Phosphorus and Nitrogen Dual Doped and Simultaneously Reduced Graphene Oxide with High Surface Area as Efficient Metal-Free Electrocatalyst for Oxygen Reduction

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Abstract: A P, N dual doped reduced graphene oxide (PN-rGO) catalyst with high surface area (376.20 m²·g⁻¹), relatively high P-doping level (1.02 at. %) and a trace amount of N (0.35 at. %) was successfully prepared using a one-step method by directly pyrolyzing a homogenous mixture of graphite oxide (GO) and diammonium hydrogen phosphate ((NH₄)₂HPO₄) in an argon atmosphere, during which the thermal expansion, deoxidization of GO and P, N co-doping were realized simultaneously. The catalyst exhibited enhanced catalytic performances for oxygen reduction reaction (ORR) via a dominated four-electron reduction pathway, as well as superior long-term stability, better tolerance to methanol crossover than that of commercial Pt/C catalyst in an alkaline solution.

Keywords: phosphorus and nitrogen dual doped; graphene; carbon catalyst; fuel cell; oxygen reduction reaction

1. Introduction

A crucial component of a fuel cell is the electrocatalyst for the cathodic oxygen reduction reaction (ORR) [1]. Pt-based precious metals are regarded as the most effective ORR electrocatalysts developed to date. However, they suffer from a number of drawbacks including the scarcity and
consequent high cost of Pt, as well as their poor durability and low tolerance to methanol crossover [2]. Accordingly, considerable effort has been devoted to developing nonprecious-metal [3–10] and metal-free [11–14] ORR catalysts. Among such candidates, carbon materials doped with heteroatoms have attracted a great deal of attention due to their relative cost-effectiveness, good long-term durability, and excellent tolerance to methanol crossover.

Graphene, a two-dimensional monolayer of sp²-hybridized carbon atoms packed in a honeycomb lattice, has recently become an attractive candidate, due to its superior electrical conductivity, high surface area and excellent mechanical properties. Both theoretical calculations and experimental studies reveal that incorporating foreign atoms into the graphene structure can effectively tailor the material’s electronic and chemical properties [15–17]. Recently, graphene doped with heteroatoms such as nitrogen, sulfur, boron, and iodine has yielded metal-free ORR electrocatalysts with enhanced electrochemical performance [13,18–22]. This performance boost is attributed to the heteroatoms, because their electronegativity (N: 3.04; S: 2.58; B: 2.04; I: 2.66) differs from that of carbon (2.55), they break carbon’s electroneutrality, creating charged sites and consequently favoring O₂ adsorption during the ORR process. Since phosphorus has a lower electronegativity (2.19) than carbon, it is well worth exploring the unique properties of P-doped graphene. Liu et al. prepared P-doped graphene by pyrolyzing graphene oxide with 1-butyl-3-methylimidazolium hexafluorophosphate, and achieved, in an alkaline solution, an ORR catalytic performance comparable to that of commercial Pt/C [23]. Zhang et al. synthesized P-doped graphene by thermally annealing a mixture of graphite oxide (GO) and triphenylphosphine (TPP), and the resultant catalyst showed remarkable catalytic activity toward the ORR [24]. However, while exciting results have been obtained with P-doped graphene, just a few investigations into this type of catalyst have been reported to date. Furthermore, it has been reported that the co-doping of P and N can further improve the carbon materials’ ORR catalytic activity, due to the synergistic effect [25].

Herein, we propose a one-step method for preparing a P, N dual-doped reduced graphene oxide (PN-rGO) catalyst, using diammonium hydrogen phosphate ((NH₄)₂HPO₄) as both phosphorus and nitrogen sources. In an alkaline medium, the as-prepared PN-rGO exhibited enhanced ORR electrocatalytic activity, good long-term stability, high tolerance to methanol crossover, and high selectivity for the four-electron reduction pathway.

2. Results and Discussion

Figure 1 shows typical SEM and TEM images of the PN-rGO catalyst. As can be seen in Figure 1a,b ultrathin, crumpled PN-rGO nanosheets are randomly arranged and overlapped with each other, these could easily have formed a slit-shaped porous structure, and indeed, such a structure was confirmed by Brunauer-Emmett-Teller (BET) testing (Figure 3). In Figure 1c, the PN-rGO nanosheets are transparent and wrinkled, like wavy silk veils. The high-resolution TEM image (Figure 1d) shows well-defined graphitic lattice fringes, indicating the good crystallization of the PN-rGO nanosheets. Actually, the morphology of our product is quite consistent with those reported previously [18,19].

Figure 2a shows the X-ray diffraction (XRD) patterns of GO, rGO and PN-rGO. GO exhibited a peak at 2θ = 11° with an interlayer distance of 0.8 nm, which is larger than the interlayer distance of graphite (0.34 nm), revealing that many different oxygen-containing groups were intercalated within
the interlayer space. The peak at 11° completely disappeared after annealing, replaced by a broad peak at 2θ = 22° for rGO and PN-rGO, with a d-spacing of 0.4 nm, implying that the successful reduction of GO to reduced graphene oxide.

**Figure 1.** Scanning electron microscopy (SEM) images (a, b) and transmission electron microscopy (TEM) images (c, d) of P, N dual doped reduced graphene oxide (PN-rGO).

**Figure 2.** X-ray diffraction (XRD) patterns of GO, rGO and PN-rGO (a), Raman spectra of rGO and PN-rGO (b).

Further structural information about PN-rGO was obtained from Raman spectroscopy. As shown in Figure 2b, similar to all sp²-carbons, two distinct peaks appeared near 1350 cm⁻¹ and 1580 cm⁻¹, corresponding to the D band and G band, respectively. The D band is resulting from the disordered carbon atoms, whereas the G band from sp²-hybridized graphitic carbon atoms. The intensity ratio of
$I_D/I_G$ generally provides a gauge for the level of disorder. Evidently, the $I_D/I_G$ value of PN-rGO (1.15) was relatively higher than that of rGO (0.94) due to the incorporated phosphorus atoms.

The N$_2$ adsorption-desorption isotherms and the corresponding pore size distribution curves of PN-rGO and rGO are shown in Figure 3. According to the International Union of Pure and Applied Chemistry classification, the N$_2$ adsorption-desorption isotherms of the two samples were type IV, with hysteresis loops type H3. A type IV adsorption-desorption isotherm indicates the presence of mesopores, while a type H3 hysteresis loop is correlated with slit-shaped pores, possibly between parallel layers. This result is consistent with the SEM observations. Surface area and pore volume for PN-rGO were 376.2 m$^2$·g$^{-1}$ and 1.50 cm$^3$·g$^{-1}$, and for rGO 260.2 m$^2$·g$^{-1}$ and 1.17 cm$^3$·g$^{-1}$, respectively. The greatly increased BET surface area and pore volume of PN-rGO may have been due to the activation effect of (NH$_4$)$_2$HPO$_4$ on carbon [26,27]. The high surface area and large pore volume of PN-rGO could have (i) exposed more active sites and (ii) favored the mass transport of reactants and products.

Figure 3. Nitrogen adsorption-desorption isotherms (a) and the corresponding pore size distribution curve of PN-rGO and rGO (b).

To further investigate the elemental composition of PN-rGO, we carried out XPS measurement. As shown in Figure 4a, the XPS survey spectrum of PN-rGO presented a dominant C1s peak (~284.5 eV), a O1s peak (~532.0 eV), a P2p peak (~132.8 eV), and a N1s peak (~400.0 eV), confirming successful P and N co-doping [28]; the corresponding atomic percentages were 92.47, 6.02, 1.16, and 0.35 at. %, respectively.

High-resolution spectra were then obtained to gain more insight into the phosphorus and nitrogen doping. As shown in Figure 4b, the high-resolution P2p spectrum can be deconvoluted into two main component peaks located at 131.7 and 133.1 eV, corresponding to P–C and P–O bonding, respectively [23]. In addition, the peak area ratio of P–C to P–O is close to 2:3. Doping phosphorus atoms into the carbon lattice (forming a P–C covalent bond) can induce negatively delocalized C atoms adjacent to P atoms; meanwhile, in P–O bonding (where an oxygen bridge is formed between C and P), the oxygen atoms can enhance electron poverty in the carbon atoms. These two kinds of structure have been reported to be advantageous for the ORR [29]. It should be pointed out that, there
is always the debate if P can access to the honeycomb crystal lattice of graphene like what N or B atom does, due to the big difference of carbon and phosphorous in radius [30].

The deconvolution results of the high-resolution N1s spectrum were shown in Figure 4c. It’s shown the prepared PN-rGO catalyst had four types of N species, corresponding to oxidized N (~403.3 eV), graphitic N (~401.4 eV), pyrrolic N (~399.6 eV), and pyridinic N (~398.3 eV) [31,32], with compositions of 31.0, 32.3, 20.8, and 15.9 at. %, respectively [33]. The total amount of active N species (graphitic N, pyrrolic N, and pyridinic N) reached 69.0 at. % [34].

![Figure 4. X-ray photoelectron spectroscopy (XPS) survey (a) and high resolution P2p (b), and N1s (c) spectra of PN-rGO.](image)

To explore the electrocatalytic activity of PN-rGO for the ORR, cyclic voltammetry (CV) experiments were carried out in an O2-saturated 0.1 M KOH solution. The CV curves of a bare GCE and rGO were also measured for comparison. As shown in Figure 5a, for all the electrodes, the CV curves displayed distinct oxygen reduction cathodic peaks. The ORR peak potential positively shifted from −0.39 V for the GCE to −0.26 V for rGO and −0.21 V for PN-rGO. In addition, the PN-rGO had the highest peak current density, at −0.96 mA cm\(^{-2}\), which was about four times higher than that of the GCE. The most positive ORR peak potential and the highest peak current density of PN-rGO, suggest that phosphorus and trace nitrogen co-doping can greatly enhance the ORR catalytic activity of graphene. The CV area of the PN-rGO was also much greater than that of the rGO, indicating the former had a much greater electroactive area, as CV area is closely related to a sample’s
capacitance, which is proportional to its specific surface area. This result is in good agreement with the BET results.

To gain further insight into the role of P, N co-doping in the ORR, the linear sweep voltammetry (LSV) curve of PN-rGO was recorded in an O$_2$-saturated 0.1 M KOH solution; for comparison, analogous LSV curves were also obtained for GCE, rGO, and commercial 20 wt. % Pt/C. As can be seen in Figure 5b, PN-rGO had a much more positive ORR onset potential and a much higher limiting current density than GCE or rGO, indicating that doping graphene with phosphorus and trace nitrogen can facilitate the ORR. The LSV results are consistent with the CV results.

**Figure 5.** Cyclic voltammetry (CV) curves (a) and linear sweep voltammetry (LSV) curves at 1600 rpm (b) for different samples, LSV curves at different rotation rates (c) and the corresponding K-L plots (d) of PN-rGO.

To gain more information on the ORR kinetics of the PN-rGO catalyst, we recorded LSV curves in an O$_2$-saturated 0.1 M KOH solution at various rotation rates, from 1600 to 3600 rpm (Figure 5c). The diffusion current density increased rapidly as the rotation rate increased. In addition, the K-L plots at different electrode potentials displayed good linearity. We used the K-L equation to calculate the electron transfer number (n) of PN-rGO in the potential range of $-0.40$ to $-0.60$ V and obtained an average n value of 3.66, indicating that the ORR proceeded via a dominated four-electron pathway.

For practical application in fuel cells, the fuel crossover effect should be considered because fuel molecules (e.g., methanol) may pass from anode to cathode through the membrane and poison the cathode catalyst. Thus, we recorded the chronoamperometric responses of PN-rGO and Pt/C upon
the addition of 3 M methanol (Figure 6a). After the methanol was introduced into an O\textsubscript{2}-saturated 0.1 M KOH solution at about 200 s, no noticeable change was observed in the ORR current for PN-rGO; in contrast, Pt/C showed a significant drop in ORR current. These results indicated that PN-rGO possessed a high immunity to methanol crossover.

As durability is also of great importance in practical applications of fuel-cell technology, the chronoamperometric durabilities of PN-rGO and Pt/C were measured at −0.3 V for 20,000 s in an O\textsubscript{2}-saturated 0.1 M KOH solution. As can be seen in Figure 6b, slight performance attenuation with high current retention (96%) was achieved with our PN-rGO catalyst. However, commercial Pt/C suffered a current loss of 12% under the same conditions, indicating that the PN-rGO electrocatalyst was much more stable in an alkaline medium.

![Figure 6.](image)

**Figure 6.** Chronoamperometric responses of PN-rGO and Pt/C at −0.3 V upon the addition of methanol (a), durability testing curves of PN-rGO and Pt/C for 20,000 s at 1600 rpm (b).

### 3. Experimental Section

#### 3.1. Catalysts Preparation

Graphite oxide (GO) was prepared from 10,000 mesh graphite powder using a modified Hummers’ Method. PN-rGO was synthesized by the thermal annealing of GO and (NH\textsubscript{4})\textsubscript{2}HPO\textsubscript{4}. In a typical procedure, 50 mg of GO was mixed with 15 mg of (NH\textsubscript{4})\textsubscript{2}HPO\textsubscript{4} in 50 mL of deionized water, at room temperature, under stirring in an open beaker. After the water was completely removed using a rotary evaporator at 50 °C, the resulting mixture was transferred into a quartz boat in the center of a tube furnace and annealed at 900 °C for 1 h, with high-purity argon as the protective atmosphere. For comparison, reduced graphene oxide without P, N doping (rGO) was also prepared using the same procedure but in the absence of (NH\textsubscript{4})\textsubscript{2}HPO\textsubscript{4}.

#### 3.2. Physical Characterization

Scanning electron microscopy (SEM) was performed on a Nova Nano 430 field emission scanning electron microscope (FEI, Hillsboro, OR, USA). Transmission electron microscopy (TEM) images were recorded on JEM-2100HR transmission electron microscope (JEOL, Tokyo, Japan). X-ray
diffraction (XRD) patterns were conducted on a TD-3500 powder diffractometer (Tongda, Liaoning, China). Raman spectroscopy measurements were carried out on a Lab RAM Aramis Raman spectrometer (HORIBA Jobin Yvon, Edison, NJ, USA) with a laser wave length of 632.8 nm. Surface area and pore characteristics were determined by recording nitrogen adsorption-desorption isotherms using a Tristar II 3020 gas adsorption analyzer (Micromeritics, Norcross, GA, USA). X-ray photoelectron spectroscopy (XPS) was performed with an ESCALAB 250 X-ray photoelectron spectrometer (Thermo-VG Scientific, Waltham, MA, USA).

3.3. Electrochemical Measurements

Electrochemical measurements were carried out on an electrochemical workstation (Ivium, Eindhoven, The Netherlands) with a standard three-electrode system at room temperature. A glassy carbon rotating disk electrode (GC-RDE) (5 mm diameter, 0.196 cm² geometric area) was used as the working electrode, while a Pt wire and an Ag/AgCl (3 M NaCl) electrode were the counter and reference electrodes, respectively. The electrolyte was 0.1 M aqueous KOH solution. For each sample, a catalyst ink was prepared by dispersing 5 mg of the corresponding catalyst in 1 mL Nafion ethanol solution (0.25·wt. %). Then 20 μL of the dispersed catalyst ink was pipetted onto the GC-RDE and dried under an infrared lamp. The mass loading of the catalyst was 0.5 mg·cm⁻². Before testing, the electrolyte solution was purged with high-purity nitrogen or oxygen gas for at least 30 min. Unless otherwise specified, the scanning rate was 10 mV·s⁻¹. The electron transfer number (n) per oxygen molecule involved was calculated on the basis of the Koutecky-Levich (K-L) equation:

\[ J^{-1} = J_L^{-1} + J_K^{-1} = B^{-1} \omega^{-1/2} + J_L^{-1} \]

\[ B = 0.62nFC_0D_0^{2/3} \gamma^{-1/6} \]

\[ J_K = nFKC_0 \]

where \( J \) is the measured current density; \( J_K \) and \( J_L \) are the kinetic current density and the diffusion limiting current density, respectively; \( \omega \) is the angular velocity of the disk (\( \omega = 2 \pi N \), where \( N \) is the linear rotation rate); \( n \) is the number of electrons transferred for the ORR; \( F \) is the Faraday constant (\( F = 96485 \text{C·mol}^{-1} \)); \( C_0 \) is the bulk concentration of \( \text{O}_2 \) (\( 1.2 \times 10^{-3} \text{mol·L}^{-1} \)); \( D_0 \) is the diffusion coefficient of \( \text{O}_2 \) in 0.1 M KOH (\( 1.9 \times 10^{-5} \text{cm}^2\cdot\text{s}^{-1} \)); \( \gamma \) is the kinetic viscosity of the electrolyte (0.01 cm²·s⁻¹); and \( \kappa \) is the electron transfer rate constant.

4. Conclusions

A metal-free phosphorus and nitrogen dual-doped reduced graphene oxide (PN-rGO) catalyst was successfully synthesized using a one-step thermal annealing method by directly pyrolyzing a homogenous mixture of graphite oxide (GO) and diammonium hydrogen phosphate ((NH₄)₂HPO₄). The specific surface area of PN-rGO, 376.2 m²·g⁻¹, was much higher than that of rGO (260.2 m²·g⁻¹). The catalyst exhibited enhanced ORR activity via a dominant four-electron reduction pathway and showed outstanding selectivity and stability in an alkaline solution. Certainly, the details of the ORR mechanism and active sites of this new catalyst require further investigation.
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Author Contributions

X. Q. performed the experiment and analyzed the data; X. Q. and S. L. wrote the paper; C. Y. and R. C analyzed the data.

Conflicts of Interest

The authors declare no conflict of interest.

References


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