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Preparation, Physicochemical Properties, and Long-Term Performance of Photocatalytic Ceramsite Sand in Cementitious Materials

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Featured Application: Photocatalytic construction.

Abstract: Incorporation of TiO₂ into cementitious materials is an important technology in the field of photocatalytic pollution mitigation; however, the photocatalytic activity of TiO₂ is limited by specific surface area, poor gas diffusion and light transmission performance of cementitious materials. In this study, a novel photocatalytic lightweight aggregate—photocatalytic ceramsite sand (PCS) was synthesized by loading TiO₂ on activated porous ceramsite sand (CS) with negative pressure method to solve problems in application of photocatalysts in cementitious materials. Photocatalytic cement material (PCM) was prepared by loading PCS on the surface of cementitious materials, which improved the photocatalytic activity and efficiency of TiO₂ in cementitious materials. It was found that the pore structure (pore volume, size distribution and interconnectivity) of ceramsite sand (CS) varies with particle size. The photocatalytic removal rate of benzene on PCS increased significantly through adjusting ceramsite sands in appropriate pore structure and TiO₂ at best coating ratio. The photocatalytic activity of PCS slightly decreased but still remained active after incorporated into concrete. 2 μL benzene was degraded completely in 200 min by 5 g 4PCS-1.25~2.35 and 300 min by PCM-5, and was still degraded over 80% in 400 min by PCM-5 after exposure to natural environment for 6 months. The results suggested that the photocatalytic activity of TiO₂ in cementitious materials was enhanced by the preparation of PCS and PCM, which could provide more gas diffusion, higher specific surface area, more TiO₂ active sites, and prevent TiO₂ particles from being influenced by the envelope of cement hydration products and the carbonation of cement.

Keywords: photocatalytic ceramsite sand; photocatalytic cementitious material; pore structure; TiO₂; photocatalytic activity

1. Introduction

In recent years, there is an increasing interest in applying photocatalysts in cementitious materials to eliminate urban air pollutants [1–5]. However, cementitious materials have many disadvantages, such as complex constituent, low specific surface area, surface carbonation and poor light transmittance, etc. Previous studies [6–8] showed that the photocatalytic activity of photocatalysts, especially the long-term photocatalytic performance, decreased obviously after being applied in cementitious materials, whether TiO₂ was coated on the surface or mixed with the substrates. This phenomenon was attributed to the surface carbonation of gas–solid interface, the peeling of photocatalyst layer

and the influence of ion species (Ca^{2+} , Na^+ , OH^- , etc.) [9]. Many studies have attempted to improve the photocatalytic activity of photocatalysts in cementitious materials. Janus et al. [10] prepared cement pastes containing nitrogen and carbon co-modified TiO_2 ($\text{TiO}_2\text{-N,C}$) to enhance photocatalytic activity of photocatalysts in cementitious materials. Vaish et al. [11] presented visible light active photocatalytic filler based on BaTiO_3 and reduced graphene oxide immobilized in Portland cement could be readily reinforced the photocatalytic performance for xanthene dye degradation. Moreover, Lee et al. [12] evaluated the influence of water-to-cement ratio (w/c) on photocatalytic efficiency in the presence of NO and NO_2 . It was found that w/c of the material altered the pore structure of cementitious materials [13], the amount of effective surface area available for photocatalytic oxidation and the binding of oxidation products [14,15]. Generally, there are two ways to improve the photocatalytic activity of photocatalytic cementitious materials effectively: increasing the absorption of air pollutant and reducing the coverage of photocatalysts by cementitious materials or hydration products [16]. Therefore, some methods such as introducing the high porosity expanded shale into concrete [17], controlling the pore structure of cement pastes [18] and regulating the microstructure of mortar cement [19] have been developed to enhance the photocatalytic activity of photocatalytic cementitious materials.

Aggregates are important components of cement concrete materials, which have potential to act as the substrate of photocatalysts due to the stable composition and structure. Among many aggregates, ceramsite is a promising aggregate to load photocatalysts such as TiO_2 because of physicochemical stability and porous structure. It has been applied in exposed aggregate concrete to build concrete pavement and sidewalk for sound absorption and noise reduction in replace of traditional aggregate. In addition, the exposed aggregate concrete exhibited better surface strength and abrasion performance compared with ordinary concrete [20,21]. However, the functional ceramsites with large particle size brought some difficulties in construction when being loaded on the surface of cement concrete. Therefore, ceramsites were crushed into ceramsite sands (CS), which could facilely being used in photocatalytic cement material (PCM) to increase the reaction areas of photocatalysis and improve the stability of TiO_2 as the intermediate.

In this paper, photocatalytic ceramsite sand (PCS) was prepared by negative pressure method and then loaded on the surface of cement pastes at the initial set. The photocatalytic activity was improved significantly by using CS as intermediate of TiO_2 and cementitious materials. Comparing with cementitious materials, PCS could not only prevent the aggregation of TiO_2 particles but also provide more active sites, high specific surface areas and more gas diffusion. The preparation of PCS increased the reacting areas and utilization rate of TiO_2 in cement concrete constructions compared with coating on the surface or mixing with substrates. Therefore benzene was chosen as target contaminant in experiments which was rarely selected in other studies because of its difficulty in degradation. As we all know, benzene is an organic solvent which is very stable compared with other organic or air contaminants. It is rather hard to degrade especially for photocatalysts applied in cementitious materials. In this study, 2 μL benzene was degraded completely in 300 min by PCM in a closed cylindrical stainless steel gas-phase reactor (5.56 L), which proved extraordinary photocatalytic property of PCM. The photocatalytic activity of specimens in simulation conditions were studied in detail.

In addition, X-ray diffraction (XRD), scanning electron microscope (SEM), energy dispersive spectrometer (EDS), X-ray photoelectron spectroscopy (XPS) and gas chromatography (GC) were conducted to comprehensively characterize the photocatalytic materials and mechanisms governing the catalytic process.

2. Materials and Methods

2.1. Materials

CS was obtained from Yi Chang and sieved to three levels of particle sizes (0.1~0.6 mm, 0.6~1.25 mm and 1.25~2.35 mm). The physical characteristics and the chemical composition are listed in Tables 1 and 2, respectively. ASTM type I was purchased from Huaxin Cement Co., Ltd. (Wuhan, China). Photocatalysts applied in this study, TiO₂ (25% rutile and 75% anatase) is Degussa P25, which has a crystal particle size of 25 and 40 nm, respectively. Sodium hydroxide (NaOH) and absolute alcohol (C₂H₅OH) were purchased from Shenshi Chem (Shanghai, China). All chemicals applied in this study were analytical grade.

Table 1. The physical characteristics of CS.

Sample	Density Grade	Bulk Density (kg·m ⁻³)	1 h Water Absorption (%)	Total Porosity (%)
Ceramsite sand	900	1100	5.6	43.57

Table 2. The chemical composition of CS.

SiO ₂ /%	Al ₂ O ₃ /%	Fe ₂ O ₃ /%	CaO/%	MgO/%	K ₂ O/%	N ₂ O/%	TiO ₂ /%
64.903	18.346	7.231	0.618	2.397	3.781	1.1	0.946

2.2. Methods

2.2.1. Preparation of Photocatalytic Ceramsite Sand

Ceramsite sand (CS) was activated by being immersed into 0.1 mol/L NaOH aqueous solution with sonication for 30 min to get more active sites. The other method for activating CS was to immerse CS into 5 mol/L NaOH at 85 °C in a reflux device for 4 h. Then, the activated CSs were put into vacuum saturation container at 0.1 MPa vacuum for 30 min to discharge the air in pores of CS. TiO₂ (P25) was firstly dispersed in 100 mL absolute alcohol with magnetic stirring in various concentrations. After 30 min of vacuum pumping and magnetic stirring, TiO₂ absolute alcohol dispersion solution was inhaled into the vacuum container with ultrasonic treatment for 1 h. Then, photocatalytic ceramsite sand (PCS) precursor was taken out, washed by distilled water, and dried at 105 °C for 3 h. PCS was thus successfully prepared. The concentrations of TiO₂ absolute alcohol dispersion solution were designed as 1, 2, 3, 4 and 5 g/L, respectively. The samples were denoted as mPCS-n, where m was the coating concentration of TiO₂ absolute alcohol dispersion (m = 1, 2, 3, 4, 5 g/L), and n was the particle size of CS (n = 0.1~0.6, 0.6~1.25, 1.25~2.35 mm). To compare PCM in this study with traditional photocatalytic cementitious materials, control groups were prepared in traditional ways by spraying TiO₂ on the surface or incorporating into cement pastes with same amount of photocatalysts (0.1 g) and same dimension compared with other samples. The photocatalytic activity of pure TiO₂ (P25) was also tested. The experimental devices and procedure are shown in Figure 1.

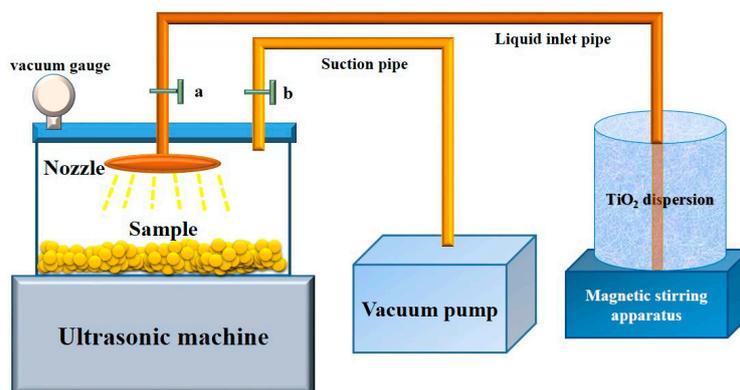


Figure 1. Diagram of the vacuum-ultrasonic-negative pressure devices.

2.2.2. Preparation of Photocatalytic Cement Materials

The photocatalytic cement materials (PCM) were prepared with 4PCS-1.25~2.35, cement and water. First, cement and water (w/c is 0.35) was mixed for 3 min in cement mixer. After the cement pastes were cast into a petri dish (diameter 90 mm) for 3 h, 1, 3, 5 g 4PCS-1.25~2.35 was sprayed on the top surface of the cement pastes, respectively. PCS was thus exposed to air. After 3 days of hydration, PCM was successfully prepared. The samples were denoted as PCM- x , where x is the weight of 4PCS-1.25~2.35 ($x = 1, 3, 5$ g).

2.3. Characterization

The crystalline compositions of PCS were identified by XRD (MAX-RB RU-200B) (Rigaku, Tokyo, Japan) at the scanning speed of 8° min^{-1} and scanning range from 10° to 80° [17]. The scanning electron microscopy (SEM) images were obtained through SEM (Quanta FEG 450, FEI) analyzer (FEI, Hillsboro, OR, USA), together with Genesis EDS for compositional analysis. Chemical valence states of PCS were investigated by X-ray photoelectron spectroscopy (XPS), which was performed on a Thermo Fisher ESCALAB 250Xi instrument (Thermo Fisher Scientific, Waltham, MA, USA) with a monochromatic Al K Alpha (1486.68 eV) X-ray source. The apparent morphology of CS, PCS and PCM were observed by digital camera. The specific surface area was measured from N₂ adsorption isotherms by Brunauer-Emmett-Teller surface area measurement (BET, ASAP 2020M, Atlanta, GA, USA). The pore size distribution, available pore volume and total porosity of CS and PCS were determined by Mercury intrusion porosimetry (Pore Master 60, Quantachrome, Boynton Beach, FL, USA). The working pressure was 3.45 kPa~413 MPa, allowing mercury intrusion in pores with diameters 1 nm~350 μm .

2.4. Evaluation of Photocatalytic Activity

The photocatalytic activity was characterized in mineralization of benzene. The source of UV light was a high pressure Hg lamp (125 W, $\lambda_{\text{nm}} > 340$ nm, light spectrum around 365, 400, 440, 550 and 580 nm), which was obtained from Shanghai Yaming, China. There was a 15 cm distance between the lamp and sample. In these tests, the UV light region on the surface of sample was 320~400 nm and the UV intensity was 0.96 mW/cm^2 .

Samples were put in a closed cylindrical stainless steel gas-phase reactor (5.6 L) with a quartz window. Gaseous environment was synthetic air and humidity was indoor humidity. For each test, liquid benzene (2 μL) was injected and evaporated in the reactor. After complete evaporation of benzene in the reactor (shown by concentration in gas chromatograph), a sample of PCM or 5 g PCS was started to be irradiated by the UV lamp. Then, the concentration of benzene in the photocatalytic progress was measured every 15 min by gas chromatograph (GC9560, Shanghai, Huaai, China).

The PCSs with coating concentrations of TiO_2 1~5 g/L were used in recycled experiments. The approach is as follows: after the first photocatalytic activity test, the sample was irradiated under a UV lamp for 1 h with the quartz window open to ensure the pollutants absorbed on the surface of sample removed completely. Then, seal the reactor again and start the next test. PCM-5 was applied in long-term photocatalytic effect test. The sample was placed on the roof (State Key laboratory of Silicate Materials for Architecture, Wuhan University of Technology, Luoshi Road, Wuhan, Hubei, China) with the face up for 6 months (September 2015 to March 2016). The outdoor temperature was $-6\sim 38\text{ }^\circ\text{C}$ and the relative humidity conditions range from 35 to 90%. The month average rainfall was 118.9 mm. After 6 months of atmospheric environment exposure, the photocatalytic activity of PCM-5~6 months was evaluated under same experimental conditions with other samples. The control group was also tested after 1 month of atmospheric environment exposure.

3. Results and Discussion

3.1. Physical and Chemical Properties

Figure 2 showed the surface morphology of CS and PCS with different particle sizes. The samples were denoted as CS-n and mPCS-n, where m was the coating concentration of TiO_2 absolute alcohol dispersion ($m = 1, 4\text{ g/L}$), n was the particle size of CS ($n = 0.1\sim 0.6, 0.6\sim 1.25, 1.25\sim 2.35\text{ mm}$). No apparent color differences were observed between CS-0.1~0.6 and 1PCS-0.1~0.6. In addition, 1PCS-0.6~1.25 was a little whiter than samples without coating. Nevertheless, all PCS of the particle size 1.25~2.35 mm were obviously whiter compared with the control samples without coating, which was consistent with Yang's study [17]. The change of color was attributed to the light absorption and reflection of TiO_2 particles [22]. As we can see, PCS with TiO_2 dispersion solution concentrations 4 g/L were significantly whiter than that of 1 g/L, which was related to more TiO_2 coating areas and thicker TiO_2 coating layer. The surface topography of PCM with 1, 3, 5 g PCS loaded and interface between PCS and matrix are shown in Figure 2 as well.

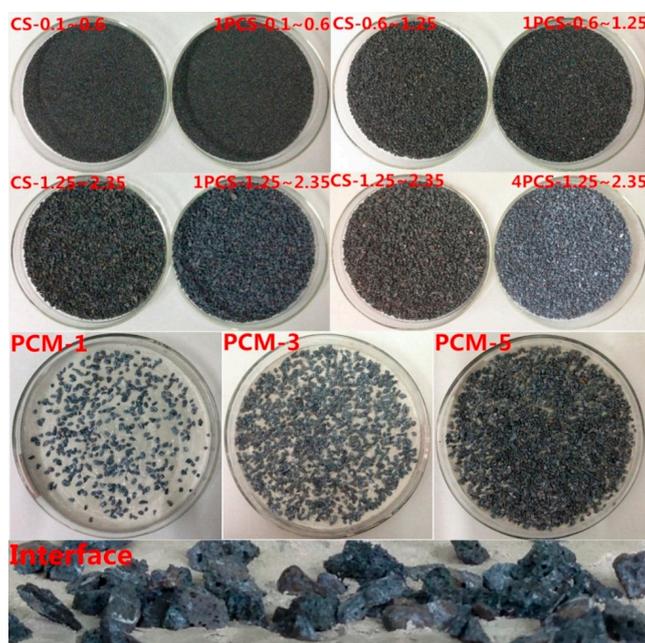


Figure 2. Surface morphology of photocatalytic ceramsite sand and photocatalytic cement materials and the interface of PCS and cement; CS-n and mPCS-n (m was the coating concentration of TiO_2 absolute alcohol dispersion, $m = 1\sim 4\text{ g/L}$, n was the particle size of CS, $n = 0.1\sim 0.6, 0.6\sim 1.25, 1.25\sim 2.35\text{ mm}$); PCM-1, 3, 5 (photocatalytic cement materials with 1, 3, 5 g PCS); PCS: photocatalytic ceramsite sand; CS: ceramsite sand; PCM: Photocatalytic cement material.

XRD patterns were applied to explore the crystalline structure of samples. Figure 3 showed the XRD patterns of CS, TiO₂ (P25) and 1PCS-1.25~2.35 with ultrasonic activation (1PCS-1.25~2.35-UA) and alkali activated (1PCS-1.25~2.35-AC). It can be seen that the main components of CS were SiO₂ and Mg_{0.7}Fe_{0.23}Al_{1.97}O₄. Comparing with the XRD pattern of TiO₂, the characteristic peaks of 1PCS-1.25~2.35-UA occurred at about 2θ = 27° and 44.3° can be assigned to the Rutile [23]. The characteristic peaks at 2θ = 25.3° can be assigned to the Anatase. The peak of TiO₂ could also be found in 1PCS-1.25~2.35-AC, but weaker than 1PCS-1.25~2.35-UA. The XRD suggested that TiO₂ particles were successfully coated on the surface of CS and ultrasonic activation could get more active sites for combination of CS and TiO₂.

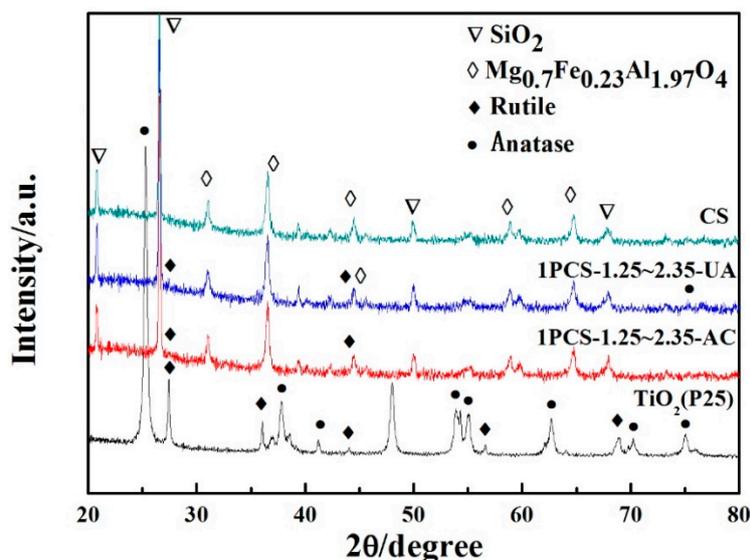


Figure 3. X-ray diffraction patterns of CS, TiO₂ (P25), 1PCS-1.25~2.35 with ultrasonic activation (1PCS-1.25~2.35-UA) and alkali activated (1PCS-1.25~2.35-AC).

SEM images in Figure 4 presented the morphological structure of CS, 4PCS-1.25~2.35, and 1PCS-1.25~2.35, respectively. From Figure 4a,b, it can be seen that the pore size of CS mainly ranged from 5 to 50 μm and there are some small pores under 1 μm. Some inter-connecting pores were also observed clearly, which could facilitate the coating of TiO₂ and the absorption of gas pollutants. Figure 4c,d showed that TiO₂ was coated on the surface of CS successfully. It can be found TiO₂ was layered homogeneously on the surface of CS and there were abundant small holes among TiO₂ layers, which could contribute to the absorption of gas pollutants. There are three factors to form those structures, including silica-based CS, ultrasonic activation and negative pressure preparation method. These factors were helpful to produce hydroxyl groups on the surface of CS and therefore lead to the formation of Ti-O-Si bonds and the accumulation of TiO₂ particles [24,25].

TiO₂ particles on the surface of 1PCS-1.25~2.35 were observed in Figure 4e, which were seen as bright spots. It was confirmed that TiO₂ particles loaded on the surface of CS agglomerated rarely. The observation in Figure 4e was in agreement with the results of EDS in Figure 4f,g. Element Ti could be identified at Area 1 where many bright spots could be observed. Meanwhile, Area 2 contained no Ti had no bright spot. The result of EDS showed that TiO₂ had deposited on the surface of PCS successfully.

BET was applied to explore the specific surface area of samples. As shown in Table 3, the specific surface area of CS and mortars are 0.0770 m²/g and 0.0013 m²/g, respectively. The specific surface area of CS is much higher than that of mortars.

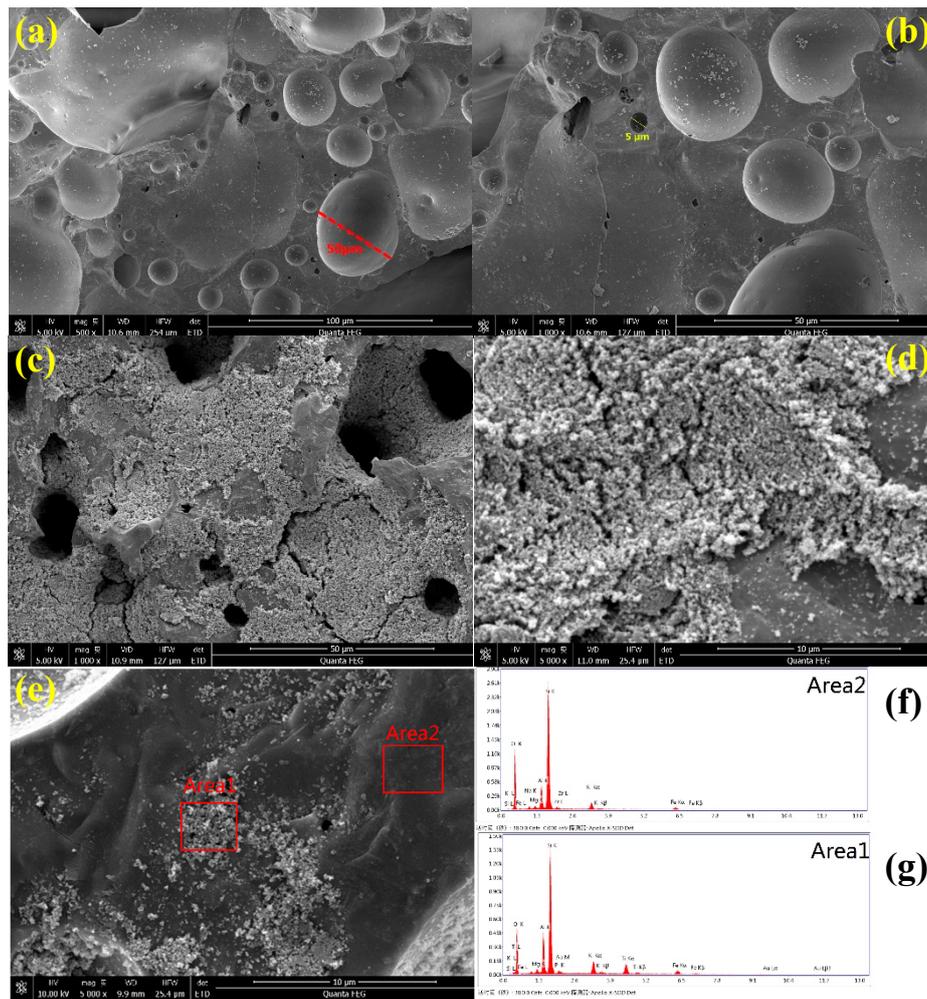


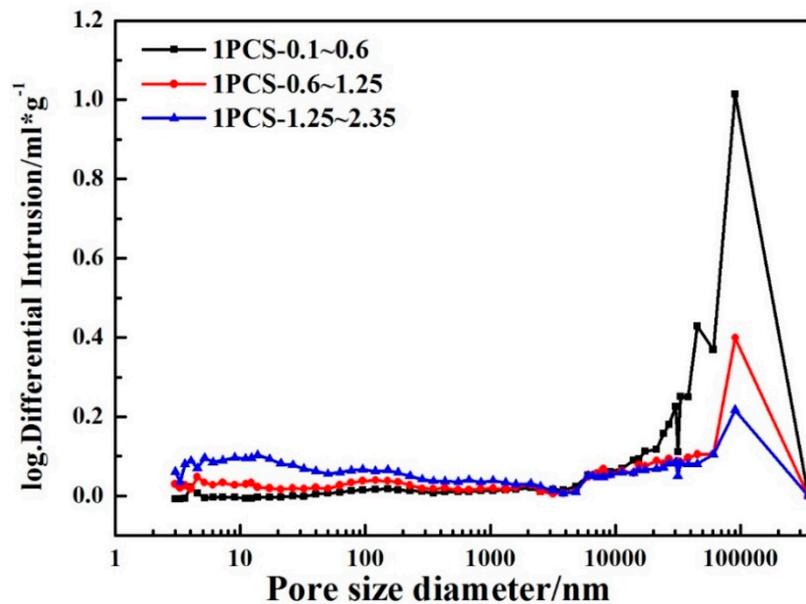
Figure 4. Surface morphology and EDS of CS and PCS; (a,b) surface morphology of CS-1.25~2.35; (c,d) surface morphology of PCS (4PCS-1.25~2.35); (e–g) surface morphology and EDS patterns of PCS (1PCS-1.25~2.35); EDS: energy dispersive spectrometer.

Table 3. BET results of CS and mortars.

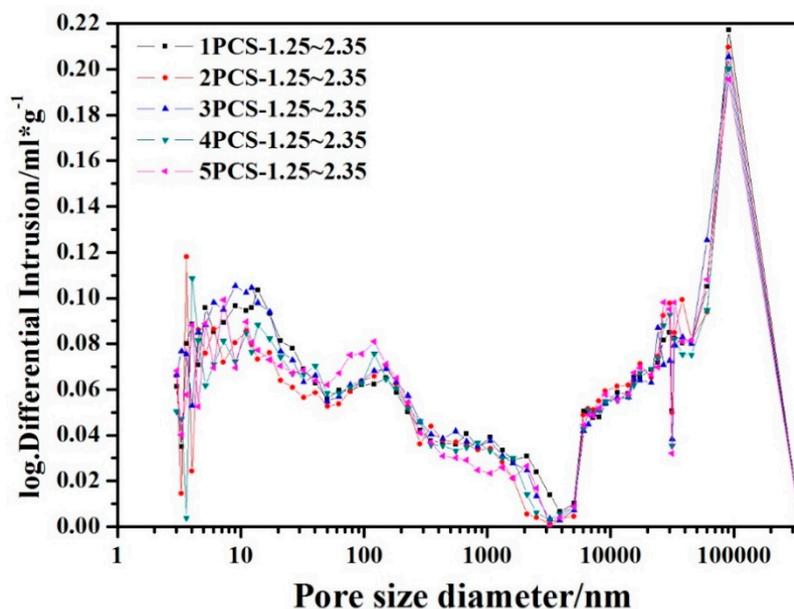
CS (m ² /g)	Mortars (m ² /g)
0.0770	0.0013

Figure 5 and Table 4 showed the pore size distribution, total porosity and available pore volume for all PCS samples. Figure 5a revealed that PCS exhibited a pore size distribution mostly in the range of 1~100 µm, with a small amount of pores from 10 to 1000 nm. The pores distributed in the range of 10~1000 nm increased with the increase of the PCS particle size while the pores distributed in the range of 1~100 µm decreased with the increase of the PCS particle size. In other words, when CS was crushed into smaller pieces, many small pores of CS were lost but the amount of larger pores increased. This might be because some small pores at section were destroyed through the crushing, while some larger closed pores turned to open. Therefore the CS with small particle size had higher total porosity and pore volume than larger ones (as shown in Table 4). More big pores, higher porosity and pore volume might cause the waste of TiO₂ particles, because the TiO₂ particles at the bottom of the pores could not accept the light. In addition, big pores were nearly useless to gas diffusion compared with small ones. Figure 5b showed the similar pore distribution curves for the samples prepared with different TiO₂ coating concentrations. The pore volumes of PCS with different TiO₂

coating concentrations had few changes, corresponding to the results of Figure 5b. With the increasing of TiO₂ coating concentrations, the pores between 90 μm and 100 μm reduced slightly while the pores in the range of 1~1000 nm increased. This phenomenon can be attributed to the formation of TiO₂ layer provided many small pores and reduced the pore size of some big pores. According to the above results, it can be inferred that PCS-1.25~2.35 might possess more excellent photocatalytic performance due to appropriate pore structure (more pores in the range of 10~1000 nm, relatively lower total porosity and pore volume), which could provide an excessive exposure of TiO₂ and more gas diffusion channels.



(a)



(b)

Figure 5. The pore distribution of all PCS samples; (a) 1PCS-n, (n = 0.1~0.6, 0.6~1.25, 1.25~2.35 mm); (b) mPCS-1.25~2.35, (m = 1, 2, 3, 4, 5 g/L).

Table 4. The total porosity and pore volume of all PCS samples.

Sample	Total Porosity (%)	Pore Volume (mL/g)
1PCS-0.1~0.6	61.19	0.8438
1PCS-0.6~1.25	43.00	0.4053
1PCS-1.25~2.35	43.38	0.3973
2PCS-1.25~2.35	41.83	0.3655
3PCS-1.25~2.35	42.72	0.4070
4PCS-1.25~2.35	41.97	0.3706
5PCS-1.25~2.35	42.84	0.3806

In order to further identify the chemical composition of PCS before and after TiO₂ loading, XPS analysis was carried out. The XPS spectra survey patterns of Mg, Fe, O, Ti, C, Si, and Al peaks are showed in Figure 6a, indicating the elemental components and their chemical states of PCS, which are consistent with the XRD results. The C1s peak correspond to organic and inorganic carbon, which represent the pollution peak. The O1s characteristic peaks can be deconvoluted into four peaks at 529.6, 530.8, 531.9 and 532.5 eV, which can be assigned to Ti–O–Ti, Al–O–Al, Ti–O–Si and Si–O–Si, respectively [26]. The Al–O–Al bonds might be caused by the calcination of CS [27]. The bonding of titanium and silicon (Ti–O–Si) is supported by peaks centered at 531.6 eV, formed by the combination of Ti–O–H bond and Si–O–H bond. The binding energy of Ti–O–Si is 531.6 eV because the O1s binding energy of Si–O bond was weakened by the formation of Ti–O–Si and the electron density of Si is higher than Ti [28]. The formation of Ti–O–Si bond confirmed that TiO₂ was loaded on the surface of CS by chemical bond, which is tighter than physical connection and could improve the catalysis durability of PCM effectively [29].

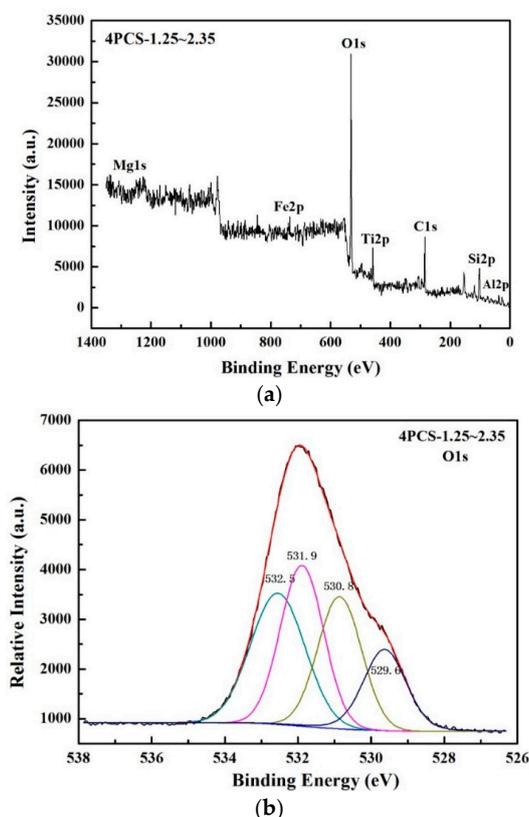


Figure 6. XPS photographs of PCS (4PCS-1.25~2.35); (a) survey pattern; (b) O1s XPS spectra; XPS: X-ray photoelectron spectroscopy.

3.2. Photocatalytic Performance

The photodecomposition process by PCSs is summarized in Figure 7. Comparing with cementitious materials, the porous structure of PCS could provide more gas diffusion and light transmittance channels which improved the reaction areas and utilization rate of photocatalysts. The photogenerated charges can generate “•OH” radicals and “O₂⁻” intermediate species by reacting with absorbed H₂O and O₂ molecules for the photodecomposition of benzene. The mechanism of photocatalytic reaction is shown as follows [4]:

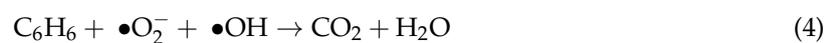


Figure 8a showed the photocatalytic degradation of benzene on the surface of PCS and control groups. 1PCS-1.25~2.35-UA presented more superior photocatalysis performance (degraded 2 μL benzene completely in 350 min) compared with other PCSs, proved to be the most efficient one. It was because CS-1.25~2.35 provided more superior pore structure to absorption of gas pollutants, which led to a larger gas-solid contact area between TiO₂ and benzene under the same condition. After 1 month of atmospheric environment exposure, 1PCS-1.25~2.35-UA-1 month could still degrade 96% of 2 μL benzene in 400 min. The control group (Surface coating) showed even better photocatalytic activity (degraded 2 μL benzene completely in 250 min) than 1PCS-1.25~2.35-UA with same amount of photocatalysts because of full exposure of TiO₂ particles. However, only after 1 month of atmospheric environment exposure, the photocatalytic activity of “Surface coating” decreased significantly by degrading 67% of 2 μL benzene in 400 min. The reason is that TiO₂ is not bound tightly with the surface of cement, which is easily washed off, abraded or enveloped by the cement hydration products. Another control group (Incorporation) showed very little photocatalytic effects because the TiO₂ incorporated into cement materials could barely touch the lights. The photocatalytic activity of TiO₂ (P25) is also shown in Figure 8a, which is lower than “Surface coating”. Despite its higher photocatalytic activity, the agglomeration of TiO₂ nanoparticles and the lower adsorption abilities are prevailing in our photocatalytic tests. Besides, the degradation of benzene by 1PCS-1.25~2.35-AC was much slower than 1PCS-1.25~2.35-UA, which indicated more active sites were provided by ultrasonic activation than alkali activation. Thus, ultrasonic activation was selected in subsequent experiments. As indicated in Figure 8b, the photocatalytic activity of PCS increased gradually when TiO₂ concentrations rose from 1 g/L to 4 g/L. However, when TiO₂ concentration increased to 5 g/L, the photocatalytic activity decreased compared with 4PCS-1.25~2.35. Higher concentration of TiO₂ solution led to more TiO₂ deposition on the surface of CS and higher photocatalytic activity. Nevertheless, when TiO₂ concentration came to 5 g/L, agglomeration problem of TiO₂ was serious, which led to more recombination of electron-hole and difficulties for TiO₂ particles to be exposed to the photocatalytic reaction. Hence, 4PCS-1.25~2.35 was chosen as the best material to prepare PCM.

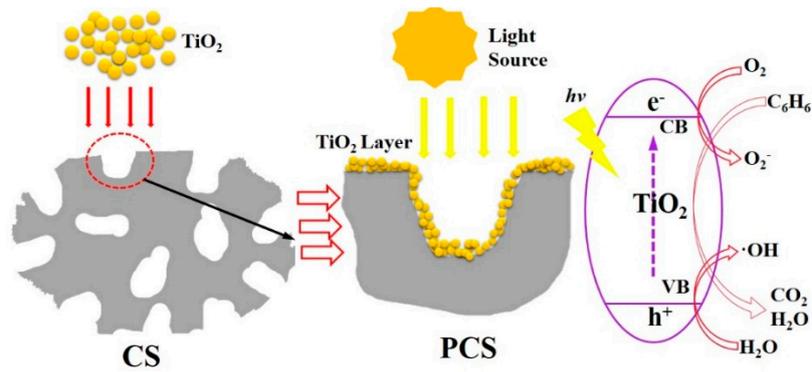


Figure 7. Preparation and catalytic mechanism of PCS.

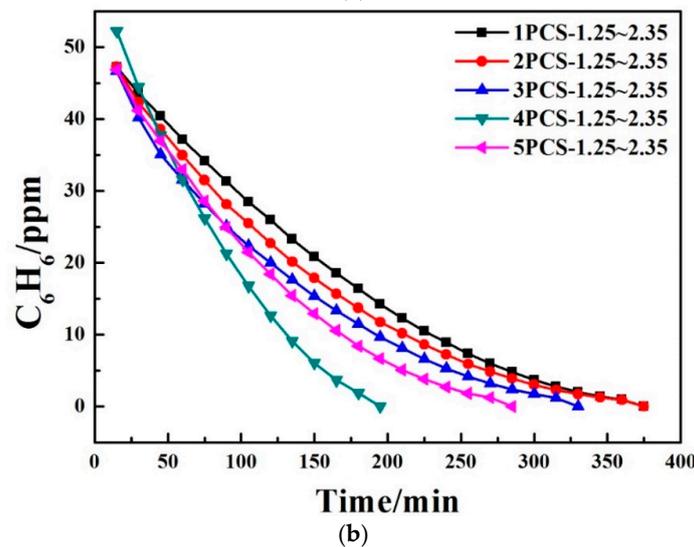
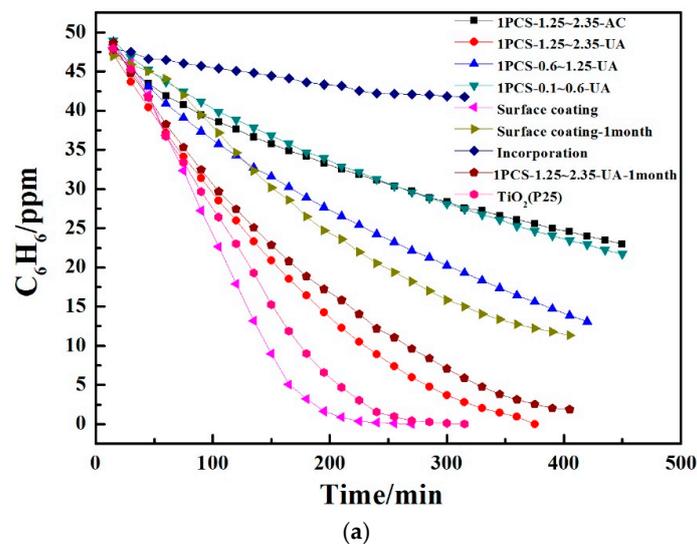


Figure 8. Photocatalytic activity of PCS to benzene; (a) 1PCS-n-UA (PCS with ultrasonic activation, $n = 0.1\text{--}0.6, 0.6\text{--}1.25, 1.25\text{--}2.35$ mm), 1PCS-1.25~2.35-AC (PCS alkali activated), surface coating (the mortars with TiO_2 on the surface), surface coating-1month (the mortars with TiO_2 on the surface after 1 month of atmospheric environment exposure), Incorporation (the mortars with TiO_2 incorporated) and TiO_2 (P25); (b) different TiO_2 coating concentrations.

Figure 9 revealed that 4PCS-1.25~2.35 still presented excellent photocatalytic effect after being loaded on the surface of cement materials. As we can see, the adsorption of benzene on surface of mortars without photocatalysts hardly had influence on experiment compared to the photocatalytic effect of samples. It was demonstrated that the surface pores of PCS played an important role in photocatalytic reaction. The presence of PCS increased the reaction area and the loading contents of TiO_2 particles, which were beneficial to improve the photocatalytic efficiency. The photocatalytic activity of PCM increased with the increase of PCS on the surface of PCM from 1 g to 5 g because the photocatalysts on the surface of PCSs were not covered by cement materials and hydration products and had no agglomeration problem. The PCM still remained high photocatalytic activities after exposure to the natural environment for 6 months (which is marked by PCM-5–6 months) which is more stable than Surface coating. It is because that the chemical bond (Si-O-Ti) between TiO_2 and CS could prevent TiO_2 particles from peeling from PCS. On the other hand, TiO_2 coated on the surface of CS was influenced slightly by the envelope of cement hydration products and the carbonation of cement. However, the prepared PCM was washed and brushed repeatedly for 6 months, and the slight coverage of macro-particles was unavoidable in atmospheric environment exposure, which could slightly decrease the photocatalytic activity of specimen [30].

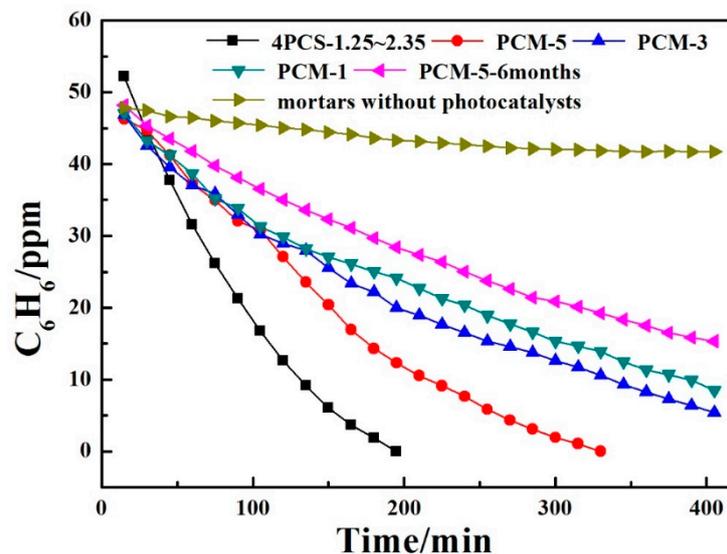


Figure 9. Photocatalytic activity of PCM and mortars without photocatalysts to benzene; PCM-1, 3, 5 (photocatalytic cement materials with 1, 3, 5 g 4PCS-1.25~2.35).

The reusability performance of catalyst is also an important parameter to evaluate the long-term catalytic performance of photocatalytic material in practical application. Figure 10 showed that time of fully degradation of $2 \mu\text{L}$ benzene by PCS with different TiO_2 concentrations and control group. All PCSs still presented excellent photocatalytic activity after being reused for 10 cycles. This might be due to the good stability of substrate and the large bonding force between TiO_2 and CS. The control group (Surface coating) showed decreased photocatalytic effects during the cycle experiment. This might be because that high surface energy of TiO_2 and low specific surface area of mortar caused the agglomeration of TiO_2 particles, which could lead to more recombination of electron-hole and difficulties for TiO_2 particles to be exposed to the photocatalytic reaction. Comparing with the previous studies [22,31], our material had an excellent long-term photocatalytic performance with higher retention rate for photocatalytic activity. The reason was that the structures of porous PCS provided higher specific surface area and more gas diffusion channels than cementitious materials and prevented TiO_2 particles from being affected seriously by the envelope of cement hydration products

and the carbonation of cement [32], which improved the reacting areas, utilization rate of catalysts, and enhanced the photocatalytic activity effectively.

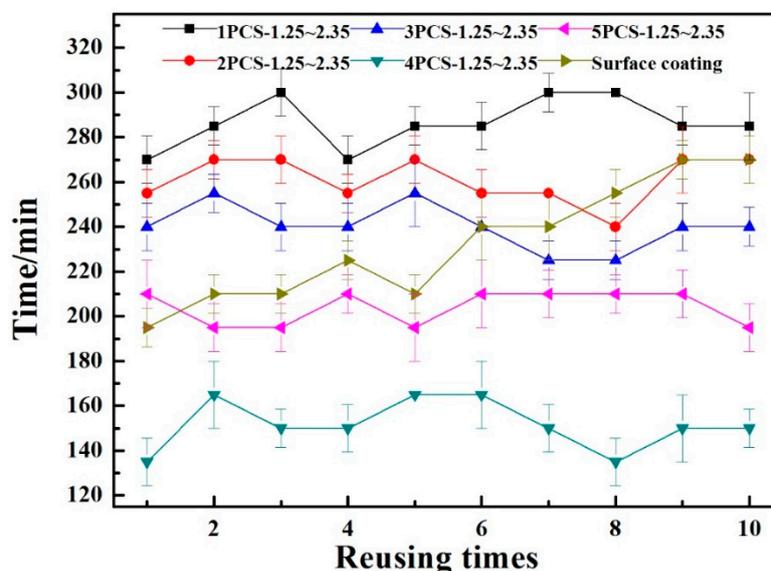


Figure 10. Time of fully degradation of 2 μ L benzene by mPCS-1.25~2.35 ($m = 1\sim 5$ g/L) for 10 cycles.

4. Conclusions

In this paper, PCS and PCM were prepared successfully. The physicochemical properties, photocatalytic activity and long-term catalytic performance were studied comprehensively. 2 μ L benzene was degraded completely in 200 min by 5 g 4PCS-1.25~2.35 and 300 min by PCM-5, and was still degraded over 80% in 400 min by PCM-5 after exposure to the natural environment for 6 months. The results showed that PCS presented an excellent photocatalytic activity due to abundant pores, stable structure and large reaction area of CS. The detailed results showed that the photocatalytic activity of PCS increased through adjusting CS in appropriate pore structure (more pores in the range of 10~1000 nm, appropriate total porosity and pore volume) and TiO₂ in appropriate ratio, which could provide an excessive exposure of TiO₂ on the surface and more gas diffusion channels. PCS exhibited an optimal degradation activity to benzene when TiO₂ concentration rose to 4 g/L. However, a fraction of the loaded TiO₂ was inaccessible to the reactants or light as the concentration of TiO₂ kept increasing (up to 5 g/L). Excessive TiO₂ might have serious agglomeration problems, leading to the decreased adsorption rate of air pollutants.

After PCS was loaded on the cement materials, PCM could still present a high catalytic activity. Even after exposure to the natural environment for 6 months, PCM still remained good photocatalytic activity comparing with the cement material with TiO₂ on the surface or incorporated. The excellent photocatalytic activity and long-term catalytic performance were attributed to the activation for surface groups and distinctive structure of PCS. The porous structure of PCS increased specific surface areas, gas diffusion and light transmittance channels of PCM, improving the reacting areas and utilization rate of TiO₂ effectively. Through this preparation route, more TiO₂ particles on PCS were exposed to environment without being influenced by the envelope of cement hydration products and the carbonation of cement.

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References

1. Mendoza, C.; Valle, A.; Castellote, M.; Bahamonde, A.; Faraldos, M. TiO₂ and TiO₂-SiO₂ coated cement: Comparison of mechanic and photocatalytic properties. *Appl. Catal. B* **2015**, *178*, 155–164. [[CrossRef](#)]
2. Yang, L.; Wang, F.; Hakki, A.; Macphee, D.E.; Liu, P.; Hu, S. The influence of zeolites fly ash bead/TiO₂ composite material surface morphologies on their adsorption and photocatalytic performance. *Appl. Surf. Sci.* **2017**, *392*, 687–696. [[CrossRef](#)]
3. Aïssa, A.H.; Puzenat, E.; Plassais, A.; Herrmann, J.-M.; Haehnel, C.; Guillard, C. Characterization and photocatalytic performance in air of cementitious materials containing TiO₂. Case study of formaldehyde removal. *Appl. Catal. B* **2011**, *107*, 1–8. [[CrossRef](#)]
4. Pérez-Nicolás, M.; Balbuena, J.; Cruz-Yusta, M.; Sánchez, L.; Navarro-Blasco, I.; Fernández, J.M.; Alvarez, J.I. Photocatalytic NO_x abatement by calcium aluminate cements modified with TiO₂: Improved NO₂ conversion. *Cem. Concr. Res.* **2015**, *70*, 67–76. [[CrossRef](#)]
5. Yang, L.; Wang, F.; Shu, C.; Liu, P.; Zhang, W.; Hu, S. An in situ synthesis of Ag/AgCl/TiO₂/hierarchical porous magnesian material and its photocatalytic performance. *Sci. Rep.* **2016**, *6*, 21617. [[CrossRef](#)] [[PubMed](#)]
6. Boonen, E.; Beeldens, A. Photocatalytic roads: From lab tests to real scale applications. *Eur. Transp. Res. Rev.* **2012**, *5*, 79–89. [[CrossRef](#)]
7. Folli, A.; Pade, C.; Hansen, T.B.; De Marco, T.; Macphee, D.E. TiO₂ photocatalysis in cementitious systems: Insights into self-cleaning and depollution chemistry. *Cem. Concr. Res.* **2012**, *42*, 539–548. [[CrossRef](#)]
8. Todorova, N.; Giannakopoulou, T.; Karapati, S.; Petridis, D.; Vaimakis, T.; Trapalis, C. Composite TiO₂/clays materials for photocatalytic NO_x oxidation. *Appl. Surf. Sci.* **2014**, *319*, 113–120. [[CrossRef](#)]
9. Folli, A.; Pochard, I.; Nonat, A.; Jakobsen, U.H.; Shepherd, A.M.; Macphee, D.E. Engineering photocatalytic cements: Understanding TiO₂ surface chemistry to control and modulate photocatalytic performances. *J. Am. Ceram. Soc.* **2010**, *93*, 3360–3369. [[CrossRef](#)]
10. Janus, M.; Zatorska, J.; Czyżewski, A.; Bubacz, K.; Kusiak-Nejman, E.; Morawski, A.W. Self-cleaning properties of cement plates loaded with N,C-modified TiO₂ photocatalysts. *Appl. Surf. Sci.* **2015**, *330*, 200–206. [[CrossRef](#)]
11. Rastogi, M.; Vaish, R. Visible light induced water detoxification through portland cement composites reinforced with photocatalytic filler: A leap away from TiO₂. *Constr. Build. Mater.* **2016**, *120*, 364–372. [[CrossRef](#)]
12. Lee, B.Y.; Jayapalan, A.R.; Bergin, M.H.; Kurtis, K.E. Photocatalytic cement exposed to nitrogen oxides: Effect of oxidation and binding. *Cem. Concr. Res.* **2014**, *60*, 30–36. [[CrossRef](#)]
13. Powers, T.C. Structure and physical properties of hardened portland cement paste. *J. Am. Ceram. Soc.* **1958**, *41*, 1–6. [[CrossRef](#)]
14. Ramirez, A.M.; Demeestere, K.; De Belie, N.; Mäntylä, T.; Levänen, E. Titanium dioxide coated cementitious materials for air purifying purposes: Preparation, characterization and toluene removal potential. *Build. Environ.* **2010**, *45*, 832–838. [[CrossRef](#)]
15. Chen, J.; Poon, C.-S. Photocatalytic activity of titanium dioxide modified concrete materials—Influence of utilizing recycled glass cullets as aggregates. *J. Environ. Manag.* **2009**, *90*, 3436–3442. [[CrossRef](#)] [[PubMed](#)]
16. Maury-Ramirez, A.; De Muynck, W.; Stevens, R.; Demeestere, K.; De Belie, N. Titanium dioxide based strategies to prevent algal fouling on cementitious materials. *Cem. Concr. Compos.* **2013**, *36*, 93–100. [[CrossRef](#)]
17. Wang, F.; Yang, L.; Wang, H.; Yu, H. Facile preparation of photocatalytic exposed aggregate concrete with highly efficient and stable catalytic performance. *Chem. Eng. J.* **2015**, *264*, 577–586. [[CrossRef](#)]
18. Wang, F.; Yang, L.; Guan, L.; Hu, S. Microstructure and properties of cement foams prepared by magnesium oxychloride cement. *J. Wuhan Univ. Technol. Mater. Sci.* **2015**, *30*, 331–337. [[CrossRef](#)]
19. Sagrañez, R.; Álvarez, J.I.; Cruz-Yusta, M.; Mármol, I.; Morales, J.; Vila, J.; Sánchez, L. Enhanced photocatalytic degradation of NO_x gases by regulating the microstructure of mortar cement modified with titanium dioxide. *Build. Environ.* **2013**, *69*, 55–63. [[CrossRef](#)]
20. Hu, S.; Yang, T.; Wang, F. Influence of mineralogical composition on the properties of lightweight aggregate. *Cem. Concr. Compos.* **2010**, *32*, 15–18.

21. Olorunsogo, F.T.; Padayachee, N. Performance of recycled aggregate concrete monitored by durability indexes. *Cem. Concr. Res.* **2002**, *32*, 179–185. [[CrossRef](#)]
22. Yang, L.; Liu, P.; Li, X.; Li, S. The photo-catalytic activities of neodymium and fluorine doped TiO₂ nanoparticles. *Ceram. Int.* **2012**, *38*, 4791–4796. [[CrossRef](#)]
23. Cravanzola, S.; Cesano, F.; Gaziano, F.; Scarano, D. Sulfur-doped TiO₂: Structure and surface properties. *Catalysts* **2017**, *7*, 214. [[CrossRef](#)]
24. Habibi, M.H.; Mikhak, M. Titania/zinc oxide nanocomposite coatings on glass or quartz substrate for photocatalytic degradation of direct blue 71. *Appl. Surf. Sci.* **2012**, *258*, 6745–6752. [[CrossRef](#)]
25. Nischk, M.; Mazierski, P.; Gazda, M.; Zaleska, A. Ordered TiO₂ nanotubes: The effect of preparation parameters on the photocatalytic activity in air purification process. *Appl. Catal. B* **2014**, *144*, 674–685. [[CrossRef](#)]
26. Kim, W.B.; Choi, S.H.; Lee, J.S. Quantitative analysis of Ti-O-Si and Ti-O-Ti bonds in Ti-Si binary oxides by the linear combination of xanes. *J. Phys. Chem. B* **2000**, *104*, 8670–8678. [[CrossRef](#)]
27. Lippmaa, E.; Mägi, M.; Samoson, A.; Engelhardt, G.; Grimmer, A.R. Structural studies of silicates by solid-state high-resolution ²⁹Si nmr. *J. Am. Chem. Soc.* **1980**, *102*, 4889–4893. [[CrossRef](#)]
28. Tokarský, J.; Čapková, P. Structure compatibility of TiO₂ and SiO₂ surfaces. *Appl. Surf. Sci.* **2013**, *284*, 155–164. [[CrossRef](#)]
29. Atuchin, V.V.; Kesler, V.G.; Pervukhina, N.V.; Zhang, Z. Ti2p and O1s core levels and chemical bonding in titanium-bearing oxides. *J. Electron. Spectrosc. Relat. Phenom.* **2006**, *152*, 18–24. [[CrossRef](#)]
30. Maury-Ramirez, A.; Demeestere, K.; De Belie, N. Photocatalytic activity of titanium dioxide nanoparticle coatings applied on autoclaved aerated concrete: Effect of weathering on coating physical characteristics and gaseous toluene removal. *J. Hazard. Mater.* **2012**, *211–212*, 218–225. [[CrossRef](#)] [[PubMed](#)]
31. Yu, C. *Deactivation and Regeneration of Environmentally Exposed Titanium Dioxide (TiO₂) Based Products*; Department of Chemistry, Chinese University of Hong Kong: Hong Kong, China, 2003.
32. Wang, F.; Yang, L.; Sun, G.; Guan, L.; Hu, S. The hierarchical porous structure of substrate enhanced photocatalytic activity of TiO₂/cementitious materials. *Constr. Build. Mater.* **2014**, *64*, 488–495. [[CrossRef](#)]



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