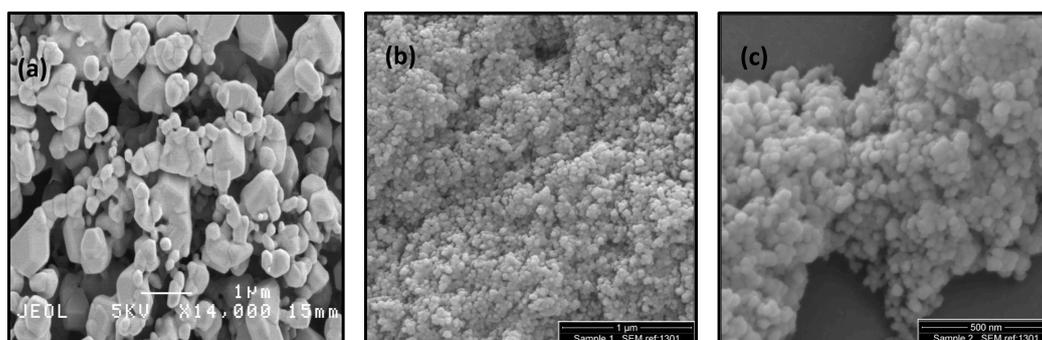


Figure 1. SEM images of (a) AMNP0; (b) AMNP1 and (c) AMNP2. AMNP denotes as Antimicrobial Nanoparticles.

In addition to the SEM analysis, EDX measurements were also performed with the aim of identifying possible elements present in these samples, and the results are presented and discussed in the atomic analysis section.

2.2. Chemical Characteristics

FTIR and Raman spectroscopy were used to identify the types of components (organic/inorganic/polymer) that may be contained in these nanoparticles. Raman and FTIR spectroscopies provided complimentary analyses to characterise the presence of intramolecular bonding in these composites. Four samples were investigated; this included the commercially available WC (<1 μm) from Alfa Aesar[®] (Lancashire, UK), AMNP0, AMNP1, and AMNP2. Figure 2a shows the vibration spectra of Alfa Aesar WC in both FTIR (top) and Raman (bottom) modes, while Figure 2b shows the two spectra obtained from AMNP0. Similarly Figure 3a,b showed both FTIR and Raman vibration spectra of the AMNP1 and AMNP2 powder samples. The FTIR spectra of all four samples (top spectrum of Figures 2a,b and 3a,b) showed only symmetrical stretches near the low frequency fingerprint region (500–1000 cm^{-1}). No distinctive asymmetrical vibration stretches were observed in any of these samples, which suggested that these samples were clean of organic contaminants (i.e., C=O, CH, H₂O).

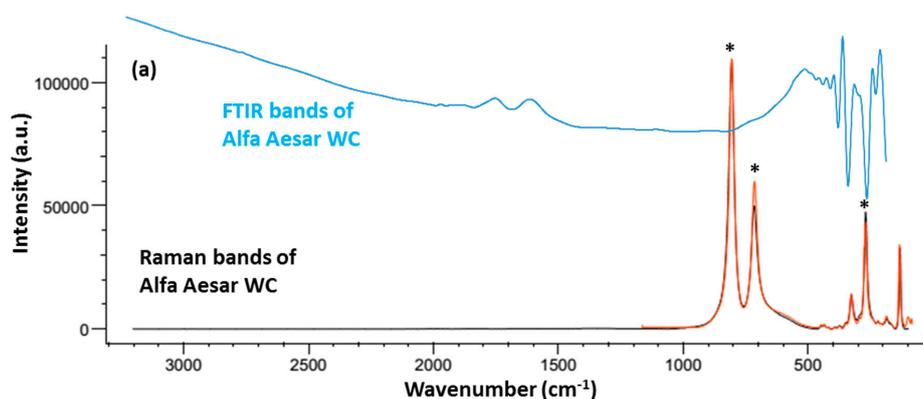


Figure 2. Cont.

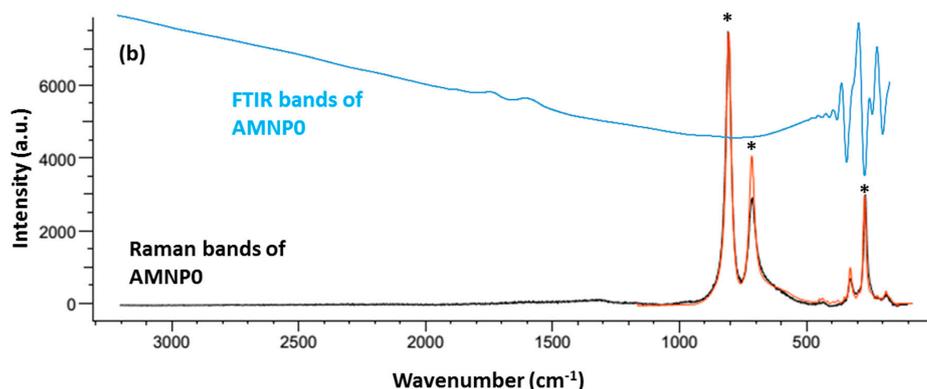


Figure 2. Raman and FTIR spectra of (a) Alfa Aesar tungsten carbide WC and (b) AMNP0. Spectra in red show reference Raman bands associate with related tungsten compounds. Symbols * denote the optimal vibrational Raman peaks observed at 803, 715 and 265 cm^{-1} .

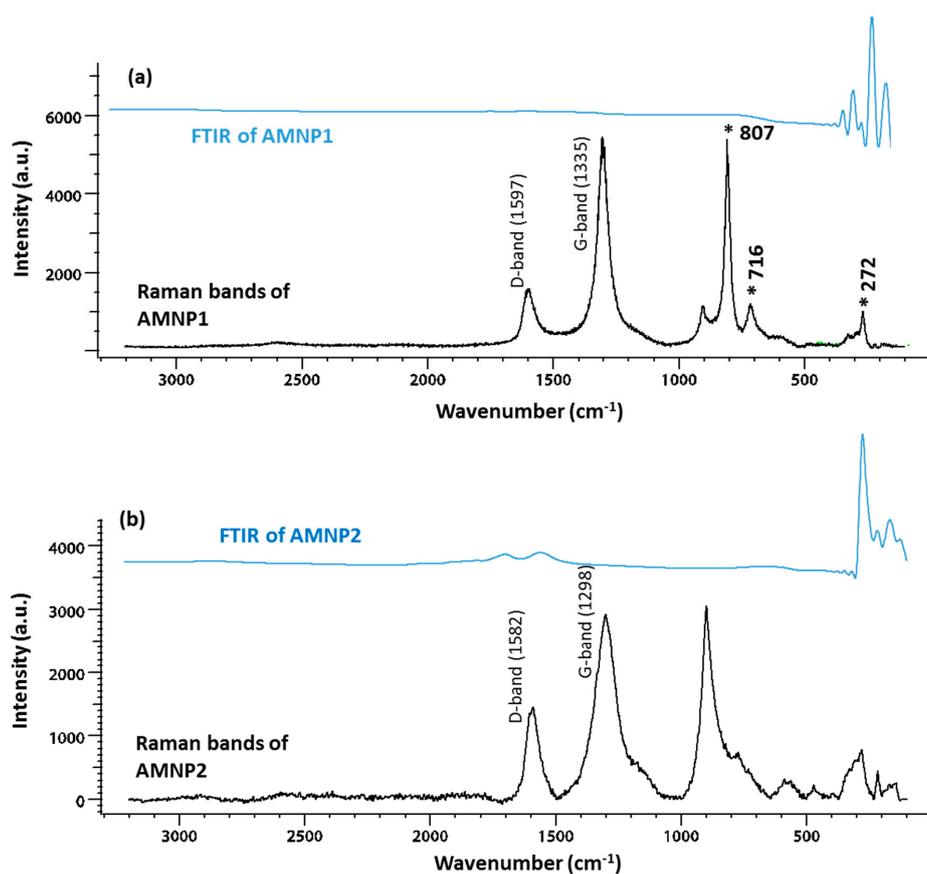


Figure 3. Raman and FTIR spectra of (a) AMNP1 and (b) AMNP2.

Upon detailed analysis of the Raman spectra acquired, additional composition information was obtained in line with the characteristic spectral profiles. In this regard, Raman active vibrational bands at ca. 265, 715, and 803 cm^{-1} observed in the AMNP0 spectrum (Figure 2b) coincide with those for the commercial WC sample [38] (Figure 2a), associated to stretching vibrational motions of the WC triple bond. In turn, the Raman spectra for AMNP1 and AMNP2 (Figure 3a,b) were observed to exhibit significant differences with respect to those of the commercial WC and AMNP0. In summary, clearly defined vibrational bands centred at ca. 1335/1298 and 1582/1597 cm^{-1} were observed for AMNP1/AMNP2, respectively, with the Raman active peaks at lower frequency associated with the

so-called D-band characteristic of the A_{1g} mode of diamond type carbon with sp^3 hybridisation. Thus, the vibrational bands at higher frequency were observed to be in agreement with the G vibrational band (E_{2g}) characteristic of sp^2 hybridised graphitic carbon [38,39]. The relative intensity of the D and G vibrational bands indicates a greater degree of graphitisation in AMNP2 [40]. In addition, Raman active vibrational bands at ca. 807, 716, and 272 cm^{-1} in the spectrum of AMNP1 (Figure 3a) are consistent with the presence of WC (vide supra). In fact, these active Raman bands may also be present in AMNP2 (Figure 3b), only it was swarmed by overlapping with other active signals due to the high complexity of the samples.

^{13}C -NMR spectroscopy is probably one of the best techniques to analyse and study carbon environments. We were unable to obtain a ^{13}C signal from the commercial WC, however, this may be due to the paramagnetic effect caused by the presence of impurities in the commercial sample which was detected later from the XRD analysis. At present, there is insufficient NMR reports relating to metal carbides, whereas here we report the first ^{13}C -NMR resonance observed in a WC sample. The lack of NMR study in metal carbides is due to the lack of aliphatic carbon (CH) present in carbide ions and also the attribution of the sp hybridised carbon, which are known to largely reduce the sensitivity of detection. With the chosen parameters (10 s relaxation delay) applied to these NMR experiments, we were able to observe three active carbon chemical signals resonating at δ_C 123, 254, and 307 ppm (top spectrum in Figure 4) in the AMNP0 sample. The one that resonated at δ_C 123 ppm is typically associated with aromatic/graphitic/charcoal type of carbon. Whereas the other two deshielded resonances at δ_C 254 and 307 ppm indicated carbon environments that are less electronegative, which match the chemical property of a carbide type carbon [41]. In contrast, the signal observed at δ_C 254 ppm was indeed comparable to the NMR shifts observed in Scandium carbide Sc_2C_2 (δ_C 253 ppm) and Yttrium carbide Y_2C_2 (δ_C 257 ppm) reported by Yamazaki et al. [41] and Zhang et al. [42]. An exceedingly rare ^{13}C -NMR resonance published by Buss et al. [43] observed a carbon resonance exhibited at δ_C 546 ppm from a novel molybdenum carbide complex, thus the unusually sharp signal observed at δ_C 307 ppm could also be interpreted as a metal carbide resonance.

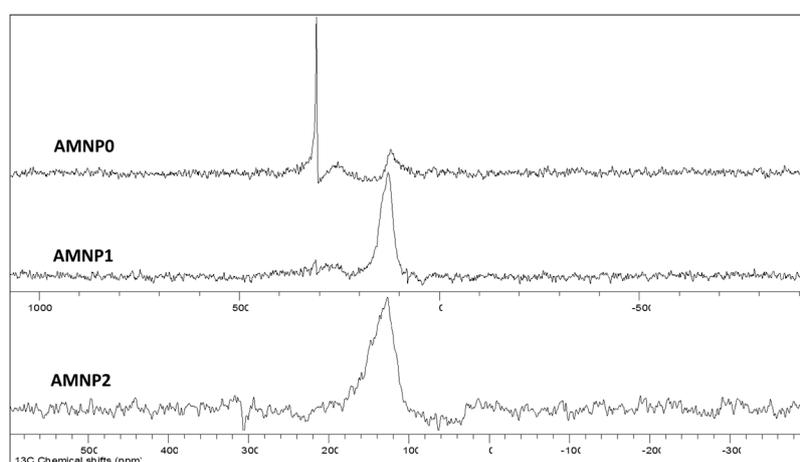


Figure 4. NMR spectra of AMNP0, AMNP1, and AMNP2.

The NMR investigations of AMNP1 and AMNP2 (Figure 4) are straight forward, and the only observable ^{13}C chemical shifts resonated at 126 ppm are again associated with the presence of graphene/graphite types of carbons in both samples. Such high aromatic carbon contents found in both AMNP1 and AMNP2 NMR studies support the hypothesis that was drawn from the previous Raman analyses.

2.3. Phase Identification

Powder X-ray analyses of the commercial WC and AMNP0 samples have unambiguously confirmed that these two samples contain single phase hexagonal WC, known as Qusongite [44]. Figure 5 shows the resulting XRD patterns of the commercial WC and AMNP0 samples against the reference supported by the International Centre of Diffraction Data (ICDD 00-051-0939) is shown as stick diagram in Figure 5). XRD analyses also identified 0.3–0.5% of the impurities as monoclinic MoO₃ and hexagonal MoS₂ (see insert diagram in Figure 5) in the commercial WC sample [45–47], these patterns match those provided by the database service (ICDD 04-004-5723 and ICDD 01-073-1544).

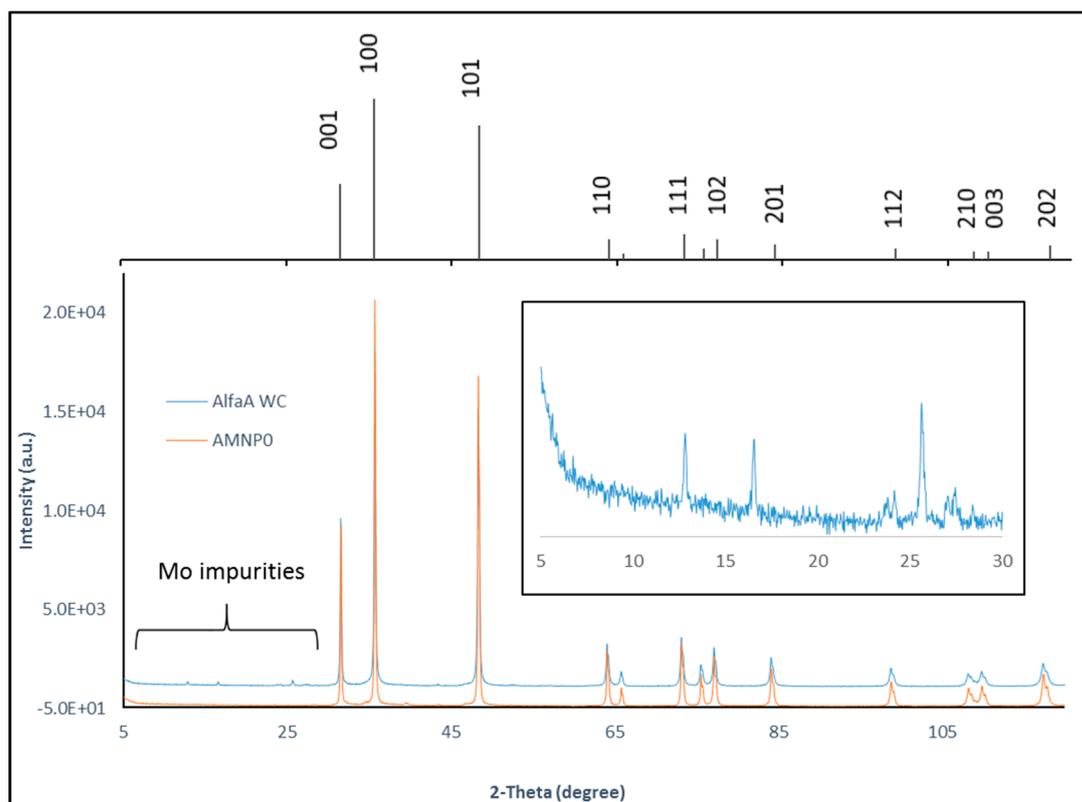


Figure 5. XRD patterns of AMNP0 and Alfa Aesar WC (insert diagram shows the presence of Mo impurities in the commercial sample). Top XRD stick pattern shows hexagonal $P\bar{6}m2$ WC obtained from the International Centre Diffraction Data service (ICDD 00-051-0939).

As expected, a higher degree of broadening effects were observed in the diffractograms obtained from AMNP1 and AMNP2 (Figure 6), which indicated decreasing particle sizes in these samples. Such broadening may also be attributed to the multi-metallic elements present in high carbon content samples. The absence of the graphitic peak at 26° 2θ implies that the carbon is likely to be mainly amorphous.

As shown in Figure 6 (blue diffractogram), multiple phase tungsten carbides were identified in AMNP1; this includes hexagonal WC ($P\bar{6}m2$) and two carbon deficient ones, $W_2C/WC_{0.5}$ and cubic $WC_{0.82}$ (Table 1) [48,49]. As well as tungsten carbide derivatives, some W and Ag in cubic forms were also found in both AMNP1 and AMNP2, and the selective patterns extracted from the ICDD references are represented as stick diagrams on the bottom of Figure 6 [50,51]. As indicated, a trace amount of cubic CuO was also detected in the AMNP2 sample [52]. However, in this case, the XRD results obtained from AMNP1 and AMNP2 were not used to estimate the atomic arrangement and to quantify the elemental ratios in these powders due to the heavy broadening effects observed in these analyses.

It should be noted that the XRD chemical analysis does not reveal all the details of small intermetallic powder samples (see our discussion in Section 2.4).

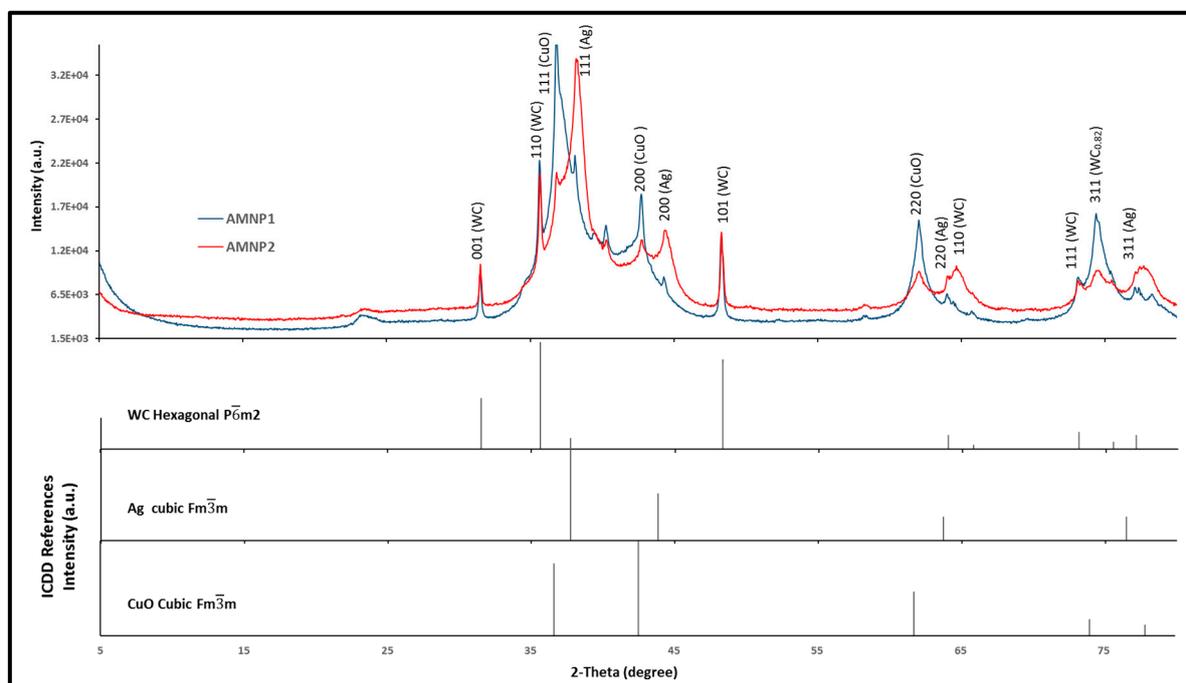


Figure 6. XRD patterns of AMNP1 in blue and AMNP2 in red. Stick diagrams on the bottom show XRD references extracted from the ICDD service.

Table 1. Visible components found in AMNP1 and AMNP2.

AMNP1		AMNP2	
Phases Detected	Details	Phases Detected	Details
Tungsten carbide (WC)	Hexagonal, $P\bar{6}m2$	Tungsten carbide (WC)	Hexagonal, $P\bar{6}m2$
Tungsten carbide ($WC_{0.82}$)	Cubic, $Fm\bar{3}m$		
Tungsten carbide (W_2C)	orthorhombic, $Pbcn$	Copper oxide (CuO)	Cubic, $Fm\bar{3}m$
Tungsten (W)	β -W, cubic $Pm\bar{3}n$	Tungsten (W)	β -W, cubic $Pm\bar{3}n$
Silver (Ag)	cubic, $Fm\bar{3}m$	Silver (Ag)	cubic, $Fm\bar{3}m$

2.4. Atomic and Chemical State Analyses

Preliminary elemental analyses were performed using Energy dispersive X-ray spectroscopy, which was equipped with the Emission field SEM. Multiple point selective analyses of AMNP1 and AMNP2 (Figure 7) using EDX confirmed the presence of all of the elements (i.e., C, W, Ag, and Cu) found in previous Raman, NMR, and XRD analyses.

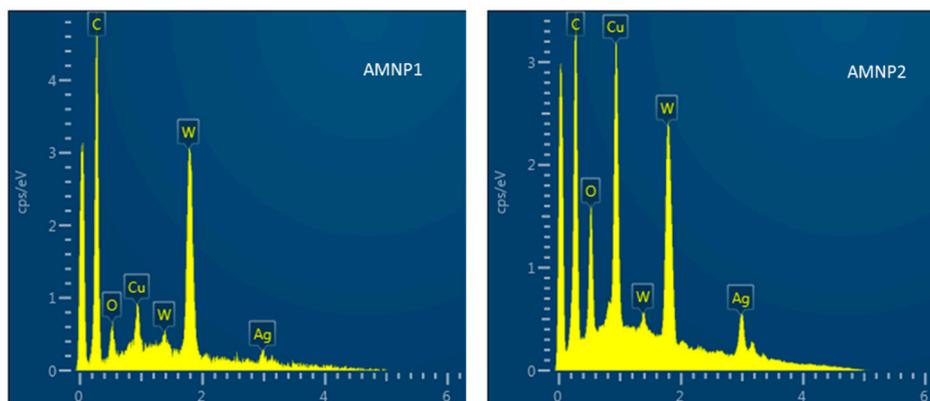


Figure 7. EDX spectra of AMNP1 and AMNP2.

In order to investigate surface composition in the AMNP series and to quantify their elemental ratio, XPS analyses were performed. Figures 8–10 show selected XPS energy profiles (C1s, W4f, Cu2p, and Ag3d) measured at different etch times (0, 30, 60, 120, 200, and 240 s) for the samples of AMNP0, AMNP1, and AMNP2 respectively.

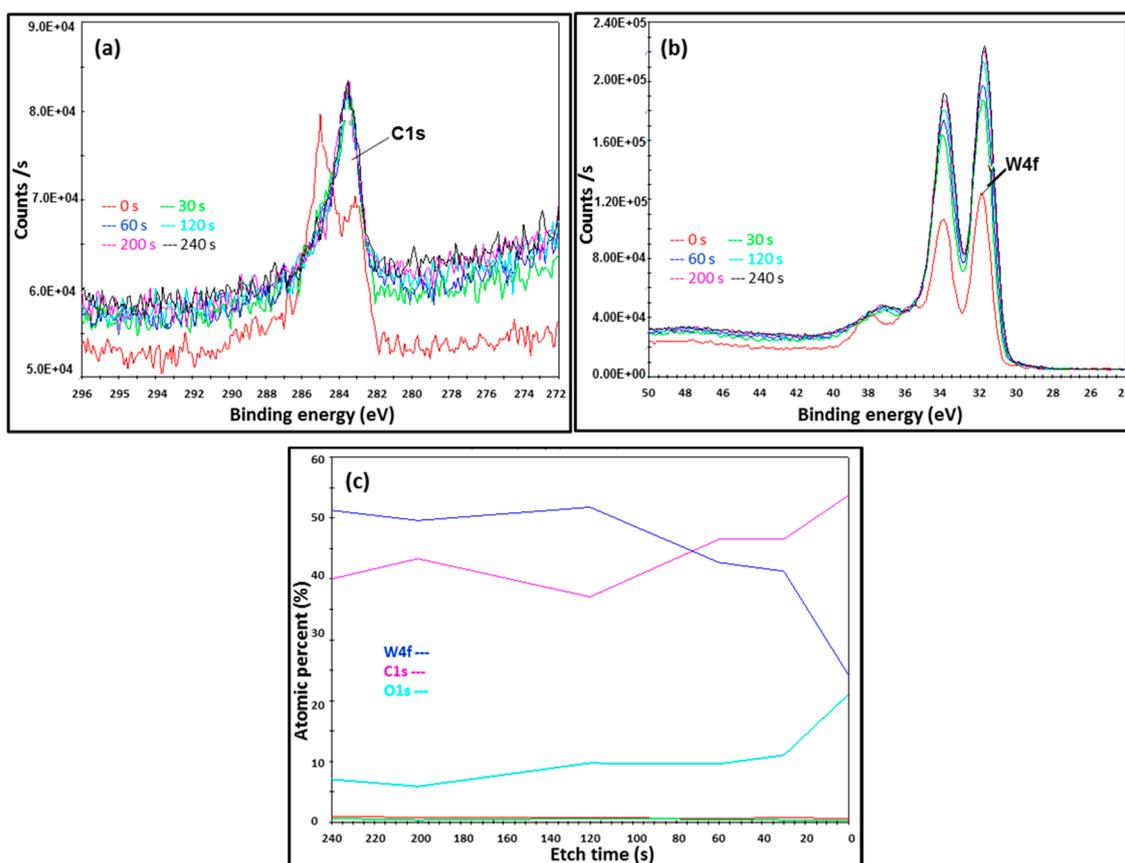


Figure 8. X-ray Photoelectron spectral Energy profiles of AMNP0 shows (a) the C1s binding energies and (b) the corresponding W4f binding energies; (c) shows the XPS atomic ratio analysis of AMNP0.

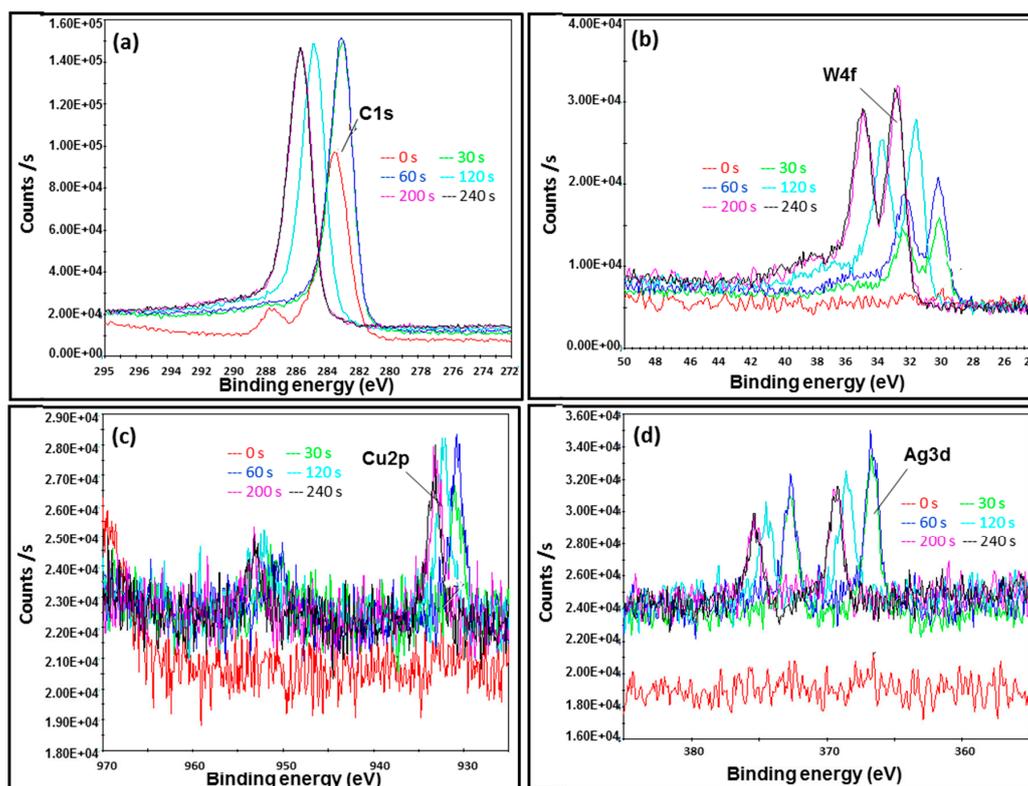


Figure 9. XPS Energy profiles of AMNP1 showing (a) binding energies of the corresponding C1s orbital; (b) the W4f; (c) the Ag3d and (d) the Cu2p, respectively.

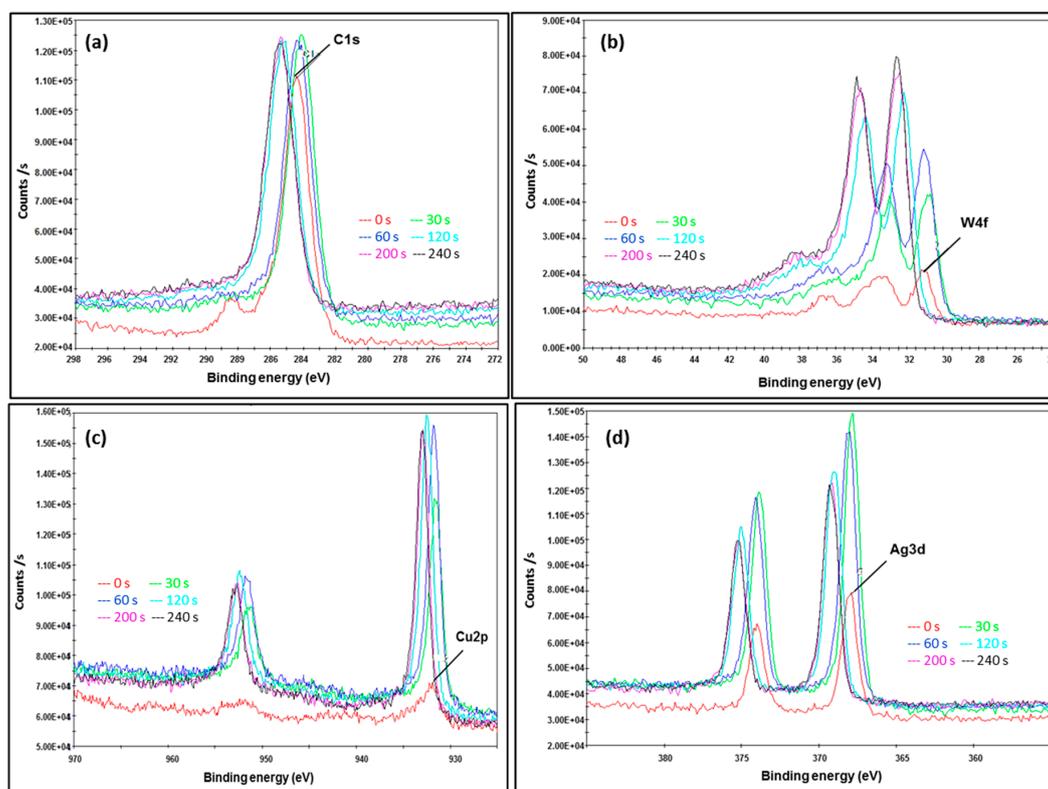


Figure 10. XPS Energy profiles of AMNP2 showing (a) the binding energies of the corresponding C1s orbital; (b) the W4f; (c) the Ag3d and (d) the Cu2p, respectively.

For AMNP0 (Figure 8), two binding energies at 285 and 283 eV (indicated in red in Figure 8a) were initially measured in the C1s spectrum at the 0 s etch time. C1s binding energies were then subsequently stabilised from etch time 30 s onwards. As expected, AMNP0 gave only singlets at 283.4 eV in the C1s spectrum, which is ascribed with the metal carbide energy state [39]. Figure 8b shows the XPS W4f spectrum acquired from AMNP0, consistent binding energies (eV) at 32, 34 (4f_{7/2} and 4f_{5/2} doublet), and 37 (5p_{3/2}) were measured throughout, and these signals again indicated the presence of the carbidic bonding associated with WC in the AMNP0 [53].

Figures 9 and 10 show the XPS energy profiles measured in AMNP1 and AMNP2. It is worth noting that the initial asymmetric peak at 284 eV with a tailed binding energy of 288 eV was found in both C1s spectra (Figures 9a and 10a) in AMNP1 and AMNP2. These are associated with the common adventitious carbon contaminant (according to Thermo Scientific instruments application note). These peaks were subsequently removed by applying Argon clusters and were not observed at the 30 s etch time and onwards. Multiple binding energies (satellite features) were measured between 183 and 285 eV in the C1s spectra (Figures 9a and 10a) indicating that both AMNP1 and AMNP2 have a high concentration of sp² carbons [54,55]. This feature is also supported by the broad and asymmetric tail towards high eV, hence, if the sample contains a high sp³ concentration, the C1s peak will appear to be more symmetrical. Again the high content of graphitic carbon present in both AMNP1 and AMNP2 samples measured in the XPS analyses agrees with the results obtained from the corresponding Raman and NMR data reported earlier in Section 3.2.

As shown in Figure 9b, several binding energies were measured at the W4f core in AMNP1. They all appeared as asymmetric doublets at 30/32 eV, 31.5/33.5 eV, and 32.5/35 eV, and these three sets of doublets are believed to be associated with the multiple phases of W identified in the XRD results (Table 1). Metal Tungsten (W) has a reference XPS binding energy of 31.6/33.5 (doublet) and this W4f_{7/2}/W4f_{5/2} feature in lost as the transition state of W changes to other W compounds (i.e., W₂C/WC), hence eV shifts to higher energy are observed in the cases of both AMNP1 (Figure 9b) and AMNP2 (Figure 10b) [53]. Again, two sets of doublets in the W4f spectrum for AMNP2 (Figure 10b) were measured at 32/33.2 eV and 32/34.2 eV, which suggest different phases of the W compounds (i.e., W and WC) are present in AMNP2.

Although Cu/CuO were not detected in AMNP1 during the XRD analysis, it was perhaps embedded beneath the particle surface. XPS analysis had clearly picked up a trace of Cu as observed in the Cu2p spectrum (Figure 9c), in which the binding energy was measured at 932 ± 1 eV as the etch level increased. Cu2p content was later quantified as 0.5% (Table 2) relative to the entire AMNP1 sample. Similar to the W4f core, significant split spin-orbit doublets ($\Delta = 19.75$ eV) [53] were found in both XPS Cu2p spectra for AMNP1 (Figure 9c) and AMNP2 (Figure 10c). In contrast, the Cu content (CuO as identified from the XRD analysis) measured at 932.8/952.6 eV in AMNP2 is substantially higher than those found in AMNP1.

Table 2. XPS atomic percentage for AMNP0, AMNP1 and AMNP2.

	C	O	W	Ag	Cu
AMNP0	42.7%	8.7%	47.3%	0.5%	0.8%
AMNP1	94.0%	2.9%	2.1%	0.5%	0.5%
AMNP2	77.7%	5.5%	6.7%	4.7%	5.4%

Similarly, a significant amount of Ag in AMNP2 was detected when compared to AMNP1, and Figures 9d and 10d represent the Ag3d spectra for AMNP1 and AMNP2, respectively. All of the Ag3d regions for both AMNP1 (368/374 eV) and AMNP2 (368/374 eV) samples, as expected, appeared as asymmetric doublets and have well separated spin-orbit components ($\Delta = 6.0$ eV). Small shifts observed in these binding energies, particularly in AMNP1, may have been associated with the presence of oxides. Finally, the atomic ratio variation between C, O, W, Ag, and Cu were investigated. The atomic ratio of samples AMNP0, AMNP1, and AMNP2 were also measured during the XPS analyses at etch

intervals of 0, 30, 60, 120, 200, and 240 s, and Figure 8c shows an illustrated sample of an atomic ratio diagram for the AMNP0. In summary, the overall atomic ratios were found and the average values were calculated using the data obtained from 30, 60, 120, 200, and 240 s intervals, as the sample surfaces at these intervals were stabilised under the conditions applied. The average atomic ratios for all three samples are given in Table 2.

Both Cu and Ag NP preparations, which are constituents of AMNP1 and AMNP2, have previously been shown to have good antimicrobial properties [56–58] and it is the presence of these compounds that is most likely to cause the antimicrobial effect seen in a previous study using these AMNPs incorporated into polymer mesh-like filters [33]. However, the exact content of these elements are not easy to determine in highly complex and heavily agglomerated nanopowders prepared using bulk forming methods.

3. Materials and Methods

3.1. Materials

Commercial Tungsten Carbide (>99.5%) with particle sizes <1 μm was purchased from Alfa Aesar (Lancashire, UK) and was used as a reference for a parallel study in this research. AMNP0, AMNP1, and AMNP2 were previously prepared by Qinetiq Nanomaterials[®] using patented Tesima[™] thermal plasma technology (Farnborough, UK), the generic details of this production method are reported elsewhere [1,31]. All materials were used as received unless indicated otherwise.

3.2. Scanning Electron Microscopy

AMNP0, AMNP1, and AMNP2 particles were assessed using Scanning Electron Microscopy (SEM). AMNP0 (3 mg) was secured onto a carbon based adhesive substrate and positioned on a specimen stage, while powder samples (3 mg) of AMNP1 and AMNP2 were dispersed in 10 mL of ethanol prior analysis. From each dispersion, 10 μL was placed on separate SEM pin stubs and left to air dry. The samples were sputter-coated with 20 nm of gold for 180 s under argon using a Quorum Q150T Turbo-Pumped Sputter Coater (Essex, UK). AMNP0 was imaged using a JEOL JSM-6301F instrument (Welwyn Garden City, UK), whilst AMNP1 and AMNP2 were imaged using a Quanta 200 FEG ESEM (OR, USA). All images were collected with an accelerating voltage of 5 kV.

3.3. Fourier Transform Infrared and Raman Spectroscopy

Infrared spectra were acquired using a PerkinElmer Frontier FT-IR/FIR spectrometer (Coventry, UK) equipped with an Attenuated Total Reflectance (ATR) accessory. Powder samples were loaded directly onto the diamond crystal stage and secured by a compressor rod. Blanks were performed prior to each sample submission, and all data were acquired at a resolution of 64 cm^{-1} using the built-in software 'IRWinLab' and 32 scans were collected.

Raman spectra were obtained by utilising a Renishaw InVia Raman microscope (Gloucestershire, UK) and associated WiRE 3.4 software supplied by the manufacturer. All measurements were performed by means of the 785 nm excitation wavelength and a 2 mW power laser. Powder samples were presented on a microscope slide with an approximate examined area of $20 \times 20\ \mu\text{m}^2$ and each measurement was taken after an average of 20 scans. The data were further analysed using the BioRed[®] (Philadelphia, PA, USA) program and all visible Raman shifts were studied against the references supported by the database within the program.

3.4. Solid State Nuclear Magnetic Resonance Spectroscopy

Carbon-13 analyses were recorded at 100.6 MHz using a Bruker Avance III NMR spectrometer (Coventry, UK) equipped with a magic-angle spinning probe, in which the samples (~250 mg) were loaded in a 5 mm rotor (o.d.). Data were obtained using cross-polarisation with a 2 s recycle delay, 3 millisecond contact time, at ambient probe temperature (~25 °C), and at a sample spin-rate of

10 kHz. Between 1000 and 1600 repetitions were accumulated. Spectral referencing was with respect to an external sample of neat tetramethylsilane (carried out by setting the high-frequency signal from adamantane to 38.5 ppm). The ^{13}C resonances apparent in the spectra represent the numbers of different chemical carbon environments emitted at specific frequencies and are measured as chemical shifts (δ) in 'part per million' (ppm). Data were analysed and processed using both Bruker Topspin (Coventry, UK) and Mestrec Nova (Santiago de Compostela, Spain).

3.5. X-ray Powder Diffraction

Multiple XRD analyses were performed on each sample using different instruments and methods, and the selected XRD results of commercial WC and AMNP0 were acquired on a Panalytical X'Pert Pro diffractometer (Panalytical, Almelo, The Netherlands). This instrument was equipped with an X'Celerator solid state detector where Cu $K\alpha$ was used as the radiation source with 0.25° divergence slits. The samples were mounted on zero background silicon single crystal substrates. Data were collected from 5 to 120° 2θ in steps of 0.033 degrees and a counting time of 200 s equivalent was used at each point. Whereas the selected XRD results of AMNP1 and AMNP2 were acquired from a Bruker D8 Advance X-ray diffractometer (Bruker AXS, Karlsruhe, Germany) using Cu $K\alpha$ radiation. Data were collected from 5 to 80° 2θ in steps of 0.04° and a counting time of 1 s equivalent was used at each point. All four data sets were analysed using the HighScore program (Panalytical B.V.2012 version 3.0.5, Almelo, The Netherlands) and all visible XRD patterns were supported by matching ICDD references.

3.6. Energy Dispersive X-ray Spectroscopy

Energy dispersive X-ray spectra of AMNP1 and AMNP2 were obtained using a JEOL field emission SEM JSM-7610F (Welwyn Garden City, UK) equipped with an Oxford Instruments (Oxfordshire, UK) 150 mm^2 XMax^N silicon drift detector for energy dispersive spectroscopy. Powder samples (2–3 mg) were secured on carbon adhesive substrates and were positioned on a Eucentric specimen stage (5 axes motor control) before submitting to the SEM chamber for analyses. Built-in JEOL control software was used for all acquisitions, data collections, and image processing. All data were collected with an accelerating voltage of 0.1–30 kV, a spot size of $70 \times 50\text{ mm}$, and at a working distance of 1–40 mm.

3.7. X-ray Photoelectron Spectroscopy

All powder samples (1–2 g) were moulded and pressed into disks ($2 \times 2\text{ mm}^2$) with a thickness of 1 mm. Each disk sample was mounted on a stage where XPS measurements were performed using a Thermo VG Scientific ESCALAB250 (Waltham, MA, USA) surface analysis system equipped with a 3 kV Argon ion sputtering device operated at 500 eV (beam $2\text{ }\mu\text{A}$, current 6×10^{-8} mbar) and at a sputter etching rate of 0.02 nm/s. The spectrometer used a monochromatic $\text{AlK}\alpha$ as the radiation source with photon energies at 1486.6 eV and a spot size of $500\text{ }\mu\text{m}$. The pass energy of 50 eV and energy step size of 0.1 eV were used in all experiments. Binding energies were recorded, and data were collected and analysed using the built-in software.

4. Conclusions

A combination of analytical methods including SEM, IR/Raman vibrational spectroscopy, ^{13}C -NMR, XRD, and XPS were used to investigate the structure and chemical components contained in three types of antimicrobial nanoparticles.

AMNP0 contains 99.8% single phase WC. As can be seen from both the FTIR/Raman spectra and the XRD analyses, no observable contaminant was detected in AMNP0 when compared with micro-metre sized commercial WC. This suggested that the production method (TesimaTM thermal plasma technology) used to manufacture these formulations have reliable quality control. To further support the chemical content found in AMNP0, XPS analyses have recorded C1s and W4f binding energies emitted by the carbidic bonding presence in AMNP0. XPS analysis also measured the atomic

ratio of the C1s:W4f content as 42.7:47.3%. It is also worth noting that a carbidic type of carbon signal (δ_C 254/307 ppm) associated with WC was detected using solid state NMR and it is the first report that has been published regarding this finding.

As for AMNP1 and AMNP2, these both contain similar atomic components (C, O, W, Cu, and Ag), but in different ratios. Different phases of tungsten components (W, WC, WC_{1-x}) were detected in AMNP1 and AMNP2, which may explain how each formulation exhibits different antimicrobial functions against specific bacteria/viruses. The high metallic ratio in AMNP2 was found to be five times more than that found in AMNP1 (17% vs. 3%) and can impart potential antimicrobial functions. The majority of the high carbon content found in both AMNP1 and AMNP2 (confirmed by Raman, NMR, and XPS analyses) may have been designed to increase bio-compatibilities and to reduce cytotoxicity effects.

The chemistries of these nano-powders, especially AMNP1 and AMNP2 produced using the thermal plasma process, were highly complex. Further analyses using more advanced characterisation methods (i.e., TEM and XRF) may allow a more in-depth understanding of these AMNP powders. In this aspect, identifications of any active sites and intermetallic properties present in the particles will provide insights for AMNP modifications and so to develop customised antimicrobial agents for selective pathogens. Extensive in vitro studies of these AMNP derivatives are being investigated, accumulating and collecting various analytical results from these nanopowders will help to understand the biological-chemical interactions between the surfaces of these particles and different microbes.

Acknowledgments: The authors would like to thank the EPSRC (EP/N034228/1) for funding this research, and we are also very grateful for the support of the EPSRC UK National Solid State NMR service in Durham University where the carbon analyses were obtained. We are very thankful to JEOL UK for access to the FE-SEM-EDX spectrometer.

Author Contributions: Y.-K.C. and G.R. conceived and designed the experiments; Y.-K.C., R.M.W., Q.K., J.C.-C., U.E.L., and R.M. performed the experiments; Y.-K.C., R.M.W., and J.C.-C. analyzed the data; G.R. contributed reagents/materials/analysis tools; Y.-K.C. wrote and revised the manuscript; G.R., L.C., S.M., E.C.-G., and M.E. reviewed the manuscript with major inputs contributed by M.E. All authors have read and approved the final manuscript.

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Ren, G.; Hu, D.; Cheng, E.W.; Vargas-Reus, M.A.; Reip, P.; Allaker, R.P. Characterisation of copper oxide nanoparticles for antimicrobial applications. *Int. J. Antimicrob. Agents* **2009**, *33*, 587–590. [[CrossRef](#)] [[PubMed](#)]
2. Jha, R.K.; Jha, P.K.; Chaudhury, K.; Rana, S.V.; Guha, S.K. An emerging interface between life science and nanotechnology: Present status and prospects of reproductive healthcare aided by nano-biotechnology. *Nano Rev.* **2014**, *5*, 22762. [[CrossRef](#)] [[PubMed](#)]
3. Pelgrift, R.Y.; Friedman, A.J. Nanotechnology as a therapeutic tool to combat microbial resistance. *Adv. Drug Deliv. Rev.* **2013**, *65*, 1803–1815. [[CrossRef](#)] [[PubMed](#)]
4. Bellare, J.R. Nanotechnology and nanomedicine for healthcare: Challenges in translating innovations from bench to bedside. *J. Biomed. Nanotechnol.* **2011**, *7*, 36–37. [[CrossRef](#)] [[PubMed](#)]
5. Zhang, L.; Pornpattananankul, D.; Hu, C.M.; Huang, C.M. Development of Nanoparticles for Antimicrobial Drug Delivery. *Curr. Med. Chem.* **2010**, *17*, 585–594. [[PubMed](#)]
6. Raffa, V.; Vittorio, O.; Riggio, C.; Cuschieri, A. Progress in nanotechnology for healthcare. *Minim. Invasive Ther. Allied Technol.* **2010**, *19*, 127–135. [[CrossRef](#)] [[PubMed](#)]
7. Munir, A.B.; Yasin, S.H. Nanotechnology in healthcare: Are existing laws adequate? *Eur. J. Health Law.* **2007**, *14*, 261–272. [[CrossRef](#)] [[PubMed](#)]
8. Venugopal, J.R.; Ramakrishna, S. Nanotechnology: 21st century revolution in restorative healthcare. *Nanomedicine* **2016**, *11*, 1511–1513. [[CrossRef](#)] [[PubMed](#)]
9. Singh, S.; Barick, K.C.; Bahadur, D. Surface engineered magnetic nanoparticles for removal of toxic metal ions and bacterial pathogens. *J. Hazard. Mater.* **2011**, *192*, 1539–1547. [[CrossRef](#)] [[PubMed](#)]

10. Salata, O. Applications of nanoparticles in biology and medicine. *J. Nanobiotechnol.* **2004**, *2*, 3. [[CrossRef](#)] [[PubMed](#)]
11. Huang, K.-S.; Yang, C.-H.; Huang, S.-L.; Chen, C.-Y.; Lu, Y.-Y.; Lin, Y.-S. Recent Advances in Antimicrobial Polymers: A Mini-Review. *Int. J. Mol. Sci.* **2016**, *17*, 1578. [[CrossRef](#)] [[PubMed](#)]
12. Sun, D.; Xu, D.; Yang, C.; Shahzad, M.B.; Sun, Z.; Xia, J.; Zhao, J.; Gu, T.; Yang, K.; Wang, G. An investigation of the antibacterial ability and cytotoxicity of a novel cu-bearing 317L stainless steel. *Sci. Rep.* **2016**, *6*, 29244. [[CrossRef](#)] [[PubMed](#)]
13. Fatoba, O.S.; Esezobor, D.E.; Akanji, O.L.; Fatoba, A.J.; Macgregor, D.; Etubor, J. The Study of the Antimicrobial Properties of Selected Engineering Materials' Surfaces. *J. Miner. Mater. Charact. Eng.* **2014**, *2*, 78–87. [[CrossRef](#)]
14. Sun, D.; Shahzad, M.B.; Li, M.; Wang, G.; Xu, D. Antimicrobial materials with medical applications. *Mater. Technol. Adv. Biomater.* **2015**, *30*, B90–B95. [[CrossRef](#)]
15. Devasconcellos, P.; Bose, S.; Beyenal, H.; Bandyopadhyay, A.; Zirkle, L.G. Antimicrobial Particulate Silver Coatings on Stainless Steel Implants for Fracture Management. *Mater. Sci. Eng. C Mater. Biol. Appl.* **2012**, *32*, 1112–1120. [[CrossRef](#)] [[PubMed](#)]
16. Aitken, R.J.; Chaudhry, M.Q.; Boxall, A.B.; Hull, M. Manufacture and use of nanomaterials: Current status in the UK and global trends. *Occup. Med. (Lond.)* **2006**, *56*, 300–306. [[CrossRef](#)] [[PubMed](#)]
17. Cloutman-Green, E.; Canales, M.; Zhou, Q.; Ciric, L.; Hartley, J.C.; McDonnell, G. Biochemical and microbial contamination of surgical devices: A quantitative analysis. *Am. J. Infect. Control* **2015**, *43*, 659–661. [[CrossRef](#)] [[PubMed](#)]
18. Ciric, L.; Brouwer, M.S.; Mullany, P.; Roberts, A.P. Minocycline resistance in an oral *Streptococcus infantis* isolate is encoded by tet(S) on a novel small, low copy number plasmid. *FEMS Microbiol. Lett.* **2014**, *353*, 106–115. [[CrossRef](#)] [[PubMed](#)]
19. Beyth, N.; Hourri-Haddad, Y.; Domb, A.; Khan, W.; Hazan, R. Alternative Antimicrobial Approach: Nano-Antimicrobial Materials. *Evid.-Based Complement. Altern. Med.* **2015**, *2015*, 1–16. [[CrossRef](#)] [[PubMed](#)]
20. Vargas-Reus, M.A.; Memarzadeh, K.; Huang, J.; Ren, G.G.; Allaker, R.P. Antimicrobial activity of nanoparticulate metal oxides against *peri-implantitis* pathogens. *Int. J. Antimicrob. Agents* **2012**, *40*, 135–139. [[CrossRef](#)] [[PubMed](#)]
21. Liu, Y.; Guan, W.; Ren, G.; Yang, Z. The possible mechanism of silver nanoparticle impact on hippocampal synaptic plasticity and spatial cognition in rats. *Toxicol. Lett.* **2012**, *209*, 227–231. [[CrossRef](#)] [[PubMed](#)]
22. Han, Y.; Xu, J.; Li, Z.; Ren, G.; Yang, Z. In vitro toxicity of multi walled carbon nanotubes in C6 rat glioma cells. *NeuroToxicol.* **2012**, *33*, 1128–1134. [[CrossRef](#)] [[PubMed](#)]
23. Yang, Z.; Liu, Z.W.; Allaker, R.P.; Reip, P.; Oxford, J.; Ahmad, Z.; Ren, G. A review of nanoparticle functionality and toxicity on the central nervous system. *J. R. Soc. Interface* **2010**, *7* (Suppl. 4), S411–S422. [[CrossRef](#)] [[PubMed](#)]
24. Liu, Z.; Liu, S.; Ren, G.; Zhang, T.; Yang, Z. Nano-CuO inhibited voltage-gated sodium current of hippocampal CA1 neurons via reactive oxygen species but independent from G-proteins pathway. *J. Appl. Toxicol.* **2011**, *31*, 439–445. [[CrossRef](#)] [[PubMed](#)]
25. Liu, Z.; Ren, G.; Zhang, T.; Yang, Z. The inhibitory effects of nano-Ag on voltage-gated potassium currents of hippocampal CA1 neurons. *Environ. Toxicol.* **2011**, *26*, 552–558. [[CrossRef](#)] [[PubMed](#)]
26. Zhao, J.; Xu, L.; Zhang, T.; Ren, G.; Yang, Z. Influences of nanoparticle zinc oxide on acutely isolated rat hippocampal CA3 pyramidal neurons. *Neurotoxicology* **2009**, *30*, 220–230. [[CrossRef](#)] [[PubMed](#)]
27. Chen, T.; Yang, J.; Ren, G.; Yang, Z.; Zhang, T. Multi-walled carbon nanotube increases the excitability of hippocampal CA1 neurons through inhibition of potassium channels in rat's brain slices. *Toxicol. Lett.* **2013**, *217*, 121–128. [[CrossRef](#)] [[PubMed](#)]
28. Chen, T.; Yang, J.; Zhang, H.; Ren, G.; Yang, Z. Multi-walled carbon nanotube inhibits CA1 glutamatergic synaptic transmission in rat's hippocampal slices. *Toxicol. Lett.* **2014**, *229*, 423–429. [[CrossRef](#)] [[PubMed](#)]
29. Han, Y.; Ren, L.; Xu, K.; Yang, F.; Li, Y.; Cheng, T.; Kang, X.; Xu, C.; Shi, Q. Supercritical fluid extraction with carbon nanotubes as a solid collection trap for the analysis of polycyclic aromatic hydrocarbons and their derivatives. *J. Chromatogr. A* **2015**, *1395*, 1–6. [[CrossRef](#)] [[PubMed](#)]
30. Allaker, R.P.; Ren, G. Potential impact of nanotechnology on the control of infectious diseases. *Trans. R. Soc. Trop. Med. Hyg.* **2008**, *102*, 1–2. [[CrossRef](#)] [[PubMed](#)]

31. Ren, G.; Oxford, J.S.; Reip, P.W.; Lambkin-Williams, R. Anti-Viral Formulations Nanomaterials and Nanoparticles. U.S. Patent 2013/0091611 A1, 18 April 2013.
32. Mahalingam, S.; Edirisinghe, M. Forming of polymer nanofibers by a pressurised gyration process. *Macromol. Rapid Commun.* **2013**, *34*, 1134–1139. [[CrossRef](#)] [[PubMed](#)]
33. Illangakoon, U.E.; Mahalingam, S.; Wang, K.; Cheong, Y.K.; Canales, E.; Ren, G.G.; Cloutman-Green, E.; Edirisinghe, M.; Ciric, L. Gyrospun antimicrobial nanoparticle loaded fibrous polymeric filters. *Mater. Sci. Eng. C Mater. Biol. Appl.* **2017**, *74*, 315–324. [[CrossRef](#)] [[PubMed](#)]
34. Zhang, C.; Liang, Z.; Hu, Z. Bacterial response to a continuous long-term exposure of silver nanoparticles at sub-ppm silver concentrations in a membrane bioreactor activated sludge system. *Water Res.* **2014**, *50*, 350–358. [[CrossRef](#)] [[PubMed](#)]
35. Matejka, V.; Tokarsky, J. Photocatalytical nanocomposites: A review. *J. Nanosci. Nanotechnol.* **2014**, *14*, 1597–1616. [[CrossRef](#)] [[PubMed](#)]
36. Beyth, N.; Yudovin-Farber, I.; Perez-Davidi, M.; Domb, A.J.; Weiss, E.I. Polyethyleneimine nanoparticles incorporated into resin composite cause cell death and trigger biofilm stress in vivo. *Proc. Natl. Acad. Sci. USA* **2010**, *107*, 22038–22043. [[CrossRef](#)] [[PubMed](#)]
37. Barron, A.R.; Allen, J.J.; Bott, S.; Jebb, M.; Jiang, C.; Kienast, W. *Physical Methods in Chemistry and Nano Science*; CONNEXIONS Rice University: Houston, TX, USA, 2012.
38. Yan, Y.; Xia, B.; Qi, X.; Wang, H.; Xu, R.; Wang, J.-Y.; Zhang, H.; Wang, X. Nano-tungsten carbide decorated graphene as co-catalysts for enhanced hydrogen evolution on molybdenum disulfide. *Chem. Commun.* **2013**, *49*, 4884–4886. [[CrossRef](#)] [[PubMed](#)]
39. Krasovskii, P.V.; Malinovskaya, O.S.; Samokhin, A.V.; Blagoveshchenskiy, Y.V.; Kazakov, V.A.; Ashmarin, A.A. XPS study of surface chemistry of tungsten carbides nanopowders produced through DC thermal plasma/hydrogen annealing process. *Appl. Surf. Sci.* **2015**, *339*, 46–54. [[CrossRef](#)]
40. Liu, C.; Zhou, D.; Zhou, J.; Xie, Z.; Xia, Y. Synthesis and characterization of tungsten carbide and application to electrocatalytic hydrogen evolution. *RSC Adv.* **2016**, *6*, 76307–76311. [[CrossRef](#)]
41. Yamazaki, Y.; Nakajima, K.; Wakahara, T.; Tsuchiya, T.; Ishitsuka, M.O.; Maeda, Y.; Akasaka, T.; Waelchli, M.; Mizorogi, N.; Nagase, S. Observation of ¹³C NMR chemical shifts of metal carbides encapsulated in fullerenes: Sc₂C₂@C₈₂, Sc₂C₂@C₈₄, and Sc₃C₂@C₈₀. *Angew. Chem. Int. Ed. Engl.* **2008**, *47*, 7905–7908. [[CrossRef](#)] [[PubMed](#)]
42. Zhang, J.; Fuhrer, T.; Fu, W.; Ge, J.; Bearden, D.W.; Dallas, J.; Duchamp, J.; Walker, K.; Champion, H.; Azurmendi, H.; et al. Nanoscale Fullerene Compression of an Yttrium Carbide Cluster. *J. Am. Chem. Soc.* **2012**, *134*, 8487–8493. [[CrossRef](#)] [[PubMed](#)]
43. Buss, J.A.; Agapie, T. Four-electron deoxygenative reductive coupling of carbon monoxide at a single metal site. *Nature* **2015**, *529*, 72–75. [[CrossRef](#)] [[PubMed](#)]
44. Kurlov, A.S.; Gusev, A.I. Oxidation of tungsten carbide powders in air. *Int. J. Refract. Met. Hard Mater.* **2013**, *41*, 300–307. [[CrossRef](#)]
45. Kalikhman, V.L. *Inorg. Mater. (Engl. Transl.)* **1983**, *19*, 957. A Primary Reference Source Provided by the International Centre for Diffraction Data (ICDD 04-004-57230). Available online: <https://link.springer.com/journal/10789> (accessed on 21 June 2017).
46. Lindqvist, I. The Crystal structure of the yellow Molybdic acid Mo₃.2H₂O. On the existence of the H₄O²⁺ ion. *Acta Chem. Scand.* **1950**, *4*, 650–657. [[CrossRef](#)]
47. Metcalfe, A.G. *J. Inst. Met.* **1947**, *73*, 591. A Primary Reference Source Provided by the International Centre for Diffraction Data (ICDD 00-051-0939). Available online: <https://catalog.hathitrust.org/Record/008893915> (accessed on 21 June 2017).
48. Ren, X.; Peng, Z.; Peng, Y.; Wang, C.; Fu, Z.; Qi, L.; Miao, H. Ultrafine binderless WC-based cemented carbides with varied amounts of AlN nano-powder fabricated by spark plasma sintering. *Int. J. Refract. Met. Hard Mater.* **2013**, *41*, 308–314. [[CrossRef](#)]
49. Sara, R.V. Phase Equilibria in the System Tungsten—Carbon. *J. Am. Ceram. Soc.* **1965**, *48*, 251–257. [[CrossRef](#)]
50. Katagiri, A.; Suzuki, M.; Takehara, Z. Electrodeposition of Tungsten in ZnBr₂-NaBr and ZnCl₂-NaCl Melts. *J. Electrochem. Soc.* **1991**, *138*, 767–773. [[CrossRef](#)]
51. Predel, B.; Bankstahl, H. Metastabile Phasen im System Silber-Wismut. *Z. Metallkd.* **1976**, *67*, 793–799.
52. Schmahl, N.G.; Eikerling, G.F.Z. Über Kryptomodifikationen des Cu(II)-Oxids. *Phys. Chem. Neue Folge. (Wiesbaden)* **1968**, *62*, 268–279. [[CrossRef](#)]

53. Briggs, D. Handbook of X-ray Photoelectron Spectroscopy C.D. Wanger, W.M. Riggs, L.E. Davis, J.F. Moulder and G.E. Muilenberg Perkin-Elmer Corp., Physical Electronics Division, Eden Prairie, Minnesota, USA, 1979. 190 pp. 195. *Surf. Interface Anal.* **1981**, *3*. [[CrossRef](#)]
54. Susi, T.; Kaukonen, M.; Havu, P.; Ljungberg, M.P.; Ayala, P.; Kauppinen, E.I. Core level binding energies of functionalized and defective graphene. *Beilstein J. Nanotechnol.* **2014**, *5*, 121–132. [[CrossRef](#)] [[PubMed](#)]
55. Susi, T.; Pichler, T.; Ayala, P. X-ray photoelectron spectroscopy of graphitic carbon nanomaterials doped with heteroatoms. *Beilstein J. Nanotechnol.* **2015**, *6*, 177–192. [[CrossRef](#)] [[PubMed](#)]
56. Syed, M.A.; Babar, S.; Bhatti, A.S.; Bokhari, H. Antibacterial Effects of Silver Nanoparticles on the Bacterial Strains Isolated from Catheterized Urinary Tract Infection Cases. *J. Biomed. Nanotechnol.* **2009**, *5*, 209–214. [[CrossRef](#)] [[PubMed](#)]
57. Nan, L.; Ren, G.; Wang, D.; Yang, K. Antibacterial Performance of Cu-Bearing Stainless Steel against *Staphylococcus aureus* and *Pseudomonas aeruginosa* in Whole Milk. *J. Mater. Sci. Technol.* **2016**, *32*, 445–451. [[CrossRef](#)]
58. Xu, Z.; Mahalingam, S.; Rohn, J.L.; Ren, G.; Edirisinghe, M. Physio-chemical and antibacterial characteristics of pressure spun nylon nanofibres embedded with functional silver nanoparticles. *Mater. Sci. Eng. C Mater. Biol. Appl.* **2015**, *56*, 195–204. [[CrossRef](#)] [[PubMed](#)]



© 2017 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<http://creativecommons.org/licenses/by/4.0/>).