

Article Modified Porous SiO₂-Supported Cu₃(BTC)₂ Membrane with High Performance of Gas Separation

Chunjing Lu¹, Gang Wang^{1,*}, Keliang Wang¹, Daizong Guo², Mingxing Bai¹ and Ying Wang³

- Key Laboratory for EOR Technology (Ministry of Education), Northeast Petroleum University, XuefuRoad 99, Daqing 163318, China; chunjinglu@outlook.com (C.L.); klwang0608@outlook.com (K.W.); baimingxing@hotmail.com (M.B.)
- ² Mechanical Science and Engineering College, Northeast Petroleum University, XuefuRoad 99, Daqing 163318, China; daizongguo@outlook.com
- ³ State Key Laboratory of Inorganic Synthesis & Preparative Chemistry, Jilin University, QianjinRoad 2699, Changchun 130012, China; ywang0707@outlook.com
- * Correspondence: g.wang@stu.nepu.edu.cn; Tel.: +86-0459-650-4105

Received: 12 June 2018; Accepted: 11 July 2018; Published: 13 July 2018



Abstract: The structures and applications of metal-organic framework materials (MOFs) have been attracting great interest due to the wide variety of possible applications, for example, chemical sensing, separation, and catalysis. N-[3-(Trimethoxysilyl)propyl]ethylenediamine is grafted on a porous SiO₂ disk to obtain a modified porous SiO₂ disk. A large-scale, continuous, and compact Cu₃(BTC)₂ membrane is prepared based on a modified porous SiO₂ disk. The chemical structure, surface morphology, thermal stability, mechanical stability, and gas separation performance of the obtained Cu₃(BTC)₂ membrane is analyzed and characterized by means of X-ray diffraction (XRD), scanning electron microscopy (SEM), thermogravimetric analysis (TGA) and a gas separation experiment. The results show that the prepared Cu₃(BTC)₂ membrane has an intact morphology with its crystal. It is continuous, compact, and intact, and has good thermal stability and mechanical stability. The result of the gas separation experiment shows that the Cu₃(BTC)₂ membrane has a good selectivity of hydrogen and can be used to recover and purify hydrogen.

Keywords: porous SiO₂ disk; N-[3-(Trimethoxysilyl)propyl]ethylenediamine; modified; Cu₃(BTC)₂ membrane; gas separation

1. Introduction

During the past few decades, the development of membrane materials has drawn research interests in both research-oriented and industrial applications [1–3]. The membrane is a barrier, which can selectively control some materials to pass through and make other substances not. As a result, the membrane can be used to separate mixtures [4–6]. Compared with traditional separation methods, the separation-based membrane has the advantage of energy saving and efficiency. Traditionally, the development and application of membranes involve documentation of the polymer membrane because it has disadvantages of short life, low working temperature, and lower chemical stability and selectivity, but researchers need to explore some new membrane materials that have stable performance and are more conducive to separation [7–9]. Metal-organic framework materials (MOFs) have become excellent candidates for membrane fabrication because they have diverse structures, uniform pore size, permanent porosity, and high thermal and chemical stability. The membrane separation is mainly based on the molecular-size sieve, which also can separate some materials that can react with the membrane through the method of adsorption and diffusion. The MOF membrane has developed



quickly, because scientists can control the pore size easily by changing the metal ions and organic ligands while also modifying the surface of their pores through some approaches. Although more MOF membranes have been successfully synthesized, how to make the membrane have higher gas permeability as well as higher selectivity also offers more challenges [10–12].

A convenient, low-cost, and universal technique of preparing MOF membranes is essential for exploring the relationships between their structures and properties [13–15]. Metal-ligand coordination bonding interactions between the MOF membrane and substrate is the most widely used strategy to construct MOF composite membranes. To date, all reported methods of preparing MOF membranes have been limited to specific MOF membranes and specific surface-functionalized substrates, producing some limited methods due to the high cost [16–19]. Thus, far, there are few reports on the preparation of large-scale continuous compact MOF membranes with low cost.

Hydrogen is the most ideal source of energy known at present. It has many advantages, such as higher heat, wide sources, and no pollution of products. Its most attractive prospect is to replace fossil fuel as a power source for vehicles, which can significantly reduce the exhaust of air pollutants such as CO₂, CO, NOx and more: removal at the source eliminates the greenhouse effect and haze threat to mankind and realizes the potential of low carbon and environmental protection [20,21]. However, due to its high production cost, difficultly of storage, transport complications and other shortcomings, its extensive popularization and application has been hindered. Presently, the main use of hydrogen is an important chemical raw material, such as ammonia gas and methanol [22,23]. The main sources of hydrogen used in the industry are coal (dry distillation, gasification of coke oven gas, gas), and petroleum or natural gas (converted to $CO + H_2$ syngas) and other fossil fuels [24,25]. These methods obviously do not get pure hydrogen. During actual industrial production, to obtain high-purity hydrogen, the above hydrogen containing gas $(CO + H_2)$ is converted to an $H_2 + CO_2$ mixture first, and then the purity of hydrogen is higher than 99% by the method of pressure variable absorption and membrane separation [26,27]. The source of hydrogen is very wide, such as the gas in the ammonia plant, the by-product coke oven gas of the coking plant, the by-product hydrogen in the chlor alkali factory and more. Generally, they are discharged into the atmosphere as exhaust gases. This causes great waste and pollution. Science can take the appropriate method to reduce the production cost of hydrogen and promote the production cost of hydrogen [28–30]. Based on this background, the authors can adsorb and separate these mixtures $(CO_2, CH_4 \text{ and } H_2)$ to recycle hydrogen. This study uses the Cu₃(BTC)₂ composite membrane to purify hydrogen, because Cu₃(BTC)₂ is a mature MOF, which has a regular pore structure, good thermal stability, and chemistry. The structures and applications of $Cu_3(BTC)_2$ membranes have been attracting great interest because of the wide variety of possible applications, for example, chemical sensing, separation, catalysis, and electromagnetism [31-33]. Concurrently, the SiO₂ substrate has a high gas flux as a very good supporting role [34,35]. It is beneficial to separate and purify hydrogen and has a long service life. It is feasible for large-scale commercial applications.

The Cu₃(BTC)₂ membrane is synthesized to be used to separate and purify hydrogen on the SiO₂ disk modified by N-[3-(Trimethoxysilyl)propyl]ethylenediamine. The method of preparing the MOF membrane or zeolite membrane using an organosiloxane agent modified base has been previously reported [36–40]. The XRD shows that the prepared Cu₃(BTC)₂ membrane has an intact morphology with its crystal, and the SEM shows that it is continuous, compact, and intact, while having a good thermal stability and mechanical stability. The separation factors of the Cu₃(BTC)₂ membrane for H₂/CO₂, H₂/N₂, H₂/CH₄ is 10.07, 10.20 and 11.34. The results show that Cu₃(BTC)₂ membrane has a good selectivity for hydrogen and can be used for recovery of hydrogen.

2. Materials and Methods

2.1. Materials

Porous SiO₂ disks (diameter = 2.0 cm) were purchased from the Sinopharm Chemical Reagent Company (Shanghai, China). Ethanol (C₂H₅OH) and N-[3-(Trimethoxysilyl)propyl]ethylenediamine (C₈H₂₂N₂O₃Si), Cupric nitrate (Cu(NO₃)₂·3H₂O) and trimesic acid (H₃BTC) were purchased from Sigma Aldrich (St. Louis, MO, USA). All products were used as received.

2.2. Pretreatment of the SiO₂ Disk

The porous SiO₂ disk (diameter = 2.0 cm) was soaked into the mixed solution of the concentrated sulfuric acid and the hydrogen peroxide with the volume ratio of 6:4 for 5 h making the surface completely oxidized. To follow, the porous SiO₂ disk was taken out and placed into the beaker of 50 mL. 30 mL of deionized water was added into the beaker. The solution in the beaker was conducted by ultrasonic treatment for 10 minutes and poured out. The washing process was repeated three times. The porous SiO₂ disk was dried at 120 °C for 2 h.

2.3. Surface Modification of the SiO_2 Disk with N-[3-(Trimethoxysilyl)propyl]ethylenediamine

N-[3-(Trimethoxysilyl)propyl]ethylenediamine (1 mL) and alcohol (50 mL) were added to the beaker and hydrolyzed for ten minutes. The oxidized porous SiO_2 disk slice was placed horizontally at the bottom of the beaker and stirred for 24 h at 25 °C. Following the reaction, the SiO_2 disk was washed with anhydrous ethanol repeatedly to remove the N-[3-(Trimethoxysilyl)propyl]ethylenediamine which was not functionalized. Then the functionalized porous SiO_2 disk was dried in a vacuum.

2.4. Synthetic $Cu_3(BTC)_2$ Membrane with the Modified SiO₂ Disk

The MOF membranes were prepared by means of the hydrothermal method. The MOF membrane chosen was $Cu_3(BTC)_2$. Then, 0.7 g of cupric nitrate was dissolved in 19.2 mL of distilled water and solution A was obtained. Similarly, 0.336 g trimesic acid was dissolved in 19.2 mL ethanol and solution B was obtained. Solution B was then poured into solution A and stirred for 1 h. The mixed solution was poured into the Teflon-lined autoclave. The substrate of the functional porous SiO₂ disk was placed in the Teflon-lined autoclave with tweezers at 100 °C for 3 d [41]. Then, the membrane was washed several times with ethanol and dried at 25 °C. The $Cu_3(BTC)_2$ crystals adhered to the membrane surface were washed away with ethanol and the dried MOF membrane based on the functional porous SiO₂ disk was prepared by air-drying. A schematic diagram of the whole synthesis process is shown in Figure 1.



Figure 1. The process of preparing MOF membranes.

2.5. Characterization of the $Cu_3(BTC)_2$ Membranes

The thermogravimetric analysis (TGA) was performed using a DTG-60 thermal analyzer system (Shimadzu Corporation, Kyoto, Japan) at the heating rate of 10 °C min⁻¹ to 900 °C in a dried air atmosphere. The air flow rate was 30 mL min⁻¹. Samples were loaded in a platinum pan. The FTIR spectra (KBr Sigma, Aldrich, St. Louis, MO, USA) were measured using a IRAFFINITY-1 Fourier transform infrared spectrometer (Shimadzu Corporation, Nakagyo-ku, Kyoto, Japan). Samples were packed firmly to obtain transparent films. PXRD studies were performed using a D/MAX2550 diffractometer (Riguku Corporation, Akishima, Tokyo, Japan) using Cu-Ka radiation, 40 kV, 200 mA with a scanning rate of 0.3° min⁻¹ (2 θ). Scanning Electron Microscopy (SEM) analysis was performed on a JSM 6700 (JEOS Corporation, Akishima, Tokyo, Japan).

2.6. Low-Pressure N₂ Sorption Measurements

Nitrogen sorption experiments were performed at 77 K up to 1 bar using a manometric sorption analyzer Autosorb iQ MP (Quantachrome Instruments, Boynton Beach, FL, USA). Prior to sorption analysis, the sample was evacuated at 150 °C for 10 h using a turbomolecular vacuum pump. Specific surface areas were calculated from nitrogen adsorption data by multipoint Brunauer-Emmett-Teller (BET) analysis. Pore size distributions were calculated from the N₂ adsorption isotherms using a quenched solid density functional theory (nitrogen on carbon, slit pore) method which gave the least fitting error.

2.7. The Gas Separation Test

Prior to gas permeation measurements, the membranes were sealed in modules and swept by using Ar (sweep gas) and detecting gas. Meanwhile, the modules were heated to 80 °C and held for 1 h, then cooled to room temperature. Regarding both single component and mixture permeation, the permeate side and the feed side pressure were both set to 1 bar. Concerning mixture permeation, both feed gas and sweep gas rates were 80 mL min⁻¹. A soap-film flow meter was used to measure the flux of the gas and the volume ratio of the mixture gas. This assembly is shown in Figure 2.



Figure 2. Schematic of gas separation process.

3. Results

3.1. The FTIR of the Modified SiO₂ Disk

The FTIR spectra of the produced porous SiO₂ disk and the modified porous SiO₂ disk, demonstrated that the N-[3-(Trimethoxysilyl)propyl]ethylenediamine groups were grafted onto the porous SiO₂ disk surface, as presented in Figure 3. Regarding the case of the porous SiO₂ disk, the sharp band at 3450 cm⁻¹ corresponded to the presence of silanol groups (Si–OH) on the silica surface. The absorption bands at

1645 cm and 1080 cm⁻¹ were related to the bending vibration of H₂O and the isolated terminal silanol (Si–OH) groups, respectively. Following modification with N-[3-(Trimethoxysilyl)propyl]ethylenediamine, the absorption of water and the Si–OH absorption peak intensity decreased, which was due to the surface of the porous SiO₂ hydroxyl (–OH) and N-[3-(Trimethoxysilyl)propyl]ethylenediamine condensation reaction reducing the number. This changed the degree of bonding of the porous SiO₂ surface to water, that is, the bonding density with hydrogen to produce hydrogen changes. The characteristic absorption peak after N-[3-(Trimethoxysilyl)propyl]ethylenediamine appeared at 2980 cm⁻¹ due to the asymmetric stretching of the C–H bond in the aminopropyl group, indicating that N-[3-(Trimethoxysilyl)propyl]ethylenediamine had been grafted onto the porous SiO₂ surface.



Figure 3. FTIR spectra of the porous SiO₂ disk (a) and the modified porous SiO₂ disk (b).

3.2. The XRD of the $Cu_3(BTC)_2$ Membranes

Figure 4 is the XRD spectrum of the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane (black) and the $Cu_3(BTC)_2$ powder (red). Figure 4 shows the apex position of the XRD peak of the $Cu_3(BTC)_2$ membrane was the same as the highest position of the XRD spectrum of the $Cu_3(BTC)_2$ powder. The phenomenon illustrates that the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane is a pure phase composed of $Cu_3(BTC)_2$ crystal.



Figure 4. XRD spectra of Cu₃(BTC)₂ powder (red) and the prepared Cu₃(BTC)₂ membrane (black).

3.3. The TGA of the $Cu_3(BTC)_2$ Membranes

The TGA was conducted to investigate the thermal stability of the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane. The results illustrate that the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane, at 63 °C, had a weight loss of 5%, which was the adsorbed water, and the $Cu_3(BTC)_2$ membrane was stable in the air to 310 °C, showing its good thermal stability. The thermogravimetric curve is shown in Figure 5.



Figure 5. TGA curve of the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane.

3.4. The Low-Pressure N_2 Sorption Measurements and the Pore Size of the $Cu_3(BTC)_2$ Membranes

The low-pressure N₂ sorption measurements and the pore size of the Cu₃(BTC)₂ membranes were revealed by nitrogen sorption isotherm measurement at 77 K (Figure 6). The samples both were activated and degassed 10 h at 150 °C and measured from 0 to 1 bar (1 bar = P₀). The result of the Cu₃(BTC)₂ membranes exhibited a type I isotherm, which is a typical feature of microporous materials. The BET surface area was evaluated, and pore diameter was consistent with those previously reported, indicating that the m as 1550 m² g⁻¹. The pore size was calculated by appropriate fitting of the density functional theory model to the isotherm yields, which was a value of 1.0 nm for the Cu₃(BTC)₂ membrane. BET surface area and pore diameter were consistent with those previously reported, indicating that the membrane material had the same adsorption performance as the powder material [42].



Figure 6. N_2 sorption isotherms (a) and the pore size (b) of $Cu_3(BTC)_2$ membrane.

3.5. The SEM of the $Cu_3(BTC)_2$ Membranes

The characterizations of the morphology of before and after the SiO₂ disk were modified and the modified porous SiO_2 disk supporting the $Cu_3(BTC)_2$ membrane was conducted after defining the structural information and thermal stability of this membrane. The SEM of the modified porous SiO_2 disk-supported Cu₃(BTC)₂ membrane is shown in Figure 7. It can be observed that there were many small holes on the surface of the SiO_2 disk (Figure 7a). Results following modification of the SiO_2 disk (Figure 7b) were the same as before modification (Figure 7a). The modification of the SiO₂ disk with the organosiloxane agent did not affect its permeability. The results show that the modified SiO_2 disk had the same permeability for H₂, CO₂, N₂ and CH₄, and were all 1.90×10^{-6} mol m⁻² s⁻¹ Pa⁻¹. Thus, the modified base without $Cu_3(BTC)_2$ membrane was not selective for H_2 , CO_2 , N_2 and CH_4 . The obtained Cu₃(BTC)₂ membrane was composed of numerous octahedron crystals inlaid and stacked to form a uniform and dense continuous defect-free membrane structure. The scale of the microscope was 200 µm, and the membrane was continuously dense. When the scale was made gradually smaller/the magnification was gradually increased, the positive octahedral structure of the Cu₃(BTC)₂ membrane became increasingly obvious but the Cu₃(BTC)₂ membrane was composed of several positive octahedral crystals intercalated and stacked to form a uniformly dense continuous defect-free membrane.



Figure 7. SEM of the prepared samples, with porous SiO₂ disk (**a**), the modified porous SiO₂ disk (**b**) and the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane (**c**–**f**).

3.6. The Gas Separation Test of $Cu_3(BTC)_2$ Membrane

The permeability of the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane to four components of H_2 , N_2 , CO_2 and CH_4 was evaluated. The dynamic diameter of the four gas molecules of H_2 , CO_2 , N_2 and CH_4 and the specific results of the membrane permeability to these four gases are listed in Table 1.

Gas	Kinetic Diameter (nm)	Permeance (mol $m^{-2} s^{-1} Pa^{-1}$)
H_2	0.29	$1.61 imes10^{-7}$
CO ₂	0.33	$1.69 imes10^{-8}$
N_2	0.36	$1.84 imes 10^{-8}$
CH ₄	0.38	$1.98 imes 10^{-8}$

Table 1. The permeable flow of the single component gas through the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane in 298 K and 0.1 MPa.

The separation experiments of the H_2/CO_2 , H_2/CH_4 and H_2/N_2 mixed gases by $Cu_3(BTC)_2$ membrane were conducted to investigate the two-component gas separation of the modified porous SiO₂ disk-supported $Cu_3(BTC)_2$ membrane. Separation test results of various mixed gases at 298 K and 0.1 MPa are shown in Table 2. These are test results of single component permeable flow and test results of two-component permeable flow. The separation factor and the ideal separation factor calculated according to the results are also shown in Table 2.

Table 2. The permeable flow of the single component gas and mixed component gas through the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane and the separation factor in 298 K and 0.1 MPa.

Gas	Single Component Flow in Mixed Gas (10 ⁻⁶ mol m ⁻² s ⁻¹ Pa ⁻¹)	Single Component Flow (10 ⁻⁶ mol m ⁻² s ⁻¹ Pa ⁻¹)	Separation Factor	Ideal Separation Factor
H ₂ N ₂	0.148 0.0145	0.161 0.0184	10.20	8.75
H ₂ CH ₄	0.152 0.0134	0.161 0.0198	11.34	8.13
H ₂ CO ₂	0.143 0.0142	0.161 0.0169	10.07	9.53

3.7. Mechanical Stability of Cu₃(BTC)₂ Membrane

To study the mechanical properties of the synthesized $Cu_3(BTC)_2$ membrane, the gas separation performance of H_2/CO_2 (red), H_2/N_2 (black), H_2/CH_4 (blue) with the $Cu_3(BTC)_2$ membrane was tested repeatedly under 298 K and 0.1 MPa. Among them, red is powder, black is membrane, and blue is methane. The results show that the separation factor of the $Cu_3(BTC)_2$ membrane, the synthesized membrane reproducibility, was not obviously changed after 24 h of repeated tests. The mechanical properties were strong, and the utilization rate was high (Figure 8).



Figure 8. The separation factor of the $Cu_3(BTC)_2$ membrane change with the time: H_2/CO_2 (red), H_2/N_2 (black), H_2/CH_4 (blue).

4. Discussion

4.1. Preparation of the MOF Membrane

 $Cu_3(BTC)_2$ crystals (Figure 2).

Here, we report a convenient and universal method to prepare MOF membranes by hydrothermal method. First, the porous SiO₂ disk is soaked in the mixed solution of the concentrated sulfuric acid and the hydrogen peroxide with a volume ratio of 6:4 for 5 h, which gets hydroxyl on the surface of the completely oxidized porous SiO₂ disk. Then the oxidized porous SiO₂ disk was modified by N-[3-(Trimethoxysilyl)propyl]ethylenediamine. The FTIR spectra of the modified porous SiO₂ disk, which could be used to grow MOF membrane. The MOF membrane used was Cu₃(BTC)₂. The stable 3D structure of the Cu₃(BTC)₂ was formed by these secondary structural units interlaced with each other, and the 3D structure had a square aperture with a regular aperture of about 1 nm. The results of N₂ adsorption show that the specific surface area of BET was about 1550 m²/g [42]. The XRD illustrates that the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane was a pure phase composed of

4.2. The Morphology and the Stability of the $Cu_3(BTC)_2$ Membrane

The TGA was conducted to investigate the thermal stability of the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane. The results show that the modified porous SiO_2 disk-supported Cu₃(BTC)₂ membrane was at 63 °C with a weight loss of 5% (the adsorbed water), and the $Cu_3(BTC)_2$ membrane was stable in the air to 300 °C, showing its good thermal stability. The thermogravimetric curve is shown in Figure 5. The characterizations of the morphology of the SiO₂ disk and the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane was conducted after defining the structural information and thermal stability of this membrane. The SEM of the modified porous SiO₂ disk-supported $Cu_3(BTC)_2$ membrane is shown in Figure 7. It can be observed that the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane was a thin, compact, and continuous membrane, closely attached to the modified SiO₂ substrate. Viewed through the scanning electron microscope, the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane also showed that the intergrowth-crystallized octahedral architectures merged tightly. To study the mechanical properties of the synthesized $Cu_3(BTC)_2$ membrane, the gas separation performance of the $Cu_3(BTC)_2$ membrane was tested repeatedly under 298 K and 0.1 MPa. The results show that the separation factor of the Cu₃(BTC)₂ membrane, the synthesized membrane reproducibility, was not obviously changed after 24 h of repeated testing. The mechanical properties were strong, and the utilization rate was high (Figure 8).

4.3. The Gas Separation Performance of $Cu_3(BTC)_2$ Membrane

The dynamic diameter of the four gas molecules of H₂, N₂, CO₂ and CH₄ and the specific results of the membrane permeability to these four gases are listed in Table 1. It can be observed that the order of the flow of the four gases through the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane was H₂ > N₂ > CH₄ > CO₂, and the flow rate was 1.61 × 10⁻⁷ mol m⁻² s⁻¹ Pa⁻¹, 1.84 × 10⁻⁸ mol m⁻² s⁻¹ Pa⁻¹, 1.98 × 10⁻⁸ mol m⁻² s⁻¹ Pa⁻¹ and 1.69 × 10⁻⁸ mol m⁻² s⁻¹ Pa⁻¹, respectively. Based on the ideal separation coefficient formula $a = J_A/J_B$, the ideal separation factor of H₂/CO₂, H₂/N₂, H₂/CH₄ were 9.53, 8.75 and 8.13, respectively. It was higher than the corresponding Knudsen diffusion coefficient (4.69 H₂/CO₂, 3.74 H₂/N₂, and 2.83 H₂/CH₄) and was also much larger than the ideal separation factor that was reported to separate the same gas through the Cu₃(BTC)₂ membrane [42]. This preliminarily determines that the Cu₃(BTC)₂ membrane synthesized in this study are suitable for the separation of H₂ in the mixed components of H₂/CO₂, H₂/N₂, H₂/CH₄. Since the pore size of the Cu₃(BTC)₂ membrane was about 1 nm, which was bigger than the dynamic diameters of H₂, CO₂, N₂ and CH₄ molecules (Table 1). Thus, there was no effect on H₂ by the molecular sieve points from the other gases. The Cu₃(BTC)₂ structure contained many Cu elements for the adsorption of CO₂, N₂ and CH₄ gases providing the active site and the Cu₃(BTC)₂ membrane by chemical

adsorption to gas diffusion the effect of separation, as far as the authors are aware. According to reports in the literature, Cu₃(BTC)₂ for CO₂, N₂ and CH₄ adsorption enthalpy is far greater than H_2 , and the preparation of the $Cu_3(BTC)_2$ membrane nitrogen adsorption performance showed the same as the previous preparation of fission material adsorption performance. Thus, in this study, some adsorption properties of the MOF powder could be on behalf of the MOF-related adsorption properties of the membrane [43–46]. Therefore, the separation of two-component gas is investigated. The separation experiments of the H_2/CO_2 , H_2/N_2 , H_2/CH_4 mixed gases by $Cu_3(BTC)_2$ membrane was conducted to investigate the two-component gas separation of the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane. Separation test results of various mixed gases at 298 K and 0.1 MPa are listed in Table 2. There are test results of single component permeable flow, and two-component permeable flow. The separation factor and the ideal separation factor calculated according to the results are also listed in Table 2. Looking at Table 2, it can be observed that the flow of H_2 in the mixed gas is 1.61×10^{-7} mol m⁻² s⁻¹ Pa⁻¹, many times higher than the flow of other gases. This phenomenon illustrates that the modified porous SiO₂ disk-supported Cu₃(BTC)₂ membrane has the function of separating and purifying H_2 and can be used to separate and purify H_2 in the mixture of H_2/CO_2 , H_2/N_2 , H_2/CH_4 . The flow of H_2 in the two-component gas is also much higher than that of the other components, it can be obtained through the calculation of the separation factors of the modified porous SiO_2 disk-supported Cu₃(BTC)₂ membrane for the H₂/CO₂, H₂/N₂, H₂/CH₄ components are 10.07, 10.20 and 11.34 in the condition of 298 K and 0.1 MPa. These values are higher than the corresponding Knudsen values. This phenomenon illustrates that the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane can be used in gas separation and has a good performance of gas separation.

5. Conclusions

This work is the first report of the synthesis of modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membranes. The obtained functioned porous SiO_2 disk-supported $Cu_3(BTC)_2$ membranes have crystal phases that coincide with $Cu_3(BTC)_2$ crystals with a high thermal stability and intact morphology. Additionally, the performances of the modified porous SiO_2 disk-supported $Cu_3(BTC)_2$ membrane for the separation of hydrogen and other gases were evaluated and the separation factor of each group of experience was calculated in detail. It was found that the membrane has a good separation performance for hydrogen and can be used in hydrogen recovery in industry. This $Cu_3(BTC)_2$ membrane fabrication method is simple and convenient and can be readily applied to a variety of other material compositions to produce functional membranes with diverse micropore structures, thus opening a host of opportunities for the development of new functional MOFs.

Author Contributions: All authors made critical contributions to the collection and interpretation of data. M.B. and G.W. conceived and designed the experiments; D.G. and Y.W. performed the experiments; C.L., K.W. wrote the paper.

Funding: This work was financially supported by the National Science Foundation of China (No. NSFC-51574085) and the Scientific Research Staring Foundation of Northeast Petroleum University (rc201716).

Conflicts of Interest: The authors declare no conflict of interest.

References

- 1. Kraytsberg, A.; Ein-Eli, Y. Review of advanced materials for proton exchange membrane fuel cells. *Energy Fuels* **2014**, *28*, 7303–7330. [CrossRef]
- 2. Wang, S.; Li, X.; Wu, H.; Tian, Z.; Xin, Q.; He, G.; Guiver, M.D. Advances in high permeability polymer-based membrane materials for CO₂ separations. *Energy Environ. Sci.* **2016**, *9*, 1863–1890. [CrossRef]
- 3. Bernardo, P.; Drioli, E.; Golemme, G. Membrane gas separation: A review/state of the art. *Ind. Eng. Chem. Res.* **2009**, *48*, 4638–4663. [CrossRef]
- 4. Padaki, M.; Murali, R.S.; Abdullah, M.S.; Misdan, N.; Moslehyani, A.; Kassim, M.A.; Ismail, A.F. Membrane technology enhancement in oil–water separation: A review. *Desalination* **2015**, *357*, 197–207. [CrossRef]

- 5. Chen, X.Y.; Vinh-Thang, H.; Ramirez, A.A.; Rodrigue, D.; Kaliaguine, S. Membrane gas separation technologies for biogas upgrading. *RSC Adv.* **2015**, *5*, 24399–24448. [CrossRef]
- Kim, S.; Lin, X.; Ou, R.; Liu, H.; Zhang, X.; Simon, G.P.; Wang, H. Highly crosslinked, chlorine tolerant polymer network entwined graphene oxide membrane for water desalination. *J. Mater. Chem. A* 2017, 5, 1533–1540. [CrossRef]
- 7. Kitagawa, S. Metal–organic frameworks (MOFs). Chem. Soc. Rev. 2014, 43, 5415–5418.
- 8. Jiang, W.L.; Ding, L.G.; Yao, B.J.; Wang, J.C.; Chen, G.J.; Li, Y.A.; Dong, Y.B. A MOF-membrane based on the covalent bonding driven assembly of a NMOF with an organic oligomer and its application in membrane reactors. *Chem. Commun.* **2016**, *52*, 13564–13567. [CrossRef] [PubMed]
- 9. Xiang, F.; Marti, A.M.; Hopkinson, D.P. Layer-by-layer assembled polymer/MOF membrane for H₂/CO₂ separation. *J. Membr. Sci.* **2018**, 556, 146–153. [CrossRef]
- 10. Lin, Y.S. Metal organic framework membranes for separation applications. *Curr. Opin. Chem. Eng.* **2015**, *8*, 21–28. [CrossRef]
- 11. Adatoz, E.; Avci, A.K.; Keskin, S. Opportunities and challenges of MOF-based membranes in gas separations. *Sep. Purif. Technol.* **2015**, *152*, 207–237. [CrossRef]
- 12. Zhang, Y.; Feng, X.; Yuan, S.; Zhou, J.; Wang, B. Challenges and recent advances in MOF–polymer composite membranes for gas separation. *Inorg. Chem. Front.* **2016**, *3*, 896–909. [CrossRef]
- Campbell, M.G.; Sheberla, D.; Liu, S.F.; Swager, T.M.; Dincă, M. Cu₃(hexaiminotriphenylene)₂: An electrically conductive 2D metal–organic framework for chemiresistive sensing. *Angew. Chem. Int. Ed.* 2015, 54, 4349–4352. [CrossRef] [PubMed]
- Al-Janabi, N.; Hill, P.; Torrente-Murciano, L.; Garforth, A.; Gorgojo, P.; Siperstein, F.; Fan, X. Mapping the Cu-BTC metal–organic framework (HKUST-1) stability envelope in the presence of water vapour for CO₂ adsorption from flue gases. *Chem. Eng. J.* 2015, *281*, 669–677. [CrossRef]
- Kim, H.K.; Yun, W.S.; Kim, M.B.; Kim, J.Y.; Bae, Y.S.; Lee, J.; Jeong, N.C. A Chemical route to activation of open metal sites in the copper-based metal–organic framework materials HKUST-1 and Cu-MOF-2. *J. Am. Chem. Soc.* 2015, *137*, 10009–10015. [CrossRef] [PubMed]
- 16. Fu, J.; Das, S.; Xing, G.; Ben, T.; Valtchev, V.; Qiu, S. Fabrication of COF-MOF composite membranes and their highly selective separation of H₂/CO₂. *J. Am. Chem. Soc.* **2016**, *138*, 7673–7680. [CrossRef] [PubMed]
- 17. Denny, M.S.; Cohen, S.M. In situ modification of metal–organic frameworks in mixed-matrix membranes. *Angew. Chem. Int. Ed.* **2015**, *54*, 9029–9032. [CrossRef] [PubMed]
- Sorribas, S.; Kudasheva, A.; Almendro, E.; Zornoza, B.; de la Iglesia, Ó.; Téllez, C.; Coronas, J. Pervaporation and membrane reactor performance of polyimide based mixed matrix membranes containing MOF HKUST-1. *Chem. Eng. Sci.* 2015, 124, 37–44. [CrossRef]
- Seoane, B.; Coronas, J.; Gascon, I.; Benavides, M.E.; Karvan, O.; Caro, J.; Gascon, J. Metal–organic framework based mixed matrix membranes: A solution for highly efficient CO₂ capture. *Chem. Soc. Rev.* 2015, 44, 2421–2454. [CrossRef] [PubMed]
- 20. Sevilla, M.; Mokaya, R. Energy storage applications of activated carbons: Supercapacitors and hydrogen storage. *Energy Environ. Sci.* **2014**, *7*, 1250–1280. [CrossRef]
- 21. Dutta, S. A review on production, storage of hydrogen and its utilization as an energy resource. *J. Ind. Eng. Chem.* **2014**, *20*, 1148–1156. [CrossRef]
- 22. Sharma, S.; Ghoshal, S.K. Hydrogen the future transportation fuel: From production to applications. *Renew. Sustain. Energy Rev.* **2015**, *43*, 1151–1158. [CrossRef]
- 23. Tremel, A.; Wasserscheid, P.; Baldauf, M.; Hammer, T. Techno-economic analysis for the synthesis of liquid and gaseous fuels based on hydrogen production via electrolysis. *Int. J. Hydrog. Energy* **2015**, 40, 11457–11464. [CrossRef]
- 24. Singh, S.; Jain, S.; Venkateswaran, P.S.; Tiwari, A.K.; Nouni, M.R.; Pandey, J.K.; Goel, S. Hydrogen: A sustainable fuel for future of the transport sector. *Renew. Sustain. Energy Rev.* **2015**, *51*, 623–633. [CrossRef]
- Uusitalo, V.; Väisänen, S.; Inkeri, E.; Soukka, R. Potential for greenhouse gas emission reductions using surplus electricity in hydrogen, methane and methanol production via electrolysis. *Energy Convers. Manag.* 2017, 134, 125–134. [CrossRef]
- Shan, X.; Qian, Y.; Zhu, L.; Lu, X. Effects of EGR rate and hydrogen/carbon monoxide ratio on combustion and emission characteristics of biogas/diesel dual fuel combustion engine. *Fuel* 2016, 181, 1050–1057. [CrossRef]

- 27. Zhang, F.; Zhao, P.; Niu, M.; Maddy, J. The survey of key technologies in hydrogen energy storage. *Int. J. Hydrogen Energy* **2016**, *41*, 14535–14552. [CrossRef]
- Yang, Q.; Li, L.; Tan, W.; Sun, Y.; Wang, H.; Ma, J.; Zhao, X. Exceptional high selectivity of hydrogen/methane separation on a phosphonate-based MOF membrane with exclusion of methane molecules. *Chem. Commun.* 2017, 53, 9797–9800. [CrossRef] [PubMed]
- 29. Hu, Y.; Dong, X.; Nan, J.; Jin, W.; Ren, X.; Xu, N.; Lee, Y.M. Metal–organic framework membranes fabricated via reactive seeding. *Chem. Commun.* **2011**, *47*, 737–739. [CrossRef] [PubMed]
- Nenoff, T.M. Hydrogen purification: MOF membranes put to the test. *Nat. Chem.* 2015, 7, 377. [CrossRef] [PubMed]
- 31. Zhang, Y.; Wang, N.; Ji, S.; Zhang, R.; Zhao, C.; Li, J.R. Metal–organic framework/poly (vinyl alcohol) nanohybrid membrane for the pervaporation of toluene/n-heptane mixtures. *J. Membr. Sci.* 2015, 489, 144–152. [CrossRef]
- Dolgopolova, E.A.; Brandt, A.J.; Ejegbavwo, O.A.; Duke, A.S.; Maddumapatabandi, T.D.; Galhenage, R.P.; Chandrashekhar, M. Electronic Properties of Bimetallic Metal–Organic Frameworks (MOFs): Tailoring the Density of Electronic States through MOF Modularity. J. Am. Chem. Soc. 2017, 139, 5201–5209. [CrossRef] [PubMed]
- 33. Gu, Z.G.; Zhang, J. Epitaxial growth and applications of oriented metal–organic framework thin films. *Coord. Chem. Rev.* **2017**. [CrossRef]
- 34. Van Gestel, T.; Hauler, F.; Bram, M.; Meulenberg, W.A.; Buchkremer, H.P. Synthesis and characterization of hydrogen-selective sol–gel SiO₂ membranes supported on ceramic and stainless steel supports. *Sep. Purif. Technol.* **2014**, *121*, 20–29. [CrossRef]
- 35. Sun, J.; Bi, H.; Su, S.; Jia, H.; Xie, X.; Sun, L. One-step preparation of GO/SiO₂ membrane for highly efficient separation of oil-in-water emulsion. *J. Membr. Sci.* **2018**, *553*, 131–138. [CrossRef]
- Huang, A.; Dou, W.; Caro, J. Steam-stable zeolitic imidazolate framework ZIF-90 membrane with hydrogen selectivity through covalent functionalization. *J. Am. Chem. Soc.* 2010, 132, 15562–15564. [CrossRef] [PubMed]
- 37. Huang, K.; Dong, Z.; Li, Q.; Jin, W. Growth of a ZIF-8 membrane on the inner-surface of a ceramic hollow fiber via cycling precursors. *Chem. Commun.* **2013**, *49*, 10326–10328. [CrossRef] [PubMed]
- Xie, Z.; Yang, J.; Wang, J.; Bai, J.; Yin, H.; Yuan, B.; Duan, C. Deposition of chemically modified α-Al₂O₃ particles for high performance ZIF-8 membrane on a macroporous tube. *Chem. Commun.* 2012, 48, 5977–5979. [CrossRef] [PubMed]
- 39. Kida, K.; Fujita, K.; Shimada, T.; Tanaka, S.; Miyake, Y. Layer-by-layer aqueous rapid synthesis of ZIF-8 films on a reactive surface. *Dalton Trans.* **2013**, *42*, 11128–11135. [CrossRef] [PubMed]
- 40. Bux, H.; Feldhoff, A.; Cravillon, J.; Wiebcke, M.; Li, Y.S.; Caro, J. Oriented zeolitic imidazolate framework-8 membrane with sharp H₂/C₃H₈ molecular sieve separation. *Chem. Mater.* **2011**, *23*, 2262–2269. [CrossRef]
- 41. Lu, C.; Ben, T.; Xu, S.; Qiu, S. Electrochemical synthesis of a microporous conductive polymer based on a Metal-Organic Framework thin film. *Angew. Chem. Int. Edit.* **2014**, *53*, 6454–6458. [CrossRef] [PubMed]
- 42. Ben, T.; Lu, C.; Pei, C.; Xu, S.; Qiu, S. Polymer-supported and free-standing metal-organic framework membrane. *Chem. A Eur. J.* **2012**, *18*, 10250–10253. [CrossRef] [PubMed]
- 43. Wu, D.; Guo, X.; Sun, H.; Navrotsky, A. Thermodynamics of methane adsorption on copper HKUST-1 at low pressure. *J. Phys. Chem. Lett.* **2015**, *6*, 2439–2443. [CrossRef] [PubMed]
- 44. Hamon, L.; Jolimaitre, E.; Pirngruber, G.D. CO₂ and CH₄ separation by adsorption using Cu-BTC metal–organic framework. *Ind. Eng. Chem. Res.* **2010**, *49*, 7497–7503. [CrossRef]
- 45. Koh, H.S.; Rana, M.K.; Wong-Foy, A.G.; Siegel, D.J. Predicting methane storage in open-metal-site metal–organic frameworks. *J. Phys. Chem. C* 2015, *119*, 13451–13458. [CrossRef]
- Mao, Y.; Li, J.; Cao, W.; Ying, Y.; Sun, L.; Peng, X. Pressure-assisted synthesis of HKUST-1 thin film on polymer hollow fiber at room temperature toward gas separation. ACS Appl. Mater. Interfaces 2014, 6, 4473–4479. [CrossRef] [PubMed]



© 2018 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (http://creativecommons.org/licenses/by/4.0/).