

Article

Numerical Simulation and Optimization of Directional Solidification Process of Single Crystal Superalloy Casting

Hang Zhang, Qingyan Xu * and Baicheng Liu

Key Laboratory for Advanced Materials Processing Technology, Ministry of Education, School of Materials Science and Engineering, Tsinghua University, Beijing 100084, China;

E-Mails: zhanghangmu@hotmail.com (H.Z.); liubc@tsinghua.edu.cn (B.L.)

* Author to whom correspondence should be addressed; E-Mail: scjxqy@tsinghua.edu.cn; Tel.: +86-10-6279-5482; Fax: +86-10-6277-3637.

Received: 17 December 2013; in revised form: 19 February 2014 / Accepted: 21 February 2014 /

Published: 28 February 2014

Abstract: The rapid development of numerical modeling techniques has led to more accurate results in modeling metal solidification processes. In this study, the cellular automaton-finite difference (CA-FD) method was used to simulate the directional solidification (DS) process of single crystal (SX) superalloy blade samples. Experiments were carried out to validate the simulation results. Meanwhile, an intelligent model based on fuzzy control theory was built to optimize the complicate DS process. Several key parameters, such as mushy zone width and temperature difference at the cast-mold interface, were recognized as the input variables. The input variables were functioned with the multivariable fuzzy rule to get the output adjustment of withdrawal rate (v) (a key technological parameter). The multivariable fuzzy rule was built, based on the structure feature of casting, such as the relationship between section area, and the delay time of the temperature change response by changing v , and the professional experience of the operator as well. Then, the fuzzy controlling model coupled with CA-FD method could be used to optimize v in real-time during the manufacturing process. The optimized process was proven to be more flexible and adaptive for a steady and stray-grain free DS process.

Keywords: numerical simulation; directional solidification; single crystal superalloy; fuzzy controlling strategy

1. Introduction

With the rapid development of computer and information technology, computer aided manufacturing technologies (computer-aided design (CAD), computer aided engineering (CAE), computer-aided manufacturing (CAM), *etc.*) are widely used in industrial production [1–10]. The numerical simulation method used in the directional solidification (DS) process of superalloys is receiving more attention in aviation and energy industries [11–18].

An important application of the numerical simulation is to predict casting defects, such as shrinkage cavity, hot cracking, and single crystal integrity during the directional solidification of the turbine blades [19–24]. The integrity of single crystal is a major index for the production of single crystal (SX) blades. Stray grain is a usual defect, which is a focal point of simulation studies [23–28].

The DS process, with many controlling parameters, is a complicated process. There are interactive effects among these parameters, which leads to a very narrow process window for the manufacturing of SX blades. Withdrawal rate is one of the important parameters influencing the DS process. A great deal of research has been done to study the relationship between withdrawal rate, microstructure, and properties [25,29–33]. A constant withdrawal rate is often used in industries because it is easy and convenient to control, however, it has less flexibility and leads to a low yield rate of SX blades. The experiment-based variable withdrawal rate for DS process develops quickly and is adopted more and more [34,35]. However, the main problem is that it needs many trials, increases cost, and enlarges time circles.

The numerical simulation used in DS process provides an effective way to lower the experimental circle and cost. In addition, some simple variable withdrawal rate processes were already proposed by simulation technology for an efficient and defect-free DS process [29,34,35]. However, these studies are still limited to modeling the experiment with a few times of rate changing. The process improvement lacked guidelines, which still meant a method of trial and error, and did not make full use of numerical simulation techniques. There are some studies [36,37] on the optimizing of DS processes, based on modeling and simulation, however, new models and algorithms need to be developed to deal with the system's variables, to instantly adjust the solidification parameters, and, finally, to improve the DS process.

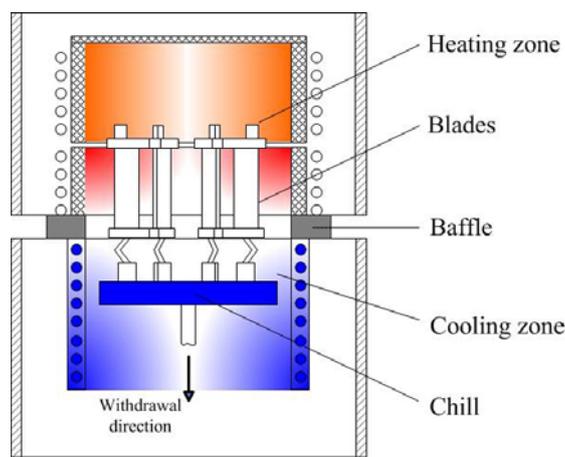
The directional solidification is a complicate nonlinear system and hard to describe using a precise mathematical model. Fuzzy controlling method, based on fuzzy set theory [38], can deal with the fuzzy relation of semantic variables easily, according to certain fuzzy rules. It is an expert in nonlinear, close coupling, and uncertain systems [39–43]. In this work, the fuzzy controlling model was built to optimize the directional solidification process. The interface temperature gradient and average mushy zone width were studied in detail. Through optimizing the withdrawal rate instantly during the calculation of the directional solidification process, the fuzzy controlling model aims to get a higher temperature gradient and improve the stability of the directional solidification process.

2. Physical and Mathematical Models

2.1. Directional Solidification Process

There are different pieces of equipment used in DS processes, and the Bridgman furnace is one of the most widely used. This furnace can be simplified and divided into five parts for modeling and simulation: heating zone, Baffle, Cooling zone, Chill, and Withdrawal unit, as shown in Figure 1. If the Cooling zone is equipped with water-cooled copper rings, the Bridgman furnace is used for the high rapid solidification (HRS) technology, which is a main DS method to produce superalloy blade castings.

Figure 1. Schematics of a Bridgman furnace.



In the HRS DS process, a group of mold shells are fixed on the chill. The liquid metal is poured into the mold and kept for minutes to make the temperature high enough. Then, the withdrawal unit starts, at certain speeds, of which the value is constant or variable. The baffle isolates the heating zone and cooling zone, then a unidirectional temperature gradient forms. When the liquid metal is drawn to pass the baffle, or entirely into the cooling zone, the mushy zone begins to freeze.

2.2. The Scheme of Optimizing DS Processes by Simulation Technology

Simulation technology was used to simulate DS processes for years. In this study, the fuzzy controlling model was built to optimize the withdrawal rate by simulation technology, as shown in Figures 2 and 3.

Firstly, the casting model was input and the withdrawal rate was pre-adjusted, based on the shape of the input model. Then, the DS process was simulated by the CA-FD method. The temperature field and microstructure growth were calculated step by step. Some key parameters, such as temperature gradient and mushy zone width, were analyzed by the fuzzy controlling model. The withdrawal rate was adjusted instantly. Then the optimized withdrawal rate was changed in the new simulation. During the simulating process, if there were stray grains appeared on the casting, the withdrawal rate would be post-adjusted and the simulating process would feed back to calculate again with a new withdrawal rate. Finally, when there is no defects predicted in the calculation, a withdrawal rate curve will be given out, which would be more effective and stable for the real DS process.

Figure 2. The framework of DS processes, optimized by simulation.

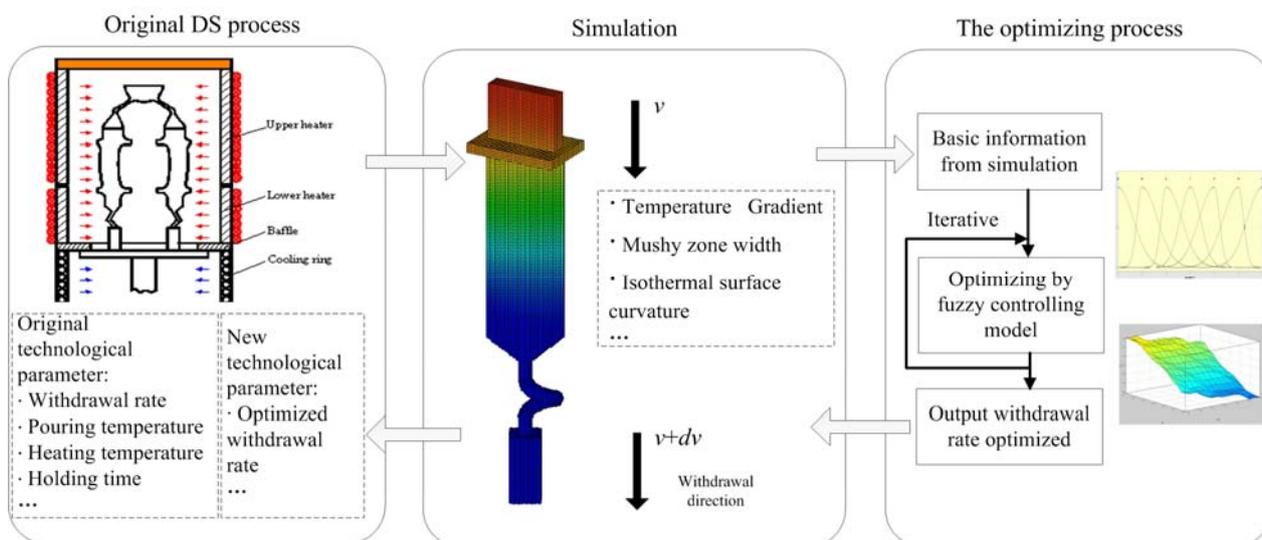
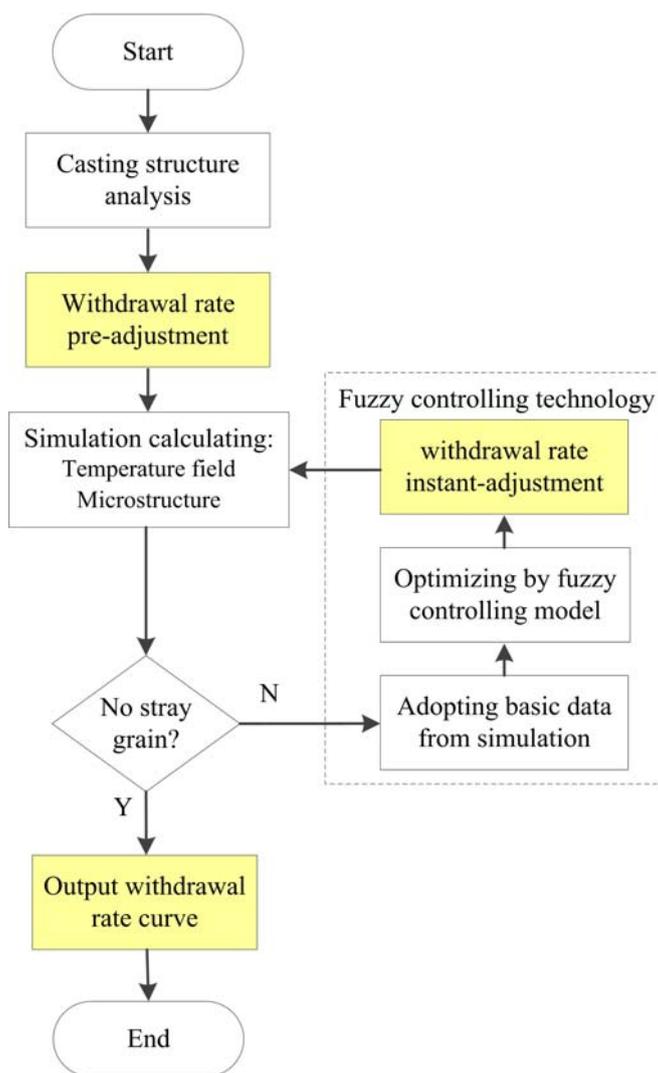


Figure 3. The flow chart of withdrawal rate optimizing by simulation.



2.3. The Fuzzy Control Model for Optimizing DS Processes

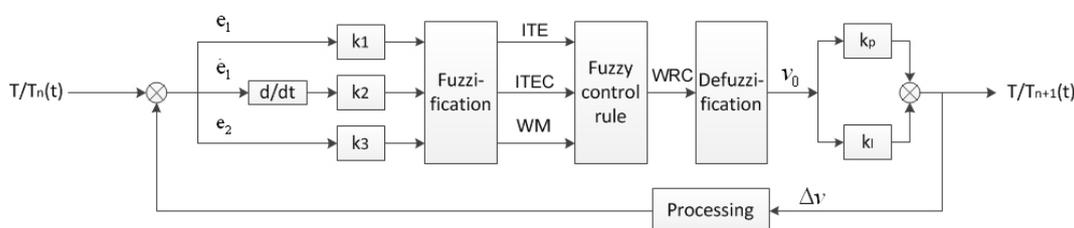
A fuzzy control model was built to analyze the real time data derived from the simulation process, and some key variables were tracked to adjust the withdrawal rate.

2.3.1. The Fuzzy Controller

The mushy zone of DS process was studied by adopting a single output and three inputs model, as shown in Figure 4.

The three input variables are follows: ITE (the casting-mold interface temperature error) is the temperature difference between most inner cell’s temperature in shell and the most outer cell’s temperature in cast; ITEC is the change of ITE; WM is the average width of mushy zone. The output variable is SC (withdrawal speed change).

Figure 4. The fuzzy controlling system.



2.3.2. The Domain of Discourse of Fuzzy Variables and Membership Functions

ITE was quantization into 13 grades, which are $\{-6, -5, -4, -3, -2, -1, +0, +1, +2, +3, +4, +5, +6\}$. The fuzzy subsets are $\{PB, PM, PS, O, NS, NM, NB\}$. e_1 , \dot{e}_1 and e_2 are the accurate values of ITE, ITEC, and WM, respectively, and the corresponding quantification factors are k_{e_1} , $k_{\dot{e}_1}$, and k_{e_2} . The membership function of ITE is shown as Equation (1). The other fuzzy variables, such as ITEC, WM, and SC, are similarly treated with ITE. In the equation, a , b , and c are the parameters of the membership function.

$$f_{ITE}(x) = \begin{cases} \exp(-\frac{(x-a)^2}{2b^2}), & x \leq a \\ \exp(-\frac{(x-a)^2}{2c^2}), & x \geq a \end{cases} \quad (1)$$

2.3.3. Fuzzy Rules for Withdrawal Rate Adjustment

Fuzzy rules could be described as follows: if WM is WM_i and ITE is ITE_j , and ITEC is $ITEC_k$, SC is SC_{ijk} .

The fuzzy relation can be written as Equations (2) and (3):

$$R_{ijk} = (WM_i \times SC_{ijk}) \wedge (ITE_j \times SC_{ijk}) \wedge (ITEC_k \times SC_{ijk}) = (WM_i \times ITE_j \times ITEC_k) \times SC_{ijk} \quad (2)$$

$$R = R_{111} + R_{112} + \dots + R_{ijk} + \dots + R_{opq} = \bigcup_{i,j,k} R_{ijk} \quad i = 1, 2, \dots, o \quad j = 1, 2, \dots, p \quad k = 1, 2, \dots, q \quad (3)$$

where WM_i , ITE_j , $ITEC_k$, and SC_{ijk} are the semantic input variables of WM, ITE, ITEC, and WRC, respectively.

During the simulation calculation of DS process, e_1 , \dot{e}_1 , and e_2 were received, as well as the semantic input variables, such as WM_i , ITE_j , and $ITEC_k$. The semantic output variable of fuzzy controller is calculated based on Equation (4):

$$\begin{aligned}
 SC^\delta &= (WM_r \times ITE_s \times ITEC_t) \circ R = (WM_r \times ITE_s \times ITEC_t) \circ \bigcup_{i,j,k} R_{ijk} \\
 &= \bigcup_{i,j,k} (WM_r \times ITE_s \times ITEC_t) \circ (WM_i \times ITE_j \times ITEC_k) \times SC_{ijk} \\
 &= \bigcup_{i,j,k} \sup[(WM_r \times ITE_s \times ITEC_t) \wedge (WM_i \times ITE_j \times ITEC_k)] \wedge SC_{ijk}
 \end{aligned}
 \tag{4}$$

$v(t)$ can be calculated based on the defuzzification of SC^δ , shown as Equation (5):

$$v(t) = \Delta v + v(t-1) = k_e \cdot \text{sgn}(SC^\delta) \cdot \text{Int}(SC^\delta + 0.5) + v(t-1)
 \tag{5}$$

k_e is the scale factor. $\text{sgn}(x)$ is sign function. $\text{Int}(x)$ is rounding function.

3. The Optimizing Process by Simulation

3.1. The Basic Simulation Condition

The SX sample blade designed has the main features of a real SX blade: the whole length is over 200 mm; the platform has an abrupt change of section; the body of the blade rotates at a certain angle respective to the tenon. The experimental material was Ni-based superalloy DD6 [44,45]. There were four schemes for simulation: group SG1 with the rate of 7.0 mm/min, group SG2 with the rate of 4.5 mm/min, group SG3 with the rate of 1.0 mm/min and group SG4 with the rate optimized by fuzzy controlling model. Two groups were selected for experimental study: Group EG1 with the rate of 7.0 mm/min and Group EG2 with the rate of 4.5 mm/min. The parameters used in these simulating groups are shown as Table 1.

Table 1. The parameters used in the calculation (Superalloy DD6 [44,45]).

Parameters	Unit	Value
Liquidus	°C	1370
Solidus	°C	1310
Density of alloy	g/cm ³	8.78
Density of shell	g/cm ³	2.50
Temperature of cooling water	°C	25

3.2. The Optimized Variable Withdrawal Rate Process

The optimized withdrawal rate of DS process was obtained in the group SG4. Figure 5 shows the curves of withdrawal rates of groups SG1–SG4. From the simulation results, it can be seen that the solidification time of SG4 was 75.5 min, which was half that of SG3, but longer than that of SG2.

The solidification processes of the four groups of DS were calculated. Then, the temperature distribution, mushy zones and temperature gradients of different processes could be analyzed by solidification times. Figure 6 is the temperature distributions of SG4 during the DS process. These

temperature cloud charts show the unidirectional heat diffusion and temperature gradient distribution. The isothermal lines basically kept horizontal and some showed a slight slant, or concave or convex.

Figure 5. Curves of withdrawal rates of SG1–SG4.

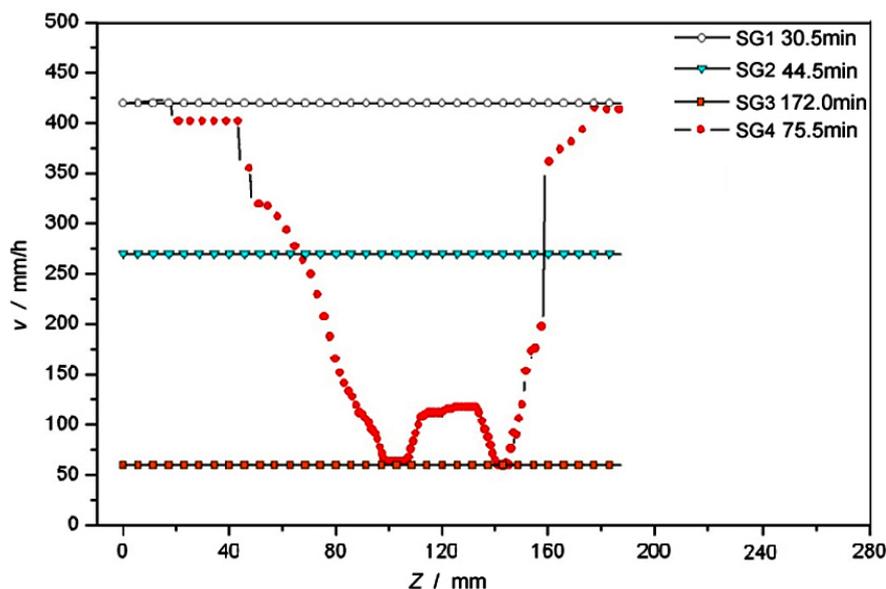
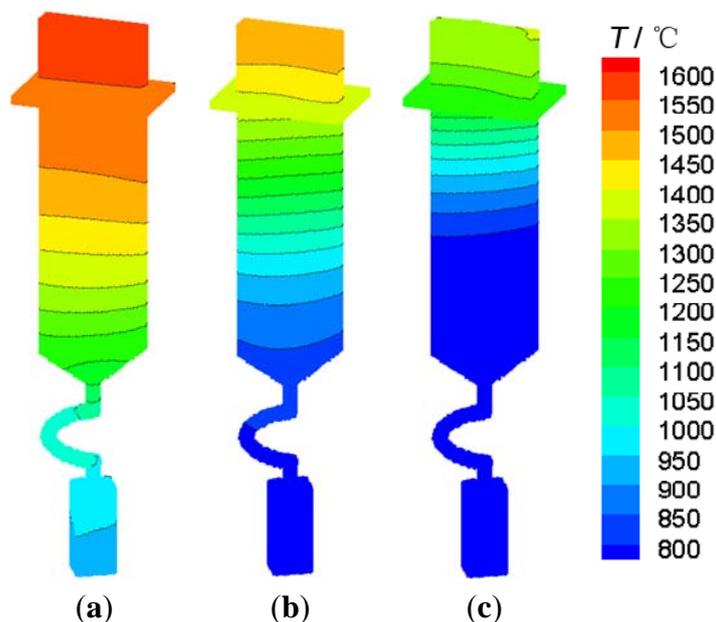


Figure 6. Temperature distribution of SG4 at different time (a) 19 min; (b) 27.5 min; (c) 36 min.

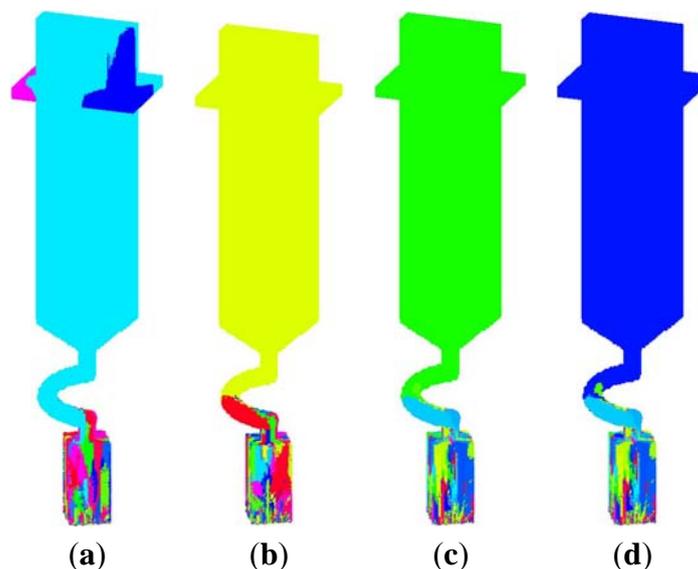


3.3. The Comparison of Constant and Optimized Withdrawal Rates Processes

Figure 7 is the microstructures of SG1–SG4 by simulation. SG1 with a constant rate of 7 mm/min appeared some stray grains in the platform, contrast to other groups. These stray grains disqualified the casting, which had different orientations from the single grain grown from the blade body. In Figure 7, although there were no stray grains at the other blades, SG2 had a very narrow process window and would tend to form stray grains at small fluctuations of other process parameters, and SG3 was of too

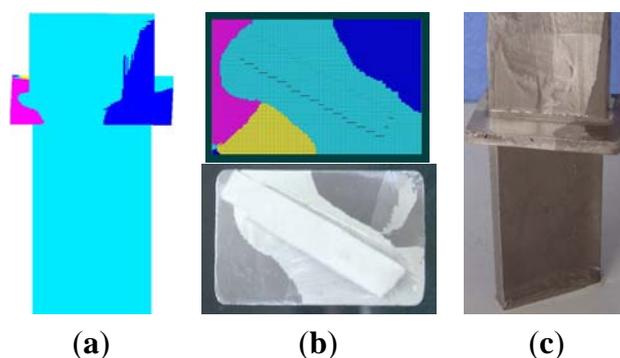
low efficiency and had too many defects to be applied in the industry production, which will be further explained below. Thus, SG4 could have a higher productivity and no stray grains.

Figure 7. Simulated Microstructures of the four groups (a) SG1; (b) SG2; (c) SG3; (d) SG4.



Stray grain is one of most severe defects of SX blade. Figure 8 is the comparison of microstructure of EG1 and SG1. Both results of EG1 and SG1 showed that the stray grains mostly started and formed in the platform, where were far from the root of the tenon and at a deep overcooling, as shown in Figure 8a,c. From Figure 8b, the locations and boundaries of stray grains predicted by simulation were similarly with that by experiment. And it's proved that the simulating models for heat transfer and microstructure evolution were accurate to predict the DS process of SX blade well.

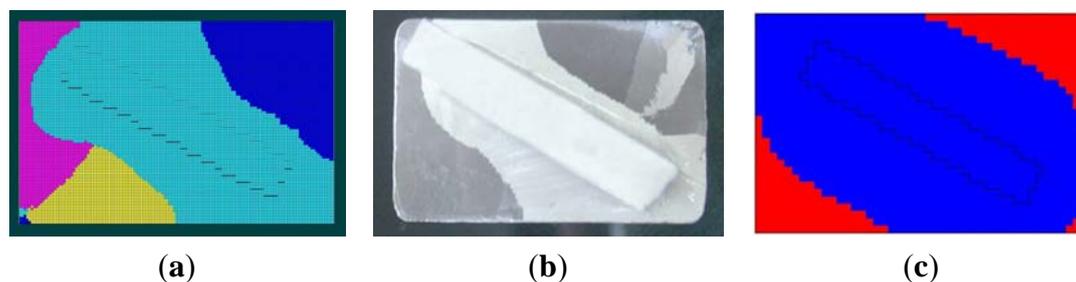
Figure 8. The comparison of microstructure of SG1 with EG1 (a) The simulation of microstructure of SX blade (partly); (b) The comparison between simulation and experiment (upward view, *i.e.*, the bottom surface of the platform); (c) The experimental result of EG1.



There are many reasons which caused stray grains, such as impurity of the melt, ceramic sunken surface, the surface of metal needles for fastening position, as well as the formation of deep undercooling zone. The deep undercooling will be the main factor leading to the stray grains in the real SX turbine blades.

In this study, SG1 and EG1 show obvious stray grains at the platform, as shown in Figure 9a,b. The undercooling zone was the main area that a stray grain nucleated and grew, which was caused by the non-uniform temperature distribution. As for SG1 and EG1, the stray grains formed at platforms. In Figure 9c, the special zones were marked in the simulated results, where the cells did not solidify but of which temperatures were lower than liquidus. These cells were named isolated undercooling zones (IUZs). IUZ provided the low temperature melt for stray grain nucleation and growth. IUZ was formed on certain condition of DS process, for example, a faster withdrawal rate.

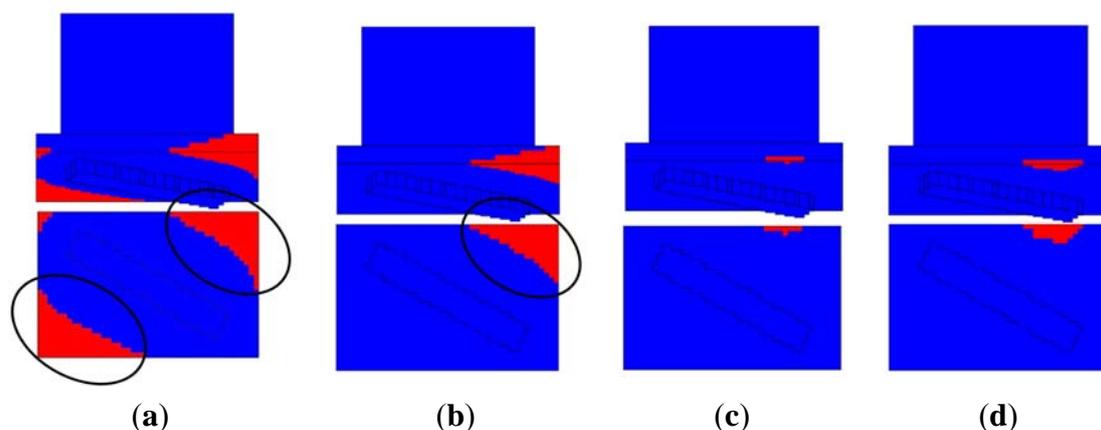
Figure 9. The comparison between the simulation and experiment to show stray grain formations (upward view, *i.e.*, the bottom surface of the platform) (a) simulation result (SG1); (b) experiment result (EG1); (c) isolated undercooling zones (IUZ) of SG1.



According to the definition above, IUZ has some features: (a) Stray grains of SX blades are mainly formed in IUZs; (b) The location and range of an IUZ are influenced and divided based on the temperature distribution; (c) IUZ often appears at some tips of the casting where heat dissipation is faster; (d) The analysis of IUZ is a convenient way to predict areas of the stray grain formation.

Figure 10 shows the IUZs formed during the calculations of SG1–SG4. The IUZs of SG1 were larger in range and more in number, as shown in Figure 10a. IUZs of SG3 and SG4 were smaller than others, which was the main reason for no stray grains formed in these groups. In addition, IUZs of SG2 were inclined to formed stray grains, but at the critical status.

Figure 10. The IUZ distributions of the four simulation groups (a) SG1; (b) SG2; (c) SG3; (d) SG4. The upper figures are side views in 3D, and the lower figures are the vertical views of the platform.



There are lots of factors influenced on the formation of IUZ, and withdrawal rate is an important one. The quick change of the cross-section area at the platform allowed more heat dissipation from the part below the platform, such as cooling zone or the water-cooled copper plate. Then at some edge or corner of the platform, the melt has a lower temperature than liquidus, but not turned to be solid, and the IUZs were formed.

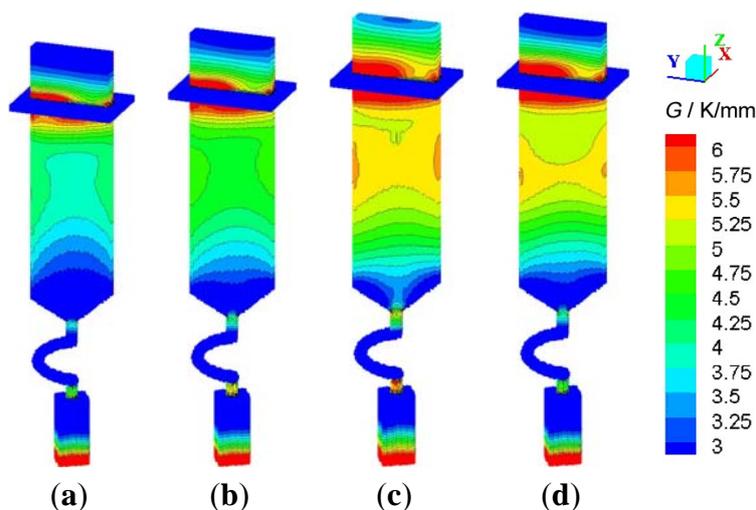
Based on the analysis above, a smaller IUZ is expected by adjusting the withdrawal rate. A lower withdrawal rate is benefit for keeping the IUZs at the heating zone of the furnace as long as possible. Then, the heat radiation would warm up these IUZs and compress their regions. Then, stray grain has no time and space to nucleate and grow. It provides a method for the fuzzy controlling model to optimize withdrawal rate dynamically, as shown in Figure 10d.

4. Advantages of the Fuzzy Optimizing Model

4.1. To Get Higher Temperature Gradient

Temperature gradient of mushy zone is a major parameter of the DS process, which influences the grain growth and quality of final castings. In the study, temperature gradients in mushy zones during solidification process were sampled (interval: 0.5 min), as shown in Figure 11. In all four groups, temperature gradients at bottom of starter block and at center of platform were higher than 6 K/mm, which resulted from the fast heat transfer of water-cooled chill plate and the sudden change of section area at the platform, respectively. Meanwhile, temperature gradients during the whole solidification process of SG3 and SG4 were higher than that of SG1 and SG2, which showed the advantages of these two processes.

Figure 11. Temperature gradients of mushy zones during DS process (a) SG1; (b) SG2; (c) SG3; (d) SG4.



In conclusion, SG4 had a short solidification time than SG3, as shown in Figure 5, but its temperature gradients were higher than others. That is to say, at a proper time span, the DS process with an optimized withdrawal rate is benefit for improving microstructure and properties, and increasing productivity and yield rate of SX blades.

4.2. To Enlarge the Process Window

DS process has many controlling parameters, which are complicate and coupled closely. Thus, the process is narrow for DS casting, which means a minor fluctuation may be amplified and lead to the formation of stray grains or other defects.

In the work, the frequencies of ΔT (the physical value of ITE) and mushy zone width were studied during the whole solidification process, and the sampling interval is 30 s. Figure 12 shows the frequencies of ΔT_s during the solidification process of the four groups. When the ΔT is zero, it means the transverse temperature gradient is eliminated, which is ideal for a DS process. In Figure 12d, most of ΔT_s were assembling around zero, which means the optimized process could keep a good unidirectional heat flux. Particularly noteworthy is that most frequencies of ΔT_s in Figure 12c were not good as those in Figure 12d. However, in Figure 12a,b, most frequencies of ΔT_s correspond to around 5 °C or more. The SG4 had a relative longer time when ΔT_s equaled to 0, which means a better DS condition than others.

Figure 13 shows the frequency distributions of WM of the four simulation groups. The frequency distributions of SG1 and SG2 were basically uniform and approximately distributed at the range of 7.5–20 mm. Most frequencies of WMs of SG4 concentrated around 10 mm and the frequency rate was above 0.25, better than that of SG3, as shown in Figure 13c. This concentration made sure the stabilities of WM and temperature gradient of mushy zone, which was benefit for a more stable DS process and allowed other solidification parameters to adjust in a lager range.

Figure 12. The frequency distributions of ΔT during the solidification time (a) SG1; (b) SG2; (c) SG3; (d) SG4.

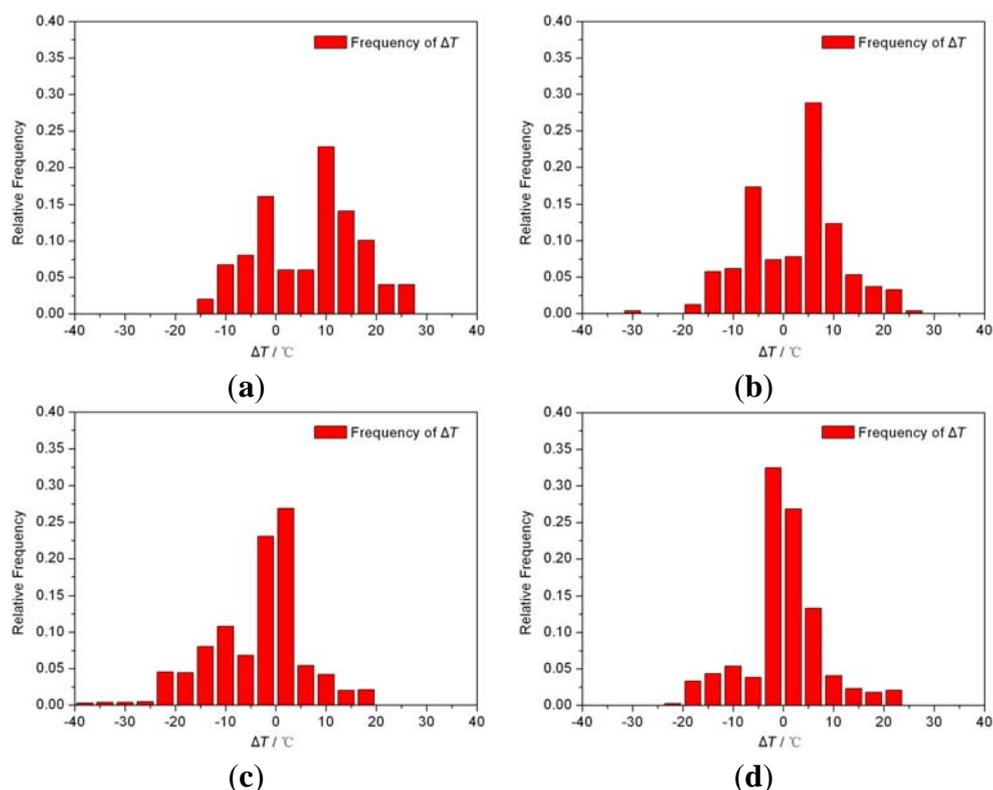
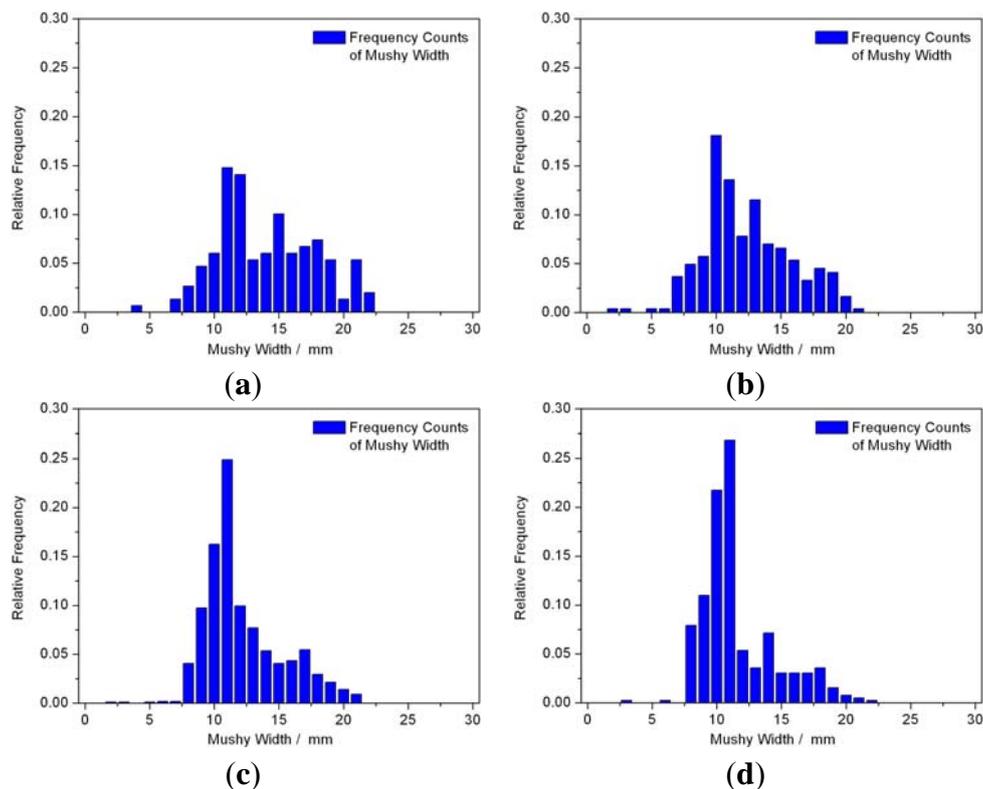


Figure 13. The frequency distributions of mushy zone width of the four simulation groups (a) SG1; (b) SG2; (c) SG3; (d) SG4.



5. Conclusions

The mathematical and physical models were built for the DS process. The SX blade was studied by numerical simulation and experimental methods. Stray grain formations were successfully predicted by simulation, and the IUZ was proposed to analyze the mechanism of stray grain formation at the platform.

A fuzzy controlling model for optimizing the DS withdrawal rate was built, and the withdrawal rate for SX blade casting was optimized. The optimized process could successfully get sound SX blade castings.

The advantages of optimized process were analyzed, based on the comparisons of temperature gradients, ITG, and WM. The proposed optimized technology in the paper that couples the intelligent controlling technology and the simulation technology is a useful way to optimize the DS process withdrawal rate and the model is useful to get a higher temperature gradient and enlarge the DS process window.

Acknowledgments

Thanks for the funds of National Basic Research Program of China (No. 2011CB706801), National Natural Science Foundation of China (Nos. 51171089 and 51374137), National Science and Technology Major Project (Nos. 2011ZX04014-052 and 2012ZX04012-011).

Conflicts of Interest

The authors declare no conflict of interest.

Author Contributions

In this research, Hang Zhang built the basic optimized model and finished the modeling and experiment. Qingyan Xu corrected the optimized model and Baicheng Liu reviewed this paper.

References

1. Lee, A. Integrating computer aids [CAD/CAM]. *Engineer* **1982**, *6572*, 35.
2. Shah, J.J.; Rogers, M.T. Functional requirements and conceptual design of the feature-based modelling system. *Comput. Aided Eng. J.* **1988**, *5*, 9–15.
3. Roy, U.; Zhang, X. Establishment of a pair of concentric circles with the minimum radial separation for assessing roundness error. *Comput. Aided Design* **1992**, *24*, 161–168.
4. De Martino, T.; Falcidieno, B.; Hassinger, S. Design and engineering process integration through a multiple view intermediate modeller in a distributed object-oriented system environment. *Comput. Aided Design* **1998**, *30*, 437–452.
5. Liu, B.C. A new generation of precision casting forming process based on physical metallurgy and mathematical physics modeling. *China Mech. Eng.* **2000**, *1*, 67–69.
6. Regli, W.C.; Cicirello, V.A. Managing digital libraries for computer-aided design. *Comput. Aided Design* **2000**, *32*, 119–132.
7. Hoffman, C.M.; Lomonosov, A.; Sitharam, M. Decomposition plans for geometric constraint systems, part I: Performance measures for CAD. *J. Symb. Comput.* **2001**, *31*, 367–408.
8. Sun, W.; Lin, F.; Hu, X. Computer-aided design and modeling of composite unit cells. *Compos. Sci. Technol.* **2001**, *61*, 289–299.
9. Kim, K.Y.; Wang, Y.; Muogboh, O.S.; Nnajia, B.O. Design formalism for collaborative assembly design. *Comput. Aided Design* **2004**, *36*, 849–871.
10. Sudarsan, R.; Fenves, S.J.; Sriram, R.D.; Wang, F. A product information modeling framework for product lifecycle management. *Comput. Aided Design* **2005**, *37*, 1399–1411.
11. Gandin, C.A.; Rappaz, M.; Tintillier, R. 3-dimensional simulation of the grain formation in investment castings. *Metall. Mater. Trans. A* **1994**, *25*, 629–635.
12. Yang, X.L.; Lee, P.D.; D'Souza, N. Stray grain formation in the seed region of single-crystal turbine blades. *JOM* **2005**, *57*, 40–44.
13. Rappaz, M.; Gandin, C.A. Probabilistic modeling of microstructure formation in solidification processes. *Acta Metall. Mater.* **1993**, *41*, 345–360.
14. Gandin, C.A.; Rappaz, M. A coupled finite-element cellular-automaton model for the prediction of dendritic grain structures in solidification processes. *Acta Metall. Mater.* **1994**, *42*, 2233–2246.
15. Pan, D.; Xu, Q.Y.; Liu, B.C. Three-dimensional microstructure simulation of Ni-based superalloy investment castings. *Sci. China Phys. Mech. Astron.* **2011**, *54*, 851–855.
16. Yu, J.; Xu, Q.Y.; Cui, K.; Liu, B.C.; Kimatsuka, A.; Kuroki, A.; Hirata, A. Numerical simulation of solidification process on single crystal Ni-based superalloy investment castings. *J. Mater. Sci. Technol.* **2007**, *23*, 47–54.
17. Zhang, H.; Xu, Q.Y.; Tang, N.; Pan, D.; Liu, B.C. Numerical simulation of microstructure evolution during directional solidification process in directional solidified (DS) turbine blades. *Sci. China Technol. Sci.* **2011**, *54*, 3191–3202.

18. Galantucci, L.M.; Tricarico, L. A computer-aided approach for the simulation of the directional-solidification process for gas turbine blades. *J. Mater. Process. Technol.* **1998**, *77*, 160–165.
19. Yang, X.L.; Dong, H.B.; Wang, W.; Lee, P.D. Microscale simulation of stray grain formation in investment cast turbine blades. *Mater. Sci. Eng. A* **2004**, *386*, 129–139.
20. Yang, X.L.; Ness, D.; Lee, P.D.; D'Souza, N. Simulation of stray grain formation during single crystal seed melt-back and initial withdrawal in the Ni-base superalloy CMSX4. *Mater. Sci. Eng. A* **2005**, *413–414*, 571–577.
21. Zambaldi, C.; Roters, F.; Raabe, D.; Glatzel, U. Modeling and experiments on the indentation deformation and recrystallization of a single-crystal nickel-base superalloy. *Mater. Sci. Eng. A* **2007**, *454–455*, 433–440.
22. He, Y.H.; Hou, X.Q.; Tao, C.H.; Han, F.K. Recrystallization and fatigue fracture of single crystal turbine blades. *Eng. Fail. Anal.* **2011**, *18*, 944–949.
23. Fang, Y.; Li, Y.; He, W.; Lu, Y.; Li, P. Numerical simulation of residual stresses fields of DD6 blade during laser shock processing. *Mater. Design* **2013**, *43*, 170–176.
24. Panwisawas, C.; Mathur, H.; Gebelin, J.; Putman, D.; Rae, C.M.F.; Reed, R.C. Prediction of recrystallization in investment cast single-crystal superalloys. *Acta Mater.* **2013**, *61*, 51–66.
25. De Bussac, A.; Gandin, C.A. Prediction of a process window for the investment casting of dendritic single crystals. *Mater. Sci. Eng. A* **1997**, *237*, 35–42.
26. Yasuda, H.; Ohnaka, I.; Kawasaki, K.; Sugiyama, A.; Ohmichi, T.; Iwane, J.; Umetani, K. Direct observation of stray crystal formation in unidirectional solidification of Sn–Bi alloy by X-ray imaging. *J. Cryst. Growth* **2004**, *262*, 645–652.
27. Zhou, Y. Formation of stray grains during directional solidification of a nickel-based superalloy. *Scr. Mater.* **2011**, *65*, 281–284.
28. Zhao, X.B.; Liu, L.; Zhang, W.G.; Qu, M.; Zhang, J.; Fu, H.Z. Analysis of competitive growth mechanism of stray grains of single crystal superalloys during directional solidification process. *Rare Met. Mater. Eng.* **2011**, *40*, 9–13.
29. Liu, S.Z.; Li, J.R.; Tang, D.Z.; Zhong, Z.G. Numerical simulation of directional solidification process of single crystal superalloys. *J. Mater. Eng.* **1999**, 40–42.
30. Liu, C.B.; Shen, J.; Zhang, J.; Lou, L.H. Effect of withdrawal rates on microstructure and creep strength of a single crystal superalloy processed by LMC. *J. Mater. Sci. Technol.* **2010**, *26*, 306–310.
31. Liu, L.; Huang, T.; Qu, M.; Liu, G.; Zhang, J.; Fu, H. High thermal gradient directional solidification and its application in the processing of nickel-based superalloys. *J. Mater. Process. Technol.* **2010**, *210*, 159–165.
32. Liu, G.; Liu, L.; Ai, C.; Ge, B.; Zhang, J.; Fu, H. Influence of withdrawal rate on the microstructure of Ni-base single-crystal superalloys containing Re and Ru. *J. Alloys Compd.* **2011**, *509*, 5866–5872.
33. Jin, T.; Li, J.G.; Zhao, N.R.; Wang, Z.; Sun, X.F.; Guan, H.R.; Hu, Z.L. Effects of withdrawal rate on solidification parameters and microstructure of a nickel-based single crystal superalloy. *J. Mater. Eng.* **2002**, *3*, 36–39.

34. Yu, J. Numerical Simulation of Directional Solidification Process and Microstructure Evolution of Superalloy Turbine Blade Castings. Master's Thesis, Tsinghua University, Beijing, China, April 2007.
35. Pan, D. Numerical Simulation of Microstructure Formation and Evolution of Directional Solidified and Single Crystal Turbine Blade Casting. Master's Thesis, Tsinghua University, Beijing, China, April 2010.
36. Monastyrskiy, V. Modeling and numerical optimization of withdrawal rate in directional solidification process. *IOP Conf. Ser. Mater. Sci. Eng.* **2012**, *33*, doi:10.1088/1757-899X/33/1/012023.
37. Hofmann, N.; Olive, S.; Laschet, G.; Hediger, F.; Wolf, J.; Sahm, P.R. Numerical optimization of process control variables for the Bridgman casting process. *Modell. Simul. Mater. Sci. Eng.* **1997**, *5*, doi:10.1088/0965-0393/5/1/002.
38. Takagi, T.; Sugeno, M.F. Fuzzy identification of systems and its applications to modeling and control. *IEEE Trans. Syst. Man Cybern.* **1985**, *SMC-15*, 116–132.
39. Lee, C.C. Fuzzy-logic in control-systems-fuzzy-logic controller. 1. *IEEE Trans. Syst. Man Cybern.* **1990**, *20*, 404–418.
40. Wang, H.O.; Tanaka, K.; Griffin, M.F. An approach to fuzzy control of nonlinear systems: Stability and design issues. *IEEE Trans. Fuzzy Syst.* **1996**, *4*, 14–23.
41. Cao, S.G.; Rees, N.W.; Feng, G. Analysis and design for a class of complex control systems. Part I: Fuzzy modelling and identification. *Automatica* **1997**, *33*, 1017–1028.
42. Hsu, Y.; Chen, G.; Tong, S.; Li, H. Integrated fuzzy modeling and adaptive control for nonlinear systems. *Inf. Sci.* **2003**, *153*, 217–236.
43. Abonyi, J.; Nagy, L.; Szeifert, F. Adaptive fuzzy inference system and its application in modelling and model based control. *Chem. Eng. Res. Design* **1999**, *77*, 281–290.
44. Li, J.R.; Zhong, Z.G.; Tang, D.Z.; Liu, S.Z.; Wei, P.; Wei, P.Y.; Wu, Z.T.; Huang, D. A low-cost second generation single crystal superalloy DD6. In Proceedings of the 10th International Symposium on Superalloys, Warrendale, PA, USA, 17 September 2000.
45. Li, J.R.; Zhao, J.Q.; Liu, S.Z.; Han, M. Effects of low angle boundaries on the mechanical properties of single crystal superalloy DD6. In Proceedings of the 11th International Symposium on Superalloys, Champion, PA, USA, 14 September 2008.