

Review

Major Impediment to Highly Efficient, Stable and Low-Cost Perovskite Solar Cells

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Abstract: Organic–inorganic hybrid perovskite solar cells (PSCs) have made immense progress in recent years, owing to outstanding optoelectronic properties of perovskite materials, such as high extinction coefficient, carrier mobility, and low exciton binding energy. Since the first appearance in 2009, the efficiency of PSCs has reached 23.3%. This has made them the most promising rival to silicon-based solar cells. However, there are still several issues to resolve to promote PSCs' outdoor applications. In this review, three crucial aspects of PSCs, including high efficiency, environmental stability, and low-cost of PSCs, are described in detail. Recent in-depth studies on different aspects are also discussed for better understanding of these issues and possible solutions.

Keywords: perovskite solar cells; conversion efficiency; stability; low-cost; large-scale

1. Introduction

Due to the low efficiency and environmental problems of fossil fuels, exploring renewable energy sources to meet the increasing energy demands has become more and more important for industrial applications and human life. Among the available renewable energy sources, solar energy is considered as the most promising source to solve the energy crisis. Since the first solar cell was created in Bell Laboratory in 1954, the photovoltaic industry has achieved very great developments, and various new types of photovoltaic technologies have emerged, such as thin-film and organic solar cells, besides conventional crystalline silicon solar cells. Owing to outstanding optoelectronic properties, PSCs have attracted great interest in recent years, and become a promising candidate for crystalline silicon solar cells.

Halide perovskite materials were first adopted as a sensitizer in dye-sensitized solar cells (DSCs). In 2009, Miyasaka and co-workers demonstrated a liquid electrolyte perovskite solar cell using $\text{CH}_3\text{NH}_3\text{PbI}_3$ (MAPbI_3) as the sensitizer, with a power conversion efficiency (PCE) of only 3.8% [1]. Park et al. fabricated a MAPbI_3 quantum-dot-sensitized solar cell using the same dye-sensitized concept, and the PCE was improved to 6.5% because of the surface modification of TiO_2 in 2011 [2]. However, the stability of PSCs, mentioned above, was found to be a few minutes, because the quantum dots dissolved gradually into the redox electrolyte. In 2012, Kim et al. reported the first all-solid-state PSC which used 2,2',7,7'-Tetrakis[*N,N*-di(4-Methoxyphenyl)amino]-9,9'-spirobifluorene (spiro-OMeTAD) as hole transport material, and gained a PCE of 9.7% [3]. In the same year, Lee et al. replaced the traditional TiO_2 mesoporous layer with insulating Al_2O_3 , resulting in an open circuit voltage (V_{OC}) exceeding 1.1 V. Then, they removed the mesoporous layer, making the perovskite layer directly contact the compact layer, and obtained the first planar heterojunction PSC with a PCE of 1.8% [4]. Afterwards, the efficiency of PSCs has increased at an incredible rate. In 2017, Seok et al.

remarkably raised the PCE to 22.1%, making the performance of PSCs comparable with commercialized crystalline silicon solar cells and thin-film photovoltaics, like CdTe and CIGS, but with extremely low cost [5]. In 2018, an extraordinary efficiency of 23.3% was achieved and, to the best of our knowledge, it is current world record [6].

Significant progress has been made in the last decade, and PSCs have been regarded as a strong competitor for silicon-based solar cells. However, the commercialization progress of PSCs remains extremely slow. This can be ascribed to the following three aspects. Firstly, the efficiency testing of some high-performance PSCs is completed under the inert atmosphere in a glove box, and the results cannot represent the actual operating condition of PSCs; in other words, the efficiency of PSCs is not enough for outdoor application. Furthermore, the stability of PSCs cannot rival other commercialized solar cells because of the short lifetime spans. Lastly, even though the cost of PSCs is already lower than other photovoltaics, the organic hole transport layer (HTL) and noble metal electrode make it possible for PSCs to further decrease the cost, so that PSCs can be more cost-effective. In this review, we will provide an overview for recent progress in efficiency, stability, and production cost of PSCs, with the expectation of accelerating the commercialization process of PSCs and offering a reference for researchers.

2. Materials and Device Structure of PSCs

2.1. Perovskite Materials

The excellent performance and rapid development of PSCs are attributed to numerous properties of perovskite materials. Perovskites have generic chemical formula ABX_3 , in which A is a monovalent cation, such as methylammonium (MA^+) $CH_3NH_3^+$, formamidinium (FA^+) $CH_2(NH_2)_2^+$, or Cs^+ ; B stands for Pb or Sn; and X for iodide (I), bromide (Br), or chloride (Cl). The crystalline perovskite structure is illustrated in Figure 1. This structure can be described as a 3D sublattice comprising B^{2+} and X^- that coordinate as a BX_6 octahedra, which is charge-balanced by A^+ filling the empty spaces. The ion in A site has significant impact on the stability of perovskite lattice, and it can be understood with the geometric tolerance factor

$$\tau = (R_A + R_X)/2^{1/2}(R_B + R_X), \quad (1)$$

where R_A , R_B , and R_C represents the ionic radius of A, B, and C site ions, respectively. Some researchers have shown that perovskite materials have a stable crystal structure when the tolerance factor is between 0.78 and 1.05 [7]. The A site ion is located at the center of BX_6 octahedral skeleton; if it is too small, the skeleton of the octahedron will be distorted. When the A site ion radius is, conversely, too large, the octahedral structure will be destroyed, and the three-dimensional symmetric structure cannot be formed.

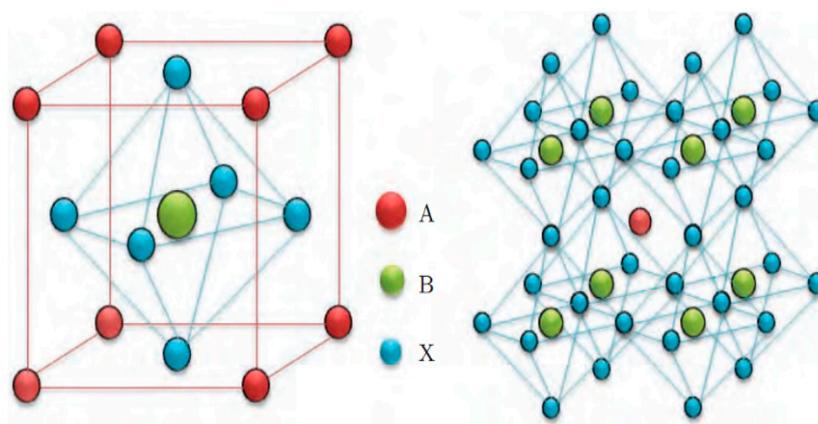


Figure 1. Crystal structure of perovskite [8].

The most typical perovskite material, MAPbI_3 , has a band gap of approximately 1.55 eV, which is the optimum value of photovoltaic materials to respond to all visible light. The extinction coefficient at 550 nm can reach $1.5 \times 10^4 \text{ cm}^{-1}$ [2], and a 400 nm-thick film of this material can absorb almost all photons in the spectral range of visible light [9]. The outstanding electrical properties guarantee superior bipolar carrier transport performance of MAPbI_3 , which enables the diffusion length of the electron and hole to exceed $1 \mu\text{m}$ [10], and allows the carrier mobility to reach $800 \text{ cm}^2/\text{V}\cdot\text{s}$ [11]. Moreover, the exciton-binding energy of MAPbI_3 film is less than 10 meV [12], while the carrier lifetime is longer than other thin-film solar cells. In terms of device fabrication, PSCs have simple architectures with low-cost solution processing methods, making them competitive with other kinds of solar cells. In addition, PSCs can be easily prepared on flexible substrates, providing wide applications for PSCs in the future.

2.2. Device Structure of PSCs

Typical PSCs consist of a perovskite active layer sandwiched between a hole transport layer (HTL) and an electron transport layer (ETL), as well as two corresponding electrodes to collect holes and electrons. The active layer absorbs sunlight to generate electron–hole pairs, which are conducted to the fluorine doped tin oxide (FTO) and the metal electrode through the ETL and the HTL, respectively, to form a photogenerated current.

There are two common configurations of PSCs, namely, mesoporous and planar heterojunction structures, as shown in Figure 2a,b. The mesoporous PSCs are generally composed of FTO/compact ETL/mesoporous layer/perovskite/HTL/metal electrode, whereas the planar heterojunction PSCs do not have a mesoporous layer, thus, the active layer contacts directly with the compact layer. Similar to DSCs, perovskite materials in mesoporous PSCs, like dyes, infiltrate in the pores of mesoporous layers. Lee et al. replaced the TiO_2 mesoporous layer with insulating Al_2O_3 to construct meso-superstructured PSCs. Furthermore, there are some other PSC structures, such as a C-electrode-based monolithic structure without an HTL (Figure 2c).

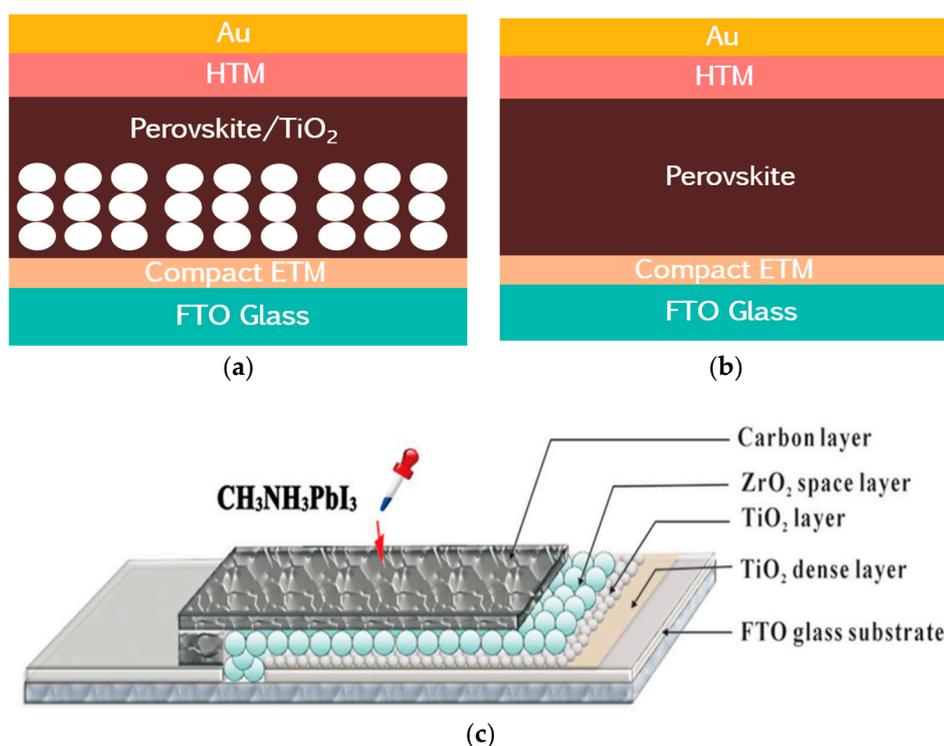


Figure 2. (a) Mesoporous perovskite solar cell (PSC); (b) Planar heterojunction PSC; (c) Monolithic hole-conductor-free PSC [13].

3. Methodologies for Efficient PSCs

The efficiency of PSCs has been raised by a variety of approaches, such as utilizing different types of ETL and HTL, employing novel device architecture and optimizing the preparation processes. In view of the perovskite active layer, several methods can be applied to fabricate highly efficient PSCs. The chemical composition of perovskite comes first on the list of elements affecting the performance of PSCs, and different combination of ions in each site could create a continuously adjustable band gap and various crystalline phases at different temperatures, which would have a specific photovoltaic effect. The quality of perovskite film can remarkably influence the transporting properties of electrons and holes; on the one hand, the coverage and homogeneity need to be taken into consideration while, on the other hand, the grain boundaries usually exist in the perovskite film and act as recombination center that needs to be passivated. The interface between the perovskite and carrier transport layer is also of great importance in improving the PCE of PSCs. In this section, we will focus on the perovskite active layer and ascribe the increasing PCE to three aspects, described above, and review recent progress to facilitate a better understanding for researchers in PSCs.

3.1. Optimization for Chemical Composition of Perovskite

A suitable band gap is significant for the optoelectronic properties of photovoltaic materials. Absorption spectra can be adjusted through changing the band gap of perovskite materials, and a narrow band gap means more incident light can be absorbed by perovskite, while a wide one signifies higher V_{OC} . Generally speaking, high-performance photovoltaic materials have a strong absorption band in the visible and near-infrared regions. The perovskite structures have a combination of different components which can be regulated to obtain different band gaps ranging from 1.15 eV to 3.06 eV through compositional control, to strike a balance between absorption spectra and V_{OC} [14–18].

The most commonly used monovalent cation at A site is MA^+ , and $MAPbI_3$ with the band gap of 1.55 eV is also ideal for single junction solar cells [19]. The ionic radius of MA^+ is favorable for stable perovskite crystal structure according to the tolerance factor described by Equation (1), but FA^+ is thermally more stable than MA^+ . Due to the narrower band gap of $FAPbI_3$, it has a better response to near-infrared light, compared to $MAPbI_3$. This is because the BX_6 octahedra creates lattice expansion, that lowers the value of the band gap. However, organic monovalent cations are not favored by commercialization, therefore, many studies have focused on inorganic cations. On account of the constraint of tolerance factor, Cs^+ , with the largest ionic radius, is often employed as a monovalent cation of perovskite. In addition, Cs^+ is a stable and non-radioactive monovalent cation, making $CsPbI_3$ one of the few options for inorganic perovskites with an attractive band gap of approximately 1.7 eV that is suitable for tandems with silicon (but suboptimal for single junctions). Nevertheless, black phase $CsPbI_3$ only occurs at higher temperatures, whereas yellow phase $CsPbI_3$ occurs at room temperatures, and PSCs based on yellow phase only have a PCE of 0.09% [20]. The diagrammatic structure of $CsPbI_3$ is shown in Figure 3a, and the absorbance spectra of black and yellow phases of $CsPbI_3$ are shown in Figure 3b. It can be clearly seen that the $CsPbI_3$ has no light absorption in most of the visible light region, leading its photovoltaic effect being extremely low. Consequently, perovskite with mixed cations has attracted researchers' attention, in order to enhance properties of different aspects of perovskite materials [21]. For example, perovskite can be high-efficiency and thermally stable, and remains as black phase at room temperatures by applying double cation materials (MA^+ , FA^+ or Cs^+ , FA^+) [22–25]. Seok et al. employed double cation perovskite $(FAPbI_3)_{0.95}(MAPbBr_3)_{0.05}$ as active layer, which showed an efficiency of 23.2%, and stabilized efficiency of 22.85%. Remarkably, they prepared a large-scaled PSC ($\sim 1 \text{ cm}^2$) with a PCE of 21.7% [26]. The resultant device showed better thermal stability, maintaining almost 95% of its initial performance for more than 500 h after thermal annealing at 60 °C.

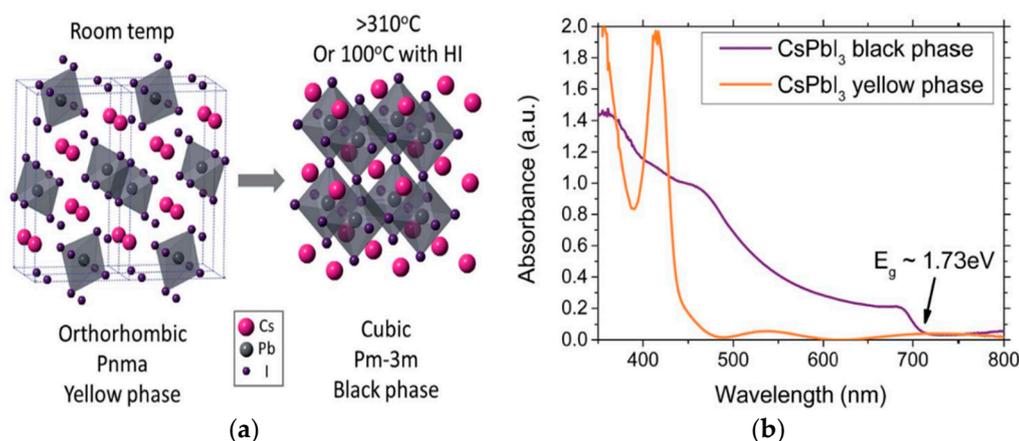


Figure 3. (a) Crystal structure of CsPbI₃; (b) Absorbance spectra of CsPbI₃ [27].

In view of the excellent optoelectronic properties of double cation perovskite materials, researchers concentrated on triple cation perovskite (Cs⁺, MA⁺, FA⁺). Triple cation perovskite materials perform better in crystal phase stability, film quality, and environmental sensitivity, which improve the reproducibility of PSCs. Saliba et al. prepared a PSC using Cs_x(MA_{0.17}FA_{0.83})_{1-x}Pb(Br_{0.17}I_{0.83})₃ as active layer, producing high-efficiency PSCs with stabilized PCE at 21.1% and output at 18%, under operational conditions, after 250 h [28]. Peng et al. obtained a high steady-state efficiency of 17.9% for MAPb₃-based cells, and 19.3% for Cs_{0.05}(MA_{0.17}FA_{0.83})_{0.95}Pb(I_{0.83}Br_{0.17})₃ cells, by optimizing ETL. Moreover, they demonstrated a high steady-state PCE of 16.6% for a semi-transparent cell, and used it to achieve a steady-state efficiency of 24.5% for a four-terminal perovskite–silicon tandem cell [29]. The radius of the A ion is limited by the tolerance factor, and inorganic Rb⁺ with a small radius is difficult to maintain in stable black phase perovskite materials. However, in the case of a multiple cation materials, Rb⁺ can also be applied to high-efficiency PSCs [30]. Saliba et al. found that Rb_{0.05}FA_{0.95}PbI₃-based PSCs exhibit a best PCE of 17.16%, which is much higher than that of the FAPbI₃ device, and this double cation PSC shows superior stability against high humidity (85%), under ambient conditions without encapsulation, retaining the high performance for 1000 h [31]. A summary of the research cited above is shown in Table 1.

Pb²⁺ is a general component employed in B site, unfortunately, the toxicity issue of Pb²⁺ remains a major obstacle in the commercialization process. With the similar properties and the same main group, Sn is an ideal substitute for Pb, but Sn²⁺ is easily oxidized to Sn⁴⁺, which is harmful to stable perovskite crystal structure. As a result, high-performance PSCs usually utilize both Sn²⁺ and Pb²⁺ to form low-toxic perovskite materials, so as to reduce the band gap and respond to the near-infrared light [32]. Moreover, the carrier density and conductivity of the perovskite materials can be increased with a certain amount of Sn²⁺.

As for halide, a different anion radius of halide result in different lattice constants of the perovskite [28], and the lattice constants become larger, along with the ionic radius, and the red shift of absorption spectrum occurs [33]. It is noteworthy that the band gap of perovskite can be regulated by adjusting the I–Br ratio.

Table 1. Summary of researches about improving the power conversion efficiency (PCE) of PSC by optimizing chemical composition.

Perovskite Material	Device Structure	PCE/%	Ref.
(FAPbI ₃) _{0.85} (MAPbBr ₃) _{0.15}	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/PTAA/Au	18.4%	[21]
FA _{0.85} Cs _{0.15} PbI ₃	FTO/compact TiO ₂ /perovskite/spiro-OMeTAD/Ag	16.21%	[22]
FA _{0.9} Cs _{0.1} PbI ₃	FTO/compact TiO ₂ /perovskite/spiro-OMeTAD/Ag	16.5%	[23]
Cs _{0.2} FA _{0.8} PbI _{2.84} Br _{0.16}	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/spiro-OMeTAD/Au	17.35%	[24]
Mixed perovskite (PbI ₂ : FAI = 1.05)	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/spiro-OMeTAD/Au	20.8%	[25]
(FAPbI ₃) _{0.95} (MAPbBr ₃) _{0.05}	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/DM/Au	22.6%	[26]
Cs _x (MA _{0.17} FA _{0.83}) _{100-x} Pb(I _{0.83} Br _{0.17}) ₃	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/spiro-OMeTAD/Au	21.17%	[28]
Cs _{0.05} (MA _{0.17} FA _{0.83}) _{0.95} Pb(I _{0.83} Br _{0.17}) ₃	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/spiro-OMeTAD/Au	19.3%	[29]
(RbCsMAFA)PbI ₃	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/spiro-OMeTAD, PTAA/Au	20.6%	[30]
Rb _{0.05} FA _{0.95} PbI ₃	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/spiro-OMeTAD/Au	17.16%	[31]

3.2. Forming High-Quality Perovskite Films by Passivating Grain Boundaries (GBs)

PSCs have frequently employed polycrystalline film as a light absorber, which creates GBs formed in the interfaces between two grains. GB tends to affect the electrical and thermal property of the material. The polycrystalline nature generates a number of GBs, as compared to their single-crystalline counterparts. In photovoltaic devices, polycrystalline materials show poor performance compared to their single-crystalline counterparts because of GBs, which are usually effective recombination centers for photon-generated electrons and holes, as well as scattering centers of free carriers. For example, silicon and CdTe-based solar cells always suffer poor performance from adverse impact induced by GBs [34,35], whereas CIS (CuInSe₂) exhibited PCE as high as 20% without special GB passivation, significantly surpassing the single-crystalline CIS-based device [36]. Therefore, the influence of GBs on photovoltaic devices is still being studied in depth. When it comes to PSCs, it has been reported that GBs in CsPbX₃ do not create mid-gap states, but induce very shallow states near the valence band maxima, hampering hole diffusion. For MAPbX₃, similarly to CIS, GBs were found to have benign effect on optoelectronic properties [37]. However, GBs in MAPbX₃ showed fast nonradiative recombination, which implies that GBs might need to be passivated.

One-step and two-step processes are commonly applied to fabricate perovskite film. Citing MAPbI₃ as example, a one-step method is performed by spin-coating a solution of CH₃NH₃I (MAI) and PbI₂ in organic solvents to form the photoactive layer of MAPbI₃, while the so-called two-step process is performed by spin-coating a mixed solution of MAI and PbI₂ onto the substrate. The perovskite film prepared by the two-step process has a higher film quality, better coverage on the substrate and, also, higher permeability in the mesoporous layer for mesoporous PSCs [38]. When considering time for the production of PSCs in industry, one-step is expected to be a more desirable process because it will reduce the time for perovskite coating and, thereby, fabrication of PCSs. Thus, retarding crystallization or controlling crystal growth is important step in a one-step procedure.

Roldan-Carmona et al. found that an excess of PbI₂ improves the average fill factor (FF) and V_{OC}, boosting the average PCE from 16.2% to values close to 18% [39]. It can be seen, from the results of scanning electron microscopy (SEM) and time resolved photoluminescence (TRPL), that an excess of

PbI_2 also improves the crystallinity of the film and the lifetime of the carriers. The same phenomenon can be discovered in perovskite with mixed cations [40]. Kim et al. also reported that the performance was greatly improved by incorporating some amount of PbI_2 into MAPbI_3 . In addition, positive effects of PbI_2 incorporation were also extended to $(\text{FAPbI}_3)_{0.85}(\text{MAPbBr}_3)_{0.15}$, yielding similar performance improvements as the MAPbI_3 cells. Nevertheless, it will have a negative impact on the quality of the perovskite film if the amount of PbI_2 is not precisely controlled [41,42].

Park and co-workers studied the effects of excess MAI on the properties of perovskite materials [43]. The non-stoichiometric precursor solutions are prepared by mixing MAI, PbI_2 , and DMSO in DMF, where the molar ratio of PbI_2/MAI is changed from 1:1 to 1:1.02, 1:1.04, 1:1.06, 1:1.08, and 1:1.1 ($x = 0, 0.02, 0.04, 0.06, 0.08, \text{ and } 0.1$). The coating process is presented schematically in Figure 4a. After the spin-coating process, intermediate $(\text{MAI})_{1+x}\text{PbI}_2\text{DMSO}$ can be formed, which provides an effective method to obtain high-quality perovskite films and achieve highly efficient device performance [44,45]. The short circuit photocurrent density (J_{SC}) hardly changes up to $x = 0.06$, but declines slightly for $x = 0.08$ and 0.1. On the other hand, both the V_{OC} and FF are increased as x is enhanced to 0.06. As a result, the average PCE reaches a maximum of 19.8% for $x = 0.06$. The results of transmission electron microscopy (TEM) showed that the stepped edges of grains disappear in the presence of excess MAI, which is indicative of a modification of the surface of perovskite grains by excess MAI. It can be seen from the results of conductive atomic force microscopy (c-AFM) (Figure 4b,c) that GBs show a brighter contrast, which indicates that the GB carries the current more efficiently.

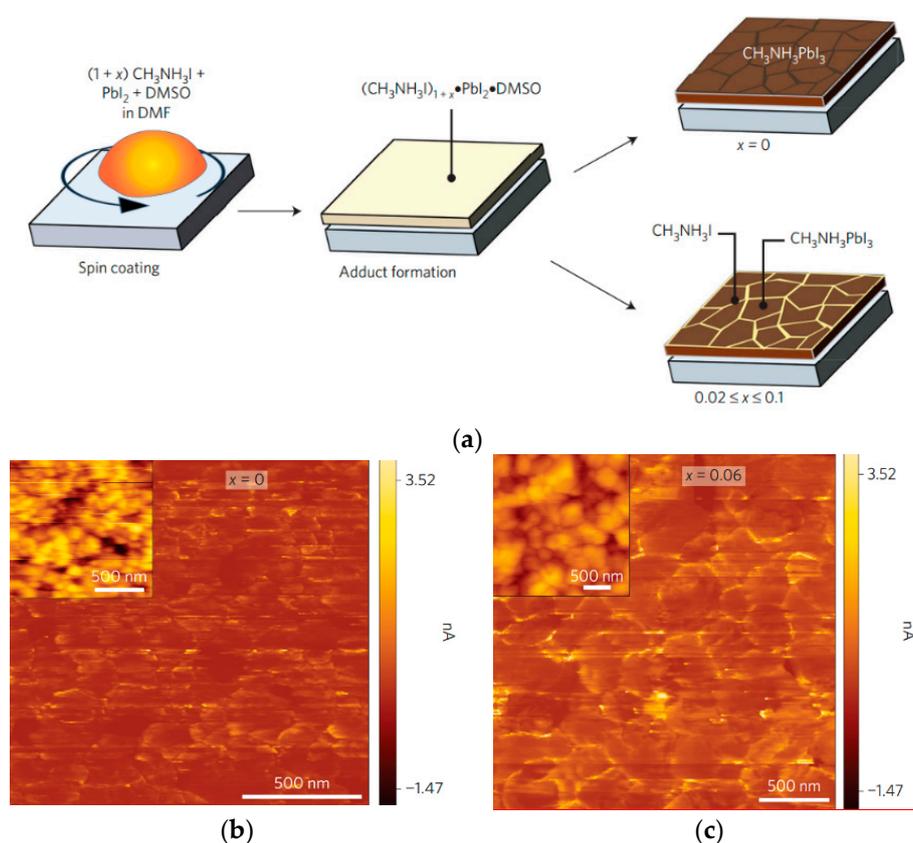


Figure 4. (a) Schematic illustration of the one-step coating process with and without MAI; c-AFM images of MAPbI_3 films with excess MAI for (b) $x = 0$; (c) $x = 0.06$ [43].

Apart from the methods of non-stoichiometric precursors, utilizing of some additives can also serve as an effective approach to achieve interface passivation, so that the performance and stability of PSCs can be improved. In 2018, Zheng et al. employed hexamethylenetetramine (HMTA) as an additive

to passivate MAPbI₃ film and, thereby, advance cell optoelectronic performance [46]. The perovskite surface, fully covered by HMTA, is modified by forming a stable insulation layer to bridge across the GBs. Additionally, HMTA can coordinate Pb²⁺ on the perovskite surface, facilitating the disappearance of PbI₂ and formation of pure perovskite nanocrystals [47], which can be seen in Figure 5a. HMTA can also serve as nucleation control reagent, facilitating perovskite to form an aligned orientation. Lastly, HMTA can cover the ZnO thin film to enhance the adhesion of perovskite with the oxide substrate. Owing to these beneficial advantages of HMTA, both PCE and reproducibility were highly improved. Li et al. introduced a large molecular organic dye AQ310 with –COOH functional group as passivator to reduce the trap states of the perovskites and promote better stability [48]. Carboxyl has been effectively used in other photovoltaic devices as an indispensable anchoring group, due to the strong coordination with metal oxide [49]. Similar to HMTA, the carboxyl of AQ310 coordinates with under-coordinated Pb²⁺, as shown in Figure 5b, thereby reducing the trap states of under-coordinated Pb²⁺, prolonging the carrier recombination time and, thus, leading to the enhancement of V_{OC}. In addition, other additives, such as fullerene derivatives [6,6]-phenyl-C61-butyric acid methyl ester (PCBM) [50], PEA⁺ [51], nitrogen-doped carbon dots (NCDs) [52], and others, are also utilized to incorporate with perovskite precursor to achieve passivation of the GBs for improved efficiency and stability. Various GB passivators used in perovskite materials, mentioned above, are summarized in Table 2.

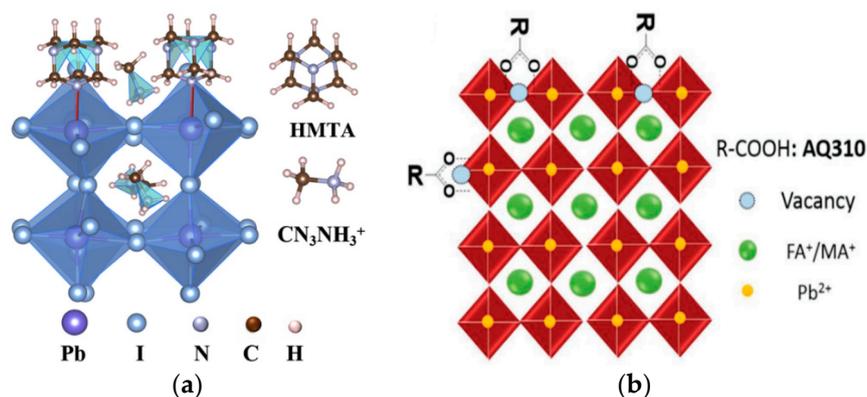


Figure 5. Crystal structure of perovskite passivated by (a) hexamethylenetetramine (HMTA) [46] and (b) AQ310 [48].

Table 2. Summary of research about improving the PCE of PSC by passivating grain boundaries (GBs).

Additive	Device Structure	PCE/%	Ref.
PbI ₂	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite(excess PbI ₂)/spiro-OMeTAD/Au	19%	[39]
MAI	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite(excess MAI)/spiro-OMeTAD/Ag	20.4%	[43]
HMTA	ITO/ZnO/HMTA-perovskite/spiro-OMeTAD/Ag	17.87%	[46]
AQ310	FTO/compact TiO ₂ /mesoporous-TiO ₂ /AQ310-perovskite/spiro-OMeTAD/Ag	19.43%	[48]
PCBM	ITO/TABC/PCBM-perovskite/PCBM/Al	15.7%	[50]
PEA ⁺	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/spiro-OMeTAD/Au	17.1%	[51]
NCDs	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite(@NCDs)/spiro-OMeTAD/Ag	15.93%	[52]

3.3. Modification of Interface

The reasonable design of the device structure, in order to obtain great properties such as energy level matching, is also one of the crucial factors affecting the performance of PSCs, apart from passivating the GBs, to achieve a certain morphology and coverage of the perovskite film. PSCs are composed of perovskite layer, carrier transport layer, and electrode, in which many interfaces exist between the different layers. There are some factors that decrease the efficiency and other properties of PSCs in these interfaces, for example, the barrier hindering the transportation of electrons or holes, the recombination center caused by surface defects, and insufficient contact between perovskite layer

and the carrier transport layer. Consequently, these problems can be solved through surface and interface engineering, thereby producing a better efficiency of PSCs.

PSCs have usually employed TiO_2 as a mesoporous layer, owing to its wide band gap, good chemical stability, and high light transmittance [53]. Meanwhile, TiO_2 with a mesoporous structure has a large specific surface area which can effectively contain perovskite materials, ensuring sufficient contact between the perovskite layer and the mesoporous layer, thereby improving the efficiency of collecting the electrons. Unfortunately, abundant oxygen vacancies, high density of electronic trap states, and low electron mobility of TiO_2 might cause undesirable charge accumulation and recombination, degrading the PCE of PSCs [54].

Metal oxides are often utilized as interface modification layers in PSCs. In 2017, Wang et al. reported an efficient PSC engineering by inserting $\text{Al}_2\text{O}_3/\text{NiO}$ layers in between the ETL and absorber layer [55], and the device structure is illustrated in Figure 6a. From the device structure, we can see that the presence of NiO accelerates the transporting of hole, and insulating Al_2O_3 limits the electrons within the perovskite layer, so that the transporting of electrons can be quite effective. Additionally, the $\text{Al}_2\text{O}_3/\text{NiO}$ interfacial engineering improves the crystalline quality of perovskite films, and effectively suppresses carrier recombination. The devices showed a maximum PCE of 18.14% and remarkable long-term air-stability, retaining ~86% of PCE over 210 days under an ambient environment. Shaikh et al. employed ZnS as an interfacial modifier between mesoporous TiO_2 and a MAPbI_3 absorber layer, via the successive ionic layer adsorption and reaction method (SILAR) [56]. The ZnS-modified layer helps in suppressing the backflow of electrons, effectively reducing interfacial charge recombination and facilitating electron transfer. Wang et al. modify the $\text{TiO}_2/\text{MAPbI}_3$ interface by introducing 3-aminopropyltrimethoxysilane (APMS) [57]. APMS hydrolysate can enhance the wettability of perovskite precursor on the TiO_2 surface, forming a perfect MAPbI_3 film, and increasing the recombination resistance at the interface.

The interface between perovskite and HTL is also a vital issue because of energy level matching and recombination of the charge carrier. Lu et al. achieved the modification of the interface between the perovskite and HTL with commercially available para-substituted benzenethiol molecules, with both improved efficiency and stability [58]. The interlayer modulates the band alignment at the perovskite/HTL interface, which could assist charge extraction and suppress carrier recombination. These effects create a high efficiency of more than 20%. During the process of preparing a perovskite film, there will frequently be non-uniformity and pinholes, with the exception of surface defects. With the aim of overcoming this problem, Wang et al. reported the improvement of the photovoltaic performance and stability of PSCs through the insertion of a solution-processed poly(methyl methacrylate) (PMMA) layer between perovskite and HTL [59]. It can be seen that the perovskite layer, without a PMMA layer, exhibited greater roughness than that with a PMMA layer. The PCE of this PSC increased to 18.1%, and the stability decreased by only about 5% during 20 days of exposure in moisture ambient conditions.

Post treatment and some additives can also effectively control the carrier's behavior of the perovskite layer/carrier transport layer interface. Singh et al. reported the incorporation of alkali metal dopants in mesoporous TiO_2 , which can effectively modulate electronic conductivity and improve the charge extraction process by counterbalancing oxygen vacancies acting as nonradiative recombination centers [60]. Sulfate bridges (SO_4^{2-}), from the alkali-bis(trifluoromethanesulfonyl)imide (alkali-TFSI) salts grafted on the surface of mesoporous TiO_2 , provide a seamless integration of absorber and electron-transporting layers that accelerates the overall transport kinetics, resulting in a high efficiency of 21.1%; the schematic illustration can be seen in Figure 6b. Zheng et al. employed crosslinkable radical copolymer poly(2,2,6,6-tetramethylpiperidinyloxy-4-yl methacrylate)-co-(4-benzoylphenyl methacrylate) (PTMA-BP) to engineer the surface of the poly(3,4-ethylenedioxythiophene): polystyrene sulfonate (PEDOT: PSS) HTL layer [61]. They found that PTMA-BP can induce better band alignment between HTL and perovskite, reduce the interfacial charge carrier recombination, and improve crystallization of perovskite hybrids that are cast on top of the HTL. Lee et al. introduced amphiphilic conjugated

Table 3. Summary of research about improving the PCE of PSC by interfacial engineering.

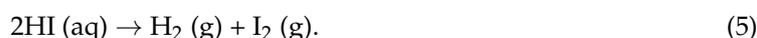
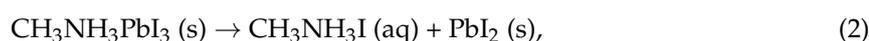
Interlayer	Device Structure	PCE/%	Ref.
Al ₂ O ₃ /NiO	FTO/compact TiO ₂ /mesoporous-TiO ₂ /Al ₂ O ₃ /NiO/perovskite-NiO/spiro-OMeTAD/Au	18.14%	[55]
ZnS	FTO/compact TiO ₂ /mesoporous-TiO ₂ /ZnS/perovskite/spiro-OMeTAD/Au	14.9%	[56]
APMS	FTO/compact TiO ₂ /APMS/perovskite/spiro-OMeTAD/Au	15.79%	[57]
Benzenethiol	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/benzenethiol/spiro-OMeTAD/Au	20.2%	[58]
PMMA	FTO/compact TiO ₂ /mesoporous-TiO ₂ /perovskite/PMMA/spiro-OMeTAD/Au	18.1%	[59]
Alkali metals	FTO/compact TiO ₂ /mesoporous-TiO ₂ -alkali metal/perovskite/Au	21.1%	[60]
PTMA-BP	ITO/PEDOT:PSS/PTMA-BP/perovskite/PCBM/AL	15%	[61]
PFN	ITO/PTPD/PFN/perovskite/PCBM/Al	17%	[62]

4. Analysis and Recent Progress of Stable PSCs

The efficiency of PSCs has been raised from 3.8% to 23.3% in just nine years, which is quite impressive compared to other thin-film solar cells that have been developed for decades. However, the stability of PSCs still remains a troublesome problem that has not been solved well, and most high-stability PSCs have a short lifetime of just around one year [63], which is a primary obstacle for the commercialization of PSCs [23]. The injection of moisture and oxygen is the main reason for the instability of PSCs, followed by thermal and ultraviolet (UV) stress. Hybrid perovskite materials may degrade in humid conditions, and suffer phase transition and thermal decomposition under fluctuation of temperature. The irradiation of UV not only reduces the stability of the perovskite materials, but also causes photocatalysis at the interface of TiO₂ when it is used as ETL. In this section, we will elaborate upon the instability of PSCs caused by ambient air, temperature, and UV irradiation, and summarize recent progress in improving the stability of PSCs. Some studies about PSC stability, cited below, are summarized in Table 4.

4.1. Moisture and Oxygen Stability

When the typical perovskite material MAPbI₃ is exposed to ambient conditions, the reactions will occur as follows [64]:



MAPbI₃ is highly sensitive to the environment, moisture and oxygen from ambient conditions, thermal stress, and even residual solvents, which can cause the decomposition of perovskite. MAPbI₃ degrades in the presence H₂O, as illustrated in Equation (2). The equilibrium of Equation (3) results in the coexistence of HI, CH₃NH₂, and CH₃NH₃I in the film. There are two ways for HI to degrade for the next step. One is the redox reaction, in the presence of oxygen, stated in Equation (4). Another consists of the photochemical reaction, whereby HI decomposes in the presence of UV radiation into I₂ and H₂, which is demonstrated in Equation (5). The consumption of HI drives the whole degradation process to move forward. We can observe, from the reactions mentioned above, that moisture and oxygen are destructive to perovskite materials and, thus, the majority of the fabrication processes must be carried out in a glove box under an inert atmosphere [65]. The performance of PSCs shows a sharp decline as the perovskite decomposes, which is a huge challenge for the application of PSCs.

Modification of the halide composition of perovskite materials was reported to be an effective way to improve the stability of the materials. The mixture of multiple halogen elements in perovskite

materials brings many positive effects, including an improvement in carrier transport and regulation of band gap. Lee et al. reported that the stability of air-processed $\text{MAPbI}_{3-x}\text{Cl}_x$ was significantly improved compared with air-processed MAPbI_3 [4]. Jeon demonstrated that the incorporation of Br into MAPbI_3 can effectively enhance its stability performance [66]. Zhu et al. employed a facile halide exchange route to incorporate Br into MAPbI_3 to form $\text{MAPbI}_{3-x}\text{Br}_x$, yielding a maximum PCE of 14.25%, and 93% of the original efficiency was retained after exposure to ambient air for 14 days without encapsulation [67].

The stability of the perovskite film has been improved with incorporating mixed halides, but halide segregation still remains a problem for the long-term stability of the PSCs [68]. As a result, cation substitution in PSCs was proposed as a further enhancement for improved device stability. It has been mentioned above that mixed cations result in a great improvement in optoelectronic properties. Actually, the incorporation of FA^+ and Cs^+ has some benefits for stability. FAPbI_3 has a larger tolerance factor and a more stable crystal structure than conventional MAPbI_3 , such that many reports have introduced perovskite of FA^+ and MA^+ hybrid cations to lift the stability of devices [69,70]. Mixed-cation materials, such as Cs/MA , Cs/FA , and Cs/MA/FA , could also improve the stability as affected by humidity and oxygen [20,22,28]. Additionally, perovskite materials based on Cs^+ with triple halide also have a remarkable stability against moisture and oxygen stress [71,72].

In order to lessen the effect of ambient air on the stability of the PSCs, some studies have introduced a waterproof tunnel layer in the PSCs. Wang et al. employed polystyrene (PS), Teflon, and polyvinylidene-trifluoroethylene copolymer (PVDF-TrFE) as tunneling materials [73]. They demonstrated that the tunneling layer can serve as an encapsulation layer to prevent perovskite film from damage caused by water. Inserting a tunneling layer between perovskite and ETL could also significantly increase device performance by suppressing carrier recombination at the point of cathode contact. Additionally, Zr acetylacetonates (ZrAcac) [74], benzylamine (BA) [75], and (*N*4,*N*4'-Di(naphthalen-1-yl)-*N*4,*N*4'-bis(4-vinylphenyl)biphenyl-4,4'-diamine) (VNPB) [76] were recently utilized as an isolated layer, to improve the stability of PSCs against ambient air.

4.2. Thermal Stability

Thermal stability also raises a lot of concerns, due to the impact on performance of PSCs. Operating temperatures of deployed solar cells can vary from $-40\text{ }^\circ\text{C}$ to $85\text{ }^\circ\text{C}$ and, depending on their location, solar cells must have the ability to work under high-temperature conditions, in particular. Conings et al. employed X-ray diffraction (XRD), X-ray photoelectron spectroscopy (XPS), PL, and AFM measurements to convincingly demonstrate that MAPbI_3 does degrade at $85\text{ }^\circ\text{C}$, even in inert conditions, indicating the poor stability under thermal stress [69]. MAPbI_3 has a low thermal conductivity [77], and the heat caused by light irradiation cannot be removed easily, thus making thermal stability worse for PSCs. The decomposition resulting from high temperature is simply due to evolution of the organic components, such as HI and CH_3NH_2 , leaving behind PbI_2 . This process is more intense when the PSCs are exposed to ambient air. Therefore, MAPbI_3 cannot be considered as a promising compound for long term operation at elevated temperatures. Snaith and co-workers successfully made PSCs using CsPbI_2Br , which exhibited greatly enhanced thermal stability, not decomposing even at $180\text{ }^\circ\text{C}$ over 30 min (Figure 7a,b), a temperature at which MAPbI_3 decomposed within 5 min [78]. However, the crystal structure of inorganic perovskite materials is highly sensitive to variations of temperature, and the performance is not eligible because inorganic perovskite materials cannot form a stable black phase with good optoelectronic properties at room temperature. Replacing MA^+ with the larger and heavier FA^+ is another option for raising the thermal stability of PSCs. Eperon et al. compared films of MAPbI_3 and FAPbI_3 heated at $150\text{ }^\circ\text{C}$ in air, to show that the FAPbI_3 did not degrade over 60 min, while the MAPbI_3 had almost completely converted to PbI_2 (Figure 7c) [79]. In addition, mixed cations with FA^+ and Cs^+ also showed an outstanding thermal stability [80].

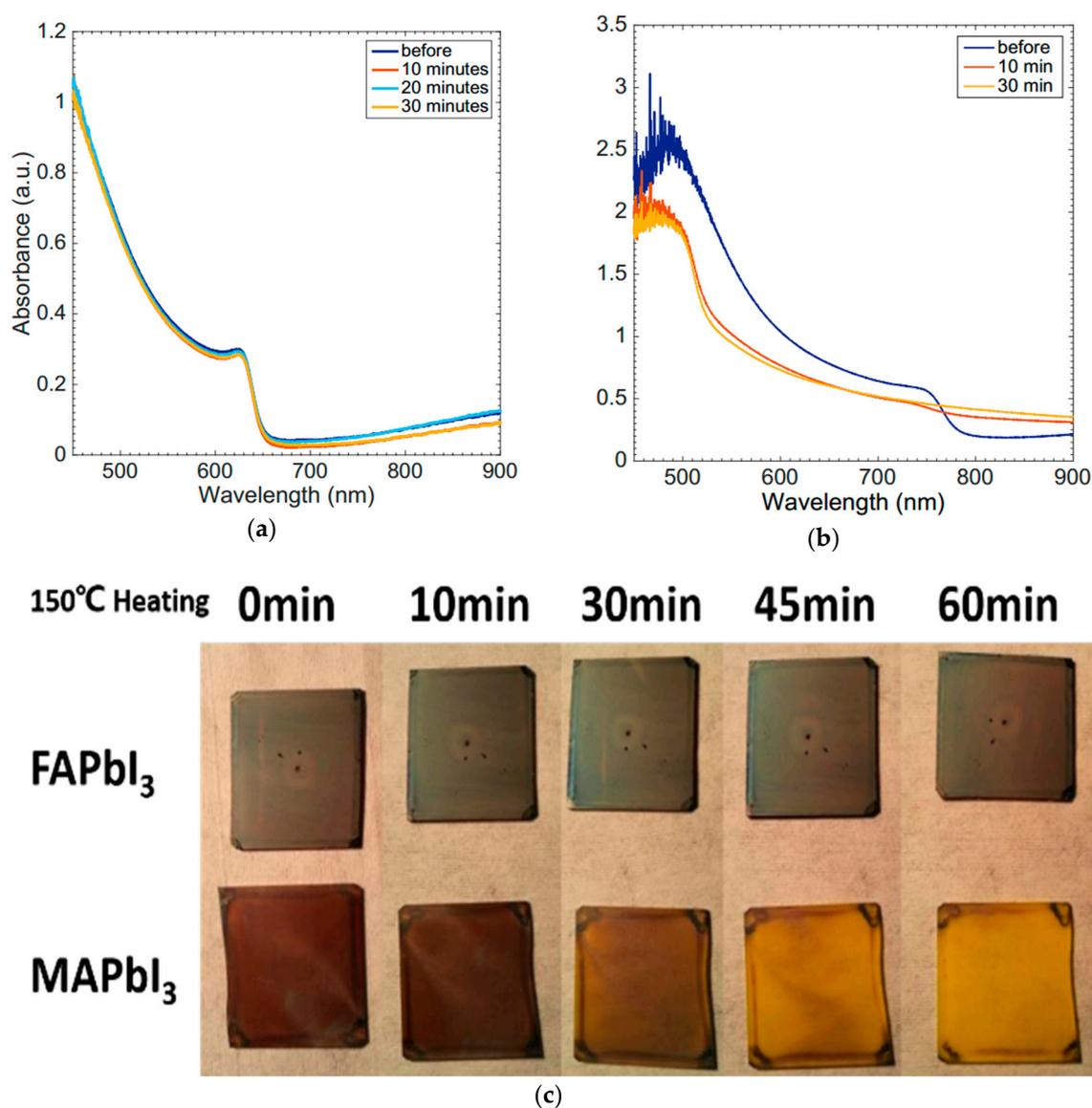


Figure 7. Absorption spectra for CsPbI₂Br (a) and MAPbI₃ (b) after heating at 180 °C [78]; (c) Thermal degradation of MAPbI₃ and FAPbI₃, when bare spin-coated films of each perovskite are heated in air at 150 °C for the times indicated [79].

The fluctuation of temperature also affects crystal structure of perovskite and, thereby, performance of PSCs. Conventional MAPbI₃ exists in a perovskite phase with tetragonal distortion at room temperature. The symmetry is further reduced at lower temperatures, as the material enters an orthorhombic phase, while it is increased with higher temperatures about 60 °C as it transitions to the cubic phase [81]. Considering the operating temperature of the solar cells, the crystal structure of MAPbI₃ is basically maintained on the tetragonal and cubic phase, and the lattice distortion has little harmful effect on PSCs based on MAPbI₃. Due to the decomposition of MAPbI₃ up to 85 °C, researchers pay more attention to crystal structure stability of FAPbI₃ and CsPbI₃, which are not easily decomposed at high temperatures. The larger size of the FA⁺, however, means that FAPbI₃ does not form a black, cubic, or tetragonal perovskite phase at room temperature, but exists as a yellow film consisting of a hexagonal non-perovskite phase. Annealing at temperatures above the tetragonal phase transition (150 °C) allows a black perovskite phase to be formed, which is retained after cooling to room temperature [79]. In order to lower the tolerance factor and prevent the appearance of a hexagonal phase, double cation perovskite with MA⁺ and FA⁺ can be employed, leading to an effective tolerance

factor close to 1, allowing a black cubic perovskite phase to be easily formed at low temperatures, and to be stable at room temperature. Inorganic CsPbI_3 does not form a stable perovskite phase at room temperature, but exists in a yellow orthorhombic phase which has an extremely low photovoltaic effect. The crystal structure of CsPbI_3 is very sensitive to temperature, and needs to be annealed at a high temperature, up to $300\text{ }^\circ\text{C}$, to form a cubic phase. CsPbBr_3 can be prepared and is structurally stable at room temperature, but has an unsuitably high band gap over 2.3 eV . Mixing both I and Br, and reducing the size of the metal halide octahedral, is an approach to stabilize CsPbX_3 structures with appropriate band gaps. Beal et al. made a fully inorganic perovskite with incorporation of Br above 20%, that exists in the cubic phase at room temperature after annealing, exhibits excellent thermal stability, and has band gaps ($1.7\text{--}1.9\text{ eV}$) suitable for perovskite–silicon or perovskite–perovskite tandems [78]. Swarnkar et al. fabricated phase-stable CsPbI_3 quantum dots, highlighting the link between structural and atmospheric stability [82].

The widely used HTL material, spiro-OMeTAD, is another reason for poor thermal stability of PSCs. Spiro-OMeTAD is prone to crystallization, which affects the contact between the absorber and the HTL inside the pores, or on the interface, as the temperature elevates [83]. On the other hand, spiro-OMeTAD requires two other additives, namely, bis(trifluoromethane) sulfonamide lithium salt (Li-TFSI) and 4-tert-butylpyridine (TBP), to process. TBP will cause degradation of the perovskite layer due to the corrosive effect [84]. For the purpose of solving the thermal stability problem caused by spiro-OMeTAD, inorganic hole-transporting material [85], or other kind of additives, could be used [86].

Low dimensional perovskite materials have been the subject of wide attention in recent years, owing to their stable crystal structure, high formation energy [87], low self-doping concentrations [88], and suppressed ion migration [89]. The low dimensional perovskite structure can be visualized to cut the three dimensional (3D) perovskite structure along the $\langle 100 \rangle$ orientation, building BX_4^{2-} octahedral layers, followed by inserting bulkier organic molecules that replace the position of small molecules such as MA^+ , forming two dimensional (2D) perovskite with a sheet structure, due to the increase in the distance between the layers (Figure 8) [90]. Bulkier organic molecules and inorganic layers are combined with van Der Waals forces. This self-assembly process results in a significant increase in the formation energy, thereby enhancing the stability of perovskite. Furthermore, hydrophobic bulkier organic molecules form a natural encapsulation layer, excluding water or oxygen from the atmosphere. Therefore, low dimensional perovskite is of great significance for stable PSCs.

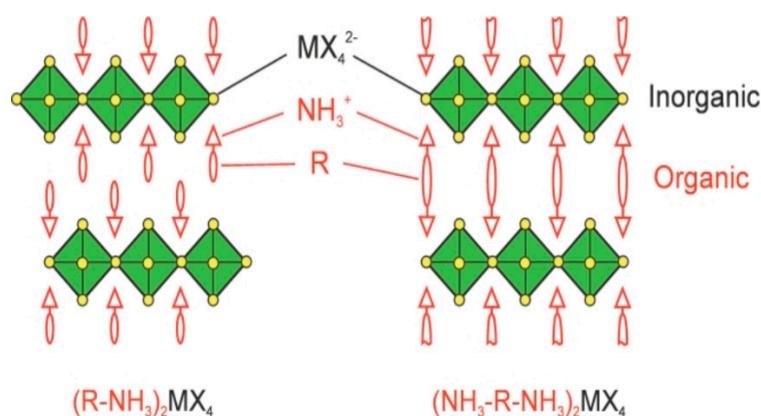


Figure 8. Schematic representation of low dimensional perovskite [90].

The number of layers of low dimensional perovskite varies by adjusting the ratio of the reactants in the precursor, and low dimensional perovskite can be easily prepared through simple one-step spin-coating. However, these layers have a large band gap [91], and bulkier organic molecules could reduce the mobility of electrons and holes [92]. These problems restrict the performance of PSCs with low dimensional materials, such that the original low dimensional PSC only achieved an efficiency

of 4.37% [93]. Finding an optimum number of inorganic layers, choosing appropriate organic amine molecules, and exploring the vertical growth of low dimensional perovskite crystals can create PSCs with a stable crystal structure and high efficiency.

4.3. UV Stability

In the PSCs, TiO₂ is frequently used as electron transport material, due to its good chemical stability, resistance to corrosion, non-toxicity, and low cost. Unfortunately, because of the redox property and band gap of 3.2 eV, TiO₂ can also serve as a photocatalytic material, oxidizing water or some organic materials to produce OH radical [94]. Leijtens et al. demonstrated that the TiO₂ layer in PSCs is susceptible to UV-induced degradation [95]. Ito et al. found that, after exposure to light, MAPbI₃ transformed to PbI₂, due to the contact between perovskite and TiO₂ [96]. The degradation in the film, caused by exposure to UV light, has been explained as follows [64]:

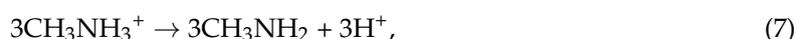


Table 4. Summary of literatures about stability of PSC.

Additive/Method	Device Structure	Stability	Ref.
Polystyrene	ITO/PTAA:F4-TCNQ/perovskite/ insulating layer/C ₆₀ /BCP/Al		[73]
ZrAcac	ITO/NiO _x /perovskite/PCBM/ZrAcac/Ag	83.6% of its initial value over 30 days (50 ± 5 RH%)	[74]
Benzylamine	FTO/compact TiO ₂ /BA-perovskite/ spiro-OMeTAD/Au	2900 h (50 ± 5 RH%)	[75]
VNPB	FTO/TiO ₂ -PCBM/perovskite/VNPB-MoO ₃ /Au	30 days (70 RH% and elevated temperatures)	[76]
Replacing cation	ITO/PEDOT:PSS/perovskite/PCBM/BCP/Al	30 min (180 °C)	[78]
Replacing cation	FTO/compact TiO ₂ /perovskite/spiro-OMeTAD/Au	60 min (150 °C)	[79]
Mixing cations	FTO/SnO ₂ -PCBM/perovskite/spiro-OMeTAD/Ag	6 h (130 °C)	[80]
Removing TiO ₂	FTO/compact TiO ₂ /mesoporous-Al ₂ O ₃ / perovskite/spiro-OMeTAD/Au	1000 h (full spectrum simulated sunlight)	[95]
Removing TiO ₂	AZO/Ga-SnO ₂ /perovskite/spiro-OMeTAD/Au	1000 h (full spectrum simulated sunlight)	[97]
Removing TiO ₂	FTO/TiS ₂ /perovskite/spiro-OMeTAD/Au	50 h (UV light)	[98]
Removing TiO ₂	ITO/HTL/perovskite/PCBM/BCP/Ag	200 min (UV light)	[99]
Blocking the UV	FTO/compact TiO ₂ /mesoporous-TiO ₂ -SM/ perovskite/HTM/Au	1000 h (UV light)	[100]
Blocking the UV	down-conversion layer/FTO/compact TiO ₂ / mesoporous-TiO ₂ /perovskite/spiro-OMeTAD/Au	100 h (UV light)	[101]

TiO₂ extracts the electron from I⁻, and then breaks the perovskite structure, creating the production of I₂, as explained by Equation (6). CH₃NH₃⁺ decomposes and releases gases of CH₃NH₂ and H⁺ (Equation (7)). I⁻ react with I₂ and H⁺, producing HI. The incessant elimination of HI and evaporation of CH₃NH₂ move Equation (6) and (7) forward, finally leading to decomposition of perovskite.

In order to overcome difficulty inducing by UV light, Snaith and co-workers replaced mesoporous TiO₂ with insulating mesoporous Al₂O₃ scaffold, showing a better stability of 1000 h with AM 1.5 illumination at 40 °C with no UV cutoff filter [95]. However, there is some initial deterioration in the FF

and V_{OC} of the cells over the first 200 h of exposure. Roose et al. employed Ga-doped mesoporous SnO_2 as a scaffold layer [97]. Gallium severely decreases the trap state density in SnO_2 , leading to a lower recombination rate and a stabilized PCE of 16.4%. PSCs based on mesoporous TiO_2 absorbed more UV light compared to Ga- SnO_2 , which can be seen in the UV/visible absorption spectra, implying a better UV stability. Some novel electron transport materials can be utilized as substitution of TiO_2 . Yin et al. applied 2D- TiS_2 as an effective electron transport layer in planar PSCs [98]. TiS_2 has similar properties to TiO_2 , including the ability of electron transportation and band alignment, ensuring high efficiency of devices. The results of Figure 9 manifest that PSCs with TiO_2 showed a strong photocatalysis, whereas PSCs with TiS_2 did not show the same phenomenon. The efficiencies of PSCs with TiS_2 were decayed by only about 10% from its maximum value after continuous illumination under UV light for 50 h. Unlike the replacement of the TiO_2 ETL, Tsai et al. introduced biaxially-extended octithiophene-based conjugated polymers which could filter UV light in inverted PSCs, in other words, the irradiation of UV light could be blocked by HTL [99]. There are some other reports that used down-conversion materials and a UV-blocking layer in PSCs to improve stability in UV light [100,101].

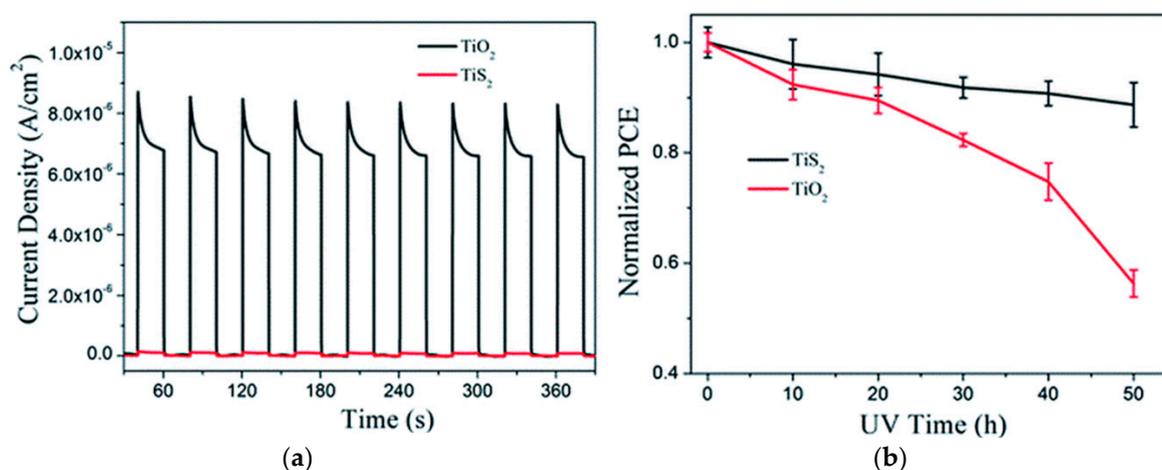


Figure 9. (a) Ultraviolet photocatalytic activity for decomposition of water in Na_2SO_4 solution by using TiO_2 and on TiS_2 film on FTO glass as photoanodes; (b) Long-term UV light stability of the PSCs with TiO_2 or TiS_2 film as the ETLs [98].

5. Routes toward Low-Cost PSCs

The trade-off between cost and efficiency is still a major obstacle for commercialization of PSCs. Although silicon solar cells that are commercialized already have an abundance of silica as raw materials, complicated manufacturing process, and efficiencies ranging between 21% and 25%, are the major drawbacks. Evolved PSCs can be prepared through a simple and low-cost solution process, making PSCs a promising substitute of silicon solar cells. The organic solid hole-transporting materials, such as spiro-OMeTAD [3], poly[bis(4-phenyl)(2,4,6-trimethylphenyl)amine] (PTAA) [102], and poly(3-hexylthiophene-2,5-diyl) (P3HT) [103], were employed because of degradation caused by liquid electrolyte, however, bring the problem of high cost to the fabrication of PSCs. It has been reported that organic HTL precursor and processing alone covers 33.9% of the device cost, due to the complicated synthesis processes (Figure 10) [104]. In the pursuit of low-cost PSCs, the inorganic hole transport materials (HTM), like CuI [105], CuSCN [106], and NiO [107–110], have been applied to replace expensive organic materials.

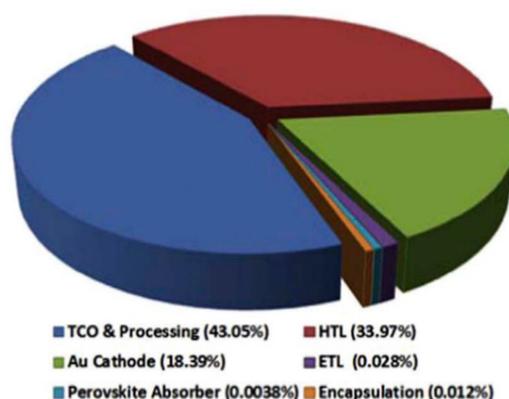


Figure 10. Price sharing of precursors of each layer of a conventional PSC [104].

5.1. Replacement of Costly Organic Hole-Transporting Materials and Hole-Conductor-Free PSCs

Materials of HTL can be divided into inorganic, organic, and hybrid materials, while organic materials can be further subdivided into small molecule and polymer [111]. Following on from the experience of solid state dye-sensitized solar cells, of the organic materials currently used to make the best devices, spiro-OMeTAD has long been the benchmark HTM against which other materials have been compared. However, its cost is about 5 or 6 times higher than the gold, so low-cost HTM is in great demand. The development in inorganic HTMs has been rather slow compared to organic HTMs, and only a limited number of inorganic materials have been tested. Copper salts, like CuI, Cu₂O, and CuSCN, and NiO, which is most widely-studied inorganic HTM, are relatively inexpensive and stable under ambient conditions; the copper salts do not require doping to achieve good hole mobilities, and the devices show good stability. Nevertheless, PSCs using inorganic HTMs cannot gain a sufficient PCE compared with PSCs based on organic HTMs, and the solvents used for inorganic HTMs can partially dissolve the active layer, thus causing device instability. Moreover, some materials, while inexpensive, must be deposited by more expensive methods, such as sputtering, and the extra fabrication costs are likely to cancel out any advantages in price of materials.

Perovskite materials offer an electrically conductive path for both p-type and n-type charge carriers [112–114], and electrons and holes both have a long diffusion length inside the perovskite [115,116], which means the perovskite layer can transport holes effectively, even if without HTL. From the view of band alignment, as long as electrode material satisfies matching the conditions of energy level and work function, that is, the HOMO is higher than the valence band of the perovskite (~5.3 eV), the holes can be effectively transported, and the band alignment of hole-conductor-free device-based Au electrode is shown in Figure 11a. Therefore, it is theoretically feasible to remove the HTL directly.

In 2012, Grätzel and co-workers reported PSC without HTL for the first time, and exhibited a promising PCE of 5.5% [117]. The synthesis of MAPbI₃ and deposition on the mesoporous TiO₂ film was carried out by spin-coating, and then Au was deposited directly onto the perovskite layer by thermal evaporation. Etgar et al. prepared hole-conductor-free PSCs, and further enhanced the efficiency to 10.85% [118]. Ma et al. fabricated a hole-conductor-free PSC based on CsPbIBr₂, achieving a PCE of 4.7% and excellent thermal stability [119].

Metal–semiconductor contact is formed in PSCs due to the removal of HTL, which is detrimental to the separation of electrons and holes in the interface. The direct contact between the perovskite layer and Au electrode leads to an energy barrier of about 0.24 eV, as well as a depleted region over 60 nm existing in the back contact region [120], causing a reversed effect in transportation of carriers and low internal quantum efficiency (IQE) in the long-wavelength region. Furthermore, interface recombination, caused by defects, also affects the performance of hole-conductor-free PSCs. Homogeneous perovskite film and thicker mesoporous layers are always required so as to form better metal–semiconductor contact and avoid short circuits. Shi et al. increased the concentration of PbI₂

in spin-coating precursors, reduced the spin-coating speed, and then prepared a uniform film with appropriate thickness. The metal electrode was directly evaporated onto perovskite, achieving a PCE of 10.47% [120].

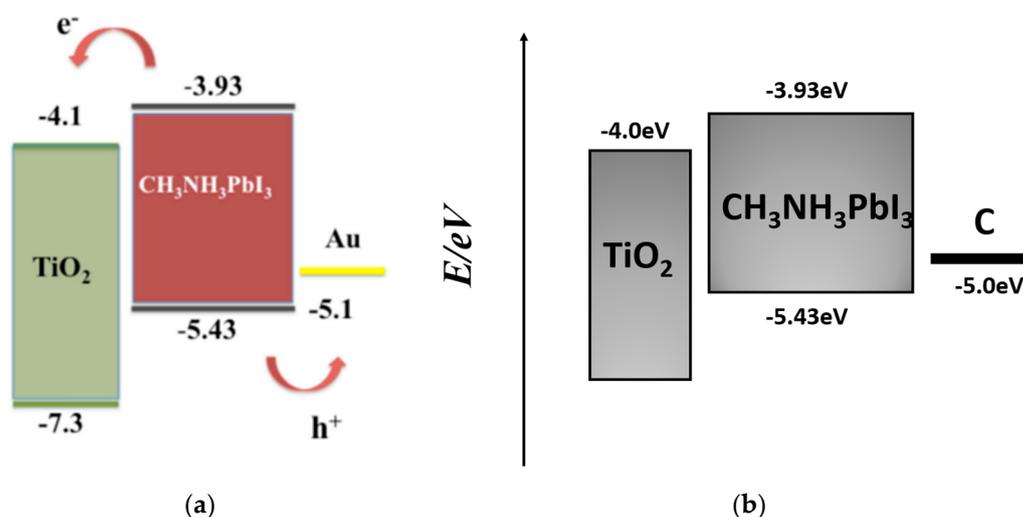


Figure 11. Band alignment of hole-conductor-free device-based (a) Au electrode [117] and (b) C electrode.

The introduction of isolated layers can overcome recombination in the interface between electrode and perovskite. Wei et al. introduced ultrathin insulator AlO_x film into the MAPbI_3/Au interface by atomic layer deposition (ALD) technology, and the ALD deposition cycles were further optimized [121]. The average PCE of the cell has been enhanced from 8.61% to 10.07%, with the highest PCE of 11.10%. The presence of AlO_x significantly enhanced the IQE of the cell in the long wavelength region, leading to the increase in photocurrent. Xu et al. employed N,N,N',N' -tetraphenyl-benzidine (TPB) by spin-coating to engineer the metal–semiconductor interface in the hole-conductor-free PSCs [122], and the thin TPB layer increased recombination resistance, resulting in a lower recombination rate in the interface.

5.2. Carbon Electrode for PSCs

Another bottleneck in PSC is the high cost and complexity involved in the Au or Ag back electrode deposition. Noble metal and processing covers 18.3% of the device cost (Figure 10) [104]. The appropriate 5.0 eV work function, near to that of Au (5.1 eV), makes carbon an ideal candidate for PSCs' back electrode, so, similar to the PSCs base on noble metal, hole-conductor-free PSCs using carbon as electrode can transport holes effectively without HTL (Figure 11b). Being abundantly available on the earth, carbon is also known for good conductivity, electrochemical activity, water resistance, high chemical stability, and flexibility in processing, leading to low-cost and high-efficiency PSCs [123]. Undoubtedly, carbon is an attractive choice for back electrode in hole-conductor-free PSCs.

To the best of our knowledge, the highest efficiency achieved by hole-conductor-free PSCs utilizing carbon as back electrode is 15.9% [124]. According to the different bonded patterns of carbon atoms and different dimensions of crystalline structures, there are several allotropes with very different chemical and physical properties, as shown in Figure 12. Carbon nanotubes (CNTs) have a particular 1D hollow tubular structure, which can be considered as the result of rolling up a monolayer graphite sheet, resulting in a strong ability charge absorption [125]. The CNTs have good conductivity and chemical stability, and the electrical properties can be adjusted by chemical doping [126]. Nanoscale CNTs can also be applied to flexible PSCs, building wearable photovoltaic devices, meanwhile, CNTs have a good transmittance of sunlight, causing more incident light for PSCs. Single-walled carbon nanotubes (SWCNTs) and multi-walled carbon nanotubes (MWCNTs) are the most common species of CNTs.

Graphite is a soft and hexagonal laminated material, which has an outstanding thermal and electrical conductivity and high corrosion resistance, however, it is often mixed with amorphous carbon black with a large specific surface to form carbon paste, in order to contact effectively with perovskite layer. Graphene could be employed in all layers of PSCs, which is called a “universal material”. Apparently, graphene is a promising candidate for noble metals, owing to the good chemical and thermal stability and high carrier mobility.

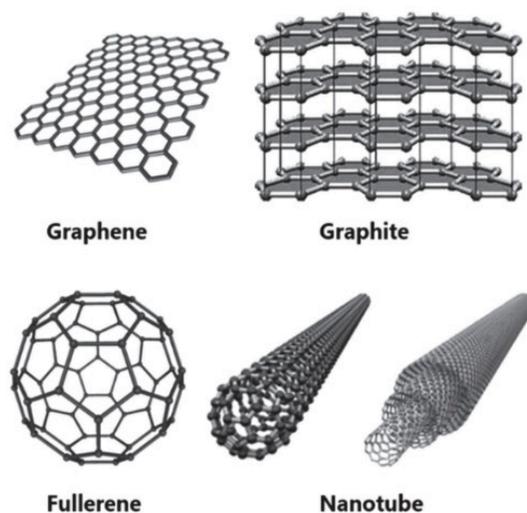


Figure 12. Structure of various carbon materials [127].

Li et al. introduced CNTs in PSCs as back electrode, for the first time, in 2014 [126]. They applied the floating catalyst chemical vapor deposition method to fabricate laminated CNT networks, and then introduced it in hole-conductor-free PSCs, obtaining a PCE of 6.87%. By adding spiro-OMeTAD to form a composite electrode, the efficiency of this device can be further improved to 9.9%, due to the better enhanced hole-extraction and reduced charge recombination. Inspired by this work, plenty of studies mixed CNTs with hole transport materials [128–131], whereafter, some researchers utilized CNT alone as back electrode, by optimizing the device structure and preparation process [132–134].

In 2013, Han et al. established a hole-conductor-free $\text{MAPbI}_3/\text{TiO}_2$ heterojunction solar cell, which first introduced carbon black/spheroidal graphite as a back electrode to substitute the noble metallic electrode [13]. Mesoporous TiO_2 and isolated ZrO_2 with a thickness of 1 μm was screen-printed on the top of the TiO_2 compact layer, then, the 10 μm carbon back electrode was deposited followed by dipping of MAPbI_3 precursor on top of the mesoscopic carbon layer, creating a structure of FTO/ TiO_2 compact layer/ TiO_2 mesoporous layer/ ZrO_2 isolating layer/carbon back electrode. This kind of structure is a so-called monolithic structure, as shown in Figure 2c, with an impressive PCE, exceeding 6.64%, and the device showed a long-term stability in the dark after 840 h. The sufficient infiltration of the perovskite precursor [135], quality of the perovskite materials [136], the size of TiO_2 particles [137], and the thickness of ZrO_2 [138], all affect the performance of monolithic PSCs.

The mass ratio of the composite materials of graphite and carbon black significantly influences the contact between the back electrode and the perovskite layer and electrical conductivity of the electrode itself. Graphite has good electrical conductivity, and plays a major role in the process of carrier conduction, but excess content of graphite will cause insufficient internal contact, thereby increasing electrical resistivity. Therefore, a moderate amount of carbon black is employed to obtain a better contact between graphite flakes [139]. The perovskite materials are filled in the porous carbon electrode of monolithic PSCs mentioned above, and an appropriate pore size can also determine the performance of monolithic PSCs. Zhang et al. found that 8 μm graphite is suitable for PSC electrodes and the optimum electrodes have a smaller square resistance and, hence, help the PCE exceed 11% [140].

Graphite, carbon black, and binder additives can be mixed to fabricate the commercial conductive carbon pastes which are widely used in PSCs. Tao et al. reported a carbon back electrode assisted by polystyrene spheres (PS-spheres) as pore-forming agent to gain microporous structure, for a better permeability with an average efficiency enhancement of 22% [141]. Liu et al. applied low-temperature carbon back electrode in fabricating large-area hole-conductor-free PSCs, achieving a PCE of 3.3%, and with no obvious decay during a testing period of 624 h [142]. Zhang et al. fabricated a monolithic PSC using carbon as back electrodes and TiO₂ nanosheet layers as photoanode. With the increasing thickness of TiO₂ nanosheet layer, more MAPbI₃ was adsorbed into the TiO₂ nanosheet photoanode, and the interfacial recombination of electrons and holes at TiO₂ nanosheet/perovskite interface was reduced, leading to the higher J_{SC} and V_{OC} [143]. Hari Bala and co-workers optimized the doping ratio of Br in perovskite, making the device retain 96.23% of the initial efficiency after being stored in the air for 56 days (1344 h) [144]. Li et al. prepared Ag/nano-TiO₂ composites and introduced them into the compact film of conventional hole-conductor-free PSCs. The strong scattering effect and the surface plasmon resonance effect of the metallic Ag nanoparticles strengthened the optical absorption of the PSCs in the visible light region, and enhanced the efficiency of electron injection, resulting in a 30% enhancement in PCE [145].

In 2004, Novoselov et al. prepared graphene by mechanical exfoliation (repeated peeling) of small mesas of graphite [146]. At present, graphene is widely used in the fields of chemistry and materials science, and plays a more and more vital part in the photovoltaic field. Snaith and co-workers employed graphene to PSCs, firstly, and utilized nanocomposites of graphene and TiO₂ nanoparticles as the electron collection layers in mesoporous PSCs, providing superior charge-collection ability [147]. Not only being used in the ETL, graphene can be a functional material of transparent electrodes [148–150] and HTL [151–153]. In terms of the back electrodes, Yan et al. synthesized single-layered graphene (SG) and multi-layered graphene (MG) from the reduction of their oxide at 1000 °C under an argon atmosphere, and then utilized these graphene materials as back electrodes [154]. After detailed investigation, they found that MG has higher hole extraction rates than SG. PSCs used MG as HTL yield about 11.5% PCE, with FF as high as 0.73, whereas SG devices yield 6.7% and 0.54. Wei et al. synthesized three-dimensional honeycomb-like structured graphene (3DHG) by the reaction of potassium with CO₂, which is coated on a perovskite layer as a back electrode with the doctor-blade method [155]. The electrical conductivity of 3DHG can be enhanced through increasing synthesis time, thereby boosting the performance, and hysteresis was not observed for the PSCs with the 3DHG back electrode. Zhu et al. synthesized high specific surface area nitrogen-doped graphene frameworks by using a facile one-step fast pyrolysis [156]. The PCE of 10.32% is achieved, higher than devices with a purely graphene framework as back electrode. The enhancement can be attributed to the increased charge-extraction efficiency and transportation ability, which reduce charge-recombination losses. Additionally, the large light transmittance of graphene makes it an excellent electrode material for top cells of tandem photovoltaics [157–159].

5.3. Scalable Deposition Methods

High-performance PSCs are fabricated through solution processing, which has not been demonstrated for other photovoltaic technologies. This makes PSCs promising candidates for a high-performance, low-cost photovoltaic technology. Many researchers turn to large-area PSCs, and even perovskite modules, in recent years, to further accelerate the process of commercialization. A perovskite module with a certified PCE of 16% and an area larger than 16 cm² has recently been demonstrated [160]. Apparently, PSCs with larger area are crucial for reducing the cost of perovskite modules. Spin-coating is the most common method for perovskite thin-film deposition. However, it is generally limited to a small scale of deposition, and a large portion (more than 90%) of the precursor ink is wasted in the process. In spin-coating, high-quality wet-solution perovskite films rely on the continuous centrifugal force from spinning, which is difficult to reproduce in scalable deposition processes, thus, a great deal of scalable coating methods have emerged.

5.3.1. Doctor-Blade Coating

The doctor-blade method uses a blade to spread precursor solution on substrates and form a wet thin film (Figure 13a). The film thickness and crystallization are generally controlled by the concentration of the precursor ink, the gap between the blade and substrate, and the speed at which the blade moves across the substrate. The fabrication process and equipment can be simplified by the doctor-blade method, and the ink waste is obviously reduced compared with spin-coating method. Many studies have employed the doctor-blade method to prepare large area PSCs [161–165] with optimized PCE greater than 19% [166].

5.3.2. Slot-Die Coating

Slot-die coating is similar to doctor-blade coating, and uses a die as ink reservoir to spread liquid through a slot (Figure 13b). Ink flow can be better controlled than doctor-blade coating, but this method normally requires larger quantities of ink to fill the ink reservoir and supply pipe and, thus, has been less explored and, at present, the PSCs prepared using a slot-die coating method have a much lower PCE than those fabricated through doctor-blade coating. However, the slot-die coating has been developed for many years, and is suitable for roll-to-roll deposition, which provides a better future for large-scale PSC fabrication [167–169]. Burkitt et al. achieved four slot-die coated layers including perovskite, HTL, ETL, and mesoporous layer, and the process is greatly simplified [170]. Kim et al. developed a single-step slot-die coating, and used gas quenching and substrate heating to achieve a defect-free dense perovskite layer, resulting in a PCE of 12.7% [171].

5.3.3. Spray Coating

Spray coating disperses tiny liquid droplets onto substrates through a nozzle (Figure 13c). According to different ways of dispersing the droplets, it can be further classified, such as pneumatic spraying (through fast gas flow), ultrasonic spraying (through ultrasonic vibration), or electrospraying (through electrical repulsion). Among these spray methods, ultrasonic spraying is commonly introduced to fabricate perovskite layer and ETL [172–175]. Several droplets need to overlap in a local area, in other words, multiple spraying needs to be carried out to ensure full coverage. The substrate should be kept at an elevated temperature during spraying, in order to increase the rate of solvent evaporation and suppress the redissolution of deposited material.

5.3.4. Inkjet Printing and Screen Printing

In inkjet printing process, nozzles with a piezo material are used to disperse the precursor ink, with fine control of the droplet size and pattern (Figure 13d). Inkjet printing can be applied to prepare mesoporous [176,177] and monolithic hole-conductor-free PSCs [178,179], which can be easily combined with a roll-to-roll process. Screen printing always needs a patterned mesh screen to hold and transfer ink to the substrate (Figure 13e). This method has good patterning ability, with a lateral resolution of around 75–100 μm [180]. Screen printing is usually used to deposit thicker films, thus, it is favored method to form mesoporous layers and carbon back electrodes [181,182].

5.3.5. Vapor-Phase Deposition

The vapor-phase method is common in solar cell fabrication, such as in the preparation of CuInGaSe_2 (CIGS) and CdTe , which relies heavily on physical vapor deposition. Liu et al. employed vapor-phase method to the fabrication of perovskite film for the first time [183]. They use dual-source vapor deposition to create uniform flat films with high coverage, compared to films prepared through a solution process. The dual-source thermal evaporation system is shown in Figure 13f. Subsequently, a lot of reports combined solution-phase and vapor-phase method to form high-quality perovskite films without sophisticated vacuum equipment [184].

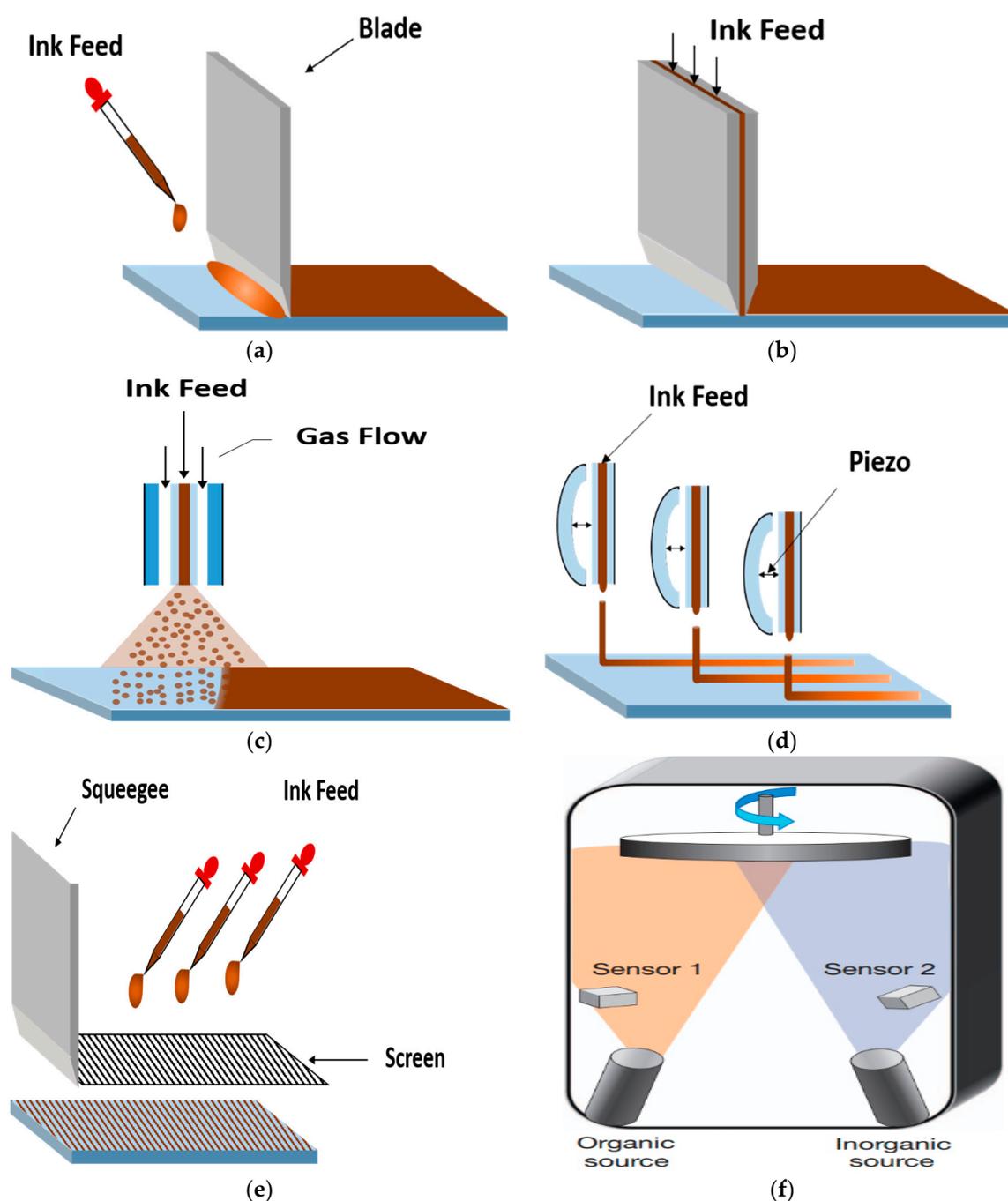


Figure 13. Scalable deposition methods: (a) Doctor-blade coating; (b) Slot-die coating; (c) Spray coating; (d) Inkjet printing; (e) Screen printing; (f) Vapor-phase deposition [183].

6. Summary and Future Outlook

PSCs have made great progress in the past nine years, thanks to the highlighted properties of organic–inorganic hybrid halide perovskite materials. Despite this rapid development, PSCs are still not comparable to other commercialized photovoltaics, such as crystalline silicon and CIGS, which we attributed to three aspects, and gave a detailed description. Firstly, there is still much room for improvement in the efficiency. Optimizing the chemical composition of the perovskite materials, introducing suitable additives to passivate the grain boundaries of the perovskite polycrystalline films, and modifying the interface by using some interlayers or passivator through interfacial engineering,

can effectively improve the optoelectronic properties of perovskite materials and transportation of carriers in the interfaces, thereby enhancing the efficiency of PSCs. Secondly, the stability of PSCs has not been well solved. Moisture, oxygen, and thermal instability can be alleviated by adjusting the composition of the perovskite and, moreover, the application of low-dimensional perovskite materials is another solution toward these issues. Remarkably, the intrinsic moisture and oxygen problem can be overcome by application of some isolating layers and additives. The poor stability caused by UV light is generally due to the presence of TiO₂, which can be settled by employing other kinds of ETL materials, and directly isolating the UV light. Finally, although perovskite has abundant low-cost raw materials, expensive organic hole transport materials and noble metals still make PSCs a high-cost photovoltaic device. For the HTL, inorganic materials can be an applicable replacement. The HTL can also be directly removed to fabricate hole-conductor-free PSCs. Regarding metal electrodes, a variety of carbon electrodes can be utilized to lower the cost of PSCs. When PSCs enter commercial large-scale production, a larger area of perovskite apparently increases productivity and reduces the cost of PSC modules. Therefore, we briefly overviewed common scalable deposition methods in which inkjet printing and screen printing represent well-developed technologies which can be easily introduced into a roll-to-roll process, making it the most promising method for large-scale production. PSCs can make a great step forward in commercialization on the condition that all the difficulties, described above, are completely resolved.

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References

1. Kojima, A.; Teshima, K.; Shirai, Y.; Miyasaka, T. Organometal Halide Perovskites as Visible-Light Sensitizers for Photovoltaic Cells. *J. Am. Chem. Soc.* **2009**, *131*, 6050–6051. [[CrossRef](#)] [[PubMed](#)]
2. Im, J.H.; Lee, C.R.; Lee, J.W.; Park, S.W.; Park, N.G. 6.5% Efficient Perovskite Quantum-Dot-Sensitized Solar Cell. *Nanoscale* **2011**, *3*, 4088–4093. [[CrossRef](#)] [[PubMed](#)]
3. Kim, H.S.; Lee, C.R.; Im, J.H.; Lee, K.B.; Moehl, T.; Marchioro, A.; Moon, S.J.; Humphry-Baker, R.; Yum, J.H.; Moser, J.E.; et al. Lead Iodide Perovskite Sensitized All-Solid-State Submicron Thin Film Mesoscopic Solar Cell with Efficiency Exceeding 9%. *Sci. Rep.* **2012**, *2*, 591. [[CrossRef](#)] [[PubMed](#)]
4. Lee, M.M.; Teuscher, J.; Miyasaka, T.; Murakami, T.N.; Snaith, H.J. Efficient Hybrid Solar Cells Based on Meso-Superstructured Organometal Halide Perovskites. *Science* **2012**, *338*, 643–647. [[CrossRef](#)] [[PubMed](#)]
5. Yang, W.S.; Park, B.W.; Jung, E.H.; Jeon, N.J.; Kim, Y.C.; Lee, D.U.; Shin, S.S.; Seo, J.; Kim, E.K.; Noh, J.H.; et al. Iodide Management in Formamidinium-Lead-Halide-Based Perovskite Layers for Efficient Solar Cells. *Science* **2017**, *356*, 1376–1379. [[CrossRef](#)] [[PubMed](#)]
6. NREL Chart. Available online: <https://www.nrel.gov/pv/assets/pdfs/pv-efficiencies-07-17-2018.pdf> (accessed on 17 July 2018).
7. Ting, H.K.; Ni, L.; Ma, S.B.; Ma, Y.Z.; Xiao, L.X.; Chen, Z.J. Progress in Electron-Transport Materials in Application of Perovskite Solar Cells. *Acta Phys. Sin.* **2015**, *64*, 038802. [[CrossRef](#)]
8. Ma, Y.Z.; Wang, S.F.; Zheng, L.L.; Lu, Z.L.; Zhang, D.F.; Bian, Z.Q.; Huang, C.H.; Xiao, L.X. Recent Research Developments of Perovskite Solar Cells. *Chin. J. Chem.* **2014**, *32*, 957–963. [[CrossRef](#)]
9. De Wolf, S.; Holovsky, J.; Moon, S.J.; Loper, P.; Niesen, B.; Ledinsky, M.; Haug, F.J.; Yum, J.H.; Ballif, C. Organometallic Halide Perovskites: Sharp Optical Absorption Edge and Its Relation to Photovoltaic Performance. *J. Phys. Chem. Lett.* **2014**, *5*, 1035–1039. [[CrossRef](#)] [[PubMed](#)]
10. Stranks, S.D.; Eperon, G.E.; Grancini, G.; Menelaou, C.; Alcocer, M.J.P.; Leijtens, T.; Herz, L.M.; Petrozza, A.; Snaith, H.J. Electron-Hole Diffusion Lengths Exceeding 1 Micrometer in An Organometal Trihalide Perovskite Absorber. *Science* **2013**, *342*, 341–344. [[CrossRef](#)] [[PubMed](#)]

11. Valverde-Chavez, D.A.; Ponseca, C.S.; Stoumpos, C.C.; Yartsev, A.; Kanatzidis, M.G.; Sundstrom, V.; Cooke, D.G. Intrinsic Femtosecond Charge Generation Dynamics in Single Crystal $\text{CH}_3\text{NH}_3\text{PbI}_3$. *Energy Environ. Sci.* **2015**, *8*, 3700–3707. [[CrossRef](#)]
12. Miyata, A.; Mitioglu, A.; Plochocka, P.; Portugall, O.; Wang, J.T.W.; Stranks, S.D.; Snaith, H.J.; Nicholas, R.J. Direct Measurement of The Exciton Binding Energy and Effective Masses for Charge Carriers in Organic-Inorganic Tri-Halide Perovskites. *Nat. Phys.* **2015**, *11*, 582–588. [[CrossRef](#)]
13. Ku, Z.L.; Rong, Y.G.; Xu, M.; Liu, T.F.; Han, H.W. Full Printable Processed Mesoscopic $\text{CH}_3\text{NH}_3\text{PbI}_3/\text{TiO}_2$ Heterojunction Solar Cells with Carbon Counter Electrode. *Sci. Rep.* **2013**, *3*, 3132. [[CrossRef](#)] [[PubMed](#)]
14. Pellet, N.; Gao, P.; Gregori, G.; Yang, T.Y.; Nazeeruddin, M.K.; Maier, J.; Gratzel, M. Mixed-Organic-Cation Perovskite Photovoltaics for Enhanced Solar-Light Harvesting. *Angew. Chem.-Int. Edit.* **2014**, *53*, 3151–3157. [[CrossRef](#)] [[PubMed](#)]
15. Yang, W.S.; Noh, J.H.; Jeon, N.J.; Kim, Y.C.; Ryu, S.; Seo, J.; Seok, S.I. High-Performance Photovoltaic Perovskite Layers Fabricated Through Intramolecular Exchange. *Science* **2015**, *348*, 1234–1237. [[CrossRef](#)] [[PubMed](#)]
16. Ogomi, Y.; Morita, A.; Tsukamoto, S.; Saitho, T.; Fujikawa, N.; Shen, Q.; Toyoda, T.; Yoshino, K.; Pandey, S.S.; Ma, T.L.; et al. $\text{CH}_3\text{NH}_3\text{Sn}_x\text{Pb}_{(1-x)}\text{I}_3$ Perovskite Solar Cells Covering up to 1060 nm. *J. Phys. Chem. Lett.* **2014**, *5*, 1004–1011. [[CrossRef](#)] [[PubMed](#)]
17. Stoumpos, C.C.; Malliakas, C.D.; Kanatzidis, M.G. Semiconducting Tin and Lead Iodide Perovskites with Organic Cations: Phase Transitions, High Mobilities, and Near-infrared Photoluminescent Properties. *Inorg. Chem.* **2013**, *52*, 9019–9038. [[CrossRef](#)] [[PubMed](#)]
18. Noh, J.H.; Im, S.H.; Heo, J.H.; Mandal, T.N.; Seok, S.I. Chemical Management for Colorful, Efficient, and Stable Inorganic-Organic Hybrid Nanostructured Solar Cells. *Nano Lett.* **2013**, *13*, 1764–1769. [[CrossRef](#)] [[PubMed](#)]
19. Shockley, W.; Queisser, H.J. Detailed Balance Limit of Efficiency of p-n Junction Solar Cells. *J. Appl. Phys.* **1961**, *32*, 510–519. [[CrossRef](#)]
20. Choi, H.; Jeong, J.; Kim, H.B.; Kim, S.; Walker, B.; Kim, G.H.; Kim, J.Y. Cesium-Doped Methylammonium Lead Iodide Perovskite Light Absorber for Hybrid Solar Cells. *Nano Energy* **2014**, *7*, 80–85. [[CrossRef](#)]
21. Jeon, N.J.; Noh, J.H.; Yang, W.S.; Kim, Y.C.; Ryu, S.; Seo, J.; Seok, S.I. Compositional Engineering of Perovskite Materials for High-Performance Solar Cells. *Nature* **2015**, *517*, 476–480. [[CrossRef](#)] [[PubMed](#)]
22. Li, Z.; Yang, M.J.; Park, J.S.; Wei, S.H.; Berry, J.J.; Zhu, K. Stabilizing Perovskite Structures by Tuning Tolerance Factor: Formation of Formamidinium and Cesium Lead Iodide Solid-State Alloys. *Chem. Mat.* **2015**, *28*, 284–292. [[CrossRef](#)]
23. Lee, J.W.; Kim, D.H.; Kim, H.S.; Seo, S.W.; Cho, S.M.; Park, N.G. Formamidinium and Cesium Hybridization for Photo- and Moisture-Stable Perovskite Solar Cell. *Adv. Energy Mater.* **2015**, *5*, 1501310. [[CrossRef](#)]
24. Yi, C.Y.; Luo, J.S.; Meloni, S.; Boziki, A.; Ashari-Astani, N.; Gratzel, C.; Zakeeruddin, S.M.; Rothlisberger, U.; Gratzel, M. Entropic Stabilization of Mixed A-Cation ABX_3 Metal Halide Perovskites for High Performance Perovskite Solar Cells. *Energy Environ. Sci.* **2016**, *9*, 656–662. [[CrossRef](#)]
25. Bi, D.Q.; Tress, W.; Dar, M.I.; Gao, P.; Luo, J.S.; Renevier, C.; Schenk, K.; Abate, A.; Giordano, F.; Baena, J.P.C.; et al. Efficient Luminescent Solar Cells Based on Tailored Mixed-Cation Perovskites. *Sci. Adv.* **2016**, *2*, e1501170. [[CrossRef](#)] [[PubMed](#)]
26. Jeon, N.J.; Na, H.; Jung, E.H.; Yang, T.Y.; Lee, Y.G.; Kim, G.; Shin, H.W.; Seok, S.I.; Lee, J.; Seo, J. A Fluorene-Terminated Hole-Transporting Material for Highly Efficient and Stable Perovskite Solar Cells. *Nat. Energy* **2018**, *3*, 682–689. [[CrossRef](#)]
27. Eperon, G.E.; Paterno, G.M.; Sutton, R.J.; Zampetti, A.; Haghighirad, A.A.; Cacialli, F.; Snaith, H.J. Inorganic Caesium Lead Iodide Perovskite Solar Cells. *J. Mater. Chem. A* **2015**, *3*, 19688–19695. [[CrossRef](#)]
28. Saliba, M.; Matsui, T.; Seo, J.Y.; Domanski, K.; Correa-Baena, J.P.; Nazeeruddin, M.K.; Zakeeruddin, S.M.; Tress, W.; Abate, A.; Hagfeldt, A.; et al. Cesium-Containing Triple Cation Perovskite Solar Cells: Improved Stability, Reproducibility and High Efficiency. *Energy Environ. Sci.* **2016**, *9*, 1989–1997. [[CrossRef](#)] [[PubMed](#)]
29. Peng, J.; Duong, T.; Zhou, X.Z.; Shen, H.P.; Wu, Y.L.; Mulmudi, H.K.; Wan, Y.M.; Zhong, D.Y.; Li, J.T.; Tsuzuki, T.; et al. Efficient Indium-Doped TiO_x Electron Transport Layers for High-Performance Perovskite Solar Cells and Perovskite-Silicon Tandems. *Adv. Energy Mater.* **2016**, *7*, 1601768. [[CrossRef](#)]

30. Saliba, M.; Matsui, T.; Domanski, K.; Seo, J.Y.; Ummadisingu, A.; Zakeeruddin, S.M.; Correa-Baena, J.P.; Tress, W.R.; Abate, A.; Hagfeldt, A.; et al. Incorporation of Rubidium Cations into Perovskite Solar Cells Improves Photovoltaic Performance. *Science* **2016**, *354*, 206–209. [[CrossRef](#)] [[PubMed](#)]
31. Park, Y.H.; Jeong, I.; Bae, S.; Son, H.J.; Lee, P.; Lee, J.; Lee, C.H.; Ko, M.J. Inorganic Rubidium Cation as an Enhancer for Photovoltaic Performance and Moisture Stability of $\text{HC}(\text{NH}_2)_2\text{PbI}_3$ Perovskite Solar Cells. *Adv. Funct. Mater.* **2017**, *27*, 1605988. [[CrossRef](#)]
32. Hao, F.; Stoumpos, C.C.; Chang, R.P.H.; Kanatzidis, M.G. Anomalous Band Gap Behavior in Mixed Sn and Pb Perovskites Enables Broadening of Absorption Spectrum in Solar Cells. *J. Am. Chem. Soc.* **2014**, *136*, 8094–8099. [[CrossRef](#)] [[PubMed](#)]
33. Kulkarni, S.A.; Baikie, T.; Boix, P.P.; Yantara, N.; Mathews, N.; Mhaisalkar, S. Band-Gap Tuning of Lead Halide Perovskites Using a Sequential Deposition Process. *J. Mater. Chem. A* **2014**, *2*, 9221–9225. [[CrossRef](#)]
34. Rohatgi, A.; Kim, D.S.; Nakayashiki, K.; Yelundur, V.; Rounsaville, B. High-Efficiency Solar Cells on Edge-Defined Film-Fed Grown (18.2%) and String Ribbon (17.8%) Silicon by Rapid Thermal Processing. *Appl. Phys. Lett.* **2004**, *84*, 145–147. [[CrossRef](#)]
35. Yan, Y.F.; Yin, W.J.; Wu, Y.L.; Shi, T.T.; Paudel, N.R.; Li, C.; Poplawsky, J.; Wang, Z.W.; Moseley, J.; Guthrey, H.; et al. Physics of Grain Boundaries in Polycrystalline Photovoltaic Semiconductors. *J. Appl. Phys.* **2015**, *117*, 112807. [[CrossRef](#)]
36. Yan, Y.; Jiang, C.S.; Noufi, R.; Wei, S.H.; Moutinho, H.R.; Al-Jassim, M.M. Electrically Benign Behavior of Grain Boundaries in Polycrystalline CuInSe_2 Films. *Phys. Rev. Lett.* **2007**, *99*, 235504. [[CrossRef](#)] [[PubMed](#)]
37. Guo, Y.G.; Wang, Q.; Saidi, W.A. Structural Stabilities and Electronic Properties of High-Angle Grain Boundaries in Perovskite Cesium Lead Halides. *J. Phys. Chem. C* **2017**, *121*, 1715–1722. [[CrossRef](#)]
38. Lee, J.W.; Park, N.G. Two-Step Deposition Method for High-Efficiency Perovskite Solar Cells. *MRS Bull.* **2015**, *40*, 654–659. [[CrossRef](#)]
39. Roldan-Carmona, C.; Gratia, P.; Zimmermann, I.; Grancini, G.; Gao, P.; Graetzel, M.; Nazeeruddin, M.K. High Efficiency Methylammonium Lead Triiodide Perovskite Solar Cells: The Relevance of Non-Stoichiometric Precursors. *Energy Environ. Sci.* **2015**, *8*, 3550–3556. [[CrossRef](#)]
40. Kim, Y.C.; Jeon, N.J.; Noh, J.H.; Yang, W.S.; Seo, J.; Yun, J.S.; Ho-Baillie, A.; Huang, S.J.; Green, M.A.; Seidel, J.; et al. Beneficial Effects of PbI_2 Incorporated in Organo-Lead Halide Perovskite Solar Cells. *Adv. Energy Mater.* **2016**, *6*, 1502104. [[CrossRef](#)]
41. Wang, H.Y.; Hao, M.Y.; Han, J.; Yu, M.; Qin, Y.J.; Zhang, P.; Guo, Z.X.; Ai, X.C.; Zhang, J.P. The Adverse Effects of Excessively Remained PbI_2 on Photovoltaic Performance, Charge Separation and Trap State Properties in Mesoporous Structured Perovskite Solar Cells. *Chem.-Eur. J.* **2017**, *23*, 3986–3992. [[CrossRef](#)] [[PubMed](#)]
42. Hsu, H.Y.; Ji, L.; Du, M.S.; Zhao, J.; Yu, E.T.; Bard, A.J. Optimization of $\text{PbI}_2/\text{MAPbI}_3$ Perovskite Composites by Scanning Electrochemical Microscopy. *J. Phys. Chem. C* **2016**, *120*, 19890–19895. [[CrossRef](#)]
43. Son, D.Y.; Lee, J.W.; Choi, Y.J.; Jang, I.H.; Lee, S.; Yoo, P.J.; Shin, H.; Ahn, N.; Choi, M.; Kim, D.; et al. Self-Formed Grain Boundary Healing Layer for Highly Efficient $\text{CH}_3\text{NH}_3\text{PbI}_3$ Perovskite Solar Cells. *Nat. Energy* **2016**, *67*, 16081. [[CrossRef](#)]
44. Li, X.; Bi, D.Q.; Yi, C.Y.; Decoppet, J.D.; Luo, J.S.; Zakeeruddin, S.M.; Hagfeldt, A.; Gratzel, M. A Vacuum Flash-Assisted Solution Process for High-Efficiency Large-Area Perovskite Solar Cells. *Science* **2016**, *353*, 58–62. [[CrossRef](#)] [[PubMed](#)]
45. Saliba, M.; Orlandi, S.; Matsui, T.; Aghazada, S.; Cavazzini, M.; Correa-Baena, J.P.; Gao, P.; Scopelliti, R.; Mosconi, E.; Dahmen, K.H.; et al. A Molecularly Engineered Hole-Transporting Material for Efficient Perovskite Solar Cells. *Nat. Energy* **2016**, *1*, 15017. [[CrossRef](#)]
46. Zheng, Y.Z.; Li, X.T.; Zhao, E.F.; Lv, X.D.; Meng, F.L.; Peng, C.; Lai, X.S.; Huang, M.L.; Cao, G.Z.; Tao, X.; et al. Hexamethylenetetramine-Mediated Growth of Grain-Boundary-Passivation $\text{CH}_3\text{NH}_3\text{PbI}_3$ for Highly Reproducible and Stable Perovskite Solar Cells. *J. Power Sources* **2017**, *377*, 103–109. [[CrossRef](#)]
47. Tong, Y.H.; Liu, Y.C.; Dong, L.; Zhao, D.X.; Zhang, J.Y.; Lu, Y.M.; Shen, D.Z.; Fan, X.W. Growth of ZnO Nanostructures with Different Morphologies by Using Hydrothermal Technique. *J. Phys. Chem. B* **2006**, *110*, 20263–20267. [[CrossRef](#)] [[PubMed](#)]
48. Li, X.; Chen, C.C.; Cai, M.L.; Hua, X.; Xie, F.X.; Liu, X.; Hua, J.L.; Long, Y.T.; Tian, H.; Han, L.Y. Efficient Passivation of Hybrid Perovskite Solar Cells Using Organic Dyes with -COOH Functional Group. *Adv. Energy Mater.* **2018**, *8*, 1800715. [[CrossRef](#)]

49. Ooyama, Y.; Harima, Y. Photophysical and Electrochemical Properties, and Molecular Structures of Organic Dyes for Dye-Sensitized Solar Cells. *ChemPhysChem* **2013**, *13*, 4032–4080. [[CrossRef](#)] [[PubMed](#)]
50. Park, C.; Ko, H.; Sin, D.H.; Song, K.C.; Cho, K. Organometal Halide Perovskite Solar Cells with Improved Thermal Stability via Grain Boundary Passivation Using a Molecular Additive. *Adv. Funct. Mater.* **2017**, *27*, 1703546. [[CrossRef](#)]
51. Lee, D.; Yun, J.S.; Kim, J.; Soufiani, A.M.; Chen, S.; Cho, Y.; Deng, X.F.; Seidel, J.; Lim, S.; Huang, S.J.; et al. Passivation of Grain Boundaries by Phenethylammonium in Formamidinium-Methylammonium Lead Halide Perovskite Solar Cells. *ACS Energy Lett.* **2018**, *3*, 647–654. [[CrossRef](#)]
52. Wang, Y.F.; Zhang, J.; Chen, S.H.; Zhang, H.Y.; Li, L.G.; Fu, Z.Y. Surface Passivation with Nitrogen-Doped Carbon Dots for Improved Perovskite Solar Cell Performance. *J. Mater. Sci.* **2018**, *53*, 9180–9190. [[CrossRef](#)]
53. Liu, L.F.; Mei, A.Y.; Liu, T.F.; Jiang, P.; Sheng, Y.S.; Zhang, L.J.; Han, H.W. Fully Printable Mesoscopic Perovskite Solar Cells with Organic Silane Self-Assembled Monolayer. *J. Am. Chem. Soc.* **2015**, *137*, 1790–1793. [[CrossRef](#)] [[PubMed](#)]
54. Rao, S.S.; Punnoose, D.; Tulasivarma, C.V.; Kumar, C.H.S.S.P.; Gopi, C.V.V.M.; Kim, S.K.; Kim, H.J. A Strategy to Enhance the Efficiency of Dye-Sensitized Solar Cells by the Highly Efficient TiO₂/ZnS Photoanode. *Dalton Trans.* **2015**, *44*, 2447–2455. [[CrossRef](#)]
55. Wang, Y.; Mahmoudi, T.; Rho, W.Y.; Yang, H.Y.; Seo, S.; Bhat, K.S.; Ahmad, R.; Hahn, Y.B. Ambient-Air-Solution-Processed Efficient and Highly Stable Perovskite Solar Cells Based on CH₃NH₃PbI_{3-x}Cl_x-NiO Composite with Al₂O₃/NiO Interfacial Engineering. *Nano Energy* **2017**, *40*, 408–417. [[CrossRef](#)]
56. Shaikh, S.F.; Kwon, H.C.; Yang, W.; Mane, R.S.; Moon, J. Performance Enhancement of Mesoporous TiO₂-Based Perovskite Solar Cells by ZnS Ultrathin-Interfacial Modification Layer. *J. Alloys Compd.* **2017**, *738*, 405–414. [[CrossRef](#)]
57. Wang, Y.Q.; Xu, S.B.; Deng, J.G.; Gao, L.Z. Enhancing the Efficiency of Planar Heterojunction Perovskite Solar Cells via Interfacial Engineering with 3-aminopropyl Trimethoxy Silane Hydrolysate. *R. Soc. Open Sci.* **2017**, *4*, 170980. [[CrossRef](#)] [[PubMed](#)]
58. Lu, J.F.; Lin, X.F.; Jiao, X.C.; Gengenbach, T.; Scully, A.D.; Jiang, L.C.; Tan, B.; Sun, J.S.; Li, B.; Pai, N.; et al. Interfacial Benzenethiol Modification Facilitates Charge Transfer and Improves Stability of cm-sized Metal Halide Perovskite Solar Cells with up to 20% Efficiency. *Energy Environ. Sci.* **2018**, *11*, 1880–1889. [[CrossRef](#)]
59. Wang, F.J.; Shirnazaki, A.; Yang, F.J.; Kanahashi, K.; Matsuldi, K.; Miyauchi, Y.; Takenobu, T.; Wakamiya, A.; Murata, Y.; Matsuda, K. Highly Efficient and Stable Perovskite Solar Cells by Interfacial Engineering Using Solution-Processed Polymer Layer. *J. Phys. Chem. C* **2017**, *121*, 1562–1568. [[CrossRef](#)]
60. Singh, T.; Oz, S.; Sasinska, A.; Frohnhoven, R.; Mathur, S.; Miyasaka, T. Sulfate-Assisted Interfacial Engineering for High Yield and Efficiency of Triple Cation Perovskite Solar Cells with Alkali-Doped TiO₂ Electron-Transporting Layers. *Adv. Funct. Mater.* **2018**, *28*, 1706287. [[CrossRef](#)]
61. Zheng, L.Y.; Mukherjee, S.; Wang, K.; Hay, M.E.; Boudouris, B.W.; Gong, X. Radical Polymers as Interfacial Layers in Inverted Hybrid Perovskite Solar Cells. *J. Mater. Chem. A* **2017**, *5*, 23831–23839. [[CrossRef](#)]
62. Lee, J.; Kang, H.; Kim, G.; Back, H.; Kim, J.; Hong, S.; Park, B.; Lee, E.; Lee, K. Achieving Large-Area Planar Perovskite Solar Cells by Introducing an Interfacial Compatibilizer. *Adv. Mater.* **2017**, *29*, 1606363. [[CrossRef](#)] [[PubMed](#)]
63. Grancini, G.; Roldan-Carmona, C.; Zimmermann, I.; Mosconi, E.; Lee, X.; Martineau, D.; Nabey, S.; Oswald, F.; De Angelis, F.; Graetzel, M.; et al. One-Year Stable Perovskite Solar Cells by 2D/3D Interface Engineering. *Nat. Commun.* **2017**, *8*, 15684. [[CrossRef](#)] [[PubMed](#)]
64. Niu, G.D.; Guo, X.D.; Wang, L.D. Review of Recent Progress in Chemical Stability of Perovskite Solar Cells. *J. Mater. Chem. A* **2014**, *3*, 8970–8980. [[CrossRef](#)]
65. O'Mahony, F.T.F.; Lee, Y.H.; Jelllett, C.; Dmitrov, S.; Bryant, D.T.J.; Durrant, J.R.; O'Regan, B.C.; Graetzel, M.; Nazeeruddin, M.K.; Haque, S.A. Improved Environmental Stability of Organic Lead Trihalide Perovskite-Based Photoactive-Layers in the Presence of Mesoporous TiO₂. *J. Mater. Chem. A* **2015**, *3*, 7219–7223. [[CrossRef](#)]
66. Jeon, N.J.; Noh, J.H.; Kim, Y.C.; Yang, W.S.; Ryu, S.; Seok, S.I. Solvent Engineering for High-Performance Inorganic-Organic Hybrid Perovskite Solar Cells. *Nat. Mater.* **2014**, *13*, 897–903. [[CrossRef](#)] [[PubMed](#)]

67. Zhu, W.D.; Bao, C.X.; Li, F.M.; Yu, T.; Gao, H.; Yi, Y.; Yang, J.; Fu, G.; Zhou, X.X.; Zou, Z.G. A Halide Exchange Engineering for $\text{CH}_3\text{NH}_3\text{PbI}_{3-x}\text{Br}_x$ Perovskite Solar Cells with High Performance and Stability. *Nano Energy* **2016**, *19*, 17–26. [[CrossRef](#)]
68. Hoke, E.T.; Slotcavage, D.J.; Dohner, E.R.; Bowering, A.R.; Karunadasa, H.I.; McGehee, M.D. Reversible Photo-Induced Trap Formation in Mixed-Halide Hybrid Perovskites for Photovoltaics. *Chem. Sci.* **2014**, *6*, 613–617. [[CrossRef](#)] [[PubMed](#)]
69. Conings, B.; Drijkoningen, J.; Gauquelin, N.; Babayigit, A.; D'Haen, J.; D'Olieslaeger, L.; Ethirajan, A.; Verbeeck, J.; Manca, J.; Mosconi, E.; et al. Intrinsic Thermal Instability of Methylammonium Lead Trihalide Perovskite. *Adv. Energy Mater.* **2015**, *5*, 1500477. [[CrossRef](#)]
70. Lee, J.W.; Seol, D.J.; Cho, A.N.; Park, N.G. High-Efficiency Perovskite Solar Cells Based on the Black Polymorph of $\text{HC}(\text{NH}_2)_2\text{PbI}_3$. *Adv. Mater.* **2014**, *26*, 4991–4998. [[CrossRef](#)] [[PubMed](#)]
71. Liang, J.; Wang, C.X.; Wang, Y.R.; Xu, Z.R.; Lu, Z.P.; Ma, Y.; Zhu, H.F.; Hu, Y.; Xiao, C.C.; Yi, X.; et al. All-Inorganic Perovskite Solar Cells. *J. Am. Chem. Soc.* **2016**, *138*, 15829–15832. [[CrossRef](#)] [[PubMed](#)]
72. Frolova, L.A.; Anokhin, D.V.; Piryazev, A.A.; Luchkin, S.Y.; Dremova, N.N.; Stevenson, K.J.; Troshin, P.A. Highly Efficient All-Inorganic Planar Heterojunction Perovskite Solar Cells Produced by Thermal Coevaporation of CsI and PbI_2 . *J. Phys. Chem. Lett.* **2016**, *8*, 67–72. [[CrossRef](#)] [[PubMed](#)]
73. Wang, Q.; Dong, Q.F.; Li, T.; Gruverman, A.; Huang, J.S. Thin Insulating Tunneling Contacts for Efficient and Water-Resistant Perovskite Solar Cells. *Adv. Mater.* **2016**, *28*, 6734–6739. [[CrossRef](#)] [[PubMed](#)]
74. Chen, W.; Xu, L.M.; Feng, X.Y.; Jie, J.S.; He, Z. Metal Acetylacetonate Series in Interface Engineering for Full Low-Temperature-Processed, High-Performance, and Stable Planar Perovskite Solar Cells with Conversion Efficiency over 16% on 1 cm^2 Scale. *Adv. Mater.* **2017**, *29*, 1603923. [[CrossRef](#)] [[PubMed](#)]
75. Wang, F.; Geng, W.; Zhou, Y.; Fang, H.H.; Tong, C.J.; Loi, M.A.; Liu, L.M.; Zhao, N. Phenylalkylamine Passivation of Organolead Halide Perovskites Enabling High-Efficiency and Air-Stable Photovoltaic Cells. *Adv. Mater.* **2016**, *28*, 9986–9992. [[CrossRef](#)] [[PubMed](#)]
76. Xu, J.X.; Voznyy, O.; Comin, R.; Gong, X.W.; Walters, G.; Liu, M.; Kanjanaboos, P.; Lan, X.Z.; Sargent, E.H. Crosslinked Remote-Doped Hole-Extracting Contacts Enhance Stability under Accelerated Lifetime Testing in Perovskite Solar Cells. *Adv. Mater.* **2016**, *28*, 2807–2815. [[CrossRef](#)] [[PubMed](#)]
77. Pisoni, A.; Jacimovic, J.; Barisic, O.S.; Spina, M.; Gaal, R.; Forro, L.; Horvath, E. Ultra-Low Thermal Conductivity in Organic-Inorganic Hybrid Perovskite $\text{CH}_3\text{NH}_3\text{PbI}_3$. *J. Phys. Chem. Lett.* **2014**, *5*, 2488–2492. [[CrossRef](#)] [[PubMed](#)]
78. Beal, R.E.; Slotcavage, D.J.; Leijtens, T.; Bowering, A.R.; Belisle, R.A.; Nguyen, W.H.; Burkhard, G.F.; Hoke, E.T.; McGehee, M.D. Cesium Lead Halide Perovskites with Improved Stability for Tandem Solar Cells. *J. Phys. Chem. Lett.* **2016**, *7*, 746–751. [[CrossRef](#)] [[PubMed](#)]
79. Eperon, G.E.; Stranks, S.D.; Menelaou, C.; Johnston, M.B.; Herz, L.M.; Snaith, H.J. Formamidinium Lead Trihalide: A Broadly Tunable Perovskite for Efficient Planar Heterojunction Solar Cells. *Energy Environ. Sci.* **2014**, *7*, 982–988. [[CrossRef](#)]
80. McMeekin, D.P.; Sadoughi, G.; Rehman, W.; Eperon, G.E.; Saliba, M.; Horantner, M.T.; Haghighirad, A.; Sakai, N.; Korte, L.; Rech, B.; et al. A Mixed-Cation Lead Mixed-Halide Perovskite Absorber for Tandem Solar Cells. *Science* **2016**, *351*, 151–155. [[CrossRef](#)] [[PubMed](#)]
81. Baikie, T.; Fang, Y.N.; Kadro, J.M.; Schreyer, M.; Wei, F.X.; Mhaisalkar, S.G.; Graetzel, M.; White, T.J. Synthesis and Crystal Chemistry of the Hybrid Perovskite $\text{CH}_3\text{NH}_3\text{PbI}_3$ for Solid-State Sensitized Solar Cell Applications. *J. Mater. Chem. A* **2013**, *1*, 5628–5641. [[CrossRef](#)]
82. Swarnkar, A.; Marshall, A.R.; Sanhira, E.M.; Chernomordik, B.D.; Moore, D.T.; Christians, J.A.; Chakrabarti, T.; Luther, J.M. Quantum Dot-Induced Phase Stabilization of $\alpha\text{-CsPbI}_3$ Perovskite for High-Efficiency Photovoltaics. *Science* **2016**, *354*, 92–95. [[CrossRef](#)] [[PubMed](#)]
83. Malinauskas, T.; Tomkute-Luksiene, D.; Sens, R.; Daskeviciene, M.; Send, R.; Wonneberger, H.; Jankauskas, V.; Bruder, I.; Getautis, V. Enhancing Thermal Stability and Lifetime of Solid-State Dye-Sensitized Solar Cells via Molecular Engineering of the Hole Transporting Material spiro-OMeTAD. *ACS Appl. Mater. Interfaces* **2015**, *7*, 11107–11116. [[CrossRef](#)] [[PubMed](#)]
84. Li, W.Z.; Dong, H.P.; Wang, L.D.; Li, N.; Guo, X.D.; Li, J.W.; Qiu, Y. Montmorillonite as Bifunctional Buffer Layer Material for Hybrid Perovskite Solar Cells with Protection from Corrosion and Retarding Recombination. *J. Mater. Chem. A* **2014**, *2*, 13587–13592. [[CrossRef](#)]

85. Zhang, J.B.; Zhang, T.; Jiang, L.C.; Bach, U.; Cheng, Y.B. 4-tert-Butylpyridine Free Hole Transport Materials for Efficient Perovskite Solar Cells: A New Strategy to Enhance the Environmental and Thermal Stability. *ACS Energy Lett.* **2018**, *3*, 1677–1682. [[CrossRef](#)]
86. Zhao, X.; Kim, H.S.; Seo, J.Y.; Park, N.G. Effect of Selective Contacts on the Thermal Stability of Perovskite Solar Cells. *ACS Appl. Mater. Interfaces* **2017**, *9*, 7148–7153. [[CrossRef](#)] [[PubMed](#)]
87. Quan, L.N.; Yuan, M.J.; Comin, R.; Voznyy, O.; Beauregard, E.M.; Hoogland, S.; Buin, A.; Kirmani, A.R.; Zhao, K.; Amassian, A.; et al. Ligand-Stabilized Reduced-Dimensionality Perovskites. *J. Am. Chem. Soc.* **2016**, *138*, 2649–2655. [[CrossRef](#)] [[PubMed](#)]
88. Peng, W.; Yin, J.; Ho, K.T.; Ouellette, O.; De Bastiani, M.; Murali, B.; El Tall, O.; Shen, C.; Miao, X.H.; Pan, J.; et al. Ultralow Self-Doping in 2D Hybrid Perovskite Single Crystals. *Nano Lett.* **2017**, *17*, 4759–4767. [[CrossRef](#)] [[PubMed](#)]
89. Lin, Y.; Bai, Y.; Fang, Y.J.; Wang, Q.; Deng, Y.H.; Huang, J.S. Suppressed Ion Migration in Low Dimensional Perovskites. *ACS Energy Lett.* **2017**, *2*, 1571–1572. [[CrossRef](#)]
90. Mitzi, D.B. Templating and Structural Engineering in Organic–Inorganic Perovskites. *J. Chem. Soc. Dalton* **2000**, *1*, 1–12. [[CrossRef](#)]
91. Kamminga, M.E.; Fang, H.H.; Filip, M.R.; Giustino, F.; Baas, J.; Blake, G.R.; Loi, M.A.; Palstra, T.T.M. Confinement Effects in Low-Dimensional Lead Iodide Perovskite Hybrids. *Chem. Mater.* **2016**, *28*, 4554–4562. [[CrossRef](#)]
92. Muljarov, E.A.; Tikhodeev, S.G.; Gippius, N.A. Excitons in Self-Organized Semiconductor/Insulator Superlattices: Pbl-Based Perovskite Compounds. *Phys. Rev. B* **1995**, *51*, 14370–14378. [[CrossRef](#)]
93. Smith, I.C.; Hoke, E.T.; Solis-Ibarra, D.; McGehee, M.D.; Karunadasa, H.I. A Layered Hybrid Perovskite Solar-Cell Absorber with Enhanced Moisture Stability. *Angew. Chem.-Int. Ed.* **2015**, *126*, 11232–11235. [[CrossRef](#)]
94. Shayegan, Z.; Lee, C.S.; Haghghat, F. TiO₂ Photocatalyst for Removal of Volatile Organic Compounds in Gas Phase—A Review. *Chem. Eng. J.* **2018**, *334*, 2408–2439. [[CrossRef](#)]
95. Leijtens, T.; Eperon, G.E.; Pathak, S.; Abate, A.; Lee, M.M.; Snaith, H.J. Overcoming Ultraviolet Light Instability of Sensitized TiO₂ with Meso-Superstructured Organometal Tri-Halide Perovskite Solar Cells. *Nat. Commun.* **2013**, *4*, 2885. [[CrossRef](#)] [[PubMed](#)]
96. Ito, S.; Tanaka, S.; Manabe, K.; Nishino, H. Effects of Surface Blocking Layer of Sb₂S₃ on Nanocrystalline TiO₂ for CH₃NH₃PbI₃ Perovskite Solar Cells. *J. Phys. Chem. C* **2014**, *118*, 16995–17000. [[CrossRef](#)]
97. Roose, B.; Johansen, C.M.; Dupraz, K.; Jaouen, T.; Aebi, P.; Steiner, U.; Abate, A. Ga-Doped SnO₂ Mesoporous Contact for UV Stable Highly Efficient Perovskite Solar Cells. *J. Mater. Chem. A* **2017**, *6*, 1850–1857. [[CrossRef](#)]
98. Yin, G.N.; Zhao, H.; Feng, J.S.; Sun, J.; Yan, J.Q.; Liu, Z.K.; Lin, S.H.; Liu, S.Z. A Low-Temperature and Facile Solution-Processed Two-Dimensional TiS₂ as an Effective Electron Transport Layer for UV-Stable Planar Perovskite solar cell. *J. Mater. Chem. A* **2018**, *6*, 9132–9138. [[CrossRef](#)]
99. Tsai, C.H.; Li, N.; Lee, C.C.; Wu, H.C.; Zhu, Z.L.; Wang, L.D.; Chen, W.C.; Yan, H.; Chueh, C.C. Efficient and UV-Stable Perovskite Solar Cells Enabled by Side Chain-Engineered Polymeric Hole-Transporting Layers. *J. Mater. Chem. A* **2018**, *6*, 12999–13004. [[CrossRef](#)]
100. Cao, J.; Lv, X.D.; Zhang, P.; Chuong, T.T.; Wu, B.H.; Feng, X.X.; Shan, C.F.; Liu, J.C.; Tang, Y. Plant Sunscreen and Co(II)/(III) Porphyrins for UV-Resistant and Thermally Stable Perovskite Solar Cells: From Natural to Artificial. *Adv. Mater.* **2018**, *30*, 1800568. [[CrossRef](#)] [[PubMed](#)]
101. Roh, H.S.; Han, G.S.; Lee, S.; Kim, S.; Choi, S.; Yoon, C.; Lee, J.K. New Down-Converter for UV-Stable Perovskite Solar Cells: Phosphor-in-Glass. *J. Power Sources* **2018**, *389*, 135–139. [[CrossRef](#)]
102. Heo, J.H.; Im, S.H.; Noh, J.H.; Mandal, T.N.; Lim, C.S.; Chang, J.A.; Lee, Y.H.; Kim, H.J.; Sarkar, A.; Nazeeruddin, M.K.; et al. Efficient Inorganic–Organic Hybrid Heterojunction Solar Cells Containing Perovskite Compound and Polymeric Hole Conductors. *Nat. Photonics* **2013**, *7*, 487–492. [[CrossRef](#)]
103. Wang, H.; Liu, G.H.; Li, X.; Xiang, P.; Ku, Z.L.; Rong, Y.G.; Xu, M.; Liu, L.F.; Hu, M.; Yang, Y.; et al. Highly Efficient poly(3-hexylthiophene) Based Monolithic Dye-Sensitized Solar Cells with Carbon Counter Electrode. *Energy Environ. Sci.* **2011**, *4*, 2025–2029. [[CrossRef](#)]
104. Maniarasu, S.; Korukonda, T.B.; Manjunath, V.; Ramasamy, E.; Ramesh, M.; Veerappan, G. Recent Advancement in Metal Cathode and Hole-Conductor-Free Perovskite Solar Cells for Low-Cost and High Stability: A Route Towards Commercialization. *Renew. Sustain. Energy Rev.* **2018**, *82*, 845–857. [[CrossRef](#)]

105. Sun, W.H.; Ye, S.Y.; Rao, H.X.; Li, Y.L.; Liu, Z.W.; Xiao, L.X.; Chen, Z.J.; Bian, Z.Q.; Huang, C.H. Room-Temperature and Solution-Processed Copper Iodide as the Hole Transport Layer for Inverted Planar Perovskite Solar Cells. *Nanoscale* **2016**, *8*, 15954–15960. [[CrossRef](#)] [[PubMed](#)]
106. Ye, S.Y.; Rao, H.X.; Yan, W.B.; Li, Y.L.; Sun, W.H.; Peng, H.T.; Liu, Z.W.; Bian, Z.Q.; Li, Y.F.; Huang, C.H. A Strategy to Simplify the Preparation Process of Perovskite Solar Cells by Co-deposition of a Hole-Conductor and a Perovskite Layer. *Adv. Mater.* **2016**, *28*, 9648–9654. [[CrossRef](#)] [[PubMed](#)]
107. Liu, Z.H.; Zhu, A.L.; Cai, F.S.; Tao, L.M.; Zhou, Y.H.; Zhao, Z.X.; Chen, Q.; Cheng, Y.B.; Zhou, H.P. Nickel Oxide Nanoparticles for Efficient Hole Transport in p-i-n and n-i-p Perovskite Solar Cells. *J. Mater. Chem. A* **2017**, *5*, 6597–6605. [[CrossRef](#)]
108. Kwon, U.; Kim, B.G.; Nguyen, D.C.; Park, J.H.; Ha, N.Y.; Kim, S.J.; Ko, S.H.; Lee, S.; Lee, D.; Park, H.J. Solution-Processible Crystalline NiO Nanoparticles for High-Performance Planar Perovskite Photovoltaic Cells. *Sci. Rep.* **2016**, *6*, 30759. [[CrossRef](#)] [[PubMed](#)]
109. Zhu, Z.L.; Bai, Y.; Zhang, T.; Liu, Z.K.; Long, X.; Wei, Z.H.; Wang, Z.L.; Zhang, L.X.; Wang, J.N.; Yan, F.; et al. High-Performance Hole-Extraction Layer of Sol-Gel-Processed NiO Nanocrystals for Inverted Planar Perovskite Solar Cells. *Angew. Chem.-Int. Ed.* **2015**, *53*, 12571–12575. [[CrossRef](#)]
110. Cui, J.; Meng, F.P.; Zhang, H.; Cao, K.; Yuan, H.L.; Cheng, Y.B.; Huang, F.; Wang, M.K. CH₃NH₃PbI₃-Based Planar Solar Cells with Magnetron Sputtered Nickel Oxide. *ACS Appl. Mater. Interfaces* **2014**, *6*, 22862–22870. [[CrossRef](#)] [[PubMed](#)]
111. Vivo, P.; Salunke, J.K.; Priimagi, A. Hole-Transporting Materials for Printable Perovskite Solar Cells. *Materials* **2017**, *10*, 1087. [[CrossRef](#)] [[PubMed](#)]
112. Giorgi, G.; Yamashita, K. Organic-Inorganic Halide Perovskites: An Ambipolar Class of Materials with Enhanced Photovoltaic Performances. *J. Mater. Chem. A* **2015**, *3*, 8981–8991. [[CrossRef](#)]
113. Stamplecoskie, K.G.; Manser, J.S.; Kamat, P.V. Dual Nature of the Excited State in Organic-Inorganic Lead Halide Perovskites. *Energy Environ. Sci.* **2014**, *8*, 208–215. [[CrossRef](#)]
114. Edri, E.; Kirmayer, S.; Mukhopadhyay, S.; Gartsman, K.; Hodes, G.; Cahen, D. Elucidating the Charge Carrier Separation and Working Mechanism of CH₃NH₃PbI_{3-x}Cl_x Perovskite Solar Cells. *Nat. Commun.* **2014**, *5*, 3461. [[CrossRef](#)] [[PubMed](#)]
115. Xing, G.C.; Mathews, N.; Sun, S.Y.; Lim, S.S.; Lam, Y.M.; Gratzel, M.; Mhaisalkar, S.; Sum, T.C. Long-Range Balanced Electron- and Hole-Transport Lengths in Organic-Inorganic CH₃NH₃PbI₃. *Science* **2013**, *342*, 344–347. [[CrossRef](#)] [[PubMed](#)]
116. Li, Y.; Yan, W.B.; Li, Y.L.; Wang, S.F.; Wang, W.; Bian, Z.Q.; Xiao, L.X.; Gong, Q.H. Direct Observation of Long Electron-Hole Diffusion Distance in CH₃NH₃PbI₃ Perovskite Thin Film. *Sci. Rep.* **2015**, *5*, 14485. [[CrossRef](#)] [[PubMed](#)]
117. Etgar, L.; Gao, P.; Xue, Z.S.; Peng, Q.; Chandiran, A.K.; Liu, B.; Nazeeruddin, M.K.; Gratzel, M. Mesoscopic CH₃NH₃PbI₃/TiO₂ Heterojunction Solar Cells. *J. Am. Chem. Soc.* **2012**, *134*, 17396–17399. [[CrossRef](#)] [[PubMed](#)]
118. Aharon, S.; Gamliel, S.; El Cohen, B.; Etgar, L. Depletion Region Effect of Highly Efficient Hole Conductor Free CH₃NH₃PbI₃ Perovskite Solar Cells. *Phys. Chem. Chem. Phys.* **2014**, *16*, 10512–10518. [[CrossRef](#)] [[PubMed](#)]
119. Ma, Q.S.; Huang, S.J.; Wen, X.M.; Green, M.A.; Ho-Baillie, A.W.Y. Hole Transport Layer Free Inorganic CsPbIBr₂ Perovskite Solar Cell by Dual Source Thermal Evaporation. *Adv. Energy Mater.* **2016**, *6*, 1502202. [[CrossRef](#)]
120. Shi, J.J.; Luo, Y.H.; Wei, H.Y.; Luo, J.H.; Dong, J.; Lv, S.T.; Xiao, J.Y.; Xu, Y.Z.; Zhu, L.F.; Xu, X.; et al. Modified Two-Step Deposition Method for High-Efficiency TiO₂/CH₃NH₃PbI₃ Heterojunction Solar Cells. *ACS Appl. Mater. Interfaces* **2014**, *6*, 9711–9718. [[CrossRef](#)] [[PubMed](#)]
121. Wei, H.Y.; Shi, J.J.; Xu, X.; Xiao, J.Y.; Luo, J.H.; Dong, J.; Lv, S.T.; Zhu, L.F.; Wu, H.J.; Li, D.M.; et al. Enhanced Charge Collection with Ultrathin AlO_x Electron Blocking Layer for Hole-Transporting Material-Free Perovskite Solar Cell. *Phys. Chem. Chem. Phys.* **2015**, *17*, 4937–4944. [[CrossRef](#)] [[PubMed](#)]
122. Xu, Y.Z.; Shi, J.J.; Lv, S.T.; Zhu, L.F.; Dong, J.; Wu, H.J.; Xiao, Y.; Luo, Y.H.; Wang, S.R.; Li, D.M.; et al. Simple Way to Engineer Metal-Semiconductor Interface for Enhanced Performance of Perovskite Organic Lead Iodide Solar Cells. *ACS Appl. Mater. Interfaces* **2014**, *6*, 5651–5656. [[CrossRef](#)] [[PubMed](#)]

123. Zhang, F.Q.; Yang, X.C.; Wang, H.X.; Cheng, M.; Zhao, J.H.; Sun, L.C. Structure Engineering of Hole-Conductor Free Perovskite-Based Solar Cells with Low-Temperature-Processed Commercial Carbon Paste as Cathode. *ACS Appl. Mater. Interfaces* **2014**, *6*, 16140–16146. [[CrossRef](#)] [[PubMed](#)]
124. Zhang, H.; Wang, H.; Williams, S.T.; Xiong, D.H.; Zhang, W.J.; Chueh, C.C.; Chen, W.; Jen, A.K.Y. SrCl₂ Derived Perovskite Facilitating a High Efficiency of 16% in Hole-Conductor-Free Fully Printable Mesoscopic Perovskite Solar Cells. *Adv. Mater.* **2017**, *29*, 1606608. [[CrossRef](#)] [[PubMed](#)]
125. Qiu, L.B.; Deng, J.; Lu, X.; Yang, Z.B.; Peng, H.S. Integrating Perovskite Solar Cells into a Flexible Fiber. *Angew. Chem.-Int. Ed.* **2015**, *53*, 10425–10428. [[CrossRef](#)] [[PubMed](#)]
126. Li, Z.; Kulkarni, S.A.; Boix, P.P.; Shi, E.Z.; Cao, A.Y.; Fu, K.W.; Batabyal, S.K.; Zhang, J.; Xiong, Q.H.; Wong, L.H.; et al. Laminated Carbon Nanotube Networks for Metal Electrode-Free Efficient Perovskite Solar Cells. *ACS Nano* **2014**, *8*, 6797–6804. [[CrossRef](#)] [[PubMed](#)]
127. Cai, Y.; Liang, L.S.; Gao, P. Promise of Commercialization: Carbon Materials for Low-Cost Perovskite Solar Cells. *Chin. Phys. B* **2018**, *27*, 27. [[CrossRef](#)]
128. Lee, J.; Menampambath, M.M.; Hwang, J.Y.; Baik, S. Hierarchically Structured Hole Transport Layers of Spiro-OMeTAD and Multiwalled Carbon Nanotubes for Perovskite Solar Cells. *ChemSusChem* **2015**, *8*, 2358–2362. [[CrossRef](#)] [[PubMed](#)]
129. Habisreutinger, S.N.; Leijtens, T.; Eperon, G.E.; Stranks, S.D.; Nicholas, R.J.; Snaith, H.J. Carbon Nanotube/Polymer Composites as a Highly Stable Hole Collection Layer in Perovskite Solar Cells. *Nano Lett.* **2014**, *14*, 5561–5568. [[CrossRef](#)] [[PubMed](#)]
130. Yang, Y.L.; Chen, H.N.; Zheng, X.L.; Meng, X.Y.; Zhang, T.; Hu, C.; Bai, Y.; Xiao, S.; Yang, S.H. Ultrasound-Spray Deposition of Multi-Walled Carbon Nanotubes on NiO Nanoparticles-Embedded Perovskite Layers for High-Performance Carbon-Based Perovskite Solar Cells. *Nano Energy* **2017**, *42*, 322–333. [[CrossRef](#)]
131. Yoon, S.; Ha, T.J.; Kang, D.W. Improving the Performance and Reliability of Inverted Planar Perovskite Solar Cells with a Carbon Nanotubes/PEDOT:PSS Hybrid Hole Collector. *Nanoscale* **2017**, *9*, 9754–9761. [[CrossRef](#)] [[PubMed](#)]
132. Li, F.R.; Xu, Y.; Chen, W.; Xie, S.H.; Li, J.Y. Nanotube Enhanced Carbon Grids as Top Electrode for Fully Printable Mesoscopic Semitransparent Perovskite Solar Cells. *J. Mater. Chem. A* **2017**, *5*, 10374–10379. [[CrossRef](#)]
133. Li, H.; Cao, K.; Cui, J.; Liu, S.S.; Qiao, X.F.; Shen, Y.; Wang, M.K. 14.7% Efficient Mesoscopic Perovskite Solar Cells Using Single Walled Carbon Nanotubes/Carbon Composite Counter Electrodes. *Nanoscale* **2016**, *8*, 6379–6385. [[CrossRef](#)] [[PubMed](#)]
134. Ryu, J.; Lee, K.; Yun, J.; Yu, H.; Lee, J.; Jang, J. Paintable Carbon-Based Perovskite Solar Cells with Engineered Perovskite/Carbon Interface Using Carbon Nanotubes Dripping Method. *Small* **2017**, *13*, 1701225. [[CrossRef](#)] [[PubMed](#)]
135. Chen, J.Z.; Xiong, Y.L.; Rong, Y.G.; Mei, A.Y.; Sheng, Y.S.; Jiang, P.; Hu, Y.; Li, X.; Han, H.W. Solvent Effect on the Hole-Conductor-Free Fully Printable Perovskite Solar Cells. *Nano Energy* **2016**, *27*, 130–137. [[CrossRef](#)]
136. Mei, A.Y.; Li, X.; Liu, L.F.; Ku, Z.L.; Liu, T.F.; Rong, Y.G.; Xu, M.; Hu, M.; Chen, J.Z.; Yang, Y.; et al. A Hole-Conductor-Free, Fully Printable Mesoscopic Perovskite Solar Cell with High Stability. *Science* **2014**, *345*, 295–298. [[CrossRef](#)] [[PubMed](#)]
137. Yang, Y.; Ri, K.; Mei, A.Y.; Liu, L.F.; Hu, M.; Liu, T.F.; Li, X.; Han, H.W. The Size Effect of TiO₂ Nanoparticles on the Printable Mesoscopic Perovskite Solar Cell. *J. Mater. Chem. A* **2015**, *3*, 9103–9107. [[CrossRef](#)]
138. Liu, T.F.; Liu, L.F.; Hu, M.; Yang, Y.; Zhang, L.J.; Mei, A.Y.; Han, H.W. Critical Parameters in TiO₂/ZrO₂/Carbon-Based Mesoscopic Perovskite Solar Cell. *J. Power Sources* **2015**, *293*, 533–538. [[CrossRef](#)]
139. Wei, H.Y.; Xiao, J.Y.; Yang, Y.Y.; Lv, S.T.; Shi, J.J.; Xu, X.; Dong, J.; Luo, Y.H.; Li, D.M.; Meng, Q.B. Free-Standing Flexible Carbon Electrode for Highly Efficient Hole-Conductor-Free Perovskite Solar Cells. *Carbon* **2015**, *93*, 861–868. [[CrossRef](#)]
140. Zhang, L.J.; Liu, T.F.; Liu, L.F.; Hu, M.; Yang, Y.; Mei, A.Y.; Han, H.W. The Effect of Carbon Counter Electrode on Fully Printable Mesoscopic Perovskite Solar Cell. *J. Mater. Chem. A* **2015**, *3*, 9165–9170. [[CrossRef](#)]
141. Tao, H.J.; Li, Y.T.; Zhang, C.X.; Wang, K.; Wang, J.Y.; Tan, B.; Han, L.X.; Tao, J. High Permeable Microporous Structured Carbon Counter Electrode Assisted by Polystyrene Sphere for Fully Printable Perovskite Solar Cells. *Solid State Commun.* **2018**, *271*, 71–75. [[CrossRef](#)]

142. Liu, Z.Y.; Shi, T.L.; Tang, Z.R.; Liao, G.L. A Large-Area Hole-Conductor-Free Perovskite Solar Cell Based on a Low-Temperature Carbon Counter Electrode. *Mater. Res. Bull.* **2017**, *96*, 196–200. [[CrossRef](#)]
143. Zhang, J.J.; Meng, Z.; Guo, D.P.; Zou, H.Y.; Yu, J.G.; Fan, K. Hole-Conductor-Free Perovskite Solar Cells Prepared with Carbon Counter Electrode. *Appl. Surf. Sci.* **2017**, *430*, 531–538. [[CrossRef](#)]
144. Zong, B.B.; Fu, W.Y.; Liu, H.J.; Huang, L.W.; Bala, H.; Wang, X.D.; Sun, G.; Cao, J.L.; Zhang, Z.Y. Highly Stable Hole-Conductor-Free $\text{CH}_3\text{NH}_3\text{Pb}(\text{I}_{1-x}\text{Br}_x)_3$ Perovskite Solar Cells with Carbon Counter Electrode. *J. Alloys Compd.* **2018**, *748*, 1006–1012. [[CrossRef](#)]
145. Li, S.H.; Hu, J.H.; Yang, Y.P.; Zhao, L.; Qiao, Y.; Liu, W.H.; Liu, P.H.; Chen, M.W. Ag/Nano-TiO₂ Composite Compact Film for Enhanced Performance of Perovskite Solar Cells Based on Carbon Counter Electrodes. *Appl. Phys. A Mater. Sci. Process.* **2017**, *123*, 628. [[CrossRef](#)]
146. Novoselov, K.S.; Geim, A.K.; Morozov, S.V.; Jiang, D.; Zhang, Y.; Dubonos, S.V.; Grigorieva, I.V.; Firsov, A.A. Electric Field Effect in Atomically Thin Carbon Films. *Science* **2004**, *306*, 666–669. [[CrossRef](#)] [[PubMed](#)]
147. Wang, J.T.W.; Ball, J.M.; Barea, E.M.; Abate, A.; Alexander-Webber, J.A.; Huang, J.; Saliba, M.; Mora-Sero, I.; Bisquert, J.; Snaith, H.J.; et al. Low-Temperature Processed Electron Collection Layers of Graphene/TiO₂ Nanocomposites in Thin Film Perovskite Solar Cells. *Nano Lett.* **2014**, *14*, 724–730. [[CrossRef](#)] [[PubMed](#)]
148. Liu, Z.K.; You, P.; Xie, C.; Tang, G.Q.; Yan, F. Ultrathin and Flexible Perovskite Solar Cells with Graphene Transparent Electrodes. *Nano Energy* **2016**, *28*, 151–157. [[CrossRef](#)]
149. Sung, H.; Ahn, N.; Jang, M.S.; Lee, J.K.; Yoon, H.; Park, N.G.; Choi, M. Transparent Conductive Oxide-Free Graphene-Based Perovskite Solar Cells with over 17% Efficiency. *Adv. Energy Mater.* **2016**, *6*, 1501873. [[CrossRef](#)]
150. Yoon, J.; Sung, H.; Lee, G.; Cho, W.; Ahn, N.; Jung, H.S.; Choi, M. Super Flexible, High-efficiency Perovskite Solar Cells Employing Graphene Electrodes: Toward Future Foldable Power Sources. *Energy Environ. Sci.* **2017**, *10*, 337–345. [[CrossRef](#)]
151. Wu, Z.W.; Bai, S.; Xiang, J.; Yuan, Z.C.; Yang, Y.G.; Cui, W.; Gao, X.Y.; Liu, Z.; Jin, Y.Z.; Sun, B.Q. Efficient Planar Heterojunction Perovskite Solar Cells Employing Graphene Oxide as Hole Conductor. *Nanoscale* **2014**, *6*, 10505–10510. [[CrossRef](#)] [[PubMed](#)]
152. Giuri, A.; Masi, S.; Colella, S.; Listorti, A.; Rizzo, A.; Gigli, G.; Liscio, A.; Treossi, E.; Palermo, V.; Rella, S.; et al. UV Reduced Graphene Oxide PEDOT: PSS Nanocomposite for Perovskite Solar Cells. *IEEE Trans. Nanotechnol.* **2016**, *15*, 725–730. [[CrossRef](#)]
153. Li, D.; Cui, J.; Li, H.; Huang, D.K.; Wang, M.K.; Shen, Y. Graphene Oxide Modified Hole Transport Layer for $\text{CH}_3\text{NH}_3\text{PbI}_3$ Planar Heterojunction Solar Cells. *Sol. Energy* **2016**, *131*, 176–182. [[CrossRef](#)]
154. Yan, K.Y.; Wei, Z.H.; Li, J.K.; Chen, H.N.; Yi, Y.; Zheng, X.L.; Long, X.; Wang, Z.L.; Wang, J.N.; Xu, J.B. High-Performance Graphene-Based Hole Conductor-Free Perovskite Solar Cells: Schottky Junction Enhanced Hole Extraction and Electron Blocking. *Small* **2015**, *11*, 2269–2274. [[CrossRef](#)] [[PubMed](#)]
155. Wei, W.; Hu, B.Y.; Jin, F.M.; Jing, Z.Z.; Li, Y.X.; Blanco, A.A.G.; Stacchiola, D.J.; Hu, Y.H. Potassium-Chemical Synthesis of 3D Graphene from CO₂ and its Excellent Performance in HTM-Free Perovskite Solar Cells. *J. Mater. Chem. A* **2017**, *5*, 7749–7752. [[CrossRef](#)]
156. Zhu, Y.Y.; Jia, S.P.; Zheng, J.F.; Lin, Y.L.; Wu, Y.R.; Wang, J. Facile Synthesis of Nitrogen-Doped Graphene Frameworks for Enhanced Performance of Hole-Transport Material-Free Perovskite Solar Cells. *J. Mater. Chem. C* **2018**, *6*, 3097–3103. [[CrossRef](#)]
157. Lang, F.; Gluba, M.A.; Albrecht, S.; Rappich, J.; Korte, L.; Rech, B.; Nickel, N.H. Perovskite Solar Cells with Large-Area CVD-Graphene for Tandem Solar Cells. *J. Phys. Chem. Lett.* **2015**, *6*, 2745–2750. [[CrossRef](#)] [[PubMed](#)]
158. Zhou, J.X.; Ren, Z.W.; Li, S.H.; Liang, Z.C.; Surya, C.; Shen, H. Semi-Transparent Cl-Doped Perovskite Solar Cells with Graphene Electrodes for Tandem Application. *Mater. Lett.* **2018**, *220*, 80–85. [[CrossRef](#)]
159. Selvakumar, D.; Murugadoss, G.; Alsalmeh, A.; Alkathiri, A.M.; Jayavel, R. Heteroatom Doped Reduced Graphene Oxide Paper for Large Area Perovskite Solar Cells. *Sol. Energy* **2018**, *163*, 564–569. [[CrossRef](#)]
160. Green, M.A.; Hishikawa, Y.; Warta, W.; Dunlop, E.D.; Levi, D.H.; Hohl-Ebinger, J.; Ho-Baillie, A.W.Y. Solar Cell Efficiency Tables (Version 50). *Prog. Photovoltaics* **2017**, *25*, 668–676. [[CrossRef](#)]
161. Deng, Y.H.; Peng, E.; Shao, Y.C.; Xiao, Z.G.; Dong, Q.F.; Huang, J.S. Scalable Fabrication of Efficient Organolead Trihalide Perovskite Solar Cells with Doctor-Bladed Active Layers. *Energy Environ. Sci.* **2015**, *8*, 1544–1550. [[CrossRef](#)]

162. Wu, H.; Zhang, C.J.; Ding, K.X.; Wang, L.J.; Gao, Y.L.; Yang, J.L. Efficient Planar Heterojunction Perovskite Solar Cells Fabricated by In-Situ, Thermal-Annealing Doctor Blading in Ambient Condition. *Org. Electron.* **2017**, *45*, 302–307. [[CrossRef](#)]
163. Back, H.; Kim, J.; Kim, G.; Kim, T.K.; Kang, H.; Kong, J.; Lee, S.H.; Lee, K. Interfacial Modification of Hole Transport Layers for Efficient Large-Area Perovskite Solar Cells Achieved via Blade-Coating. *Sol. Energy Mater. Sol. Cells* **2016**, *144*, 309–315. [[CrossRef](#)]
164. Zhong, Y.F.; Munir, R.; Li, J.B.; Tang, M.C.; Niazi, M.R.; Smilgies, D.M.; Zhao, K.; Arnessian, A. Blade-Coated Hybrid Perovskite Solar Cells with Efficiency >17%: An In Situ Investigation. *ACS Energy Lett.* **2018**, *3*, 1078–1085. [[CrossRef](#)]
165. Kong, W.G.; Wang, G.L.; Zheng, J.M.; Hu, H.; Chen, H.; Li, Y.L.; Hu, M.M.; Zhou, X.Y.; Liu, C.; Chandrashekar, B.N.; et al. Fabricating High-Efficient Blade-Coated Perovskite Solar Cells under Ambient Condition Using Lead Acetate Trihydrate. *Sol. RRL* **2018**, *2*, 1770153. [[CrossRef](#)]
166. Tang, S.; Deng, Y.H.; Zheng, X.P.; Bai, Y.; Fang, Y.J.; Dong, Q.F.; Wei, H.T.; Huang, J.S. Composition Engineering in Doctor-Blading of Perovskite Solar Cells. *Adv. Energy Mater.* **2017**, *7*, 1700302. [[CrossRef](#)]
167. Hwang, K.; Jung, Y.S.; Heo, Y.J.; Scholes, F.H.; Watkins, S.E.; Subbiah, J.; Jones, D.J.; Kim, D.Y.; Vak, D. Toward Large Scale Roll-to-Roll Production of Fully Printed Perovskite Solar Cells. *Adv. Mater.* **2015**, *27*, 1241–1247. [[CrossRef](#)] [[PubMed](#)]
168. Gu, Z.W.; Zuo, L.J.; Larsen-Olsen, T.T.; Ye, T.; Wu, G.; Krebs, F.C.; Chen, H.Z. Interfacial Engineering of Self-Assembling Monolayer Modified Semi-Roll-to-Roll Planar Heterojunction Mixed Halide Perovskite Solar Cells on Flexible Substrates. *J. Mater. Chem. A* **2015**, *3*, 24254–24260. [[CrossRef](#)]
169. Lee, D.; Jung, Y.S.; Heo, Y.J.; Lee, S.; Hwang, K.; Jeon, Y.J.; Kim, J.E.; Park, J.; Jung, G.Y.; Kim, D.Y. Slot-Die Coated Perovskite Films Using Mixed Lead Precursors for Highly Reproducible and Large-Area Solar Cells. *ACS Appl. Mater. Interfaces* **2018**, *10*, 16133–16139. [[CrossRef](#)] [[PubMed](#)]
170. Burkitt, D.; Searle, J.; Watson, T. Perovskite Solar Cells in N-I-P Structure with Four Slot-Die-Coated Layers. *R. Soc. Open Sci.* **2018**, *5*, 172158. [[CrossRef](#)] [[PubMed](#)]
171. Kim, J.E.; Jung, Y.S.; Heo, Y.J.; Hwang, K.; Qin, T.S.; Kim, D.Y.; Vak, D. Slot Die Coated Planar Perovskite Solar Cells via Blowing and Heating Assisted One Step Deposition. *Sol. Energy Mater. Sol. Cells* **2018**, *179*, 80–86. [[CrossRef](#)]
172. Hsieh, H.C.; Yu, J.F.; Rwei, S.P.; Lin, K.F.; Shih, Y.C.; Wang, L.Y. Ultra-Compact Titanium Oxide Prepared by Ultrasonic Spray Pyrolysis Method for Planar Heterojunction Perovskite Hybrid Solar Cells. *Thin Solid Films* **2018**, *659*, 41–47. [[CrossRef](#)]
173. Kumari, N.; Patel, S.R.; Gohel, J.V. Optical and Structural Properties of ZnO Thin Films Prepared by Spray Pyrolysis for Enhanced Efficiency Perovskite Solar Cell Application. *Opt. Quantum Electron.* **2018**, *50*, 180. [[CrossRef](#)]
174. Chang, W.C.; Lan, D.H.; Lee, K.M.; Wang, X.F.; Liu, C.L. Controlled Deposition and Performance Optimization of Perovskite Solar Cells Using Ultrasonic Spray-Coating Photoactive Layers. *ChemSusChem* **2016**, *10*, 1405–1412. [[CrossRef](#)] [[PubMed](#)]
175. Chai, G.D.; Wang, S.Z.; Xia, Z.G.; Luo, S.Q.; Teng, C.; Yang, T.B.; Nie, Z.X.; Meng, T.J.; Zhou, H. PbI₂ Platelets for Inverted Planar Organolead Halide Perovskite Solar Cells via Ultrasonic Spray Deposition. *Semicond. Sci. Technol.* **2017**, *32*, 074003. [[CrossRef](#)]
176. Li, S.G.; Jiang, K.J.; Su, M.J.; Cui, X.P.; Huang, J.H.; Zhang, Q.Q.; Zhou, X.Q.; Yang, L.M.; Song, Y.L. Inkjet Printing of CH₃NH₃PbI₃ on a Mesoscopic TiO₂ Film for Highly Efficient Perovskite Solar Cells. *J. Mater. Chem. A* **2015**, *3*, 9092–9097. [[CrossRef](#)]
177. Peng, X.J.; Yuan, J.; Shen, S.; Gao, M.; Chesman, A.S.R.; Yin, H.; Cheng, J.S.; Zhang, Q.; Angmo, D. Perovskite and Organic Solar Cells Fabricated by Inkjet Printing: Progress and Prospects. *Adv. Funct. Mater.* **2017**, *27*, 1703704. [[CrossRef](#)]
178. Hashmi, S.G.; Tiihonen, A.; Martineau, D.; Ozkan, M.; Vivo, P.; Kaunisto, K.; Ulla, V.; Zakeeruddin, S.M.; Gratzel, M. Long Term Stability of Air Processed Inkjet Infiltrated Carbon-Based Printed Perovskite Solar Cells under Intense Ultra-Violet Light Soaking. *J. Mater. Chem. A* **2017**, *5*, 4797–4802. [[CrossRef](#)]
179. Hashmi, S.G.; Martineau, D.; Li, X.; Ozkan, M.; Tiihonen, A.; Dar, M.I.; Sarikka, T.; Zakeeruddin, S.M.; Paltakari, J.; Lund, P.D.; et al. Air Processed Inkjet Infiltrated Carbon Based Printed Perovskite Solar Cells with High Stability and Reproducibility. *Adv. Mater. Technol.* **2017**, *2*, 1600183. [[CrossRef](#)]

180. Bao, Z.N.; Rogers, J.A.; Katz, H.E. Printable Organic and Polymeric Semiconducting Materials and Devices. *J. Mater. Chem.* **1999**, *9*, 1895–1904. [[CrossRef](#)]
181. Raminafshar, C.; Dracopoulos, V.; Mohammadi, M.R.; Lianos, P. Carbon Based Perovskite Solar Cells Constructed by Screen-Printed Components. *Electrochim. Acta* **2018**, *276*, 261–267. [[CrossRef](#)]
182. Cao, K.; Zuo, Z.X.; Cui, J.; Shen, Y.; Moehl, T.; Zakeeruddin, S.M.; Grazel, M.; Wang, M.K. Efficient Screen Printed Perovskite Solar Cells Based on Mesoscopic TiO₂/Al₂O₃/NiO/carbon Architecture. *Nano Energy* **2015**, *17*, 171–179. [[CrossRef](#)]
183. Liu, M.Z.; Johnston, M.B.; Snaith, H.J. Efficient Planar Heterojunction Perovskite Solar Cells by Vapour Deposition. *Nature* **2013**, *501*, 395–398. [[CrossRef](#)] [[PubMed](#)]
184. Chen, Q.; Zhou, H.P.; Hong, Z.R.; Luo, S.; Duan, H.S.; Wang, H.H.; Liu, Y.S.; Li, G.; Yang, Y. Planar Heterojunction Perovskite Solar Cells via Vapor-Assisted Solution Process. *J. Am. Chem. Soc.* **2014**, *136*, 622–625. [[CrossRef](#)] [[PubMed](#)]



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