



# Thermodynamic and Experimental Study on Efficient Extraction of Valuable Metals from Polymetallic Nodules

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**Abstract:** Polymetallic nodules are promising resources for the extraction of valuable metals such as copper, nickel, and cobalt, as well as manganese alloys. To achieve efficient extraction of useful metals from the emerging resource, high-temperature carbothermic reduction of nodules was investigated by optimizing the reductant addition, slag and alloy systems. Thermochemical software FactSage was used to predict the liquidus temperature of the slag system, which is not sensitive to FeO, CaO and Al<sub>2</sub>O<sub>3</sub>, but decreases significantly with decreasing MnO/SiO<sub>2</sub> mass ratio. The experiments were designed to reduce the oxides of Cu, Co and Ni completely, and reduce FeO<sub>x</sub> partially depending on the amount of graphite addition while leaving the residual slag for further processing into ferromanganese and/or silicomanganese alloys. Co, Cu and Ni concentrations in the alloy decreased with increasing graphite addition. The optimal reduction condition was reached by adding 4 wt% graphite at the MnO/SiO<sub>2</sub> mass ratio of 1.6 in slag. The most effective metal-slag separation was achieved at 1350 °C, which enables the smelting reduction to be carried out in various furnaces.

Keywords: polymetallic nodules; carbothermic reduction; valuable metals; liquidus temperature

# 1. Introduction

The manganese and iron concretions on the sea bottom formed of concentric layers around a core are known as manganese nodules or polymetallic nodules. Polymetallic nodules contain many valuable elements such as Cu, Ni and Co, besides Mn and Fe. Trace amounts of Mo, V, and rare earth elements (REEs) are also present in the nodules. As the land-based resources for Mn, Cu, Ni and Co are depleting very fast, the exploration and exploitation of manganese nodules are essential to meet the future demands of Cu, Ni, Co and Mn in the world [1]. In recent years, the exploitation of manganese nodules from the deep sea has once again become a political and economic topic [2,3]. One of the biggest and economically most important areas, where polymetallic nodules can be found, is situated between the Clarion and Clipperton Zones in the tropical northern Pacific Ocean. The nodules could serve as an enormous reserve for these metals since they are found in relative abundance. The nodules in the Clarion Clipperton Zone alone constitute an estimated reserve of up to 220 million tons of copper, up to 260 million tons of nickel, up to 50 million tons of cobalt, and up to 5 billion tons of manganese [2,3]. These estimated amounts currently surpass universal land-based reserves in mines for Ni, Cu, Co and Mn. Recently, the increased interest in electrical energy storage and rapid growth in the market for electric vehicles will make the polymetallic nodules promising resources to produce component metals of batteries.

Polymetallic nodules are composed of complex Mn-oxides and Fe-oxyhydroxide minerals with mineral particles smaller than one  $\mu$ m. Ni, Cu, Co and Mo are found in the manganese- or iron-oxide matrix [4]. Therefore, beneficiation with classic techniques (density separation, floatation, etc.) to



generate a metal-bearing concentrate, which could be directly used in existing industries, is not effective [5]. Considerable research has focused on the extraction of valuable metals from ocean nodules. The nodule processing techniques are broadly divided into three categories: (1) hydrometallurgical processing (only); (2) pyrometallurgical treatment followed by hydrometallurgical processing; (3) hydrometallurgical processing followed by pyrometallurgical treatment for recovery of Mn from leaching residue [6–8]. Various hydrometallurgical approaches included reductive ammonia leaching [9–14], reduction and sulphuric acid leaching [15–18], reduction and hydrochloric acid leaching [19,20], and other integrated or synergistic solvent extractions [21–23]. Reductive ammonia and sulphuric acid leaching cannot recover manganese, which makes up nearly half of the nodules' value. Reductive hydrochloric acid leaching can recover manganese, however, the corrosive nature of chloride is the main disadvantage [24]. With time, the environmental regulations are becoming more and more stringent, it is mandatory to properly deal with the disposal of solid/liquid effluents generated during the hydrometallurgical processes. Some researchers reported the pyrometallurgical route to recover the Cu, Ni, Co, as well as Mn metals from polymetallic nodules at temperatures above 1400 °C [25–30]. Recently, the pyrometallurgical process with "zero-waste" has been proposed [28,30], based on the "Inco-process" [7]. Valuable metals such as Cu, Co, Ni are recovered first by smelting reduction at high temperature or sulfurizing into matte. The slag generated contains high manganese can be used in the existing production plants for ferromanganese or silico-manganese alloys. Since the (Cu + Ni + Co) grade of nodules is similar to nickel laterite, the pyrometallurgical treatment route of nickel laterites [31], which is a well-established industrial method, namely, the rotary kiln-electric arc furnace (RKEF) process, can be used to treat nodules to save a new capital investment to build a new process for nodules. However, limited experimental data on the extraction of valuable metals have been reported due to a limited supply of nodules. The operating parameters need to be optimized to prepare industrial scale-up. Of particular significance is that the melting temperatures of slags over 1500 °C were typically quoted significantly higher, which results in high consumption of energy and refractory. Therefore, in this study, the pyrometallurgical process of extraction of useful minerals from deep-sea nodules will be developed and optimized to produce Fe-Cu-Co-Ni master alloys and Mn-rich slag.

#### 2. Materials and Methods

#### 2.1. Nodule Materials

The nodules employed in this study were dredged from the Clarion-Clipperton Zone of the Pacific Ocean. The chemical composition of the nodule sample is shown in Table 1.

MnO	SiO <sub>2</sub>	Fe <sub>2</sub> O <sub>3</sub>	Al <sub>2</sub> O <sub>3</sub>	MgO	Na <sub>2</sub> O	CaO	NiO	CuO	CoO
40.25	13.48	8.28	4.67	3.48	3.01	2.16	1.71	1.65	0.30
K <sub>2</sub> O	TiO <sub>2</sub>	P <sub>2</sub> O <sub>5</sub>	BaO	ZnO	SO <sub>3</sub>	PbO	Cr <sub>2</sub> O <sub>3</sub>	LOI	MnO/SiO <sub>2</sub>
1.24	0.52	0.32	0.22	0.20	0.17	0.03	0.01	17.28	3.0

Table 1. Chemical composition of the nodules (wt%).

The microstructure and elemental mapping of a local area of an example nodule were determined by scanning electron microscopy (SEM) with energy dispersive X-ray spectrometry (EDS), which is shown in Figure 1. The nodule consists of concentrically banded zones of micro-layers around a nucleus. The microstructure of the shell is dense, while it is looser in the core of the nodule scanned. The maps of the distribution of each element across the local area show the distribution of Mn and Fe is in striation. Si is incorporated with Al, Na, K and Ca as impurity minerals, likely detrital deep-sea ooze [1]. Valuable metal elements such as Cu, Co and Ni in low concentrations are uniformly distributed in the nodule matrix. Ba is found to be associated with S, which is most likely considered to be precipitated barite (BaSO<sub>4</sub>).



Figure 1. Microstructure and element distribution measured by SEM-EDS.

## 2.2. Thermodynamic Consideration

## 2.2.1. Effect of Flux on the Liquidus Temperature of Resulted Slags

Thermodynamically, it is possible to reduce the oxides of nickel, copper, cobalt, and iron to metals while leaving the manganese oxide over a wide range of temperatures. The selective reduction of NiO, CuO, CoO and Fe<sub>2</sub>O<sub>3</sub> over MnO by carbon reductant can be simplified in Reactions (1)–(4) with a ratio of CO and CO<sub>2</sub> gases between critical limits. The iron oxide can be fully or partially reduced in the smelting process to adjust the slag composition and to control the alloy grade. For the sake of convenience, the stoichiometric amount of carbon needed to reduce the oxides is 2.4 g per 100 g ore, assuming only CO is generated during carbon gasification instead of a mixer of CO/CO<sub>2</sub>.

$$NiO + C \rightarrow Ni + CO/CO_2, \tag{1}$$

$$CuO + C \rightarrow Cu + CO/CO_2, \tag{2}$$

$$CoO + C \rightarrow Co + CO/CO_2,$$
 (3)

$$Fe_2O_3 + C \rightarrow FeO/Fe + CO/CO_2.$$
 (4)

In similar commercial scaled operations, some common oxides such as CaO, SiO<sub>2</sub>, and Al<sub>2</sub>O<sub>3</sub> are normally added as a flux to adjust the slag composition and keep the smelting slag in a fully molten state at a suitable temperature. The thermodynamic software package FactSage 7.2 [32] with the suitable optimized compounds and solution databases was used to predict the liquidus temperatures of the MnO-SiO<sub>2</sub>-"FeO"-Al<sub>2</sub>O<sub>3</sub>-MgO-Na<sub>2</sub>O-CaO-K<sub>2</sub>O system at iron saturation condition based on the normalization of major components of Table 1 and assumption of full reduction of NiO, CuO and CoO. The composition of the slag used for the FactSage calculation is given in Table 2.

The FactSage databases including FactPS (for pure compounds), FToxide (for solid and liquid oxide solutions), and FTmisc (for liquid alloys), were used. The effects of the  $MnO/SiO_2$  ratio and CaO,  $Al_2O_3$ , and "FeO" on the liquidus temperatures of the slag are shown in Figure 2. It can be seen from Figure 2a that the liquidus temperature increases with increasing CaO concentration in the monoxide

primary phase field at different MnO/SiO<sub>2</sub> ratios but decreases in the olivine primary phase field at the MnO/SiO<sub>2</sub> ratio of 1.0. Figure 2b shows the Al<sub>2</sub>O<sub>3</sub> concentration has a little effect on the liquidus temperature at the MnO/SiO<sub>2</sub> ratio of 3.0. The liquidus temperature increases slightly with increasing Al<sub>2</sub>O<sub>3</sub> concentration at the MnO/SiO<sub>2</sub> ratio of 2.0 in the monoxide primary phase field but decreases slightly at the MnO/SiO<sub>2</sub> ratio of 1.0 in the olivine primary phase field. It shows in Figure 2c that the FeO concentration has little effect on the liquidus temperature at different MnO/SiO<sub>2</sub> ratios, which indicates the liquidus temperature of slag will not change significantly when FeO in slag is partially or fully reduced.

It is clear from Figure 2 that the MnO/SiO<sub>2</sub> ratio has a significant impact on the primary phase field and liquidus temperature. SiO<sub>2</sub> can be added as a flux to control the liquidus temperature of the slag. The effect of SiO<sub>2</sub> or MnO/SiO<sub>2</sub> on the liquidus temperature is shown in Figure 3. It can be seen that the autogenic residue without SiO2 flux has a liquidus temperature of around 1570 °C, which requires an operation temperature of above 1600 °C to keep the slag fully liquid. By adding SiO<sub>2</sub> flux, the primary phase field changes from mono-oxide to olivine 2(Ca, Mn)O·SiO<sub>2</sub> and rhodonite (Ca, Mn)O·SiO<sub>2</sub>. A large operating window with the liquidus temperature below 1300 °C can be obtained with 10 to 50 wt% SiO<sub>2</sub> addition.



**Figure 2.** Effects of CaO,  $Al_2O_3$ , and "FeO" on liquidus temperatures at different MnO/SiO<sub>2</sub> ratios in equilibrium with iron calculated with FactSage 7.2, (**a**) CaO in the slag,(**b**)  $Al_2O_3$  in the slag,(**c**) FeO in the slag.



**Figure 3.** The effect of weight percent of  $SiO_2$  per 100 g nodules added as a flux on the liquidus temperature at a fixed ratio of other components.

CaO	MnO/SiO <sub>2</sub>	"FeO"	Al <sub>2</sub> O <sub>3</sub>	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	1.0	9.6	6.2	4.6	4.0	1.6
CaO	MnO/SiO <sub>2</sub>	"FeO"	Al <sub>2</sub> O <sub>3</sub>	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	2.0	9.6	6.2	4.6	4.0	1.6
CaO	MnO/SiO <sub>2</sub>	"FeO"	Al <sub>2</sub> O <sub>3</sub>	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	3.0	9.6	6.2	4.6	4.0	1.6
Al <sub>2</sub> O <sub>3</sub>	MnO/SiO <sub>2</sub>	"FeO"	CaO	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	1.0	9.6	2.9	4.6	4.0	1.6
Al <sub>2</sub> O <sub>3</sub>	MnO/SiO <sub>2</sub>	"FeO"	CaO	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	2.0	9.6	2.9	4.6	4.0	1.6
Al <sub>2</sub> O <sub>3</sub>	MnO/SiO <sub>2</sub>	"FeO"	CaO	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	3.0	9.6	2.9	4.6	4.0	1.6
"FeO"	MnO/SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	CaO	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	1.0	6.2	2.9	4.6	4.0	1.6
"FeO"	MnO/SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	CaO	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	2.0	6.2	2.9	4.6	4.0	1.6
"FeO"	MnO/SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	CaO	MgO	Na <sub>2</sub> O	K <sub>2</sub> O
0–15	3.0	6.2	2.9	4.6	4.0	1.6

Table 2. The composition of the slag used for FactSage calculation (wt%).

# 2.2.2. Effect of Carbon Addition on the Reduction Degree

The effect of carbon addition on the extension of the reduction was examined by FactSage 7.2 with the SiO<sub>2</sub> addition of 10 wt% of ore weight at 1350 °C. The recovery rate and alloy composition as a function of carbon addition are shown in Figure 4. It can be seen from Figure 4 that the NiO and CuO are first reduced and Ni and Cu concentrations in the alloy reached a maximum. With increasing carbon addition, FeO<sub>x</sub> and CoO are reduced and their concentrations in the alloy gradually increase. At 1.2 wt% carbon addition, all NiO, CuO and CoO are reduced and the alloy composition is 50Ni-12Cu-6Co-32Fe. Further addition of carbon will continuously reduce FeO<sub>x</sub> and increase Fe in the alloy. Accordingly, the concentrations of Ni, Co and Cu will decrease as the total alloy increases. At carbon 2.5 wt%, all

FeO<sub>x</sub> is reduced and Fe in the alloy reaches maximum. MnO starts to be reduced and carbon starts to dissolve into the alloy. Further addition of carbon will increase Mn and carbon concentrations in the alloy, and decrease the concentrations of other components in the alloy. Over 6.5 wt% carbon addition, the alloy composition will be constant, which is considered as the critical amount of graphite. Thus, adding graphite greater than 6.5 wt% would not change the composition of alloy significantly.



**Figure 4.** Reduction degree and alloy composition as a function of carbon addition for 100 g ore and 10 g SiO<sub>2</sub> at 1350 °C.

#### 2.3. Experimental Procedure

A vertical tube furnace with lanthanum chromate heating elements was used for the experiments. For each experimental run, 10 g of nodules were crushed into powder, and well mixed with flux and reductant in an agate mortar then pressed into briquette. Pure graphite powder (99.99 wt%) was used as the reductant. The flux  $SiO_2$  and reductant graphite were added as per ore weight. The addition of SiO<sub>2</sub> was adjusted based on the MnO/SiO<sub>2</sub> ratios and the liquidus temperatures showing in Figure 3. The experimental conditions are shown in Table 3. The pelletized sample was then placed into an alumina crucible (ID: 18 mm, H: 25 mm). The sample was heated and pre-reduced under an Ar atmosphere at 1000 °C for 60 min. Then, each pre-reduced sample was heated immediately to the smelting temperature and held there for 60 min. A Pt-30% Rh/Pt-6% Rh thermocouple placed in an alumina sheath was located adjacent to the sample to monitor the temperature. The temperature was controlled to an accuracy of  $\pm 2$  K. The samples were directly quenched into the water after the smelting reduction. After drying, they were mounted in resin and polished for metallographic analysis. The microstructures were examined by scanning electron microscopy coupled with X-ray energy-dispersive spectroscopy analysis (SEM-EDS). Compositions of the liquid and solid phases were measured by electron probe X-ray microanalysis (EPMA). The EPMA operation was conducted at an accelerating voltage of 15 kV and a probe current of 15 nA. The standards used for analysis were spinel (MgAl<sub>2</sub>O<sub>4</sub>) for Al and Mg, wollastonite (CaSiO<sub>3</sub>) for Ca and Si, chalcopyrite (FeCuS<sub>2</sub>) for Cu, rutile (TiO<sub>2</sub>) for Ti, chromite (FeCrO<sub>4</sub>) for Cr, albite (NaAlSi<sub>3</sub>O<sub>8</sub>) for Na, orthoclase (KAlSi<sub>3</sub>O<sub>8</sub>) for K, olivine (Ni<sub>2</sub>SiO<sub>4</sub>) for Ni, spessartine (Mn<sub>3</sub>Al<sub>2</sub>Si<sub>3</sub>O<sub>12</sub>) for Mn, hematite (Fe<sub>2</sub>O<sub>3</sub>) for Fe, and pure Co for Co. The ZAF correction procedure, which is a series of multiplicative factors that account for the effects of atomic number (Z-stopping power, back-scattering factor and X-ray production power), absorption (A) and fluorescence (F), built-in the electron-probe software was applied automatically to the matrix correction. The average accuracy of the EPMA measurements is within ±1 wt%. The alloy droplets were collected to be measured by Inductively Coupled Plasma Optical Emission Spectrometer (ICP-OES).

Nodules	Flux	Reductant	Crucible	Pre-Re	duction	Reduction Smelting		
(G)	SiO <sub>2</sub>	Graphite	crucible	Temp. (°C)	Time (Min)	Temp. (°C)	Time (Min)	
10	9%	0.8%	Al <sub>2</sub> O <sub>3</sub>	1000	60	1420	60	
10	9%	1.2%	$Al_2O_3$	1000	60	1420	60	
10	9%	12%	$Al_2O_3$	1000	60	1420	60	
10	17.5%	4%	$Al_2O_3$	1000	60	1350	60	
10	17.5%	8%	$Al_2O_3$	1000	60	1350	60	
10	35%	4%	$Al_2O_3$	1000	60	1300	60	
10	35%	8%	$Al_2O_3$	1000	60	1300	60	
10	15%	2%	$Al_2O_3$	1000	60	1350	60	
10	15%	4%	$Al_2O_3$	1000	60	1350	60	
10	15%	6%	$Al_2O_3$	1000	60	1350	60	
10	15%	8%	$Al_2O_3$	1000	60	1350	60	
10	10%	4%	$Al_2O_3$	1000	60	1350	60	
10	5%	4%	$Al_2O_3$	1000	60	1350	60	
	Nodules (G) 10 10 10 10 10 10 10 10 10 10 10 10 10	Nodules (G) Flux   10 9%   10 9%   10 9%   10 9%   10 17.5%   10 17.5%   10 35%   10 35%   10 15%   10 15%   10 15%   10 15%   10 15%   10 15%   10 10%   10 5%	Nodules (G) Flux Reductant   10 9% Graphite   10 9% 1.2%   10 9% 1.2%   10 9% 12%   10 9% 12%   10 17.5% 4%   10 17.5% 8%   10 35% 8%   10 35% 8%   10 15% 2%   10 15% 6%   10 15% 8%   10 15% 8%   10 15% 8%   10 15% 8%   10 15% 8%   10 15% 8%   10 5% 4%	$\begin{array}{c c c c c c c c c c c c c c c c c c c $	$\begin{array}{c c c c c c c c c c c c c c c c c c c $	$ \begin{array}{c c c c c c c c c c c c c c c c c c c $	$ \begin{array}{ c c c c c c c c c c c c c c c c c c c$	

Table 3. Experimental conditions for nodule reduction.

Note: Exp., experienmnet; Temp., temperature.

#### 3. Results and Discussion

### 3.1. Pre-Reduction

Pre-reduction was conducted to calcinate nodules and enable nickel, copper, cobalt and iron oxides to be partly reduced. The microstructures of samples ND2 and ND9 after pre-reduction at 1000 °C for 60 min are shown in Figure 5. In Figure 5a, the sample ND2 contains pyroxene ((Si, Ti, Mn)O<sub>2</sub>·(Al, Fe)<sub>2</sub>O<sub>3</sub>·(Mn, Fe, Mg, Ca, Ni)O), rhodonite ((Mn, Ca, Mg)O·SiO<sub>2</sub>), spinel ((Al, Fe)<sub>2</sub>O<sub>3</sub>·(Mn, Fe, Mg, Ca, Ni)O),), olivine 2(Mn, Mg, Ca, Ni, Fe)O·SiO<sub>2</sub>, and glass phases. At 1000 °C, the mineral SiO<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub> incorporated with low melting point oxides K<sub>2</sub>O and Na<sub>2</sub>O may diffuse and react with MnO-FeO phases. Some reactions occurred in the local area near the pore, where olivine and liquid together with rhodonite were formed. Metals were not observed in the samples after pre-reduction due to the insufficient reductant addition. In Figure 5b, alloy droplets were observed by the reduction with more reductant addition. Olivine and liquid phases were formed in the sample ND9. The alloy and liquid compositions are 38.4Ni-36.1Fe-19.7Cu-3.4Co-2.3Mn and 49.7SiO<sub>2</sub>-13.4MnO-11.8Al<sub>2</sub>O<sub>3</sub>-12.9Na<sub>2</sub>O-5.5K<sub>2</sub>O-2.5FeO-2.8CaO-0.7MgO-0.6TiO<sub>2</sub>, respectively.

## 3.2. Reduction Smelting

## 3.2.1. Effect of Reductant Addition on the Alloys and Slags

The smelting process was conducted at high temperatures to further reduce nickel, copper, cobalt, and iron oxides to form the required alloys and slags. Figure 6 shows the pictures and microstructures of samples ND8, ND9, and ND10 after smelting at 1350 °C with a carbon addition of 2, 4, and 6 wt%, respectively. It can be seen from Figure 6a that in sample ND8, the slag contains spinel crystals and the alloy is not observed. In samples ND9 and ND10, the slag is fully liquid and alloy sphere (around 5 mm in diameter) settled at the bottom of the crucible can be easily separated from the slag, and small alloy droplets are also present in the slag. A small amount of graphite was left after smelting reduction in sample ND10, which indicates the addition of graphite was in excess. Comparing Figure 6a–c, when the addition of the reductant was insufficient, the liquidus temperature of slag was much higher due to the presence of Fe<sup>3+</sup>. Consider the results that no alloy was observed with 2 wt% graphite addition and excess graphite was observed with 6 wt% graphite addition. Therefore, an optimal addition of graphite is around 4 wt% per residues in terms of alloy morphology and full-liquid slag. The effect of the addition of graphite on the reduction extent of copper, nickel, cobalt, iron and manganese oxides will be discussed in the following section.



**Figure 5.** Microstructure of reduction sample after pre-reduction at 1000 °C for 60 min: (**a**) sample ND2 with 1.2% graphite addition; (**b**) sample ND9 with 4% graphite addition.



(a) Sample ND8 (2 wt% graphite addition)



(b) Sample ND9 (4 wt% graphite addition)

Figure 6. Cont.



(c) Sample ND10 (6 wt% graphite addition)

**Figure 6.** Pictures and microstructures of quenched samples after smelting at 1350 °C. (**a**) Sample ND8 (2 wt% graphite addition); (**b**) sample ND9 (4 wt% graphite addition); (**c**) sample ND10 (6 wt% graphite addition).

# 3.2.2. Effect of Flux Addition

 $SiO_2$  as a flux was added to decrease the liquidus temperature of slag, which enables the slag to be a fully-liquid at the smelting temperature. FactSage calculations shown in Figure 3 demonstrated that the liquidus temperature of the slag decreases rapidly with the addition of up to 50 wt% SiO<sub>2</sub>. Different amounts of SiO<sub>2</sub> were added to verify the predictions at a given temperature of 1350 °C. The dependence of SiO<sub>2</sub> addition on the formation of slags with the same amount of graphite additive is shown in Figure 7. It is evident from Figure 7a-c that the slags were fully liquid at the experimental temperature with  $SiO_2$  addition over 10 wt%. However, as shown in Figure 7d, with the  $SiO_2$  addition down to 5 wt%, the liquidus temperature of the slag was above the experimental temperature resulting in the presence of solid. The primary monoxide phase precipitated in the slag, and tiny reduced alloy blebs are distributed in the periphery of the monoxide solids. In addition to the liquidus temperature, the viscosity of slag can influence the settlement of alloys and tapping of slags. The viscosity of slags in the full-liquid state at 1350 °C with the addition of different amounts of SiO<sub>2</sub> was calculated by FactSage 7.2. Figure 8 shows the viscosity of slag increases with increasing  $SiO_2$ . The addition of  $SiO_2$  needs to be minimized to keep the slags in a full liquid state with a relatively low viscosity. Overall, SiO<sub>2</sub> addition can be optimized by several factors including the liquidus temperature of slag for operational temperature, the viscosity of slag for separation of alloy and slag tapping, and MnO-grade of the residue slag for further production of ferromanganese or silicomanganese alloys.



(c) Sample ND12 (adding 10 wt% SiO<sub>2</sub>)

(**d**) Sample ND13 (adding 5 wt% SiO<sub>2</sub>)

Figure 7. Microstructures of slags of the quenched samples with different SiO<sub>2</sub> addition.



Figure 8. The viscosity of slags as a function of SiO<sub>2</sub> addition (wt%) of ore weight.

# 3.3. Effect of Graphite Addition on the Alloy Grade

The liquidus temperature of slags has been predicted by FactSage, and verified by the experiments with a variation of SiO<sub>2</sub> flux and carbon addition in the above sections. The liquidus temperature of slags decreases with the SiO<sub>2</sub> addition and reduction of Fe<sup>3+</sup> to Fe<sup>2+</sup> in the slag by carbon. The reduction degree of FeO in the slag and reduced Fe in the alloy as a function of carbon are shown in Figure 9, where experimental results of samples ND8 to ND11 with 15 wt% SiO<sub>2</sub> addition at 1350 °C are

compared with FactSage calculation. The compositions of alloys and phases present in the quenched samples are listed in Table 4. Figure 9 shows FeO concentration in the slag decreases along with carbon addition, while Fe concentration in the alloy increases for experimental results. A significant difference was found between experimental results and FactSage calculation for the FeO reduction. FactSage calculation indicates FeO in the slag is fully reduced by 3 wt% carbon, while more carbon is needed in the experiments, in which approximately 8 wt% carbon can fully reduce the FeO in the slag. One of the reasons is that a small fraction of MnO was reduced during the carbothermic reduction, which resulted in carbon more than theoretical is needed. Moreover, the metallic oxides could present in multivalent forms (e.g., MeO<sub>x</sub>) rather than bivalent form (e.g., MeO) that require additional carbon to reduce them.



**Figure 9.** A comparison between experimental results and FactSage calculation on the FeO and Fe concentration in the slag and alloy with the addition of carbon.

Full reduction of CuO, NiO, and CoO are desired to allow for high recovery rates, while the iron oxide can be partly/sufficiently reduced depending on the amount of reductant. Recovery of Cu, Ni, and Co is taken as a whole, and the alloy grade is defined as the weight percentage of the sum of Cu, Ni, and Co, (Cu + Ni + Co). The effect of graphite addition on the alloy grade is shown in Figure 10, where the data is plotted from Table 4. The alloy grade decreases with increasing graphite addition. The sequential ease of reduction of metal oxides is (Cu, Ni, Co) > Fe > (Mn, Si). The (Cu + Ni + Co) concentration gradually decreases in the alloy along with the reduction of Fe and Mn oxides. Figure 10 can be divided into three areas, where "Area I" represents the insufficient addition of reductant and no alloys are reduced, "Area III" means excess addition of reductant, beyond that, MnO (or even SiO<sub>2</sub>) in the slag will be significantly reduced into the alloys. SiO<sub>2</sub> and MnO are not required to be reduced at the smelting reduction stage to produce high-grade Fe-Ni-Cu-Co alloy. "Area II" in Figure 10 is thus the working area to optimize the addition of reductant to control the alloy grade and recovery of valuable metals. Within "Area II", iron oxide is gradually reduced into the alloy along with graphite additive. Iron oxide is almost fully reduced with 8 wt% graphite addition.



Figure 10. Relationship between alloy grade (Cu + Ni + Co) and graphite addition.

Sample No.	Slag	MnO	$SiO_2$	$Al_2O_3$	"FeO"	MgO	CaO	Na <sub>2</sub> O	K <sub>2</sub> O	TiO <sub>2</sub>	Cu <sub>2</sub> O	NiO	CoO
ND1	Glass	36.4	29.8	8.2	5.6	3.8	3.0	7.6	1.7	0.6	1.9	1.2	0.2
	Spinel	62.2	0.5	8.2	19.7	3.3	0.1	0.0	0.0	0.8	0.7	4.1	0.4
ND2	Glass	36.0	30.4	9.4	4.4	3.7	3.1	7.9	1.6	0.5	1.9	0.8	0.2
	Spinel	63.1	0.4	6.8	20.9	3.0	0.0	0.0	0.0	0.8	0.5	4.0	0.4
ND3	Glass	36.1	32.1	15.2	0.1	4.8	3.3	6.2	1.4	0.8	0.0	0.0	0.0
ND4	Glass	41.4	31.7	10.5	3.8	4.1	2.4	4.1	1.1	0.7	0.1	0.0	0.0
ND5	Glass	44.0	34.3	6.8	1.2	4.3	2.6	4.8	1.3	0.7	0.0	0.1	0.0
ND6	Glass	36.1	41.7	5.0	6.2	3.3	2.0	3.7	1.1	0.5	0.1	0.1	0.0
ND7	Glass	38.0	44.7	4.8	0.7	3.6	2.2	4.1	1.2	0.5	0.0	0.0	0.0
NIDO	Glass	40.3	32.2	6.3	6.3	3.9	2.5	3.1	1.7	0.5	1.4	1.5	0.3
ND8	Spinel	47.1	0.4	9.8	31.2	2.8	0.0	0.0	0.0	1.5	0.2	6.7	0.4
ND9	Glass	42.0	31.4	11.6	3.5	3.9	2.4	3.0	1.5	0.6	0.1	0.0	0.0
ND10	Glass	42.9	32.8	7.6	2.7	4.2	2.5	5.5	1.1	0.6	0.0	0.0	0.0
ND11	Glass	43.1	33.0	8.9	0.4	4.2	2.5	5.8	1.1	0.6	0.1	0.0	0.0
ND12	Glass	44.5	28.1	11.2	2.1	4.2	2.5	5.5	1.0	0.6	0.0	0.0	0.0
	Glass	47.1	22.9	10.5	3.4	4.2	2.8	6.7	1.1	0.7	0.0	0.0	0.0
ND13	Monoxide	86.5	0.0	0.3	6.8	5.9	0.1	0.1	0.0	0.1	0.1	0.1	0.1
Sample No.	Alloy	Mn	Si	Al	Fe	Mg	Ca	Na	К	Ti	Cu	Ni	Со
ND3	Alloy	60.7	1.0	0.0	25.9	0.0	0.0	0.0	0.0	0.1	5.6	5.9	0.8
ND4	Alloy	0.3	0.0	0.0	53.2	0.0	0.0	0.0	0.0	0.0	20.2	23.1	3.2
ND5	Alloy	1.1	0.0	0.0	67.9	0.0	0.0	0.0	0.0	0.0	12.8	15.5	2.6
ND6	Alloy	0.1	0.0	0.0	36.1	0.0	0.0	0.0	0.0	0.0	24.8	34.5	4.4
ND7	Alloy	0.3	0.0	0.0	68.6	0.0	0.0	0.0	0.0	0.0	10.6	17.6	2.8
ND9	Alloy	0.2	0.0	0.0	57.7	0.0	0.0	0.0	0.0	0.0	18.3	20.7	3.1
ND10	Alloy	0.4	0.0	0.0	64.4	0.0	0.0	0.0	0.0	0.0	14.4	17.7	2.8
ND11	Alloy	0.9	0.0	0.0	66.2	0.0	0.0	0.0	0.0	0.0	13.8	16.5	2.5
ND12	Alloy	0.5	0.0	0.0	57.3	0.0	0.0	0.0	0.0	0.0	13.5	25.1	3.6
ND13	Alloy	0.5	0.0	0.0	59.2	0.0	0.0	0.0	0.0	0.0	25.1	12.6	2.5

Table 4. The compositions of the phases present in the quenched samples (wt%).

## 4. Conclusions

An efficient process for the smelting reduction of polymetallic nodules at a lower temperature was investigated by the optimization of both slag and alloy systems. Essential findings are summarized as follows:

- (1) The as-received nodules mainly contain Mn and Fe oxides in striation. Si is incorporated with Al, Na, K and Ca as impurity minerals. Valuable metal elements such as Cu, Co and Ni in low concentrations are uniformly distributed in the nodule matrix. Ba is found to be associated with S, most likely as barite.
- (2) FactSage software was used to simulate/predict the liquidus temperature of the slag system, which is not sensitive to FeO, CaO and Al<sub>2</sub>O<sub>3</sub>, but decreases most significantly with decreasing MnO/SiO<sub>2</sub> mass ratio. The SiO<sub>2</sub> flux to decrease the liquidus temperature of slag was experimentally verified.
- (3) The alloy grade was found to decrease with increasing graphite addition. The optimal reduction was achieved by adding a 4 wt% graphite at the MnO/SiO<sub>2</sub> mass ratio of 1.6 in slag. The most effective metal-slag separation was achieved at 1350 °C. This relatively low-temperature smelting process for the utilization of polymetallic nodules is promising for industrial applications.

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