Abstract: A new series of conjugated carbazole polymers based on the 1,8-carbazolylene unit was synthesized by the Pd-catalyzed polycondensation between the 1,8-diiodocarbazole derivative and various bifunctional counter comonomers. An alkyne spacer was found to be a key to increasing the molecular weight of the resulting polymers. All the obtained polymers showed good solubilities in the common organic solvents, and they were fully characterized by Gel permeation chromatography (GPC), and $^1$H NMR and infrared (IR) spectroscopies. The UV-vis absorption and fluorescence spectra revealed the relationship between the chemical structure and effective conjugation length. The efficiency order of the carbazole connectivity was 2,7-carbazolylene > 1,8-carbazolylene > 3,6-carbazolylene. The electrochemical properties of these polymers suggested the relatively facile oxidation at ca. +0.5–0.7 V vs. Fc/Fc$^+$ or a high potential as p-type semiconductors. The combination of the electrochemical oxidation potentials and the optical band gaps allowed us to estimate the HOMO and LUMO levels of the polymers. It was shown that the energy levels of the 1,8-carbazole-based conjugated polymers can be tunable by selecting the appropriate comonomer structures.
Keywords: carbazole; conjugated polymer; electrochemistry; UV-vis absorption

1. Introduction

In recent years, conjugated carbazole polymers have attracted increasing attention because of their strong fluorescence, excellent hole transporting ability, and the related application possibilities in organic electronic devices [1,2]. It is known that the connectivity of the carbazole moiety significantly affects the polymer properties. Poly(3,6-carbazole)s are an initially reported class of conjugated carbazole polymers, in which the highest electron density positions are connected either directly or through π-spacers (Figure 1) [3-5]. They are suitable for electrochemical and phosphorescent applications due to the relatively stable oxidized or excited states [6,7]. When an efficient synthetic route for achieving 2,7-dihalocarbazole derivatives was established [8,9], poly(2,7-carbazole)s emerged as an alternative to blue-light fluorescent polyfluorene derivatives, which possessed a photodegradation problem leading to a different color emission and optical instability [10,11]. Later, the conjugated copolymers of the electron-donating 2,7-carbazolylene and electron-accepting π-systems were found to be very suitable as a p-type semiconductor for bulk heterojunction solar cells [12-16]. These findings stimulated renewed interests in the development of conjugated carbazole polymers with different connectivities. For example, poly(2,9-carbazole)s and poly(3,9-carbazole)s were recently synthesized and their properties were comprehensively investigated [17-20].

In 2007, we started a project focused on carbazole polymers [21-24]. During the course of the study, we realized that the 1,8-positions of the carbazole had not been noted in polymer chemistry. Therefore, we designed some new poly(1,8-carbazole)s, e.g., alkyne-linked conjugated poly(1,8-carbazole)s, and it was found that these carbazole polymers possess characteristics of both the poly(3,6-carbazole)s and poly(2,7-carbazole)s, as revealed by the long wavelength absorption, strong fluorescence, and significant electrochemical activities [25,26]. To diversify the list of the promising poly(1,8-carbazole)s, we also reported the directly-linked poly(1,8-carbazole) [27]. In this paper, conjugated copolymers of 1,8-carbazolylene and various arylene units are reported. Their optical and electrochemical properties were investigated in detail and their chemical structure-property relationships are summarized.

Figure 1. Conjugated carbazole polymers with various connectivities.
2. Results and Discussion

2.1. Polymer Synthesis

A key monomer, the 1,8-diodocarbazole derivative 1, was designed and synthesized according to a previous method [25]. The 3, 6, and 9 positions of the carbazole moiety were substituted by alkane groups in order to improve their chemical stability toward oxidation and solubility in organic solvents.

This bifunctional monomer was subjected to polycondensation with other bifunctional monomers under the Suzuki coupling conditions or the Sonogashira coupling conditions. In a typical Suzuki polycondensation procedure, a mixture of 1 and a bis-boronic ester functionalized comonomer in THF/aq. K$_2$CO$_3$ solution was heated under reflux for 72 h in the presence of Pd(PPh$_3$)$_4$. The previous synthesis of a directly-linked poly(1,8-carbazole) was based on the Yamamoto polycondensation of 1 [Ni(cod)$_2$, cod, bpy, DMF] [27]. In contrast to this method, which only produces homopolymers or random copolymers, the Suzuki polycondensation allows for the incorporation of various comonomers in an alternating sequence. Thus, copolymers P1–P3 were synthesized from 1 and different comonomers 2-4 (Scheme 1). The molecular weights of the resulting polymers were determined by Gel permeation chromatography (GPC) (Table 1). The results revealed the oligomeric molecular weights of these polymers, which are similar to previous reports [27]. This might be caused by unstable intermediates in the catalytic cycle, leading to the hydro-dehalogenated oligomers, or the steric hindrance of the long alkyl chain attached to the carbazole nitrogen.

<table>
<thead>
<tr>
<th>Polymer</th>
<th>Yield (%)</th>
<th>$M_n$ $^a$</th>
<th>$M_n$ $^a$</th>
<th>$M_w/M_n$ $^a$</th>
<th>$T_{5%}$ (°C) $^b$</th>
<th>$T_g$ (°C) $^c$</th>
</tr>
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<tbody>
<tr>
<td>P1</td>
<td>28</td>
<td>1,400</td>
<td>1,300</td>
<td>1.1</td>
<td>259</td>
<td>-</td>
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<tr>
<td>P2</td>
<td>61</td>
<td>1,700</td>
<td>1,400</td>
<td>1.2</td>
<td>276</td>
<td>-</td>
</tr>
<tr>
<td>P3</td>
<td>15</td>
<td>2,000</td>
<td>1,800</td>
<td>1.1</td>
<td>276</td>
<td>63</td>
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<tr>
<td>P4</td>
<td>99</td>
<td>6,900</td>
<td>4,300</td>
<td>1.6</td>
<td>340</td>
<td>77</td>
</tr>
<tr>
<td>P5</td>
<td>100</td>
<td>13,000</td>
<td>7,200</td>
<td>1.8</td>
<td>380</td>
<td>80</td>
</tr>
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</table>

$^a$ Determined by GPC (THF eluent, calibrated by polystyrene standards).

$^b$ Temperature at which a 5% weight loss occurred upon heating.

$^c$ Glass transition temperature determined by the second heating scan of the DSC measurements.

It was previously shown that an alkyne spacer is effective for improving the molecular weights of the poly(1,8-carbazole)s [27]. Therefore, new alkyne-linked 1,8-carbazolylene-based copolymers were prepared from 1 and diethynylcarbazole derivatives 5/6 under the Sonogashira coupling conditions (Scheme 1). In these cases, the molecular weight ($M_w$) reasonably increased to 6900 for P4 and 13000 for P5.

The obtained polymers showed good solubilities in the common organic solvents as well as film formation capability. Therefore, they were fully characterized by NMR and Infrared (IR) spectroscopies. As an example, the $^1$H NMR and IR spectra of P4 are shown in Figure 2. The $^1$H NMR spectrum showed complex multiplet peaks in the aromatic region due to the two different connectivity modes of the carbazole unit. Alkyl peaks ascribed to the 3, 6, and 9-positions of the carbazole unit were reasonably detected. The IR spectrum was consistent with these data. The C-H stretching
vibration of the alkyl chains appeared in the range from 2953 to 2851 cm$^{-1}$. Moreover, a very weak alkyne vibration peak ascribed to the C≡C moieties was observed at 2199 cm$^{-1}$.

**Scheme 1.** Synthesis of conjugated carbazole polymers. (a) Pd(PPh$_3$)$_4$, THF, 2 M aq. K$_2$CO$_3$, $\Delta$, 72 h. (b) PdCl$_2$(PPh$_3$)$_2$, CuI, toluene, iPr$_2$NH, $\Delta$, 24 h.
Figure 2. (a) $^1$H NMR spectrum of P4 in C$_6$D$_6$ and (b) IR spectrum of P4 (KBr).

2.2. Thermal Properties

Thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) measurements were performed to estimate the decomposition temperatures and glass transition temperatures ($T_g$). Both analyses and measurements were conducted at the scanning rate of 10 °C min$^{-1}$ under flowing nitrogen, and the data are summarized in Table 1. The directly-linked carbazole polymer P1 started to decompose at ca. 250 °C, and after the stepwise decomposition, the residual soot amount was ca. 10% at 500 °C (Figure 3). The incorporation of different π-chromophores, such as fluorene and bithiophene derivatives, improved the thermal stability. Thus, the decomposition temperatures ($T_{5\%}$) of P2 and P3 were higher by 17 °C than that of P1. The decomposition temperature increased when alkyne groups were introduced in the polymer main chain. The $T_{5\%}$ of P4 was much higher (340 °C) than that of P1. The thermal stability of P5 showed a similar improvement due to the presence of the alkyne groups. The different decomposition temperatures between P4 and P5 are consistent with the previous result that the 2,7-carbazolylene unit generally shows a higher thermal stability than the 3,6-carbazolylene unit [27]. It is postulated that these decomposition processes include thermally induced rearrangements of the alkyne moieties. The DSC curves of P1 and P2 did not show well-defined transition temperatures, while P3–P5 displayed $T_g$s in the range from 63 to 80 °C.

Figure 3. Thermogravimetric analysis (TGA) curves of P1–P5 at a heating rate of 10 °C min$^{-1}$ under flowing nitrogen.

2.3. Electrochemistry

Cyclic voltammograms (CVs) of all the polymers were measured both in solutions and in the solid film states. Figure 4 shows the CV curves of P1–P5 measured in CH$_2$Cl$_2$ with 0.1 M (nC$_4$H$_9$)$_4$NClO$_4$ at 20 °C, and the oxidation potentials are summarized in Table 2.
Figure 4. Cyclic voltammograms of (a) P1, (b) P2, (c) P3, (d) P4, and (e) P5 in CH$_2$Cl$_2$ with 0.1 M ($n$C$_4$H$_9$)$_4$NClO$_4$ at the scanning rate of 0.1 V s$^{-1}$ under flowing nitrogen. Potentials vs. Fc/Fc$^+$ couple.

![Cyclic voltammograms of P1-P5](image)

Table 2. Summary of the electrochemistry data of poly(1,8-carbazole)\textsuperscript{a}.

<table>
<thead>
<tr>
<th>polymer</th>
<th>E$_{onset}$ (V)</th>
<th>E$^{\circ}$ (V)\textsuperscript{d}</th>
<th>E$_{pa}$ (V)\textsuperscript{d}</th>
<th>E$_{pc}$ (V)\textsuperscript{d}</th>
<th>E$_{onset}$ (V)</th>
<th>E$_{pa}$ (V)</th>
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<tr>
<td>P1</td>
<td>0.44</td>
<td>0.41</td>
<td>0.57</td>
<td>0.25</td>
<td>0.58</td>
<td>0.70</td>
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<td>0.45</td>
<td>0.92</td>
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<td></td>
<td></td>
<td></td>
<td>1.12</td>
<td>0.76</td>
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<td></td>
<td>1.28</td>
<td>0.60</td>
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<tr>
<td>P2</td>
<td>0.55</td>
<td>0.48</td>
<td>0.67</td>
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<td></td>
<td>0.58</td>
<td>1.03</td>
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<td></td>
<td>1.22</td>
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<tr>
<td>P3</td>
<td>0.48</td>
<td>0.51</td>
<td>0.73</td>
<td>0.30</td>
<td>0.75</td>
<td>0.70</td>
</tr>
<tr>
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<td></td>
<td></td>
<td></td>
<td></td>
<td>1.11</td>
<td>0.70</td>
</tr>
<tr>
<td>P4</td>
<td>0.60</td>
<td>0.77</td>
<td>0.86</td>
<td>0.67</td>
<td>1.15</td>
<td>1.25</td>
</tr>
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<td></td>
<td></td>
<td>1.05</td>
<td>0.80</td>
</tr>
<tr>
<td>P5</td>
<td>0.69</td>
<td>0.90</td>
<td>1.03</td>
<td>0.76</td>
<td>1.41</td>
<td>0.97</td>
</tr>
</tbody>
</table>

\textsuperscript{a} Potentials vs. Fc/Fc$^+$ couple. Working electrode: glassy carbon electrode; counter electrode: Pt wire; reference electrode: Ag/Ag$^+$.\n
\textsuperscript{b} Measured in CH$_2$Cl$_2$ with 0.1 M ($n$C$_4$H$_9$)$_4$NClO$_4$ at a scanning rate of 0.1 V s$^{-1}$ under flowing nitrogen.\n
\textsuperscript{c} Measured in CH$_3$CN with 0.1 M ($n$C$_4$H$_9$)$_4$NClO$_4$ at a scanning rate of 0.1 V s$^{-1}$ under flowing nitrogen.\n
\textsuperscript{d} $E^{\circ}$ = $(E_{pc} + E_{pa})/2$, where $E_{pc}$ and $E_{pa}$ correspond to the cathodic and anodic peak potentials, respectively.
Carbazole polymers are redox active polymers in the anodic direction. The first oxidation potentials of all the polymers were almost reversible on the CV time scale. All polymers showed multiple oxidation peaks ascribed to the 1,8-carbazolylene and comonomer moieties. The first oxidations are presumably derived from the carbazole moieties. The first oxidation potentials of P1–P3 were lower than those of P4 and P5. It is postulated that direct substitution of the carbazole 1,8-positions by electron-donating groups, such as carbazole and bithiophene units, lowered the oxidation potential of the 1,8-carbazolylene unit. In contrast, this activation effect by electron-donating substituents is weak in the case of P4 and P5 due to the alkyne spacers. The higher oxidation potential of P5 than that of P4 is consistent with a previous report that poly(2,7-carbazole) derivatives possess deeper HOMO levels than the poly(3,6-carbazole) derivatives [28].

To study the redox activities of P1–P5 in the solid state, the polymer cast films were prepared on a glassy carbon working electrode and subjected to measurements in CH2CN with 0.1 M (nC4H9)4NCIO4. All the polymer films were insoluble in the electrolyte solution and displayed irreversible oxidation waves. The first oxidation peak potentials of P1–P3 shifted anodically compared to those in the solutions, while P4 and P5 exhibited lower first oxidation peaks or onset potentials than in the solutions. This result suggests different packing modes of the solid states between the directly-linked poly(1,8-carbazole) derivatives and the alkyne-linked poly(1,8-carbazole) derivatives. The coplanarity of the former derivatives would decrease due to the twisted linkage of the π-chromophores, while the free rotation of the alkyne spacers was supposed to facilitate relatively planar packings of the latter derivatives, leading to a decrease in the oxidation potentials in the solid state.

2.4. UV-vis and Fluorescence Spectroscopies

The UV-vis absorption and fluorescence spectra of P1–P5 were measured both in CH2Cl2 and in thin film state (Figure 5). The spectroscopic data are summarized in Table 3.

The longest wavelength absorption maximum (λmax) of P1 was detected at 362 nm in CH2Cl2. P2 and P3 with a linearly conjugated spacer of the fluorene or bithiophene unit showed a slight bathochromic shift in λmax (366 nm). This bathochromic shift was more significant when the most intense emission peak maxima (λem) were noted. The λem of P1 was observed at 392 nm, and the value shifted to 412 nm for P2 and 467 nm for P3. Accordingly, the Stokes shift calculated from the λmax and λem values increased in the order of P1 < P2 < P3. There was a tendency that the quantum yields (Φem) in CH2Cl2 increased with the decreasing Stokes shift values.

It was previously demonstrated that an alkyne group is a superior spacer to expand the effective conjugation length of conjugated carbazole polymers [23,27]. Based on these reports, P4 and P5 displayed the λmax at 378 and 408 nm, respectively. It is also interesting to compare the different connectivity modes of the carbazole unit. The λmax of poly[(3,6-di-tert-butyl-N-hexadecyl-1,8-carbazolylene)ethynylene] P7 (Mw = 9,400, Mn = 4,100) was reported to be 389 nm, which is intermediate between these values (Figure 6) [27]. Although the molecular weight of P5 is slightly larger than those of P4 and P7, the data suggested that the order of the effective conjugation length of the alkyne-linked poly(1,8-carbazole) is P4 < P7 < P5. In other words, the most efficient π-spacer is the 2,7-carbazolylene unit, followed by the 1,8-carbazolylene and 3,6-carbazolylene units. A similar tendency was observed for the fluorescence spectra. Both P4 and P5 exhibited two emission maxima
in CH₂Cl₂. The higher energy emission peaks originated from single polymers, while the lower energy emission peaks were presumably derived from the excimers. In the thin film states, the single polymer-based emission completely disappeared and a broad low energy emission was detected at 616 nm for P4 and 569 nm for P5, suggesting the excimer formation.

**Figure 5.** Normalized UV-vis absorption in CH₂Cl₂ measured at 10⁻⁶ M repeat unit⁻¹ (a) and in film (c) and fluorescence spectra in CH₂Cl₂ measured at 10⁻⁶ M repeat unit⁻¹ (b) and in film (d) of P1–P5.

Table 3. Summary of the optical properties of poly(1,8-carbazole)s.

<table>
<thead>
<tr>
<th>polymer</th>
<th>in CH₂Cl₂</th>
<th></th>
<th></th>
<th>Film</th>
</tr>
</thead>
<tbody>
<tr>
<td>polymer</td>
<td>λ&lt;sub&gt;max&lt;/sub&gt; (nm)</td>
<td>λ&lt;sub&gt;em&lt;/sub&gt; (nm)</td>
<td>Stokes shift (cm⁻¹)</td>
<td>Φ&lt;sub&gt;f,sol&lt;/sub&gt;&lt;sup&gt;a&lt;/sup&gt;</td>
</tr>
<tr>
<td>P1</td>
<td>362</td>
<td>392 [302]</td>
<td>2114</td>
<td>0.250</td>
</tr>
<tr>
<td>P2</td>
<td>366</td>
<td>412 [317]</td>
<td>3050</td>
<td>0.419</td>
</tr>
<tr>
<td>P3</td>
<td>366</td>
<td>467 [366]</td>
<td>5909</td>
<td>0.125</td>
</tr>
<tr>
<td>P4</td>
<td>378</td>
<td>410, 497 [315]</td>
<td>2065</td>
<td>0.352</td>
</tr>
<tr>
<td>P5</td>
<td>408</td>
<td>432, 457&lt;sup&gt;b&lt;/sup&gt; [348]</td>
<td>1362</td>
<td>0.496</td>
</tr>
</tbody>
</table>

<sup>a</sup> Determined against quinine sulfate in 0.05 M aqueous H₂SO₄ (Φ<sub>f</sub> = 0.546) as a standard.

<sup>b</sup> Shoulder.
The energy diagrams of P1–P5 were determined from the electrochemistry data and UV-vis absorption spectra (Figure 7). The HOMO levels were calculated from the \(E_{\text{onset}}\) values of the solution CVs based on the assumption of \(\text{Fc/Fc}^+ = -4.80\) eV. The LUMO levels were estimated from the HOMO levels and the optical band gaps in CH\(_2\)Cl\(_2\). The diagrams revealed that the HOMO levels of P1–P5 are similar at \(-5.37 \pm 0.12\) eV, while the LUMO levels are quite dependent on the comonomer or spacer structures. In particular, the presence of alkyne spacers led to a considerable lowering of the polymer LUMO levels. To obtain further insight into the chemical structure-energy level relationship, the effects of the different connectivities of the carbazole units was considered. The HOMO and LUMO levels of poly(3,6-di-tert-butyl-N-hexadecyl-1,8-carbazolylene) P6 were reported to be \(-5.56\) and \(-2.27\) eV, respectively [27]. A comparison of this energy level to that of P1 suggested that the copolymerization of 1 with the 3,6-carbazolylene unit 2 results in an elevation of both the HOMO and LUMO levels. A higher HOMO level generally facilitates oxidation or p-type doping. Judging from the estimated energy levels, it is thus expected that P1 is a better p-type semiconductor than P6. The energy levels of P4, P5, and P7 were similarly compared. The HOMO and LUMO levels of P7 were reported to be \(-5.05\) and \(-2.58\) eV, respectively [27]. Replacement of half of the 1,8-carbazolylene units of P7 by the 3,6-carbazolylene or 2,7-carbazolylene units resulted in an elevation of the HOMO level and a lowering of the LUMO level, namely decreased band gaps. The extent of the lowered LUMO level is more significant for P4 than for P5. A comparison between P1 and P4 also suggested that the insertion of one alkyne spacer into the polymer repeat unit lowers the LUMO level by ca. 1.3 eV.

**Figure 6.** Chemical structure of P6 and P7.

**Figure 7.** Schematic energy level diagrams of P1–P5.
Although P1–P5 did not exhibit solvatochromic behaviors in homogeneous solutions, solvent effects on the optical properties were further pursued. The addition of methanol to the homogeneous polymer solution is a general method to investigate the aggregation behaviors [29,30]. The emission spectra of the alkyne-linked polymers P4 and P5 were completely different between the solutions and the solid films. Therefore, the UV-vis absorption and fluorescence spectra of these polymers were measured as a function of the methanol content. The addition of methanol to the homogeneous polymer solutions in CH$_2$Cl$_2$ led to a gradual bathochromic shift in the absorption spectra (Figures 8a and 8c). This behavior became more significant in the fluorescence spectra. The initial single polymer-based emission peak started to decrease and a new low energy peak ascribed to an excimer formation appeared when more methanol was added (Figures 8b and 8d). The extent of this aggregation behavior is more remarkable for P5 than for P4, presumably reflecting the ready intermolecular packing due to the relatively linear 2,7-carbazolylene unit. Note that the zigzagged conformation of the conjugated carbazole polymers is suitable for aggregation suppression and, accordingly, stable blue light emission in the solid state [27,31]. Therefore, P4 composed of the 1,8-carbazolylene and 3,6-carbazolylene units possessed a feature of aggregation suppression similar to P7. However, the excimer formation could not be perfectly inhibited in the CH$_2$Cl$_2$:MeOH mixtures.

**Figure 8.** Solvatochromic behaviors of UV-vis absorption (a, P4; c, P5) and fluorescence spectra (b, P4; d, P5) in CH$_2$Cl$_2$/MeOH mixtures.

3. Experimental Section

3.1. Materials

All reagents including 2,2’-(9,9-dihexyl-9H-fluorene-2,7-diyl)bis(1,3,2-dioxaborinane) (3) and 2,2’-(2,2’-bithiene-5,5’-diyl)bis(4,4,5,5-tetramethyl-1,3,2-dioxaborolane) (4) were purchased from
Kanto, Tokyo Kasei, and Wako and used as received. 3,6-Di-tert-butyl-9-hexadecyl-1,8-diiodo-9H-carbazole (1) [25], 3,6-diethynyl-9-hexadecyl-9H-carbazole (5) [32], and 2,7-diethynyl-9-hexadecyl-9H-carbazole (6) [33] were prepared according to the literature methods.

3.2. Measurements

$^1$H NMR and $^{13}$C NMR spectra were measured on a JEOL model AL400 spectrometer at 20 °C. Chemical shifts are reported in ppm downfield from SiMe$_4$, using the solvent’s residual signal as an internal reference. Coupling constants ($J$) are given in Hertz. The resonance multiplicity is described as s (singlet), d (doublet), t (triplet), and m (multiplet). Infrared (IR) spectra were recorded on a JASCO FT/IR-4100 spectrometer. MALDI-TOF MS spectra were measured on an Applied Biosystems model Voyager-DE STR in reflector mode. UV-vis spectra were recorded on a JASCO V-550 spectrophotometer. The solution spectra were measured in a quartz cuvette of 1 cm. Fluorescence spectra were measured on a JASCO FP6500 spectrophotometer. Quantum yields were determined against quinine sulfate in 0.05 M aqueous H$_2$SO$_4$ ($\Phi_F = 0.546$) [34]. Gel permeation chromatography (GPC) was measured on a JASCO system equipped with polystyrene gel columns using THF as an eluent at a flow rate of 1.0 mL min$^{-1}$ after calibration with standard polystyrenes. Thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) measurement were carried out on a Rigaku Thermoplus TG8120 and DSC8230, respectively, under flowing nitrogen at a scanning rate of 10 °C min$^{-1}$. Electrochemistry measurements were carried out at 20 °C in a classical three-electrode cell. The working, reference, and auxiliary electrodes were a glassy carbon disk electrode (0.07 cm$^2$), Ag/Ag$^+$/CH$_3$CN/(nC$_3$H$_7$)$_2$NClO$_4$ and Pt wire, respectively.

9-Hexadecyl-3,6-bis(4,4,5,5-tetramethyl-1,3,2-dioxaborolan-2-yl)-9H-carbazole (2). To a solution of 3,6-dibromo-9-hexadecyl-9H-carbazole [25] (3.00 g, 5.46 mmol) in THF (50 mL) at −78 °C, 1.6 M $n$-butyllithium in hexane (7.8 mL) was added dropwise. After the solution was stirred for 20 min, isopropoxy-4,4,5,5-tetramethyl-1,3,2-dioxaborolan-2,3,3-triol (2.33 g, 12.5 mmol) was added. The mixture was stirred at −78 °C for 5 min, warmed to 20 °C, and stirred for 12 h. After the solution was poured into an ice-water mixture, the organic phase was extracted with CH$_2$Cl$_2$ and washed with brine. Evaporation and column chromatography (SiO$_2$, hexane/EtOAc = 10:1) afforded the desired compound 2 (1.44 g, 41%). $^1$H NMR (400 MHz, CDCl$_3$): δ 0.88 (t, $J = 6$ Hz, 3 H), 1.22 (s, 6 H), 1.25 (s, 6 H), 1.40 (s, 26 H), 1.85 (t, $J = 6$ Hz, 2 H), 4.30 (t, $J = 6$ Hz, 2 H), 7.38 (d, $J = 8$ Hz, 2 H), 7.90 (d, $J = 8$ Hz, 2 H), 8.66 (s, 2 H). $^{13}$C NMR (100 MHz, CDCl$_3$): δ 14.11, 22.68, 24.93, 27.22, 28.90, 29.35, 29.38, 29.46, 29.56, 29.57, 29.61, 29.65, 29.68, 31.91, 43.12, 83.50, 108.10, 122.80, 128.05, 131.95, 142.65. IR (KBr): 3046, 2978, 2922, 2851, 1628, 1598, 1574, 1469, 1351, 1274, 1261, 1145, 1083, 1054, 1027, 963, 903, 894, 860, 810, 749, 720, 702, 674, 638, 580, 420 cm$^{-1}$. MALDI-TOF MS [2-(4-hydroxyphenylazo)benzoic acid]: m/z: calculated for C$_{40}$H$_{63}$B$_2$NO$_4$$^+$: 643.49 g mol$^{-1}$; found: 643.71 g mol$^{-1}$ [M$^+$].

P1. To a 100 mL Schlenk tube, 1 (121 mg, 0.160 mmol), 2 (103 mg, 0.160 mmol), dehydrated THF (1.3 mL), and 2 M aq. K$_2$CO$_3$ (0.315 mL) were added. Under nitrogen atmosphere, a catalytic amount of Pd(PPh$_3$)$_4$ was added, and the mixture was refluxed for 72 h. After cooling to 20 °C, the reaction mixture was diluted with CH$_2$Cl$_2$ and the organic phase was washed with water and dried over Na$_2$SO$_4$. Evaporation of the solvents and precipitation into CH$_3$OH afforded the desired polymer. The
precipitate was filtered and dried in vacuo (40.3 mg, 28%). $^1$H NMR (400 MHz, C$_4$D$_8$): $\delta$0.90–1.16 (m, 80n H), 3.57–3.87 (m, 4n H), 7.19–7.22 (m, 4n H), 7.66–7.75 (m, 2n H), 8.13–8.61 (m, 4n H). IR (KBr): 2953, 2923, 2851, 1599, 1479, 1467, 1351, 1329, 1260, 1213, 1145, 1083, 1048, 1024, 964, 867, 809 cm$^{-1}$.

**P2.** To a 100 mL Schlenk tube, 1 (130 mg, 0.171 mmol), 3 (86.1 mg, 0.171 mmol), dehydrated THF (2.0 mL), and 2 M aq. K$_2$CO$_3$ (0.335 mL) were added. Under nitrogen atmosphere, a catalytic amount of Pd(PPh$_3$)$_4$ was added, and the mixture was refluxed for 72 h. After cooling to 20 °C, the reaction mixture was diluted with CH$_2$Cl$_2$ and the organic phase was washed with water and dried over Na$_2$SO$_4$. Evaporation of the solvents and precipitation into CH$_3$OH afforded the desired polymer. The precipitate was filtered and dried in vacuo (94.7 mg, 61%). $^1$H NMR (400 MHz, C$_4$D$_8$): $\delta$0.80–1.67 (m, 75n H), 3.64–3.65 (m, 2n H), 7.72–8.50 (m, 10n H). IR (KBr): 3054, 2954, 2852, 1608, 1567, 1540, 1481, 1465, 1434, 1412, 1362, 1311, 1275, 1245, 1183, 1123, 1096, 1049, 869, 825, 805, 750, 721, 694 cm$^{-1}$.

**P3.** To a 20 mL round-bottle flask, 1 (165 mg, 0.214 mmol), 4 (89.9 mg, 0.214 mmol), dehydrated THF (2.0 mL), and 2 M aq. K$_2$CO$_3$ (0.425 mL) were added. Under nitrogen atmosphere, a catalytic amount of Pd(PPh$_3$)$_4$ was added, and the mixture was refluxed for 72 h. After cooling to 20 °C, the reaction mixture was diluted with CH$_2$Cl$_2$ and the organic phase was washed with water and dried over Na$_2$SO$_4$. Evaporation of the solvents and precipitation into CH$_3$OH afforded the desired polymer. The precipitate was filtered and dried in vacuo (20.8 mg, 15%). $^1$H NMR (400 MHz, C$_4$D$_8$): $\delta$0.90–1.44 (m, 49n H), 4.20 (br s, 2n H), 6.64–8.41 (m, 8n H). IR (KBr): 3065, 2956, 2922, 2851, 1520, 1480, 1465, 1443, 1416, 1380, 1362, 1335, 1268, 1243, 1201, 1143, 1049, 1002, 871, 800 cm$^{-1}$.

**P4.** To a 30 mL round-bottle flask, 1 (100 mg, 0.133 mmol), 5 (58.8 mg, 0.133 mmol), iPr$_2$NH (0.88 mL), and toluene (2.6 mL) were added. After bubbling N$_2$ gas for 1 h, PdCl$_2$(PPh$_3$)$_2$ (3.7 mg, 0.053 mmol) and CuI (1.5 mg, 0.0078 mmol) were added and the reaction mixture was heated to 80 °C for 24 h. After cooling to 20 °C, the solution was diluted with CH$_2$Cl$_2$ and poured into CH$_3$OH. The precipitate was collected, and the polymer was purified by reprecipitation into CH$_3$OH and then dried in vacuo (99.5 mg, 99%). $^1$H NMR (400 MHz, C$_4$D$_8$): $\delta$ 0.83–1.48 (m, 76n H), 2.10–2.14 (m, 4n H), 3.64–3.69 (m, 4n H), 6.87–8.59 (m, 10n H). IR (KBr): 3055, 2953, 2922, 2851, 2199, 1481, 1382, 1362, 1283, 1149, 1132, 870, 804 cm$^{-1}$.

**P5.** To a 30 mL round-bottle flask, 1 (100 mg, 0.133 mmol), 6 (58.8 mg, 0.133 mmol), iPr$_2$NH (0.88 mL), and toluene (2.6 mL) were added. After bubbling N$_2$ gas for 1 h, PdCl$_2$(PPh$_3$)$_2$ (3.7 mg, 0.053 mmol) and CuI (1.3 mg, 0.0068 mmol) were added and the reaction mixture was heated to 80 °C for 24 h. After cooling to 20 °C, the solution was diluted with CH$_2$Cl$_2$ and poured into CH$_3$OH. The precipitate was collected, and the polymer was purified by reprecipitation into CH$_3$OH and then dried in vacuo (115 mg, 100%). $^1$H NMR (400 MHz, C$_4$D$_8$): $\delta$ 0.82–1.75 (m, 76n H), 2.40 (br s, 4n H), 3.91 (br s, 4n H), 7.00–8.34 (m, 10n H). IR (KBr): 3065, 2953, 2922, 2851, 2196, 1600, 1482, 1456, 1434, 1387, 1362, 1322, 1345, 1188, 871, 849, 804 cm$^{-1}$.

4. Conclusions

A variety of 1,8-carbazole-based conjugated copolymers was synthesized by the Suzuki or Sonogashira polycondensation between the 1,8-diiodocarbazole derivative and the corresponding
bifunctional comonomers. The insertion of alkyne spacers was found to increase the molecular weights of the resulting polymers and improve the thermal stability. The polymer energy levels were comprehensively investigated by optical and electrochemical measurements. The data revealed that the polymer LUMO levels are variable using different comonomer or spacer structures, while the HOMO levels are almost constant with a high quality p-type semiconducting feature. This result suggests that the obtained polymers fit the standard anode materials such as ITO with the ionization potential of ca. 5.0 eV as well as the various cathode materials such as Al with the work function of ca. 3.5 eV. It was also demonstrated that the presence of alkyne spacers expands the effective conjugation length, but it also enhances the aggregation propensity of the copolymers. Further diversification of the comonomer structures will lead to realization of the 1,8-carbazole-based conjugated polymers suitable for specific applications in organic electronic devices.

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References and Notes


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