



Article

NO₂ Selective Sensor Based on α-Fe₂O₃ Nanoparticles Synthesized via Hydrothermal Technique

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Abstract: In the present work, hematite $(\alpha\text{-Fe}_2O_3)$ nanopowders were successfully prepared via a hydrothermal route. The morphology and microstructure of the synthesized nanopowders were analyzed by using scanning and transmission electron microscopy (SEM and TEM, respectively) analysis and X-ray diffraction. Gas sensing devices were fabricated by printing $\alpha\text{-Fe}_2O_3$ nanopowders on alumina substrates provided with an interdigitated platinum electrode. To determine the sensor sensitivity toward NO_2 , one of the main environmental pollutants, tests with low concentrations of NO_2 in air were carried out. The results of sensing tests performed at the operating temperature of $200~^{\circ}\text{C}$ have shown that the $\alpha\text{-Fe}_2O_3$ sensor exhibits p-type semiconductor behavior and high sensitivity. Further, the dynamics exhibited by the sensor are also very fast. Lastly, to determine the selectivity of the $\alpha\text{-Fe}_2O_3$ sensor, it was tested toward different gases. The sensor displayed large selectivity to nitrogen dioxide, which can be attributed to larger affinity towards NO_2 in comparison to other pollutant gases present in the environment, such as CO and CO_2 .

Keywords: α -Fe₂O₃; hydrothermal; NO₂; selectivity; p-type behavior

1. Introduction

Metal oxides, such as ZnO, NiO, SnO $_2$, and others, have been used for a long time as an active layer in gas sensors to detect toxic and hazardous environmental gases, such as CO, CO $_2$, and NO $_2$, because they have a low cost in comparison to other sensing technologies, are robust, have a small size, and are lightweight. Recently, the growth of industries and the improvements in quality of life have led to the increased importance of the problem of air pollution. As a consequence, the quality of urban air has lately become a problem of public health. Among the air pollutants, nitrogen dioxide (NO $_2$) is one of the most harmful and highly toxic gases [1,2]. The sources of nitrogen dioxide are car exhausts and house and industrial combustion processes. The threshold limit value of NO $_2$ is up to 25 ppm [3]; hence, its detection at this low concentration is necessary.

Widely used commercial gas sensors have, however, severe limitations. These concerns originate from their large power consumption, low sensitivity, and high working temperature [4,5]. Therefore, there is an urgent demand for new gas sensors with better performances. For this purpose, several metallic oxides have attracted increasing interest and are intensively investigated, such as ZnO [6], SnO₂ [7], CdO [8], NiO [9], and Fe₂O₃ [10] for gas sensing. However, more research is still necessary to develop further improvements in gas-sensing properties, especially the sensitivity and selectivity [11]. These characteristics are of utmost importance for environmental applications where the target gas is usually present in low concentrations in a complex gaseous mixture.

Conductometric gas sensors can be fabricated by printing a film of the sensing material on the sensor substrate coated with interdigitated electrodes [12]. It is well known that the sensor performance is closely related to the morphology of the sensor surface exposed to the target gas. Obviously, sensors based on nanopowders perform better than thin, film-based ones; this is mainly due to the larger specific surface and reacting sites available in the powder in comparison to thin films, where the surface roughness is a limiting parameter.

Among metal oxide semiconductors, α -Fe₂O₃ is a good candidate as a sensing layer for different gases. α -Fe₂O₃ is an n-type metal oxide semiconductor with a band gap of 2.1 eV [13]. Iron oxide has been a commonly studied transitional metal oxide material due to its variable oxidation state, low cost, and remarkable magnetic properties [14,15]. Several methods have been used to synthetize α -Fe₂O₃ nanomaterials with different morphologies and nanostructures, including solvothermal synthesis [16], sol-gel method [17], thermal decomposition method [18], hydrothermal synthesis [19], and sonoelectrochemical anodization method [20]. Among these techniques, hydrothermal route is considered advantageous because of its simplicity and low cost [21].

Many reports about the detection of different gases and volatile organic compounds, such as NO₂ [22], ethanol [23], H₂S [24], and acetone [25], by pristine or doped α -Fe₂O₃ can be found in literature. Nowadays, many efforts have been aimed at enhancing the gas-sensing performance of α -Fe₂O₃ sensors by controlling its microstructure through many physical/chemical preparation techniques. Recently, it has been outlined that the hematite morphology plays an important role on gas-sensing properties. α -Fe₂O₃ prepared as thin films showed a selective detection for NO₂ [22], while α -Fe₂O₃ nanowires are more sensitive to CO gas [26], and flowerlike α -Fe₂O₃ exhibited better performance regarding ethanol [27]. This shape dependence of sensor sensitivity towards gases is attributed to the variation of the available nanocrystal facets in contact with target gas [28]

On this basis, in this work, we report efforts made in synthesizing α -Fe₂O₃ nanoparticles by hydrothermal method, focusing on the development of a selective sensor for NO₂ gas for applications in the field of environmental monitoring.

2. Experimental Procedure

2.1. Preparation of α -Fe₂O₃ Nanoparticles

The hydrothermal route was used to prepare Fe_2O_3 nanoparticles. The starting material is $FeCl_3$ powder. This precursor was dissolved in water after magnetic stirring for 1 h to obtain a 0.1 M solution. Then, 2 mL of ammonia was added to the solution. The final solution was poured into a Teflon lined steel autoclave which was heated in a programmed furnace at 200 °C for 10 h. The resultant material was washed several times with a mixture of ethanol and water for purification and then dried at 60 °C for 1 h in an oven. Then, the obtained powder was annealed for 2 h at 500 °C.

2.2. Characterization

The microstructure of the material was analyzed using the X-ray diffraction technique (XRD, D8 Advance, manufacturer Bruker Corporation, Brucker, Germany). Scanning and transmission electron microscopy (SEM (Zeiss, Oberkochen, Germany) and TEM (JEOL. LTD., Tokyo, Japan), respectively) were used to evaluate the size and morphology of the particles. To perform SEM analysis, nanoparticles were attached onto an SEM holder by a carbon tape, utilizing an acceleration voltage of 15 kV. TEM analysis was performed by using a JEOL JSM-200F atomic resolution microscope (JEOL Legacy, Peabody, MA, USA) operating at 200 kV. Small amount of nanopowdered samples were dispersed in ethanol by using ultrasonic bath (type Branson CPX5800H-E, GT Sonic, Meizhou city, China) for several minutes, and the solution was dropped onto a carbon-coated Cu grid and left to dry in a clean room at room temperature.

2.3. Preparation of Sensor

First, a proper amount of α -Fe₂O₃ nanoparticles was mixed with several drops of distilled water until a paste was formed. The sensor was prepared by coating paste on a substrate of alumina, which is chemically inert with respect to hematite and is considered a refractory material, allowing sensors to work at high operating temperatures. Geometric dimensions of the substrate were 6 mm \times 3 mm. On the front side, platinum electrodes were fixed, and on the back side, platinum resistances were used as a thermal heater. Sensing tests were performed under a dry air flow. The sensor resistance data were collected in the four points mode using an Agilent 34970A multimeter (KEYSIGHT, Santa Rosa, CA, USA). The operating temperature range of the tested sample was from 150 to 300 °C. Tests were performed in an experimental test bench which operates at a controlled temperature and reads resistance measurements by varying the gas concentrations. The response of the sensor is calculated as R_0/R for oxidizing gas, where R_0 is the baseline resistance of the sensor in air, while R is the sensor resistance when the target gas is turned on. Dynamic characteristics, such as response time, τ_{res} , defined as the time required for the sensor resistance to reach 90% of the equilibrium value after NO_2 is injected, and recovery time, τ_{rec} , taken as the time needed for the sensor resistance to reach 90% of the baseline value in air, were also evaluated.

3. Results and Discussion

3.1. Microstructure and Morphology

Figure 1 shows the X-ray diffraction (XRD) pattern of the prepared sample annealed at 500 °C for 2 h. The diffraction peaks can be readily indexed to the rhombohedral (hexagonal) structures of hematite (JCPDS No: 33-0664) [29] with the lattice parameters a = b = 5.0387 Å and c = 13.767 Å. The peaks of hematite phase are located at $2\theta = 24.11^{\circ}$, 33.16° , 35.56° , 40.87° , 49.43° , 53.96° , 62.46° , and 64.02° ; they are assigned to (012), (104), (110), (113), (024), (116), (214), and (300) planes, respectively. No other phases were observed, indicating the purity of the prepared powder and the α -Fe₂O₃ single phase. No diffraction peak of the FeOOH phase was detected, indicating that the annealing had completely changed the FeOOH to α -Fe₂O₃ according to the following reaction:

$$2 \text{ FeOOH} \rightarrow \alpha\text{-Fe}_2\text{O}_3 + \text{H}_2\text{O}$$

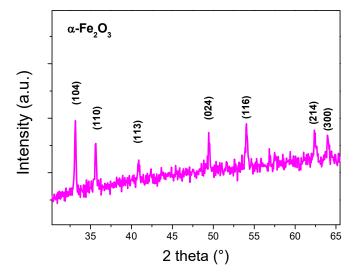


Figure 1. X-ray diffraction pattern of α -Fe₂O₃ nanoparticles.

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The average crystallite size (*D*) of the particles is calculated by the Scherrer equation [30]:

$$D = \frac{0.9\lambda}{B\cos\theta_B}$$

where λ is the X-ray wavelength (1.5406 Å), B is the full width at half maxima (FWHM) of the concerned peaks, and θ_B is the diffraction angle in degree. The most intense peak corresponding to (104) reflection was used to determine the crystallite size, which was estimated to be 55 nm. The average crystallite size is largely dependent on the synthesis method. Cuong et al. [28] have obtained a crystallite size of 40 nm in α -Fe₂O₃ nanopowder prepared by the same hydrothermal procedure of our work, followed by a heat treatment at 500 °C. However, they used ferric nitrate as the starting salt. As the crystallite size of the metal oxide products obtained is well known to depend on the type of salt used, this discrepancy might be due to the easier dissociation of chloride salt by comparison to nitrate ones, leading to different nucleation and growth kinetics. The same findings have been reported in the synthesis of other metal oxides. For example, Lehraki et al. have reported that ZnO prepared by using zinc chloride as a precursor displays large crystallite size, while a zinc nitrate precursor leads to smaller crystallite size [31].

The morphological properties of the sample have been investigated by SEM. The grains have irregular shapes, as shown in Figure 2a. The grains are formed by the coalescence of smaller grains. This may explain the observed irregular spherical and elongated shapes of the final grains. The same remarks have been outlined by several authors [32–34], highlighting that α -Fe₂O₃ powders prepared via the hydrothermal process are composed of particles of 1.5–2.5 μ m in diameter because of the agglomeration of smaller nanoparticles.

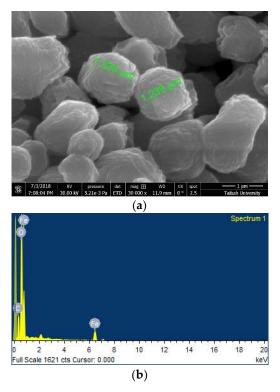


Figure 2. (a) Scanning electron microscopy (SEM) micrograph of α -Fe₂O₃ nanoparticles; (b) Energy dispersive x-ray EDX analysis.

Energy dispersive x-ray EDX analysis reported in Figure 2b indicates the presence of iron and oxygen as the main elements. Small amounts of C were also detected at impurity levels. The carbon impurities may originate from the solution during powder formation.

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TEM images of the nanoparticles are presented in Figure 3. The synthesized nanoparticles are almost hexagonal in shape. The average crystallite size is about 50 nm, which is in agreement with the value indicated by XRD analysis. The lattice fringes as seen in the high resolution transmission electron microscopy HRTEM image (Figure 3b) indicate clearly the high crystallinity of the grains.

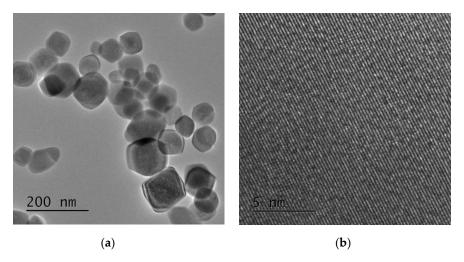


Figure 3. (a) Transmission electron microscopy (TEM) and (b) High resolution transmission electron microscopy HRTEM images of iron oxide nanoparticles synthesized.

3.2. Sensing Properties

The synthesized α -Fe₂O₃ nanoparticles were used as the sensing layer on devices with a planar-layer configuration. It can be assumed that the three-dimensional network of the iron oxide nanoparticles deposited and randomly oriented over the electrodes is responsible for the electrical paths between the adjacent Pt electrodes.

Before being exposed to the target gas, α -Fe₂O₃ sensor must reach a stable resistance, taken as R₀. First, the optimal operating temperature of the sensors was evaluated. The operating temperature is an important parameter for a semiconductor metal oxide sensor as it determines the sensitivity of the device and can be used also to modulate the selectivity towards different gases. The temperature-dependent response measurements to 5 ppm NO₂ gas was performed from 150 to 300 °C, as seen in Figure 4. At low temperatures, the response is restricted by the kinetics of the chemical reaction, and at high temperatures, by gas molecular diffusion. At intermediate temperatures, the two processes' kinetics become equal, leading to the maximization of the sensor response [35]. From these results, we concluded that the operating temperature is around 200 °C. At this temperature, the sensor response presents a maximum value of 3.4 for 5 ppm of NO₂.

It is well known that α -Fe₂O₃ gas sensors have been used for many applications, such as for detecting NO, NO₂, NH₃, SO₂, H₂, H₂S, CO and a variety of volatile organic compounds (VOCs), such as ethanol, methanol, and acetone. Consequently, it is very important to do further studies on the selectivity of our developed α -Fe₂O₃ sensor, i.e., testing the sensors with these other gaseous substances. For this purpose, we have tested three environmental pollutant gases: NO₂, CO, and CO₂.

Figure 5a–c displays the response of the sensor to these target gases. As can be seen, the α -Fe₂O₃ sensor exhibits a high sensitivity regarding NO₂. Instead, for CO and CO₂, the sensor sensitivity is too low. Figure 5d shows the bar chart of selectivity of the sensor. Responses to other gases, i.e., NH₃ and methane, are also added. Tests revealed that the sensor response to 5 ppm NO₂ was higher than that of other gases. One can conclude that the α -Fe₂O₃-based sensor has a good selectivity to NO₂ gas. This finding is in agreement with the results of Navale et al. [22]. They have exposed the α -Fe₂O₃-based sensor to different gases, such as NO₂, H₂S, CH₃OH, C₂H₅OH, and NH₃, and noticed the large response to NO₂ with respect to other gases. This remarkable selectivity of the α -Fe₂O₃-based sensor may be due to the high affinity of NO₂ for hematite at the operating temperature of the sensor.

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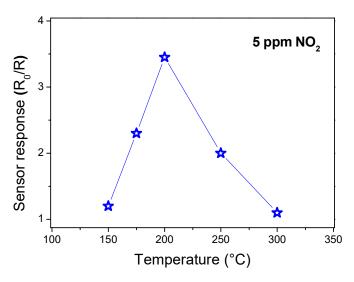


Figure 4. Response to 5 ppm NO_2 of the α -Fe₂O₃ nanoparticles as a function of the temperature.

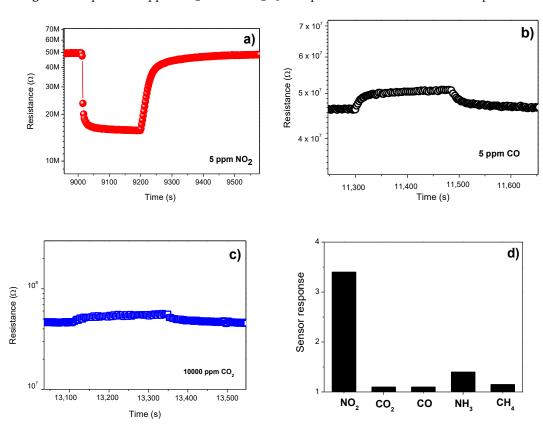


Figure 5. Response of α -Fe₂O₃ sensor to (a) NO₂; (b) CO; (c) CO₂; (d) Selectivity pattern to different gases at 200 °C.

From Table 1, we can observe that the realized α -Fe₂O₃ sensor has the largest response towards NO₂ compared to sensors reported by other authors. A response, ($\Delta R/R_0$)% of 59.9 at 1 ppm, was obtained, which indicates the high sensitivity of our device as an NO₂ sensor.

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Material	Target Gas	Operating Temperature (°)	Tested Gases	Preparation Technique	Response to NO ₂ $(\Delta R/R_0)\%$	Ref.
α-Fe ₂ O ₃	NO ₂	200	NO ₂ , H ₂ S, acetone methanol, NH ₃	Sol-gel	17.2 (200 ppm)	[22]
α-Fe ₂ O ₃	Ethanol	225	NO ₂ , CO, acetone, CO ₂ , NH ₃ , H ₂ , O ₂	Sol-gel	90 (100 ppm)	[36]
Polypyrrole/α-Fe ₂ O ₃	NO ₂	200	NO ₂ , NH ₃ , ethanol, H ₂ S, methanol, Cl ₂	Sol-gel	54 (100 ppm)	[37]
graphene/α-Fe ₂ O ₃	NO ₂	120	NO ₂	Hydrothermal	8.2 (5 ppm)	[38]
α-Fe ₂ O ₃	NO ₂	200	NO ₂ , H ₂ S, acetone methanol, NH ₃	Spray pyrolysis	17.2 (200 ppm)	[39]
α-Fe ₂ O ₃	NO ₂	200	NO ₂ , CO, CO ₂	Hydrothermal	59.9 (1 ppm)	This work

Table 1. Deposition techniques, operating temperatures, and gas sensitivity of α -Fe₂O₃-based, NO₂ sensors reported in literature.

To formulate a hypothesis on the sensing mechanism, we consider here that for semiconducting metal oxide sensors, it is usually based on the sensor resistance change with the adsorbed gas [40,41]. When the sensor surface is exposed to air, oxygen molecules are adsorbed on the surface, to be ionized by electrons trapped from the semiconductor conduction band to produce oxygen ions such as O_2^- , O^- or O^{2-} . In the case of the reducing gas, the latter reacts with oxygen ions to release the captured electrons back to the semiconductor conduction band, causing the reduction in the resistance in the n-type metal oxide semiconductor or increasing the resistance in the p-type one, while in the oxidizing gas, more electrons are trapped from the conduction band. In this situation, the resistance is enhanced in an n-type semiconductor or reduced in the p-type semiconductor.

According to Figure 5, since NO_2 is an oxidation gas and both CO and CO_2 are reducing gases, the behavior of the resistance suggests that the synthesized α -Fe₂O₃ behave as a p-type semiconductor. This seems to be in contradiction with the commonly reported n-type behavior of α -Fe₂O₃. In fact, it is well established that, due to the oxygen deficiency, α -Fe₂O₃ is a native n-type semiconductor. Moreover, α -Fe₂O₃ has been known to exhibits an n- to p-type transition behavior. This intriguing phenomenon was a serious concern, particularly for gas-sensing studies [42]. This n- to p-type transition is ascribed to the formation of a surface inversion layer due to oxygen adsorption causing an increase in the holes concentration [42].

In Figure 6a we have plotted the dynamic response of the sensor to NO₂ exposed to different concentrations from 1 to 5 ppm.

In the investigated concentration range, the sensitivity varies almost linearly with the gas concentration, as shown in Figure 6b. As can be seen, the sensor remains highly sensitive to a low concentration of gas up to 1 ppm, suggesting that it is able to detect lower NO_2 concentrations (in the ppb range) so as to be indicated for the sensing of nitrogen dioxide in the environment [3].

The response and recovery times of the α -Fe₂O₃ sensor as a function of NO₂ concentrations at the operating temperature of 200 °C are presented in Figure 7. The measured response times and recovery times are fast. The response time is reduced from 72 to 10 s with increasing the gas concentration from 1 to 5 ppm, this variation is due to the increase in the gas adsorption with the concentration. However, the recovery time, which depends on the gas desorption kinetic, has a reverse behavior: it increases from 8 to 180 s with the concentration. Navale et al. [22] has observed the same trend in the response and the recovery time variations as a function of NO₂ concentration. Recently, Zhang et al. [39] have reported a longer time response of up to 2.6 min at an operating temperature of 125 °C and 4.6 min at an operating temperature at 100 °C. These large response times are mainly due to the gas adsorption kinetic slowing with a reduction in the operating temperature.

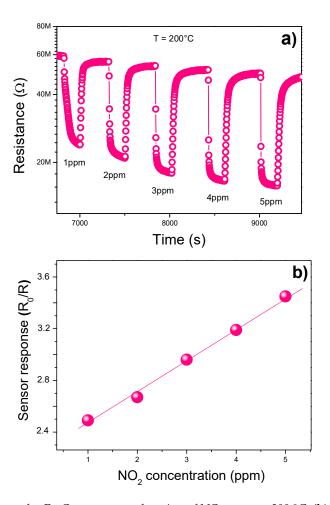


Figure 6. (a) Response of α -Fe₂O₃ sensor as a function of NO₂ conc. at 200 °C. (b) Calibration curve.

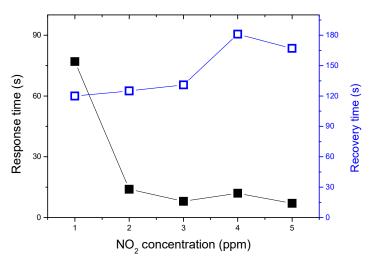


Figure 7. Response and recovery times of α -Fe₂O₃ sensor as a function of NO₂ concentrations at the operating temperature of 200 $^{\circ}$ C.

Reproducibility and stability are important parameters for the performance of the sensor. Figure 8 shows, furthermore, the reproducibility of the sensor when exposed to four consecutive pulses of 5 ppm of NO_2 gas. As clearly shown in the figure, the response of the material is almost constant. This confirms the reproducibility of the sensor material.

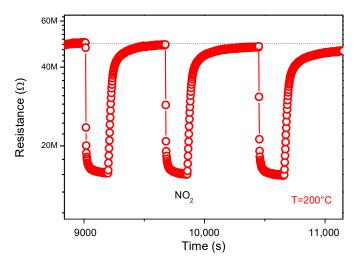


Figure 8. Reproducibility tests to NO₂ gas.

To check the stability of the sensor, the resistance change was studied after one month, and we observed for any changes in the behavior of the material resistance.

4. Conclusions

Hematite Fe_2O_3 nanoparticles were successfully prepared by hydrothermal technique. Crystallinity, particle size, and morphology of nanopowders were characterized by XRD, TEM, and SEM respectively. Results demonstrated that α -Fe $_2O_3$ nanoparticles synthesized by hydrothermal route are highly promising materials for NO_2 detection at operating temperature of 200 °C. Moreover, the realized sensor is selective against a variety of interfering gases and show good characteristics of fast response/recovery times and reproducibility.

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References

- 1. Brunet, J.; Dubois, M.; Pauly, A.; Spinelle, L.; Ndiaye, A.; Guérin, K.; Varenne, C.; Lauron, B. An innovative gas sensor system designed from a sensitive organic semiconductor downstream a nanocarbonaceous chemical filter for the selective detection of NO₂ in an environmental context: Part I: Development of a nanocarbon filter for the removal of ozone. *Sens. Actuators B* **2012**, *173*, 659–667.
- 2. Patnaik, P. *A Comprehensive Guide to the Hazardous Properties of Chemical Substances*; John Wiley & Sons, Inc.: Hoboken, NJ, USA, 2007; p. 1060.
- 3. Bochenkov, V.E.; Sergeev, G.B. Preparation and chemiresistive properties of nanostructured materials. *Adv. Colloid Interface Sci.* **2005**, *116*, 245–254. [CrossRef] [PubMed]
- 4. Qureshi, A.; Mergen, A.; Altindal, A. Preparation and characterization of Li and Ti codoped NiO nanocomposites for gas sensors applications. *Sens. Actuators B* **2009**, *135*, 537–540. [CrossRef]
- 5. Soleimanpour, A.M.; Hou, Y.H.; Jayatissa, A. Surface and gas sensing properties of nanocrystalline nickel oxide thin films. *Appl. Surf. Sci.* **2011**, 257, 5398–5402. [CrossRef]
- 6. Zhu, L.; Zeng, W. Room-temperature gas sensing of ZnO-based gas sensor: A review. *Sens. Actuators A* **2017**, 267, 242–261. [CrossRef]

7. Liu, X.; Ma, T.; Xu, Y.; Sun, L.; Zheng, L.; Schmidt, O.G.; Zhang, J. Rolled-up SnO₂ nanomembranes: A new platform for efficient gas sensors. *Sens. Actuators B* **2018**, 264, 92–99. [CrossRef]

- 8. Velusamy, P.; Babu, R.R.; Ramamurthi, K.; Elangovan, E.; Viegas, J.; Sridharan, M. Spray deposited ruthenium incorporated CdO thin films for opto-electronic and gas sensing applications. *J. Phys. Chem. Solids* **2018**, 112, 127–136. [CrossRef]
- 9. Swati, R.; Gawali, V.L.; Patil, V.G.; Deonikar, S.S.; Patil, D.R.; Patil, P.S.; Patil, J.P. Ce doped NiO nanoparticles as selective NO₂ gas sensor. *J. Phys. Chem. Solids* **2018**, *114*, 28–35.
- 10. Liang, S.; Li, J.; Wang, F.; Qina, J.; Lai, X.; Jiang, X. Highly sensitive acetone gas sensor based on ultrafine α -Fe₂O₃ nanoparticles. *Sens. Actuators B* **2017**, 238, 923–927. [CrossRef]
- 11. Neri, G. First Fifty Years of Chemoresistive Gas Sensors. Chemosensors 2015, 3, 1–20. [CrossRef]
- 12. Hjiri, M.; el Mir, L.; Leonardi, S.G.; Pistone, A.; Mavilia, L.; Neri, G. Al-doped ZnO for highly sensitive CO gas sensors. *Sens. Actuators B* **2014**, *196*, 413–420. [CrossRef]
- 13. Balouria, V.; Kumar, A.; Samanta, S.; Singha, A.; Debnath, A.K.; Mahajan, A.; Bedi, R.K.; Aswal, D.K.; Gupta, S.K. Temperature Dependent H₂S and Cl₂ Sensing Selectivity of Cr₂O₃ Thin Films. *Sens. Actuators B* **2013**, *181*, 471–478. [CrossRef]
- 14. Liu, Z.; Lv, B.; Wu, D.; Sun, Y.; Xu, Y. Preparation and Properties of Octadecahedral α -Fe₂O₃ Nanoparticles Enclosed by {104} and {112} Facets. *Eur. J. Inorg. Chem.* **2012**, 2012, 4076–4081. [CrossRef]
- 15. Cai, J.; Chen, S.; Ji, M.; Hu, J.; Ma, Y.; Qi, L. Organic additive-free synthesis of mesocrystalline hematite nanoplates via two-dimensional oriented attachment. *Cryst. Eng. Comm.* **2014**, *16*, 1553–1559. [CrossRef]
- 16. Tao, B.; Zhang, Q.; Liu, Z.Z.; Geng, B.Y. Cooperative effect of pH value and anions on single-crystalline hexagonal and circular α-Fe₂O₃ nanorings. *Mater. Chem. Phys.* **2012**, *136*, 604–612. [CrossRef]
- 17. Reda, S.M. Synthesis of ZnO and Fe₂O₃ nanoparticles by sol-gel method and their application in dye-sensitized solar cells. *Mater. Sci. Semicond. Process* **2010**, *13*, 417–425. [CrossRef]
- 18. Darezereshki, E.; Bakhtiari, F.; Alizadeh, M.; Vakylabad, A.B.; Ranjbar, M. Innovative impregnation process for production of γ-Fe₂O₃-activated carbon nanocomposite. *Mater. Sci. Semicond. Process* **2012**, *15*, 91–97. [CrossRef]
- 19. He, K.; Xu, C.Y.; Zhen, L.; Shao, W.Z. Fractal growth of single-crystal α-Fe₂O₃: From dendritic micro-pines to hexagonal micro-snowflakes. *Mater. Lett.* **2008**, *62*, 739–742. [CrossRef]
- 20. Zhang, Z.H.; Hossain, M.F.; Takahashi, T. Fabrication of shape-controlled α-Fe₂O₃ nanostructures by sonoelectrochemical anodization for visible light photocatalytic application. *Mater. Lett.* **2010**, *64*, 435–438. [CrossRef]
- 21. Thangadurai, V.; Kopp, P. Chemical synthesis of Ca-doped CeO₂—Intermediate temperature oxide ion electrolytes. *J. Power Sources* **2007**, *168*, 178–183. [CrossRef]
- 22. Navale, S.T.; Bandgar, D.K.; Nalage, S.R.; Khuspe, G.D.; Chougule, M.A.; Kolekar, Y.D.; Sen, S.; Patil, V.B. Synthesis of Fe₂O₃ nanoparticles for nitrogen dioxide gas sensing applications. *Ceram. Int.* **2013**, *39*, 6453–6460. [CrossRef]
- 23. Pandeeswari, R.; Karn, R.K.; Jeyaprakash, B.G. Ethanol sensing behaviour of sol-gel dip-coated TiO₂ thin films. *Sens. Actuators B* **2014**, 194, 470–477. [CrossRef]
- 24. Wang, Y.; Cao, J.; Kong, F.; Xia, H.; Zhang, J.; Zhu, B.; Wang, S.; Wu, S. Low-temperature H₂S sensors based on Ag-doped α-Fe₂O₃ nanoparticles. *Sens. Actuators B* **2008**, *131*, 183–189. [CrossRef]
- 25. Biswal, R.C. Pure and Pt-loaded gamma iron oxide as sensor for detection of sub ppm level of acetone. *Sens. Actuators B Chem.* **2011**, *157*, 183–188. [CrossRef]
- Liao, L.; Zheng, Z.; Yan, B.; Zhang, J.X.; Gong, H.; Li, J.C.; Liu, C.; Shen, Z.X.; Yu, T. Morphology controllable synthesis of α-Fe₂O₃ 1D nanostructurs: Growth mechanism and nanodevice based on single nanowire.
 J. Phys. Chem. C 2008, 112, 10784–10788. [CrossRef]
- 27. Wang, L.; Fei, T.; Lou, Z.; Zhang, T. Three-Dimensional Hierarchical Flowerlike α-Fe₂O₃ Nanostructures: Synthesis and Ethanol-Sensing Properties. *ACS Appl. Mater. Interfaces* **2011**, *3*, 4689–4694. [CrossRef] [PubMed]
- 28. Cuong, N.D.; Khieu, D.Q.; Hoa, T.T.; Quang, D.T.; Viet, P.H.; Lam, T.D.; Hoa, N.D.; Hieu, N.V. Facile Synthesis of α -Fe₂O₃ Nanoparticles for High-Performance CO Gas Sensor. *Mater. Res. Bull.* **2015**, *68*, 302–307. [CrossRef]
- 29. Mathevulaa, L.E.; Notoa, L.L.; Mothudia, B.M.; Chithambob, M.; Dhlamini, M.S. Structural and optical properties of sol-gel derived α -Fe₂O₃ nanoparticles. *J. Lumin.* **2017**, *192*, 879–887. [CrossRef]

- 30. Cullity, B.D. Elements of X-ray Diffraction; Addison-Wesley: Boston, MA, USA, 1978; p. 102.
- 31. Lehraki, N.; Aida, M.S.; Abed, S.; Attaf, N.; Attaf, A.; Poulain, M. ZnO thin films deposition by spray pyrolysis: Influence of precursor solution properties. *Curr. Appl. Phys.* **2012**, *12*, 1283–1287. [CrossRef]
- 32. Jin, W.X.; Ma, S.Y.; Tie, Z.Z.; Jiang, X.H.; Li, W.Q.; Luo, J.; Xu, X.L.; Wang, T.T. Hydrothermal synthesis of mono disperse porous cube, cake and spheroid-like a-Fe₂O₃ particles and their high gassensing properties. *Sens. Actuators B* **2015**, 220, 243–254. [CrossRef]
- 33. Zeng, Q.Z.; Ma, S.Y.; Jin, W.X.; Yang, H.M.; Chen, H.; Ge, Q.; Ma, L. Hydrothermal synthesis of monodisperse α -Fe₂O₃ hollow microspheroids and their high gas-sensing properties. *J. Alloy. Compd.* **2017**, 705, 427–437. [CrossRef]
- 34. Hung, C.M.; Hoa, N.D.; Duy, N.V.; Toan, N.V.; Le, D.T.T.; Hieu, N.V. Synthesis and gas-sensing characteristics of α-Fe₂O₃ hollow balls. *J. Sci. Adv. Mater. Devices* **2016**, *1*, 45–50. [CrossRef]
- 35. Chougule, M.A.; Sen, S.; Patil, V.B. Fabrication of nanostructured ZnO thin film sensor for NO₂ monitoring. *Ceram. Int.* **2012**, *38*, 2685–2692. [CrossRef]
- 36. Mirzaei, A.; Janghorban, K.; Hashemi, B.; Bonyani, M.; Leonardi, S.G.; Neri, G. Highly stable and selective ethanol sensor based on α-Fe₂O₃ nanoparticles prepared by Pechini sol-gel method. *Ceram. Int.* **2016**, 42, 6136–6144. [CrossRef]
- 37. Navale, S.T.; Khuspe, G.D.; Chougule, M.A.; Patil, V.B. Room temperature NO2 gas sensor based on PPy/α-Fe₂O₃ hybrid nanocomposites. *Ceram. Int.* **2014**, *40*, 8013–8020. [CrossRef]
- 38. Zhang, B.; Liu, G.; Cheng, M.; Gao, Y.; Zhao, L.; Li, S.; Liu, F.; Yan, X.; Zhang, T.; Sun, P.; et al. The preparation of reduced graphene oxide-encapsulated α-Fe₂O₃ hybrid and its outstanding NO₂ gas sensing properties at room temperature. *Sens. Actuators B* **2018**, *261*, 252–263. [CrossRef]
- 39. Saritas, S.; Kundakci, M.; Coban, O.; Tuzemen, S.; Yildirim, M. Ni: Fe₂O₃, Mg: Fe₂O₃ and Fe₂O₃ thin films gas sensor application. *Phys. B Condens. Matter* **2018**, *541*, 14–18. [CrossRef]
- 40. Yamazoe, N.; Sakai, G.; Shimanoe, K. Oxide semiconductor gas sensors. *Catal. Surv. Asia* **2003**, *7*, 63–75. [CrossRef]
- 41. Dayan, N.J.; Sainkar, S.R.; Karekar, R.N.; Aiyer, R.C. Formulation and characterization of ZnO:Sb thick-film gas sensors. *Thin Solid Films* **1998**, 325, 254–258. [CrossRef]
- 42. Gurlo, A.; Sahm, M.; Oprea, A.; Barsan, N.; Weimar, U. A p- to n-transition on α-Fe₂O₃-basedthick film sensors studied by conductance and work function change measurements. *Sens. Actuators B* **2004**, 102, 291–298. [CrossRef]



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